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Mechanical and electrochemical properties of multiple-layer diode laser cladding of 316L stainless steel

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Abstract

In the present investigation, a detailed mechanical and electrochemical properties of multiple-layer laser clad 316L stainless steel (from the powders produced by gas atomized route) has been carried out. Multiple-layer laser cladding of 316 L stainless steel has been conducted using a diode laser. The mechanical property (microhardness) of the fabricated product has been evaluated using a microhardness testing machine and correlated with the process parameters. The electrochemical property, mainly pitting corrosion resistance of the fabricated layer corresponding to maximum microhardness (in a 3.56% NaCl solution) has been evaluated using standard potentiodynamic polarization testing. The microhardness of the laser assisted fabricated layers was found to vary from 170 to 278 VHN, increased with decrease in applied power density and increase in scan speed and was higher than that of conventionally processed 316L (155 VHN). The superior microhardness value is attributed to grain refinement associated with laser melting and rapid solidification. The critical potential to pit formation (E_{pp1}) was measured to be 550 mV saturated calomel electrode (SCE) and superior to the conventionally processed 316L stainless steel (445 mV (SCE)).

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1. Introduction

Laser, as a clean and directed source of energy has a wide scope of applications in materials processing [1,2]. Laser assisted surface melting and alloying of metals and alloys has been found to improve wear,

corrosion and oxidation resistance [3–6]. High power laser beam may also be used to melt metals in the form of particles/wire, deposition of molten layer and thereby, building of a full component from the computer aided design (CAD) and it can be used to produce complex geometry [7]. Multiple-layer laser cladding has been reported to fabricate components made of TiAl intermetallics, Ni-based superalloys, steel and titanium [8–10]. 316L Stainless steel is an important class of stainless steel having a wide scope of application in oil and gas industry, refineries,

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chemical and petro-chemical plants and as biomaterials because of its excellent corrosion properties [11]. Pinkerton and Li [12] studied the influence of pulse frequency on the microstructure, surface roughness and hardness of fabricated 316L stainless steel using a pulsed wave CO₂ laser. As an extension of previous attempt, a detailed study on laser assisted fabrication of AISI 316L has been carried out. It has been found that under a very narrow range of laser parameters a homogeneous microstructure with minimum area fraction of porosity can be achieved [13]. As a part of the study, in the present investigation, a detailed microhardness analysis of the fabricated layers have been made and correlated with microstructures. In addition, the pitting corrosion resistance of the formed layer has been evaluated and compared with the conventionally available 316L stainless steel.

2. Experimental procedure

In the present study, gas atomized 316L stainless steel (C <0.03%, Si 0.7%, Mn 1.7%, Mo 3%, Ni 11%, Cr 18%, balance Fe) powders of particle size 50–150 μm was used as feedstock material. A Laserline diode laser of wavelength 940 nm (maximum power of 1.5 kW) was used for the materials processing. Fabrication was done by melting the feedstock powder (delivered by an external powder feeder) using the laser, deposition of the melt on the substrate (mild steel) in a layer by layer fashion using an applied power density (P) of 0.031–0.165 kW/mm², scan speed (ν) of 2.5–12.5 mm/s and powder feed rate (F_p) of 65 to 340 mg/s, respectively. After the laser processing, detailed microstructural study of the top surface and cross section of the fabricated component was carried out by optical and scanning electron microscopy. Microhardness of the fabricated layer (both on the top surface and along cross section) was carefully measured using a Tukon 2100 Vicker's microhardness tester with a 300 g applied load and correlated with process parameters. Finally, corrosion test was carried out for the sample corresponding to maximum microhardness using a ACM Gill AC potentiostat with SEQUENCER software. Pitting corrosion behaviour was evaluated by standard cyclic potentiodynamic polarization study [14]. The specimen was carefully polished using 6 μm diamond paste

prior to corrosion study. Standard cyclic potentiodynamic polarization test was performed at a sweep rate of 0.5 mV/s in aerated 3.56% NaCl solution. The experiment was conducted at 20 °C. A saturated calomel electrode (SCE) was employed as the reference electrode. A platinum electrode was used as counter electrode for current measurement. The solution was prepared from analytically pure chemicals and deionised water. The specimen was exposed to the test conditions for 2 h open circuit potential (OCP) before commencing the test.

3. Results and discussions

Microstructural homogeneity and defect free matrix are the pre-requisite for the superior mechanical and electrochemical properties of any component. A detailed study on the variation of microstructure and its homogeneity with the laser parameters showed that under a very narrow range of processing conditions a uniform and homogeneous microstructure may be achieved [13]. Fig. 1 shows the microstructure of laser assisted fabricated 316L stainless steel lased with a power density of 0.073 kW/mm², scan speed of 5 mm/s and a powder feed rate of 203 mg/s (one of the optimum processing conditions). The microstructure is predominantly cellular, with an average grain size of 10 μm . However, the morphology and grain size of the microstructure were found to vary with laser parameters [13]. The microhardness analysis of the



Fig. 1. Scanning electron micrograph of laser assisted fabricated 316L stainless steel lased with a power of 0.073 kW/mm², scan speed of 5 mm/s and a powder feed rate of 203 mg/s.

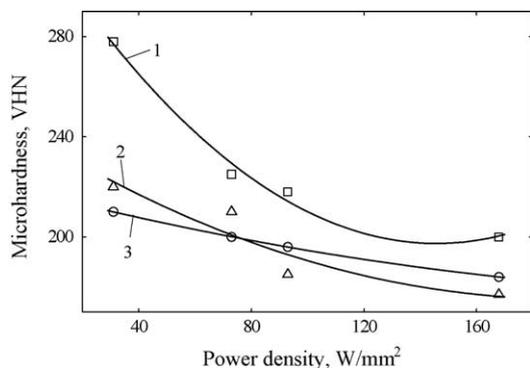


Fig. 2. Variation of average microhardness with applied power density for laser assisted fabricated AISI 316L stainless steel lased with a (1) scan speed of 5 mm/s, powder feed rate of 203 mg/s; (2) scan speed of 2.5 mm/s, powder feed rate of 203 mg/s and (3) scan speed of 2.5 mm/s, powder feed rate of 136 mg/s, respectively.

fabricated product showed that the average microhardness varied from 170 to 280 VHN which is significantly higher than the microhardness values of conventionally processed annealed 316L stainless steel (155 VHN) and even that of the same fabricated by pulsed wave CO₂ laser [11,12]. The microhardness value was however, found to vary with laser parameters. Fig. 2 shows the effect of applied laser power density on the average microhardness of laser assisted fabricated AISI 316L stainless steel lased with a (1) scan speed of 5 mm/s, powder feed rate of 203 mg/s; (2) scan speed of 2.5 mm/s, powder feed rate of 203 mg/s; (3) scan speed of 2.5 mm/s, powder feed rate of 136 mg/s, respectively. From Fig. 2, it is relevant that average microhardness of the fabricated layers decreases with increase in applied power density. This effect is attributed to coarsening of grains as was evident in detailed microstructural investigation [13]. A close comparison of plot 1 with plot 2 shows that average microhardness value increases with increase in scan speed. With increasing the scan speed, due to a shorter time of interaction a low energy is supplied during melting resulting in refinement of grains, and hence, increase in average microhardness. The effect of powder flow rate on the microhardness does not however, show any specific trend (plot 2 vis-à-vis plot 3). From the variation of microhardness with laser parameters it may be concluded that hardening of the formed parts is mainly because of grain refinement and for an

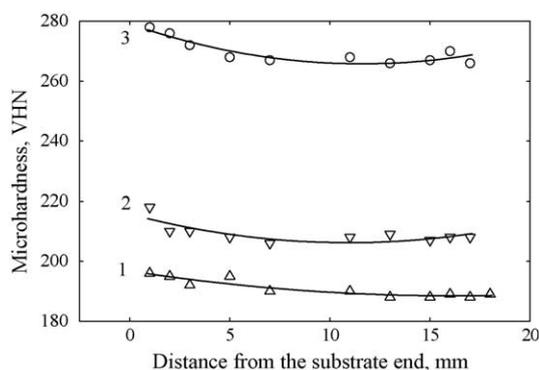


Fig. 3. Variation of microhardness with length for laser assisted fabricated AISI 316L stainless steel processed with (1) a power density of 0.091 kW/mm², scan speed of 2.5 mm/s; (2) a power density of 0.091 kW/mm², scan speed of 5 mm/s and (3) a power density of 0.031 kW/mm², scan speed of 5 mm/s (with a powder feed rate of 136 mg/s), respectively.

improved microhardness, a low power and high scan speed combinations should be chosen.

Fig. 3 shows the variation of average microhardness along the wall growth direction for laser assisted fabricated AISI 316L stainless steel processed with (1) a power density of 0.091 kW/mm², scan speed of 2.5 mm/s; (2) 0.091 kW/mm², 5 mm/s; (3) 0.031 kW/mm², 5 mm/s (with a powder feed rate of 136 mg/s), respectively. It is relevant from Fig. 3 that the microhardness is almost uniform throughout the cross section along the wall height with a marginal higher value near to the substrate and a lower value at the intermediate region. The marginal higher value of microhardness near the substrate region is mainly because of refinement of microstructure due to a high quenching rate from the underlying substrate. On the other hand, lower level of microhardness at the intermediate region is attributed to the grain coarsening effect. Application of a lower power density (plot 3 vis-à-vis plot 2) increases the average microhardness. Similarly, application of a lower scan speed (plot 1 vis-à-vis plot 2), increases the average microhardness of the fabricated layer. Hence, the microhardness of the fabricated layer marginally varies with the position and highly dependent on the applied laser parameters.

Fig. 4 presents the results of the potentiodynamic polarization test conducted in a 3.56 wt.% NaCl solution for the sample lased with a power density of 0.031 kW/mm², scan speed of 5 mm/s and powder feed rate of 136 mg/s (the condition corresponding to

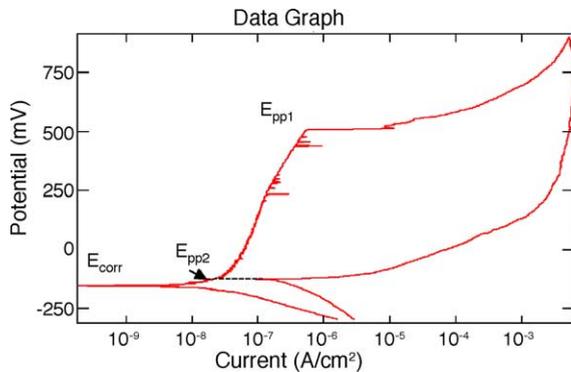


Fig. 4. Potentiodynamic cyclic polarization behaviour of direct laser fabricated 316L stainless steel (lased with a power of 0.031 kW/mm², scan speed of 5 mm/s and powder feed rate of 136 mg/s) in a 3.56% NaCl solution.

maximum microhardness value). The polarization test was conducted both in forward and reverse potential cycles to determine the critical potential for pit formation (E_{pp1}) and pit growth (E_{pp2}) values. Critical potential for pit formation is represented by (E_{pp1}) which is the potential at which the current density increases very rapidly with a slight change in potential and higher the E_{pp1} greater is the resistance of the material to pit formation [15]. On the other hand, E_{pp2} is the potential where the reverse cycle intersects the forward cycle. Nobler the E_{pp2} to E_{corr} , superior is the resistance of the material to pit growth. Table 1 summarizes the pitting potentials of the present samples under the given study. For comparison, the E_{pp1} value of AISI 316L stainless steel under annealed condition is also shown in Table 1 [16]. From Table 1 it may be noted that E_{pp1} value achieved for the present sample is 550 mV(SCE), which is higher than that for the conventionally processed AISI 316L stainless steel, i.e. 425 mV(SCE). Moreover, E_{pp2} is more nobler than the corresponding oxygen evolution

Table 1
Comparison of corrosion properties of laser assisted fabricated vis-à-vis conventional 316L stainless steel

Sample	E_{corr} (mV/SCE)	E_{pp1} (mV/SCE)	E_{pp2} (mV/SCE)
Conventional	0	425	NA
Laser assisted fabricated (present study)	-150	550	-125

potential (E_{corr}), hence, pit propagation is also precluded for the laser assisted fabricated 316L stainless steel at the oxygen evolution potential.

4. Summary and conclusions

In the present study, the mechanical (microhardness) and electrochemical (pitting corrosion resistance) analysis of laser assisted fabricated AISI316L stainless steel has been carried out (with a power density of 0.031–0.168 kW/mm², scan speed of 5–7.5 mm/s and powder feed rate of 136–203 mg/s).

From the results, the following conclusions may be drawn:

1. The average microhardness of the fabricated layer was significantly improved to as high as 170–280 VHN as compared to 155 VHN of conventionally processed and 175 VHN of CO₂ laser assisted fabricated 316L stainless steel. The enhanced microhardness is attributed to grain refinement achieved during laser processing.
2. Microhardness was found to be marginally higher at the near substrate region and reduced at the intermediate region. Average microhardness of the fabricated layer decreased with increase in applied laser power density and decrease in scan speed.
3. Pitting corrosion resistance was marginally improved in terms of critical potential of pit formation (which is 550 mV (SCE) in the present study as compared to 440 mV (SCE) of the conventionally processed one).

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