

PROPERTIES OF LARGE 7079 ALUMINUM ALLOY FORGINGS IN A CRYOGENIC ENVIRONMENT

by F. T. Inouye

Prepared by

AEROJET-GENERAL CORPORATION Sacramento, Calif. for Lewis Research Center

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for Lewis Research Center

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FOREWORD

The research described herein, which was conducted by Aerojet-General Corporation, Liquid Rocket Operations, was performed under NASA Contract NAS 3-2555 with Mr. J. M. Kazaroff, Chemical Rocket Division, NASA Lewis Research Center, as Technical Manager. The report was originally issued as Aerojet-General Report No. 8800-20, November 1965.

ABSTRACT



Large 7079-T652 hand forgings were evaluated for pump impeller and inducer applications in liquid oxygen/liquid hydrogen rocket engines.

The results of mechanical property tests, reheattreatment experiments, and microstructural studies are presented. The application of 7079 alloy in a cryogenic environment is discussed based upon the test results, and recommendations are made for metallurgical analysis of the more promising alloys.

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TABLE OF CONTENTS

				Page
I.	SUMM	ARY		1
II.	INTR	DDUCT:	ION	1
III.	TECH	NICAL	DISCUSSION	21
	A.	MATE	RIAL	21
	В.	TEST	ING PROCEDURE	21
	C.	TEST	RESULTS	21
		1.	Mechanical Properties of Hand Forging "A"	21
		2.	Mechanical Properties of Hand Forging "B"	43
		3.	Mechanical Properties of Hand Forging "C"	43
		4.	Mechanical Properties of Hand Forging "D"	63
		5.	Effects of -T6 Temper Reheat Treatment (After Rough Machining) on Properties of	6.0
			Hand Forging "E"	63
		6.	Microstructure	72
		7.	Fatigue Properties	74
		8.	Inducer and Impeller Blanks and Finish- Machined Parts	82
IV.	CONC	LUSIO	NS	82
V.	RECO	MEND	ATIONS	82

BIBLIOGRAPHY

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I. SUMMARY

Several large 7079-T652 aluminum alloy forgings, which were candidate materials for use as pump impellers and inducers in liquid oxygen-liquid hydrogen rocket engines, were evaluated.

A number of factors were determined, including the influence of cryogenic temperatures upon smooth- and notched-bar tensile properties, the effect of -T6 temper reheat treatment (after rough machining) upon tensile properties, mechanical fatigue strengths under completely reversed bending and tension/ compression axial stressing, and microstructure.

The 7079 forgings were characterized by good notch toughness at ambient temperature. These forgings were slightly notch sensitive at -320° F and extremely notch sensitive at -423° F. Notch sensitivity varied with specimen orientation as well as material strength and ductility. The poor notch toughness of 7079 forgings also appears to be influenced by impurities (inclusions), a cored microstructure, and high alloy content. The -T6 reheat treatment significantly increased tensile strength but decreased smooth-bar ductility as well as notch toughness. In reversed bending, fatigue strengths closely approximated those for commercial-size products and were higher than the fatigue strengths obtained in tension/compression stressing.

These studies show that the 7079-T652 forgings appear to have satisfactory properties for impeller service in liquid engines at temperatures down to -320° F; however, application in inducers at -423° F is not recommended for the sizes being considered because of poor notch toughness and low fatigue strength. Nickel-base alloys, alpha A-110-AT-ELI titanium alloy, and aluminum alloys developed specifically for cryogenic service are considered more promising for inducer service. However, the nickel and titanium alloys are only applicable if their higher densities can be tolerated. All of these alloys must be metallurgically analyzed to determine their suitability for large -423° F inducers.

II. INTRODUCTION

Aluminum alloys offer unique advantages as a material of a construction because of their low density, good strength, and ductility. They also have excellent corrosion resistance. Certain aluminum alloys possess good toughness and they are used in liquid rocket engine hardware as well as ground equipment exposed to liquid oxygen (-297°F) and liquid hydrogen (-423°F). Several of the typical grades commonly used are 2014 (Thor and Titan structural tankage), 5456 (first-stage Saturn C-1 kerosene and liquid oxygen tanks), 2219 (Saturn S-1C launch vehicle), 5083 (liquid hydrogen storage tanks), and 6061 (used extensively in systems and controls hardware, such as valve bodies, seals, and conduits, at temperatures down to $-423^{\circ}F$).

The 7000-series alloys are the highest strength aluminum alloys. These alloys have been used at temperatures as low as $-320^{\circ}F$, but not generally below that point because of their relatively poorer toughness. Grade 7075 is used in Titan I pump impellers, which operate in kerosene and liquid oxygen (-297°F). The major factors for selecting Grade 7075 over other alloys and steels were

material availability, its high strength-to-density ratio, its propellant compatibility, and its excellent machinability. Because of earlier success-ful experience with 7075 in Titan I, 7075 was again used in Titan II pump impellers which operate in noncryogenic propellant combinations of AeroZINE 50⁽¹⁾ and nitrogen tetroxide.

A major problem in large 7075 impeller forgings is the low strength found near the center. This is caused partly by the low depth-of-hardening at the reduced cooling rates that result from the hot water quench (see Table 1). Low center properties have limited design allowables.

In 1957, a study⁽²⁾ was made to determine the minimum properties obtainable in a "special product" forging (13-in. diameter x 9-in.) of 7079 aluminum alloy. The 7079 aluminum has higher magnesium and lower zinc, chromium, and copper contents than 7075, which results in greater depth of hardening. The 7079 properties presented in Table 2 and Figures 1 through 8 show conclusively that the 7079 depth of hardening is superior to that of 7075. The trend of decreasing properties with forging size and the superiority of 7079 over 7075 is further shown in Tables 3 and 4, which are the results from recent George C. Marshall Space Center tests.⁽³⁾

Experience has demonstrated that 7079 aluminum alloy has the highest mechanical properties of the 7000 series alloys. When a strength-to-density basis is considered, the 7079 alloy appeared promising for pump impellers and inducers of the newer liquid oxygen-liquid hydrogen engines because the lower strength 7079 alloy operated successfully in liquid rocket engines at temperatures down to -320° F. However, impeller and inducer sizes have increased, and mechanical properties of large 7079 forgings were nonexistent for design analysis.

An investigation was undertaken to obtain design mechanical property data for five large 7079-T652 forgings of interest for pump applications in a liquid rocket engine operating in liquid oxygen and liquid hydrogen environments. It is this data that is delineated in this report. Conclusions and recommendations regarding the usage of 7079-T652 forgings exposed to a cryogenic environment are also included.

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⁽¹⁾ AeroZINE 50 is a 50/50 fuel blend of unsymmetrical dimethyl hydrazine (UDMH) and hydrazine (N_2H_4) .

^{(2) &}lt;u>Mechanical Properties of 7075-T6 and 7079-T6 Aluminum Alloy Forgings</u>, Report MM-58, Aerojet-General Corp., 1957.

^{(3) &}lt;u>Aluminum Alloy Forgings, 7075-T652 and 7079-T652</u>, Summary Report R-RE-MMP, George C. Marshall Space Center, 1964.

Elongation	ω	ささ	4	11.5 12	00 20 20 20 20 20 20 20 20 20 20 20 20 2	10.5 7.8 14.0 10.9	0.0 7.0 7.0	NGFFFN NNN00
0.2% Offset Yield Strength (ksi)	42	55 52	62	49 50	50 57 57	558 675 868 77	£ 874	ጚ ት ይ <u></u> ሸባ ጚያ
Ultimate Strength (ksi)	62	66 65	8	67 67	57 66 67	88 88 7 7 7 7	68 68 73	45 71 75 75 75 75 75
Temperature (OF)	RT	RT	RT	RT RT	RT - 320 RT - 320	RT RT RT	RT RT	RT RT RT RT RT RT
Orientation*	A	A 450 to axis	A	A A	ላ ላ ዊ ዊ	4 H 4 H	4 H H	4 4 K K H H
Size (in.)	13 D x 9	ł		8 D x 4 11.3 D x 8.5	12 D x 6.7	11.3 D x 8.5	5 D x 4	5-1/4 D × 4-1/2
Condition	л6	1652	'1652	T652 T652	Ţ6	т652	т65	9H

TABLE 1

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MINIMUM TENSILE PROPERTIES OF 7075 ALUMINUM ALLOY IMPELLER FORGINGS

*A = axial, R = radial, T = tangential

TABLE 2

MECHANICAL PROPERTIES OF STANDARD 7075-T6 AND SPECIAL PRODUCT 7079-T6 ALUMINUM ALLOY FORGINGS*

A. Room Temperature

Specimens	Tensile ks	Strength i	Yield S 0.2% Off	trength set, ksi	Elongat 1 inch	ion In (4D) %	Hardnes	s, BHN
	7075-тб	<u>7079-T6</u>	7075-T6	7079-T6	7075-T6	7079-T6	<u>7075-16</u>	7079-T6
l-A	53.2	69.3	42.6	59.0	2.5	6.0	122	143
щ	57.5	62.6	43.6	53.5	**	N. 1	113	137
U	51.0	70.5	41.5	59.6	о. С	6.5	113	140
2-A	54.8	69.6	42.8	59.5		6.5	115	140
д	57.8	67.8	144.2	53.6	5.0	7.0	115	140
υ	52.8	69.5	41.3	59.7	с. •	6.0	113	143
4-A	59.5	69.4	46.7	58.7	4.0	5 • •	113	148
ф	57.8	68.7	43.2	55.6	5.0	6.5	104	138
Ð	57.3	69.1	42.7	58.8	t. 5	5.0	104	146
6-A	61.9	70.0	51.6	57.7	М	5.0	140	145
щ	58.1	70.0	42.5	57.2	5.0	7.0	118	137
U	58.8	6.69	43.8	57.1	*	6 •5	111	145
8-A	62.5	71.9	53.2	60.6	2.5	.+ .5	137	143
щ	59.2	70.0	45.4	58.1	5.0	6. 2	124	140
U	58.5	71.4	47.1	60.3	د. م	4.5	127	140
10-A	58.5	71.3	44.5	60.3	4.0	11.0	115	142
ф	64.5	71.7	54.6	57.4	Э. 5 С	9.5	129	140
U	65.1	72.7	57.4	58.2	2.5	7.0	140	140
11-A	. 1	73.4	I I	60.2	ı	8.5 .5	I	140 140
Ē	!	73.9	:	6.09	I	8.5	1	145 ,
12-A	59.2	74.4	43.2	65.8	5.0	7.0	118	145
f	58.5	72.9	40.9	64.2	5.0	0.7	115	140
10	58.7	68.9	41.8	57.9	0°5	0.0	109	140
Q	58.8	70.6	42.5	58.4	4.5	0.0	111	13/

*Dimensions of 13-in. diameter by 9-in.

and in the

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Tensil Specimens	7075-T6	12-E 59.5	F 60.1	G 60.1	Н 60.5	T 60.1	J 59.3	K 58.6	г 59.0	M 61.4	N 60.5	13-A 59.5	B 59.7	c 59.1	D 60.7	E 59.7	Fi 60.5	G 60.1	н 59.6	I 61.5	J 59.9	K 59.4	L 60.1	M 62.8	N 60.1	15-A 63.2	B 62.2	c 60.8	D 60.0	
e Strength ƙsi	7079-T6	70.8	69.1	68.5	69.I	70.5	73.0	71.5	66.5	72.5	73.0	76.0	73.0	70.0	72.0	70.0	70.0	69.0	69.5	70.5	71.6	71.2	71.2	72.1	74.8	75.1	71.3	72.0	71.7	
Yield S 0.2% Off	7075 - T6	42.6	43.0	43.4	43.6	43.8	43.5	43.6	144.3	48.1	144.2	43.9	42.2	42.5	43.8	48.8	43.8	43.6	42.7	44.5	144.0	0.444	45.0	48.3	41.5	47.5	46.1	144.5	43.5	
trength set, ksi	7079 - T6	57.2	55.6	55.1	55.7	56.5	57.5	57.5	58.0	62.5	64.0	66.8	62.5	57.5	59.0	56.0	56.5	56.0	57.0	57.1	57.3	59.2	60.5	61.5	65.4	65.3	58.8	59.5	59.4	
Elongat: 1 inch	7075-T6	5.0	7.5	7.5	7.0	6.5	5.0	5.0	*	5.0	5.5	4.0	6.0	5.0	7.5	5 1 1	7.0	7.0	7.0	7.5	7.0	5.0	5.0	6.J	8.5	7.5	7.0	6.5	0.7	1
ion In (4D) %	7079-T6	0.6	10.5	12.5	12.0	12.0	0.11	10.0	*	6.5	6.5	7.0	7.5	8°0	7.5	11.0	12.5	11.5	12.5	13.0	13.0	7.0	6.0	6.0	6.5	7.5	10.0	0.0	8.5	•
Hardnes	7075-T6	JIZ	115	115	111	115	118	118	118	124	118	118	113	113	116	113	107	113	113	112	115	118	120	122	113	122	120	120	118	3
s, BHN	7079-T(137	136	130	130	130	130	136	137	140	140	742	137	136	136	136	135	134	134	136 1	135 135	134	136	138	140	143	138	137	137	1

Specimens	Tensile ks	Strength i	Yield S 0.2% Off	trength set, ksi	Elongat: 1 inch	ion In $(l_{tD}) \not \ll$	Hardnes	s, BHN
4	7075 - T6	7079-T6	7075-тб	7079-T6	<u>7075-T6</u>	7079-T6	7075-T6	7079-T6
15 - G	60.8	70.2	43.3	57.7	7.5	12.5	118	135
Н	61.2	70.5	144.1	58.1	0.7	12.5	118	136
н	60.7	70.0	44.5	58.0	6.5	0.11	111	134
Ŀ	61.2	71.5	46.0	57.5	5.0	12.5	115	136
К	62.0	71.5	47.3	58.0	5.0	10.5	122	135
Г	62.2	73.0	49.0	0.09	5.0	8.0	115	138
М	65.7	71.5	53.2	60.5	5.0	7.0	118	138
N	68.4	74.5	54.5	65 . 5	7.0	7 . 0	120	138
17-A	65.6	76.0	50.1	66.5	7.5	С. С	104	140 140
д	63.6	73.0	48.4	62.0	6.5	0°.0	115	138
U	63.2	73.0	46.5	60.5	0.000	8.5	104	138
A	62.3	72.0	45.2	57.5	0.0	12.5	109	136
ы	62.0	71.5	44.5	57.0	7.0	13.0	100	138
ſщ	61.7	71.5	44.5	58.5	0.0	10.5	TOT	136
IJ	61.2	72.0	43.7	59.5	7.5	10.5	101	138
н	62.1	72.0	6.44	0.09	7.5	10.5	107	138
н	62.9	71.8	46.5	59.4	7.5	11.5	115	140 1
Ŀ	64.1	72.5	48.5	59.3	7.5	12.5	109	130 130
М	65.5	71.6	51.0	58.1	7.0	14.0	104	7.5.1 7.0
1	67.2	72.8	53.4	60.1	7.5	12.0	120	130
W	69.0	70.0	56.1	57.0	J.0	11.0	113	137
Z	69.0	75.0	57.5	65.4	6.5	<u>7.5</u>	02T	
19 - A	6.9	74.2	53.5	65.2	**	7.5	725 725	143 150
Ē	66.5	74.5	52.8	62.1	0	11.0		
U D	65.8	73.8	52.8	62.1	4 0	12.0	124 124	0.1 0.1
Ē	62.9	74.8	53.1	63.0	- - -	10.5	727	140 140
1 [E]	64.5	73.8	52.1	62.0	4	11.5	62T	0)11
1 म	62.6	74.2	52.0	62.1	4.5	10.0	120	
, IJ	64.7	72.2	52 . 1	62.1	** -	10.0	07.T	001 r
Н	63.3	73.8	51.7	62.4	t.5	C.01	17T	104

TABLE 2 (cont.)

(cont.)	
2	
TABLE	

	nsile S ksj	Strength I	Yield S 0.2% Off	trength set, ksi	Elongat 1 inch	ion In $(4D) %$	Hardnes	s, BHN
	5-T6	7079-T6	<u>7075-T6</u>	7079-T6	7075-T6	7079-T6	7075-T6	7079-T6
ō	6.3	72.5	53.1	61.2	4.5	10.5	129	138
Õ	6.4	72.7	54.6	60.09	**	11.0	129	138
Ō	7.7	72.5	56.4	59.5	**	11.0	124	138
Ō	7.6	73.0	57.0	61.8	**	10.0	129	140
Õ	8.5	73.0	57.7	61.0	**	11.0	127	142
Ē.	0.8	74.5	59.2	63.0	5.5	11.5	129	138
0	3.4	76.0	55.5	65.3	2.5	12.0	129	143
Ō	7.5	75.5	55.6	65.0	3.5	12.5	144	148
9	6.5	78.5	56.9	66.5	**	0.11	140	143
Ó	4.6	79.0	55.2	70.0	3.5	7.5	120	140
Ó	4.6	79.0	56.0	69.6	Э•5	**	111	143
9	5.4	79.0	56.5	69.3	3•5	6 .0	127	143
Ó	4.7	76.2	56.2	66.2	3.5	6. 0	127	143
Ó	5.9	78.0	55.3	67.0	3.5	6.5	133	142
Õ	8.1	78.1	57.8	67.4	4.5	7.0	124	137
Ó	4.1	78.0	54.6	67.8	4.5	7.5	124	142
Ō	6.8	77.0	57.0	65.6	4.5	0.6	113	143
9	5.9	76.4	56.9	65.0	4.5	11.0	133	142
Ó	8 0	73.2	54.5	61.9	2.5	13.0	118	137
9	3.4	72.3	54.2	61.0	2.5	0.11	118	143

**No value obtained - specimen failed outside the gage mark.

B. Minus 320°F

Specimens	Tensile ks	Strength i	Yield S 0.2% Off	trength set, ksi	Elongat 1 inch	ion In (4D) %
	7075-тб	7079-T6	7075-T6	7079-T6	7075-T6	7079-76
3 - A	63.9	75.6	54.4	70.6	3.0	2.0
д	63.9	77.4	52.5	63.1	2.0	2 . 8
Ð	61.6	78.4	51.5	69.2	2.5	**
5 - A	66.4	78.2	58.5	70.4	**	1 . 8
Д	61.8	78.2	52.0	65.1	2.0	2.5
Ð	65.0	77.6	53.4	*	2.5	2.0
7-A	68.8	78.8	62.2	62.4	1.0	2.0
щ	64.1	79.8	52.0	54.3	2.5	5. 0
U	65.0	77.4	53.7	64.1	2.5	2°0
9- A	65.5	81.0	63.4	69.7	1.0	**
, М	67.4	77.0	63.8	67.0	1 . 5	**
U	6.9	81.0	61.2	70.6	2.0	2.0
14-A	67.4	82.0	55.1	72.6	3.5	1.0
щ	68.4	80.5	53.6	73.4	4.5	**
Ð	67.2	79.5	53.5	61.9	3.5	**
D	68.0	81.2	53.3	59.6	4.0	2.5 .5
ы	68 . 80	82.2	53.4	*	4.0	5.0
Ę۲	69.2	81.7	53.4	66.1	5.0	6.0
Ċ	68.2 68.2	80.5	53.2	65.2	5.0	5.3
н	69.0	81.7	54.0	68.0	0. 4	5.0
н	69.3	81.2	54.2	66.4	4.0	4.0
L P	67.2	83.1	54.5	6.89	Э•5	5.0
- X	66.9	81.1	56.4	67.4	2.5	**
1	66 . 8	80.2	57.7	71.1	2.5	**
ıΣ	4.17	80.0	61.4	73.1	**	Т. 2
N	70.0	81.5	58.3	75.4	ЭЛ	1.0
16-A	71.1	82.6	58.9	68.1	3.5	**
щ	70.5	7.67	56.8	68.4	**	2.J
U	70.5	82.5	55.9	70.0	**	o m
D	68.1	83.0	53.5	70.5	3.5	

8

7079-T6 4.9 4.0 0 0 0 0 N N th N N 4.6 3.5 5.0 ы С 2.0 2.0 * ** Elongation In 1 inch $(hD) \frac{\pi}{6}$ 7075-T6 * 2°0 00100 0044400 005000 0000 1000 1000 ** * * ł 7079-T6 71.9 74.2 74.7 71.9 77.9 73.0 68.5 66.05 66.05 66.05 66.05 66.05 66.05 66.05 67 .51 0.2% Offset, ksi Yield Strength 7075-T6 69.2 7079**-**T6 Tensile Strength ksi 7075-T6 69.0 68.6 68.9 68.9 69.5 71.5 74.05 76.0 775.5 772.4 772.5 773.5 775.5 775.5 775.5 77.5 77.5 80.6 80.6 77.0 773.5 772.8 70.5 Specimens 18-A 20-A щ υA Z \mathbb{Z} ЧU 16-E ᆔ ΣZ 되 E. ரு

7079-T6 Elongation In 1 inch $(\mu D) \ %$ 7075-T6 2•0 00000 * 2.0 * 7079-T6 75.2 74.8 74.6 74.6 74.6 75.0 75.0 0.2% Offset, ksi Yield Strength 7075-T6 63.9 62.7 65.2 65.2 65.2 65.2 74.5 7079-T6 86.5 86.5 886.6 87.0 87.0 87.0 87.0 Tensile Strength ksi 7075-T6 Specimens 20-G жньхлхг

*No value obtained - extensometer slipped. **No value obtained - specimen failed outside the gage mark. ۰,

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SECTIMENTS

C. Statistic Contactor



Figure 1 Specimen Locations of Standard 7075-T6 and Special Product 7079-T6 Hand Forgings

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Figure 2 Mechanical Properties vs Specimen Locations of Standard 7075-T6 and Special Product 7079-T6 Hand Forgings





Figure 3 Mechanical Properties vs Specimen Locations of Standard 7075-T6 and Special Product 7079-T6 Hand Forgings



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Figure 4 Mechanical Properties vs Specimen Locations of Standard 7075-T6 and Special Product 7079-T6 Hand Forgings

Figure 5 Mechanical Properties vs Specimen Locations of Standard 7075-T6 and Special Product 7079-T6 Hand Forgings

Figure 6 Mechanical Properties vs Specimen Locations of Standard 7075-T6 and Special Product 7079-T6 Hand Forgings

Figure 7 Mechanical Properties vs Specimen Locations of Standard 7075-T6 and Special Product 7079-T6 Hand Forgings

Figure 8 Mechanical Properties vs Specimen Locations of Standard 7075-T6 and Special Product 7079-T6 Hand Forgings

TABLE 3

EFFECT OF FORGING THICKNESS ON THE ROOM TEMPERATURE MECHANICAL PROPERTIES OF 7075-T652 HAND FORGINGS(1)

(in.)	Direction	Ftu (ksi)	Fty (ksi)	Elongation (% in 4D)
	Longitudinal Long Trans. Short Trans.	78 - 88 74 - 80 68 - 81	65 - 77 63 - 68 55 - 63	10 - 12 6 - 10 3 - 6
	Longitudinal Long Trans. Short Trans.	74 - 83 69 - 81 65 - 77	60 - 73 59 - 70 52 - 63	603 4
Q	Longitudinal Long Trans. Short Trans.	63 - 83 60 - 81 57 - 76	48 - 73 46 - 70 43 - 60	8 15 9 8 8 8 8
Q	Longitudinal Long Trans. Short Trans.	61 - 87 57 - 79 53 - 76	46 - 73 45 - 71 43 - 64	9 - 14 9 - 14 9 - 18 9 - 19 9 - 14 9 - 14 10 10 - 14 10 10 - 14 1
12	Longitudinal Long Trans. Short Trans.	57 - 81 55 - 77 56 - 73	43 - 68 43 - 69 39 - 63	7 - 14 2 - 7 3 - 8
12	Longitudinal Long Trans. Short Trans.	53 - 81 53 - 76 52 - 77	36 - 71 37 - 68 33 - 66	0 4 0 1 1 1 1 2 0 0 0 0 0 0 0
18	Longitudinal Long Trans. Short Trans.	55 - 82 52 - 82 52 - 82	36 - 73 34 - 66 35 - 71	4 - 15 5 - 12 12 12

(1) <u>Aluminum Alloy Forgings, 7075-T652 and 7079-T652</u>, Summary Report R-RE-MMP, George C. Marshall Space Center, 1964.

TABLE 4

EFFECT OF FORGING THICKNESS ON THE ROOM TEMPERATURE MECHANICAL PROPERTIES OF 7079-T652 HAND FORGINGS(1)

		н, , , ,		
Size (in.)	Direction	(ksi)	(ksi)	(% in 4D)
2 x 32 x 12	Longitudinal	80 - 83	71 - 74	10 - 12
	Long Trans.	81.5 - 84	71 - 75	8 - 13
	Short Trans.	80 - 82	66 - 68	6 - 10
4 x 32 x 12	Longitudinal Long Trans. Short Trans.	77 - 84 77 - 80 72 - 78	67 - 76 65.5 - 70 60 - 65	 9
6 x 32 x 12	Longitudinal	74 - 84	64 - 75	9 - 13
	Long Trans.	73 - 78	61 - 70	7 - 11
	Short Trans.	70 - 77	57 - 63	6 - 10
8 x 32 x 12	Longitudinal	73 - 81	61 - 73	9 - 14
	Long Trans.	70 - 79	60 - 71	6 - 10
	Short Trans.	77 - 77	57 - 65	11 - 7
10 x 32 x 12	Longitudinal	69 - 78	56.5 - 70	10 - 14
	Long Trans.	69 - 79	57 - 70	5 - 11
	Short Trans.	68 - 79	54.5 - 68	7 - 2
12 x 32 x 12	Longitudinal Long Trans. Short Trans.	69 - 80 67 - 79 66.5 - 78	57 - 72 55 - 69 52 - 67	9 - 12 - 12 - 12 - 12 - 12 - 12 - 12 - 12
18 x 32 x 18	Longitudinal	59 - 78	44 - 70	01 - 1
	Long Trans.	55 - 75	44 - 70	71 - 1
	Short Trans.	57 - 77	44	71 - 1

(1) <u>Aluminum Alloy Forgings, 7075-T652 and 7079-T652</u>, Summary Report R-RE-MMP, George C. Marshall Space Center, 1964.

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III. TECHNICAL DISCUSSION

A. MATERIAL

The forging material condition and chemistry are listed in Table 5 along with the forgings dimensions and test section dimensions. The test sections are shown in Figures 9 through 11.

B. TESTING PROCEDURE

The general test procedure consists of cutting test rings from the peripheral area of the forgings, as shown in Figures 9 and 10. These test rings were machined into tensile specimens which were subsequently tested; the test results were then compared with those of like tensile specimens, which had been taken from the interior areas of the sectioned forging.

Control specimens in the -T652 temper were tested for base-line properties; the test results were compared with those of specimens that were taken from test rings parted from the forging and given the -T6 heat-treatment (per MIL-H-6088) after rough machining. The samples used in the reheattreatment studies were taken from the areas shown in Figure 11.

The configuration and dimensions of tensile specimens are shown in Figure 12. They are standard designs and the stress concentration of the notched specimen is approximately 6.3. Specimen orientations were axial, radial, and tangential.

Tension tests were conducted using standard test equipment at 0.005 in./in./min strain rate at ambient temperature, at -320° F by immersion in liquid nitrogen, and at -423° F by immersion in liquid hydrogen.

Fatigue tests were conducted under tension/compression and bending stress at ambient temperature using standard test equipment. The fatigue specimens were taken from the interior areas of the sectioned forging; their configuration and dimensions are shown in Figures 13 and 14.

C. TEST RESULTS

1. Mechanical Properties of Hand Forging "A"

The tension data for the test ring and center section are listed in Tables 6 and 7 and shown graphically in Figures 15 through 20. The relative location of these sections is seen in Figure 9.

The test results substantiated the following conclusions regarding the large 7079-T652 impeller forging.

a. The increase of the 0.2% offset yield strength with decreasing temperature was gradual over the entire temperature range from ambient to -423^{0} F.

		10. 03		.05	•05				.05	0.02
		-1 .03 0		0 10 .0	•056 c				0.046 (0.05
		 		0.15 0	0.17 0				0.18 (0.16 0
	VI STRY	4 Zr 4.8		t. 45	8°¶				h.6	h.8
	цн. Тче	3.05 3.05		3.51	2.85				3.25	3 . 4
	Tast M	MN 12.0		0.195	0.21				0.20	0.23
'sNO	្ច	0.1		0.11	0.18				0.18	0.21
IMENSI	ORGING	<u>s.</u>		0.10	60•0				0.08	60.0
CTION D	HAND F	0.65 0.65		0.54	0.73				0.80	0.68
CONDITION, TEST SE	L CHEMISTRY OF 7079 TEET SECTION	DIMENSIONS DIMENSIONS IN INCHES TEST RING 31D x 211D x 7L	CENTER 31D x 4.8L	Test Ring 31D × 251D × ¹ 4L	Test Ring 20D x 181D x 1.5L	TOP SECTION A 20D × 4.3L	CENTER SECTION B 20D x 4.3L	BOTTOM SECTION C 20D × 4.3L	TEST RING 20D x 181D x 1.5L	CENTER AND Periphery
FORGING SIZE,	AND MATERIA	<u>Condition</u> -T652		-1652	-T 652				-1652	-7652, тнем гоибн маснімер амр генеат- треатер то -76
		SIZE IN INCHES DIA X LENGTH 31 X 21.5		31 × 21.5	20 × 13				20 × 13	20 × 13
		Hand Forging Oxidizer Impeller		OX I D I ZER IMPELLER	Fuel Inducer				F uel I noucer	FUEL Inducer
22		PART		ß	U				۵	ធ

TABLE 5

Specimen location, Impeller Blank, Oxidizer Pump, Forgings "A" and "B" specimens from Test Ring and Center Section were of axial, radial, and tangential orientations. Specimens were smooth and notched types per Figure 12.

Figure 9 Specimen Location, Impeller Blank, Oxidizer Pump, Forgings "A" and "B"

Specimen location, Inducer Blank, Fuel Pump, Forgings "C" and "D" specimens from Test Rings were of tangential orientation. Those from test sections were of axial, radial, and tangential orientations. Specimens were smooth and notched types per Figure 12. Specimens from Inducer, Fuel Pump, Forging "D" were of like tangential orientation as Test Ring specimens and were taken from similar locations.

Figure 10 Specimen Location, Inducer Blank, Fuel Pump, Forgings "C" and "D"

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Figure 11 Specimen Location, Inducer Blank, Fuel Pump, Forging "E"

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Figure 12 Tensile Specimen Configurations

NOT TO SCALE

Dimensions: $D = 0.5000 \pm 0.0005$ im. $A = 1.25 \pm 0.125$ $B = 4.00 \pm 0.125$ $C = 1.50 \pm 0.005$ $R = 2.3125 \pm 0.002$ Notes: Diameter "D" and "D/2" shall be conce

es: Diameter "D" and "D/2" shall be concentric within 0.001 in. Adjust dimension "D/2" to identify through all specimens <u>+</u> 0.0002 in.

Specimen not to scale

"C" section shall have to max. finish. All other surfaces 32/

Figure 14 Tension/Compression Fatigue Specimen

Notch Yield Ratio		3) 1.47		3) 1.46		3) 1.30
Notch Tensile Ratio		(Kt = 6. 1.13		(Kt = 6. 1.13		(Kt = 6. 1.08
Reduction of Area (\mathscr{A})	25.8 26.4 23.9	;	11.5 10.8 9.9 10.7	;	13.7 13.0 13.0 13.2	;
Elongation (% in 4D)	13.5 14.0 13.8	ensile specimen ensile specimen ensile specimen	10.0 9.5 0.5 0.0 0.0 0.0	ensile specimen ensile specimen ensile specimen	007-00 NNN0	ensile specimen ensile specimen ensile specimen
0.2% Yield Strength (ksi)	49.7 53.7 60.1	Notched t Notched t Notched t	48.4 51.9 54.8 51.7	Notched t Notched t Notched t	59.9 56.6 57.0 57.8	Notched t Notched t Notched t
Ultimate Strength (ksi)	67.3 70.5 71.1	78.4 78.9 84.4	63.8 67.5 66.7 66.7	72.3 74.5 80.4 75.7	71.9 68.9 69.9 69.9	73.9 78.7 73.2 75.3
Test Temp (°F)	RT RT RT Avg.	RT RT RT Avg.	RT RT RT Avg.	RT RT RT Avg.	RT RT RT Avg.	RT RT RT Avg
Orientation	Axial Axial Axial	Axial Axial Axial	Radial Radial Rad ia l	Radi al Radial Radial	Tangential Tangential Tangential	Tangential Tangential Tangential
Specimen	н о м	11	19 20 21	8 8 8 8 8 9 8	38 39 39	9 1- 8 7 1- 1- 29

MECHANICAL PROPERTIES OF 7079-T652 FORGING "A" TEST RING

TABLE 6

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Notch Yield Ratio	(1.41		1.08		1.14
Notch Tensile Ratio	(Kt = 6.3 1.05		Kt = 6.3 0.91		Kt = 6.3 0.97
Reduction of Area (%) 15.2 9.3 5.5			;	0 0 1- 0 0 0 1- 0	;
Elongation (% in 4D) 8.0 1.0	6.3 ensile specimen ensile specimen ensile specimen	NOON NOON	ensile specimen ensile specimen ensile specimen	0.00 0.00 0.00 0.00	ensile specimen ensile specimen ensile specimen
0.2% Yield Strength (ksi) 50.1 63.8 69.6	61.2 Notched t Notched t Notched t	57.6 65.0 63.1 63.6	Notched t Notched t Notched t	70.1 61.2 58.7 63.3	Notched t Notched t Notched t
Ultimate Strength (ksi) 77.8 82.6 85.6	882 82.0 86.59 86.59 86.59 86.59 86.59 86.59 86.59 86.59 86.59 86.59 86.59 86.59 86.50 86.59 86.50 86.	73.5 76.8 75.7 75.3	75.7 67.4 62.0 68.4	77.7 73.7 72.0 74.5	74.6 70.7 71.7 72.3
Test Temp (°F) - 320 - 320	Avg. - 320 - 320 Avg.	-320 -320 -320 Avg.	-320 -320 -320 Avg.	-320 -320 -320 Avg.	-320 -320 Avg.
<u>Orientation</u> Axial Axial Axial	Axial Axial Axial	Radial Radial Radial	Radial Radial Radial	Tangential Tangential Tangential	Tangential Tangential Tangential
Specimen 4	Ч Ц Ц С Ц Ц С	22 24 24	32 32 32 32 33 5	ひ ト ト ト	50 70 1

TABLE 6 (cont.)

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Notch Yield <u>Ratio</u>		3 1.27		1.02		0.95
Notch Tensile Ratio		Kt = 6.3 0.98		Kt = 6.3 0.89		Kt = 6.3 0.78
Reduction of Area (%)	4.01.2 7.096	;	10401 10401	;	0.00 0.44 w	;
Elongation (% in 4D)	++77+ 7.000 7.000	censile specimen censile specimen censile specimen		ensile specimen ensile specimen ensile specimen	0 N N N 0 N N N	ensile specimen ensile specimen ensile specimen
0.2% Yield Strength (ksi)	6.9 6.9 6.9	Notched t Notched t Notched t 	63.1 65.6 64.4	Notched t Notched t Notched t 	 56.8 57.3 57.1	Notched t Notched t Notched t
Ultimate Strength (ksi)	82.4 82.7 84.0 83.0	73.5 88.9 84.3 82.2	75.2 76.4 71.6 74.4	73.9 54.5 66.0	73.3 69.8 69.9	39.2 69.7 54.7 54.5
Test Temp (°F)	-423 -423 -423 Avg.	-423 -423 -423 Avg.	-423 -423 -423 Avg.	-423 -423 -423 Avg.	-423 -423 -423 Avg.	-423 -423 -423 Avg.
Orientation	Axial Axial Axial	Axial Axial Axial	Radial Radial Radial	Radial Radial Radial	Tangential Tangential Tangential	Tangential Tangential Tangential
Specimen	r-∞ 0	16 17 18	25 26 27	365 355 26	44 44 45	5 53 44
<u>~</u>	ŀ					
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TABLE						

MECHANICAL PROPERTIES OF 7079-T652 FORGING "A" CENTER-SECTION

Notch Yield Ratio		1.97		1.59	
Notch Tensile Ratio		Kt = 6.3 1.27		Kt = 6.3 1.07	
Reduction of Area (β)	32.9 36.1 36.1 33.6	;	18.9 20.1 19.5 19.5	ł	13.8 15.1 16.1 16.3
Elongation $(\phi \text{ in } \mu D)$	19.0 0.81 0.0 0.81 180 7.0 7.0	tensile specimen tensile specimen tensile specimen tensile specimen	15.5 13.0 13.0 13.4	tensile specimen tensile specimen tensile specimen tensile specimen	
0.2% Yield Strength (ksi)	 36.8 37.5 37.5	Notched Notched Notched Notched 	39.0 89.2 9.6 9.7 9.7 7 9.7 7	Notched Notched Notched Notched 	44777777777777777777777777777777777777
Ultimate Strength (ksi)	58.1 57.8 57.2 57.9 57.8	80.5 72.3 70.8 73.4	528 28 28 28 28 28 28 28 28 28 28 28 28 2	63.5 63.6 61.3 62.5 62.7	60.1 60.3 61.8 60.8 60.8
Test Temp (°F)	RT RT RT RT Avg.	RT RT RT RT AVG.	RT RT RT RT AVG.	RT RT RT RT Avg.	RT RT RT AVG.
Orientation	Axial Axial Axial Axial	Axial Axial Axial Axial	Radial Radial Radial Radial	Radial Radial Radial Radial	Tangential Tangential Tangential Tangential
Specimen	4 0 M 4	40 70 70 88 70 70 70	111 14 90	4 5 7 7 4 7 7 7 0 7 7 7 0	25 27 28

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TABLE

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Notch Yield Ratio	1.45		1.78		1.32
Notch Tensile Ratio	Kt = 6.3 1.01		Kt = 6.3 1.09		Kt = 6.3 0.92
Reduction of Area (ϕ)	1	24.3 24.3 18.9 21.1 21.1	;	800 600 600 600 600 600 600 600 600 600	:
Elongation (% in 4D)	tensile specimen tensile specimen tensile specimen tensile specimen	19.0 21.0 16.0 17.0 28.3	tensile specimen tensile specimen tensile specimen tensile specimen	0.0 1.0 1.0 2.0 2.0	tensile specimen tensile specimen tensile specimen tensile specimen
0.2% Yield Strength (ksi)	Notched Notched Notched Notched	50.00 50.00 50.00 50.00 50 50 50 50 50 50 50 50 50 50 50 50 5	Notched Notched Notched Notched	48.3 50.1 49,9	Notched Notched Notched Notched
Ultimate Strength (ksi)	60.7 60.3 61.0 61.3	73.3 74.6 74.0 73.9	81.0 80.6 82.3 82.3	70.2 72.6 73.0 72.2 72.0	64.0 66.2 60.5 66.1 66.1
Test Temp (°F)	RT RT RT RT Avg.	-320 -320 -320 -320 Avg.	-320 -320 -320 -320 Avg.	- 320 - 320 - 320 - 320 Avg.	- 320 - 320 - 320 Avg.
Orientation	Tangential Tangential Tangential Tangential	Axial Axial Axial Axial	Axial Axial Axial Axial	Radi a l Radial Radial Radial	Radial Radial Radial Radial
Specimen	61 64 64	w⊿ ⊳∞	サとって	17 19 20	らうりう のよう のの

Notch Yield Ratio		1.26		1.48
Notch Tensile Ratio		Kt = 6.3 0.92		Kt = 6.3 0.915
Reduction of Area (ϕ)		1	10.8 13.7 13.8 13.8 13.8	;
Elongation (\$\verthing\$ in \$\text{hD}\$)	N7 N N7 0 N 0 0 0	ensile specimen ensile specimen ensile specimen ensile specimen	10.0 10.0 10.0	ensile specimen ensile specimen ensile specimen ensile specimen
0.2% Yield Strength (ksi)	51.9 52.7 54.8 56.7	Notched t Notched t Notched t Notched t	700.64 70.00 70.00 70.00	Notched t Notched t Notched t Notched t
Ultimate Strength (ksi)	72.1 73.0 74.4 75.1 73.7	66.3 69.0 68.8 67.7	80.3 82.1 81.3 81.3	71.1 76.8 71.0 71.0
Test Temp (°F)	- 320 - 320 - 320 - 320 - 320 - 320 - 320	- 320 - 320 - 320 Avg.	-423 -423 -423 -423 Avg.	-423 -423 -423 Avg.
Orientation	Tangential Tangential Tangential Tangential	Tangential Tangential Tangential Tangential	Axial Axial Axial Axial	Axial Axial Axial Axial
Specimen	50 93 93 93 93 93 93 93 93 93 93 93 93 93	65 66 68	6 0 H 8 H H H	84 60 4 4 4 4 4

TABLE 7 (cont.)

Notch Yield Ratio		3 0.94		3 1.06
Notch Tensile Ratio		Kt = 6. 0.71		Kt = 6. 0.84
Reduction of Area (\mathscr{A})	0.07.00 4.00.00 4.00	;	೦.೦.೦.೦.೦ ಸಸಸಸಸ	1
Elongation $(\% \text{ in } ^{\text{l}}D)$	+ 7 N Y + 4 N 0 0 0 0	ensile specimen ensile specimen ensile specimen	~~~~~~~~~~~~~~~~~~~~~~~~~~~~~~~~~~~~~~	censile specimen censile specimen censile specimen
0.2% Yield Strength (ksi)	52.5 58.5 58.5 58.5	Notched t Notched t Notched t	59.6 55.2 57.4	Notched t Notched t Notched t
Ultimate Strength (ksi)	75.5 77.1 76.0 79.2 76.95	63.0 16.4 54.9 54.8	70.2 73.4 71.8 72.5	59.1 70.0 60.8 60.8
Test Temp (°F)	-423 -423 -423 -423 Avg.	-423 -423 -423 Avg.	-423 -423 -423 -423 AVB.	-423 -423 -423 Avg.
Orientation	Radial Radial Radial Radial	Radial Radial Radial	Tangential Tangential Tangential Tangential	Tangential Tangential Tangential
Specimen	22 23 24 24	0,0 0,0 0,0	30.55 30.55 30.55	69 70 71

TABLE 7 (cont.)

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Figure 15 The Effect of Temperature on the Smooth-Bar Mechanical Properties of Forging "A" Test Ring



Figure 16 The Effect of Temperature on the Notched-Bar Mechanical Properties of Forging "A" Test Ring



The Effect of Temperature on the Smooth-Bar Mechanical Properties of Forging "A" Center Section



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Comparison of Forging "A" Test Ring and Center Section Mechanical Properties

Axial Thomastine HAH Cadial Maria Canalita -320 Tangential. Corrigina. Poncinc ### Post Ping Podd ol A Page contration (RT 1.h (+320) 4 P(+420 1.2 **4** (-320) -420) (-420) 0.8 35 1:0 65 50 ៩៩ 60

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VIELD SUBBIORH (PST x 1000)

Figure 20 Forging "A" Notch-Yield Ratio vs Yield Strength

b. The ultimate strength increased as the testing temperature was decreased and at approximately the same gradual rate as the yield strength. The 7079 alloy experienced a low work hardening rate, as indicated by the gradual increase with a lowering of temperature.

c. Ductility, as measured by elongation and area reduction, exhibited a reverse trend -- a decrease with lowering of temperature. This is uncommon for a wrought, face-centered-cubic alloy where ductility does not change appreciably with a temperature decrease.

d. The notch-tensile strength and the notch-tensile (notched tensile strength/smooth bar tensile strength) and notch-yield (notched tensile strength/smooth bar yield strength) ratios decreased as the testing temperature was decreased. This is an indication of reduced toughness. The toughness, as measured by notch-yield ratio, is quite good at ambient temperature, is marginal at -320° F, and is poor at -423° F. There was considerable scatter of notch-tensile data at very low temperature, partly because of the relatively poor toughness of Grade 7079.

e. The forging exhibited an anisotropic condition, which is illustrated by the variance of the smooth-bar properties of specimens in the axial, radial, and tangential orientations. The properties were generally highest in the grain flow (axial) direction as is common in other alloys.

f. The mechanical properties were strongly influenced by the mass-quench effect. This is illustrated by the significant variation of the center-section and test ring properties at the corresponding testing temperatures, as shown in Figure 19. The test ring generally had higher strength and lower ductility than the center section. The variation was consistent in the axial, radial and tangential specimen orientations. In a subsequent section of this report it will be shown that the properties are influenced by grain size, which is controlled by hot-working.

g. The notch-yield ratio is an inverse function of yield strength, as shown in Figure 20. Yield strength and ductility are also inversely related. For cryogenic service, a material with a high notch-yield ratio (greater than unity) as well as high yield strength is desired. This condition is not satisfied with Grade 7079 in forgings of the size under consideration.

Recent data (4) related the notch-yield ratio and yield strength for 6061, 5456, 2014, 2019, 7039, and X7106 in a similar notch-yield ratio versus yield-strength plot. Highest notch-yield ratios were obtained with the lowest yield strengths in these aluminum alloys. Adequate combination of high notch-yield ratios and high yield strengths are not observed in these commercial aluminum alloys at very low temperatures.

⁽⁴⁾ Campbell, J.E., <u>Properties and Applications of Aluminum Alloys at Low</u> <u>Temperatures</u>, Battelle Memorial Institute, 1964.

2. Mechanical Properties of Hand Forging "B"

Tensile data for the test ring are listed in Table 8 and shown graphically in Figure 21.

The strengths are slightly higher and the ductilities are lower than those of the Hand Forging "A" test ring. The relative position of the test rings with respect to the periphery of the forging is shown in Figure 9. Specimens of Hand Forging "B" were taken closer to the surface where greater hardening response and mechanical working resulted in higher strength.

In agreement with previous data, strength and ductility are influenced in a manner similar to that described previously upon exposure to low temperatures. At low temperatures, the YS/UTS ratios (yield strength to ultimate strength in a smooth-bar test specimen) were notably high, which is undesirable in a cryogenic alloy.

Based upon notch-yield criteria, the test ring is relatively notch tough at ambient temperature and when axially oriented at -320° F. Notchsensitivity is apparent in the radial and tangential orientations at -320° F, and in all orientations at -423° F.

3. Mechanical Properties of Hand Forging "C"

The low temperature tests described earlier were performed using a slightly smaller forging (see Figure 10). Properties at the test ring, top, center, and bottom locations were studied. These data are listed in Tables 9 and 10 and illustrated graphically in Figures 22 through 24.

The test results indicate the following significant features:

a. The data are in agreement with previous findings regarding the effect of temperature upon mechanical properties. It confirmed that when the temperature is lowered, strengths and notch sensitivity are increased while ductility is decreased.

b. Highest strengths were obtained in the test ring. Lowest strength, superior toughness, and highest ductilities were obtained at the center section. The top and bottom section properties were essentially equivalent.

Generally, when comparing the center-section properties of the impeller and inducer forgings (viz., Forgings "A" and "C", respectively), Forging "A" has inferior strength but superior ductility (see Figures 25 and 26). Highest strengths are obtained near the quenched surface while ductility variations are smaller (see Figure 27).

All of the above observations are based upon experimental results and allow the following general conclusions to be drawn. The 7079-T652 forgings have relatively poor toughness at a temperature of -423°F. Also,

Notch Yield Ratio		3 1.52		3 1.41	
Notch Tensile Ratio		K _t = 6. 1.25		Kt = 6 1.25	
Reduction of Area $(\frac{d}{ ho})$	22.4 19.8 16.1 17.4				9.50 7.50 7.50 7.50 7.50 7.50 7.50 7.50 7
Elongation (% in 4D)	10.5 12.0 12.0 10.4	ensile specimen ensile specimen ensile specimen ensile specimen	NNN44 00000	ensile specimen ensile specimen ensile specimen ensile specimen	00000 80000
0.2% Yield Strength (ksi)	62.5 62.4 61.0 62.0	Notched t Notched t Notched t Notched t	63.5 63.5 62.2 63.5 63.5 63.5 63.5 63.5 63.5 63.5 63.5	Notched t Notched t Notched t Notched t	64.4 61.1 60.9 62.0 62.0
Ultimate Strength (ksi)	75.5 76.3 73.1 73.1	93:9 94:11 94:12 94:12	71.1 72.0 72.1 79.6 71.2	83.4 85.8 87.3 887.3	74.4 71.2 71.3 70.9 72.0
Test Temp (°F)	RT RT RT RT Avg.	RT RT RT RT AVG.	RT RT RT RT Avg.	RT RT RT RT Avg.	RT RT RT RT AVG.
Orientation	Axial Axial Axial Axial	Axial Axial Axial Axial	Radial Radial Radial Radial	Radial Radial Radial Radial	Tangential Tangential Tangential Tangential
Specimen	よる ちょ	4000 1000	002 th 1 t t t	4 5 5 5 6 7 7 7 7 7 7 7 7 7 7 7 7 7 7 7 7	25 26 28

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MECHANICAL PROPERTIES OF 7079-T652 FORGING "B" TEST RING

TABLE 8

Notch Yield Ratio	1.23		1.19		0.87
Notch Tensile Ratio	Kt = 6.3		Kt = 6.3 1.03		K _t = 6.3 0.86
Reduction of Area (%)	1.10	00000 00000		9.5.4.7.0 9.5.4.7.6	
Elongation (% in 4D)	censile specimen censile specimen censile specimen	u u u u u u n n n n n n	ensile specimen ensile specimen ensile specimen	00000 00000 00000	ensile specimen ensile specimen ensile specimen
0.2% Yield Strength (ksi)	Notched t Notched t Notched t	72.5 71.1 75.2 72.9	Notched t Notched t Notched t	 78.1 71.8 71.3 73.7	Notched t Notched t Notched t
Ultimate Strength (ksi)	77.8 76.9 74.4 76.4	83.3 80.9 87.4 86.7 86.7	88.5 83.4 86.8	74.7 78.1 75.4 73.5 73.5	64.1 67.9 61.6 64.5
Test Temp (°F)	RT RT RT Avg.	- 320 - 320 - 320 - 320 Avg.	- 320 - 320 - 320 Avg.	- 320 - 320 - 320 Avg.	
Orientation	Tangential Tangential Tangential	Axial Axial Axial Axial	Axial Axial Axial Axial	Radial Radial Radial Radial	Radial Radial Radial Radial
Specimen	61 62 63	50 0 0 0	4 4 4 4 4 4 4 4 4 4 4 4 4 4 4 4 4 4 4	17 19 20 20	2025 2025 2020 2020 2020 2020 2020 2020

TABLE 8 (cont.)

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Notch Yield Ratio		3 0.67		3 0 . 86	
Notch Tensile Ratio		K _t = 6. 0.65		K _t = 6. 0.80	
Reduction of Area (β)	๛๛๛๛๛ ๛๛๛๛๛๛		10000 10000 10000		00000
Elongation (% in 4D)	waaaa orooro	censile specimen censile specimen censile specimen censile specimen	(1) (1) (1) (1) (1) (1) (1) (1) (1) (1)	censile specimen censile specimen censile specimen censile specimen	1-1-0 0-0 0.0 2.0
0.2% Yield Strength (ksi)	73.2 73.4 78.1 74.9	Notched t Notched t Notched t Notched t	77.0 84.3 83.6 81.9 81.9	Notched t Notched t Notched t Notched t	77.8 (2) (2) (2) (2) 77.8
Ultimate Strength (ksi)	75.8 77.1 75.8 75.8 76.8	50.2 50.7 50.7 50.7 50.7	86.9 89.9 88.1 0	49.2 68.5 87.1 70.0	78.1 73.2(3) 76.4(3) 78.7(3) 78.1
Test Temp (°F)	- 320 - 320 - 320 - 320 Avg.	- 320 - 320 - 320 Ave.	-423 -423 -423 -423 Avg.	-423 -423 -423 -423 Avg.	-423 -423 -423 -423 Avg.
Orientation	Tangential Tangential Tangential Tangential	Tangential Tangential Tangential Tangential	Axial Axial Axial Axial	Axial Axial Axial Axial	Radial Radial Radial Radial
Specimen	8 N N 0 9 N 0 0	69 66 66 66 66 66 66 66 66 66 66 66 66 6	6 1 1 1 1 1 1 1 1	84 02 たたた	21 25 25 25 25

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TABLE 8 (cont.)

Notch Yield Ratio	3 0.695		3 0.428
Notch Tensile Ratio	Kt = 6. 0.69		K _t = 6. 0.426
Reduction of Area (%)		000000	
Elongation (% in 4D)	censile specimen censile specimen censile specimen ensile specimen	000000000000000000000000000000000000000	ensile specimen ensile specimen ensile specimen ensile specimen
0.2% Yield Strength (ksi)	Notched t Notched t Notched t Notched t	(2) (2) (2) 81.7 81.7	Notched t Notched t Notched t Notched t
Ultimate Strength (ksi)	СС4 СС4 СС С С С С С С С С С С С С С С	75.8(3) 60.6(3) 78.1(3) 82.3 82.3	27.6 46.9 35.0 37.0
Test Temp (°F)	-423 -423 -423 -423 Avg.	-423 -423 -423 -423 Avg.	-423 -423 -423 -423 Avg.
Orientation	Radial Radial Radial Radial	Tangential Tangential Tangential Tangential	Tangential Tangential Tangential Tangential
Specimen	ううい での の の	965544 30334	69 71 72

TABLE 8 (cont.)

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NOTES:

Specimen fractured outside of the gage mark. No yield obtained in specimen during stressing. Not included in average of results. H N M





	Notch Yield Ratio		1.29			0.805
	NOTCH TENSILE Ratio		1.12			0.75
	STRESS CONCENTR. (KT)		יידי מידי מ מידי מי			7.3 6.6 6.2
	REDUCTION OF AREA (\$)	1.0 1.0 1.0 1.0 1.0 1.0 1.0 1.0 1.0 1.0	2 7 7			z z z z
	ELONGATION (2 in ⁴ D)	0.9 0.7 7.7	ENSILE SPECIMEI Ensile Specimei Ensile Specimei		8.0 2.0 3.0 3.0	ENSILE SPECIME Ensile Specime Ensile Specime Ensile Specime
0.2% ҮІЕLD Strength (ksi) 66.6 63.9 67.9 66.1	NOT CHED T Not CHED T Not CHED T		87.9 82.5 83.2 87.7	NOT CHED T NOT CHED T NOT CHED T NOT CHED T		
	ULT IMATE Strength (KSI)	76.4 74.5 76.4 75.8	84.6 83.4 87.1 85.0		93.4 91.7 94.8 94.5	61.8 62.7 77.9 79.0 70.6
	TEST TEMP (^O F)	RT RT RT Avg.	RT RT RT Avg.		-423 -423 -423 -423 Ave.	- 423 - 423 - 423 - 423 - 423
	AREA	РЕК І РНЕ КҮ РЕК І РНЕ КҮ РЕК І РНЕ КҮ	РЕ В І РНЕ В Ү РЕ В І РНЕ В Ү РЕ В І РНЕ В Ү		PERIPHERY PERIPHERY PERIPHERY PERIPHERY	РЕКІРНЕКҮ РЕКІРНЕКҮ РЕКІРНЕКҮ РЕКІРНЕКҮ
ING NO. 1	ORIENTATION	Tangent i al Tangent i al Tangent i al	Tangent i al Tangent i al Tangent i al	RING NO. 2	Tangential Tangential Tangential Tangential	Tangential Tangential Tangential Tangential
A. TEST R	SPECIMEN	24-C 26-C 28-C	23-C 25-C 27-C	В. Тезт Р	10-4 12-4 14-5 16-8	9-8 11-8 13-8

MECHANICAL PROPERTIES OF 7079-T652 FORGING "C" TEST RING

TABLE 9

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MECHANICAL PROPERTIES OF 7079-T652 FORGING "C" TOP, CENTER, AND BOTTOM SECTIONS

TOP SECTION

A. TOP	SECTION		TEST	ULTIMATE	0.2% YIELD		REDUCTION	STRESS	NOTCH	Not CH
SPECIMEN	ORIENTATION	AREA	TEMP (^o f)	STRENGTH (KSI)	STRENGTH (KSI)	ELONGATION (% IN ¹ 4D)	OF AREA (%)	CONCENTR. (KT)	TENSILE RATIO	YIELD Ratio
~	TANGENT I AL	1/2 RADIUS	RT	63.4	54.6	0.6	13.0			
4	TANGENTIAL	1/2 RADIUS	RT	6•99	NOT CHED	FENSILE SPECIME	z	ti • ti	96.0	1.22
2	AXIAL	1/2 RADIUS	RT	65.4	51.1	6.5	13.1			
ŝ	AXIAL	1/2 RADIUS	RT	64.6	NOTCHED	ENSILE SPECIME	z		66•0	1.26
16	AXIAL	3/4 RADIUS	RT	64.1	1.9th	11.0	14.6			
m	RADIAL	1/2 RADIUS	RT	67.3	53.1	11.5	15.9			
9	RADIAL	1/2 RADIUS	RT	78.7	NOT CHED	ENSILE SPECIME	z	5•2	7.1.	1.48
18	RADIAL	3/4 RADIUS	RT	60.2	NOT CHED	ENSILE SPECIME	z	h•2		
1A 2A	Tangent i al Tangent i al	1/2 RADIUS 1/4 RADIUS	-320 -320 Avg.	78.4 75.6 77.0	 4.99 66.4	0 0 0 9 9 0 9 9 9	3.3 4.0 3.7			
7A 8A	Tangent ial Tangent ial	1/4 RADIUS 1/4 RADIUS	-320 -320 Avg.	78.3 80.1 79.2	NOT CHED 1 Not CHED 1	ENSILE SPECIME.	zz	6•3 6•3	1.03	1.19
44	AXIAL Axial	1/4 RADIUS 1/4 RADIUS	-320 -320 Ave.	78.3 79.9 75.8	68.3 60.2 64.3	2.5 2.5 2.5	ა ა ა ი ი ი ი ი ი			

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SPECIMEN	OR I ENTATION	AREA	TEST TEMP (^o F)	ULT IMATE Strength (KSI)	0.2% YIELD Strength (KSI)	ELONGATION (& IN 4D)	REDUCTION OF AREA (%)	STRESS CONCENTR. (KT)	NOTCH Tensile Ratio	NOTCH YIELD RATIO
\$.\$	AXIAL AXIAL	1/4 RADIUS 1/4 RADIUS	-320 -320 Avg.	72.7 65.0 68.9	NOT CHED NOT CHED	TENSILE 3PECIME Tensile specime	ZZ	6•3 6•3	0.91	1.07
9 8 104	RADIAL Radial	1/4 RADIUS 1/4 RADIUS	-320 -320 Avg.	78.7 78.5 78.6	68.5 66.7 67.6	0°04 0°04 0°1	4.7 4.8 4.75			
11A 12A	RADIAL Radial	1/4 RADIUS 1/4 RADIUS	-320 -320 Avg.	87.6 84.7 86.2	NOT CHED NOT CHED	TENSILE SPECIMI Tensile Specimi	Z Z W W	6.3 6.3	1.10	1.27
13	Tangent i al Tangent i al	1/2 RADIUS 1/2 RADIUS	-423 -423 Avg.	83.9 80.3 82.1	72.0 71.0 71.5	3.0 2.0 2.5	ביביב מיטיט מיטיט			
10	TANGENTIAL	1/2 RADIUS	-423	56.6	NOT CHED	TENSILE SPECIM	Z	۲°8	0.69	0.791
17	TANGENTIAL	3/4 RADIUS	-423	80.7	64.4	3•0	2.4			
8 14(1)	Ax1aL Ax1aL	1/2 RADIUS 1/2 RADIUS	-423 -423 Avg.	19.4 78.6 79.0	65.2 65.2	8.5 2.0 8.5				
:1	AXIAL	1/2 RADIUS	-423	58.0	NOT CHED	TENSILE SPECIM	Z W	4°9	0.73 ⁴	0.89
19	AXIAL	3/4 RADIUS	-423	74.1	NOTCHED	TENSILE SPECIM	NB			
12	RADIAL	1/2 RADIUS	-423	6.07	NOF CHED	TENSILE SPECIM	EN	7.T	0.86	1.01

TABLE 10 (CONT.)

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SPECIMEN	ORIENTATION	AREA	TEST TEMP (^o f)	ULTIMATE Strength (ksi)	0.2% YIELD Strength (#Si)	ELONGATION (\$ IN 4D)	REDUCTION OF AREA (\$)	STRESS Concentr. (K _T)	NOTCH TENSILE RATIO	NOTCH YIELD RATIO
6 <u>(</u>	RADIAL RADIAL	1/2 Radius 1/2 Radius	-423 -423 AVG.	83.0 82.6 82.8	72.2 68.4 70.3	2.0 3.0 2.5	3.1 3.2 3.15			
B. CENI	FER SECTION									
26	TANGENTIAL	1/4 RADIUS	RT	64.6	48.7	8.5	13.8			
28	TANGENTIAL	1/2 RADIUS	RT	66.6	50.9	12.0	16.7			
31	TANGENTIAL	1/2 RADIUS	RT	76.7	NOTCHED 1	LENSILE SPECIM	E N	6.1	1.15	1.5
27	AXIAL	1/4 RADIUS	RT	64.0	48 . 5	10.0	11.5			
59	AXIAL	1/2 RADIUS	RT	62.0	7.44	0•6	13.1			
32	AXIAL	1/2 RADIUS	RT	62.1	NOT CHED	LENSILE SPECIM	U.S.	₽ + 5	1.0	1.39
1 13	AXIAL	3/4 RADIUS	RT	62.3	45.9	8.5	14.5			
30	RADIAL	1/2 RADIUS	RT	64.8	4.84	12.0	16.7			
33	RADIAL	1/2 RADIUS	RT	61.9	NOTCHED	LENSILE SPECIM	ĒN	4 .9 5	1.15	1.39
¹⁴⁵	RADIAL	3/4 RADIUS	RT	62.2	1+6.7	0-1	13.7			
178 188	Tangent I al. Tangent I al	1/4 RADIUS 1/4 RADIUS	-320 -320 Avg.	79.05 79.1	66.1 66.0 66.05	0 0 0 6 0 0	₩ • • • 8			

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TABLE 10 (CONT.)

Not CH Y I ELD RAT I O	1.21		1.28		1.25		0.914		1.01
NOTCH TENSILE RATIO	1.01		1.02		1.03		47.0		0.786
STRESS Concentr. (KT)	6.3 6.3		6•3 6		6.3 6.3		4.6		4 •9
REDUCTION OF AREA (\$)		০ ব ব ব ব ব ব ব		4.0 4.7 4.35		h.o			
ELONGATION (\$ IN 4D)	FENSILE SPECIMEN Tensile Specimen	۵۰5 ۳۰5 ۳۰5	rensille specimen Fensille specimen	4.0 3.5 3.75	ENSILE SPECIMEN Ensile specimen	2.0	ENSILE SPECIMEN	2°0 5°0	ENSILE SPECIMEN
0.2% YIELD Strength (KSI)	NOT CHED 1 Not ched 1	56.0 57.4 56.7	NOT CHED T Not ched T	63.6 64.2 63.9	NOTCHED T Notched T	69•0	NOT СНЕВ Т	65.8 65.2 65.5	NOT CHED T
ULT IMATE Strength (ksi)	82.0 77.7 79.9	71.2 71.4 71.3	69.4 75.6	77.3 77.2 77.25	4°0,4 4°08 79.9	84.9	63•0	84.7 84.18 4.48	66.4
TEST TEMP (^o F)	-320 -320 Avg.	-320 -320 Avg.	-320 -320 Avg.	-320 -320 Avg.	-320 -320 Avg.	-423	-423	-423 -423 Avg.	-423
AREA	1/4 RADIUS 1/4 RADIUS	1/4 RADIUS 1/4 RADIUS	1/4 RADIUS 1/4 RADIUS	1/4 RADIUS 1/4 RADIUS	1/¼ RADIUS 1/¼ RADIUS	1/4 RADIUS	1/4 RADIUS	1/2 RADIUS 1/2 RADIUS	1/2 RADIUS
ORIENTATION	Tangent i al Tangent i al	AXIAL Axial	AXIAL AXIAL	RADIAL Radial	Radial Radial	TANGENTIAL	TANGENTIAL	Tangent i al Tangent i al	TANGENTIAL
SPECIMEN	198 208	138 146	158 168	218 228	238 248	20	23	34 10	37

TABLE 10 (CONT.)

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			TEST TEMP	ULTIMATE Strength ////////////////////////////////////	0.2% YIELD Strength	ELONGATION	REDUCTION OF AREA	STRESS Concentr. (K+)	NOT CH TENSILE RATIO	NOTCH YIELD Ratio
		AREA	<u> </u>	, us	TIEN	TOL NI ST	727			
1 11	TANGENTIAL	3/4 RADIUS	-423	88.6	67.5	3•0	5.0			
22(2)	AXIAL	1/4 RADIUS	-423	74 . 0	59.4	2•0	2°7			
25	AXIAL	1/4 RADIUS	-423	54.1	Noт снер т	ENSILE SPECIM	EN	4.5	0.73	0.91
35(2) 41(3)	AXIAL Axial	1/2 RADIUS 1/2 RADIUS	-423 -423 AVG.	74.0 73.2 73.6	59.0 60.7 59.9	0.0 v. 0.0 v. 0.0 v.	ב ב ב מ מ מ מ מ מ			
38	AXIAL	1/2 RADIUS	-423	60.6	NOT CHED T	ENSILE SPECIM	EN	5.8	0.823	1.01
91	AXIAL	3/4 RADIUS	-423	87.2	66.3	3•0	3•2			
21	RADIAL	1/4 RADIUS	-423	82.1	71.3	2.0	2.4			
24	RADIAL	1/4 RADIUS	-423	65.6	NOT CHED T	ENSILE SPECIM	Z	4.8	0.80	0,92
36 42	RADIAL Radial	1/2 RADIUS 1/2 RADIUS	-423 -423 Avg.	80.9 81.7 81.3	64.4 65.0 64.7	3•0 2•5 2•5	4.0 3.2 3.6			
39	RADIAL	1/2 RADIUS	-423	60 . 4	NOT CHED T	ENSILE SPECIM	Z	5.1	447.0	۰۰93 ۵
С. Вотт	OM SECTION									
Ļμ	TANGENTIAL	1/2 RADIUS	RT	61.5	2.92	8.5	11.5			
ጽ	TANGENTIAL	1/2 RADIUS	RT	71.1	NOTCHED 1	ENSILE SPECIM	EN	t • t	1.05	1.25

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TABLE 10 (CONT.)

TABLE 10 (CONT.)

			TEST TEMP	ULTIMATE Strength	0.2% YIELD Strength	ELONGATION	REDUCTION OF AREA	STRESS Concentr.	NOT CH TENSILE	Νοτ сμ Υιειρ
SPECIMEN	OR I ENTATION	AREA	(°F)	(KSI)	(KSI)	(CH NI %)	(%)	(K ₁)	RATIO	RATIO
81	AXIAL	1/2 RADIUS	RT	64.4	6.94	1.5	13.7			
51	AXIAL	1/2 RADIUS	RT	66.0	NOTCHED .	LENSILE SPECIM	Z	4.7	1.03	1.32
62	AXIAL	3/4 RADIUS	RT	64 . 2	50.1	8.5	13.7			
611	RADIAL	1/2 RADIUS	RT	68.4	57.2	8.0	10.8			
52	RADIAL	1/2 RADIUS	RT	70.2	NOT CHED	LENSILE SPECIM	Z	4.2	1.03	1.23
64	RADIAL	3/4 RADIUS	RT	65.1	51•3	10.0	15.3			
33 34	Tangent i al Tangent i al	1/4 RADIUS 1/4 RADIUS	-320 -320 Avg.	77.3 78.4 77.9	63.6 63.1 63.4	ب د. م. م. م.	5.9 5.9			
35 36	Tangent i al Tangent i al	1/4 RADIUS 1/4 RADIUS	-320 -320 Avg.	80.7 83.8 82.3	NOT CHED . NOT CHED .	FENSILE SPECIMI	ζ Ζ	6.3 6.3	1.06	1.30
30 53	AXIAL Axial	1/4 RADIUS 1/4 RADIUS	-320 -320 Avg.	75.1 76.3 75.7	65.6 62.9 64.3	0.0 W. W. 4. W.	0 9 8 8 6 0 0			
31 32	AXIAL AXIAL	1/4 RADIUS 1/4 RADIUS	-320 -320 Avg.	84.7 83.0 83.9	NOT CHED	FENSILE SPECIMI FENSILE SPECIMI	5 Z	6.3 6.3	1.11	1.31
X) 55 55	RADIAL Radial	1/4 RADIUS 1/4 RADIUS	-320 -320 Avg.	73.7 71.8 72.8	61.6 59.1 60.4	0 0 0 8 8 8	າ. 8 1. 0			

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۲. ۲.	GAGE MAR	I FAILED AT	SPECIMEN	ۍ ف		SUIC	I THE RAD	AS UNDERCUT IN	3. SPECIMEN W	
	ED	.NI IETER SLIPP	EXTENSON	5.	ED WITH LH2	ETELY COVERI ISOR	GUID SEN	AY NOT HAVE BE Hort in the Li	2. SPECIMEN M. DUE TO A SI	
EXTENSOMETER	POINT OF	I FAILED AT	SPECIMEN	• 17			-	ER FROZE	1. EXTENSOMET	NOTES:
								1		
			0.8	ł	71.2	78.7	-423	1/2 RADIUS	RADIAL	(9)
·6 0.745	0.67	3.2		TENSILE SPECIMEN	NOTCHED	58.0	-423	1/2 RADIUS	RADIAL	58
			1.6	2.0	6•11	85.7	-1;23	1/2 RADIUS	RADIAL	55
			1.6	1.0	68 . 4	11.5	-423	3/4 RADIUS	Axial	65
15 0.87	17.0	3•5		TENSILE SPECIMEN	NOTCHED	54.2	-423	1/2 RADIUS	Axial	57
			ьт 0 0 0	1.0 2.0	64.2 60.7 62.5	75.7 76.0 75.9	-423 -423 Avg.	1/2 Radius 1/2 Radius	AXIAL Axial	54(4) 60
			1.6	1.0	21.5	81.7	-423	3/4 RADIUS	TANGENTIAL	63
3 0 . 84	0.73	3.1		TENSILE SPECIMEN	Not сн ер	57.2	-423	1/2 RADIUS	TANGENTIAL	56
			トトト の ろろろ	3.0 2.0	68 . 2 68 . 2	79.1 77.3 78.2	-423 -423 Avg.	1/2 RADIUS 1/2 RADIUS	Tangential Tangential	53 59(5)
1.20	6.0	6.3 6.3		TENSILE SPECIMEN Tensile specimen	NOT CHED NOT CHED	72.5 72.7 72.6	-320 -320 Avg.	1/4 RADIUS 1/4 RADIUS	Radial Radial	27 cx 28 c x
H NOTCH LE YIELD O RATIO	Nord	STRESS Concentr (K _T)	REDUCTION OF AREA (%)	ELONGATION (& IN ¹⁴ D)	0.2% Y1ELD Strength (ksi)	ULTIMATE Strength (KSI)	TEST TEMP (°F)	AREA	ORI ENTATION	N N N N N N N N N N N N N N N N N N N

TABLE 10 (CONT.)

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SPECIMEN WAS UNDERCUT IN THE RADIUS

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The Effect of Temperature on the Smooth Bar Mechanical Properties of Forging "C" at the Top Section Area



The Effect of Temperature on the Smooth Bar Mechanical Properties of Forging "C" at the Center Section Area



The Effect of Temperature on the Smooth Bar Mechanical Properties of Forging "C" at the Bottom Section Area







Figure 26

Comparative Forging "C" Test Ring Mechanical Properties



Figure 27 Effect of Forging Thickness on the Room- and Low-Temperature Mechanical Properties of Forging "C"

the toughness does not vary greatly in forgings of the two different sizes; the impeller (30-in. diameter) and the inducer (20-in. diameter). Properties of extremely large 7079 forgings are influenced in part by orientation, specimen location, and mass-quench effect.

4. Mechanical Properties of Hand Forging "D"

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The tensile data for Hand Forging "D" (Figure 10) are listed in Table 11. Pronounced variations in strength and ductility at ambient temperature and -423°F are evident. The effect of cryogenic temperatures upon the test ring mechanical properties is in agreement with previous results.

Although the 7079 test ring was notch tough (based on notchyield ratio) at ambient temperature, it was notch sensitive at $-423^{\circ}F$.

5. <u>Effects of -T6 Temper Reheat-Treatment (After</u> Rough Machining) on Properties of Hand Forging "E"

As previously stated, the tensile strength of the center sections was considerably lower than the tensile strength of the peripheral test rings. Therefore, tests were performed to determine whether improvements in strength and ductility would result when the -T6 temper reheat treatment was used after rough machining to remove extraneous material.

The properties of the control specimens and the -T6 reheattreatment specimens are listed in Table 12. The control specimen test results compared favorably with those of the Forging "C" center section, which is of equivalent "as forged" dimensions. Both strength and ductility are good.

As shown in Figure 28, reheat treatment produced the following effects upon axial properties:

a. At ambient temperature, ultimate and 0.2% offset yield strengths were increased approximately 12% and 28%, respectively.

b. A similar percentage increase in strength was noted at cryogenic temperatures.

c. Ductilities were lowered. Ambient temperature elongation and reduction of area were lowered approximately 33% and 20%, respectively. Axial reduction of area was lowered approximately 42% at -320° F.

d. Notch-yield ratio was lowered from approximately 1.28 to 0.87 at $320^{\circ}\mathrm{F}$.

The increase of strength is ascribed to the mass-quench effect. The -T6 treatment of a smaller rough machined section resulted in faster cooling from solution treatment; therefore, higher properties were obtained after aging.

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FCHANICAL PROPERTIES OF 7079-T652 FORGING "D" TEST RING

TABLE 11

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TABLE 12

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MECHANICAL PROPERTIES OF 7079-T652 FORGING "E" HEAT-TREATED TO THE -T6 TEMPER CONDITION AFTER ROUGH MACHINING

A. Control Specimens (-T652 Temper Condition)

rientation	Test Temp (^{OF})	Ultimate Strength (ksi)	0.2% Offset Yield Strength (ksi)	Elongation (% in 4D)	Reduction of Area (%)
	RT RT	67.4 67.4	51.0		134. 134. 134. 134. 134. 134. 134. 134.
	Avg.	e6.1	50.9	7•4	16.7
러려	- 320 - 320	76.6 76.9	63.4 63.4	0 • • •	
	Avg.	76.8	63.4	2.4	7.5
	RT RT	63.5 62.4	45.6 46.0	6•3 7•0	7.4 11.9
	Avg.	65.9	45.8	6.7	7.6
	- 320	65.4 67.7	56.0 52.7	1.6 3.1	.0 .0 .0
	Avg.	66.6	54 .4	2.4	ł, "8

B. HEAT	TREATED SF	CIMENS FROM TE	ST RINGS	(-Тб Темрек	CONDITION) (HEAT	TREATED AFTER	Коисн Маснік	1 I NC)		
SPECIMEN	TEST RING NUMBER	SPECIMEN ORIENTATION	TEST TEMP (^o F)	ULTIMATE Strength (ksi)	0.2% OFFSET Yield Strength (ksi)	ELONGATION	REDUCTION OF AREA (%)	STRESS CONC. (KT)	NOT CH TENSILE RATIO	NOT CH YIELD RATIO
- N M	~ ~ ~	AXIAL AXIAL AXIAL	RT RT RT	71.1 71.8 71.3	63.6 63.8 63.7	1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1	9.9 4.9 7.8			
			Avg.	71 . 4	63.7	4.5	7.8			
5 7 N 5	₩ ₩ ₩	AXIAL AXIAL AXIAL	RT RT RT	81.5 78.4 84.7	NOTCHED TENSI Notched tensi Notched tensi	LE SPECIMEN LE SPECIMEN LE SPECIMEN		6.3 6.3 6.3		
			A vg.	81.5					1.14	1.28
すらるて		AXIAL AXIAL AXIAL AXIAL	-320 -320 -320	79.2 78.8 79.9	74.6 74.4 74.8	0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0	~~~~~ ~~~~~~~~~~~~~~~~~~~~~~~~~~~~~~~~			
			Avg.	79.3	74.6	2.5	2.8			
15 15 18		AXIAL AXIAL AXIAL AXIAL	-320 -320 -320	61.8 64.0 63.3 69.4	NOTCHED TENSI NOTCHED TENSI NOTCHED TENSI NOTCHED TENSI	LE SPECIMEN LE SPECIMEN LE SPECIMEN LE SPECIMEN		6.3 6.3 6.3 6.3 6.3 7 6.3 7 7 7 7 7 7 7 7 7 7 7 7 7 7 7 7 7 7 7		
			Avg.	64.6					0.81	0.866

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TABLE 12 (CONT.)

HEAT TREATED SPECIMENS FROM TEST RINGS (-T6 TEMPER CONDITION) (HEAT TREATED AFTER ROUGH MACHINING)

	TEST		Test	ULTIMATE	0.2% OFFSET		REDUCT I ON	Stress	Notch	Notch
SPECIMEN NUMBER	RING	SPECIMEN ORIENTATION	TEMP (°F)	STRENGTH (KSI)	YIELD STRENGTH (KSI)	ELONGATION	OF AREA (\$)	CONC.	TENSILE	YIELD RATIO
8	-	AXIAL	-423	83.5	80.1	1.0	0.8			
6	~	AXIAL	-423	82.9	82.5	1.0	1.6			
01	-	AXIAL	-423	83.7	80.1	1.0	1.6			
11	-	AXIAL	-42 <u>3</u>	83.5	80.9	1.0	1.6			
			Avg.	83.4	80.9	1.0	t, . r			
19	-	AXIAL	- 423	55.2	NOTCHED TENSI	LE SPECIMEN		6.3		
20	-	AXIAL	-423	53.2	NOTCHED TENSI	LE SPECIMEN		6 . 3		
21	-	AXIAL	-423	63.4	NOTCHED TENSI	LE SPECIMEN		6.3		
22	-	AXIAL	-423	56.6	NOTCHED TENSI	LE SPECIMEN		6.3		
			A VG.	57.1					0.684	0.706
23	cu	RADIAL	RT	72.8	70.4	6.3	13.1			
54	2	RADIAL	RT	72.8	67.2	2.0	13.2			
25	N	RADIAL	RT	73.1	68.4	6.3	1,4			
26	0	RADIAL	RT	72.1	66.2	6.3	13.4			
27	Q	RADIAL	RT	73.9	67.4	5.5	13.4			
28	N	RADIAL	RT	72.3	68.1	5.5	12.3			
			Avg.	72.8	61.9	6.2	13.3			
59	Q	RADIAL	-320	82.7	76.4	1.6	2•5			
õ	ง	RADIAL	-320	80.3	73.9	1.6	5.0			
5	ง	RADIAL	-320	81.3	73.6	1.6	6.0			
32	5	RADIAL	-320	79.6	74.9	1.6	5.0			
33	5	RADIAL	-320	78.1	74.6	1.6	5.0			
34	0	RADIAL	-320	81.1	76.1	1.6	6.0			
			A VG.	80.5	74.9	1.6	4.9			

TABLE 12 (CONT.)

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NOTCH Yield Ratio				1.37				1.05				0.72
NOTCH TENSILE RATIO				1.2				16.0				0.67
STRESS Conc. (KT)				6.3			6.3 6.3				6.3 6.3	
REDUCTION OF AREA (%)	5 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5	2•35	12.3		2.4 3.2	2.8			0.8 1.6	1.2		
ELONGATION	2.0	1.5	0.7	LE SPECIMEN	2.0 2.0	2•0	LE SPECIMEN Le Specimen		1.0	1.0	ILE SPECIMEN	
0.2% OFFSET Yield Strength (ksi)	82.1 79.7	80.9	66.6	NOTCHED TENSI	80.2 78.2	79.2	NOTCHED TENSI Notched tensi		86.1 82.8	84.5	NOTCHED TENSI Notched tensi	
ULTIMATE Strength (ksi)	89 .9 89.5	89.7	75.6	1.1	86.2 83.8	85.0	78.6 86.9	82.8	92.4 88.9	7.06	66 . 4 55 . 5	6.09
TEST TEMP (^O F)	-423 -423	Avg.	RT	RT	-320 -320	A VG.	-320 -320	AVG.	-423 -423	A VG.	-423 -423	. A vg.
SPECIMEN	Tangent i al Tangent i al		TANGENTIAL	TA NGENT I AL	Tangent i al Tangent i al		Tangent i al Tangent i al		Tangent i al Tangent i al		Tangent i al Tangent i al	
TEST Ring Number	ო ო		4	ħ	ন ন		ন ক		7 7		オオ	
SPECIMEN Number	45 46		۲ţ	52	48 149		53 54		5 20		55 56	

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TABLE 12 (CONT.)

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Z z	TEST Ring Number	SPECIMEN ORIENTATION	TEST TEMP (^o f)	ULTIMATE Strength (KSI)	0.2% OFFSET Yield Strength (ksi)	ELONGATION	REDUCTION OF AREA (%)	STRESS CONC. (KT)	NOTCH TENSILE Ratio	NOTCH YIELD RATIO
	ß	TANGENTIAL	RT	٥٠ ۴۲	65.2	7.5	14.6			
	5	TANGENT I AL	RT	4.98	NOTCHED TENSI	LE SPECIMEN		6.3	1.2	1.37
	in in	Tangent i al Tangent i al	-320 -320	81.6 82.1	75.6 80.1	5°0 5	1.6 2.3			
			AVG.	81.9	6-11	2•0	1.95			
	5	Tangent i al	-320	73.7	NOTCHED TENSII	E SPECIMEN		6.3	6.0	0.946
	ഗന	Tangent i al Tangent i al	-423 -423	88 . 1 38.5	84.3 83.8	1.0	1.6 1.6			
			A VG.	88.3	84.05	1.0	1.6			
	n n	Tangent i al Tangent i al	-423 -423	53.8 63.9	NOTCHED TENSIL Notched tensil	E SPECIMEN E SPECIMEN		6.3 6.3		
			Avg.	58.9	-				٥.67	0.702
	VANES Vanes Vanes	RADIAL RADIAL RADIAL	RT RT RT	76.0 76.1 73.8	69.8 69.7 69.6	7.0 7.8 7.9	16.7 13.4 11.3			
			AVG.	75.3	69.7	7.5	13.8			

০° ন १° ন १	4.9	15.7 19.1 16.9	17.2	9. 4 . 4 . 4 . 4 . 4 . 4 . 4 . 4 . 4 . 4	3.9
1.6 1.6	1.6	7.0 4.7	8.1	1.6 1.6	1.6
7.77 7.77 7.67	78.4	 66.4 68.4	67 . lt	77.9 76.2	77.05
81.4 81.6 82.1	81.7	75.0 73.3 72.1	73.5	81.6 72.6 77.2	1.17
-320 -320 -320	A VG.	RT RT RT	Avg.	-320 -320 -320	AVG.
RADIAL Radial Radial		Tangent i al Tangent i al Tangent i al		Tangential Tangential Tangential	
VANES VANES VANES		VANES VANES VANES		VANES VANES VANES	
69 71		75 76 77		78 79 80	
	69 Vanes Radial -320 81.4 77.7 1.6 4.9 70 Vanes Radial -320 81.6 77.7 1.6 4.9 71 Vanes Radial -320 82.1 79.7 1.6 4.9 71 Vanes Radial -320 82.1 79.7 1.6 4.9	69 Vanes Radial -320 81.4 77.7 1.6 4.9 70 Vanes Radial -320 81.6 77.7 1.6 4.9 71 Vanes Radial -320 81.6 77.7 1.6 4.9 71 Vanes Radial -320 82.1 79.7 1.6 4.9	69 Vanes Radial -320 81.4 77.7 1.6 4.9 70 Vanes Radial -320 81.6 77.7 1.6 4.9 71 Vanes Radial -320 81.6 77.7 1.6 4.9 71 Vanes Radial -320 82.1 79.7 1.6 4.9 71 Vanes Radial -320 82.1 79.7 1.6 4.9 75 Vanes Tangential RT 78.4 1.6 4.9 75 Vanes Tangential RT 73.3 66.4 9.4 19.1 77 Vanes Tangential RT 72.1 68.4 7.8 16.9	69 Vanes Radial -320 81.4 77.7 1.6 4.9 70 Vanes Radial -320 81.6 77.7 1.6 4.9 71 Vanes Radial -320 81.6 77.7 1.6 4.9 71 Vanes Radial -320 82.1 79.7 1.6 4.9 7 Vanes Radial RT 75.0 81.7 78.4 1.6 4.9 75 Vanes Tangential RT 75.0 7.0 15.7 76 Vanes Tangential RT 73.3 66.4 9.4 19.1 77 Vanes Tangential RT 72.1 68.4 7.8 16.9 77 Vanes Tangential RT 72.1 68.4 7.8 16.9 77 Vanes Tangential RT 72.1 68.4 7.8 16.9 70 Yab Yab 7.6 7.6 17.2 19.1 70 Yab 7.6	69 Vanes Radial -320 81.4 77.7 1.6 4.9 70 Vanes Radial -320 81.6 77.7 1.6 4.9 71 Vanes Radial -320 81.6 77.7 1.6 4.9 71 Vanes Radial -320 82.1 79.7 1.6 4.9 75 Vanes Tangential RT 75.0 7.0 15.7 76 Vanes Tangential RT 75.0 7.0 15.7 77 Vanes Tangential RT 73.3 66.4 7.8 16.9 77 Vanes Tangential RT 73.3 66.4 7.8 16.9 77 Vanes Tangential RT 73.1 68.4 7.8 16.9 77 Vanes Tangential RT 72.1 68.4 7.8 16.9 78 Vanes Tangential <t< td=""></t<>

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TABLE 12 (CONT.)



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The Effect of -T6 Reheat-Treatment After Rough Machining on Room- and Low-Temperature Mechanical Properties of Forging "E"

The data shows that toughness was impaired by -T6 reheattreatment temper. Formerly notch tough at room temperature and -320° F, notchtoughness is apparent only at room temperature in the -T6 reheat-treatment temper. Both conditions display notch sensitivity at -423° F.

In considering the above described influence of -T6 reheat-treatment temper on alloy properties, the 7079 forging, in the section size and physical condition as tested, was superior in toughness when -T652 tempered.

6. <u>Microstructure</u>

The microstructure of Forging "C" at three locations is shown in Figures 29 through 31. The important features illustrated in these micrographs are a finer grain structure at the peripheral area, a slightly cored center section, and inclusions in the matrix.

The coarse, cored-grain structure apparently contributed to the lower strength of the forging center section. Although an investigation was not performed to study the effect of hot working, it appears that -T6 reheat treatment would have been more beneficial if the center-section grain had been refined by hot working prior to the -T6 reheat treatment. This is evident when the test ring properties of Forging "C" are compared with the reheat treated center section properties of Forging "E". The only apparent difference is the grain size because the two sections are at their maximum heat-treatment strength level; however, the test ring had superior properties.

It is concluded that grain size is equally as important as heat-treatment in controlling the properties of large 7079 forgings. Because grain size is primarily controlled by hot working, it is essential that billet stock be grain refined extensively at all areas to obtain a forging with higher properties.

Inclusions were found in the 7079 forgings. The poor toughness of 7079 forgings are ascribed to these inclusions as well as the cored structure and the high alloy content. Work of other investigators⁽⁵⁾ with 7000 series aluminum alloys also related the poor toughness of these microstructural conditions. These investigations also observed that 7079-T6 sheet (0.080-in. thick) possessed higher toughness, as measured by notch-tensile ratio, than a 7079-T6 forging (5.0-in. billct). This is probably because of the greater dispersal, orientation, and refinement of inclusions in the sheet (by the rolling operation) which minimized their notch effect in the high-strength matrix. The inclusions in the large 7079 forgings were generally segregated at grain boundaries and they appear to be less effective as stress-risers in a low strength matrix as compared with a high strength matrix. The superior

 ⁽⁵⁾ Christian, J. L., and Watson, J. F., "Properties of 7000 Series Aluminum Alloys at Cryogenic Temperatures," <u>Advances in Cryogenic Engineering</u>, vol. 6, pp. 604-621, 1960.



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Figure 29 Microstructure of Forging C (Peripheral Location) Magnification: 100X Etchant: Flick's



Figure 30 Microstructure of Forging C (One-Half Radius Location) Magnification: 100S Etchant: Flick's



Figure 31 Microstructure of Forging C (Three-Quarter Radius Location) Magnification: 100X Etchant: Flick's

toughness found at the center section of these large forgings appears to be partly caused by these factors of low matrix strength and inclusion distribution.

7. Fatigue Properties

The results of rotating beam and tension/compression fatigue tests are listed in Tables 13 through 15 and plotted in Figure 32. The S-log N curve for 7079 conventional size wrought product, as reported in the literature⁽⁶⁾, is included in Figure 32. Analysis of the data revealed the following:

a. The fatigue strengths in reversed bending closely approximated those reported in the literature⁽⁷⁾ for commercial-size wrought products. There was good data agreement in the 10^6 and 10^8 cycle range; the widest disparity occurred at 10^5 cycle.

b. The fatigue strengths in tension/compression were lower than those obtained in reversed bending. For example, at 10^8 cycles, the difference is approximately 40%; this is not unusual. One investigator(8) found that for round specimens of 2014-T4, bending stress gave approximately 41% higher results than axial stress.

c. The agreement of fatigue test data with literature data indicates that although the latter might not be directly applicable to final design of hardware, they serve as good first-approximation design values.

The results of the tension-tension fatigue tests are listed in Table 16. These results permitted construction of the S-Log N curve shown in Figure 33 and the Stress Range Diagram shown in Figure 34. The diagram is based upon the assumption that the ability of a material to withstand combined alternating (F_a) and steady (F_m) stresses can be defined by a straight-line function of the tensile ultimate (F_u) strength and the fatigue (F_e) strength; it is assumed that the following relationship is applicable.

$$F_a = F_e (1 - F_m/F_u)$$

The diagram in Figure 34 is for axial tests. The plotted points for 10^6 and 10^7 cycles are well below the Goodman line, while those for the 10^5 cycle are above the Goodman line (see Figure 32).

^{(6) &}lt;u>Reynolds Aluminum Data Book</u>, p. 37, 1961.

⁽⁷⁾ ibid.

⁽⁸⁾ Saver, J. A., and Lemon, D. C., "Effect of Steady Stress of Fatigue Behavior of Aluminum," <u>Trans ASM</u>, vol. 42, p. 559, 1950.

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CYCLES TO FAILURE FOR 7079-T652 FORGING "C" AT VARIOUS STRESS LEVELS UNDER CONDITIONS OF COMPLETE BENDING STRESS REVERSAL AT ROOM TEMPERATURE

Stress (ksi)	Cycles to Failure
65	1,300
60	2,450
50	8,900
45	16,300
40 40	23,900
37.5	49,100
35	46,100
30	314,100
30	484,000
30	1,871,600
30	131,700
27.5	1,129,000
25	5,742,000
22.5	100,000,000

CYCLES TO FAILURE FOR 7079-T652 FORGING "A" AT VARIOUS STRESS LEVELS UNDER CONDITIONS OF COMPLETE ALTERNATING TENSION/COMPRESSION STRESS REVERSAL AT ROOM TEMPERATURE

Stress	(ksi)	Cycles	to Failure
55			2,400
45		1	6,000
38		1	8,500
35		3	7,700
30		7	3,000
30		5	8,000
25		9	8,000
25		4	9,000
20		30	7,000
20		25	3,000
15		59	7,000
15		10 , 76	3,000

GRADE 7079-T652 FATIGUE STRENGTHS AND FATIGUE STRENGTH - ULTIMATE TENSILE AND YIELD STRENGTH RATIOS AT ROOM TEMPERATURE

A. Forging "C" in completely reversed stress

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Cycles	Fatigue Strength (ksi)	Fatigue Strength - Ultimate Tensile Strength Ratio	Fatigue Strength 0.2% Offset Yield Strength Ratio
104	48	0.726	0.927
10 ⁵	31	0.470	0.600
106	28	0.424	0.540
107	25	0.379	0.484
108	22	0.334	0.425

B. Forging "A" in tension/compression

Cycles	Fatigue Strength (ksi)	Fatigue Strength - Ultimate Tensile Strength Ratio	Fatigue Strength 0.2% Offset Yield Strength Ratio
104	38	0.535	0.696
10 ⁵	21	0.296	0.386
10 ⁶	15	0.211	0.275
107	15	0.211	0.275



Stress vs Log-Cycle Curve for 7079-T652 Forging

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TENSION-TENSION FATIGUE TEST RESULTS OF GRADE 7079-T652 FORGING AT ROOM TEMPERATURE

Steady Stress	Alternating Stress	
(KS1)	<u>(ksi)</u>	Cycles to Failure
30	15	34,000
30	12.5	67.000
30	10	119.000
30	10	183,000
30	7.5	8,086,000
30	6.25	787,000
30	5	10,725,000
20	15	71.300
20	12.5	95,700
20	10	2,173,000
20	10	256,000
20	7.5	799,000
20	7.5	6,958, 000
20	5	15,490,000
10	20	37,000
10	17.5	53,000
10	15	3,224,000
10	12.5	1,140,000
10	12.5	187,000
10	12.5	250,000
TO	10	2,893,000





Figure 34 Stress Range Diagram for 7079-T652 Forging

The data points for fatigue properties showed considerable scatter; these properties were apparently influenced by flow lines and grain size. The results of the fatigue tests emphasized the importance of determining individual forging fatigue strength. ÷

These tests serve to demonstrate that large 7079 forgings have low fatigue strength. This property factor, together with its low toughness, make 7079 a poor candidate for inducer application in liquid hydrogen.

8. Inducer and Impeller Blanks and Finish-Machined Parts

The impeller and inducer forging blanks are shown in Figures 35 and 37, respectively. The partially machined impeller and finished machined subscale inducer are shown in Figures 36 and 38. A comparison between the photographs of the forgings and the machined parts indicates the amount of metal removed during machining.

IV. CONCLUSIONS

The results were analyzed and the conclusions as regards the large 7079-T652 hand forgings are as follows:

A. The forgings were notch tough at ambient temperature. They were notch sensitive at -423° F, and slightly notch sensitive at -320° F. The notch sensitivity appeared to be partially dependent upon matrix strength and ductility levels, as well as grain size and orientation.

B. The forgings appeared satisfactory for service at temperatures down to -320° F for the M-1 impellers. However, Grade 7079 is not considered satisfactory for liquid hydrogen (-423°F) service in this size and application because of poor notch toughness; if used, conservative design values must be applied.

C. Ambient temperature fatigue properties closely approximated those of commercial size wrought products, but additional fatigue testing is required to obtain data at liquid hydrogen temperature.

V. RECOMMENDATIONS

The following recommendations are based upon the investigation described herein:

A. Continue evaluation of 7079-T652 to obtain fatigue data to -423°F.

B. Consider higher toughness titanium (Ti-5A1-2.5Sn) and nickel-base (Inconel 718) alloys for impeller service at -423°F. Several of the new aluminum alloys developed specifically for cryogenic service are of interest, including X7106, X7005, and X7039. The mechanical properties of these alloys should be assessed in large forgings.



Fuel Pump Inducer Blank

Figure 35





Figure 37

Oxidizer Impeller Blank



Figure 38

Partially Machined Oxidizer Impeller

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