(CODE) 22

NASA - CR - 72019 WCAF - 2972



IRRADIATION OF REFRACTORY FUEL COMPOUNDS, UO₂ AND UC, AT HIGH SPECIFIC POWER TO HIGH BURNUPS

POST IRRADIATION EXAMINATION OF CAPSULE 1

by RICHARD E. SCHREIBER

GPO PRICE \$ _____

CFSTI PRICE(S) \$ ____

Microfiche (MF) ________

prepared for

ff 653 July 65

NATIONAL AERONAUTICS AND SPACE ADMINISTRATION CONTRACT NAS 3-3392 PB

WESTINGHOUSE ELECTRIC CORPORATION ATOMIC POWER DIVISIONS

NOTICE

This report was prepared as an account of Government sponsored work. Neither the United States, nor the National Aeronautics and Space Administration (NASA), nor any person acting on behalf of NASA:

- A.) Makes any warranty or representation, expressed or implied, with respect to the accuracy, completeness, or usefulness of the information contained in this report, or that the use of any information, apparatus, method, or process disclosed in this report may not infringe privately owned rights; or
- B.) Assumes any liabilities with respect to the use of, or for damages resulting from the use of any information, apparatus, method or process disclosed in this report.

As used above, "person acting on behalf of NASA" includes any employee or contractor of NASA, or employee of such contractor, to the extent that such employee or contractor of NASA, or employee of such contractor prepares, disseminates, or provides access to, any information pursuant to his employment or contract with NASA, or his employment with such contractor.

Requests for copies of this report should be referred to

National Aeronautics and Space Administration Office of Scientific and Technical Information Attention: AFSS-A Washington, D.C. 20546

First Interim Report

IRRADIATION OF REFRACTORY FUEL COMPOUNDS, UO, AND UC, AT HIGH SPECIFIC POWER TO HIGH BURNUPS - POST-IRRADIATION EXAMINATION OF CAPSULE 1

Ъу

Richard E. Schreiber

Prepared for

National Aeronautics and Space Administration
August 1, 1966
Contract NAS-3-3392-PB
Technical Management
NASA Lewis Research Center
Cleveland, Ohio
Technical Liaison Office
Dominic C. Dilanni

Westinghouse Electric Corporation
Atomic Power Division
P. 0. Box 355
Pittsburgh, Pennsylvania 15230

NASA Distribution List

NASA-Lewis Research Center 21000 Brookpark Road Cleveland, Ohio 44135

R. M. Caves	MS 105-1
D. C. Dilanni (3)	49-2
P. L. Donoughe	49-2
L. V. Humble	49-2
S. J. Kaufman	49-2
A. F. Lietzke	49-2
Dr. B. Lubarsky	500-201
J. F. Mondt	500-309
F. E. Rom	49-2
E. W. Sams	49-2
N. D. Sanders	302-1
W. R. Thielke	54-2
Report Control	5 - 5
Library (2)	60-3

NASA-Lewis Research Center Plum Brook Station Sandusky, Ohio 44871

R. F. Barrows R. J. Koch Library (2)

AEC-NASA Space Nuc. Prop. Office U. S. Atomic Energy Commission Washington, D. C. 20545

F. C. Schwenk J. J. Lynch

NASA Scientific and Technical Information Facility (6) Box 5700 Bethesda, Maryland

U. S. AEC Technical Information Service (3) P. O. Box 65 Oak Ridge, Tennessee

U. S. Atomic Energy Commission (3) Technical Report Library Washington, D. C. 20545

Irradiation of Refractory Fuel Compounds, UO2 and UC, at High Specific Powers to High Burnups

Post-Irradiation Examination of Capsule 1

bу

Richard E. Schreiber

35219

ABSTRACT

The first of thirteen capsules, each containing four fuel pins, has been irradiated and examined. Three of the pins are UO₂ and the fourth is UC. Mean power levels of 33 kw/ft and burnups of 2.26 x 10²⁰ fission/cc were achieved in each of the 0.3000 inch diameter, 4.0 inch long fuel pellet columns. The AISI 348 stainless steel clad was 10, 20, or 30 mils thick on the UO₂ pins and 20 mils on the UC. All pins were gas bonded, surrounded by NaK, and operated at approximately 760°C fuel surface temperature. Fission gas release, density changes, melting behavior and dimensional changes agreed well with predictions. The 20 mil clad UO₂ pin failed, evidently by local NaK deprivation.

Table of Contents

	Page
Abstract	- iii-
Table of Contents	-iv-
List of Tables	-v-
List of Figures	-vi-
Summary	-1-
Introduction	-3-
Description of Irradiation Test Assembly	_14_
Presentation of Data	-1+-
Characterization of Materials	
Fuel Clad Tensile Wires Fuel Pin Assembly Procedure	-4- -5- -5- -5-
Design Calculation	-7-
Heat Transfer	-7-
Fission Gas Release and Clad Expansion	-11-
Exposure History	-17-
Post-Irradiation Examination	
Fission Gas Release Dimensional Change Fuel Sectioning Burnup Analysis Metallography Fuel Density	-17- -19- -20- -20- -21- -22-
Data Interpretation	- 23-
Conclusions	-27-
References	-28-
Tables I-XIV	29-42
Figures 1-39	43-81

List of Tables

<u>Table</u>	
I	Nominal Power and Burnup for UO, Fuel Pins in Each Capsule
II	Materials, Enrichment, and Clad Thickness - Each Fuel Pin
III	Analysis of UO ₂ Impurities
IA	Analysis of Rare Earth Content of UO2
V	Analysis of AISI $3^{14}8$ Stainless Steel Used for Cladding all Fuel Pins
VI	Designed Behavior of Capsule 1 Based on Maximum (Initial) Power = 49.6 kw/ft
VII	Designed Behavior of Capsule 1 Based on Mean (Initial) Power = 33.19 kw/ft
VIII	Fission Gas Samples Taken at Plumbrook Hot Cells
IX	Analysis of Gas Samples
X	Quantity of Individual Isotopes of Kr and Xe Present in the Fuel Pins
XI	Post-Irradiation Dimensions of Fuel Pins 1-1, 1-3, 1-4
XII	Fission Densities from U-isotope and Cs-137 Measurements on Pellets 1 and 6 from Pins 1-1, 1-3, $1-4$
XIII	Atom Percent Burnup from Mass Spectrographic Measurements on Pellets 1 and 6 from Pins 1-1, 1-3, 1-4
XIV	Density Measurements on Fuel Samples from Capsule 1

List of Figures

Figure	
1	Plumbrook Reactor Facility In-Pile Irradiation Experiment General Assembly
2	Plumbrook Reactor Facility In-Pile Irradiation Experiment Capsule Assembly
3	Pre-Irradiation Data Fuel Rod Can No. 1-1
14	Pre-Irradiation Data Fuel Rod Can No. 1-2
5	Pre-Irradiation Data Fuel Rod Can No. 1-3
6	Pre-Irradiation Data Fuel Rod Can No. 1-4
7	Pre-Irradiation Data Fuel Pin 1-1
8	Pre-Irradiation Data Fuel Pin 1-2
9	Pre-Irradiation Data Fuel Pin 1-3
10	Pre-Irradiation Data Fuel Pin 1-4
11	Pre-Irradiation Data Capsule 1
12	Capsule Insertion Record Cycle 20 P
13	Capsule Insertion Record Cycle 21 P
14	Capsule Insertion Record Cycle 22 P
15	Capsule Insertion Record Cycle 23 P
16	Capsule Insertion Record Cycle 24 P
17	Capsule Insertion Record Cycle 25 P
18	Capsule Insertion Record Cycle 26 P
19	Plumbrook Reactor Power During Irradiation of Capsule 215-001
20	Ruptured Fuel Pin 1-2
21	Macrophotograph of Fuel Pin 1-1
22	Macrophotograph of Fuel Pin 1-3
23	Macrophotograph of Fuel Pin 1-4
24	Pellet 1-1-2
25	Pellet 1-1-5
26	Top Guard Pellet, Pin 1-2
27	Pellet 1-2-1
28	Pellet 1-2-2
29	Pellet 1-2-3

List of Figures (cont'd)

Figure	
30	Pellet 1-2-4
31	Pellet 1-2-5
32	Pellet 1-2-6
33	Pellet 1-2-8
34	Lower Guard Pellet, Pin 1-2
35	Pellet 1-3-2
36	Pellet 1-3-5
37	Pellet 1-4-2
38	Pellet 1-4-5
39	Pellet 1-4-5, Near Center, 500X

Irradiation of Refractory Fuel Compounds, UO₂ and UC, at High Specific Power to High Burnups

Post-Irradiation Examination of Capsule 1

SUMMARY

A joint Westinghouse (APD)-NASA program has been undertaken for the irradiation of refractory fuel compounds, both UC and UO₂, at high specific power to high burnups in the NASA Plumbrook Reactor Facility (PBRF).

<u>WAPD</u> has designed and fabricated fuel bearing capsules, twelve containing three UO₂ fuel pins and one UC fuel pin and one containing one UO₂ pin and three UC pins. The power levels expected in each capsule have been calculated from flux perturbation measurements made in NASA's Mockup Reactor (MUR) at the Plumbrook site. Scheduling the irradiation sequences to give desired burnup and power levels is being done in cooperation with NASA and according to available reactor space based on program priority.

Design and safeguards and operating procedures manuals have been prepared. The majority of post-irradiation examination is being conducted at PBRF. In addition to disassembly, this includes photography, dimensioning, fission gas collection, sectioning of fuel pins, preliminary mounting of metallographic specimens, density measurement of fuel samples and burnup analysis by Cs-137 radiochemistry and U and Pu isotopic distribution by mass spectrograph. Final preparation and photography of metallographic specimens and analysis of fission gas are being done by WAPD in its own hot cells. In addition to the fuel pins, a number of tensile wires representing a variety of potential cladding materials are built into the capsules. These will be tested by WAPD.

Nominally, four power levels will be investigated between 25 and 50 kw/ft, each at three burnups, ranging from 10,000 to 80,000 MWD/MTU. In order to achieve power level control in the available fluxes, a set of neutron absorber sleeves has been prepared to surround the fuel capsules. The sleeves are made of either

aluminum, stainless steel, or boronated stainless steel. As depletion of U-235 occurs, and thus power drops in the capsules, the absorber sleeve is exchanged for one which depresses the flux less. As the power drops off again, the capsule position in the reactor is changed to provide a higher flux. The effect of this stepwise adjustment of power is to simulate operation of the fuel in a normally cycled core.

At present, WAPD and NASA personnel are evaluating the nuclear requirements and reactor space availability to achieve the program goal in the shortest time possible.

Each fuel pin is composed of a 4 inch long column of 0.300 inch diameter, half inch long, undished pellets. The UC is arc cast, hyperstoichiometric (excess C). The UO2 is of nominal composition, 96% of theoretical density. In each pin there is a suitable diametral gap to accommodate thermal expansion of fuel operating at a nominal power of 25 (Capsules 1-3), 30 (Capsules 4-6), 40 (Capsules 7-9, 13), or 50 (Capsules 10-12) kw/ft. The pins are clad with annealed AISI 348 stainless steel. In each capsule the clad on the UO2 pins is 10, 20, and 30 mils thick and the clad on the UC pin is 20 mils thick. Each pin is surrounded by either NaK (Capsules 1-6) or Na (Capsules 7-13) contained in stainless steel cans. Three tensile wires are fitted into the NaK or Na annulus of each can and four tensile wires are trapped in each of the four flow dividers in the capsule for a total of 28 wires. In addition to the fuel in each pin, there is a tubular spacer above the fuel to provide a plenum for fission gas, as well as to restrict the free aixal expansion of the short fuel column, thus making the pin equivalent to a much longer fuel rod. There is a deformable expansion marker below the fuel which records the maximum relative axial motion of clad and fuel.

INTRODUCTION

Uranium monocarbide and uranium dioxide are two currently available fuel materials of high uranium density. Both fuels have the potential of being utilized at high specific power levels to high burnups in a variety of reactor types. However, additional irradiation data on both carbides and oxides will be required to establish with confidence the performance limit of these fuels.

Uranium monocarbide has proved to be a very promising fuel for use in liquid metal cooled reactors with high specific power density cores. The metallic thermal conductivity and high uranium density of UC enables it to be used as the fuel in small, high power density reactors. UC, however, has a temperature and burnup limit due to fuel volume changes $\frac{2.3}{2}$ and reaction with fuel rod cladding materials at high temperatures. Carburization of the cladding is a serious problem in high temperature reactors particularly if a liquid metal is used as a heat transfer media between hyperstoichiometric UC and cladding.

UO₂ has been developed to an advanced stage primarily for use in central station power reactors. Although UO₂ fuel is inert to reactor coolants, refractory, dimensionally stable under irradiation and easy to fabricate, it has not been considered for use in reactors requiring high specific power cores because of its very low thermal conductivity in the solid state. This limit may be overcome if the fuel is allowed to operate partially molten. As power levels are allowed to exceed the limit for center melting, a question arises about the long term stability of the fuel rod. Extensive melting releases large amounts of fission gas, affects overall swelling of the fuel, and can redistribute the fuel axially in the central void that develops. It is also possible that the molten core of the fuel, if large enough, may come in contact with the clad, causing it to rupture.

In order to demonstrate the ability of UO₂ to be used as a high specific power fuel when irradiated to high burnups, and to establish operating limits for

both UO₂ and UC, an irradiation test program has been designed. The independent variables are: a) fuel rod power level, b) fuel burnup, and c) degree of restraint offered by the clad.

Table I shows the combinations of nominal power and burnup planned for each capsule. Table II gives the fuel, enrichment and clad thickness (AISI Type 348, annealed) for each fuel pin.

Description of Irradiation Test Assembly

The capsules are irradiated in tandem pairs in a holder. The holders are designed to fit the "L" holes in the primary beryllium reflector. This assembly is shown in Figure 1. The individual capsule is shown in Figure 2. The cans positioned in Capsule 1 - the subject of this report - are illustrated in data sheet Figures 3, 4, 5, 6. The tensile wire identifications and liquid metal loadings are listed, in addition to the dimensions. The fuel pins contained in their respective cans are shown in data sheet Figures 7, 8, 9, 10. The pellet densities listed in Figures 7, 8, 9, 10 were computed from the weights and dimensions given. Data sheet Figure 11 identifies the tensile wires located in the capsule flow dividers and the positions of the fuel cans in the capsule.

Presentation of Data

Characterization of Materials

Fuel

The ${\rm UO}_2$ was supplied by United Nuclear Corporation and was fabricated by conventional cold pressing and hydrogen sintering. The ${\rm UO}_2$ pellets had densities ranging around 96% of theoretical. The ratio of oxygen to uranium atoms ranged from 1.992 to 1.995. The analysis of impurities is given in Tables III and IV. All pellets were centerless ground to a diameter of 0.2995 \pm 0.0005 inches with a surface finish of 32

microinches RMS or better. The cylindrical pellets are 0.450 to 0.500 inch long; the ends are flat.

The UC was supplied by Atomics International and was made by arc melting and casting. The composition varies from stoichiometric up to 1.13 wt. % excess C. No impurity analysis is available. The pellets were centerless ground to the same tolerances as the UO₂. The length is approximately the same; the pellet ends are also flat.

Clad

The analysis of the weldrawn, annealed AISI 348 stainless steel is given in Table V. The end plugs are type 304 stainless steel which gives a tough, ductile weld to the type 348. The minimum mechanical properties were determined from axial tension tests on the tubing:

Yield Strength = 39,800 psi Ultimate Strength = 87,000 psi Elongation in 2 in. = 40 %

The 304 stainless steel end plug material is stronger and more ductile than 348 stainless steel, so the weld zones are stronger and more ductile than the clad. All clad tubing was 100% eddy current tested for defects.

Tensile Wires

None of the wires from the capsules were tested, but have been stored until the wires from several capsules are available for test. The pre-irradiation characterization of the wires will appear in a later report when the irradiated wires have been tensile tested.

Fuel Pin Assembly Procedure

To insure the integrity of the fuel pin cladding, a sequence of nondestructive tests were performed during the fabrication of the pins. In the following procedure, the parentheses, (), represent an assembly operation, and a square [], refers to an inspection step:

- (1) Bottom 304 SST end plug is TIG welded to 348 SST tubing.
- Bottom weld is radiographed.*
- Bottom weld is helium leak tested.**
- (4) Fuel pellets, previously identified in separate containers, are loaded into the tubes. UC fuel handling is all done in a helium atmosphere glove box.
- (5) Upper 304 SST end plug is TIG welded in a helium atmosphere. Both UO₂ and UC pins now contain helium at one atmosphere pressure.
- 6 Upper weld is radiographed.*
- [7] Entire rod is helium leak tested.**
- (8) Overall length and diameter measurements are taken prior to loading pin in outer can.

Radiography Technique

Two radiographs are taken of each weld 90° apart using SST correction forms. The sensitivity of the process is better than 2% revealing any defect or lack of weld penetration.

Helium Leak Test

All welds are first examined at 45X for gross defects such as holes or under cutting prior to leak testing. The fuel pins are then placed in a chamber which is evacuated and backfilled with helium to 45 psi. This pressure is maintained for 30 minutes. The pins are removed from the chamber, and after 15 minutes, placed in a vacuum retort connected to a helium mass spectrometer. The retort is evacuated through the spectrometer to detect helium loss from the pins. The sensitivity of this test is better than 1×10^{-6} cc/sec.

Design Calculations

Heat Transfer - Temperature Distributions

Capsule equivalent diameter, De = 0.198 in.

Coolant velocity, V = 27.2 fps

Coolant flow, F = 60 gpm

Coolant inlet Temperature, $T_{in} = 52^{\circ}C$

Coolant temperature rise, $\Delta T_{rise} = \frac{3.78P}{F}$, °C

where $P = P_L \times \frac{4'' \text{ fuel length}}{12''/'} \times \frac{4 \text{ pins}}{\text{cap.}}$, kw.

 $P_{T} = kw/ft$, each pin.

Coolant outlet temperature, Tout Tin + ATrise, °C

Mean coolant temperature, T_{mean} = T_{in} + T_{out}, oc

Coolant film temperature, $T_f \simeq \frac{T_{mean} + T_{surf}}{2}$, oc

Surface temperature of can, $T_{surf} = T_{mean} + \frac{(Q/A)_{can}}{h}$, oc

Can surface heat flux, $\frac{Q}{A}$ = $P_L \times \frac{4''}{12''/'} \times \frac{3415}{\pi D/12}, \frac{BTU}{hr-ft^2}$

Can outside diameter, D = 0.625 in.

Heat transfer coefficient, h = $N(T_f) \frac{v^{8}}{D_0^{2}} \times \frac{10^{4}}{5.67} \times 1.8$, $\frac{BTU}{hr-ft^{\circ}C}$

V and De are in the units given above and $N(T_f)$ is a water properties function of film temperature only.

Maximum fuel pin power, P_T = 49.6 kw/ft

Mean fuel pin power, P_L = 33.19 kw/ft

Then:
$$\Delta T_{rise}$$
 = 2.79 C°, (based on mean P_L)
= 0.3451 x 10⁶ Btu/hr-ft²
(Based on max P_L)

Assume $T_f = 62.8$ °C, then $N(T_f) = 0.215$ and h = 17,930 Btu/hr-ft²-C°

$$T_{mean} = 52 + \frac{2.79}{2} = 53.4^{\circ}C$$

$$T_{surf} = 53.4 + \frac{.3451 \times 10^{6}}{.01793 \times 10^{6}} = 72.7^{\circ}C$$

Checking $T_f = \frac{53.4 + 72.7}{2} = 63.0 \, ^{\circ}\text{C} \, ^{\circ}$ 62.8, 0.K.

The temperature drop across the successive annulii - can, NaK, clad - are approximately expressed by:

$$\Delta T_{ann.}$$
 = $\frac{P_L}{2\pi k}$ $\ln \left(\frac{OD}{ID}\right)$, °C
 P_L = max linear power, 49.6 kw/ft
k = thermal conductivity

Region	k, kw/ft-°C	OD, in.	ID, in.	Δ^{T} ann., °C
can (304 SST)	0.00501	0.625	0.525	267.9
			.324 pin 1	481.5
NaK	0.00791	0.525	.344 pin 2	422.1
(78% K)			.364 pin 3	365.3
			.342 pin 4	422.1
		.324	.304 pin 1	73.4
clad	0.00685	.344	.304 pin 2	142.4
(348 SST)		.364	.304 pin 3	207.4
		.342	.302 pin 4	142.4

The annular temperature differences for a mean $P_L = 33.19 \text{ kw/ft}$ are pro-

portionally less. The can surface temperature for the mean $P_{\rm L}$ is about 66°C.

The fuel surface temperature depends on the heat flux and gas gap conductance. The latter can vary - depending on interfacial pressure and surface roughness of clad and fuel - over the range 500 to 4000 Btu/hr-ft-°F. We initially select a value of 2000 for estimating fuel temperatures. The heat flux at the gap is about the same for all pins:

$$\left(\frac{Q}{A}\right)_{gap} = \left(\frac{Q}{A}\right)_{can} \times \frac{.625}{.300}$$

$$= 0.719 \times 10^6 \text{ Btu/hr-ft}^2 \text{ for } P_L \text{ max.}$$

$$= 0.481 \times 10^6 \text{ Btu/hr-ft}^2 \text{ for } P_L \text{ mean}$$

The temperature drop across the gas gap in each pin,

$$\Delta T \text{ gap} = \frac{0.719 \times 10^6}{2000 \times 1.8} = 200 \text{ C}^{\circ} \text{ for } P_L \text{ max.}$$

$$\Delta T$$
 gap = 134 C° for P_L mean

The central temperature of the fuel pins may be characterized by the sum of two integrals:

$$\int_{0}^{T_{center}} k d\theta = \int_{0}^{T_{surface}} k d\theta + \int_{T_{s}}^{T_{c}} k d\theta$$

where the fuel thermal conductivity, $k = k(\theta)$

The method of Robertson/6 may be used to evaluate the terms in the expression and to determine the fuel center temperature (if all solid) or the radius of melting. At a power level of 49.6 kw/ft, the fuel surface temperature is about 1100° C. The value of the first integral, $\int_{0}^{T_{s}} k \ d \ \theta \simeq 50 \ \text{w/cm.*}$

The value of the second integral depends on the flux depression factor, f, as well as the power level. At 8.5% enrichment and 0.3000 inch diameter, f = 0.8825.*

^{*}WAPD UO, thermal conductivity and flux depression data are used rather than Robertson's.

$$\int_{T_{s}}^{T_{c}} k d \theta = 0.8825 x \frac{49.6 \text{ kw/ft}}{4\pi} x \frac{10^{3} \text{ w}}{\text{kw}} x \frac{1 \text{ ft}}{30.5 \text{ cm}}$$

= 114.2 w/cm

If the sum of these integral, $\int_{0}^{T_{c}} k d\theta > 97 \text{ w/cm}$, there is center

melting. This is clearly the case for the UO, fuel pins.

For f = 0.8825, a dimensionless neutron diffusion parameter, Ka = 1.5.

The fuel radius, a = 0.381 cm. The integral to the radius of the molten

zone,
$$r_m$$
, $\int_{T_s}^{2800^{\circ}C} k d\theta = \frac{49.6 \times 10^3 f (r_m)}{4 \pi \times 30.5} = 97-50 = 47 \frac{W}{cm}$

where 97 is the value from 0 to 2800°C 50 is the value from 0 to T_s

and $f(r_m)$ is the flux depression into r_m

Solving, $f(r_m) = 0.363$, for which the dimensionless radius of melting,

$$\frac{r_{m}}{a} = 0.8 , \text{ for } Ka = 1.5$$

The radius of melting, $r_{\rm m}$ = .303 cm is nearly the same for all three UO₂ pins.

The center temperature of the UC is simpler to calculate since k_{UC} does not vary strongly with temperature and there is no melting. In the temperature range above 300°C, $k_{UC} \simeq 0.2249 \frac{w}{cm - oc}$.

The melting point of UC is about 2450°C.

$$\int_{T_{S}}^{T_{C}} k_{UC} d\theta \approx k_{UC} \Delta T = .2249 (T_{C}-1105) = 114.2 \frac{W}{cm}$$

Solving, $T_c = 1613^{\circ}C$ (<2450°C) for $P_L = 49.6$ kw/ft. The same procedure is followed for $P_L = 33.19$ kw/ft. The temperature distributions are given in Tables VI and VII. The temperature profile is used to estimate the fission gas release and dimensional changes of the pins.

Fission Gas Release and Clad Expansion

The U-235 burnup is an effective full power time of 3.24×10^6 seconds is 0.2264×10^{21} fissions/cc fuel. The fission gas yield (Kr and Xe), including decay, is 0.295. At STP (0°C and 1 atm), one cc of (monatomic) fission gas is composed of 2.69×10^{19} atoms. The volume of gas generated in each of the pins per unit volume of fuel,

$$V = \frac{0.2264 \times 10^{21} \text{ fissions/cc fuel } \times 0.295 \text{ atoms of Kr and Xe per fission}}{2.69 \times 10^{19} \text{ atoms/cc gas at STP}}$$

$$V = \frac{2.48 \text{ cc gas at STP}}{\text{cc fuel}}$$

The fuel volume in each pin is 4.268 cc. Therefore, the volume of gas generated in each pin is $2.48 \times 4.268 = 10.58$ cc.

The fraction of gas release is estimated by the method of Booth $\frac{7}{7}$. The activation energy for fission gas diffusion in UO_2 is taken as 82 Kcal/mole and the diffusion constant at 1400° C,

$$D'(1400^{\circ}C) = 2 \times 10^{13} \text{ sec}^{-1}$$

This gives

$$D'(2800^{\circ}C) = 1.5645 \times 10^{-8} \text{ sec}^{-1}$$

and in general,

eral,
$$D'(T) = 2 \times 10^{13} \exp \left[-4.13 \left(\frac{10^4}{T + 273} - 5.98\right)\right]$$
where $T = {}^{\circ}C$

The parameter, $\frac{1}{\pi^2 D^* (2800)} = 6.476 \times 10^6 \text{ sec}$

which is greater than the effective full power time, 3.24×10^6 sec. In this case, the fraction of gas released may be approximated by,

$$F = 4\sqrt{D' t/\pi - 3D' t/2}$$

for 2800° C, F = .432

D' for $T \le 1100^{\circ}C$ is 9.2 x 10^{-16} sec⁻¹.

For temperature this low, f ? o.

The mean temperature in the solid fuel annulus, $T_m = \frac{T_s + 2800}{2}$

At $P_{T} = 49.6 \text{ kw/ft}$, $T_{g} \approx 1100^{\circ}\text{C}$

to give $T_m = 1948$ °C. Then

$$D^{-}(1948^{\circ}C) = 9 \times 10^{-11} \text{ sec}^{-1}$$

and $F_{(1948°C)} = .0381$

For the whole solid portion, ignoring axial temperature gradient,

$$F_{\text{solid}} = 1/6 [F_{2800} + 4F_{1948}] = 0.0974$$

The volume of solid UO₂, weighted for flux depression, is $f_{(rm)} = .363$. The fraction of gas released from the molten fuel is 1.0. The total for the whole pin,

$$F = 1 - .363 + .0974 \times .363 = .672$$

The volume of gas released,

$$V_{gas} = .672 \times 10.58 = 7.11 \text{ ce, STP.}$$

A method analogous to this for UC is not available, but it is believed that a negligible quantity of gas would be released.

At mean $P_{T} = 33.19 \text{ kw/ft}$, the fuel is partially molten, though not to the extent calculated for $P_L = 49.6 \text{ kw/ft}$. At the mean power, $f_{(rm)} =$.6525 and $T_s \approx 757^{\circ}C$. The surface is less than $1100^{\circ}F$, so diffusion processes are negligible in that part of the solid which is less than 1100°C, but recoil processes, which are independent of temperature, release fission gasses from the solid as though it were at a uniform temperature of 1100°C. Thus, Booth's method will underestimate the fission gas release from fuel with less than 1100°C surface temperature.

This is partly compensated, however, by using the initial mean power rather than the mean power over the exposure, 31.24 kw/ft.

$$T_{m} = \frac{2800 + 757}{2} = 1779^{\circ}C$$

$$D'_{(1779^{\circ}C)} = 1.96 \times 10^{-11} \text{ sec}^{-1}$$

$$D'_{t} = 6.35 \times 10^{-5}$$

$$F_{(1779)} = 0.0179$$

$$F_{\text{Total,solid}} = 0.0839$$

$$F_{\text{pin}} = 0.402$$

The 40.2% fission gas release corresponds to 4.25 cc (STP).

Retained fission products contribute to fuel swelling. If the temperatures are less than 1100°C, the contribution of bubble expansion to fuel swelling is negligible. The atomic volume effect, however, is operative and amounts to about 1.3 volume per cent for the 2.264 x 10²⁰ fission/cc experienced in the capsule 1 fuel. This is the maximum swelling possible in the solid part of the fuel. The correction for fission gas lost from the solid part is negligible.

The volume change of UO $_2$ upon melting is 10.6%; the thermal expansion of liquid UO $_2$ is uncertain and will be ignored. The fractional volume of melted UO $_2$ at P_L = 49.6 kw/ft is 64%; the molten part @ P_L = 33.19 kw/ft is 30.25%. The initial void (porosity) is 5% or less; this serves to absorb part of the expansion.

The thermal expansion of the UO_2 fuel relative to the clad is more than offset by the 0.040 inch initial axial gap and 0.004" diametral gap.

$$\frac{\Delta D}{D} = 2.9 \times 10^{-9} (T)^{2} + 6.8 \times 10^{-6} T + 1.72 \times 10^{-4}$$

$$T = ^{\circ}C$$

$$\frac{\Delta D}{D} = 19.8 \times 10^{-6} \text{ (T-25)}$$

$$T = ^{\circ}C$$

At
$$P_L$$
 = 49.6 kw/ft, T_{UO_2} , solid = 1948°C, T_{clad} $\stackrel{\sim}{\sim}$ 834°C

$$\Delta D_{UO_2}$$
, solid = 0.300 x 24.42 x 10⁻³ = 0.00733 inch

$$\Delta L_{UO_2}$$
, solid = 3.7 x 24.42 x 10⁻³ = 0.0904 inch

$$\Delta D_{\text{clad ID}} = 0.304 \times 16.02 \times 10^{-3} = 0.00487 \text{ inch}$$

$$\Delta L_{UO_2}$$
, solid - ΔL clad - axial gap = .0904 - .0593 - .040 = -.009 inch

No interference.

At
$$P_L = 33.19 \text{ kw/ft}$$
, $T_{UO_2 \text{ solid}} = 1779^{\circ}\text{C}$, $T_{clad} = 576^{\circ}\text{C}$.

To translate the volumetric terms into diametral ones, the expansion is assumed isotropic:

$$@ P_{T} = 49.6 \text{ kw/ft}$$
:

$$\Delta D_{\text{molten UO}_2 \text{ part}} = .64 \times \frac{.106}{3} \times .300 = .0068 \text{ inch}$$
 of pellet

$$\Delta D_{\text{fission products}} = (1-.64) \times \frac{.013}{3} \times .300 = .0005 \text{ inch}$$
 (in solid UO_2)

$$\Delta D_{UO_2}$$
 (minus porosity) = .0096 inch

$$\Delta D_{clad}$$
 (including initial gap) = $\frac{.0089}{.0007}$ inch @ $P_{L} = 49.6$ kw/ft

$$@ P_{t} = 33.19 \text{ kw/ft}$$
:

$$\Delta D)_{\text{molten UO}_2} = .3025 \text{ x} \cdot \frac{.106}{3} \text{ x} \cdot .300 = .00325 \text{ inch}$$

$$^{\Delta D)}$$
f.p.'s = (1-.3025) x $\frac{.013}{3}$ x .300 = .00091 inch

$$\Sigma\Delta D_{UO_2}$$
 (less = .0055 inch porosity)

$$\Sigma \Delta D$$
 clad (+ cold = $\frac{.0073}{\text{gap}}$ inch

diametral hot gap= .0018 inch

$$\Delta L_{UO_2}$$
 solid - ΔL clad - axial gap = 3.7 (21.45 - 10.91) x 10^{-3} - .040 = -.001 No interference.

No net change in clad diameter or length is expected since the initial void is larger than the permanent ${\rm UO}_{2}$ expansion.

The total expansion of UC is described by
$$\frac{\Delta L}{Lo}$$
 = (3.487 + 1.143 ln T) (T-20) x 10⁻⁶ T = °C

$$T_{mean}$$
 (@ P_{L} = 49.6 kw/ft) = 1359°C

$$T_{mean}$$
 (@ P_{L} = 33.19 kw/ft) = 927°C

$$\Delta D_{f.p.'s} = \frac{.013}{3} \times .3 = .0013$$

Initial porosity = 0

$$\Delta D$$
 Therm. = 0.0157 x .3 = $.0047$

$$\Sigma \Delta D_{UC} = .0060 \ \text{@ P}_{L} = 49.6 \text{ kw/ft}$$

Cold gap
$$= 0.0020$$

AD therm., clad= 0.0049

$$\Sigma\Delta$$
 clad = $.0069$ @ P_L = 49.6 kw/ft

diametral hot gap = .0009 inch

$$\Delta L_{UC} = \left(\frac{.013}{3}\right) \times 3.7 + .0157 \times 3.7 = 0.0741 \text{ inch}$$
 axial hot gap $\Delta L_{clad} + \underset{(cold)}{\text{axial gap}} = .0593 + .042 = 0.1013$ = 0.027 inch

$$\Delta D$$
 therm. UC = 0.01035 x .3 = 0.0031 inch ΔD f.p.'s = $\frac{0.0013}{\Delta \Sigma D_{\rm HC}}$ = 0.0044 inch

 $\theta P_{\tau} = 33.19 \text{ kw/ft}$

$$\Delta D$$
 therm, clad = .0033
cold gap = .0020
 ΔL clad = 0.0053 inch
diametral hot gap = 0.0009

$$\Delta L_{UC} = 0.01035 \times 3.7 + \frac{.013}{3} \times 3.7 = .0543 \text{ inch}$$

$$\Delta L_{clad} + \text{axial gap} = 10.91 \times 10^{-3} \times 3.7 + .042 = 0.0824$$
axial hot gap = 0.0281

If these hot gaps existed, the UC would probably increase in temperature until it touched the clad since only under the condition of clad-fuel contact can the high conductance postulated be achieved. The clad temperature would not change since this depends only on the heat flux through it.

The UO₂ would tend to expand for the same reason, but, having touched the clad, would exert little force because of the large central void and molten region which cannot support radial loads. The UC, however, is not molten, develops no central void, and thus is able to support radial loads. The higher temperature in the UC that is necessary for adequate thermal expansion to touch the clad raises it significantly above the 1100°C threshold for the formation of fission gas bubbles in the fuel. These exert pressure on the UC, causing it to creep and thus swell with time. The force exerted on the clad by the fuel may be sufficient to cause the clad to creep and thus have a permanent diametral increase when examined after irradiation. Under these circumstances, it is impossible to predict either

the operating temperature in the UC fuel or amount of clad swelling associated with it.

The expected fission gas release and estimated clad dimension changes computed in this section are presented in Tables VI and VII.

Exposure History

The exposure history for capsule 1 is given for each cycle in data sheet figures 12 through 18. The reactor power variation for the entire exposure is given in figure 19. The calculated mean power variation in a typical fuel pin is also shown in figure 19.

Figures 12 through 18 are included in this report, but may be omitted in later reports. This group of figures illustrates the method of management employed to control an increasingly complex series of irradiation experiments. Considering the number of capsules to be irradiated, the variety of reactor positions available, and the variation in equipment used, this represents a convenient way to keep track of the tests.

Post-Irradiation Examination

Fission Gas Release

After removal of the four cans from Capsule 1 and cursory examination, the first destructive test was to puncture each can and extract the cover gas over the NaK to see if the pin had failed, releasing fission gas into the can. The activity of the gas samples was checked and, if negligible, was released.

The gas sample taken from can 1-2 had substantial activity. When the pin was removed from the can it was found to be ruptured. The other fuel pins were intact and were punctured to extract their fission gas after dimensions and photographs were taken.

Scale in Figure 19 is based on initial U-235 content. Actual mean power (at 60 MW) drops about 2 kw/ft during exposure.

The amount of gas in the pin or can was determined from the pressures recorded in the sampling system before and after puncture, and the known volume of the various components. The basic expression is:

$$P_1V_1 = P_3 (V_1 + V_2) - P_2V_2$$

where

 P_{γ} = gas pressure in pin or can prior to rupture, μ

 V_1 = volume of gas in pin or can, cc

 P_{o} = pressure in system, including sample bulb, prior to puncture, μ

 V_{o} = volume of system and sample bulb = 2015 cc

 P_{q} = pressure in system after puncture, μ

 $V_3 = V_1 + V_2$, cc

The amount of gas shipped to Westinghouse Atomic Power Division is given by:

$$P_B V_B = P_1 V_1 - P_{i_1} V_{i_1}$$

where

 $P_{\rm R}$ = pressure in bulb after sweeping gas from system into bulb, μ

 V_R = sample bulb volume = 100 cc

 P_h = pressure in system after sweep, μ

 $V_4 = V_3 - V_B$, ce

If the pressure in the can is desired, as in 1-2, the free volume in can and pin, $V_1 = 6.189$ eq. The free volume in each of the unruptured pins, V_1 , varies slightly. The recorded and calculated quantities mentioned above for capsule 1 are given in Table VIII. All pressures are at room temperature.

After arrival of the gas samples at WAPD, the quantity of gas that had been present in each pin was again determined, this time by analysis. Mass spectrographic analysis revealed some air contamination of the gas samples. In addition to air and fission gas, there is considerable helium, most of which is the original cover gas in the pin (or pin and can), and some decomposition and reaction products of containinants originally present in the fuel, such as

water. The glass bulb containing the gas from can 1-2 was damaged in separating it from the sampling apparatus; the gas was lost. The analyses of the three remaining gas samples are reported in Tables IX and X.

The amount of gas and the pressure calculated from the analysis is in good agreement with the P_1V_1 calculated from the Plumbrook sampling data. The respective quantities are given in Tables IX and VIII.

Dimensional Changes and Appearance

In general appearance, pins 1-1, 1-3 and 1-4 were unchanged after irradiation. Pin 1-2 failed by melting of the clad along one side of the lower third of the enriched fuel column. Some of the fuel was extruded to form a crust. Chemical and physical analysis of the crust showed it to be a porous mass of UO₂, NaK residue, and beads of cladding. There was no material which could defintely be characterized as reaction product between clad and fuel. No gross swelling was associated with the rupture. No dimensions were taken, but the axial expansion marker was examined and found to be undeformed. Figure 20 presents several views of the rupture, with and without the crust.

The dimensions of pins 1-1, 1-3, 1-4 are given in Table XI. The maximum diametral increase in pin 1-1 was 1.424%. This was accompanied by an axial shrinkage of 0.06% and small diametral shrinkage at the top and base of the pin. The maximum diametral increase in pin 1-3 was 0.659%. This was accompanied by an axial shrinkage of 0.06% and small diametral changes at the top and base of the pin. The diametral measurements on pin 1-4 are apparently in error; they suggest a 2.2% shrinkage. Checks on metallographic specimens from pin 1-4 show clad diametral increases as large as 4.4%. The increase in diameter was also confirmed by fuel density measurements, reported fully in a later section. Pin 1-4 apparently increased in length 0.07%.

The volume change of the pins may be estimated from the expression.

$$\frac{\Delta V}{Vo} \quad \tilde{z} \quad 2 \, \frac{\Delta D}{Do} \, + \, \frac{\Delta L}{Lo}$$
 where $\frac{\Delta D}{Do}$ is the mean diametral change

Fuel Pin	$\frac{\Delta D}{Do}$	$\frac{\Delta L}{Lo}$	$\frac{\Delta V}{Vo}$	Displacement Volume Change
1-1	+0.00031	-0.000596	√ 0	+18%
1-3	+0.00165	-0.00062	0.00268	0
1-4	(+0.022)*	+0.00066	0.04466	+27%

Half of maximum estimated from metallographic specimen.

It is evident that the displacment measurement is worthless; it has been dropped from the procedure.

Fuel Sectioning and Preliminary Examination

After fission gas samples and dimension were taken, the pins were sectioned at the enriched pellet interfaces. The top and bottom of each pellet were photographed while the section was held by manipulators. One pellet was crushed and some material lost from the others. A new method of supporting the sections for photographing is being developed. The photos of sectioned pins 1-1, 1-3, 1-4 are presented in Figures 21, 22, 23. The central void and columnar grain growth regions are clearly visible, but the photos are too crude for quantitative evaluation.

The sections of each pin were allocated as follows for further examination:

Pellet Number	<u>Use</u>
1,6	Burnup Analysis
2,5	Metallography
4,8	Density
3,7	Spare

Burnup Analysis

A radiochemical analysis for Cs-137 was performed on pellets 1 and 6 of each pin at Plumbrook, in addition to a mass spectrographic analysis for uranium isotopic distribution. The dissolved samples were shipped to WAPD where confirmatory Cs-137 measurements were made. Consensus values of fission density are presented in Table XII.

Calculations of the atom percent burnup, based on mass spectrographic results, are given in Table XIII.

Metallography

Pellets 2 and 5 from pins 1-1, 1-3, 1-4 were mounted at Plumbrook, and ground and polished at WAPD. Pin 1-2 was sent intact to WAPD for mounting, sectioning, grinding and polishing. None of the specimens were etched. Macrophotographs were prepared to show extent of melting, grain growth, swelling, and central void development. These are presented in figures 24 through 38.

The size of the central void in pellet 1-1-2 is difficult to estimate because of fuel loss during handling. Pellet 1-1-5 is also badly fractured but the cavity is about 25% of the fuel diameter. Pellets 1-3-2 and 1-3-5 are more intact, so some estimate may be made of fuel redistribution during operation. On volume basis, the void in 1-3-2 is 9.6% and the void in 1-3-5 is 0.61%. The average of these quantities is close to the initial porosity of 5%. If the closed initial porosity, as well as the fission gas, migrates to a central void which extends over the length of the fuel, the UO₂ density should slightly increase. In fact, it did: from 95% to 97%.

The difference in void size between pellets 1-3-2 and 1-3-5 is due to fuel slumping. Pellet 2 operated in a 12% lower flux (unperturbed) than pellet 5, but the fuel slumping reduced the perturbation sufficiently to cause the upper part of the fuel to operate at a higher rate of fission density. The fission density measurements confirm this. The actual linear power distribution is proportional to the product of fission density rate and fuel cross sectional area at each level. The linear power is responsible for the temperature distribution, which in turn affects the amount of fuel slumping to the base of the pin. The complexity of the situation makes it difficult to predict power and temperature distributions, or reactivity effects, in fuel columns which are largely molten. This is the reason molten UO₂ operation is usually avoided in commercial reactor operation. The possibility of clad rupture due to contact with molten UO₂ aggravates this problem.

The central void everywhere above the rupture zone in pin 1-2 is nearly uniform: 46% of the fuel diameter. Since some fuel was lost through the opening at the base of the enriched fuel column, it may be assumed that most of the molten portion above the rupture has left that region. Thus, the void diameter above the rupture in pin 1-2 is nearly as large as the region which operated in the molten state. The predicted molten zone, based on the mean linear power, was 55% of the fuel diameter. This number, however, did not consider loss of molten fuel from this region. The maximum void that can exist due to the 5% initial porosity is only 22.4% of the fuel diameter. The actual cavity of twice this amount clearly represents molten fuel loss.

Microphotographic surveys at 500X, after etching, were made of pellets 1-4-2 and 1-4-5 to see if a significant amount of bubbles had formed. A typical central region of 1-4-5 is shown in Figure 39. Point count analysis of several grain boundaries gave the estimate that the grain boundary bubbles alone accounted for as much as 2.8% of the volume in pellets 1-4-2 and 1-4-5. The mean value for all areas surveyed was 2%. Some bubbles are also evident in the matrix of the UC, but these were not counted.

Fuel Density

Pellet densities were determined by the ${\rm CCl}_{\downarrow}$ displacement method at Plumbrook. The weights, dry and immersed, of the fuel plus clad and the clad alone were measured. The temperature of the ${\rm CCl}_{\downarrow}$ was recorded and its density calculated according to the expression:

$$d_{CCl_{\frac{1}{4}}} = 1.63255 - 1.9110 \times 10^{-3} t$$
 $= 0.690 \times 10^{-6} t^{2} + 0.0002, g/cc, 0 < t < 40^{\circ}C$

The fuel density was calculated from the expression:

where

Two measurements were made of each quantity, so the probable error was taken as half the difference between them. The fuel density error was determined in the standard manner:

The formula for fuel density is of the form: $d_{fuel} = \frac{A \times B}{C}$

This leads to the density error expression:

$$e_{d \text{ fuel}} = \sqrt{\left(\frac{B}{C}\right)^2} e_{A}^2 + \left(\frac{A}{C}\right)^2 e_{B}^2 + \left(\frac{AB}{C^2}\right)^2 e_{C}^2$$

where e_A , e_B , e_C are the errors in A, B, C.

Table XIV gives the measured weights and calculated fuel densitities. A slight densification of the UO₂ is evident and the UC is significantly less dense than non-irradiated material. There is apparently some error in the density of pellet 1-3-8.

The UC densities, pre- and post-irradiation may be used to calculate the swelling:

$$\frac{\Delta V}{Vo} = \frac{\frac{1}{d_f} - \frac{1}{d_o}}{\frac{1}{d}} = \frac{d_o}{d_f} - 1$$

where subscripts o and f refer to initial and final densities.

Pellet	1-4-4	1-4-8
d _o , g/cc d _f , g/cc	13.3170 12.8086	13.2982 12.8225
$\frac{\Delta V}{Vo}$	0.03969	0.03710

Data Interpretation

Despite the wide variations in air contamination of the fission gas samples, it

was possible to determine the amount of fission gas released and the pressure it exerted in the pin prior to puncture to a very acceptable degree of accuracy. This is attested to by the excellent agreement of P_1V_1 (and P_1) values for UO_2 pins 1-1 and 1-3 which operated under nearly identical conditions. Also, fair agreement between the P_1V_1 product calculated from the PBRF data (Table VIII) and the P_1V_1 product calculated from the gas analysis (Table IX) implies fair gas sampling technique in the PBRF hot cells.

The fraction of fission gas released from the $\rm UO_2$ in pins 1-1 and 1-3 (Table X) is in good agreement with the predicted release based on mean pin power (Table VII), given the limitations of the diffusion model below $1100^{\rm o}$ C and the uncertainty in the gap conductance. The gas release from the UC pin 1-4 was low, as expected. The internal pressure, $\rm P_1$, in the pins was in all cases too low to cause significant stress in the cladding.

The clad dimensional measurements were not entirely satisfactory. The apparent diametral shrinkage of the annealed stainless steel cald has no reasonable explanation. Certainly the diameter of UC pin, 1-4, did not shrink since the UC swelled noticeably during irradiation, as shown by density measurements and by examination of the mounted metallographic specimens. Greater care with the optical gage is recommended, such as interspersing the pin diametral measurements with checks on standard bars. It may be that the plunger is not hitting the top of the pin. The overall length changes were so small that no judgement regarding their accuracy can be made.

The nature of the failure of UO₂ pin 1-2 cannot be conclusively determined. Since the failure was unaccompanied by general swelling or chemical reaction, and was exposed to the same conditions as the other UO₂ pins, which showed no signs of incipient failure, it must be concluded that the failure was a freak. Two mechanisms of failure may be conjectured:

1. Molten ${\rm UO}_2$ touching the clad via a random crack in the solid ${\rm UO}_2$ annulus, may have so weakened the clad that the gas pres-

sure forced a small leak. At the next reactor shutdown, the gas pressure - now reversed - forced NaK into the pin. At startup, the boiling, churning mass of NaK and molten UO₂ was forced out of the pin by the NaK vapor pressure. As the mixture left the pin and flowed down the outside of the clad, the UO₂ froze, forming the crust observed. This crust deprived the clad locally of heat transfer medium, causing it to melt, further exposing the fuel. Successive power cycling caused more fuel to leave the pin until the structure we observed was established.

2. Movement in and out of the reactor of the holder containing the capsule, including the time it was sent to the hot cells for gamma scanning, provided an opportunity for a bubble of helium cover gas to be trapped underneath the pin. This is possible since the annulus is partly constricted by a support disc over the end plug, called a spider, which holds the tensile wires in place. If such a bubble later escapes - say, due to hydraulic vibration - while the pin is operating at substantial power, the clad is momentarily deprived of NaK heat transfer medium. The clad overheats, burnsout, and molten UO₂ fuel is released which freezes to form a crust.

The second of these conjectured mechanisms depends on the heat transfer medium being liquid at room temperature. Thus, if this is the failure mode, it should not occur in the Na capsules (7 through 13).

If the first mechanism is operative, i.e., failures occur in Na as well as NaK capsules, then failures is random, and suitable statistics cannot be established unless a great many high power tests are run.

The results of burnup analyses (Table XII) for UO₂ fission density are very close to the predicted value for the given exposure time (Table VII).

The axial variation of calculated UO₂ fission density based on Cs-137 is not meaningful due to the considerable axial redistribution of Cs-137. The UC, which saw the same axial (unperturbed) flux gradient - higher flux in pellet

6 than in pellet 1 - reflects this in the calculated fission density based on Cs-137. The axial fission density distribution in the UO₂, however, should not be identical to that in the UC, largely due to the molten UO₂ slumping and its effect on flux perturbation.

The metallographic examination confirmed the $\rm UO_2$ fuel slumping, the UC swelling, the $\rm UO_2$ densification, and the radius of $\rm UO_2$ melting predicted on the basis of mean linear power and the assumed gap conductance. The central void size in the unruptured $\rm UO_2$ pins was entirely accounted for in terms of the initial closed porosity. The axial variation in void size explained the difference between the fission density gradient and the unperturbed flux gradient. Measurement of the UC clad diameter in the metallographic mounts at WAPD and bubble counts at high magnification confirmed PBRF's density measurement and rejected their diametral measurement of the UC. The release of the $\rm UO_2$ closed porosity to form a central void explained the measured increase in $\rm UO_2$ density. The void size above the rupture in pin 1-2 was very nearly equal to that predicted for the radius of melting at mean linear power. Apparently the rupture allowed most of the molten $\rm UO_2$ to flow out of the pin.

The fuel density measurements considered the whole pellet and thus represent a mean radial fuel density. The ${\rm CCl}_{\downarrow}$ does not penetrate the closed porosity; this made it possible to estimate volumetric changes in the UC fuel due to fission gas bubbles. It also permitted an estimate of the UO $_2$ fuel densification due to loss of initial porosity. The 4% volume change of the UC was confirmed by metallographic examination.

CONCLUSIONS

In general, it was possible - despite some experimental difficulties - to confirm the predicted fuel pin behavior based on the mean linear power of 33.19 kw/ft. The fission gas release, radius of melting, and density change of the UO₂ were in good agreement with predicted values even though there was considerable axial redistribution of fuel. The irradiation behavior of the UC could not be predicted quantitatively, but the qualitative expectations were confirmed.

The nature of the failure of pin 1-2 remains the thorniest problem to be solved. Further study of that failure will be justified only if a pattern of failures develops as the entire experiment proceeds.

At this low burnup, even at this high power, it is not possible to develop an empirical model of the UO₂ fuel swelling, nor to determine the restraining effect of varying clad thickness. It seems safe to say that the threshold burnup for UO₂ fuel swelling has not been reached.

REFERENCES

- 1. Rough, F. A., and Chubb, W., "An Evaluation of Data on Nuclear Carbides," BMI-1441, May 1960.
- 2. Hare, A. W., and Rough, F. A., "Irradiation Effects on Massive Uranium Carbide," BMI-1452, July 1960.
- 3. Sinizer, D. I., Webb, B. A., and Berger, S., "Irradiation Behavior of Uranium Carbide Fuels," International Atomic Energy Symposium on Radiation Damage in Solids and Reactor Materials, Venice, Italy, May 1962.
- 4. Webb, B. A., "Carburization of Austenitic Stainless Steel by Uranium Carbides in Sodium Systems," NAA-SR-6242, February 1962.
- 5. Belle, J., ed., "Uranium Dioxide: Properties and Nuclear Applications," Naval Reactors, Division of Reactor Development, USAEC: p. 177-189, July 1961.
- 6. Robertson, J. A., "k d θ in Fuel Irradiations," CRFD-835, April 1959.
- 7. Booth, A. H., "A Method of Calculating Fission Gas Diffusion From UO₂ Fuel and its Application to the X-2-f Loop Test," CRDC-721, January 1960.

<u>Table I</u>

Nominal Power and Burnup for UO2* Fuel Pins in Each Capsule

			Power Level, kw/f		
		<u>25</u>	<u>30</u>	40	<u>50</u>
	10	Capsule 1	14	7	
Burnup MWD/MTU	25	2	5	8	10
	40	3	6	9	11
Bur	60			13	
103	80		that Allies		12

The UC pins are designed to give the same power level as the UO₂ pins, but being denser in U, their enrichment must be less. This makes their burnup in terms of total uranium less than the UO₂.

Table II

Materials, Enrichment and Clad Thickness - Each Fuel Pin

Identification Capsule - Pin	Fuel	Enrichment U-235, \$	348 Stainless Steel clad wall, 10 ⁻³ Inch
1-1	υo ₂	8,5	10
1-2	[ı	20
1-3	Į.	+	30
1-4	UC	6.8	20
2-1	υ0 ₂	8.5	10
2-2	j		20
2-3	ŧ	+	30
2-4	UC	6.8	50
3-1	vo ₂	8.5 †	10
3-2		İ	20 30
3-3	*	*	20
3-4	UC	6.8	10
4-1	υ ₀ 2	12.5 	20
4-2	l l		30
4-3 1. 1.	UC.	♥ 10	20
4-4 5-1		12.5	10
5-2	^{UO} 2	i	20
5-3	Ţ	ļ	30
)−3 5–4	υc	10	20
6–1	υo ₂	12.5	10
6-2	1	1	20
6-3	1	ļ	30
6-4	UC	10	20
7-1	UO ₂	17.5	10
7-2	1	1	20
7-3	↓	.	30
7-h	UC	14	20
8-1	υo ₂	17.5	10
8–2		İ	20
8–3	+	+	30
8- 4	UC	14	20
9-1	902	17.5	10
9-2	1		20
9-3		*	30 20
9- 4	UC	14	10
10-1A 10-2A	1002	12.5 	20
	1	ļ.	30
10-3A 10-4A	uc	10	20
11-1A	υο ₂	12.5	10
11-2A	12	1	50
11-3A	1	1	30
11-4A	uc	10	20
12 -1A	UO ₂	12.5	10
12-2A	ا	1	20
12-3A	1	 	30
12-4A	υc	10	20
13(10)-3	υo ₂	8.5	30
13(10)-4	ບຕົ້	6.8	20
13(11)-4			20
13(12)-4	Ť	t	20

The A stands for the alternate (higher) enrichment finally selected for the manufacture of these fuel pins.

^{**} The number in parenthesis refers to the capsule originally designated for fuel of this enrichment.

<u>Table III</u>

<u>Analysis of UO</u>2 <u>Impurities</u>

Four samples of pellet chips from each enrichment were spectrographically analyzed. The range of the results on all the samples are reported below in parts per million of ${\rm UO}_2$.

Element	Amount, ppm	Element	Amount, ppm
Ag	<.4	Mn	2.9-7.6
Al	<10-23	Мо	3.1-25.0
Au	<2	Na	<1-2.8
В	<.233	Ni	18-200
Ba	<10	P	<50
Ве	<.0204	Рb	<1
Bi	<.4	Rb	<2
Ca	<.5-36	Sb	<2
Cđ	<.4	Si	86-340
Co	1.0	Sn	<1-2.7
Cr	19-51	Sr	<20
Cs	<10	Ti	<1-3.6
Cu	7.3-75	Tl	<10
Fe	73-160	V	<2
In	<2	W	<20
K	<1	Zn	<20
Li	<.2	Zr	<20
Mg	3.7-18		

Table IV

Analysis of Rare Earth Content of UO

One sample was spectrographically analyzed from each enrichment. Results are in parts per million of ${\tt UO}_2$. The results were the same for all three enrichments.

La	<.3
Се	< 1
Pr	< 1
Nd	<.3
Sm	<.1
Eu	<.03
Gđ	<.03
Tb	<.1
Dy	<.1
Но	<.1
Er	<.03
Tm	<.3
Yb	<.01
Lu	<.3

Table V

Analysis of AISI 348 Stainless Steel

Used for Cladding all Fuel Pins

Element	Weight %
С	0.074
Mn	1.70
Si	0.77
S	0.012
P	0.022
Cr	18.39
Ni	10.39
Мо	0.08
Cu	0.06
Co	0.013
Cb + Ta	0.87
Ta	0.079
Fe	Balance

Table VI

Designed Behavior of Capsule 1

Based on Maximum (Initial) Power* = 49.6 kw/ft

	Fuel Pin			
	1-1	1-2	<u>1-3</u>	1-4
Fuel	U0 ₂	υ0 ₂	υ0 ₂	UC
Enrichment, %	8.5	8.5	8.5	6.8
Burnup, 10 ²¹ Fissions/cc (Calculated for actual ti	0.2264 me)	0.2264	0.2264	0.2264
Effective fuel power time, 10 ⁶ sec (Actual)	3.24	3.24	3.24	3.24
Temperature Profile Clad OD, °C Clad ID, °C Fuel Surface, °C Fuel Center (solid), °C	821.9 895.3 1095.2	904.9		762.5 904.9 1104.8 1613
Relative radius of melting, rm/a	.8	.8	.8	
Fission gas release % STP cc	67.2 7.11	67.2 7.11	67.2 7.11	0 0
Clad diameter change, %	0	0	0	**

^{*}Includes 1 kw/ft > heat.

^{**}Cannot be predicted

Table VII

Designed Behavior of Capsule 1

Based on Mean (Initial) Power* = 33.19 kw/ft

	Fuel Pin				
	1-1	1-2	<u>1-3</u>	1-4	
Fuel	uo ₂	U0 ₂	υ0 ₂	UC	
Enrichment %	8.5	8.5	8.5	6.8	
Burnup, 10 ²¹ fission	.2264	.2264	.2264	.2264	
(Calculated for Actual Time	:)				
Effective fuel power time, 10 ⁶ sec (Actual)	3.24	3.24	3.24	3.24	
Temperature Profile Clad OD, °C Clad ID, °C Fuel Surface, °C Fuel Center (solid), °C	567.5 616.6 750.6		489.7 628.5 762.5	-	
Relative radius of melting, rm/a	•55	•55	•55		
Fission gas release % STP cc	40.2 4.25	40.2 4.25	40.2 4.25	0	
Clad diameter change, %	0	0	0	**	

^{*}Includes 1 kw/ft 8 heat.

^{**} Cannot be predicted.

Table VIII

Fission Gas Samples Taken at Plumbrook Hot Cells

NOTE: All operations and pressure measurements made at room temperature.

	Fuel Pin or Can			
	1-1	1-2	<u>1-3</u>	1-4
System pressure before puncture, P ₂ , µ	12	14	7	7
Pin or can free volume, V ₁ , cc	2.065	6.189	2.024	1.993
System pressure after puncture, P3, µ	3000	4500	2350	710
Total volume, V ₃ , cc	2017	2021.2	2017	2017
Pressure after sweep, $P_{l_{\!4}}$, μ	200	250	160	46
Volume after sweep, V ₁₄ , cc	1917	1921.2	1917	1917
Pressure in pin or can, P ₁ , atm	3.94	1.93	3.07	0.935
Amount of gas shipped, P_BV_B , atm-cc	7.44	11.322	5.8133	1.747
Amount of gas in pin (or can), P_1V_1 , atm-cc	7.94	11.954	6.2168	1.863

Table IX

Analysis of Gas Samples, Mole %

		<u>Fuel Pin</u>	
	1-1	<u>1-3</u>	1-4
Species			
H ₂	0.37	2.30	2.73
Не	14.86	22.95	88.79
N_2	25.91	2.38	3.12
02	6.45	0.19	1.29
Ar	0.32	0.05	0.07
сн ₁₄	0.16	0.18	0.30
Other Hydrocarbons	-	-	0.90
Kr	7.05	9.84	0.43
Xe	44.88	62.11	2.37
"Fission Gas" (Xe + Kr)	51.93	71.95	2.80
"Air" (N ₂ + O ₂ + Ar)	32.68	2.62	4.48
Calculated P ₁ V ₁ (room temp.), atm-cc	6.4838	7.3999	2.1627
Calculated P _l (room temp.), atm	3.14	3.66	1.085

Quantity of Individual Isotopes of Kr and Xe

Present in the Fuel Pins, atm-cc @ 0°C

		Fuel Pin	
	1-1	1-3	1-4
Kr-83	0.0902	.0934	1.07×10^{-3}
K r- 84	0.1793	.1850	2.77×10^{-3}
Kr-85	0.0478	.0494	0.75×10^{-3}
Kr-86	0.3420	.3526	4.26×10^{-3}
Xe-131	0.4266	.4353	4.80×10^{-3}
Xe-132	0.6590	.6774	7.68×10^{-3}
Xe-134	1.2508	1.3002	14.61×10^{-3}
Xe-136	1.8617	1.8821	21.75×10^{-3}
Total released	4.8574	4.9754	57.69×10^{-3}
Total generated	10.58	10.58	10.58
% Fission gas released	45.91	47.03	0.55

Table XI

Post-Irradiation Dimensions of Fuel Pins 1-1, 1-3, 1-4

		Fuel Pin	
	1-1	1-3	1-4
Length, in. (3 readings taken) mean	8.7138 8.7134	8.7028 8.7025 8.7025 8.7026 <u>+</u> 0.0002	8.7079 8.7072
Diameter, in. (mas and min)			
top mear	0.31612	0.36280 0.36249 0.3626 <u>+</u> 0.0002	0.33299
	0.32728	0.36640 0.36728 0.3668 <u>+</u> 0.0004	0.33348
	0.32234	0.36544 0.36570 0.3656 <u>+</u> 0.0001	0.33581
Displacement volume, cc	. 13	14	14

Fission Densities from U-isotope and Cs-137 Measurements on Pellets 1 and 6 from Pins 1-1, 1-3, 1-4

	Fission f/cc	Mean* fission	137 Fissi	Mean* fission density for p
	Fission density, f/cc	Mean* fission density for pin f/cc	Fission density, f/cc	lean# fission density for pin f/cc
1-1-1	2.38 x 10 ²⁰	2.37	3.20 × 10 ²⁰	2,36
1-1-6	2.36 x 10 ²⁰	2.37 x 10 ²⁰	2.00 × 10.20	x 10 ²⁰
1-3-1	4.75 x 10 ²⁰	2.73	3.17 x 10 ²⁰	2,19
1-3-6	1.86 x 10 ²⁰	2.73 x 10 ²⁰	1.77 × 10 ²⁰	2.19 x 10 ²⁰
1-4-1	7.51 x 10 ²⁰	4.11		84.8
1-4-6	2.65 x 10 ²⁰	4.11 × 10 ²⁰	2.28 x 10 ²⁰ 2.57 x 10 ²⁰	2.48 x 10 ²⁰
	•			

* Assuming linear axial gradient.

Table XIII

Atom Percent Burnup from Mass Spectrographic

Measurements on Pellets 1 and 6 from Pins 1-1, 1-3, 1-4

Pellet	Atom Percent U	Burnup
	Spectrograph	Mean* for pin
1-1-1 1-1-6	$ \begin{array}{c} 1.16 \pm 0.21 \\ 1.15 \pm 0.20 \end{array} $	1.15
1-3-1 1-3-6	$\begin{array}{c} 2.26 \pm 0.14 \\ 0.90 \pm 0.16 \end{array}$	1.31
1-4-1 1-4-6	$ \begin{array}{c} 2.29 \pm 0.05 \\ 0.81 \pm 0.07 \end{array} $	1.25

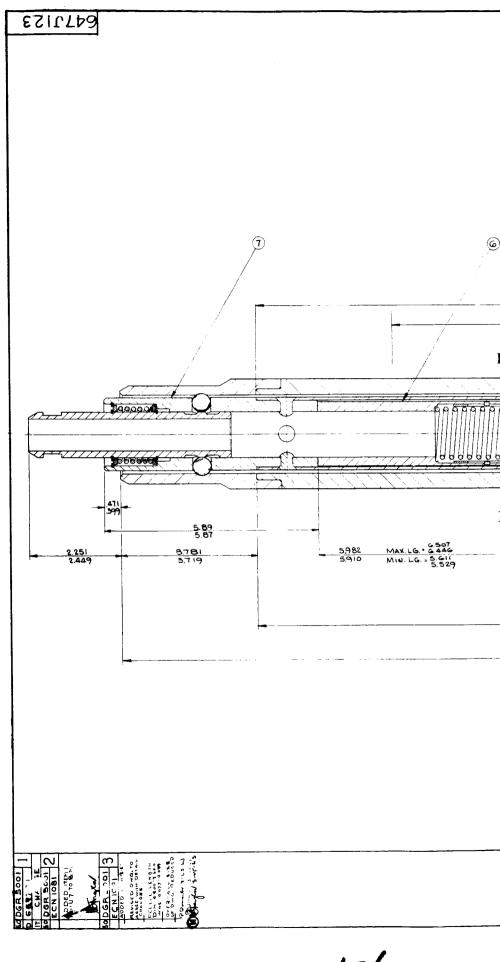
^{*} Assuming linear axial gradient.

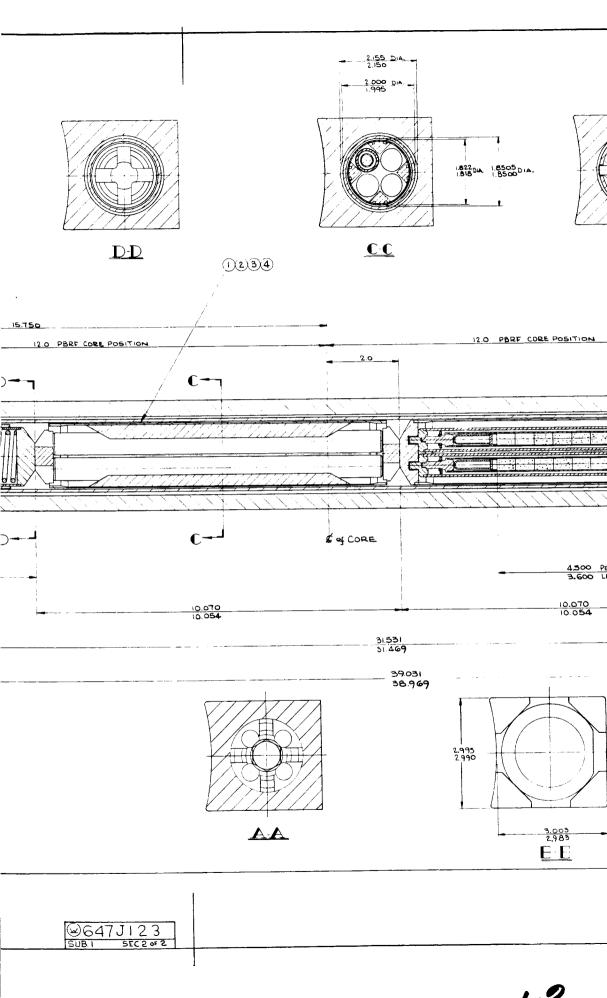
Table XIV

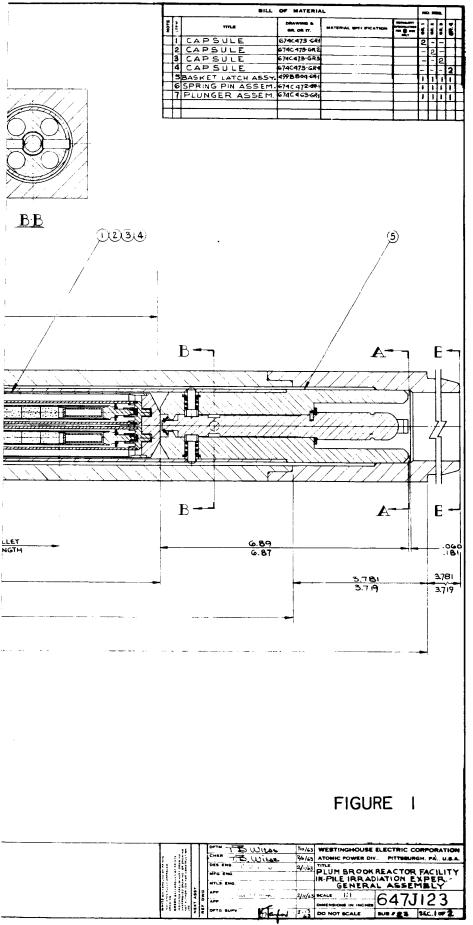
Density Measurements on Fuel Samples from Capsule 1

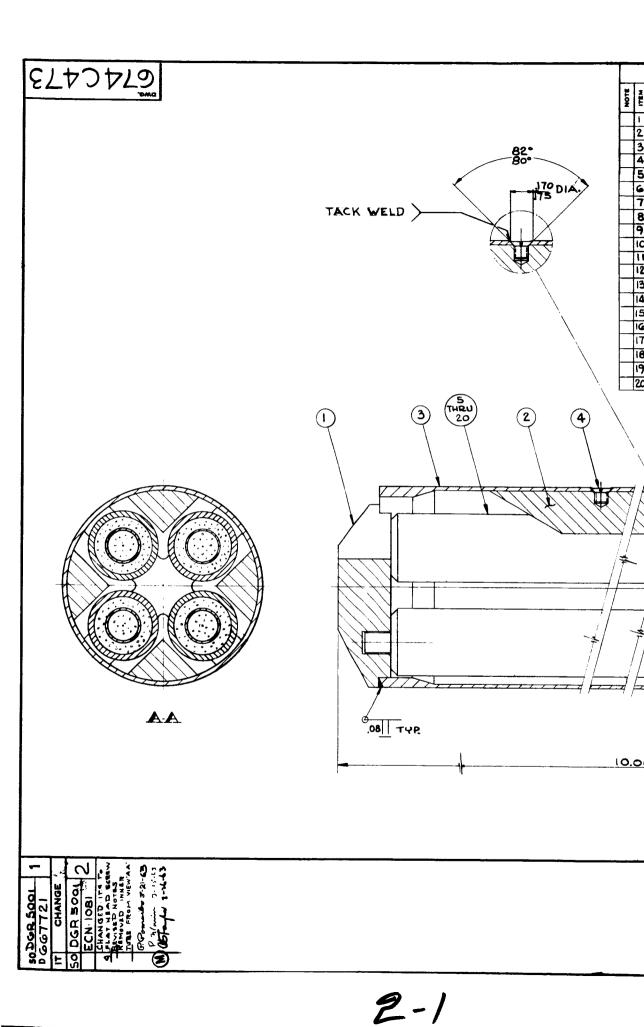
Fuel Pellet

	1-1-3	1-3-4	1-4-4	1-1-8	1-3-8	1-4-8
Fuel Material	00 ²	υ0 ₂	UC	no ⁵	U0 ₂	UC
Clad + fuel, dry weight (mean), grams	2.78670 <u>+</u> 0	5.19855 <u>+</u> 0.00005	7.16685 + 0.00015	4.86865 + 0.00005	9.34975 <u>+</u> 0.00005	8.0061 <u>+</u> 0.0001
Clad, dry weight (mean), grams	0.58904 <u>+</u> 0.00005	1.4206 ± 0	1.10095 <u>+</u> 0.00005	0.52725 <u>+</u> 0.00005	2.43060 <u>+</u> 0.0001	1.1848 <u>+</u> 0
Clad + fuel, im- mersed weight (mean), grams	2.34065 <u>+</u> 0.00005	4.34065 <u>+</u> 0.00035	6.18975 <u>+</u> 0.00015	4.10635 <u>+</u> 0.00025	7.74925 + 0.00025	6.9194 <u>+</u> 0.0003
Clad, immersed weight (mean) grams	0.47065 <u>+</u> 0.00015	1.13205 <u>+</u> 0.00015	0.87825 <u>+</u> 0.00005	0.41845 <u>+</u> 0.00005	1.9400 <u>+</u> 0	0.9461 <u>+</u> 0.0001
CCl ₄ tempera- ture, °C	20.56	20.00	20.56	20.00	20.56	20.0
CCl ₄ density, g/cc	1.59297 <u>+</u> 0.00020	1.59405 + 0.0002	1.59297 <u>+</u> 0.0002	1.59405 <u>+</u> 0.0002	1.59297 <u>+</u> 0.0002	1.59405 <u>+</u> 0.0002
Fuel density, g/cc	10.68454 <u>+</u> 0.00558	10.5774 + 0.00725	12.8086 + 0.00414	10.5898 + 0.00449	9.9306 <u>+</u> 0.00275	12.8225 + 0.00527
Percent of Theoretical Density	97.40	96.42	93.97	96.53	90.53	94.08









BJLL OF MATERIAL					NO. REG.											
TITLE	DRAWING & GR. OR'IT.	MATERIAL SPEC.	EQUIV,	- K	9 . A	OR 3	4.4									T
CAPSULE END PLUG	674C472-17.1		+	2	2	2	<u> </u>	H	\vdash	-	-	Н	\vdash		+	+
FLOW BLOCKER	G14C472-IT.2			4	4				t	-		Н	Н	\vdash	+	十
CAPSULE TUBE	G74C472-17.3			Ť	ī	ī	Ť	+	t	Г		Н	Н	H	+	╅
SCREW!	G74C472-IT4			12	12	12	12	+	\vdash	\vdash	-	Н	\vdash	-	+	+
FUEL ROD CAN ASSEM.	881D852.6c.1			tī	Γ-		۳	H	t^-	\vdash	H	Н	\vdash	\vdash	+	十
A	-Gr.2			1	Н				t	H	-	Н	Н	\forall	+	+
	-Gr.3			1	Н	Г		<u> </u>	t^-			Н	\vdash	\vdash	+	+
	-Gr.4			Ī	Н			T	T	-	Н	H			┪`	十
	-Gr.5			Ė	T			┪	T		Н	H		+	+	十
	-Gr.6			Г	T	П	_	Н	 		П		Ħ	\dashv	+	十
	- Gr.7				1		Н	Ι	\vdash	Н	Н	H			+	╅
	- 61.8				T				Π	П	П				1	十
	-Gr:9				Г	ī					П				+	十
	-Gr.10					1	Т		1		Н			\vdash	+	十
	-Gr.II			_		1					П			+	+	十
	-Gr.12					ī	Г	Г	T		H			\dashv	+	十
	-Gr.13						ī	Г		Г	П				+	+
	-Gr.14				Г	П	T		Г	Г	П			\dashv	+	\top
	-Gt15			Г	Г		T		\vdash		Н	\vdash	\Box	+	+	+
FUEL ROD CAN ASSEM.	881D852Gr.K				1	Н	Ť	\vdash	1	\vdash	Н	Н	\vdash	-+	+	+

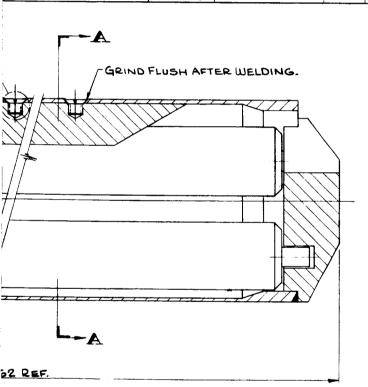


FIGURE 2

I	20 0 2325	3 2	DPTE. BUPY. Garlo	2-13	DO NOT SCALE SUB # 2
		5 5	MP.		DIMERNO IN INCHES 6 14 4 1 2
Ŀ	i, iii	3 3	APP. PMapm	2/11/43	SCALE 2:1 (7/1/17)
1			MTLS, ENG.		CAPSULE ASSEM.
ı		4	MFG. ENG.		IN-PILE IRRADIATION EXPER:
j		5	DES. ENG. R. DUNCAN	2/11/63	PLUM BROOK REACTOR FACILITY
		23	CHKR. T.D. WIESE	71/6:	ATOMIC POWER DIV PITTSBURGH, PA., U.S.A.
- 1			T.B. WULLE	Make.	WESTINGHOUSE ELECTRIC CORPORATION

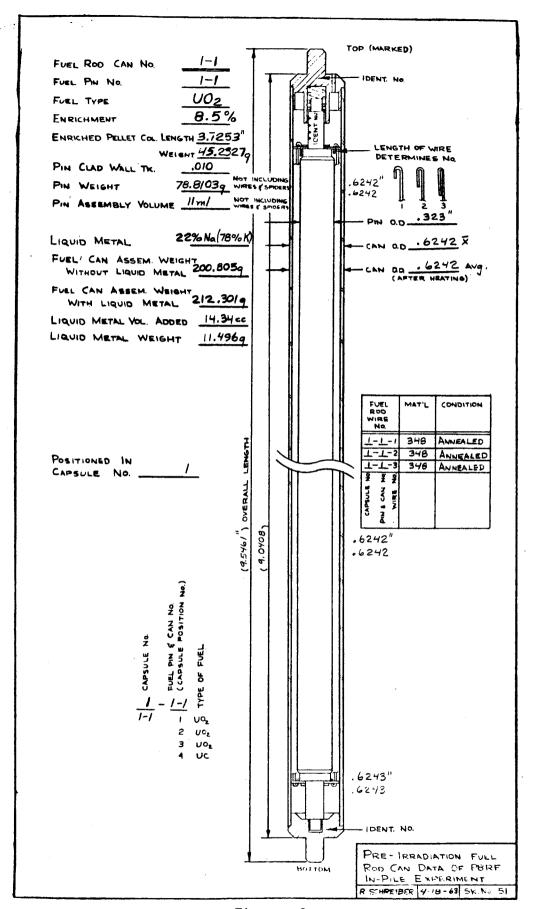
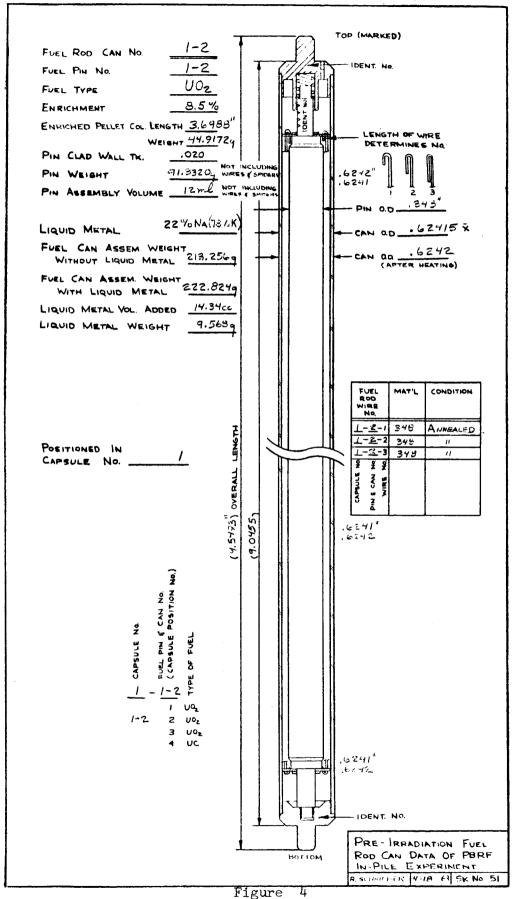


Figure 3



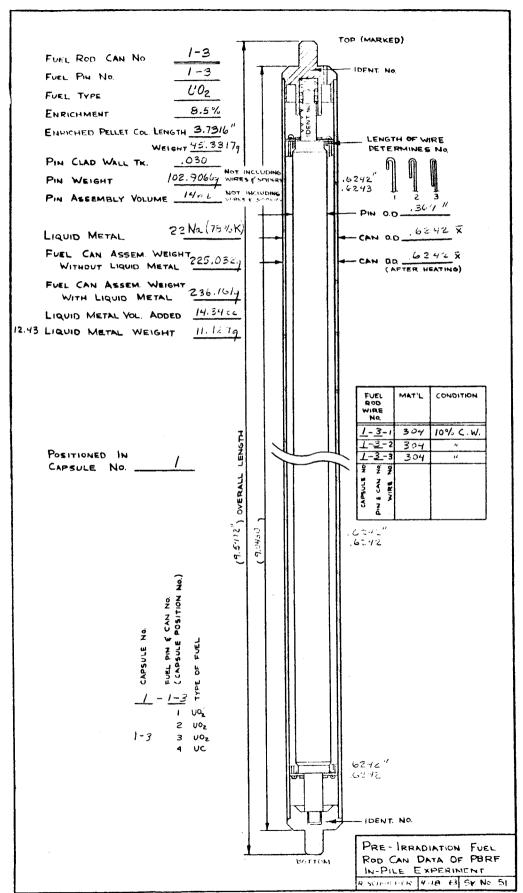


Figure 5

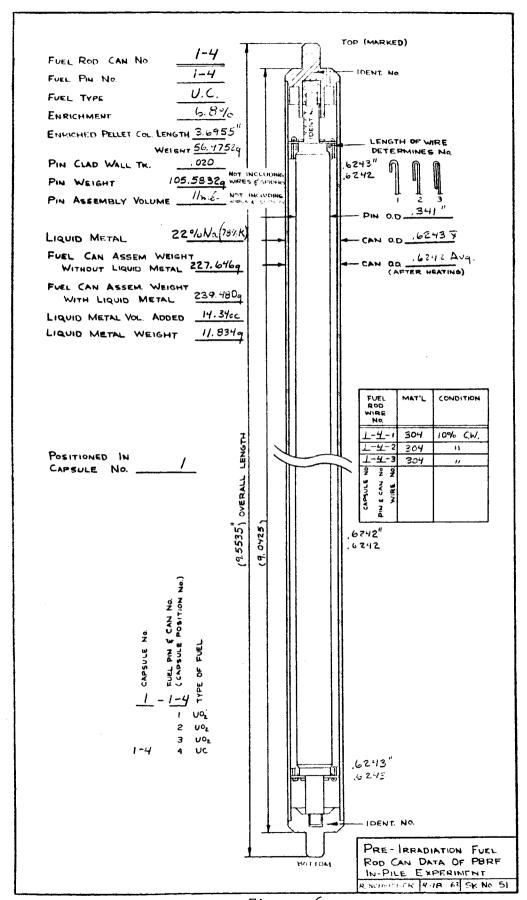


Figure 6

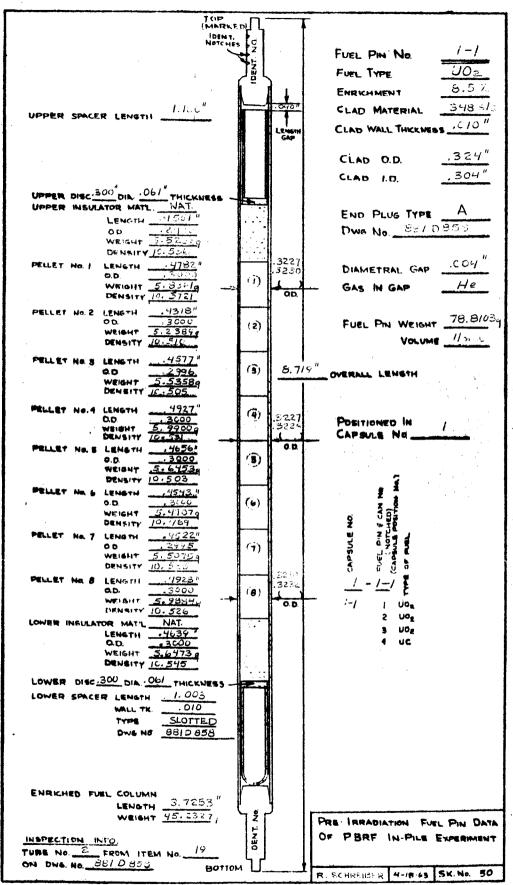
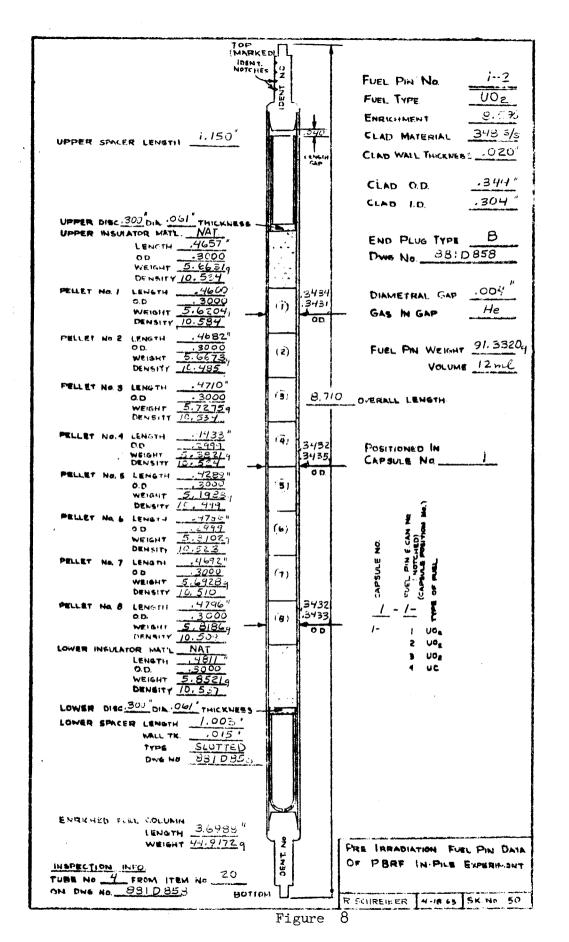


Figure '



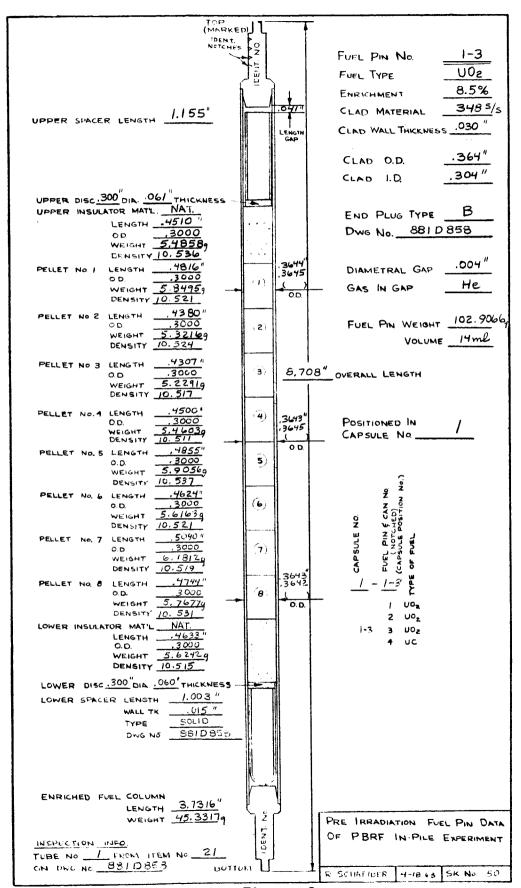
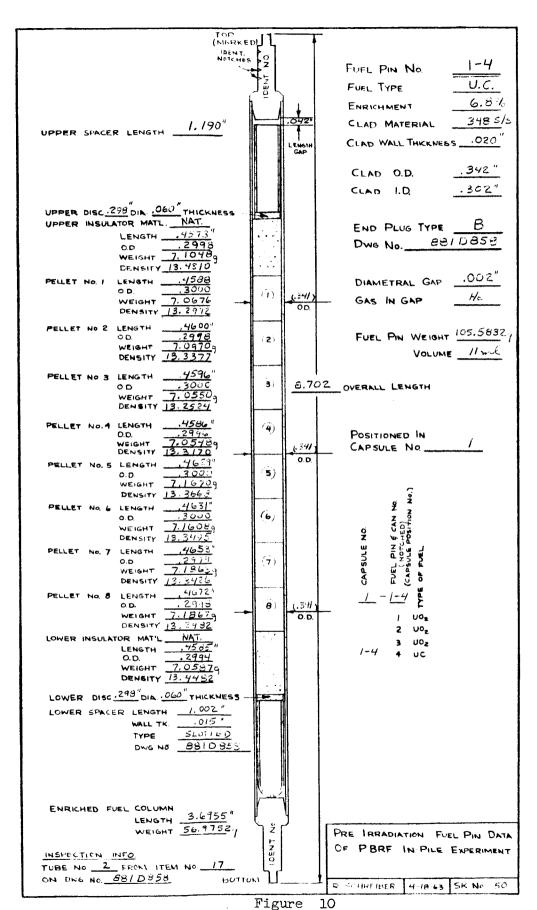
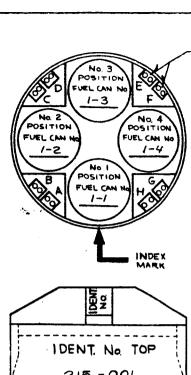


Figure 9

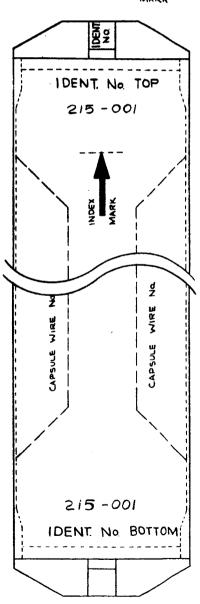


_



CAPSULE TEST
WIRE POSITIONS
(IDENTIFY POSITION
ON TOP OF FLOW
BLOCKER)

CAPSULE No. / ALUMINIUM BODY



CAPSULE TEST WIRE NO	MATE	RIAL	CONDITION					
<u> </u>	(4)	48	ANNE	ALED				
1-4-2*				i .				
1-B-1								
1-B-2*								
1-c-1								
1-C-2*								
1-D-1								
1-D-2"		İ						
1-E-1	(U) 16	-20	10%	C.W.				
1-E-2*								
1-F-1								
1-F-2*								
1-6-1								
1-G-2"								
<u>H-1</u>								
<u>/</u> -H-2*	•		1					
CAPSULE NO.								

No. 2 WIRES ARE 1/2 INCH SHORTER THAN No. 1

PRE- IRRADIATION CAPSULE DATA OF PBRF IN-PILE EXPERIMENT #. 23-63 SK No. 52

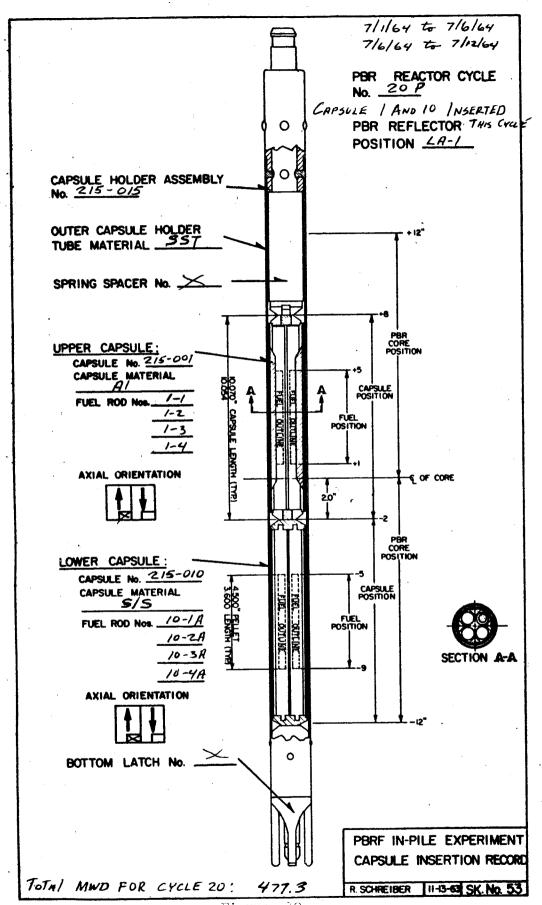


Figure 32

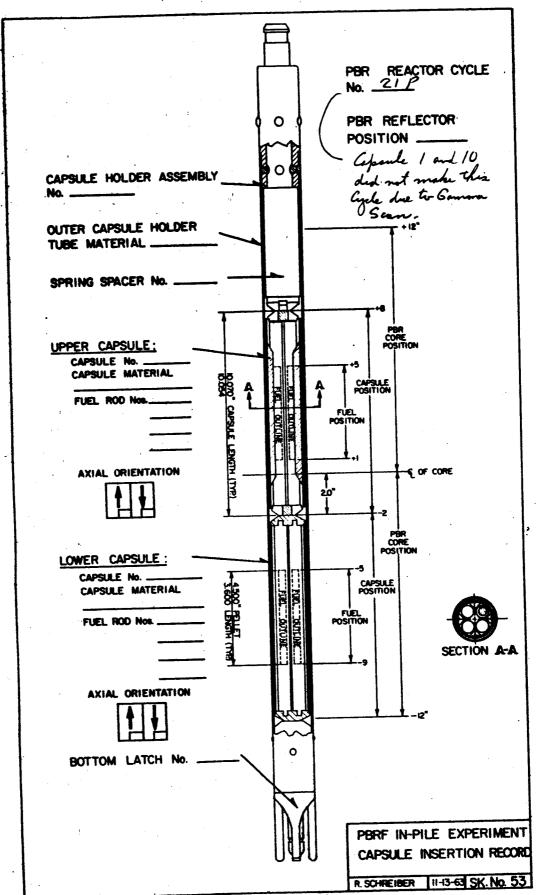
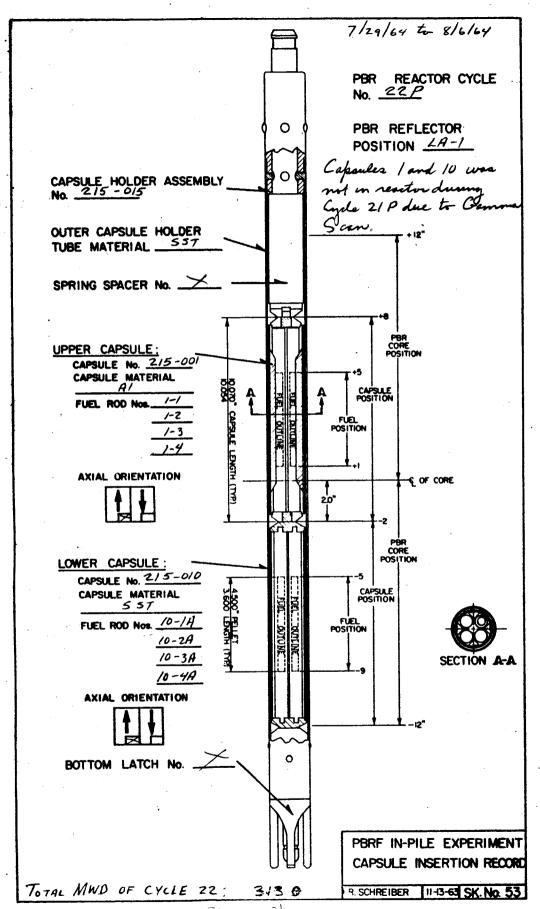


Figure 13



Tigura 14

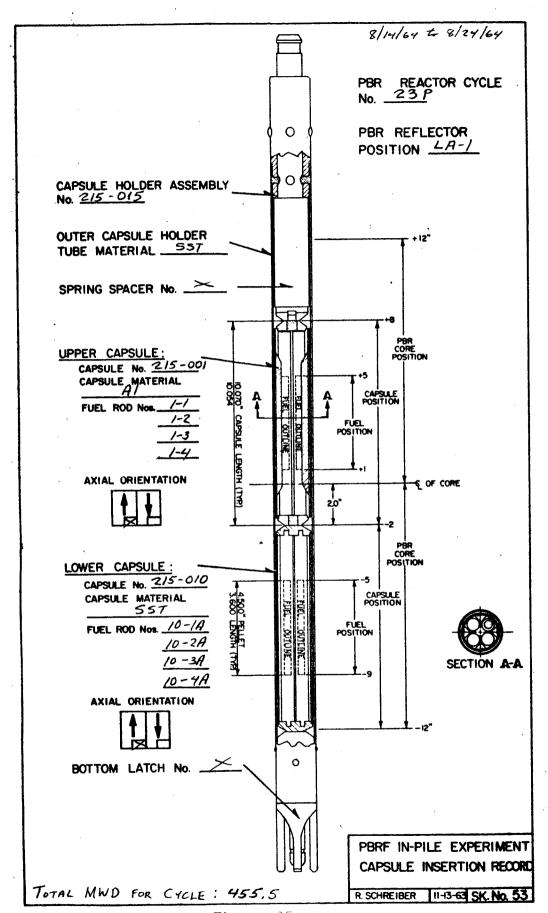


Figure 15

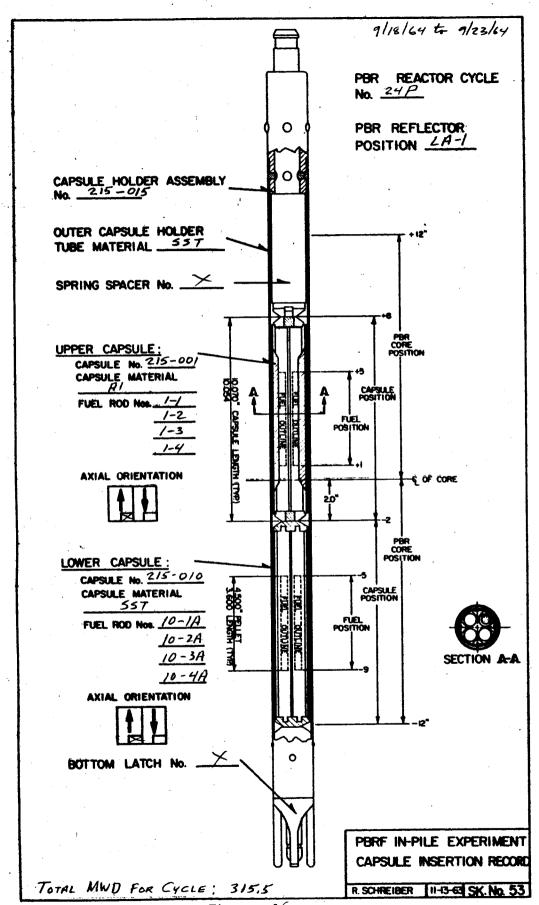


Figure 16

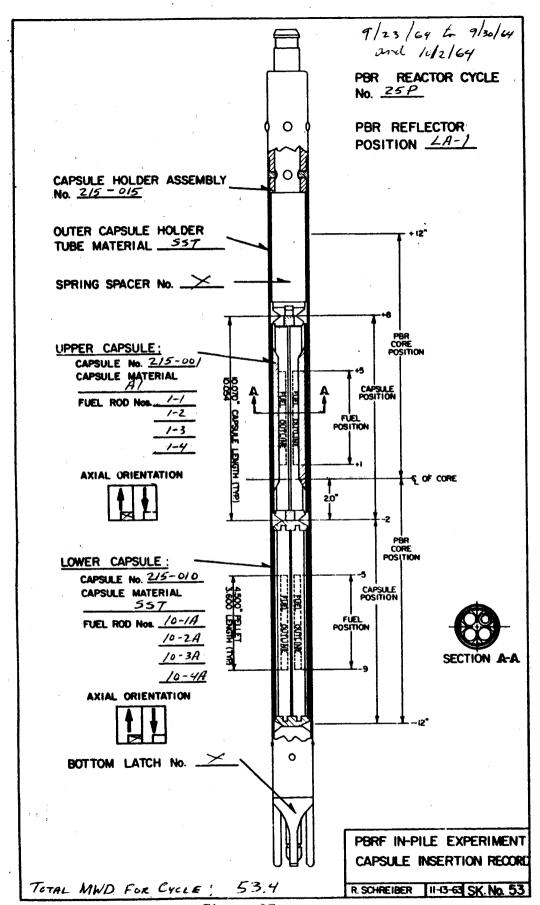


Figure 17

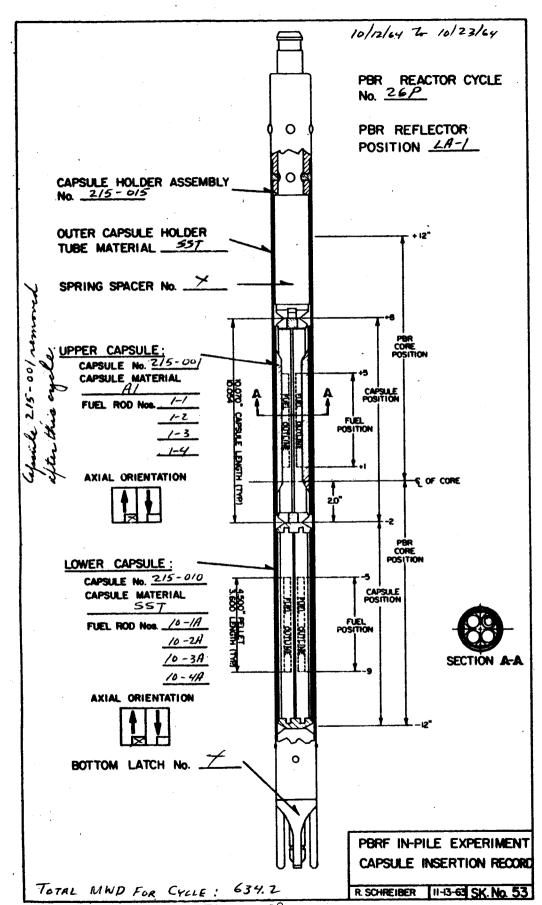
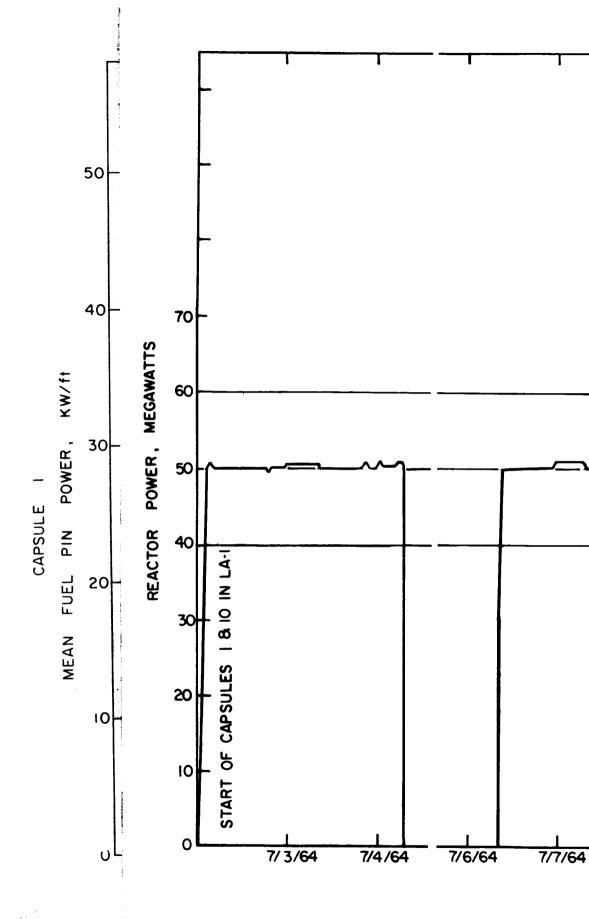
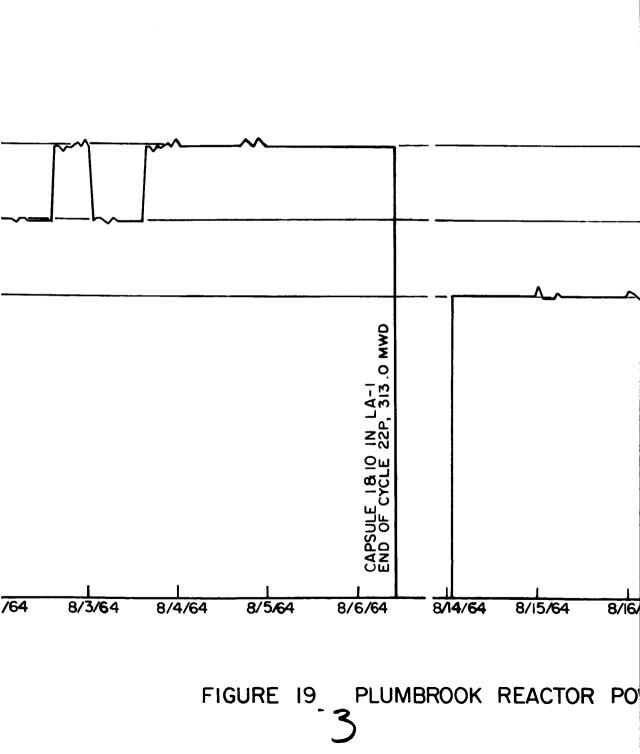


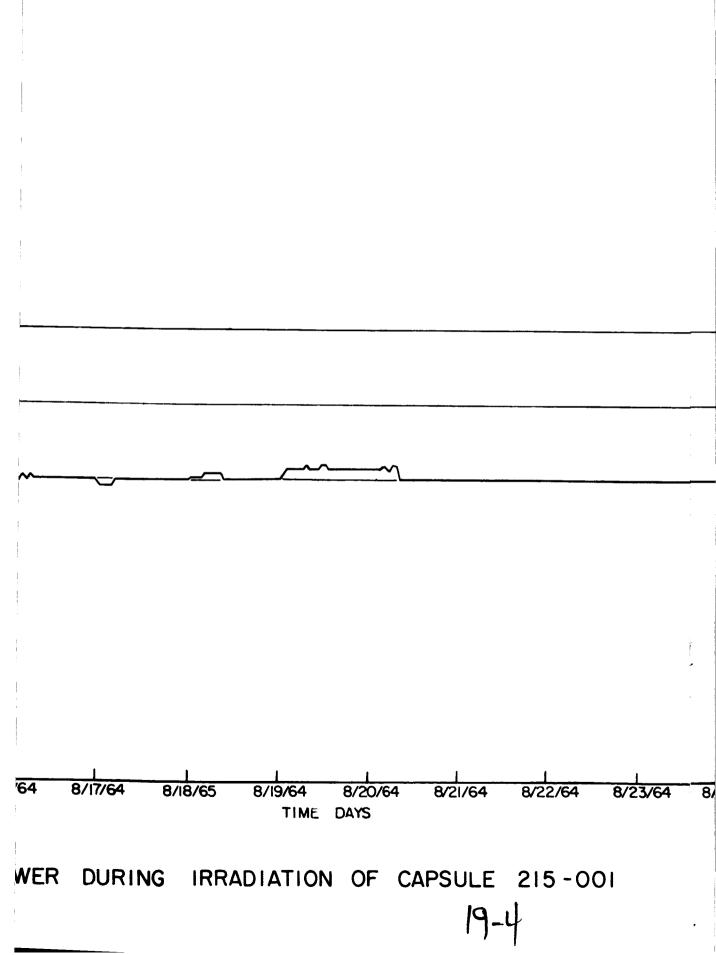
Figure 1

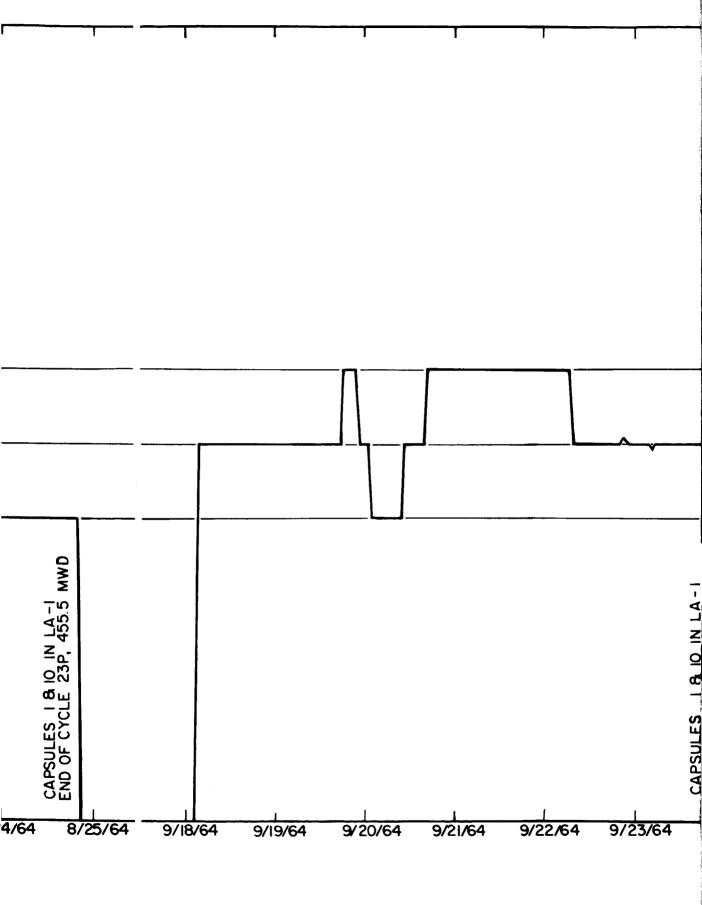


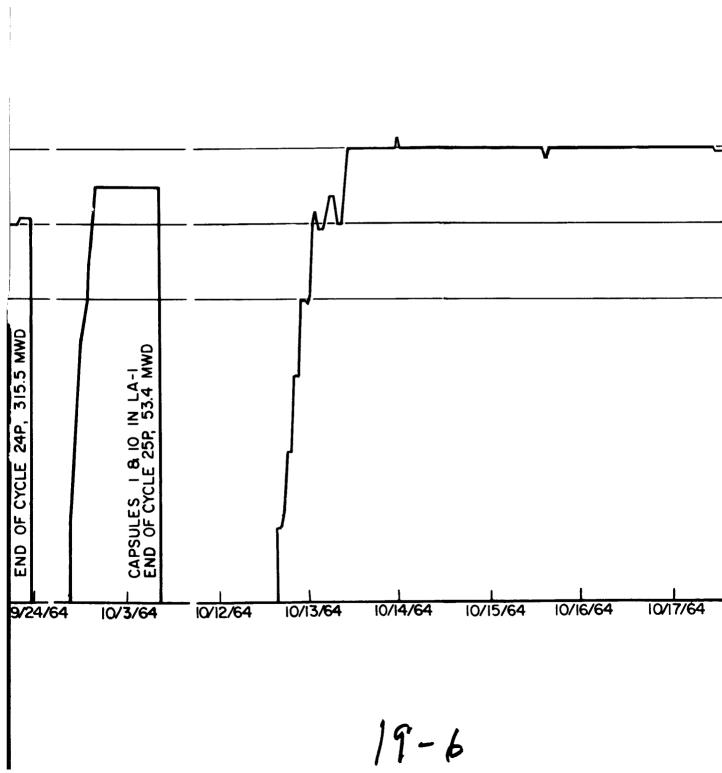


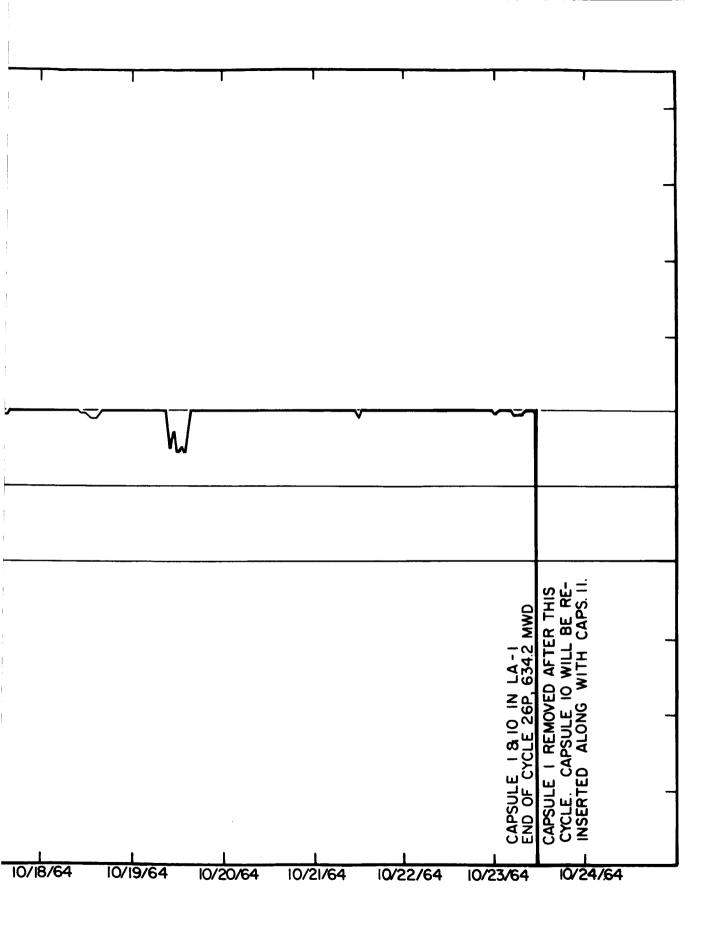
8/

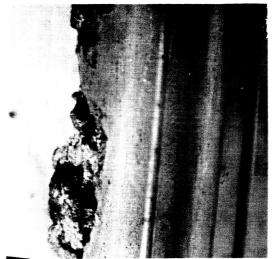




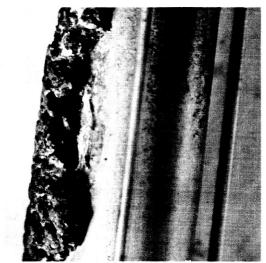








SIDE VIEW, END OF DEPOSIT.



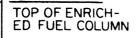
SIDE VIEW, CENTER OF DEPOSIT.





CLAD C

CLOSE UP VIE LEFT TO RIC



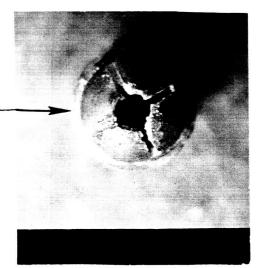




FRONT VIEW, CLOSE-UP OF RUPTURE NEAR END.



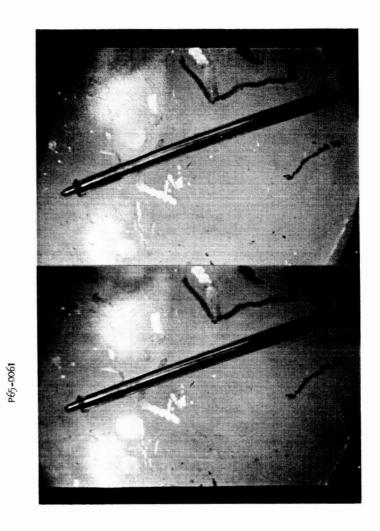
FRONT VIEW, CLOSE-UP OF RUPTURE CENTER SECTION.



END VIEW OF AXIAL EXPANSION MARKER. (PARALLEL EDGES INDICATES NO AXIAL EXPANSION OF FUEL COLUMN)

.D. 0.344"

WS ARE REVERSED HT.



STEREO PHOTOGRAPH OF FUEL PIN 1-1 AFTER IRRADIATION.

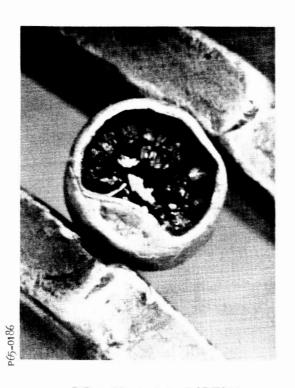
21.1



PELLET 1-1-1, TOP



PELLET 1-1-2, TOP



PELLET I-I-I, BOTTOM



PELLET 1-1-2, BOTTOM



NOT TAKEN

PELLET 1-1-3, TOP

PELLET 1-1-4, TOP

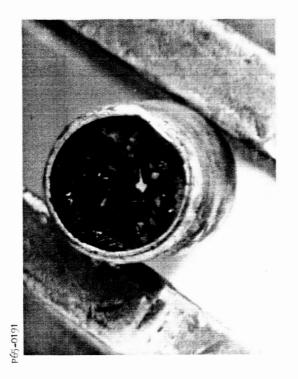


NOT TAKEN

PELLET 1-1-3, BOTTOM

PELLET 1-1-4, BOTTOM

FIGURE 21 ' MACROPHOTOGRAPHS OF FUEL PIN



PELLET 1-1-5, TOP



PELLET 1-1-6, TOP



PELLET 1-1-5, BOTTOM



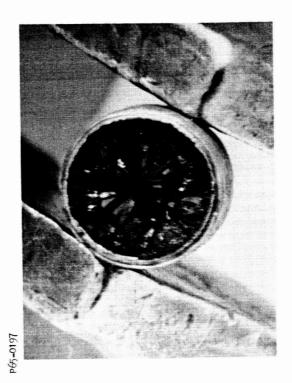
PELLET 1-1-6, BOTTOM

-I FROM CAPSULE 215-001. (UO₂, .010"CLAD).

21-4



PELLET I-1-7, TOP



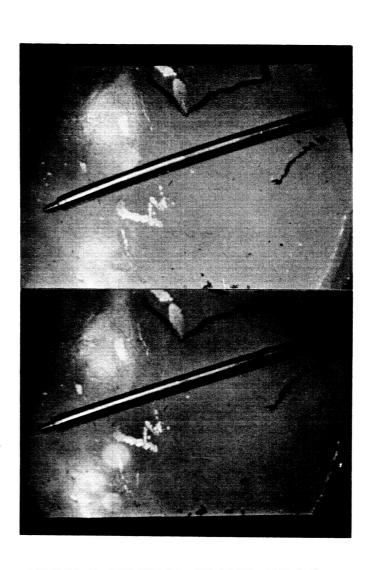
PELLET 1-1-8, TOP



PELLET 1-1-7, BOTTOM



PELLET 1-1-8, BOTTOM



STEREO PHOTOGRAPH OF FUEL PIN 1-3 AFTER IRRADIATION.



PELLET 1-3-1, TOP



PELLET 1-3-2, TOP



PELLET 1-3-1, BOTTOM



PELLET I-3-2, BOTTOM



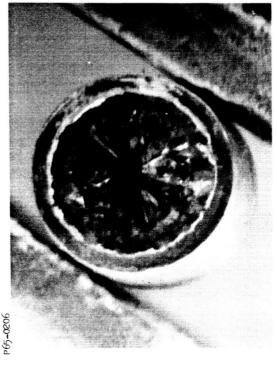
PELLET 1-3-3, TOP



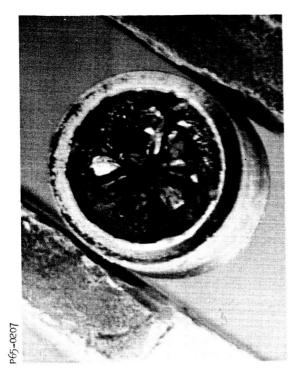
PELLET 1-3-4, TOP



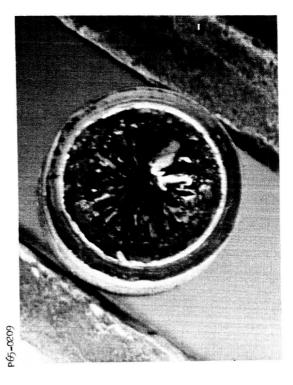
PELLET 1-3-3, BOTTOM



PELLET 1-3-4, BOTTOM



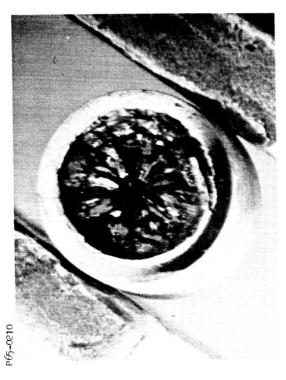
PELLET 1-3-5, TOP



PELLET 1-3-6, TOP

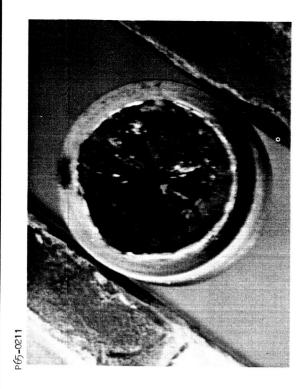


PELLET 1-3-5, BOTTOM

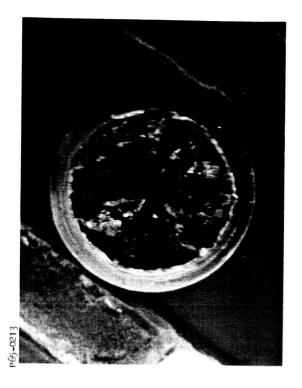


PELLET 1-3-6, BOTTOM

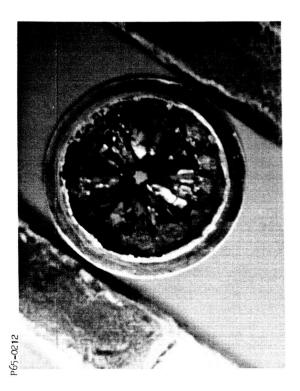
M CAPSULE 215-001. (UO₂, .030" CLAD).



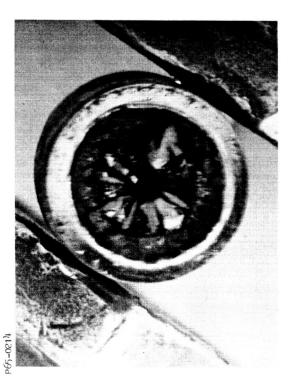
PELLET I-3-7, TOP



PELLET 1-3-8, TOP

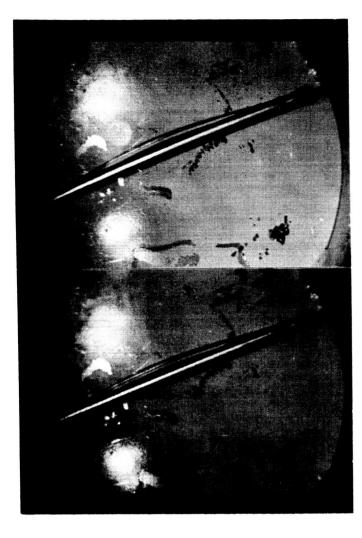


PELLET 1-3-7, BOTTOM



PELLET 1-3-8, BOTTOM

22-5



P65-0063

STEREO PHOTOGRAPH OF FUEL PIN 1-4 AFTER IRRADIATION.



PELLET 1-4-1, TOP



PELLET 1-4-2, TOP



PELLET 1-4-1, BOTTOM



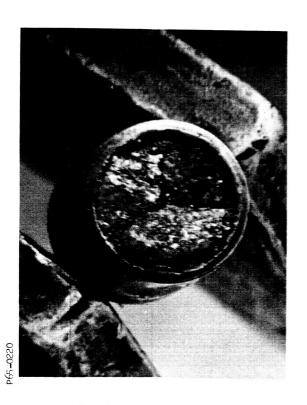
PELLET 1-4-2, BOTTOM



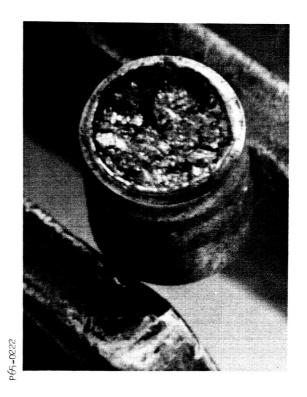
PELLET 1-4-3, TOP



PELLET 1-4-4, TOP



PELLET 1-4-3, BOTTOM



PELLET 1-4-4, BOTTOM

FIGURE 33 MACROPHOTOGRAPHS OF FUEL PIN 1-4 FRON



PELLET 1-4-5, TOP



PELLET 1-4-6, TOP



PELLET 1-4-5, BOTTOM



PELLET 1-4-6, BOTTOM

CAPSULE 215-001. (UC, .020" CLAD).

23-4



PELLET 1-4-7, TOP



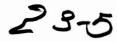
PELLET 1-4-8, TOP



PELLET 1-4-7, BOTTOM



PELLET 1-4-8, BOTTOM



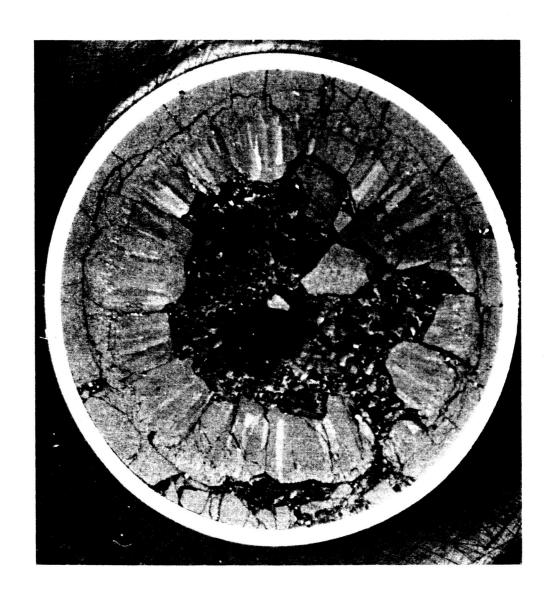


Fig. 24 PELLET 1-1-2



Fig. 25 PELLET 1-1-5

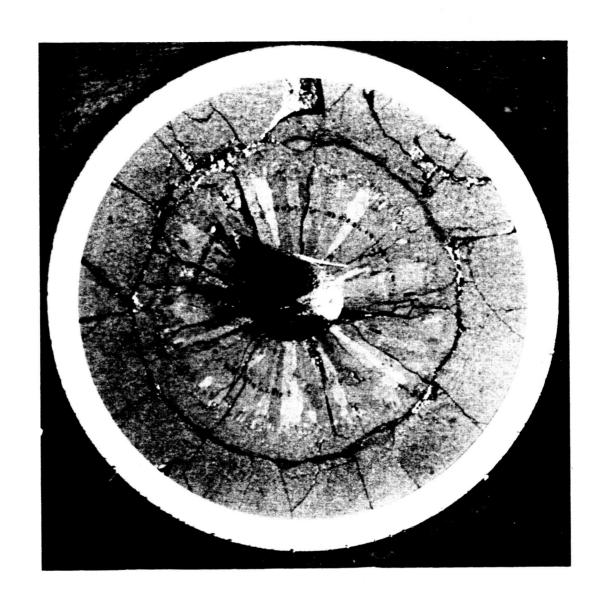


Photo 2814

Fig. 26 TOP GUARD PELLET, 2-3/8" From Top, Bottom Surface, Pin 1-2



Photo 2813

Fig. 27 PELLET 1-2-1, 2-5/8" From Top, Bottom Surface



Photo 2808

Fig. 28 PELLET 1-2-2, 2-5/8" From Top



Fig. 29 PELLET 1-2-3, 3-1/16" From Top

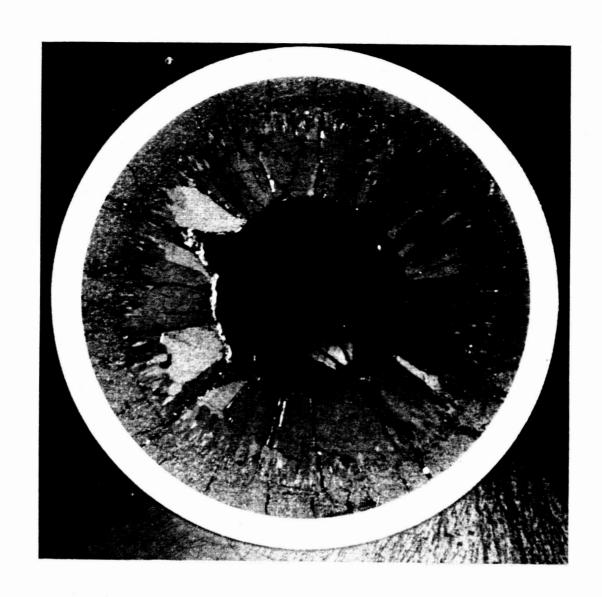


Photo 2810

Fig. 30 PELLET 1-2-4, 3-7/16" From Top



Photo 2811

Fig. 31 PELLET 1-2-5, 3-7/8" From Top



Photo 2812

Fig. 32 PELLET 1-2-6, 4-5/16" From Top

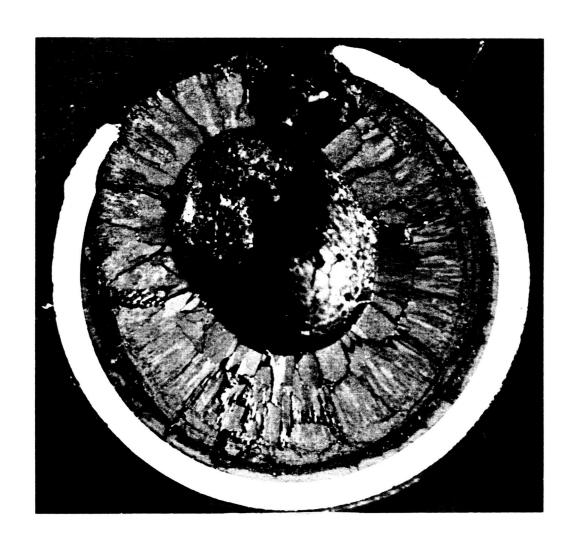


Fig. 33 PELLET 1-2-8, ~ 5-7/8" From Top

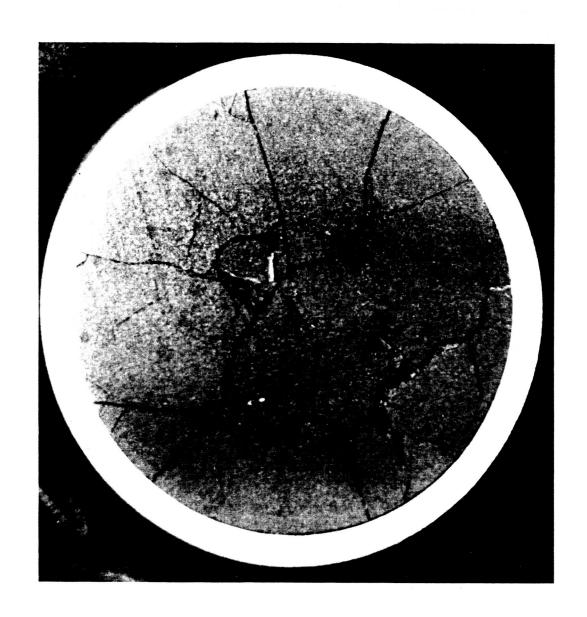


Photo 2818

Fig. 34 LOWER GUARD PELLET, 6-3/8" From Top, Top Surface, Pin 1-2



Fig. 35 PELLET 1-3-2



Fig. 36 PELLET 1-3-5

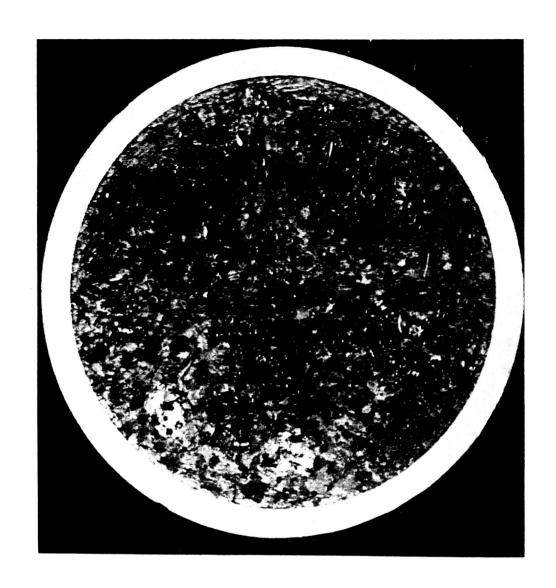


Fig. 37 PELLET 1-4-2



Fig. 38 PELLET 1-4-5

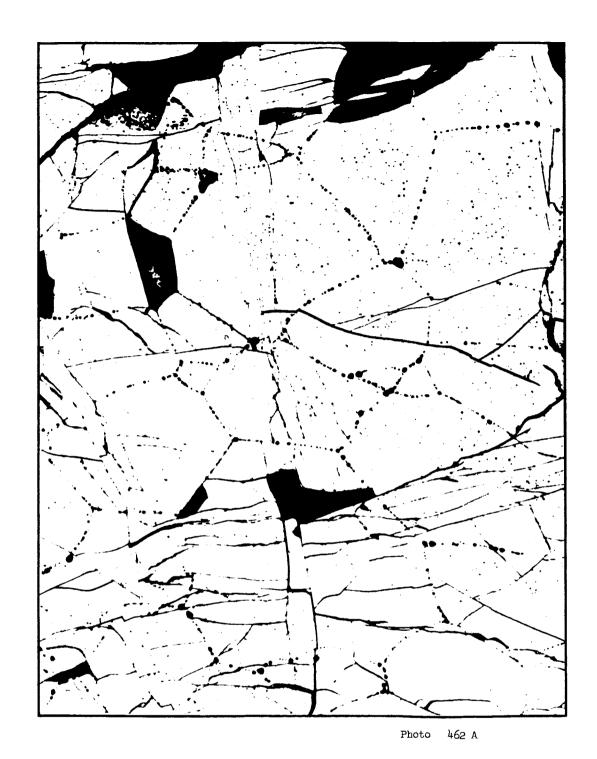


Fig. 39 PELLET 1-4-5, NEAR CENTER, ~ 500X