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Procedia Engineering 141 (2016) 23 - 26

Procedia Engineering

www.elsevier.com/locate/procedia

# MRS Singapore - ICMAT Symposia Proceedings

8th International Conference on Materials for Advanced Technologies

# On the mechanism of BaSi<sub>2</sub> thin film formation on Si substrate by vacuum evaporation

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# Abstract

We report on the formation mechanism of  $BaSi_2$  thin film on Si substrate grown by vacuum evaporation using  $BaSi_2$  granules as source materials. Since the vapor flux at the initial stage of evaporation is known to be Ba-rich, Si supply from the substrate is of crucial importance to obtain homogeneous  $BaSi_2$  thin film. In fact, low substrate temperature and/or thick film deposition led to formation of rough film with voids, and the oxidation proceeded upon exposure to air. We revealed that appropriate choice of substrate temperature, film thickness, and post-growth *in-situ* annealing can provide enough diffusion of Si and Ba, leading to realization of homogeneous  $BaSi_2$  thin film.

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Keywords: barium desiliside, vacuum evaporation;

# 1. Introduction

Orthorhombic BaSi<sub>2</sub> is attractive material as an absorption layer for single junction thin-film solar cells because it has suitable band gap (1.3 eV [1,2]) and large absorption coefficients ( $3 \times 10^{-4}$  cm<sup>-1</sup> at 1.5 eV[1]). Furthermore, both

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of Ba and Si are earth-abundant and suitable for large-scale production. Fundamental studies clarified various promising properties to certify its potential as the absorption layer of thin film solar cells. These include long diffusion length (10  $\mu$ m [3]), long minority-carrier lifetime (14  $\mu$ s [4]), wide range control of electron and hole densities [5-7], application of MoO<sub>x</sub> hole transport layer to undoped n-type BaSi<sub>2</sub>[8], proposal of tin sulfide/BaSi<sub>2</sub> heterojunction solar cell to have a similar band diagram of pn homojunction [9], and so on.

The most of these fundamental studies have been done using  $BaSi_2$  epitaxial films grown on Si(111) or Si(100) substrates by molecular beam epitaxy [10-12]. As for practical application, development of alternative method to permit realization of large-scale homogeneous  $BaSi_2$  thin film is demanded. For this purpose, polycrystalline  $BaSi_2$  was fabricated on glass substrate by radio-frequency sputtering [13, 14]. In our previous study, we proposed to utilize vacuum evaporation method, which enables us to fabricate films simply, quickly and stably. In fact, we have shown that single-phase  $BaSi_2$  films can be formed on Si(111) substrate [15] and alkali-free glass substrate [16] by this method. In addition, the mechanism of film growth was discussed and we found that the vapor flux is Ba-rich and the supply of Si to the film is important to obtain homogeneous  $BaSi_2$ . Then, we speculated that high substrate temperature ( $\geq$ 500 °C) wound be necessary to fabricate  $BaSi_2$  film. However, high substrate temperatures set a limit to choice of the substrate. It would be useful if we could somehow decrease the substrate temperature based on fundamental understanding of formation mechanisms.

In this paper, we report on the impact of the film thickness, substrate temperature and post-growth annealing times on the formation of BaSi<sub>2</sub> thin film on Si substrate grown by vacuum evaporation using BaSi<sub>2</sub> granules as source materials. These three factors were found to affect the composition in the film, and the oxidation was found to occur if the diffusion of Ba and Si is not enough. We revealed that appropriate choice of substrate temperature, film thickness, and post-growth *in-situ* annealing is necessary to realize homogeneous BaSi<sub>2</sub> thin film.

#### 2. Experimental procedure

The films were deposited by vacuum evaporation under heating to form stoichiometric BaSi<sub>2</sub> film.  $20 \times 20$ -mm<sup>2</sup> Floating Zone n-type Si (111) ( $\rho$ >1000  $\Omega$ ·cm) wafers were used for substrates after cleaning with 1% HF solution to remove SiO<sub>2</sub> on the surface. BaSi<sub>2</sub> (99% in purity, Kojundo Chemical Lab.) granules were used as sources and melted by heating for approximately 2 minute. The film thickness was varied by changing the source weight. The distance between source and substrate was set as 19 cm. The base pressure was lower than  $1 \times 10^{-5}$  mbar.

The grown films were characterized by Raman spectroscopy (Nanofinder, Tokyo Instruments, Inc.) with  $Ar^+$  ion laser ( $\lambda$ = 488 nm), energy dispersive X-ray spectrometry (EDX), and scanning electron microscopy (SEM; JSM-7001FA, JEOL).

# 3. Results and discussion

Figure 1 shows cross-sectional SEM images of samples deposited under various conditions. Figures 1 (a) and (b) show the film deposited at 600 and 400 °C, respectively. Both of the films were fabricated by using the same weight of source. The film deposited at 600 °C is dense and the surface is flat. On the other hands, the film deposited at 400



Fig. 1. Cross-sectional SEM images of the film grown at (a) 600 and (b), (c) 600 °C.

 $^{\circ}$ C includes voids and the surface is very rough. Consequently, the film thickness is slightly thicker than that deposited at 600  $^{\circ}$ C. Figure 1 (c) shows the film deposited at 400  $^{\circ}$ C with smaller weight of source compared to the film shown in Fig. 1 (a) and (b). In spite of low growth temperature, the film morphology was improved.

Figure 2 shows Raman spectra of the films with various thicknesses deposited at 400 °C. For comparison, Raman spectrum of the 1300 nm thick film deposited at 600 °C is also shown. It is seen that all samples show peaks originating from phonons associated with  $BaSi_2$  in 200-500 cm<sup>-1</sup>[17]. Crystalline Si peak at 521 cm<sup>-1</sup> is seen only in the spectrum of the 1600 nm thick film deposited at 400 °C. This result indicates that the film contains crystalline Si. It is probably due to oxidation of the film upon expose to air to produce  $BaO_x$  and crystalline Si. Raman spectra of the 630 and 320 nm thick films contain broad peak around 400 to 550 cm<sup>-1</sup>, which suggests that amorphous Si is included in the films. Formation of amorphous Si would be due to the Si-rich vapor flux at the final stage of evaporation since the most of Ba are consumed at the earlier stage [15]. These results indicate that there is a limitation in the thickness for realization of homogeneous  $BaSi_2$  films by vacuum evaporation depending on the substrate temperature, and the critical thickness increases with increasing substrate temperature.

In order to clarify underling mechanisms, we used EDX measurement to compare the film composition. Figure 3 shows the oxygen content defined by the intensity ratio of O  $K\alpha$  peak to Ba  $L\alpha$  peak as a function of the film thickness. It is seen that the thick film deposited at low substrate temperature includes high amount of oxygen, which suggests that the film was oxidized upon exposure to air. On the other hand, oxidization can be suppressed by decreasing film thickness. This result indicates that homogeneous BaSi<sub>2</sub> films can be formed under conditions to permit sufficient diffusion of Ba and Si atoms. From these results, it is considered that film thickness and substrate temperature variation affect the diffusion length, and the diffusion of Ba and Si atom is one of the key factors to fabricate homogeneous BaSi<sub>2</sub> film.

To investigate the effect of annealing time to control the diffusion length, we prepared three samples at 400  $^{\circ}$ C. The samples are annealed *in-situ* after deposition in the vacuum chamber. Annealing time was set at 0, 30 and 60 minute. Figure 4 shows effect of annealing on the oxygen ratio and film thickness. It is found that oxidation ratio decrease with increasing annealing times accompanied by decrease of film thickness. This suggests that *in-situ* annealing may suppress oxidation by sufficient diffusion of Ba and Si atoms. However, reevaporation of the films occurs as evidenced by the decrease of film thickness.



From these results, the mechanism of film growth is discussed as follows. Inhomogeneous film composition is eliminated by diffusion assisted by thermal energy. Thus, when the thin films are deposited at low substrate temperature or the thick films are deposited at high substrate temperature, Ba and Si atoms can be diffused enough. However, when the thick films are deposited at low substrate temperature, remaining inhomogeneous part causes the oxidation. Annealing and high substrate temperature may enhance the diffusion distance. Therefore, if we need to fabricate homogeneous BaSi<sub>2</sub> films at lower temperature, it is required to reduce the film thickness or increase annealing time.

# 4. Conclusion

We fabricated the  $BaSi_2$  film under various conditions with vacuum evaporation and investigated the effect of substrate temperature, film thickness, and annealing times on the homogeneity of the film. We clarified the formation mechanism of homogeneous  $BaSi_2$  films, and diffusion of Ba and Si atoms was found to be important. When we need to fabricate homogeneous  $BaSi_2$  at low substrate temperature, thin film thickness or long time anneal is required to permit sufficient diffusion of Ba and Si atoms.

#### Acknowledgement

This work was supported by the Japan Science and Technology Agency (JST-CREST).

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