The Influences of Elastomer toward Crystallization of Poly(lactic acid)

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Abstract

Poly (lactic acid)/elastomer blends were prepared via direct injection molding. In non-isothermal crystallization scan, the crystallinity of PLA increased with a decrease in the heating and cooling rate. The melt crystallization of PLA appeared in the low cooling rate (1, 5 and 7.5 °C/min). The presence of elastomer tended also to increase the crystallinity of PLA. However, it started to decrease in 30% of elastomer. It was also showed by the decreasing of cold crystallization activation energy. Elastomer also gave plasticization effect in PLA properties. Thermal treatment through annealing completed after 1 h at 80 °C. In isothermal crystallization scan, the cold crystallization rate increased with increasing crystallization temperature in the blends. The Avrami analysis showed that at low temperatures, the cold crystallization had two regime processes whereas at high temperature only one stage was observed.

Keywords: Poly(lactic acid); elastomer; crystallinity; cold crystallization; crystallization kinetics.

1. Introduction

Recently, environmental issues have encouraged researchers from academia and industry to develop biodegradable polymer. Poly (lactic acid)/PLA, linear aliphatic polyester, has been well acknowledged as having a better potential of mechanical properties, biocompatibility and thermal plasticity. These properties were comparable to polystyrene (PS), polyethylene (PE) and polyethylene terephthalate (PET) [1]. PLA was synthesized from renewable resources and degraded to nontoxic of simple and natural product in a relative short time. Therefore, they were able to encourage its use in many applications [2-3].

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PLA has also drawbacks that limited its range of applications. Several methods have been developed to obtain desirable properties of PLA such as: high flexibility, ductility and high impact without altering its tensile properties. One of the most economical and practical strategies to tailor the properties of PLA is melt blending with another polymer. Although, copolymerization exhibited better performance than melt blending, it required more knowledge of reaction and polymerization during the process [4-5].

Elastomers are polymers which have low young modulus and high yield strain. If an amorphous polymer has a temperature glass transition (T_g) below room temperature, it will have elastomer properties, i.e. soft and rubbery. By introducing of elastomer via melt blending, a new class of PLA can be expected with enhanced elastic and ductile properties. In previous study, Zhang *et al.* [6] and Zaman *et al.* [7] reported the depletion of tensile modulus and enhancement of yield strain of PLA in the addition of elastomer through melt blending. However, they have not detailed the crystallization behavior of its attendant properties.

Considering the relationship between molecular structure and property, crystallinity reflected to mechanical and thermal properties [2,8]. Thus, understanding of crystallization kinetic is also important allowing for the design of materials with desirable properties. For example, the lower crystallization rate required longer time to obtain the certain crystallinity, the amorphous phase was obtained in short time. Liao *et al.* [9] concluded that crystallization rate decreased as increasing temperature in isothermal scan. Due to the most process were operated under non isothermal. The study of non-isothermal crystallization kinetic was also important [3].

In this investigation, details of crystallization behavior of PLA/elastomer are reported. PLA and elastomer is blended by injection molding directly. By using differential scanning calorimetry (DSC), dynamic mechanical analysis (DMA) and X-ray diffraction (XRD), the cold and melt crystallization phenomenon is elucidated. The cold crystallization during non-isothermal and isothermal mode is also described, including the effect of thermal treatment on cold crystallization.

2. Experiment

2.1. Materials

Poly(lactic acid)/PLA, commercial name "PLI003", was kindly supplied by Natureplast[TM] (France). It is an injection molding grade of PLA which has a density of 1.25 g/mL and melt flow index (MFI) of 12-20 g/10 min (190 °C/2.16 kg). Elastomer, commercial name "NPEL001", was also supplied by Natureplast[TM] (France). It has density of 1.23 g/mL and opaque. PLA was melt blended with elastomer via injection molding directly. Then, the PLA/elastomer samples were named as PLA, PLA10, PLA20 and PLA30 that refer to the composition of elastomer as follows: 0, 10, 20, and 30 wt%, respectively.

2.2. Melt blending

PLA and elastomer were previously dried in oven during 4 h at 80 °C. ASTM samples of the blends were prepared by an injection molding machine, Super Master Series SM120 made by Asian Plastic Machinery Co. Its inside has six controlling zone through the length of injection molding. The screw speed used (from feed to die zone) was 20, 20, 25, 30, 40 and 60 rpm while the temperature profile was 140, 160, 170, 190, 190 and 180 °C. During the process, the nozzle of injection molding was opened at 60% with pressure 30 bars. Cooling time and cycle time are 30 and 40 sec, respectively. Cold water as a quencher was at 10-11 °C.

2.3. Characterization

The thermal properties of PLA were performed by A Shimadzu DSC60 in non isothermal and isothermal scans, their procedures as follows:

Non isothermal scan. The procedure of this part was: first heating with a ramp of 10 °C/min from room temperature up to 200 °C and kept at this temperature for 5 min to discard any anterior thermal history, and cooled down at a ramp of 10 °C/min down to 25 °C, second heating scan from 25 to 300 °C with a ramp of 10 °C/min. The various ramps were employed at 1, 5, 7.5, 15 and 20 °C/min for all blend samples.

Isothermal scan. The procedure of this part was: first heating with a ramp of 10 °C/min from room temperature up to 200 °C and kept at this temperature for 5 min to discard any anterior thermal history and subsequently cooling at 10 °C/min to 25 °C. The second heating scan from 25 °C to various T_{iso} at 50 °C/min and held for 5 min, then heated at 10 °C/min to 250 °C to get the melt temperature.

The plasticization effect of elastomer was investigated by using a dynamic mechanical analyzer (AR G2, TA instrument). It was done within temperature range of 26-90 °C at a frequency 1Hz, strain 1% and heating rate 2 °C/min to determine temperature dependence of storage modulus (G') of the solid samples.

The degree of crystallinity was also evaluated by XRD analysis. It was performed on XMD 300 (Unisantis, Germany) with Cu-K_{α} radiation (wavelength, λ=1.5406 Å), operated at 45 kV and 0.8 mA. The scanning rate was operated at 0.16°/sec with angle scanning from 5 to 80°.

3. Results and Discussion

3.1. Non isothermal scan

It could be highlighted from Table 1. The melt crystallization tended to appear in cooling rate lower than 7.5 °C/min while the cold crystallization was absent in this rate. Melt crystallization temperature shifted to lower temperature when cooling rate was increased. It meant that PLA started to crystallize slower at high cooling rate. In other hand, cold crystallization temperature had a tendency to shift to higher temperature with increasing heating rate. As a whole, there was not enough time for PLA chains to rearrange into a crystalline structure in the fast cooling/heating rate. Thus, it was obtaining PLA more amorphous. Further, this statement was proved and discussed later.

By increasing the heating rate from 7.5 to 20 °C/min, the exothermic cold crystallization peaks became wider and shifted to a higher temperature. It indicated that there was not enough time during crystallization at high heating rate [3]. In the same heating rate, the cold crystallization temperature decreased significantly as increasing elastomer contents in PLA/elastomer blends. This result suggests that PLA and elastomer are thermodynamically compatible as reported by others [4,7].

The result data from DSC was also supported by DMA analysis, Figure 1. The storage modulus of all the samples is almost constant at low temperature. At certain temperature, the storage modulus decreased sharply and then increased again until a new relative constant value. The depletion of PLA modulus around 50 °C represents the glass transition temperature (T_g) while the onset of the modulus increase coincided with the onset of the cold crystallization [8].

Compared to neat PLA, the storage modulus of the PLA/elastomer blends gradually decreased as the elastomer content increased. A significant decrease occurred at PLA30. It indicated the plasticization

effect at the addition of elastomer. There was a similar result between DSC and DMA analysis about the depression of cold crystallization temperature due to an increase of elastomer in blends.

Table 1. Characteristics data of non isothermal scan for PLA and its blends

Sample	Θ °C/min	Melt crystallization		Cold crystallization		Melting		X_c %
		$T_{mc}\pm0.2\%$	$\Delta H_{mc}\pm0.15\%$	$T_{cc}\pm0.2\%$	$\Delta H_{cc}\pm0.15\%$	$T_m\pm0.2\%$	$\Delta H_m\pm0.15\%$	
PLA	1	104.4	23.19	-	-	167.6	-31.34	33.45
	5	91.2	20.12	95.4	24.72	169.0	-53.75	30.98
	7.5	62.3	10.60	98.9	29.05	169.2	-51.42	23.87
	10	-	-	102.9	29.79	170.1	-49.27	20.79
	15	-	-	106.2	27.64	170.9	-38.29	11.37
	20	-	-	113.9	29.25	172.5	-34.33	5.42
PLA10	1	112.3	33.77	-	-	166.1	-30.41	36.06
	5	95.6	32.22	-	-	167.1	-29.53	35.02
	7.5	91.0	10.87	87.1	8.12	168.3	-36.02	33.08
	10	-	-	93.5	20.03	168.9	-47.17	32.18
	15	-	-	101.2	19.03	169.3	-39.90	24.75
	20	-	-	104.8	27.82	170.9	-34.42	7.83
PLA20	1	110.0	35.15	-	-	168.7	-29.74	39.67
	5	93.2	33.65	-	-	166.0	-28.85	38.48
	7.5	88.2	15.49	81.1	9.21	167.5	-37.53	37.78
	10	-	-	90.0	15.58	167.8	-43.28	36.95
	15	-	-	94.8	13.24	169.1	-29.21	21.30
	20	-	-	99.4	18.19	169.6	-23.04	6.47
PLA30	1	102.3	20.60	-	-	160.4	-22.44	34.21
	5	88.8	19.53	-	-	162.1	-21.80	33.23
	7.5	62.7	12.14	71.8	6.89	163.4	-27.58	31.54
	10	-	-	81.3	13.02	164.5	-32.09	29.07
	15	-	-	85.8	9.04	164.9	-22.47	20.48
	20	-	-	90.2	16.04	165.1	-20.63	6.99

Θ = heating/cooling rate (°C/min); X_c = degree of crystallinity (%); ΔH = melting enthalpy (J/mol)
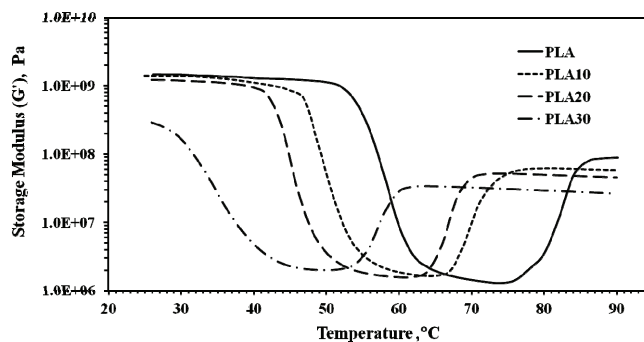

Fig. 1. Dynamic storage modulus of PLA and its blends as a function of temperature

Degree of crystallinity. The high crystallinity of PLA could be obtained in slow cooling/heating rate that giving enough time for PLA chains to rearrange its crystalline structure, Table 1. In the PLA30 blend, crystallinity of PLA started to decrease. It may be due to the addition of elastomer at 10 and 20% may act as diluent agent on the crystallization of PLA. However, at PLA30, the elastomer induced high levels of free volume and tended to inhibit the crystal formation of PLA [7,10]. In addition, the dispersion of

formed PLA crystals in blends could be also responsible for the decrease of crystallinity [6]. This result was also found in XRD scans as presented in Figure 2.
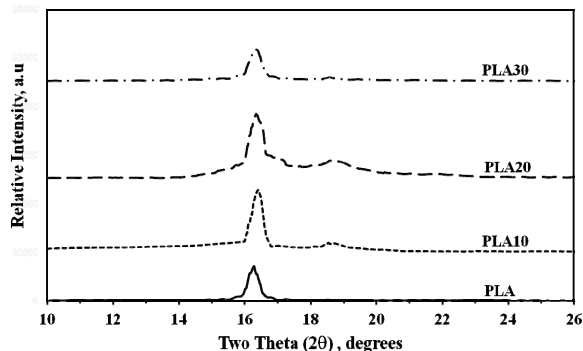

Fig. 2. Typical XRD scans of neat PLA and its blends

The degree of crystallinity calculated from DSC with normal ramp (10 °C/min) and XRD showed the different values, although it has same tendency. The values of crystallinity calculated from XRD were always higher that those calculated from DSC, Table 2. This phenomenon occurred due to difference scanned areas of the same sample [2]. XRD diffractograms just scanned the sample surface, while the DSC evaluated the bulk crystallinity. Therefore, the crystallinity values of XRD scan were strongly influenced by injection molding. It provoked morphology of surface and shear zones, thus leading to higher crystallinity, as a result of the orientation.

Table 2. The crystallinity of PLA obtained from both DSC and XRD scans

No	Sample	Degree of crystallinity, X_c (%)	
		DSC(10 °C/min)	XRD
1	PLA	20.79	24.35
2	PLA10	32.18	36.21
3	PLA20	36.95	40.54
4	PLA30	29.07	31.14

Cold crystallization activation energy (ΔE_{cc}). Kissinger's equation was the most widely used model to calculate this energy by considering the effect of the various heating rate in the non isothermal cold crystallization. Table 3 shows the presence of elastomer in blends tends to decrease the activation energy. Due to the low ΔE value exhibits the high crystallization ability [11]. It could be concluded that elastomer may give an extra movement ability of chain segments, thereby making it easier for crystallization.

Table 3. Calculation data of ΔE_{cc} by Kissinger's equation

No	Sample	ΔE_{cc}, J/mol
1	PLA	81643.5 (R^2=0.981)
2	PLA10	55254.8 (R^2=0.983)
3	PLA20	53176.3 (R^2=0.941)
4	PLA30	49468.3 (R^2=0.922)

Non isothermal cold crystallization kinetic. The kinetic of cold crystallization can be explained by making relationship between fractional crystallinity versus crystallization time. Quantification of non-isothermal crystallization rate could be proposed by half time of crystallization. The half time crystallization ($t_{1/2}$) was obtained directly by tracing the fractional crystallinity of 0.5 to required time and tabulated in Table 4. The crystallization time becomes longer as decreasing the heating rate. It indicated that the lower heating rate promoted crystallization by given an appropriate crystallization time and were proved that higher crystallinity can be obtained in lower heating rate, Table 1. However, crystallization rate was also lower than that of higher heating rate. The same tendency was also obtained for PLA/elastomer blends.

Table 4. The half time crystallization ($t_{1/2}$) at different heating rate

Sample	Half time crystallization ($t_{1/2}$), minutes			
	7.5 °C/min	10 °C/min	15 °C/min	20 °C/min
PLA	1.56	1.13	0.86	0.76
PLA10	2.01	1.45	1.02	0.84
PLA20	1.71	1.24	0.93	0.75
PLA30	1.22	1.03	0.85	0.68

Effect of annealing on cold crystallization. Annealing is a thermal treatment process at higher temperature than glass temperature transition (T_g). It was intended to recovery of crystals that were lost during injection molding process. Figure 3 shows the different annealing time of PLA at 80 °C. Clearly, the longer annealing time decreased gradually the enthalpy of cold crystallization (ΔH_{cc}). The cold crystallization peak did not appear at annealing time of 1 h. It meant that rearranging of crystalline structure of PLA would be completed after one hour at 80 °C.
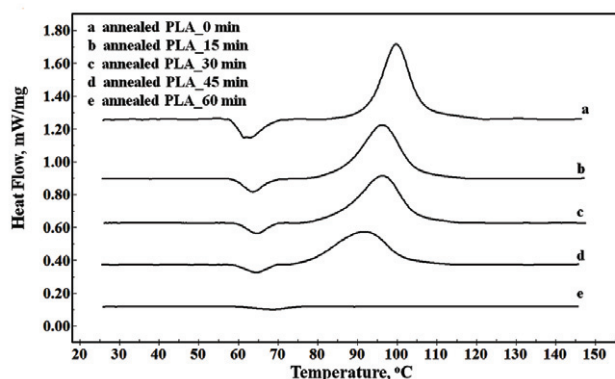

Fig. 3. The heating DSC scans (10 °C/min) of neat PLA with various annealing time at 80 °C

3.2. Isothermal scan

Evaluation of isothermal cold crystallization kinetic was presented in Figure 4a. At fractional crystallinity of 0.9, the cold crystallization rate decreased significantly. In this condition, the crystallite collision might detain the crystal growth. The different T_{cc} did not change the cold crystallization mechanism that illustrated by similar shapes of the curves in Figure 4a. From the half time of crystallization ($t_{1/2}$) as tabulated in Table 5, the presence of elastomer in PLA blends decreased the value

of $t_{1/2}$ at the same T_{cc}. It meant that elastomer may act as diluent which enhancing the crystallization rate of PLA [10]. Besides that, elastomer shifted down the crystallization peak to the lower temperature.
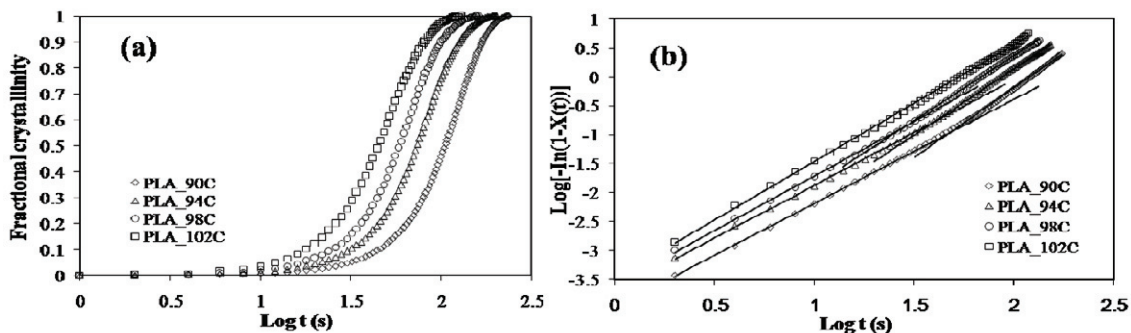

Fig. 4. Isothermal cold crystallization of PLA at different temperature: (a) fractional crystallinity vs. time and (b) Avrami plots

The Avrami model shows the powerful model ($R^2 = 0.99$) to describe an isothermal crystallization kinetic by assuming a constant growth and nucleation rate in random nucleation, Figure 4b. There were two stages of cold crystallization of poly(lactic acid) and its blends. It was clearly visible when lower temperature was employed during isothermal cold crystallization.

Table 5. Kinetics parameters obtained from isothermal cold crystallization experiments and Avrami analysis

Sample	T_{cc} (°C)	n	K (s⁻¹).10⁻³	$t^{1/2}$ (s)
PLA	90	1.80	0.104	132.961
		2.39[a]	**0.010**	**104.615**
	94	1.83	0.202	86.264
		2.30	**0.035**	**73.768**
	98	1.88	0.256	66.685
		2.24	**0.075**	**58.400**
	102	1.99	0.376	43.847
PLA10	82	1.77	0.118	135.222
		2.71	**0.002**	**122.262**
	86	1.84	0.164	92.758
		2.13	**0.072**	**73.949**
	90	1.88	0.431	50.682
		2.54	**0.037**	**47.989**
	94	1.99	0.998	26.884
PLA20	78	1.59	0.398	108.886
		2.35	**0.018**	**90.285**
	82	1.61	0.984	59.175
		2.09	**0.237**	**45.999**
	86	1.89	1.230	28.795
	90	2.00	2.636	16.170
PLA30	74	1.77	0.387	68.498
		2.69	**0.015**	**53.721**
	78	1.83	0.855	38.940
		2.21	**0.305**	**33.102**
	82	1.85	2.393	21.522
	86	1.87	4.266	15.192

[a]Bold number = second stage of crystallization

The Avrami exponent (n) at first stage was varying from 1.59 to 2 which described a one-dimensional crystallization growth with homogeneous nucleation, whereas at second stage was varying from 2.09 to 2.71 which described the transitional mechanism to three dimensional crystallization growth with heterogeneous nucleation mechanism [11]. The first and second stage referred to primary and secondary crystallization stage. The *n* value of PLA and PLA/elastomer blends at the same temperature are almost same. Indicating the presence of elastomer did almost not influence the nucleation mechanism.

Considering the *K* value, crystallization rate of PLA could be enhanced by increasing cold crystallization temperature (T_{cc}) and introducing elastomer in PLA blends. Comparison the first and second stage crystallization in lower temperature for each sample proved the previous statement. In the second stage, crystallization rate was detained by the present of crystallite collision. It was described by lower *K* value than that in the first stage.

4. Conclusion

There were two kind of crystallizations appeared in PLA during isothermal scanning, i.e. melt crystallization and cold crystallization. The appeared crystallization was only melt crystallization at low heating/cooling rate, indicating there was enough time to rearrange its crystalline structure. In other hand, the cold crystallization tended to appear at higher heating/cooling rate, indicating more amorphous structure was obtained. The presence of elastomer tended to increase the crystallinity of PLA at low composition as well as its cold crystallization kinetics during non-isothermal scan. The cold crystallization peaks shifted down to lower temperature which addressed by decreasing their activation energy. The enhancement of crystallization rate could be done by increasing heating/cooling rate. Thermal treatment through annealing completed after 1 h at 80 °C. Nucleation phenomenon of PLA was almost not influenced by presence of elastomer during isothermal mode. The higher crystallization rate could be obtained by increasing crystallization temperature and introducing elastomer in PLA blends.

Acknowledgement

References

[1]   Auras R., Harte B., Selke S., *Macromol. Biosci*. 2004; **4**: 835-864.

[2]   Carrasco F., Pagès P., Gámez-Pérez J., Santana O.O., Maspoch M.L., *Polym. Degrad. Stability* 2010; **95**: 116-125.

[3]   Zhao Y., Qiu Z., Yan S., Yang W., *Polym. Eng. Sci.*, (2011); **51**: 1564-1573.

[4]   Ali F., Chang Y.-W., Kang S.C., Yoon J.Y., *Polym. Bull.* 2009; **62**: 91-98.

[5]   Rasal R.M., Janorkar A.V., Hirt D.E., *Prog. Polym. Sci.* 2010; **35**: 338-356.

[6]   Zhang W., Chen L., Zhang Y., *Polymer* 2009; **50**: 1311-1315.

[7]   Zaman H., Song J., Park L.S., Kang I.K., Park S.Y., Kwak G., Park B.S., Yoon K.B., *Polym. Bull.* 2011; **67**: 187-198.

[8]   Suryanegara L., Nakagaito A.N., Yano H., *Compos. Sci. Technol.* 2009; **69**: 1187-1192.

[9]   Liao R., Yang B., Yu W., Zhou C., *J. Appl. Polym. Sci.* 2007; **104**: 310-317.

[10] Chen J.H., Tsai F.C., Nien Y.H., Yeh P.H., *Polymer* 2005; **46**: 5680-5688.

[11] Liu Q., Peng Z., Chen D., *Polym. Eng. Sci.* 2007; **47**: 460-466.