

# Acetone sensor based on solvothermally prepared ZnO doped with Co<sub>3</sub>O<sub>4</sub> nanorods

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**Abstract** This paper describes a reliable and sensitive method for sensing dissolved acetone using doped nanomaterials. Large-scale synthesis of ZnO nanorods (NRs) doped with Co<sub>3</sub>O<sub>4</sub> was accomplished by a solvothermal method at low temperature. The doped NRs were characterized in terms of their morphological, structural, and optical properties by using field-emission scanning electron microscopy coupled with energy-dispersive system, UV-Vis., Fourier transform IR, X-ray diffraction, and X-ray photoelectron spectroscopy. The calcinated (at 400 °C) doped NRs are shown to be an attractive semiconductor nanomaterial for detecting acetone in aqueous solution using silver electrodes. The sensor exhibits excellent sensitivity, stability and reproducibility. The calibration plot is linear over a large concentration range (66.8 μM to 0.133 mM), displays high sensitivity (~3.58 μA cm<sup>-2</sup> mM<sup>-1</sup>) and a low detection limit (~14.7±0.2 μM; at SNR of 3).

**Keywords** ZnO doped Co<sub>3</sub>O<sub>4</sub> nanorods · Solvothermal method · Acetone determination · I-V nanomaterials

## Introduction

Doped nanomaterial has been recognized as a promising host for transition material revealing a stable morphological structure and is composed of a number of irregular phases with geometrically coordinated metal and oxide atoms stacked alternately along the axes [1]. Transition metals doped-semiconductor nanomaterials have been used in insightful research efforts due to their exceptional optical and structural properties as well as versatile applications [2]. Recently, extensive developments have been made in the research leading to ZnO-based doped nanomaterials initiated by both fundamental sciences and advanced technologies [3]. This doped nanostructure material exhibits promising uses as field consequence transistors, UV photo-detectors, gas-sensors, field-emission electron sources, nanomaterials, nanoscale power generators, and many other functional devices [4–10]. Preparation of doped material is also a competent method to regulate the energy-level surface states of zinc oxide, which can further progress by changing the doping concentration of semiconductor materials. As one of the most interesting magnetic p-type doping materials, cobalt-oxide nanostructure is also recognized as an attractive material with broad applications in various fields, such as doping, catalysts, solid-state sensors, and electrochemical devices [11–13]. Undoped zinc oxide semiconductor material (band-gap~3.37 eV) is a substantial functional nanomaterial applied in various fields, such as solar cells, photocatalysis, optoelectronic, electrical, electromagnetic shielding, bio- and chemo-sensors [14]. It has gained much interest owing to proficient synthesis, easy handling, economical way of preparation, reproducibility and applications as a gas-sensor for the detection of toxic materials such as

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hydrogen, ammonia, liquid petroleum gas, ethanol, formaldehyde, ammonium hydroxide, alcohols, and phenol [15, 16].

Semiconductor nanomaterials have attracted considerable interest owing to their extrinsic and remarkable properties in the electrical, optical, thermal, and mechanical fields as compared to their undoped materials. It is essential for the preparation of nanomaterials to achieve exceptional quality of doped semiconductor structures. To this end an extensive range of chemical and physical methods has been used for the preparation of  $\text{Co}_3\text{O}_4$  nanomaterials, including polymer ignition, gel-technology, co-precipitation, ball-milling, chemical vapor deposition and others [17–21]. Co-doped nanomaterials uncovered several applications because of their attractive properties resulting from the changeable oxidation state of the transition-semiconductor metals. The impressive nano-structural morphology of transition metal oxides is the amendment of magnetic and electrical properties. It is well recognized that  $\text{Co}_3\text{O}_4$  has an extraordinary nano-structure, and doped  $\text{Co}_3\text{O}_4$  exhibits anti-ferromagnetism at lower temperatures [22]. In industries and research laboratories, highly volatile and toxic acetone is frequently used. It is harmful to human health and living organisms in health care and environmental fields. Exposure to air having high concentrations of acetone vapor may cause headache, allergy, fatigue, and even narcosis [23]. For diabetes mellitus patients, it is very harmful to inhale the toxic acetone, thus urgently requiring to investigate the sugar-level regularly [24]. Hence early detection and quantification of acetone in the environment is necessary and important for our safety in the research laboratory and industrial premises. Recently, metal oxides have been exploited as artificial mediators for the development of reliable and efficient chemo-sensors [25]. Therefore doped semiconductor materials have been utilized as a redox mediator for the detection and quantification of acetone using reliable electrochemical approaches at room temperature.

Doped metal oxides are also the model materials for sensing due to their highly active surface areas, and they are extensively employed as chemosensor for the detection, recognition, and quantification of various carcinogenic pollutants as well as toxic chemicals or biomolecules [26]. In the presence of organic volatile chemicals, acetone causes damage to the brain and specific diseases of the stomach, liver, and erythrocytes. Consequently it is a significant and big challenge to detect acetone proficiently and hence protect human health from insidious diseases and save the environment using analytical techniques [27]. A large amount of metal oxides have been considered as chemosensors for the detection of different hazardous pollutants and toxic chemicals [28, 29]. Therefore ZnO-doped  $\text{Co}_3\text{O}_4$  NRs have been offered as mediator to detect and detoxify the organic acetone chemical. The main focus of this present investigation is to fabricate and develop a highly sensitive chemical sensor for the detection and

quantification of hazardous pollutants using NR semiconductor nanomaterials. Doped nanomaterials are extensively employed for the detection of hazardous molecules in chemical control process due to their numerous benefits over conventional chemical analysis in terms of response, large-surface area, portability, and monitoring of toxic chemical in the environmental field. Conventional electrochemical methods with uncoated nanomaterial electrodes for acetone detection exhibit a slower-response, surface-fouling, unstable signals, noise, and lower sensitivity. Hence, modification of the sensor surface with doped nanomaterials is very important to achieve more sensitive, repeatable, and stable responses. Therefore, a simple and reliable I-V electrochemical approach paired with relatively easy, convenient and inexpensive instrumentation is required to obtain higher sensitivity and lower detection limits compared to conventional methods. Here, we introduced a very reliable, large-scale, highly sensitive technique for the electrochemical detection of acetone using ZnO-doped  $\text{Co}_3\text{O}_4$  NRs at room temperature. The present approach depicts a simple, highly sensitive, low-sample volume, easy to handle, and precise I-V technique.

## Experimental

### Chemical reagents and apparatus

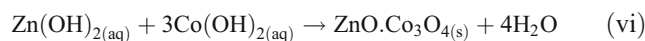
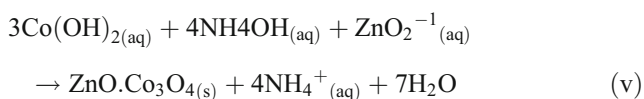
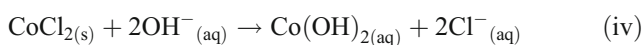
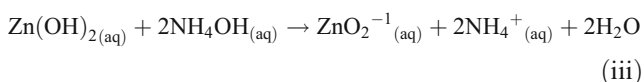
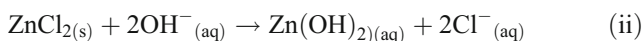
Zinc chloride ( $\text{ZnCl}_2$ ), cobalt chloride ( $\text{CoCl}_2$ ), ethyl acetate, disodium phosphate, butyl carbitol acetate, monosodium phosphate, and all the other chemicals used were of analytical grade and obtained from Sigma-Aldrich Company (<http://www.sigmaaldrich.com>). 0.1 M phosphate buffer solution at pH 7.0 was prepared by mixing equimolar concentrations of 0.2 M  $\text{Na}_2\text{HPO}_4$  and 0.2 M  $\text{NaH}_2\text{PO}_4$  solution in 100.0 mL deionized water at room temperature. The  $\lambda_{\text{max}}$  (397.0 nm) of ZnO-doped  $\text{Co}_3\text{O}_4$  nanorods was evaluated with UV/visible spectroscopy (UV/Vis., Lamda-950, Perkin Elmer, Germany; <http://www.perkinelmer.com>). Fourier-transform infrared spectroscopy (FT-IR spectra) was performed with a spectrophotometer (Spectrum-100 FT-IR) in the mid-IR range, which was purchased from Perkin Elmer, Germany (<http://www.perkinelmer.com>). A Raman station 400 (Perkin Elmer, Germany; <http://www.perkinelmer.com>) was used to measure the Raman shift of doped nanorod materials using radiation source ( $\text{Ar}^+$  laser line,  $\lambda$ ; 513.4 nm). The powder X-ray diffraction (XRD) prototypes were assessed with an X-ray diffractometer (XRD; X'Pert Explorer, PANalytical diffractometer; [www.panalytical.com](http://www.panalytical.com)) equipped with  $\text{Cu-K}_{\alpha 1}$  radiation ( $\lambda=1.5406$  nm) using a generator voltage of 40.0 kV and a generator current of 35.0 mA applied for the purpose. The XPS measurements were executed on a Thermo Scientific K-Alpha KA1066 spectrometer (Germany). Monochromatic  $\text{AlK}\alpha$  x-ray radiation sources were used as excitation sources,

where the beam-spot size was kept at 300.0 μm. The spectra were recorded in the fixed analyzer transmission mode, where the pass energy was kept at 200 eV. Scanning of the spectra was performed at lower pressures (<10<sup>-8</sup> Torr). The morphology of doped nanomaterial was investigated by field-emission scanning electron microscopy (FESEM; JSM-7600F, Japan; [www.jeol.com](http://www.jeol.com)). I-V technique was used for the investigation of acetone using an Electrometer (Kethley, 6517A, Electrometer, USA; [www.keithley.com](http://www.keithley.com)) at room temperature.

#### Synthesis and growth mechanism of ZnO-doped Co<sub>3</sub>O<sub>4</sub> NRs

Initially, cobalt chloride (0.1 M) and zinc chloride (0.1 M) solution were prepared separately with de-ionized water at room temperature. The equimolar solution of both reactants was mixed gently at 1:1 ratio. Then the reducing agent (i.e., NH<sub>4</sub>OH) was added dropwise into the reactant mixture to produce a white precipitate. The resultant solution was adjusted to pH~10.15 using NH<sub>4</sub>OH solution. The mixture was transferred into a hydrothermal cell (Teflon autoclave) and placed in the oven at a temperature of about 150.0 °C for 16 h (active solution temperature, ~92.0 °C). Then the solution was left for cooling down to room temperature for several hours. The precipitate was separated and consecutively washed thoroughly with ethanol, acetone and water. Finally, the grown ZnO-doped Co<sub>3</sub>O<sub>4</sub> NRs were placed for calcinations into the furnace at 400.0 °C for 5 h (Barnstead Thermolyne, 6000 Furnace, USA). The calcinated products were characterized in detail in terms of their morphological, structural, optical properties, and applied for acetone drug detection.

The growth mechanism of the doped nanomaterials could be explained on the basis of chemical reaction and nucleation as well as the development of host nanocrystals. The apparent reaction mechanism leading to the doped nanomaterials is anticipated as follows:



The precursors of CoCl<sub>2</sub> and ZnCl<sub>2</sub> are reacted in the alkaline medium according to the Eqs. (i) to (iv). After addition of reducing agent (NH<sub>4</sub>OH) into the mixture of reactant solution, the reaction is slowly progressed to growth the co-doped nanomaterials according to the Eqs. (v)–(vi). During the total synthesis process, NH<sub>4</sub>OH acts as a pH buffer to regulate the pH value of the solution and slow release of OH<sup>-</sup> ions [30]. When the concentration of Zn<sup>2+</sup> and OH<sup>-</sup> ions exceeded a critical value, the precipitation of ZnO nuclei begins to start. As there is a high concentration of Co<sup>2+</sup> ions in the reaction medium, the nucleation of ZnO crystals is also started to co-precipitate, owing to the lower activation energy barrier of heterogeneous nucleation. However, due to the low concentration of Co<sup>2+</sup> ions in the reaction medium, the larger ZnO crystals with a rod-like morphology are developed among the nanostructures composed of ZnO co-doped Co<sub>3</sub>O<sub>4</sub> NRs. The shape of the NRs is approximately consistent with the growth habit of transition-metal doped ZnO crystals [31, 32].

#### Fabrication of the sensor and detection technique for acetone

The silver electrode (AgE) is prepared with ZnO-doped Co<sub>3</sub>O<sub>4</sub> NRs, where butyl carbitol acetate (BCA) and ethyl acetate (EA) act as conducting binders. A small amount (0.1 μg each) of mixture (EA, BCA, and ZnO-doped Co<sub>3</sub>O<sub>4</sub>) is uniformly pasted onto the AgE surfaces, and then placed into the oven at 70.0 °C for 12 h until the film is completely dry. An electrochemical cell is mounted with doped nanomaterials coated AgE as a working electrode and Pd wire is used as counter-electrode. Stock solution of acetone (1.33 M) is diluted to various concentrations and used as a target. An amount of 0.1 M phosphate buffer solution is kept constant in the beaker as 10.0 mL throughout the chemical investigation. The acetone analyte solution is made with various concentrations in the range of 66.8 μM to 1.33 M. The sensitivity is calculated from the slope of current vs. concentration (I vs. C) from the calibration plot divided by the value of active-surface area of fabricated AgE electrodes. An electrometer is used as a voltage source for I-V technique in two electrode systems.

## Results and discussion

### Investigation of optical properties

The optical property of the ZnO-doped Co<sub>3</sub>O<sub>4</sub>NRs is one of the significant characteristics for the assessment of its photocatalytic activity. The optical absorption spectra of ZnO co-

doped  $\text{Co}_3\text{O}_4$  NRs are measured by using a UV-Vis spectrophotometer in the visible range (200.0 to 800.0 nm). UV/visible absorption is a technique in which the outer electrons of atoms or molecules absorb radiant energy and undergo transitions to high energy levels. In this phenomenon, the spectrum obtained due to optical absorption can be analyzed to acquire the energy band gap of the semiconductor. The optical absorption measurement was carried out at ambient conditions. From the absorption spectrum, the absorbance of the ZnO-doped  $\text{Co}_3\text{O}_4$  NRs was measured and found to be about 397.0 nm, as shown in Fig. 1a. The band gap energy ( $E_{\text{bg}}$ ) is calculated on the basis of the maximum absorption band of NRs and was found to be 3.1234 eV, according to the following Eq. (vii).

$$E_{\text{bg}} = \frac{1240}{\lambda} \text{ (eV)} \quad (\text{vii})$$

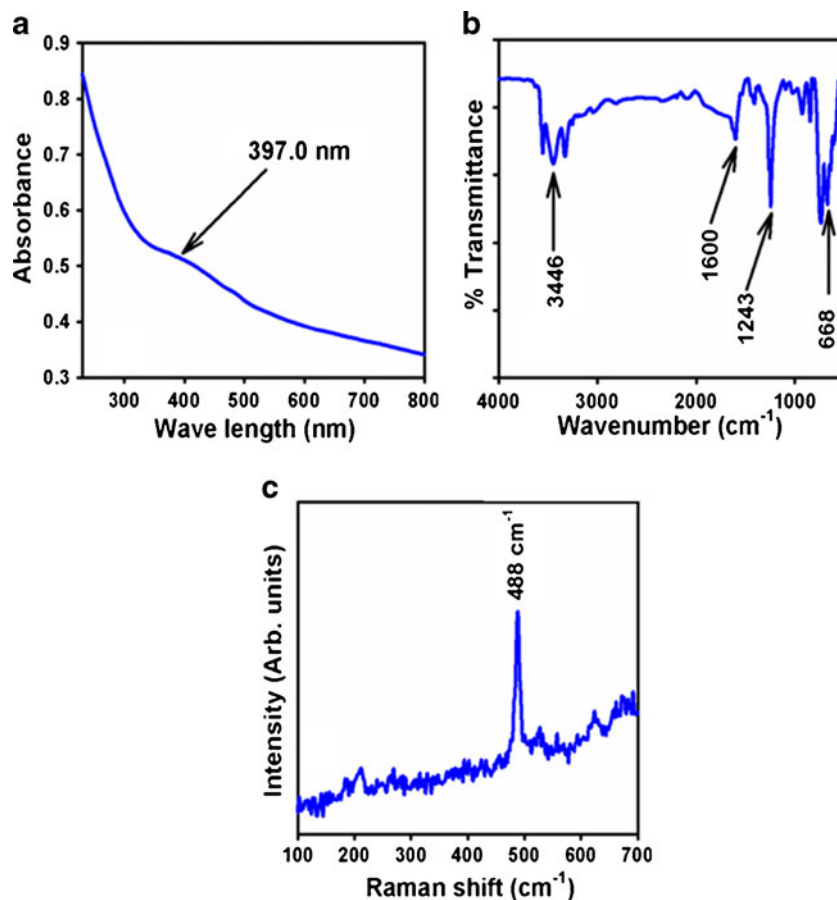
where  $E_{\text{bg}}$  is the band-gap energy, and  $\lambda_{\text{max}}$  is the wavelength (397.0 nm) of the NRs. No extra peak associated with impurities and structural defects were observed in the spectrums, which proved that the synthesized NRs control the crystallinity of ZnO-doped  $\text{Co}_3\text{O}_4$  nanomaterials [33].

The ZnO-doped  $\text{Co}_3\text{O}_4$  NRs were also studied in terms of atomic and molecular vibrations. To predict the motivated

identification, FT-IR spectra with fundamental absorption in the range of 400–4,000  $\text{cm}^{-1}$  are investigated at room temperature. Figure 1b displays the FT-IR spectrum of the ZnO-doped  $\text{Co}_3\text{O}_4$  nanostructures. It represents bands at 668, 1,243, 1,600, and 3,446  $\text{cm}^{-1}$ . This broad vibration band (at 668  $\text{cm}^{-1}$ ) could be assigned to metal-oxygen (Co–O and Zn–O modes) stretching vibrations [34], which demonstrated the configuration of doped nanostructure materials. The supplementary vibrational bands may be assigned to O–H bending vibration (1,600  $\text{cm}^{-1}$ ), C–O absorption (1,243  $\text{cm}^{-1}$ ), and O–H stretching (3,446  $\text{cm}^{-1}$ ). The absorption bands at 1,243, 1,600, and 3,446  $\text{cm}^{-1}$  usually results from  $\text{CO}_2$  and water, which due to their high surface-to-volume ratio of mesoporous nature are absorbed by semiconductor nanostructure materials from the environment [35]. Finally, the experimental vibration bands in low frequency regions suggested the formation of ZnO-doped  $\text{Co}_3\text{O}_4$  NRs.

Raman spectroscopy is a spectroscopic technique used to disclose vibrational, rotational, and other low-frequency phases in Raman active compounds. It is based on inelastic scattering of monochromatic light (called Raman scattering), usually from a laser in the visible, near-infrared, or near-ultraviolet range. The laser light interacts with molecular vibrations, which caused an upward or downward shift of

**Fig. 1** Typical **a** UV–vis spectrum, **b** FTIR spectrum, and **c** Raman spectrum of ZnO-doped  $\text{Co}_3\text{O}_4$  NRs



the energy of the laser photons. Raman spectroscopy is usually recognized and utilized in material chemistry, since the information is explicit to the chemical bonds and symmetry of metal–oxygen stretching or vibrational modes. Figure 1c confirms the Raman spectrum where key aspects of the wave number are measured at  $488\text{ cm}^{-1}$  for metal–oxygen (Co–O and Zn–O) stretching vibrations. These bands can be assigned to a cubic phase of ZnO-doped Co<sub>3</sub>O<sub>4</sub> NRs [36].

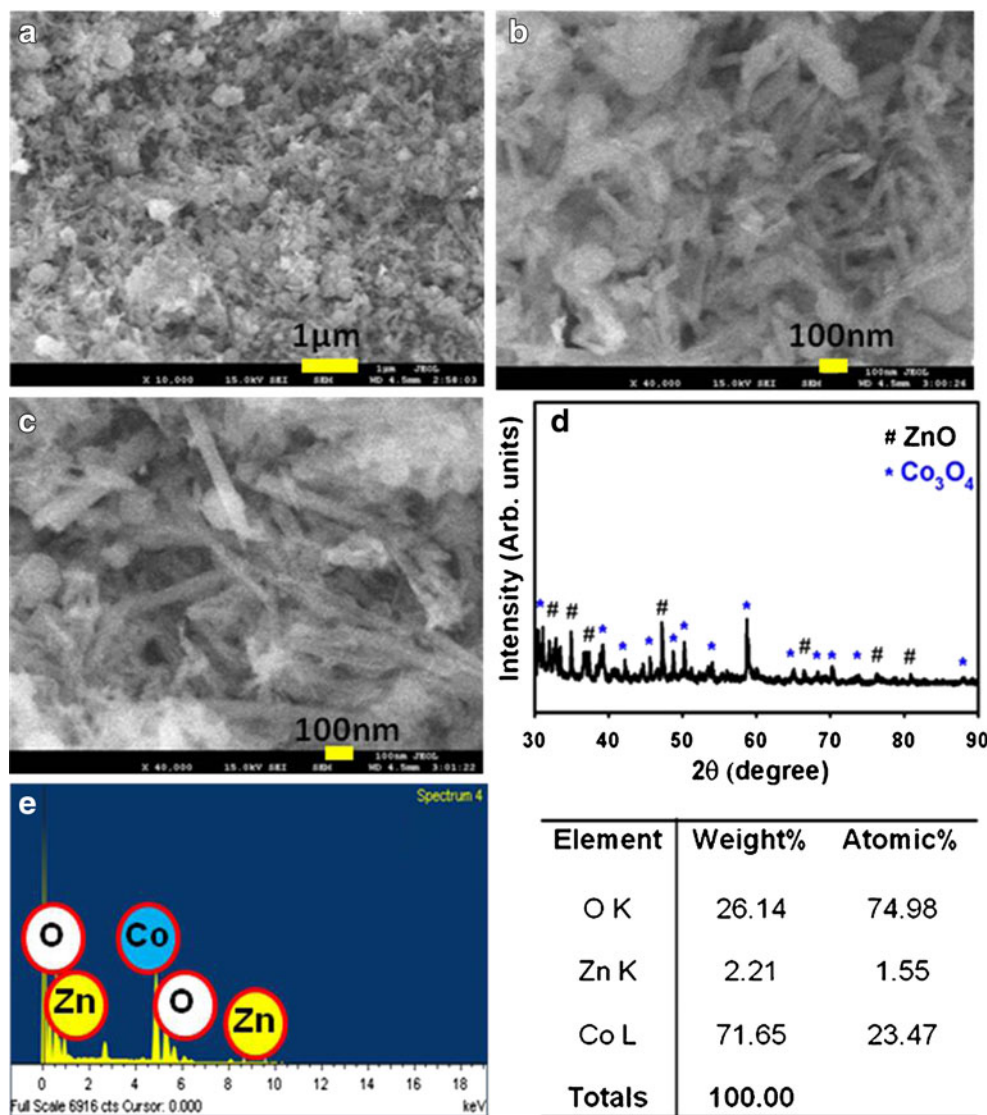
Investigation of morphological, elemental, and structural properties

High resolution FESEM images of ZnO-doped Co<sub>3</sub>O<sub>4</sub> NRs are shown in Fig. 2a to c (magnified). The images are composed of nanostructure materials with aggregated nanostructure in rod shapes. The average diameter is calculated in the range of 35.15 nm to 51.23 nm, which is close to  $42.98 \pm 5.0$  nm. It is shown clearly from the FESEM images

that the simple hydrothermal methodology of synthesized doped products creates the nanostructure of ZnO-co-doped Co<sub>3</sub>O<sub>4</sub>, which flourished to form an extraordinarily rod-shape of high-density.

The crystallinity and crystal phases of the ZnO-doped Co<sub>3</sub>O<sub>4</sub> NRs were examined. The X-ray diffraction patterns of doped NRs are shown in Fig. 2. The ZnO-doped Co<sub>3</sub>O<sub>4</sub> NRs were investigated and shown as face-centered cubic. The as-grown sample was heated in a furnace at  $400.0\text{ }^\circ\text{C}$  to start the formation of nano-crystalline doped phases. Figure 2d reveals characteristic crystallinity of the combined ZnO-doped Co<sub>3</sub>O<sub>4</sub> NRs and their aggregative crystal arrangement. The reflection peaks in this prototype were initiated to correspond with the Co<sub>3</sub>O<sub>4</sub> phase (star symbol, \*) having face-centered-cubic geometry [Joint Committee on Powder Diffraction Standards (JCPDS) # 078-1970]. The phases demonstrated the key feature peaks (blue-color with \*) with indices for crystalline Co<sub>3</sub>O<sub>4</sub> at  $2\theta$  values of  $31.27(220)$ ,  $38$ .

**Fig. 2** a–c low to high magnified FE-SEM images, d powder X-ray diffraction pattern, and e EDS investigation of ZnO-doped Co<sub>3</sub>O<sub>4</sub> NRs



54(222), 44.80(400), 49.08(331), 55.65(422), 59.35(511), 65.22(440), 68.62(531), and 74.11(620) degrees. The face-centered-cubic lattice parameters are ( $a$ , 8.085), point group (Fd-3m), and radiation ( $\text{CuK}_{\alpha 1}$ ,  $\lambda=1.5406$ ). This confirmed that there is a major number and amount of crystalline  $\text{Co}_3\text{O}_4$  present in doped nanomaterials. Beside this, all the reflection peaks in this pattern were also found to match with the ZnO phase (Zincite) having hexagonal geometry [JCPDS # 074-9942]. The phases showed the major characteristic peaks (black-color, hash symbol, #) with indices for crystalline ZnO at  $2\theta$  values of 35.01(002), 36.8(101), 48.32(102), 67.43(200), 78.3(202), and 82.99(104) degrees. The hexagonal (unit cell) lattice parameters are  $a=3.2049$ ;  $c=5.1216$ ;  $Z=2$ , point group: P63mc (186), and radiation:  $\text{CuK}_{\alpha 1}$  ( $\lambda=1.5406$ ). These indicate that there is a considerable amount of crystalline ZnO present in doped nanostructured materials. Finally, this x-ray pattern corresponds to the ZnO-doped  $\text{Co}_3\text{O}_4$  NRs sample, which may be attributed to the doping of  $\text{Co}_3\text{O}_4$  and the ZnO lattice site of doped aggregated NRs semiconductor materials [37]. Furthermore, no other impurity peak was observed in the XRD pattern showing the ZnO-doped  $\text{Co}_3\text{O}_4$  phase formation. The electron dispersive spectroscopy (EDS) investigation of ZnO-doped  $\text{Co}_3\text{O}_4$  NRs indicates the presence of Zn, Co, and O composition in the pure doped material. It is clearly shown that synthesized materials controlled only zinc, cobalt, and oxygen elements, which are presented in Fig. 2e. The composition of zinc, cobalt, and oxygen is 2.21 %, 71.65 %, and 26.14 %, respectively. No other peak related with any impurity has been detected in the EDS, which confirms that the doped products are composed only with zinc, cobalt, and oxygen.

X-ray photoelectron spectroscopy (XPS) is a quantitative spectroscopic method that determines the chemical-states of the elements that are present within co-doped materials. XPS spectra are acquired by irradiating on a nanomaterial with a beam of X-rays, while simultaneously determining the kinetic energy and number of electrons that get away from the top one to ten nm of the material being analyzed. Here, XPS measurements were measured for ZnO-doped  $\text{Co}_3\text{O}_4$  NRs to investigate the chemical states of  $\text{Co}_3\text{O}_4$  and ZnO. The XPS spectra of Co2p, Zn2p, and O1s are presented in Fig. 3a. Figure 3b presents the XPS spectra (spin orbit doublet peaks) of the  $\text{Co}2p_{(3/2)}$  and  $\text{Co}2p_{(1/2)}$  regions recorded with semiconductor co-doped nanomaterials. The binding energy of the  $\text{Co}2p_{(3/2)}$  and  $\text{Co}2p_{(1/2)}$  peak at 788.3. eV and 804.7 eV respectively denotes the presence of  $\text{Co}_3\text{O}_4$  since their binding energies are similar to the reported work [38]. In Fig. 3c, the spin orbit peaks of the  $\text{Zn}2p_{(3/2)}$  and  $\text{Zn}2p_{(1/2)}$  binding energy for co-doped nanomaterials appeared at around 1050.8 eV and 1028.2 eV respectively, which is in good agreement with the reference data for ZnO [39]. The O1s spectrum shows a main peak at 532.1 eV in Fig. 3d. The peak at 532.1 eV is assigned to lattice oxygen may indicate oxygen

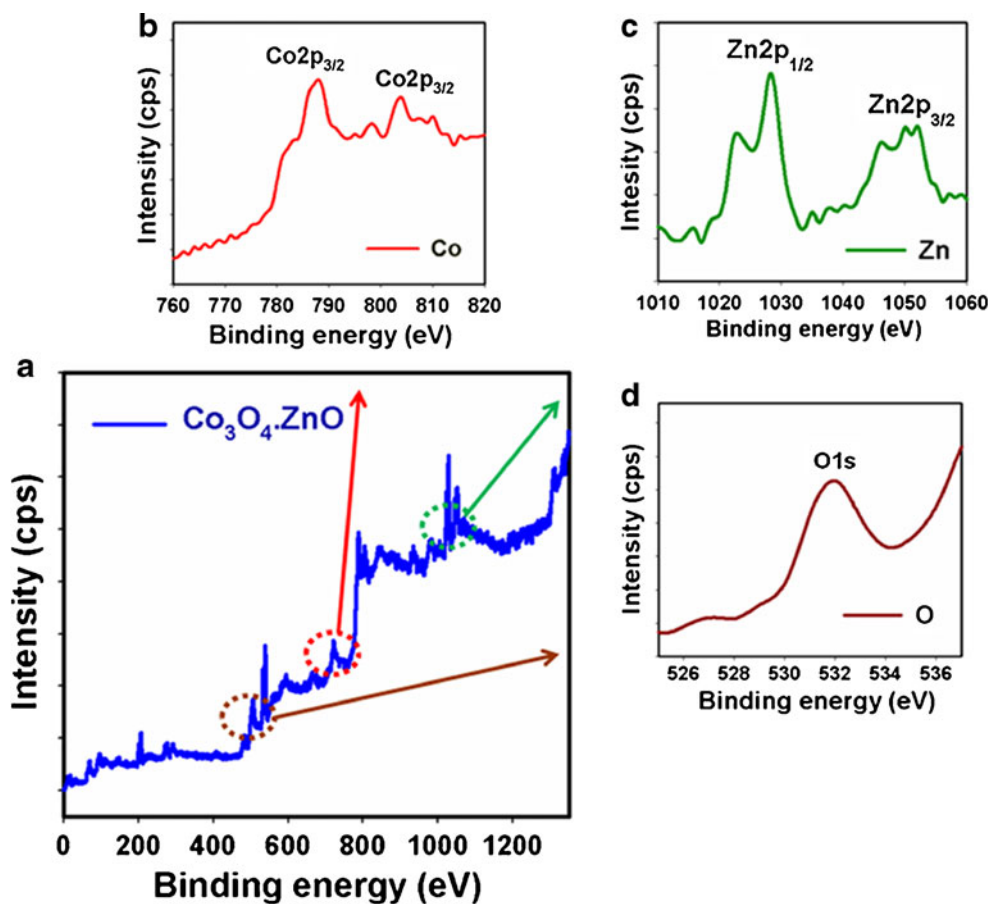
(i.e.,  $\text{O}_2^-$ ) presence in the co-doped nanomaterials [40]. XPS compositional analyses confirmed the co-existence of the two single-phases of  $\text{Co}_3\text{O}_4$  and ZnO materials. Therefore, it is concluded that the solvothermally prepared  $\text{Co}_3\text{O}_4$ -ZnO materials have NRs phase contained two different materials. This conclusion is supported by XRD data (see Fig. 3).

#### Detection of acetone using I-V technique

Due to their mechanical strength, good conductivity, large surface area, and extremely miniaturized size of ZnO-doped  $\text{Co}_3\text{O}_4$  NRs have been widely employed in electrode modification and toxic chemical detection. The doped NRs were applied for the detection of acetone in a liquid phase system at room temperature. Initially, the thin-film was fabricated using conducting binder (BCA/EA) and embedded on the AgE. The fabrication process and detection techniques are presented in the schematic diagram (Scheme 1). The PdE and NRs doped AgE were used as counter-electrode and working electrodes respectively, as presented in Scheme 1a and b respectively. The acetone was used as a target chemical in the liquid phase. The electrical responses in the presence of the toxic chemical were measured using I-V method according to Scheme 1c. The physico-sorption behaviors (adsorption and absorption) as well as the detection mechanism of ZnO-doped  $\text{Co}_3\text{O}_4$  NRs are presented in Scheme 1d. Here the acetone molecules are absorbed as well as adsorbed onto the fabricated surfaces in huge amounts due to their mesoporous nature and the large active surface area of nanostructure material in liquid phase respectively.

Figure 4a displays the current responses of the uncoated (gray-dotted) and coated (dark-dotted) AgE working electrode (surface area,  $0.0216 \text{ cm}^2$ ) with ZnO-doped  $\text{Co}_3\text{O}_4$  NRs at room temperature. With the NRs fabricated surface, the current signal is reduced compared to NRs without fabricated surface, which reveals that the surface is slightly blocked with doped nanomaterials. The current changes for the doped nanomaterials modified electrodes before (dark-dotted) and after (blue-dotted) injecting 50.0  $\mu\text{L}$  acetone (66.8  $\mu\text{M}$ ) in 10.0 mL phosphate buffer solution is presented in Fig. 4b. These considerable changes of surface current are measured with every injection of the target acetone into the electrochemical solution using an electrometer. 10.0 mL of 0.1 M phosphate buffer solution is originally transferred into the cell and the low to high concentration of acetone is added drop-wise consecutively from the stock solution. I-V responses with the ZnO-doped  $\text{Co}_3\text{O}_4$  NRs modified electrode surface are evaluated from the different concentrations (66.8  $\mu\text{M}$  to 1.33 M), as shown in Fig. 4c. It shows the current changes of fabricated films as a function of acetone concentration at room temperature. It is also observed that with increasing concentration of acetone, the resultant currents also increase significantly, which confirms

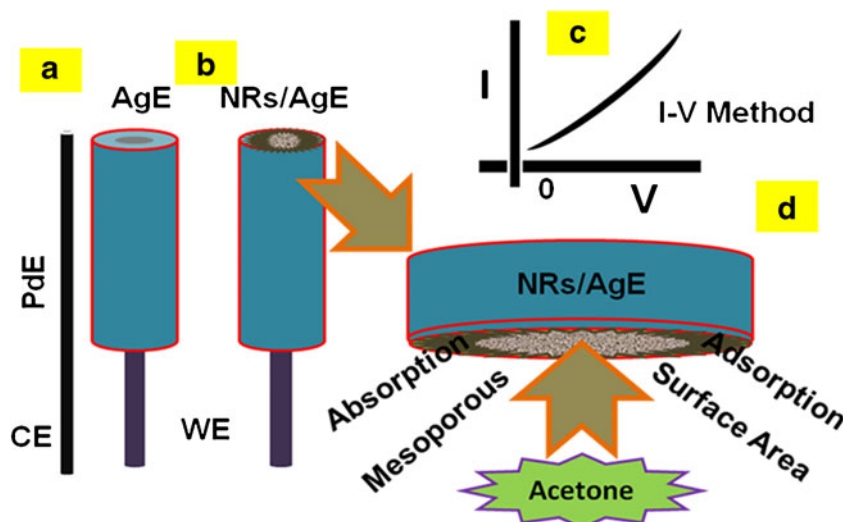
**Fig. 3** XPS of calcined a doped ZnO-doped Co<sub>3</sub>O<sub>4</sub> NRs, b Co2p, c Zn2p level, and d O1s level acquired with MgK $\alpha$  radiations



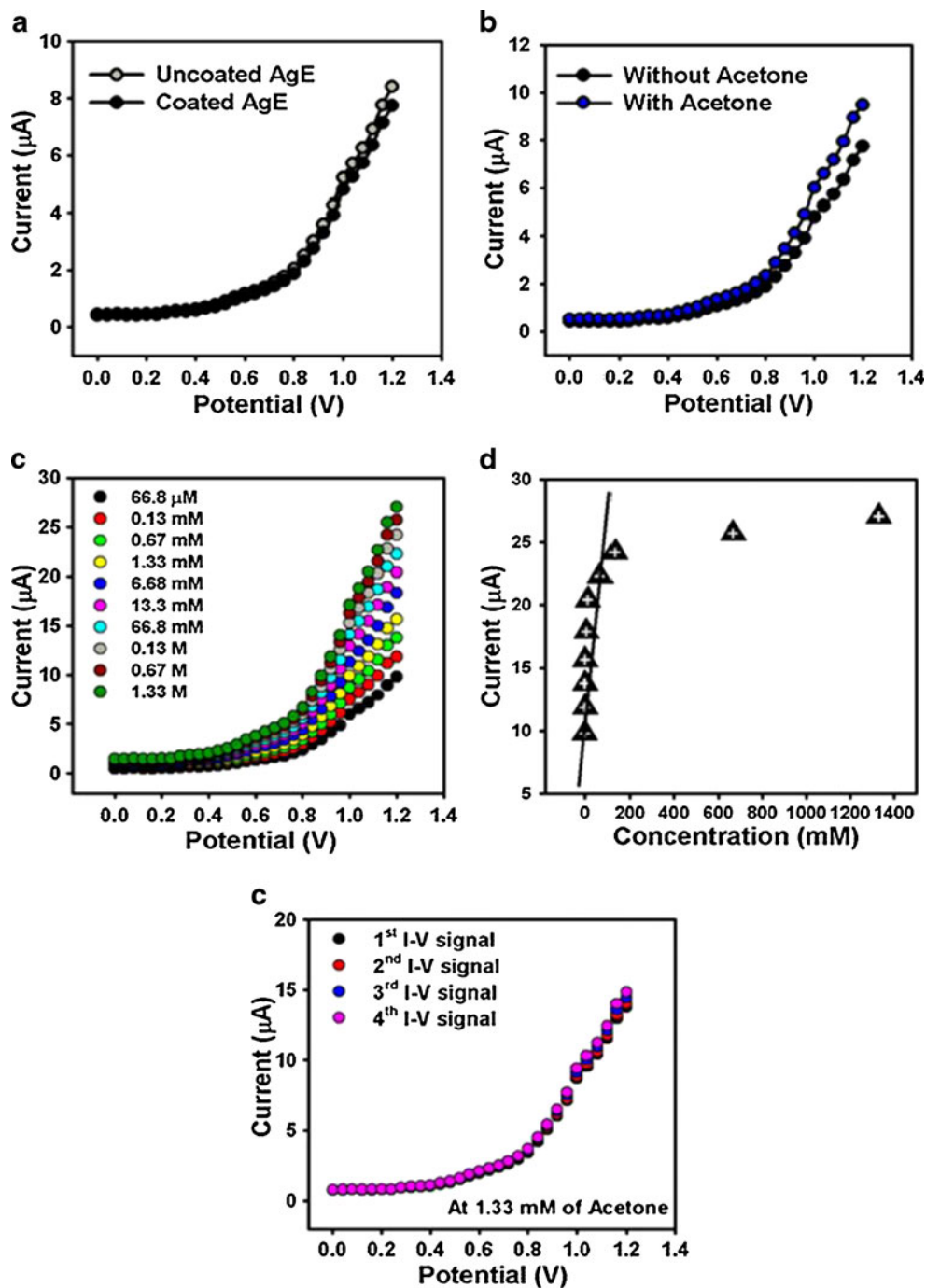
that the response is a surface process. It shows the response of doped NRs as a function of acetone concentration at room temperature. A large range of acetone concentrations is selected to study the probable analytical parameters, which are calculated in 66.8  $\mu$ M to 1.33 M. The calibration curve was plotted from the variation of acetone concentrations

(at +0.7 V), which is presented in Fig. 4d. It exhibits a calibration curve for the response current versus acetone concentration of fabricated ZnO-doped Co<sub>3</sub>O<sub>4</sub> NRs on the AgE sensor. It is observed from the calibration curve that, as the concentration of acetone increases, the current response also increases, and finally at a high-acetone concentration,

**Scheme 1** Fabrication process and methodology of acetone sensors using ZnO-doped Co<sub>3</sub>O<sub>4</sub> nanorods



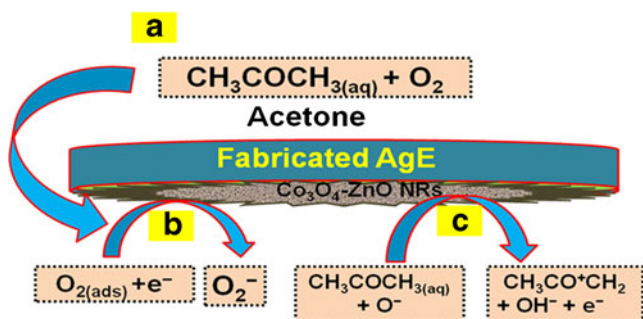
**Fig. 4** I-V responses of **a** uncoated and coated AgE; **b** without and with acetone (66.8  $\mu\text{M}$ ) in 10.0 mL phosphate buffer solution; **c** concentration variations (66.8  $\mu\text{M}$  to 1.33 M) of acetone; **d** calibration plot (at +0.7 V), and **e** repeatability signals of ZnO-doped  $\text{Co}_3\text{O}_4$  NRs fabricated on AgE surfaces. Potential was between 0.0 and +1.2 V



**Table 1** Comparison of the performances for acetone detection based on various material electrodes

Materials	LDR	Sensitivity	Detection limit	Linearity, $r^2$	Response time (s)	References
ZnO nanoparticles	0.13 mM to 0.13 M	$0.14 \mu\text{Ac}^{-2} \text{mM}^{-1}$	$0.068 \pm 0.01 \text{ mM}$	–	10.0	[41]
Mn-ZnS QDs	1.0 to 600.0 $\text{mgL}^{-1}$	–	$0.2 \text{ mgL}^{-1}$	–	–	[42]
$\text{Ag}_2\text{O}$ microflowers	0.13 $\mu\text{M}$ to 0.67 M	$1.60 \mu\text{Ac}^{-2} \text{mM}^{-1}$	0.11 $\mu\text{M}$	0.9462	10.0	[43]
Lead foil electrode	50 to 250 ppm	$2.07 \mu\text{A} \text{ppm}^{-1} \text{cm}^{-2}$	50.0 ppm	0.9980	18.0	[44]
ZnO- $\text{Co}_3\text{O}_4$ nanorods	66.8 $\mu\text{M}$ to 0.133 M	$3.58 \mu\text{Ac}^{-2} \text{mM}^{-1}$	14.7 $\mu\text{M}$	0.9684	10.0	Current work





**Scheme 2** Reaction mechanism of acetone on ZnO-doped Co<sub>3</sub>O<sub>4</sub> NRs nanomaterial fabricated surfaces over AgE at room temperature. **a** Main reactive part of sensing reaction, **b** Reaction proceeded by adsorbing oxygen from reaction system (interior or exterior) in the initial step, **c** Reaction proceeded by releasing electrons to the reaction medium in the second step

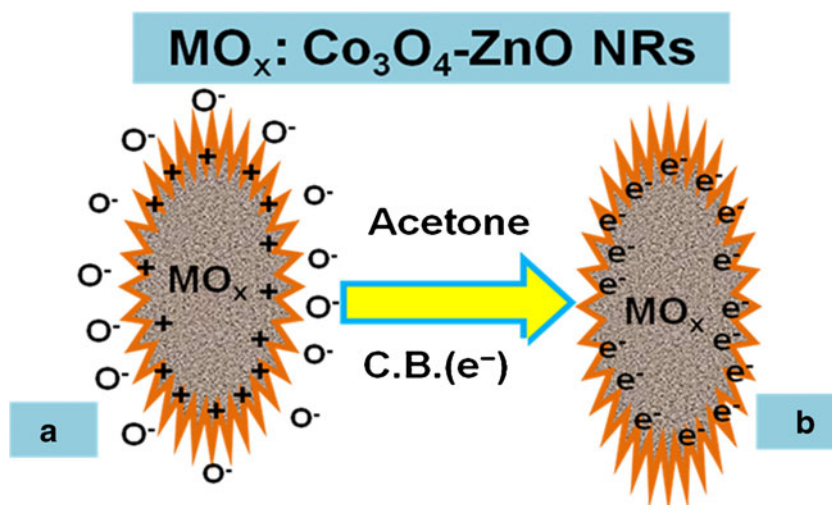
the current reaches a saturated level which suggests that the active surface sites of NRs are saturated with acetone. The sensitivity is calculated from the calibration curve, which is close to  $3.58 \mu\text{Acm}^{-2} \text{mM}^{-1}$ . The sensitivity value on the ZnO-doped Co<sub>3</sub>O<sub>4</sub> NRs sensor is 25 times higher than on the previously reported un-doped ZnO nanostructure materials [41]. The linear dynamic range of this sensor ranges from  $66.8 \mu\text{M}$  to  $0.133 \text{ M}$  (linearity,  $r^2=0.97$ ), and the detection limit was calculated to be  $14.7 \pm 0.2 \mu\text{M}$  [ $3 \times \text{noise (N)}/\text{slope(S)}$ ].

The response time for the ZnO-doped Co<sub>3</sub>O<sub>4</sub> NRs coated-sensor to achieve a saturated steady state current was approximately 10.0 s. The prominent sensitivity of the sensor can be attributed to the good absorption (porous surfaces fabricated with binders) and adsorption ability, high catalytic activity, and good biocompatibility of the doped nanomaterials. Due to the large surface area, the nanomaterials displayed a beneficial nano-environment for acetone detection and recognition with excellent sensitivity (Scheme 1). The sensitivity of ZnO-doped Co<sub>3</sub>O<sub>4</sub> NRs

affords high electron communication features which improved the direct electron movement between the active sites of NRs and the sensor electrode surfaces. Table 1 compares the analytical parameters for acetone sensors based on the performance of electrodes fabricated with various materials [41–44]. The modified thin ZnO-doped Co<sub>3</sub>O<sub>4</sub> NRs sensor film exhibited better reliability and stability. However, due to the high dynamic surface area, the ZnO-doped Co<sub>3</sub>O<sub>4</sub> NRs offered a productive environment for acetone adsorption and detection. The high sensitivity of NRs implies high electron-communication features which progress the direct electron communication between the active sites of ZnO-doped Co<sub>3</sub>O<sub>4</sub> NRs and the sensor electrode [45]. To check the repeatability and storage stabilities, I-V response for the doped nanomaterial coated sensor was examined (up to 4 weeks), and the results are presented in Fig. 4e. After each experiment, the fabricated sensor was washed thoroughly with the phosphate buffer solution and no significant decrease on current responses was obtained. The sensitivity remained almost constant over a period of 4 weeks; after that the response of the fabricated sensor gradually decreases.

Redox reaction takes place on the surface of the semiconductor ZnO-doped Co<sub>3</sub>O<sub>4</sub> NRs by reacting with the dissolved O<sub>2</sub> in bulk solution as well as with the surface-air of the neighboring atmosphere (interior or exterior). The acetone sensitivity towards ZnO-doped Co<sub>3</sub>O<sub>4</sub> NRs could be attributed to the oxygen deficiency and enhance the oxygen adsorption on the surface of doped nanomaterials. The oxidation of acetone depends on the amount of adsorbed oxygen. If the adsorbed oxygen on the sensor surface is high, then the oxidizing potential will be higher be the oxidation of acetone will be fast [46]. When acetone reacts with negatively charged adsorbed oxygen, it oxidizes to small components, releasing free electrons ( $e^-$ ) to the conduction band of ZnO-doped Co<sub>3</sub>O<sub>4</sub> NRs, according to

**Scheme 3** Reaction mechanism of acetone in the presence of semiconductor ZnO-doped Co<sub>3</sub>O<sub>4</sub> NRs at room temperature **a** Oxygen adsorbed on the surrounding of nanomaterials in the reaction system, **b** Reduction of acetone by releasing electrons to enhance the sensitivity of fabricated sensors



Scheme 2. Scheme 2 presents the main reactant (acetone & oxygen) in A, initial reaction of acetone with oxygen in B, secondary step of reaction proceeds by releasing electrons to the sensor surface in C.

The possible reaction could also be expressed by the following Scheme 3. Scheme 3a and b display the oxygen adsorbed on the surroundings of semiconductor doped nanomaterials in the reaction system and reduction of acetone by releasing electrons to enhance the sensitivity of fabricated sensors at room temperature respectively. These produced electrons contribute to a rapid increase in conductance of the fabricated thick-film of ZnO-doped  $\text{Co}_3\text{O}_4$  NRs and hence enhance the sensitivity [47]. ZnO-doped  $\text{Co}_3\text{O}_4$  NRs exhibit several approaches to providing chemical-based sensors and encouraging improvement has been accomplished in the research section. In spite of this development, there are still a number of important questions that are in need of additional investigation before this chemical sensor can be moved to commercial levels for established applications.

## Conclusions

Finally, a sensor is fabricated using hydrothermally prepared ZnO-doped  $\text{Co}_3\text{O}_4$  NRs with practically controlled shape structure, which exposed a constant morphological improvement in nanostructure materials and potential environmental applications. Semiconductor doped nanomaterials allow for a very sensitive transduction of the liquid/surface interactions for the chemical detections. This opportunity provides a variety of structural morphology assessments based on different approaches to the modification of the acetone molecules with metal oxide nanostructures. ZnO-doped  $\text{Co}_3\text{O}_4$  NR is used to fabricate a simple and efficient chemical detection method consisting of side-polished AgEs and measuring the particular acetone detection at room temperature. To the best of our knowledge, this is the first report for highly sensitive detection of acetone with ZnO-doped  $\text{Co}_3\text{O}_4$  NRs using simple and reliable I-V method in short response time. This approach has been described for the detection of acetone with doped nanomaterials and shows various attractive and potential features such as simplicity, reproducibly, rapidity, reliability, and inexpensiveness. We hope that this I-V method with semiconductor doped nanomaterials can be attractive for simultaneous detection of toxic chemicals in environmental and healthcare fields.

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