Porous LiFePO₄/NiP Composite Nanospheres as the Cathode Materials in Rechargeable Lithium–Ion Batteries

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ABSTRACT

We report the synthesis of porous LiFePO₄/NiP composite nanospheres and their application in rechargeable lithium-ion batteries. A simple one-step spraying technique was developed to prepare LiFePO₄/NiP composite nanospheres with an electrical conductivity 10^3 – 10^4 times that of bulk particles of LiFePO₄. Electrochemical measurements show that LiFePO₄ nanospheres with a uniform loading of 0.86 wt%–1.50 wt% NiP exhibit high discharge capacity, good cycling reversibility, and low apparent activation energies. The superior electrode performance of the as-prepared composite nanospheres results from the greatly enhanced electrical conductivity and porous structure of the materials.

KEYWORDS

Nanospheres, LiFePO₄/NiP, spraying, electrical conductivity, rechargeable lithium-ion batteries

Introduction

Developing new advanced materials is the most critical challenge in the fields of energy storage and energy conversion with high efficiency [1–4]. As an example, LiFePO₄ with the olivine structure has attracted great interest as the cathode material in rechargeable lithium-ion batteries because of its high-energy density, low cost, and safety [5–10]. However, the poor intrinsic electrical conductivity and the critical requirements of the synthesis procedures limit the large scale application of LiFePO₄. Various methods such as adding conductive carbon [11, 12], coating with a π -bonded polymer [13], reducing particle size [14], and doping with supervalent cations [15] have been employed in order to improve the electrical conductivity of LiFePO₄. It has also

been found that a nano-network of conductive iron phosphide (Fe₂P) in the grain boundaries of LiFePO₄ can enhance its electronic conductivity [16]. Synthesis routes based on both solid phase and wet chemistry reactions have been used to prepare LiFePO₄ [17–19]. Generally, the as-prepared LiFePO₄ powders show irregular particle morphology with a broad particle size distribution, which results in low tapping density and energy density.

Herein, we report the incorporation of a nickel phosphorus alloy (NiP) in porous $LiFePO_4/NiP$ composite nanospheres through a facile spraying technique. The method allows low reaction temperature and easy control over the preparation process, and hence can produce $LiFePO_4/NiP$ composite nanospheres in high yield as well as with low cost and low energy consumption. Electrochemical

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tests show that the as-prepared LiFePO₄ nanospheres with the addition of 0.86 wt%–1.50 wt% NiP are promising candidates as the cathode material of rechargeable lithium-ion batteries.

1. Experimental

1.1 Sample preparation

LiFePO₄/NiP composite nanospheres were prepared through a spraying technique with a modified Yamato Pulvis minispray instrument (Model GA-32) at a relatively low temperature of 350 °C (Electronic Supplementary Material (ESM), Fig. S-1) [20, 21]. The apparatus is divided into three parts comprising the droplet generator, pyrolysis reactor, and product collector. The pressure and temperature of the reaction system can be controlled. The pyrolysis reactor is a sloping chamber with an angle of about 45°, which can prolong the period of heating favoring the formation of fine particles. All the chemicals were of analytical grade and used as received without further purification. In a typical synthesis, $0.1 \text{ mol/L Fe}(NO_3)_3$ ·9H₂O and 0.1 mol/L LiH₂PO₄ were mixed and then a solution containing 5 mmol/L NiSO₄· $6H_2O_1$ 6 mmol/L sodium citrate (Na₃C₆H₅O₇·2H₂O), and 5 mmol/L NaH₂PO₂·H₂O was added. The mixed solution was drawn into the apparatus by the fluid pump and transported to the outlet of the spray nozzle by pressurized high purity N_2 (99.999%, Air Products & Chemical Co. Ltd.) which also serves to control the system temperature. The sprayed droplets were heated at about 350 °C under the protection of N_2 gas during their descent of the pyrolysis reactor. The resulting particles were collected, washed with deionized water and ethanol, and then vacuum-dried at 80 °C for 4 h. By following the above procedures and changing the molar ratio of Fe to Ni, we obtained a series of LiFePO₄ nanospheres that contained 0, 0.35 wt%, 0.86 wt%, and 1.50 wt% NiP, as measured by inductively coupled plasma spectroscopy [22]. For comparison, LiFePO₄ bulk particles were prepared through a conventional solid-phase reaction [23] using Li_2CO_3 , $NH_4H_2PO_4$, and $Fe(CH_3CO_2)_2$.

1.2 Characterization

Phase structures of the as-prepared samples were

characterized using a Rigaku INT-2000 X-ray powder diffractometer (XRD) with Cu Ka radiation. The structural refinement for LiFePO₄ was performed with the RIETAN-2000 program [24]. Transmission electron microscopy (TEM) and high-resolution TEM (HRTEM) images were taken on a Philips Tecnai-F20 transmission electron microscope equipped with an energy dispersive X-ray (EDX) spectrometer. The tapping density of the samples was measured by tapping the powder 20 times in a 10 mL cylinder. Resistivity measurements were performed at room temperature using a four-point contact arrangement on a powered pellet (\emptyset 8 mm × 10 mm). The specific surface areas of the products were measured by the Brunauer-Emmett-Teller (BET) method (Micromeritics ASAP 2010). Elemental analysis was carried out using a Hitachi P-5200 inductively coupled plasma spectrometer.

1.3 Electrochemical measurement

Electrochemical measurements were carried out using two-electrode cells with lithium metal as the counter electrode and reference electrode. The working electrodes were fabricated by compressing a mixture of 85 wt% active materials, 10 wt% acetylene black, and 5 wt% polytetrafluoroethylene onto an Al foil. The cell assembly was operated in a glovebox (Mikrouna China Universal 2240/750) filled with high purity argon (99.999%, Air Products & Chemical Co. Ltd.). The electrolyte was 1 mol/L LiPF₆ dissolved in a mixture of ethylene carbonate (EC) and diethyl carbonate (DEC) with the volume ratio of EC:DEC=1:1. Electrochemical performance was investigated using a Parstat 2273 potentiostat/ galvanostat analyzer (Princeton Applied Research & AMETEK Company) and a modified Arbin chargedischarge unit at controlled temperatures [25, 26]. Cyclic voltammograms (CVs) were carried out at a scan rate of 1 mV/s and a temperature of 20 °C. Electrochemical impedance spectroscopy (EIS) was investigated at a voltage of 3.30 V in the frequency range 0.1-1×10⁶ Hz. The electrode capacity was measured by a galvanostatic charge-discharge method at a constant current density of 50 mA/g to a cut-off voltage of 2.5 V at 20 °C. Charge-discharge cycles were tested with a current density of 50 mA/g



in the potential range 2.5–4.3 V. The capacity was calculated based on the amount of the active material, excluding the mass of the additives in the electrode.

2. Results and discussion

2.1 Characterization of the as-synthesized product

Figure 1(a) shows the XRD pattern and Rietveld analysis of the LiFePO₄ nanospheres containing 0.86 wt% NiP, while Fig. S-2 (in the ESM) displays the XRD patterns of the as-prepared LiFePO₄ bulk particles and nanospheres with different NiP contents. The diffraction peaks can be readily indexed to an orthorhombic phase with Pmnb space group (JCPDS-ICDD card No. 40-1499). No impurity peaks were observed. The cell parameters obtained from the Rietveld refinement are consistent with the standard value (Table S-1 in the ESM), indicating that addition



Figure 1 (a) Rietveld refinement of the XRD pattern of LiFePO₄ containing 0.86 wt% NiP. Red cross: observed intensity; green line: calculated intensity; blue solid line: the difference between the observed intensity and the calculated intensity; short green vertical bars indicate the positions of all possible reflections. (b) The crystal structure of LiFePO₄ viewed along the *c* axis

of a small amount of amorphous NiP does not have an effect on the crystal structure of $LiFePO_4$. Figure 1(b) shows the olivine structure of $LiFePO_4$, in which the dark-red octahedra represent Fe(II) coordinated by six oxygen atoms and the blue tetrahedra represent P(V) coordinated by four oxygen atoms. The Fe–O octahedra share corners with neighboring octahedra, forming a mesh layer which alternates with a layer of lithium chains. A P–O tetrahedron shares corners with its neighboring octahedra, embedded in the trough of the corrugated Fe–O layer. The structural characteristics facilitate the insertion/extraction of lithium ions.

Figure 2 shows representative TEM images of the as-prepared LiFePO₄ bulk particles and nanospheres. As shown in Fig. 2(a), the sample prepared by the conventional solid phase reaction route is composed of irregular shaped particles with sizes of up to micrometers. The as-synthesized LiFePO₄ nanospheres without any NiP (Fig. 2(b)) and with the addition of different amounts of NiP (Figs. 2(c)-(e)) consist mainly of nanospheres with an average diameter of about 200 nm. The proportion of the material with a nanosphere morphology in the LiFePO₄ nanospheres with the addition of 0.35 wt% and 0.86 wt% NiP is higher than that in LiFePO₄ nanospheres without NiP or with 1.5 wt% NiP. The HRTEM image (Fig. 2(f)) of one of the nanospheres in Fig. 2(d) shows the presence of a porous network structure with some disorder. Figures 2(g) and (h) show the elemental mapping of Ni and P in the LiFePO₄/NiP composite nanospheres, revealing that Ni and P are distributed homogeneously and the atomic ratio of Ni/P is close to 1:1.

The specific surface areas of the as-prepared LiFePO₄ bulk particles, nanospheres without NiP, and nanospheres containing 0.35 wt%, 0.86 wt%, and 1.5 wt% NiP are 6.2, 11.7, 15.8, 19.2, and 18.5 m²/g, respectively. The tapping densities of the LiFePO₄ irregular particles and nanospheres are in the range 1.6–1.8 g/cm³. The electrical conductivities of the as-prepared LiFePO₄ bulk particles, nanospheres without NiP, and nanospheres containing 0.35 wt%, 0.86 wt%, and 1.5 wt% NiP are 8.9×10^{-9} , 7.2×10^{-5} , 5.7×10^{-5} , and 1.2×10^{-4} S/cm, respectively. Therefore, the addition of conductive NiP leads to an



Figure 2 TEM images of the as-prepared LiFePO₄ bulk particles (a), nanospheres without NiP (b) and containing 0.35 wt% (c), 0.86 wt% (d), and 1.50 wt% (e) NiP. HRTEM image (f) of the middle sphere in (d), and the EDX mapping of Ni (g) and P (h)

increase in the electrical conductivity of LiFePO₄ by three to four orders of magnitude. The homogeneous distribution of amorphous NiP in the porous LiFePO₄ nanospheres contributes to the enhanced electrical conductivity.

2.2 Electrochemical measurements

Figure 3 shows the CVs of the electrodes made from the as-prepared LiFePO₄ bulk particles and LiFePO₄/ NiP composite nanospheres in the first cycle. Similar voltammograms were also obtained for the following five cycles. Reversible peaks were observed for the electrodes composed of bulk particles and nanospheres, but the observed peak positions and currents were different. The results shown in Table 1 confirm that the addition of 0.86 wt%–1.50 wt% NiP is very effective in increasing the kinetics of reversible Li⁺ insertion/extraction in the electrodes.

EIS has been proven to be a valuable and widely

Table 1 Summary of the cyclic voltammetry measurements

used technique to evaluate electrode kinetics [27, 28]. Therefore, the activation energies (E_a) of lithium extraction for the as-prepared LiFePO₄/NiP composite nanospheres were measured by EIS. Figure 4 shows the EIS of the as-prepared LiFePO₄/NiP composite nanospheres. The impedance curves for the samples have a similar profile with one semicircle with different radii in the high frequency region and a sloping line in the low frequency region. The equivalent circuit for the impedance spectra is shown in Fig. 4(f). After fitting the equivalent circuit, we can obtain the charge transfer impedance (R_{ct}). The apparent activation energies (E_a) of lithium-ion insertion/extraction for the

samples can be calculated by the Arrhenius equation (Eq. 1),



Figure 3 Cyclic voltammograms of the electrodes made from the as-prepared LiFePO₄ bulk particles (a), nanospheres without NiP (b) and containing 0.35 wt% (c), 0.86 wt% (d), and 1.50 wt% NiP (e) in the first cycle. The scan rate was 1 mV/s and the temperature 20 °C

Electrode	Potential values (V)			Current densities (mA/cm ²)	
	E _R	Eo	$E_{\rm O}-E_{\rm R}$	I _R	I _O
LiFePO ₄ bulk particles	3.343	3.643	0.300	4.36	4.07
Pure LiFePO ₄ nanospheres	3.380	3.630	0.250	4.82	4.53
LiFePO ₄ /0.35 wt% NiP nanospheres	3.385	3.629	0.244	5.18	4.89
LiFePO ₄ /0.86 wt% NiP nanospheres	3.404	3.601	0.197	6.31	6.01
LiFePO ₄ /1.50 wt% NiP nanospheres	3.390	3.615	0.225	5.59	5.25

 $E_{\rm R}$: reduction potential; $E_{\rm O}$: oxidation potential; $I_{\rm R}$ and $I_{\rm O}$: the corresponding current densities.





Figure 4 EIS of the as-prepared LiFePO₄/NiP composite nanospheres measured with a discharging voltage of 3.30 V and temperatures of 298, 323, and 333 K: LiFePO₄ bulk particles (a), nanospheres without NiP (b) and containing 0.35 wt% (c), 0.86 wt% (d), and 1.50 wt% NiP (e). The equivalent circuit for the impedance spectra of the samples is shown in (f)

ln $(T/R_{ct}) = -E_a/RT + \ln A'$ (1) where *T* is the absolute temperature, R_{ct} is the charge transfer impedance, E_a is the activation energy, *R* is the gas constant, and *A'* is a temperature-independent coefficient. The Arrhenius plots are shown in Fig. S-3 (in the ESM). The activation energies of the as-prepared LiFePO₄ bulk particles, nanospheres without NiP, and nanospheres containing 0.35 wt%, 0.86 wt%, and 1.5 wt% NiP are 24.5, 23.8, 22.5, 9.0, and 13.9 kJ/mol, respectively. It can be seen that LiFePO₄ nanospheres with the addition of 0.86 wt% NiP have lowest activation energy, indicating the fastest lithium insertion/extraction.

Figure 5 shows the charge-discharge curves of the electrodes made from various LiFePO₄ samples at a current density of 50 mA/g and 20 °C. Compared with the electrode composed of LiFePO₄ bulk particles, the electrodes made of nanospheres display lower charge plateaus and higher discharge plateaus, indicating their capability of high power output. The discharge capacities for the electrodes composed of the as-prepared LiFePO₄ bulk particles, nanospheres without NiP, and nanospheres containing 0.35 wt%, 0.86 wt%, and 1.5 wt% NiP are 125.9, 144.0, 158.4, 162.3, and 159.8 mAh/g, respectively. Therefore, the charge and discharge performance of LiFePO₄ electrode is significantly improved by the addition of NiP and the optimum content of NiP is 0.86 wt%.



Figure 5 Charge-discharge curves of the electrodes for the asprepared LiFePO₄ bulk particles (a), nanospheres without NiP (b) and containing 0.35 wt% (c), 0.86 wt% (d), and 1.50 wt% NiP (e) at a current density of 50 mA/g and temperature of 20 °C

Figure 6 shows the curves of discharge capacity versus cycle number for the electrodes made from different LiFePO₄ samples at a current density of 50 mA/g and 20 °C. After 50 cycles with 100% depth of charge and discharge, the electrodes of the asprepared LiFePO₄ bulk particles (a), nanospheres without NiP (b), and nanospheres containing 0.35 wt% (c), 0.86 wt% (d), and 1.5 wt% NiP (e) are 119.1, 140.8, 155.9, 160.8, and 158.2 mAh/g, respectively. The rate of capacity decay in Fig. 6 displays the following order: (d) < (e) < (c) < (b) < (a), showing that the cycling capability of LiFePO₄ nanospheres can be improved by the addition of NiP.



Figure 6 Cycle life of the electrodes for the as-prepared LiFePO₄ bulk particles (a), nanospheres without NiP (b) and containing 0.35 wt% (c), 0.86 wt% (d), and 1.50 wt% NiP (e) at a current density of 50 mA/g and temperature of 20 °C

The LiFePO₄/NiP composite electrodes display superior electrochemical properties to those of the LiFePO₄ electrode composed of bulk particles. This enhancement results from the increased electrical conductivity, larger specific surface areas, porous nanostructures with disorder, and low internal and charge-transfer resistance. Overall, the optimum NiP content in the LiFePO₄ nanospheres is around 0.86 wt%; this material has a low apparent activation energy of only 9.0 kJ/mol for lithium insertion/ extraction. Furthermore, the as-prepared nanospheres have higher tapping density and better fluidity and dispersivity than the powders composed of irregular particles, which may be beneficial in the manufacture of high quality electrodes. Therefore, LiFePO₄/NiP composite nanospheres are promising candidates for

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use as the cathode materials in rechargeable lithiumion batteries with high energy, high power, high safety, and long life, of the type which are needed for pure or hybrid electric vehicles.

3. Conclusions

The synthesis of LiFePO₄/NiP composite nanospheres with porous structures has been realized through a facile one-step spraying technique under mild conditions with low cost inorganic salts as precursors. It is found that by doping LiFePO₄ with 0.35 wt%-1.50 wt% amorphous NiP, the electrical conductivity of the composite nanospheres is dramatically improved by three or four orders of magnitude compared to bulk LiFePO₄. Cyclic voltammetry and galvanostatic results illustrated that both pure LiFePO₄ and LiFePO₄/NiP composite nanospheres exhibit superior electrochemical performance to that of bulk LiFePO₄. The optimum amount of NiP in the porous nanospheres is around 0.86 wt%. This study indicates that LiFePO₄/NiP composite nanospheres are promising cathode materials for use in rechargeable lithium-ion batteries with high capacity, high power, and long life.

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