Patil *et al. International Nano Letters* 2012, **2**:17 http://www.inl-journal.com/content/2/1/17

# **ORIGINAL ARTICLE**

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# Preparation and characterization of SnO<sub>2</sub> nanoparticles by hydrothermal route

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# Abstract

This paper demonstrates the synthesis of SnO<sub>2</sub> nanoparticles using a simple hydrothermal route in the presence of the surfactant hydrazine at 100 °C for 12 h. X-ray diffraction (XRD), field emission scanning electron microscopy, and transmission electron microscopy (TEM) were employed to characterize the as-prepared product, and optical property was studied by UV-visible diffuse reflectance spectroscopy (DRS). The XRD pattern of the as-prepared sample is indexed to the tetragonal structure of SnO<sub>2</sub>, and the calculated particle size is 22.4 nm, which is further confirmed by TEM. The selected area electron diffraction patterns showed continuous ring patterns without any additional diffraction spots and rings of secondary phases, revealing their crystalline structure. Analysis of the DRS spectrum showed the bandgap of the synthesized SnO<sub>2</sub> to be 3.6 eV. The anionic surfactant hydrazine plays a key role in the formation of the SnO<sub>2</sub> nanostructures. A probable reaction for the formation of SnO<sub>2</sub> nanoparticles is proposed.

Keywords: SnO<sub>2</sub> nanoparticles, Hydrothermal route, FESEM, TEM

# Background

Nanomaterials have attracted great interest due to their intriguing properties, which are different from those of their corresponding bulk state. In the past few years,  $SnO_2$  is an important n-type wide-energy-gap semiconductor (Eg = 3.64 eV, 330 K) which has a wide range of applications such as in solid-state gas sensors [1], transparent conducting electrodes [2], rechargeable Li batteries [3], and optical electronic devices [4]. During the past decade,  $SnO_2$  nanostructures have been one of the most important oxide nanostructures due to their properties and potential applications [5,6].

Many processes have been developed to the synthesis of SnO<sub>2</sub> nanostructures, e.g., spray pyrolysis [5], hydrothermal methods [6-8], evaporating tin grains in air [9], chemical vapor deposition [10], thermal evaporation of oxide powders [11], rapid oxidation of elemental tin [12], the sol–gel method [13], etc. Davar et al. [14] reported the synthesis of SnO<sub>2</sub> nanoparticles by thermal decomposition using [bis(2-hydroxyacetophenato)tin(II)], [Sn(HAP)<sub>2</sub>], as precursor. Salavati-Niasari et al. [15] synthesized zinc blend ZnS nanoparticles by a thioglycolic

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acid (HSCH<sub>2</sub>COOH)-assisted hydrothermal technique via the reaction between a new inorganic precursor [bis(2-hydroxyacetophenato)zinc(II)], [Zn(HAP)<sub>2</sub>], and thio-acetamide (CH<sub>3</sub>CSNH<sub>2</sub>). Gnanam and Rajendran [16] synthesized nanocrystalline tin oxide powders of about 8 to 13 nm in size using different surfactants such as cetyltrimethyl ammonium bromide, sodium dodecyl sulphate, and polyethylene glycol via hydrothermal reaction at 160°C for 12 h and studied their structural and photoluminescence properties.

A simple hydrazine-assisted hydrothermal route was employed to synthesize nanocrystalline  $SnO_2$  powders in this study, and structural, morphological, microstructural, and optical properties were discussed.

#### Methods

All reagents used were of analytical grade without further purification. First, 3.505 g of  $SnCl_4 \cdot 5H_2O$  (0.1 M) was dissolved in 100 ml of distilled water, and then 1.2800 g of hydrazine hydrate (0.01 M) was added with stirring.  $N_2H_4 \cdot H_2O$  immediately reacted with  $SnCl_4$  in the solution to form a slurry-like white precipitate of the hybrid complex between  $N_2H_4$  and  $SnCl_4$ . After 10 min of stirring, the solution was transferred into a Teflonlined stainless steel autoclave with a capacity of 200 ml

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and then sealed. The autoclave was maintained at  $100^{\circ}$ C for 12 h and cooled naturally to room temperature. The product was centrifuged, filtered out, and rinsed with methanol and distilled water several times, and then dried at  $120^{\circ}$ C for 1 h in air.

The possible reaction of SnCl<sub>4</sub>·5H<sub>2</sub>O with hydrazine produced SnO<sub>2</sub> nanoparticles via Sn<sup>4+</sup> reaction with NH<sub>4</sub>OH. The process can be expressed as follows:

$$mSnCl_4 + nN_2H_4 \rightarrow (SnCl_4)_m (N_2H_4)_n$$
(1)

$$(\mathrm{SnCl}_4)_m(\mathrm{N}_2\mathrm{H}_4)_n \to m\mathrm{Sn}^{4+} + n\mathrm{N}_2\mathrm{H}_4 + 4m\mathrm{Cl}^- \quad (2)$$

$$3N_2H_4 + 4H_2O \rightarrow 4NH_4OH + N_2 \tag{3}$$

$$Sn^{4+} + 4NH_4OH \rightarrow SnO_2 \downarrow + 4NH_4^+ + 2H_2O. \quad (4)$$

Prior to the hydrothermal process, the  $(\text{SnCl}_4)_m(\text{N}_2\text{H}_4)_n$  complex clusters were formed via reaction (1), and at the same time, the clusters were agglomerated into the slurry-like white precipitate mentioned above. As represented in reaction (2), the  $(\text{SnCl}_4)_m(\text{N}_2\text{H}_4)_n$  clusters underwent dissociation when the solution was heated to 100°C during



Figure 2 FESEM image of the SnO<sub>2</sub> sample.

the hydrothermal stage. In reaction (3),  $OH^-$  ions were formed via the dissociation of  $N_2H_4$  into  $NH_4OH$  and  $N_2$ [17]. Reaction (4) represents the formation of the  $SnO_2$ nanoparticles via the reaction between  $Sn^{4+}$  and  $OH^-$  ions formed in reaction (3).

The synthesized sample was characterized by X-ray powder diffraction (XRD) using the XRD Make-Bruker D-8 model (Bruker AXS, Inc., Madison, WI, USA) with CuK $\alpha$  radiation with a wavelength  $\lambda =$ 1.5418 Å at 2 $\theta$  values between 20° and 80°. Transmission electron microscopy (TEM) images were recorded from a transmission electron microscope (CM-200, Make-PHILIPS, Amsterdam, The Netherlands). The UV-visible (UV–vis) diffuse reflectance spectrum (DRS) was obtained from a JASCO UV–vis/NIR spectrophotometer V-670 model (Easton, MD, USA).

# **Results and discussion**

# Structural properties by XRD

The XRD pattern of the product is shown in Figure 1. The peaks at  $2\theta$  values of 26.6°, 33.8°, 37.9°, 51.8°, 54.7°,



Figure 3 SAED pattern of the SnO<sub>2</sub> sample.



61.9°, and 65.9° can be associated with  $(1 \ 1 \ 0)$ ,  $(1 \ 0 \ 1)$ ,  $(2 \ 0 \ 0)$ ,  $(2 \ 1 \ 1)$ ,  $(2 \ 2 \ 0)$ ,  $(3 \ 1 \ 0)$ , and  $(3 \ 0 \ 1)$ , respectively. A matching of the observed and standard (*hkl*) planes confirmed that the product is of SnO<sub>2</sub> having a tetragonal structure, which are in good agreement with the literature values (JCPDS card no. 41–1445). The average particle size (*D*) was estimated using the Scherrer equation [18]:

$$D = \frac{0.9\lambda}{\beta\cos\theta},\tag{5}$$

where *D* is the crystallite size,  $\lambda$  is the X-ray wavelength,  $\beta$  is the full width at half maximum of the diffraction peak, and  $\theta$  is the Bragg diffraction angle of the diffraction peaks. The average particle size was found to be 22.4 nm.

#### Morphological properties by FESEM

Figure 2 shows the field emission scanning electron microscopy (FESEM) micrograph of the synthesized  $SnO_2$  sample. Clustering of particles seems to have occurred on the surface. In this image, cubic structures can be easily seen.

Table 1 d values obtaine	ed from XRD and TEM
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#### Microstructural properties by TEM and SAED pattern

Figure 3 shows the electron diffraction patterns of the sample. It is clear from the figure that the  $SnO_2$  particles are crystalline in nature. The electron diffraction patterns show continuous ring patterns without any additional diffraction spots and rings of secondary phases, revealing their crystalline structure. Seven fringe patterns corresponding to planes (1 1 0), (1 0 1), (2 0 0), (2 1 1), (2 2 0), (3 1 0), and (3 0 1) are consistent with the peaks observed in the XRD patterns. XRD and TEM studies confirmed pure tetragonal structure of  $SnO_2$  as evidenced from Figures 1 and 4, respectively. The ring-to-the-center distance of each ring is measured as 3.01, 4.23, 4.41, 5.61, 6.52, 7.10, and 7.71 and expressed in terms of nm<sup>-1</sup>. The reciprocal of these values gives the interplanar distance *d*. Details are given in Table 1.

## Optical properties by UV-vis DRS

To determine the optical bandgap of synthesized  $SnO_2$ , the reflectance spectra of the  $SnO_2$  thick film prepared by screen printing technique [19] on a glass substrate was measured. The reflectance (*R*) spectra of the  $SnO_2$  thin film were shown in Figure 5.

As seen in Figure 5, the reflectance spectra show a strong decrease after 360 nm. This decrease is related to optical transitions occurring in the optical bandgap. In order to determine the precise value of the optical bandgap of the  $SnO_2$ , the reflectance values were converted to absorbance by application of the Kubelka-Munk function [20,21].

The Kubelka-Munk theory is generally used for the analysis of diffuse reflectance spectra obtained from weakly absorbing samples. The Kubelka-Munk formula is expressed by the following relation:

$$F(R) = \frac{(1-R)^2}{2R} = \frac{K}{S},$$
 (6)

where F(R) is the Kubelka-Munk function which corresponds to the absorbance, R is the reflectance, K is the absorption coefficient, and S is the scattering coefficient.

Reported <i>d</i> values (Å)	XRD <i>d</i> values (Å)	Electron diffraction (TEM)		Planes
		Reciprocal of <i>d</i> values $\delta_{hkl}$ (nm <sup>-1</sup> )	d values d <sub>hkl</sub> (Å)	(hkl)
3.35	3.342	3.01	3.320	(1 1 0)
2.64	2.604	4.23	2.364	(1 0 1)
2.37	2.372	4.41	2.267	(2 0 0)
1.76	1.770	5.61	1.782	(2 1 1)
1.67	1.653	6.52	1.533	(2 2 0)
1.50	1.501	7.10	1.408	(3 1 0)
1.41	1.418	7.71	1.297	(3 0 1)



It is well known that the optical transitions in semiconductor materials are taken place by direct and indirect transitions. The absorption coefficient  $\alpha$  for direct transitions is expressed by the following relation [22]:

$$\alpha h v = A (h v - Eg)^n, \tag{7}$$

where  $\alpha$  is the linear absorption coefficient of the material, A is an energy-independent constant, Eg is the optical bandgap, and n is a constant which determines the type of optical transitions: for indirect allowed transition, n = 2; for indirect forbidden transition, m = 3; for direct allowed transition, n = 1/2; and for direct forbidden transition, m = 3/2. The F(R) values of the SnO<sub>2</sub> film were obtained using the  $\frac{(1-R)^2}{2R}$  relation in Equation 6 [23,24] and the Kubelka-Munk function F(R) is directly proportional to the absorbance. Therefore, F(R) values were converted to the linear absorption coefficient by means of the  $\alpha = \frac{F(R)}{t} = \frac{\text{Absorbance}}{t}$  relation [25], where *t* is the thickness of the SnO<sub>2</sub> film. The curve of  $\left(\frac{F(R)h\nu}{t}\right)^2$  vs. hv for the SnO<sub>2</sub> film was plotted, as shown in Figure 6. The optical bandgap (Eg) of the SnO<sub>2</sub> film was determined from the curve of  $\left(\frac{F(R)h\nu}{t}\right)^2$  vs. hv and was found to be 3.6 eV. The optical bandgap of the SnO<sub>2</sub> studied is similar to that of undoped SnO<sub>2</sub> materials obtained by various methods [26,27]. This suggests that the optical bandgap of SnO<sub>2</sub> semiconductors changes with respect to the synthesis method used.

## Conclusions

 $SnO_2$  nanoparticles have been successfully synthesized by a simple hydrothermal method at low temperature using hydrazine hydrate as a mediator. The structural, morphological, microstructural, and optical properties of a  $SnO_2$  sample were investigated. XRD spectra indicated that the as-prepared product is polycrystalline in nature.



It was also shown from these spectra that the crystallite structure was observed to be tetragonal. The surface morphology was investigated by FESEM. The crystallite size (22.4 nm) of the SnO<sub>2</sub> nanoparticles, estimated by XRD, is confirmed by TEM. The optical bandgap of the SnO<sub>2</sub> film was found to be 3.6 eV.

#### **Competing interests**

The authors declare that they have no competing interests.

#### Authors' contributions

GEP synthesized the nanocrystalline SnO<sub>2</sub> materials, carried out the characterization, and drafted the manuscript. DDK participated in the discussions and interpretation of all characterization results. VBG and GHJ gave the final approval of the version to be published. All the authors read and approved the final manuscript.

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#### Acknowledgements

The financial support for this work through the INSPIRE Fellowship for doctoral degree from DST, New Delhi, is gratefully acknowledged. The authors thank the Sophisticated Analytical Instrument Facility, Indian Institute of Technology (IIT), Bombay, for carrying out TEM characterization and C-MET, Pune, for providing the FESEM facility.

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#### Received: 1 April 2011 Accepted: 28 February 2012 Published: 27 July 2012

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#### doi:10.1186/2228-5326-2-17

Cite this article as: Patil *et al.*: Preparation and characterization of SnO<sub>2</sub> nanoparticles by hydrothermal route. *International Nano Letters* 2012 2:17.

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