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## STRESS CORROSION CRACKING EVALUATION OF SEVERAL PRECIPITATION HARDENING STAINLESS STEELS

By T. S. Humphries and E. E. Nelson Astronautics Laboratory

September 12, 1969

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The stress corr	osion cracking resistance of	precipitation hardening
stainless steels PH	13-8 Mo, PH 14-8 Mo, 15-5 PH,	PH 15-7 Mo, 17-4 PH,
17-7 PH, A-286, Alma	r 362, AM-350, and Unitemp 21 round tensile, flat tensile,	2 are presented. Three
types of specimens (	bjected to alternate immersion	n in a 3.5 percent
salt solution. The	results indicated that under	these test conditions,
PH 13-8 Mo, PH 14-8	Mo, 15-5 PH, 17-4 PH, A-286,	Almar 362, and Unitemp
212 stainless steels	are highly resistant to stre	ss corrosion cracking
in practically all f	orms and heat treat condition	s. Alloys PH 15-7 Mo
and 17-7 PH are susc	ceptible to stress corrosion c	racking in nearly all
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#### STRESS CORROSION CRACKING EVALUATION OF SEVERAL PRECIPITATION HARDENING STAINLESS STEELS

#### SUMMARY

Numerous stress corrosion failures of space vehicle components fabricated from precipitation hardening stainless steels have been encountered. Most of these problems were solved by changing to a more stress corrosion resistant alloy or heat treat condition. Unfortunately only a limited amount of published data is available on the stress corrosion cracking characteristics of the precipitation hardening (PH) stainless steels. Because of this, a test program was initiated to evaluate the stress corrosion resistance of the more frequently used PH stainless steels in an accelerated test solution (alternate immersion in 3.5 percent salt solution) representative of the general service environment of space vehicles prior to launch.

The results of this investigation indicated that PH 13-8 Mo, PH 14-8 Mo, 15-5 PH, 17-4 PH, A-286, Almar 362, and Unitemp 212 stainless steels are highly resistant to stress corrosion cracking in 3.5 percent salt water in all heat treat conditions tested. PH 15-7 Mo and 17-7 PH stainless steels were susceptible to stress corrosion cracking in all conditions except 17-7 PH-CH 900, and PH 15-7 Mo is reportedly highly resistant in the CH 900 condition (1). AM-350 stainless steel was susceptible in the SCT 850 condition but resistant in the SCT 1000 condition.

#### INTRODUCTION

High strength stainless steels and the so called superalloys have played an important role in the advent and growth of jet engines, missiles, supersonic aircraft and space vehicles. The precipitation hardening stainless steels are among the more important materials that have made possible the rapid growth in these fields.

The PH stainless steels combine the excellent corrosion resistance of the austenitic chromium - nickel steels with the heat hardening characteristics of the straight chromium, martensitic steels (1). Many of the problems encountered with normal high temperature heat treatments are eliminated by the relatively low temperature aging treatment of the PH stainless steels, which are basically of two types: (1) martensitic and (2) semi-austenitic. The PH stainless steels like the hardenable chromium stainless steels may under certain conditions of tensile stress and corrosive environment suffer stress corrosion. As with most metals, the stress corrosion cracking susceptibility normally increases with increasing hardness or strength. However, susceptibility is not governed solely by hardness or strength of the material, but in certain cases appears to be associated with the process procedure used to obtain these properties. For example, both 17-7 PH and PH 15-7 Mo stainless steels exhibit the highest resistance to stress corrosion cracking in Condition CH 900 although this condition gives the highest hardness of any of the available heat treat conditions of these two alloys (2).

#### EXPERIMENTAL PROCEDURES

The PH stainless steels evaluated in this investigation were PH 13-8 Mo, PH 14-8 Mo, 15-5 PH, PH 15-7 Mo, 17-4 PH, 17-7 PH, A-286, Almar 362, AM-350, and Unitemp 212 in the form of bar stock and/or sheet. Three types of specimens were required to test the material in at least two directions of grain orientation. Flat tensile specimens, loaded by constant deflection, were used for testing sheet material; round tensile specimens, stressed in direct tension, were used for testing the longitudinal direction of all bar stock and the transverse direction of two inch or greater diameter bar; and C-rings, utilizing the constant deflection method, were used for testing the transverse direction of bar stock of less than two inch diameter. In one case, (17-4 PH Condition A) flat tensile specimens were stressed in direct tension, and this is illustrated along with the other methods of loading in Figures 1, 2, and 3.

The specimens were deflected or strained the calculated amount to give the desired stress levels, wiped with acetone, and placed in the alternate immersion tester until failure or until the test was terminated (approximately six months). A detailed description of the test specimens, formulas for calculating deflection and strain, and methods of loading and testing is given in Reference 3. Mechanical properties of the alloys were measured in both directions of testing. The chosen stress level was from 25 to 100 percent of the directional yield strength except as noted for small diameter bar. Duplicate unstressed tensile specimens were exposed under identical conditions for comparative control. The tests were conducted in a ferris wheel type alternate immersion tester (Figure 4) containing a 3.5 percent solution (deionized water) of sodium chloride, with an immersion cycle of 10 minutes in solution followed by 50 minutes of drying above the solution.

#### RESULTS AND DISCUSSION

The compositions of the alloys evaluated in this program are listed in Table I. In some cases, the typical analysis is given because the composition of the specific material was not available. The mechanical properties and the heat treatments used to obtain the properties of the PH stainless steels are shown in Tables II and III, respectively. Listed in Table IV are the complete stress corrosion results obtained in this investigation.

Armco's\* martensitic PH stainless steels (PH 13-8 Mo, 15-5 PH, and 17-4 PH) were found to possess a higher degree of stress corrosion resistance than their semi-austenitic type (PH 15-7 Mo, and 17-7 PH). No failures were encountered in the three martensitic stainless steels in most forms and heat treat conditions tested at loads up to 100 percent of their yield strengths. The only stress corrosion failure occurred in 17-4 PH-H 900 bar stock stressed in the transverse grain direction to 100 percent of the yield strength. This agrees with information published by Armco(2), which states that, for maximum resistance to stress corrosion cracking, 17-4 PH should be hardened at the highest temperature that will yield required properties, but not less that 1000°F. However, the results obtained with Condition-A material does not agree with Armco's information that the structure of 17-4 PH Condition A is untempered martensite having poor resistance to stress corrosion cracking. No stress corrosion failures were encountered with either sheet or bar stock of 17-4 PH Condition A at stress loads up to 100 percent yield strength exposed to both alternate immersion in salt water and salt spray.

The semi-austenitic type PH 15-7 Mo and 17-7 PH suffered stress corrosion cracking in all conditions of heat treat except the CH 900. The only condition in which 17-7 PH steel was found to possess a high resistance to stress corrosion cracking was the cold rolled and hardened Both PH 14-8 Mo and PH 15-7 Mo are, reportedly, resistant to CH 900. stress corrosion cracking in Condition CH 900 (1,2). This is somewhat surprising because the highest mechanical properties obtainable from these three semi-austenitic stainless steels are produced by Condition The test results indicated that the transverse grain direction CH 900. of both sheet and bar of PH 15-7 Mo and 17-7 PH was more susceptible to stress corrosion cracking than the longitudinal direction. In addition, both alloys were more susceptible to stress corrosion cracking when hardened at 950°F than 1100°F (lower mechanical properties). It was reported that PH 15-7 Mo is more resistant to stress corrosion than 17-7 PH and the TH condition is more resistant than the RH(1). The

\*Armco Steel Corporation

results of this investigation did not substantiate these findings. From a stress corrosion standpoint, the semi-austenitic PH 14-8 Mo was similar to the martensitic stainless steels in that this alloy was highly resistant to stress corrosion cracking in all heat treat conditions tested.

Of the remaining precipitation hardening stainless steels tested, A-286, Almar 362, and Unitemp 212 were highly resistant to stress corrosion cracking in all test tempers or conditions. Alloy AM-350 was found to be resistant to stress corrosion cracking in the SCT 1000 condition but susceptible in the higher strength SCT 850 condition. As with alloys PH 15-7 Mo and 17-7 PH, alloy AM-350 was more susceptible to stress corrosion cracking in the transverse grain direction than in the longitudinal direction.

#### CONCLUSIONS AND RECOMMENDATION

The results obtained with this accelerated test indicated that:

1. The precipitation hardening PH 13-8 Mo, PH 14-8 Mo, 15-5 PH, 17-4 PH, A-286, Almar 362, and Unitemp 212 stainless steels are highly resistant to stress corrosion cracking.

2. Alloys PH 15-7 Mo and 17-7 PH were susceptible to stress corrosion cracking in all heat treat conditions tested except CH 900, and AM-350 was resistant in Condition SCT 1000 but susceptible in Condition SCT 850.

3. Precipitation hardening stainless steels appear to be less resistant to stress corrosion cracking in the transverse direction of grain orientation than in the longitudinal direction.

4. The stress corrosion cracking susceptibility of the precipitation hardening stainless steels generally increased with increasing hardness or strength, but in certain cases appears to be associated with the process procedure used to obtain these properties. Alloy 17-7 PH stainless steel exhibited the highest resistance to stress corrosion cracking in Condition CH 900 and this condition gave the highest hardness and strength of any of the conditions tested.

The stress corrosion cracking resistance should be determined for the precipitation hardening stainless steels in all recommended process and heat treat conditions because of the effect of these conditions on the resistance to stress corrosion cracking.

#### REFERENCES

- 1. E. E. Denhard, Jr. "Stress Corrosion Cracking of High Strength Stainless Steels," Stress Corrosion Cracking in Aircraft Structural Materials, AGARD Conference Proceedings No. 18, 1967.
- 2. Armco Product Data S-6a (17-4 PH), S-21a (15-5 PH), S25 (PH 14-8 Mo), S-30 (17-7 PH), S-33b (PH 13-8 Mo), S-37 (PH 15-7 Mo), Armco Steel Corporation

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3. Humphries, T. S.: Procedures for Externally Loading and Corrosion Testing Stress Corrosion Specimens, NASA TM X-53483, June 1966.

<u>A110y</u>	Source & Heat No.	- Form	U	띬	<b>M</b>	νI	<u>S1</u>	립 임	Composition Wt % N1 Mo	Wt % Mo	<u>A1</u>	zl	깅	ଶ	Ta	됍
PH 13-8 Mo	Armco Steel Corp.**	Bar	.05*	.10*	.010*	.008*	.10*	13.00	8.0	2.25	1.2	.01*				
PH 14-8 Mo Air Melt	Armco Steel Corp. 33024	Sheet	. 033	.39	.007	.005	.38	14.61	8.14	2.27	1.23					
PH 14-8 Mo Vacuum Melt	Armco Steel Corp. 5432	Sheet	.032	. 05	. 003	. 004	.05	15.05	8.32	2.21	1.22	.004				
15-5 PH	Armco Steel Corp.**	Bar	*0.	1.00*	* 70 .	.03*	1.00*	15.00	5.00				3.50	Cb + <b>Ta</b> .30	6	
PH 15-7 Mo	Armco Steel Corp. 44570	Bar	0.74	.70	.018	600.	.50	15.17	7.33	2.22	1.20					
PH 15-7 Mo	Armco Steel Corp. 850395	Sheet	.070	.66	.018	.012	.38	15.27	7.27	2.21	1.16					
17-4 PH	Armco Steel Corp. 64091	Sheet	.035	. 28	.017	.010	. 54	15.97	4.37				3.28	. 25	.02	
17-4 PH	Armco Steel Corp.**	Bar	.07*	1.00*	.04*	.03*	1.00*	17.00	4.00				4.0	Cb + T .30	-	
17 <b>-</b> 7 PH	Armco Steel Corp. 850268	Sheet	.069	.50	.012	.015	.41	16.94	7.26		1.18					
17-7 PH	Armco Steel Corp. 63742	Bar	.067	. 65	.014	.010	. 28	16.83	7.29		1.23					
<b>A -</b> 286	**	Ваг	.05	1.35			. 50	15.00	26,00	1.25	0.2					2.0
AM-350	Allegheny Ludlum 29721**	Sheet	. 08	.80			. 25	16.50	4.30	2.75		. 10				
Almar 362	Allegheny Ludlum A65500	Bar	.029	.35	.017	.018	. 20	14.73	6.60	.095	.011		. 12			77.
Unitemp 212	Universal Cyclops KH 3765	Bar	.06		<b>ن</b> 10	.01		15.94	24.99		<del>د</del> . 02		¥ 01	.57		4.10
* Maximum Allowable	owable															

TABLE I. CHEMICAL COMPOSITION OF PRECIPITATION HARDENING STAINLESS STEEL

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\*\* Typical Analysis or Composition

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TABLE 11. MECHANICAL PROPERTIES OF PRECIPITATION HARDENING STAINLESS STEEL

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Alloy	Form	Heat Treatment	Grain Direction	Tensile Strength (ksi)	Yield Strength (ksi)	Percent Elongation
РН 13-8 Мо	Bar Bar Bar Bar	Н 950 Н 1000	Trans. Long. Trans. Long.	226 231 219 222	211 217 210 212	16 15 10
PH 14-8 Mo Air Melt	Sheet Sheet Sheet Sheet	SRH 950 SRH 1050	Trans. Long. Trans.	216 218 221 218	194 198 205 202	د 01 م م
PH 14-8 Mo Vacuum Melt	Sheet Sheet Sheet Sheet	SRH 950 SRH 1050	Trans. Long. Trans.	247 242 241 238	231 225 223 230	v ç 4 v
PH 15-5 Mo	Bar Bar Bar	H 900 H 925 H 1025	Long. Long. Long.	189 187 164	173 163 156	11 11 9
15-7 РН	Sheet Sheet Sheet Sheet Sheet Sheet Bar Bar Bar	RH 950 RH 1050 RH 1075 RH 950 RH 1075	Trans. Long. Trans. Long. Long. Trans. Long. Long.	240 224 205 203 203 203 203 203	225 221 220 217 201 197 186 186	ら
17-7 РН	Sheet Sheet Bar Bar Bar	H 900 Anne <b>a</b> led H 900	Trans. Long. Trans. Long. Long. Long.	215 213 156 155 200	207 206 124 151 184	4 4 1 0 8 4 1 0 8 1 8 1 8 1 8 1 8 1 8 1 8 1 8 1 8 1

TABLE II. MECHANICAL PROPERTIES OF PRECIPITATION HARDENING STAINLESS STEELS (Continued)

Percent <u>Elongation</u>	6 1	o o	Ŋ	ъ	6	9	7	7	8	7	80	5	4	4	4	4	5	5	ę	9	6	30	13	11	11	10	12	16	25	23
Yield Strength (ksi)	228	233 233	220	214	212	193	190	209	205	185	181	186	176	171	170	154	152	167	160	148	145	95	188	187	189	164	162	161	135	136
Tensile Strength (ksi)	243 225	234 234	230	219	217	198	193	216	213	192	190	187	184	176	174	158	155	173	170	154	152	147	203	210	208	177	178	167	187	185
Gr <b>a</b> in Direction	Trans.	Long Trans.	Long.	Trans.	Long.	Trans.	Long.	Trans.	Long.	Trans.	Long.	Trans.	Long.	Trans.	Long.	Trans.	Long.	Trans.	Long.	Trans.	Long.	Long		Trans.	Long.	Trans.	Long.	Long.	Long.	Long.
Heat Treatment	CH 900*	RH 950		RH 1050		RH 1100		<b>TH 1050</b>		TH 1100		RH 950		RH 1050		RH 1100		TH 1050		TH 1100		No Cold Work	>40% Cold Work	SCT 850		SCT 1000		1000°F, 3 Hrs.	Double Aged	Single Aged
Form	Sheet	Sheet	Sheet	Sheet	Sheet	Sheet	Sheet	Sheet	Sheet	Sheet	Sheet	Bar	Bar	Bar	Bar	Bar	Bar	Bar	Bar	Bar	Bar	Bar	Ваг	Sheet	Sheet	Sheet	Sheet	Bar	Bar	Bar
Alloy	17-7 PH																					A -286		<b>AM -</b> 350				Almar 362	Unitemp 212	

\* The tensile and yield strengths are approximately 10 percent below published typical properties.

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1. PH 13-8 Mo, 15-5 PH, and 17-4 PH H 900 - 900F, 1 hour, A.C. H 925, H950, H1025 - heat at indicated temperature for 4 hours 2. PH 14-8 Mo - SRH 950 and SRH 1050 Austenite conditioning: 1700F, 1 hour, A.C. Transformation cooling: Minus 100F, 8 hours Precipitation hardening at indicated temperature for 1 hour 3. PH 15-7 Mo a. Condition RH 950 (heat treated per Boeing specification BAC 5619) Austenite conditioning: 1725, 10 min. per 0.1 inch section thickness (10 minutes minimum) Transformation cooling: Minus 100F, 8 hours Precititation hardening: 950F, 1 hour b. Condition RH 1075 (Heat Treated per NAR specification MA0111-009) Austenite conditioning: 1750, 10 minutes plus 1 minute for each 0.01 inch section thickness, A.C. Transformation cooling: Minus 100F, 4 hours Precipitation hardening: 1075F, 1 hour 4. 17-7 PH Condition CH 900 (material cold reduced to Condition C) a. Heated at 900F for one hour b. Condition RH 950, RH 1050, and RH 1100 Austenite conditioning: 1750F, 10 minutes, A.C. Transformation cooling: Minus 100F, 8 hours Precipitation hardening at indicated temperature for one hour c. Condition TH 1050 and TH 1100 Austenite conditioning: 1400F, 1.5 hour Transformation cooling: Within 1 hour to 60F, 1/2 hour Precipitation hardening at indicated temperature for 1.5 hour

- 5. A-286
  - **a**. A-286

Solution treated 1800F, 2 hours, W.Q., aged 1325F, 16 hours, by vendor

b. A-286 with cold work

Solution treated 1800F, 2 hours, W.Q., cold worked 40 percent minimum, aged 1200F, 16 hours, by vendor

6. Almar 362

Aged 1000F, 3 hours

- 7. AM 350 SCT 850 and SCT 1000 LTA 1710F, 5 minutes, A.C., minus 100F, 3 hours, temper at indicated temperature for 3 hours.
- 8. Unitemp 212
  - a. Double aged Solution treated 1850F, 2 hours, aged 1425F, 2 hours, A.C., 1250F, 16 hours
  - b. Single aged Solution treated by vendor, aged 1325F, 16 hours, by MSFC
- \* All material received in the solution treated or annealed condition and heat treated by MSFC except as noted.

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Material Form	Heat Treat Condition	Stress Direction	Applied ksi	Stress % YS	Failure Ratio	D <b>a</b> ys to Failure	% Loss in T.S.
		<u>PH 1</u>	3-8 Mo Stai	nless Ste	<u>e1</u>		
Bar Stock (10" Dia.)	Н 950	Long.	163 196	75 90	0/2 0/2		N N
(10 D1a.)		Trans.	158 190	75 90	0/2 0/2		N N
	H 1000	Long.	159 191	75 90	0/2 0/2		N N
		Trans.	158 190	75 90	0/3 0/3		N N
		<u>15-</u>	5 PH Stainl	ess Steel			
Bar Stock (2 " Dia.)	Н900	Long.	130 173	75 100	0/3 0/3		N N
	Н 925	Long	0	0			N
			123 163	75 100	0/3 0/3		N N
	Н 1025	Long	0 117 156	0 75 100	0/3 0/3		N N N
		17-4	PH Stainle	ss Steel			
Sheet	Condition A	Long	110	75	0/6(2)		N
(,062")	(Solution Treated -		$\frac{148}{110}$	100 75	0/6(2) 0/6(2)		N N
	Anne <b>a</b> led)	Trans.	110 148	75 100	0/6 <sup>(2)</sup> 0/6 <sup>(2)</sup>		N N
		Long.	110 148	75 100	0/3(3) 0/3(3)		N N
		Trans.	110 148	75 100	0/5(3) 0/3(3)		N N
Bar Stock (2.5" Dia.)	Condition A (Solution Treated - Annealed)	Long.	0 113 151	0 75 100	0/9(4) 0/9(4)	  	N N N
		Trans.	0 93	0 75	$\frac{1}{0/9(4)}$		N N
			124	100	0/9(4)		N
Sheet (.062")	н 900	Long.	0 155 206	0 75 100	0/3 0/3	 	N N N
		Tr <b>a</b> ns.	0 156 207	0 75 100	0/3 0/3		N N N

## TABLE IV. STRESS CORROSION CRACKING TEST RESULTS<sup>(1)</sup>

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Material Form	Heat Treat Condition	Stress Direction	Applied <u>ksi</u>	Stress % YS	<u>Failure Ratio</u>	Days to Failure	% Loss <u>In T.S.</u>
Bar Stock	н 900	Long.	0	0			N
(2.5" Dia.)		-	140	75	0/3		N
			186	100	0/3		N
					-,•		IN
		Trans.	0	0			N
			146	75	0/6		N
			195	100	2/6	50,90	N
	Н 925	Trans.	0	0			N
			152	75	0/3		N
			202	100	0/3		N
Bar Stock	11 1005	_					
(2.5" Dia.)	H 1025	Trans.	0	0		~-	N
(2.5 DIA.)			130	75	0/3		N
			174	100	0/3		N
		<u>PH 14-8</u>	Mo Stainl	ess Steel			
Sheet	SRH 950	Long.	0	0			6
(.058" Thick)	Vacuum Melt	U	182	75	0/3		16
			243	100	0/3		6
					-, •		0
		Trans.	0	0			17
			185	75	0/3		21
			247	100	0/3		N
	SRH 1050	Long.	0	0			14
	Vacuum Melt	Ū	173	75	0/3		14
			231	100	0/3		N N
		Trance	0	0			
		Trans.	0	0		÷-	21
			185	75	0/3		17
			246	100	0/3		7
	SRH 950	Long.	0	0			N
	Air Melt		163	75	0/3	-	19
			218	100	0/3		15
		Trans.	0	0			
		LLallS.	0 161	0			7
				75	0/3		10
			216	100	0/3		N
	SRH 1050	Long.	0	0			N
	Air Melt		161	75	0/3		13
			218	100	0/3		N
		Trans,	0	0			
			166	0 75	0/3		N
			221	100	0/3		18
				100	610		N

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## TABLE IV. STRESS CORROSION CRACKING TEST RESULTS<sup>(1)</sup> (Continued)

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Material Form	Heat Treat Condition	Stress Direction	Applied ksi	Stress % YS	Failure Ratio	D <b>a</b> ys to F <b>ai</b> lure	% Loss <u>In T.S.</u>
		<u>PH 15-</u>	7 Mo Stainle	ss Steel			
Sheet (.062" Thick)	RH 900	Long.	0 55 110 165 220	0 25 50 75 100	0/3 0/3 0/6 3/5	  3(2),156	N N N
		T <b>ra</b> ns,	0 56 112 168 224	0 25 50 75 100	0/3 0/3 3/3 3/3	  3(2),52 3(2),4	N N - -
	RH 1075	Long.	0 54 108 151 202	0 25 50 75 100	0/3 0/3 3/3 3/3		N N 8 -
		Trans.	0 55 110 151 201	0 25 50 75 100	0/3 3/3 3/3 3/3	 47,58,100 6,17(2) 4,5(2)	N N - -
Bar Stock (2.5" Dia.)	RH 950	Long.	0 50 101 151 201	0 25 50 75 100	0/3 3/3 3/3 3/3	40,48,70 1(2),2 1(3)	N 8 - -
		Trans.	0 56 112 150	0 25 50 75	2/3 3/3 2/2	1,40 1 (2),4 1 (2)	10 N -
Ba: Stock (2.5" Dia.)	RH 1075	Long.	0 46 92 139 185	0 25 50 75 100	0/3 0/3 3/3 3/3	 5,6(2) 5,15,23	N 18 12 -
		Trans.	0 45 91 136	0 25 50 75	2/3 3/3 2/2	29,34 4(3) 6(2)	53 N -
		<u>17-7</u>	PH St <b>a</b> inles	s Steel			
Sheet (.050" Thick)	СН 900	Long.	114 171 205	50 75 90	0/3 0/3 0/3	  	N N N
		Trans.	114 171 205	50 75 90	0/3 0/3 0/3		N N N

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## TABLE IV. STRESS CORROSION CRACKING TEST RESULTS <sup>(1)</sup> (Continued)

Material Form	Heat Treat Condition	Stress Direction	Applied <u>ksi</u>	Stress % YS	<u>Failure Ratio</u>	Days to Failure	% Loss <u>In T.S.</u>
Sheet							
(.062" Thick)	RH 950	Long.	0	0			N
			100	43	0/3		N
			140	61	0/3		N
			180	80	4/9	1,165(3)	N
		Trans.	0	0			N
			100	45	0/3		N
			140	64	1/3	71	N
			180	81	6/6	1 to 117	-
	RH 1050	Long.	0	0			N
			140	65	0/3		N
			180	84	2/6	165(2)	N
		Trans.	0	0			N
			140	65	0/3		N
			180	84	5/6	1,51,165(3)	N
Sheet (.062" Thick)	RH 1100	Long.	180	95	0/3		N
(1002 111200)		Trans.	180	93	0/3		N
	тн 1050	Long.	0	0			N
		U	140	66	0/3		N
			180	87	1/6	165	N
		Trans.	0	0			N
			140	67	0/3		N
			180	88	2/6	165(2)	N
	<b>T</b> H 1100	Long.	180	99	0/3		N
		Trans.	180	97	0/3		N
Bar Stock	RH 950	Long	0	0			N
(2.5" Dia.)		. 5	50	24	2/3	1,22	N
			100	48	3/3	1(2), 4	-
			140	70	3/3	1,5(2)	-
		Trans.	o	0			23
			50	25	3/3	1(3)	-
			100	50	2/2	1(2)	-
			140	70	3/3	1(3)	-
	RH 1050	Long.	0	0			N
		-	50	26	0/3		N
			100	52	0/3		N
			140	73	2/3	13,29	N
		Trans.	0	0			11
			50	27	1/3	22	24
			100	55	3/3	1(2),4	-
			140	77	3/3	1(3)	-

## TABLE IV. STRESS CORROSION CRACKING TEST RESULTS(1) (Continued)

Material Form	Heat Treat Condition	Stress Direction	Applie <u>ksi</u>	d Stress % YS	<u>Failure Ratio</u>	Days to Failure	% Loss In T.S.
	RH 1100	Long.	0	0			N
		0	50	32	0/3		N
			100	63	0/3		N
			140	89	1/3	39	N
Bar Stock							
(2.5" Dia.)	RH 1100	Trans.	0	0			N
			50	33	0/3		N
			100	65	1/3	48	N
			140	91	2/3	1(2)	N
	тн 1050	Long.	140	78	0/3		Ν
		Trans.	50	27	3/3	4(2),28	-
			100	55	3/3	1(2),4	-
			140	77	3/3	1(3)	-
			1.10			- (0)	
	ТН 1100	Long.	140	93	0/3		N
		Irans.	0	0			N
		110.001	50	33	2/3	34,37	N
			100	66	2/3	1,82	N
			140	92	3/3	4,8,11	-
		A -286	Stainle	ss Steel			
					o ( o		
Bar Stock	No Cold	Trans.	48	50	0/3		-
(1" Di <b>a</b> .)	Work	(C-ring)(5)		75	0/3		-
			95	100	0/3		-
	40% Min.	Long.	0	0			N
	Cold Work		140	75	0/9		N
			170	90	0/9		N
			188	100	0/9		N
		Trans.	94	50	0/9		N
		(C-ring) <sup>(5)</sup>	140	75	0/9		-
		(	170	90	0/9		-
			188	100	0/9		-
		<u>Almar 3</u>	62 Stai	nless Steel	<u>.                                     </u>		
Bar Stock	Aged 1000°F,	Long.	0	0			N
(1.75" Dia.)	3 Hours	Long.	120	75	0/3		N
(1.75 514.)	J Hours		161	100	0/3		N
		Trene	120	75	0/3		_
		Trans. (C-ring)(5)	161	100	0/3		-
					0/0		
		<u>AM 350</u>	Stainl	ess Steel			
Sheet	SCT 850	Long.	141	75	0/3		N
(.025" Thick)	•/•	0.	188	100	2/2	5.20	-
					D / D	<b>A</b> (1) <b>A</b> = -	
		Trans.	140	75	2/3	20,153	N
			187	100	2/2	1(2)	-

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### TABLE IV. STRESS CORROSION CRACKING TEST RESULTS<sup>(1)</sup> (Continued)

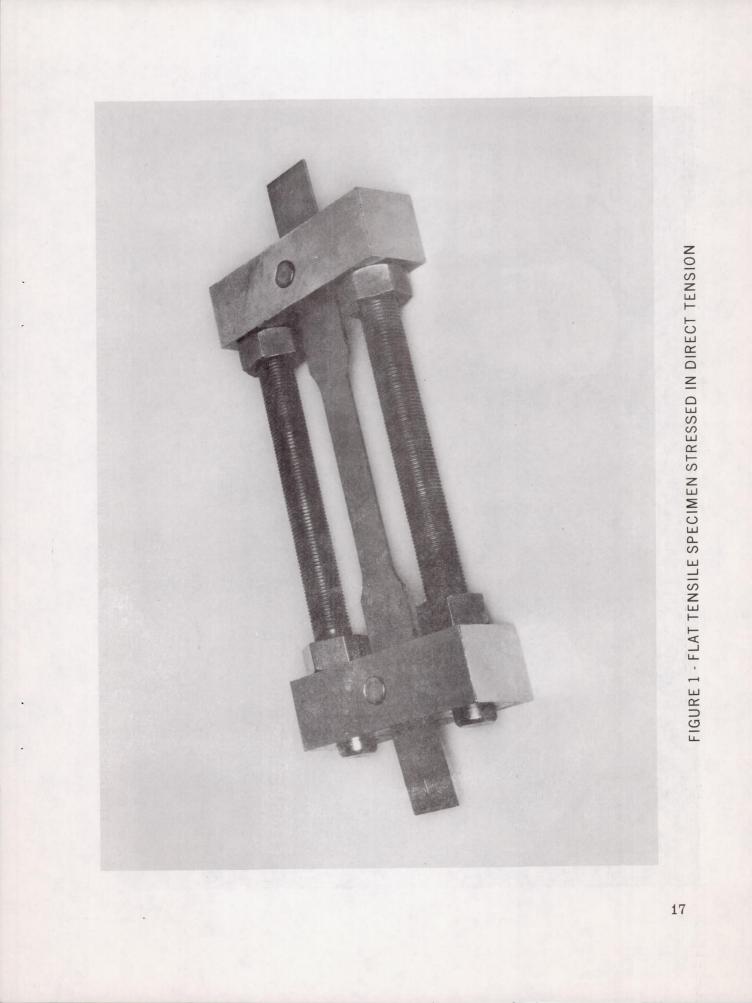
TABLE IV. STRESS CORROSION CRACKING TEST RESULTS (1) (Continued)

at Treat Stress Applied Stress Days to % Loss pudition <u>Direction</u> ksi % YS Failure Ratio Failure <u>In T.S</u> .	27 1000 Long. 122 75 0/3 N 162 100 0/2 N	Trans. 123 75 0/3 N 164 100 0/2 N	Unitemp 212 Stainless Steel	Trans. 34 25 (C-ring)(5) 70 50 102 75 122 90 136 100	Trans. 34 (C-ring)(5)67 95 121
Heat Treat S Condition D	SCT 1000 I	E		Single-Aged (	Double-Aged (
Material Form				Bar Stock (1.5" Thick)	

N - Negligible change in tensile properties

Note (1) Test Data

- Specimen: Round tensile (C-ring where noted) for bar stock and flat tensile for sheet.
- b. Stress method: Direct tension for round tensile and constant deflection for flat tensile and C-rings
- c. Medium: Alternate immersion in 3.5 percent NaCl solution
- d. Exposure time: Until failure or six months
- (2) Three of the specimens were tested in the standard medium and the remaining three were exposed to 5 percent salt spray.
- (3) The specimens were loaded in direct tension rather than by constant deflection
- (4) Six of the specimens were tested in the standard medium and the remaining three specimens were exposed to 5 percent salt spray.
- (5) Load calculations were based on longitudinal rather than transverse yield strength.



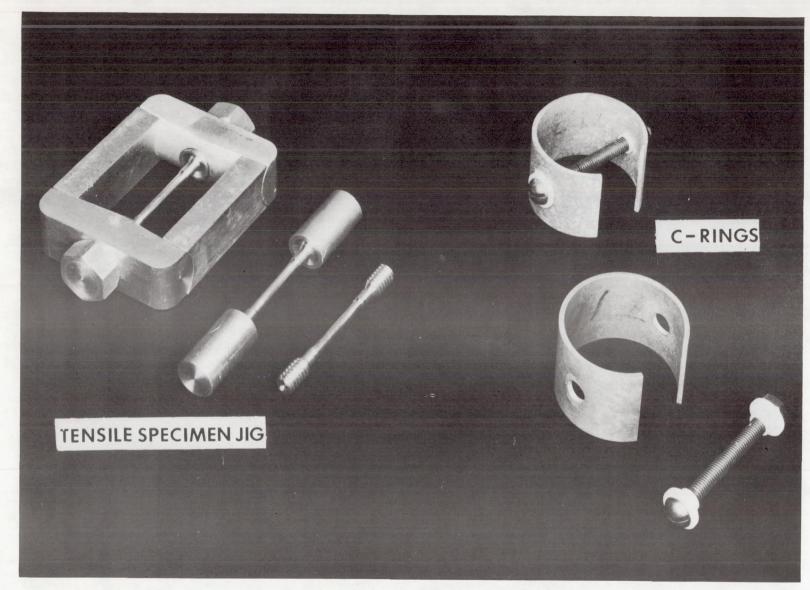
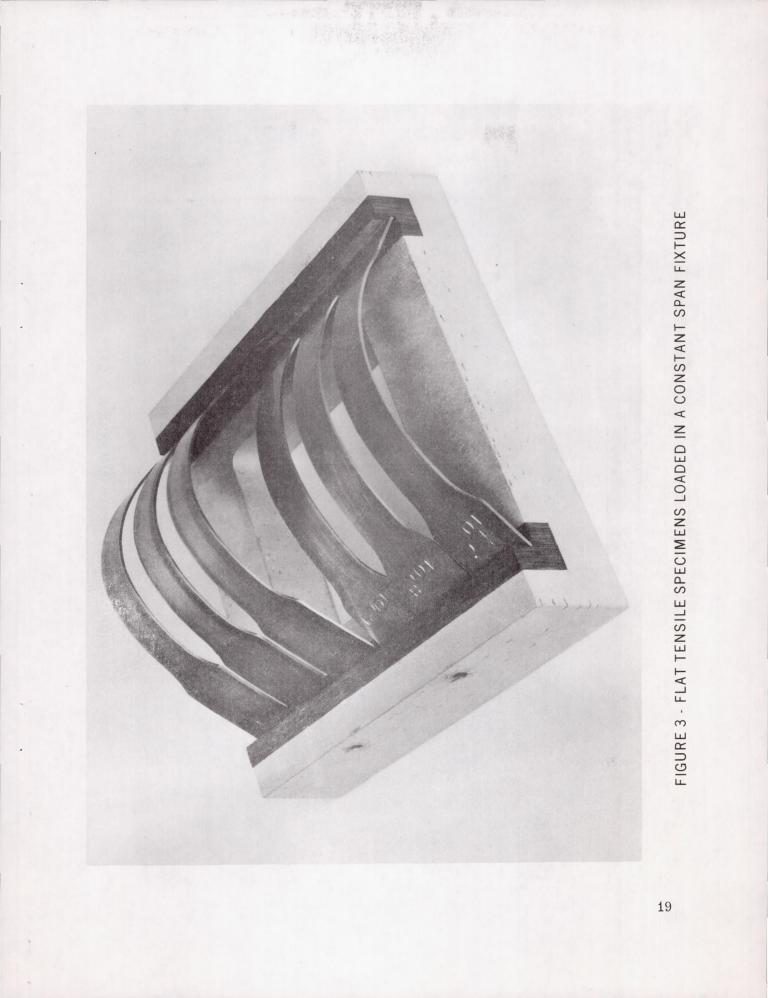


FIGURE 2 - ROUND TENSILE AND C-RING TYPE STRESS CORROSION TEST SPECIMENS



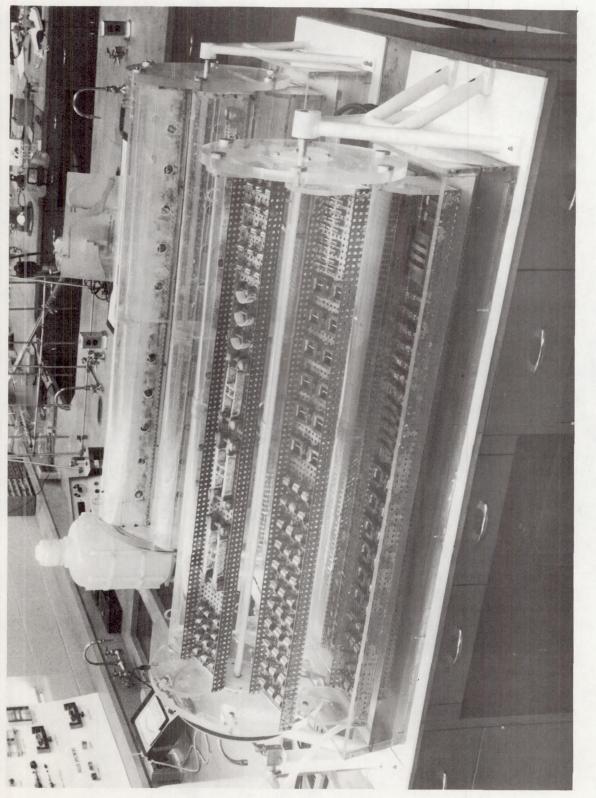


FIGURE 4 - ALTERNATE IMMERSION TESTER

#### APPROVAL

TM X-53910

#### STRESS CORROSION CRACKING EVALUATION OF SEVERAL PRECIPITATION HARDENING STAINLESS STEELS

By

#### T. S. Humphries and E. E. Nelson

The information in this report has been reviewed for security classification. Review of any information concerning Department of Defense or Atomic Energy Commission programs has been made by the MSFC Security Classification Officer. This report, in its entirety, has been determined to be unclassified.

This document has also been reviewed and approved for technical accuracy.

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