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Effect of ethanol on water permeability of controlled release films composed of ethyl cellulose and hydroxypropyl cellulose

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27 **Abstract**

28 Films for controlled release, composed of ethyl cellulose and hydroxypropyl cellulose were
29 prepared in various compositions and the effect of ethanol in the dissolution medium on the
30 water permeability of the films was studied using a modified Ussing chamber and tritiated
31 water. It was found that the effect of ethanol on the film permeability varied depending on the
32 composition of the films. The results were interpreted in terms of swelling of the ethyl
33 cellulose in the films, where the swelling increased with increasing ethanol concentration.
34 Thus, for films with low HPC content (non interconnected pores) the water permeability of
35 the films increased with increasing ethanol concentration as the diffusion through the ethyl
36 cellulose increased due to swelling. However, for films with higher HPC content (having
37 interconnected pores through the films) the permeability decreased, likely due to the swelling
38 of the ethyl cellulose blocking the pores. The interpretation of the results was supported by
39 dynamic mechanic analysis and SEM analysis.

40

41 **Keywords:** Ethyl cellulose, Hydroxypropyl cellulose, Permeability, Films, Controlled release

42

43 **1. Introduction**

44 Polymer film coatings are commonly used for controlling drug release from pellets and tablets
45 (Shah and Sheth, 1972; Donbrow and Samuelov, 1980; Sakellariou and Rowe, 1995;
46 Hyppoelae et al., 1996; Hjartstam and Hjertberg, 1999; Marucci et al., 2009;). In order to
47 modify the release, different ratios of water insoluble film forming polymer and water soluble
48 pore forming agent are used. Ethyl cellulose (EC) is a commonly used film forming agent
49 (Hyppoelae et al., 1996) because it has good film forming properties and is generally regarded
50 as non-toxic and non-allergenic (Hjartstam and Hjertberg, 1998; Marucci et al., 2009). Water
51 soluble cellulose derivates, such as hydroxypropyl cellulose (HPC) and hydroxypropyl
52 methylcellulose (HPMC) are commonly used as pore forming agents in controlled release
53 films (Sakellariou and Rowe, 1995; Hjartstam and Hjertberg, 1998; Marucci et al., 2009).
54 HPC has low toxicity, is biodegradable and has good film forming properties (Marucci et al.,
55 2009). Furthermore, HPC holds a great benefit over HPMC in industrial film spraying
56 processes. HPC can be co-dissolved with EC in ethanol, while HPMC requires the use of
57 other more hazardous solvents. Films composed of EC and HPC have been studied both with
58 regard to structure and permeability. It has been shown that EC and HPC phase separate in the
59 films (Sakellariou et al., 1986, 1988) and that the permeability of the films and the release rate
60 from formulations increases with increasing HPC content (Thombre et al., 1994; Marucci et
61 al., 2009). Furthermore, the permeability of the films as well as the release of the pore
62 forming HPC has been shown to be low below a critical HPC content, this being explained by
63 that at low HPC concentrations the pore forming network is not interconnected through the
64 film (Sakellariou et al., 1988; Marucci et al., 2009).

65 Recently, regulatory authorities have expressed concerns over the effects of alcohol on
66 extended release dosage (ER) forms, with potential dose dumping as a consequence (FDA,

67 2005; Meyer and Hussain, 2005). A regulatory framework regarding the effects of ethanol on
68 product performance and classification is currently being developed (Meyer and Hussain,
69 2005). Levina et al. (2007) have investigated the influence of ethanol on drug release from
70 HPMC based ER matrices, and reported that none of the investigated formulations exhibited
71 dose dumping when exposed to hydro-alcoholic solutions. However, Fadda et al. (2008)
72 investigated the influence of ethanol in the dissolution media on modified release tablets
73 coated with enteric methacrylic acid and methyl methacrylate ester copolymer. They found
74 that the influence of ethanol on the drug release was complicated, and concluded that several
75 tests are needed before making a decision on a formulations susceptibility to ethanol
76 impairment. Given the recent interest in the effect of ethanol on controlled release
77 formulations and the fact that EC (ETHOCEL™) is soluble in alcohol (Dow Cellulosics,
78 2005), it was considered interesting to investigate the effects of hydro-alcoholic solutions on
79 EC-HPC films for controlled delivery.

80 The aim of this study was to evaluate how the water permeability of EC films with varying
81 HPC content was affected by the presence of ethanol in the dissolution medium, and explain
82 the mechanism by which the permeability was affected. The water permeability of the films
83 was determined using a modified Ussing chamber, utilizing tritiated water as the diffusing
84 probe. Swelling of the films and changes in mechanical properties upon exposure to ethanol
85 solutions were studied using dynamic mechanic analysis (DMA). Finally, the structure of
86 films exposed to ethanol solutions and subsequently dried was studied using scanning electron
87 microscopy (SEM).

88

89 **2. Materials and methods**

90 *2.1 Materials*

91 The films in this study were prepared from ethyl cellulose (ETHOCEL™ 10, Dow Chemical,
92 USA) and hydroxypropyl cellulose (HPC LF, Ashland, USA). Ethanol used was of 95 % v/v
93 concentration (Kemetyl AB, Sweden), water used was ultra-pure deionized (Maxima USF,
94 Elga, UK). Tritiated water (PerkinElmer, USA) was used as the diffusant in the water
95 permeability measurements.

96

97 *2.2 Film preparation*

98 Films of EC with varying amount of HPC (0, 20, 30, 35, 40, 50 % w/w) were prepared as
99 follows. Desired amounts of EC and HPC were dissolved in ethanol (95 % v/v) to a total
100 polymer content of 6 % w/w. The solution was sprayed, using a moving nozzle (Schlick 970-
101 0, nozzle diameter 0.8 mm, Schlick, Germany), onto a rotating Teflon cylinder (In-house
102 manufactured, length 100 mm diameter 65 mm) in a controlled air flow. The process
103 conditions are presented in Table 1. The dry film was peeled off the cylinder and cut into
104 suitable geometries for permeability and DMA analysis. The thicknesses of the films were 70-
105 90 µm as measured with a micrometer (IP 54, Mitutoyo, Japan).

106

107 *2.3 Water permeability analysis*

108 The water permeability of the films was analyzed using a modified Ussing chamber with the
109 setup previously described (Hjartstam and Hjertberg, 1999). Briefly, a film sample was placed
110 between a donor and acceptor compartment. The film thickness was determined as the
111 average of five measurements. Initially 15 ml of dissolution medium with 0, 5, 20 or 40 % v/v
112 ethanol concentration was added to both the donor and the acceptor compartments and two
113 paddles were used to stir the dissolution medium at a speed of 200 rpm. After 5 minutes a

114 small amount of tritiated water (10 μ l, 400 kBq) was added to the donor compartment. At
115 specified times 500 μ l sample was taken from the acceptor compartment and was replaced by
116 the same amount of dissolution medium. The temperature was maintained at 37 °C through
117 the analysis. The samples extracted at the different times were weighed and analyzed in a
118 scintillator counter (1414 LSC, Win Spectral, Wallac). From the tritium activity registered in
119 the acceptor compartment at the different times, the amount of water that had diffused
120 through the film at each time could be determined, and thus the film permeability. Due to the
121 large difference in tritium activity between the donor and acceptor compartment any counter
122 diffusion was neglected.

123

124 *2.4 Dynamic mechanic analysis*

125 DMA measurements were performed using a Rheometrics RSAII (Rheometrics Scientific,
126 Piscataway, USA), equipped with an in house designed submersion cell (Edrud et al., 2003).
127 Samples were prepared to a width of 3 mm using a razor-edged punch. The sample thickness
128 was recorded as the average of three measurements. The effective initial sample length in the
129 DMA was 22-23 mm. The samples were mounted in the DMA and after about 3 minutes 40
130 ml of dissolution medium with 0, 5, 20 or 40 % v/v ethanol concentration was added.

131 The samples were analyzed in strain controlled stretching mode with a static force, keeping
132 the samples stretched, set to just exceed the amplitude of the harmonic dynamic force. The
133 deformation and the force response of the samples were monitored and from those parameters
134 the elastic modulus G' and the loss factor, $\tan(\delta)$, were calculated. Equilibrium value of $\tan(\delta)$
135 was taken as the average of the plateau values. The swelling of the samples was monitored as
136 the percent length change.

137

138 *2.5 Scanning electron microscopy*

139 Free polymer films with varying EC and HPC content were placed in beakers with dissolution
140 media containing 0, 5, 20 and 40 % ethanol for two days. The films were subsequently dried
141 and analysed using a scanning electron microscope (Quanta200, FEI, Czech Republic).

142

143 **3. Results and discussion**

144 *3.1 Water permeability analysis*

145 In order to investigate the influence of ethanol in the dissolution medium on water
146 permeability of EC-HPC films for controlled drug delivery, film samples were subjected to
147 permeability analysis. The analyses were conducted in dissolution media with different
148 ethanol concentrations using a modified Ussing chamber. The film samples were placed
149 separating the two compartments of the cell, and from the transport of tritiated water from the
150 donor to the acceptor compartment the volume of water that had diffused across the
151 membrane was calculated at each time. As seen in the exemplifying graph in Fig. 1, the
152 volume of water that had diffused across the membrane showed a linear dependence on time.
153 From the slope of the graphs the volume flow was calculated and the water permeability,
154 normalized versus film thickness, was determined as:

$$155 \quad P_N = \frac{J \cdot h}{A} \quad (1)$$

156 where P_N is the water permeability, J is the volume flow, h is the film thickness and A is the
157 area.

158 The permeability data (see Table 2 and Fig. 2) revealed that, in general, the water
159 permeability of the films increased with increasing HPC content. This is expected as HPC is
160 widely soluble both in water and in ethanol (Rowe et al., 2009), and thus should dissolve and
161 leave pores. Furthermore, the permeability was very low up to 20 % HPC content. This is in
162 agreement with a previous study on the permeability of EC-HPC films in non-ethanol
163 containing dissolution media (Marucci et al., 2009). For samples analyzed in the presence of
164 40 % v/v ethanol in the dissolution medium, some deviations from the trend were seen. If
165 instead looking at the permeability for a fixed HPC content and varying ethanol concentration
166 in the dissolution media, the permeability of films with low HPC contents (0 and 20 % w/w)
167 increases with increasing ethanol concentrations. However, at HPC contents higher than 20 %
168 w/w the permeability instead decreases with increasing concentration of ethanol in the
169 dissolution medium. Only one exception from the trend is seen, the permeability of the
170 samples with 30 % w/w HPC in 5 % v/v ethanol is higher than suggested by the trend.

171 We propose the following explanation for the changes in film permeability in ethanol
172 containing dissolution medium. At low HPC concentrations the dissolution of HPC will not
173 form a coherent pore network through the film, in accordance with previous studies (
174 Sakellariou et al., 1988; Marucci et al., 2009). Thus, the permeability of the films will be low.
175 Ethyl cellulose is however soluble in ethanol. It would be expected that upon increasing the
176 ethanol content in the dissolution medium, the solubility of EC is increased, with gelling and
177 in the extreme dissolution as a consequence. It is well known that the diffusion coefficient in
178 polymeric materials increases with decreasing polymer concentration (Masaro and Zhu,
179 1999). Thus, due to the swelling of the EC, the permeability of films with low HPC content is
180 expected to increase with increasing ethanol concentration in the dissolution medium. For
181 films with higher HPC content, the dissolution of HPC will lead to the formation of a
182 coherent pore network through the films, with dramatic increase in permeability as a

183 consequence. When the EC swells in the presence of ethanol the pores will become smaller,
184 and in the extreme cases filled with EC gel. As such, the permeability of the films with a high
185 HPC content is expected to decrease with increasing ethanol concentration in the dissolution
186 medium.

187

188 *3.2 Dynamic mechanic analysis*

189 In order to test the hypothesis that EC swells in ethanol containing dissolution medium, EC
190 film samples were subjected to DMA during submersion in dissolution media with varying
191 ethanol content. The length change, the elastic modulus, G' , and the loss factor, $\tan(\delta)$, of the
192 samples were recorded during the analyses. In Fig. 3 the percent length change is plotted
193 versus time after submersion for different concentrations of ethanol in the dissolution
194 medium. The film samples expanded more with increasing concentration of ethanol. The
195 effect was rather small for samples exposed to dissolution medium with 0 and 5 % v/v
196 ethanol, but more significant for 20 % v/v ethanol. For films swollen in dissolution medium
197 containing 40 % v/v ethanol the samples displayed a quick and accelerating length expansion
198 up to about 20 % (result not shown), at which point the analyses were terminated due to the
199 instrumental expansion limit. The elastic modulus of the film samples was relatively
200 unaffected by the addition of dissolution medium for ethanol concentrations in the range 0 to
201 20 % v/v. However, for the films exposed to the dissolution medium containing 40 % v/v
202 ethanol G' decreased dramatically even before reaching the expansion limit of the instrument.
203 Further decrease of G' would be expected if the measurements could have been continued.
204 The large decrease in G' clearly indicates that the samples undergo a transition from a solid to
205 a gelled state in the presence of 40 % v/v ethanol. Thus, the dramatic change in sample length

206 is most likely a combination of swelling of the film and irreversible strain under the small
207 force applied in the DMA.

208 In Fig. 4 the $\tan(\delta)$ values for EC films in dissolution media with different ethanol
209 concentrations are shown. The $\tan(\delta)$ values are equilibrium values for samples exposed to
210 dissolution media with ethanol concentration ranging from 0 to 20 % v/v. For the films
211 exposed to the dissolution media with 40 % v/v ethanol, the reported $\tan(\delta)$ value is acquired
212 from the last measurable values before reaching the instrumental expansion limit. It is seen
213 that the loss factor is small for low concentrations of ethanol, increasing slightly as the
214 ethanol concentration increases. However, for higher ethanol concentrations, the increment is
215 large. The loss factor correlates to the elastic and viscous, G'' , modulus as (Craig and
216 Johnson, 1995):

$$217 \quad \tan(\delta) = \frac{G''}{G'} \quad (2)$$

218 As such $\tan(\delta)$ can be regarded as a measurement of how much of a viscous liquid character a
219 material has, as compared to an elastic solid. An increase in $\tan(\delta)$ is equivalent with that a
220 sample dissipates more of the applied energy of deformation as frictional heat, rather than
221 storing the energy as in purely elastic deformation. From the increase in $\tan(\delta)$ with increasing
222 ethanol concentration in the dissolution medium, it can be concluded that the presence of
223 ethanol increases the mobility of the EC polymer chains, causing the samples to dissipate
224 more of the applied deformation energy as heat. The increase of $\tan(\delta)$ with increasing
225 concentration of alcohol is coherent with the increase in sample length, discussed above.

226

227 *3.3 Scanning electron microscopy*

228 To investigate if any difference could be detected in the structure of films exposed to
229 dissolution media with varying ethanol content, film samples composed purely of EC and
230 containing 35 % w/w HPC were submerged in dissolution media with different ethanol
231 concentrations for two days. The samples were subsequently dried and analysed using SEM.
232 If the EC was unaffected by the dissolution medium, pure EC films would be unaffected by
233 the treatment and no change in structure should be detected. For the analysed EC-HPC films
234 the HPC should dissolve, leaving a porous network. As for pure EC films, the structure should
235 not be altered by the treatment if EC was not affected by the dissolution medium. However, if
236 the EC swelled in the dissolution medium, altered structures would be expected both for pure
237 EC films and for EC-HPC films.

238 As seen in Fig. 5 A-D, the structure of pure EC films is clearly affected by the ethanol
239 concentration in the dissolution medium. The films that were exposed to dissolution medium
240 containing 20 and 40 % v/v ethanol exhibit a different surface than films exposed to
241 dissolution medium with less ethanol, probably due to swelling and subsequent drying. For
242 HPC containing films it is hard to draw any certain conclusions due to the inherent
243 heterogeneity of the films (Fig 5 E-H). However, there is some resemblance between the pure
244 EC film and the HPC containing film submerged in dissolution medium containing 40 % v/v
245 ethanol. The results from the SEM analysis support the hypothesis that the performance of
246 EC-HPC films in ethanol containing dissolution medium is influenced by the ethanol
247 concentration through the swelling, and at sufficiently high ethanol concentrations the gelling,
248 of EC.

249

250 **4. Conclusion**

251 In this study it was shown that the water permeability of EC-HPC films for controlled drug
252 delivery was influenced by the concentration of ethanol in the dissolution medium. Ethanol
253 increased the water permeability for no and low amount of HPC in the films, but reduced the
254 permeability at higher HPC content. The effects on the permeability can be explained by that
255 the EC swells in the presence of ethanol, leading to an increased diffusion through the EC in
256 the films, but more importantly decreasing the size of the pores left by the dissolved HPC.
257 This explanation is supported by results from DMA and SEM analysis. The findings are
258 interesting and of great relevance, as the effect of ethanol on controlled release formulations is
259 a current concern expressed by regulatory authorities. The results in this study indicate that
260 the performance of EC-HPC films in controlled delivery applications is at risk with regard to
261 co-ingestion with ethanol. The results here presented mainly showed on a decrease in water
262 permeability. However, given that in most formulations there will be an osmotic pressure
263 difference over the films and that drug molecules are considerably larger than water
264 molecules; our results should not be seen as a proof that dose dumping will not occur.
265 Interesting further studies would be to investigate how the concentration of ethanol in the
266 dissolution medium affects the performance of coated pellets and tablets with different
267 osmotic pressure.

268

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330

331 **Figure legends**

332

333 Fig. 1. Exemplifying plot of the volume of water having diffused across the film, here for a
334 sample with 35% HPC and a thickness of 86 μm in dissolution medium with 5 % ethanol.

335

336 Fig. 2. Plot of the water permeability, normalized versus film thickness, for EC films with
337 varying HPC content, in dissolution media with the following ethanol concentrations (■) 0,
338 (○) 5, (□) 20 and (●) 40 %. Error bars indicate one standard deviation.

339

340 Fig. 3 The percent length change over time for EC films during dynamic mechanic analysis in
341 dissolution media with different ethanol concentration. Dashed lines indicate min/max, $n = 2$,
342 where not visible the deviations are too small to be displayed.

343

344 Fig. 4. The loss factor of EC films in dissolution media with different ethanol concentration,
345 detected using DMA. The values are equilibrium values for ethanol concentrations ranging
346 from 0 to 20 % and the last value before reaching the expansion limit of the instrument for 40
347 % ethanol. Insert is magnification of the region with ethanol concentrations of 0-20 % Error
348 bars indicate min/max, $n = 2$.

349

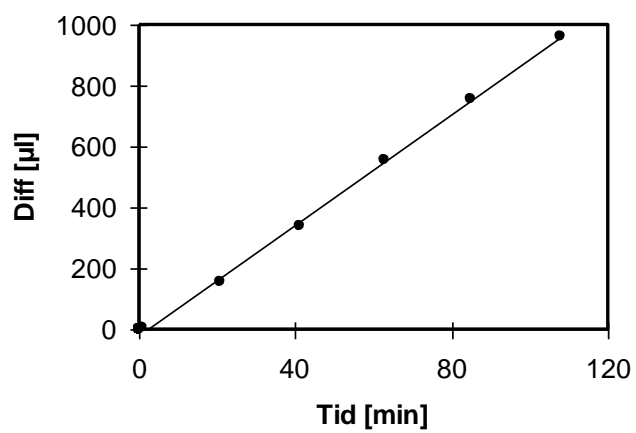
350 Fig. 5. SEM images of pure EC films (A-D) and films containing 35 % HPC (E-H) after being
351 submerged in dissolution media with different ethanol concentrations (A, E = 0 %;
352 B, F = 5 %; C, G = 20 % and D, H = 40 %) for two days. Scale bar is 20 μm .

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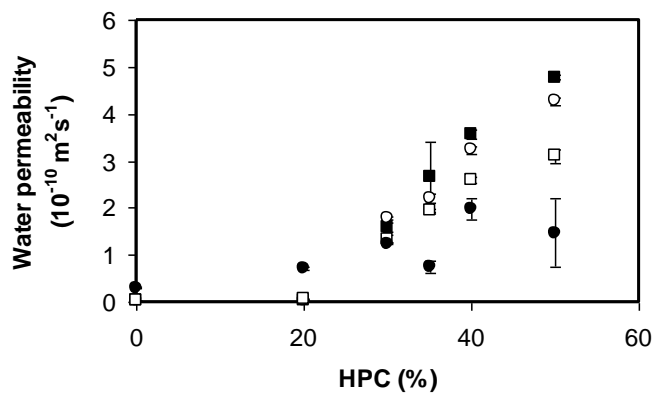
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356 Figure 1.
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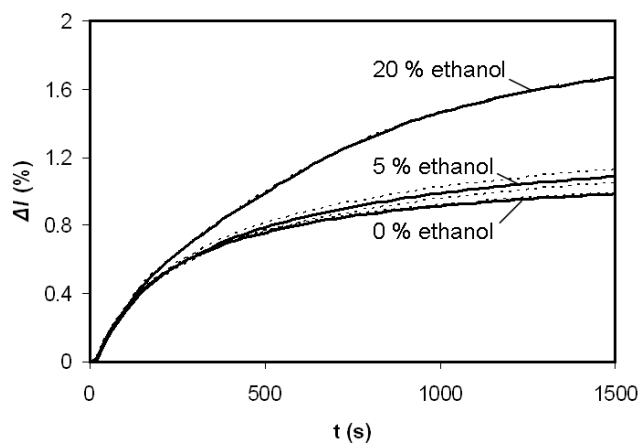
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360 Figure 2.
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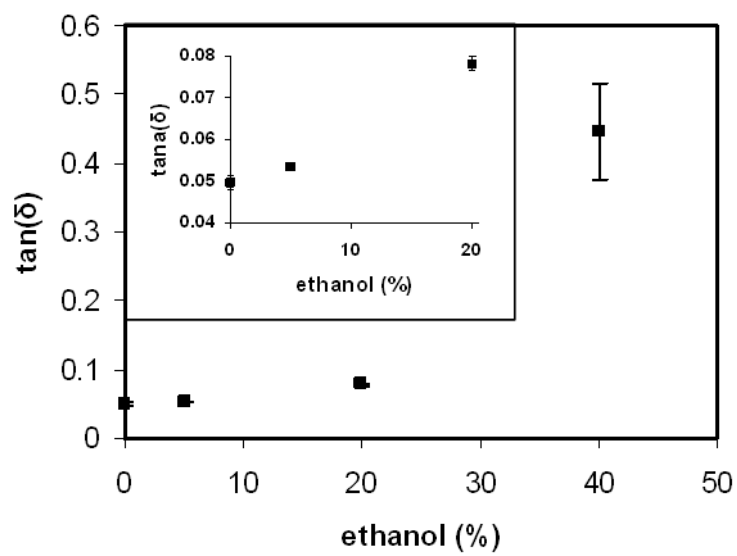
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364 Figure 3.
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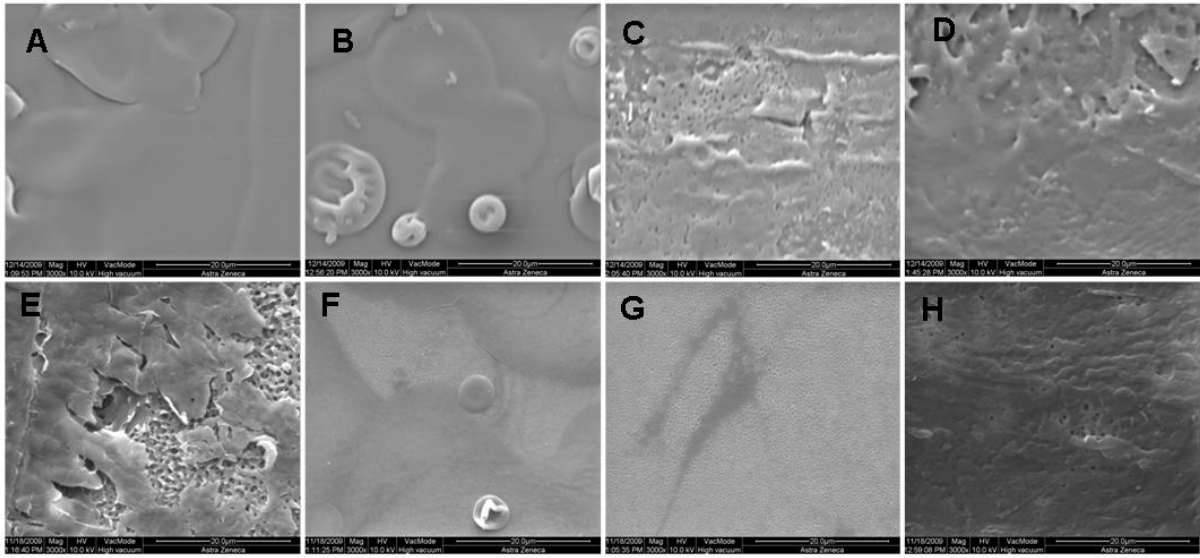
369 Figure 4.
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Figure 5.



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380 Table 1. Process parameters used in the preparation of films.

Process parameter	value
Inlet air temperature	72 °C
Outlet air temperature	45-47.5 °C
Fluidizing air flow rate	40 Nm ³ /h
Atomizing air pressure	2.0 bar
Spraying rate	10 g/min

381

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385

386 Table 2. Water permeability normalized versus film thickness ($10^{-12} \text{ m}^2\text{s}^{-1}$) for EC films with
387 varying HPC content, in dissolution media with different ethanol concentrations. One
388 standard deviation within parentheses, n = 2-5.

HPC (%)	0% EtOH	5% EtOH	20% EtOH	40% EtOH
0	2.3 (0.27)	2.40 (0.04)	3.9 (0.18)	30 (1.3)
20	3.4 (0.34)	3.61 (0.05)	7 (1.0)	70 (3.4)
30	158 (9.9)	177 (3.1)	134 (8.7)	124 (2.0)
35	270 (73)	221 (9.8)	193 (3.7)	70 (13)
40	356 (9.0)	324 (8.1)	260 (5.6)	200 (22)
50	478 (5.0)	427 (7.1)	310 (15)	150 (74)

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