Student Paper

InGaAs n-MOS devices integrated using ALD-HfO₂/metal gate without surface cleaning and interfacial layer passivation

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The aggressive scaling of Si CMOS device has called for high κ dielectrics and metal gates. However, the phonon issue related to the high κ gate dielectrics has lead to degraded channel mobility. Extensive research activities are now being taken on channel materials, such as InGaAs, with mobility higher than that of Si. In_{0.53} Ga_{0.47}As, lattice matched to InP, and In_{0.15}Ga_{0.85}As, strained growth on GaAs, have been used as a backbone for almost all the high-speed electronic devices in high electron mobility transistor (HEMT) with very high cutoff frequency. Oxide gates may improve device performance, e.g. in reducing gate leakage and increasing I_{on}/I_{off} ratio. Atomic layer deposited (ALD) high κ dielectric HfO₂ films on air-exposed In_{0.53}Ga_{0.47}As/InP and In_{0.15}Ga_{0.85}As /GaAs were found to have an atomically sharp interface, free of arsenic oxides, which is believed to attribute to the Fermi level un-pinning. The energy band parameters were measured for HfO₂/In_{0.53}Ga_{0.47}As and HfO₂/In_{0.15}Ga_{0.85}As, respectively.

Figure 1 (a)-(b) show the HR-TEM for air-exposed $In_{0.53}Ga_{0.47}As$ and ALD-HfO₂/ In_{0.53}Ga_{0.47}As, respectively. The detailed chemical state and distribution, including arsenic or arsenic oxides are studied using XPS, as shown in Fig. 2 (a)-(c). After the ALD-HfO₂ growth, As₂O₃ was removed from the oxide/InGaAs interface. No arsenic oxides (As₂O₃ or As₂O₅) were found to be on top of the as-grown ALD-HfO₂. The detection of Ga₂O₃, In₂O₃, and In(OH)₃ at the HfO₂-InGaAs interface was made possible with HR-XPS using synchrotron radiation or with AR (angle-resolved)-XPS. An abrupt transition from InGaAs to ALD-HfO₂ with a thin interfacial layer was observed using HR-TEM. The similar results were observed for air-exposed In_{0.15}Ga_{0.85}As and ALD-HfO₂/In_{0.15}Ga_{0.85}As, respectively. The removal of the arsenic oxides from HfO₂/InGaAs heterostructures during ALD process ensures the Fermi level unpinning, which was observed in the C-V measurements.

As illustrated in Fig. 3, C-V curves for $In_{0.53}Ga_{0.47}As$ show accumulation and inversion. The C-V at 1 kHz shows a similar behavior as the C-V of Si MOS diodes at 1 or 10Hz, namely the occurrence of inversion. In contrast, C-V curves show much larger frequency dispersion for $In_{0.15}Ga_{0.85}As$ sample (Fig. 4).

Low leakage current densities of ~ 10^{-7} to 10^{-9} A/cm² at electrical fields less than 4 MV/cm (Fig.5) were measured for the 7.8 nm HfO₂ on In_{0.53}Ga_{0.47}As/InP, annealed at 375°C for 60 min in forming gas. The current transport of the MOS device can be explained by the Fowler-Nordheim tunneling mechanism. Similar transport properties were measured on In_{0.15}Ga_{0.85}As /GaAs. The conduction-band offset of ~ 1.8 and 1.48eV were determined for HfO₂/In_{0.53}Ga_{0.47}As and HfO₂/In_{0.15}Ga_{0.85}As, respectively. The valence band offset at the HfO₂/In_{0.53}Ga_{0.47}As and HfO₂/In_{0.15}Ga_{0.85}As interface was determined using XPS to be ~2.9 eV and ~2.65 eV, respectively, as shown in Fig. 6. The energy band parameters determined from XPS and the transport measurement are listed in Fig. 7.

There is no surface cleaning and interfacial passivation layer prior to the ALD-HfO₂. However, the oxide/InGaAs interface is atomically sharp without the existence of arsenic oxides, strongly indicating self-cleaning of the ALD process. Excellent well-behaved J-E_G and C-V characteristics of ALD-HfO₂/In_{0.53}Ga_{0.47}As/InP have been demonstrated in this work.



Fig. 1 Cross-sectional HRTEM images of the airexposed $In_{0.53}Ga_{0.47}As$ sample (a) prior to (b) after deposition of HfO₂. Most of the native oxide (3.6nm) was annihilated after the ALD growth. The thickness of HfO₂ is about 7.8nm.



Fig. 2 (a) In $3d_{5/2}$, (b)As 3p, (c) Ga 3p, and core-level spectra of 7.8 nm thick HfO₂ film grown on In_{0.53}Ga_{0.47}As. (I) and (II) are designated for regions at the surface and in the bulk of air-exposed In_{0.53}Ga_{0.47}As.(III). (V), and (VI) are those at the surface, in the bulk of the ALD oxide, and at the oxide/In_{0.53}Ga_{0.47}As Interface.



Fig. 3 C-V curves of TiN/ALD-HfO₂ (7.8 nm) /In_{0.53}Ga_{0.47}As MOS diode under frequencies from 1kHz to 1 MHz, with inset showing the C-V curve measured at 1 MHz in comparison with the theoretical curve (D_{it} =0). The dielectric constant deduced from the 10 kHz CV curve is about 15.



Fig. 4 C-V curves of a MOS diode consists of Au/Ti/ALD-HfO₂ (8.3 nm)/In_{0.15}Ga_{0.85}As. The dielectric constant deduced from the 10 kHz CV curve is about 9.



Fig. 5 Current density (J) as a function of gate electrical field (E_G) for the TiN/ALD-HfO₂/In_{0.53}Ga_{0.47}As MOS diode of t_{ox}=7.8 nm, with the inset showing the electrical data in terms of $\ln(J/E_{ox}^2)$ versus $1/E_{ox}$. The linear region at high E_{ox} for both forward (+) and reverse (-) biase of inset is where the FN tunneling occurs.



Fig. 6 The valence band offsets ΔE_V are defined as the energy difference between the valence band maximum (VBM) of the HfO₂ and InGaAs.

ALD-HfO₂/In_xGa_{1-x}As



Fig. 7 Energy band parameters of ALD-HfO₂/ $In_xGa_{1-x}As$ (x = 0.15 and 0.53)