



Temperature dependence of infrared reflectance spectra of InN

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Abstract

To investigate both the optical and electrical properties of InN, we have measured the infrared reflectance spectra of InN thin films and performed the fitting analyses of the infrared spectra to obtain not only phonon frequencies and the damping factors but also the carrier concentration of InN. In this paper, we extend the aim of those analyses to the electron mobility and demonstrate that the temperature dependence of the electron mobility can be discussed using the infrared reflectance spectra analyses.

Keywords: InN, infrared reflectance spectra, electric mobility ;

1. Introduction

III-V nitride semiconductors and their mixed crystal semiconductors are used for promising high intensity light emitting diodes (LED) and laser diodes (LD). Especially, InN has a matched bandgap for the fiber communication wavelength, and it also has small effective mass and high electron mobility which are required for high speed electronic devices. However, it is difficult to grow bulk InN crystals. The high quality crystalline InN thin films have been fabricated by the MBE or MOCVD methods, but the lattice mismatch between InN and the substrate (normally Al₂O₃) requires the buffer layer. This layer is usually made of the low temperature growth InN or GaN, which means that the InN sample at least consists of three layers, InN crystalline layer, buffer

layer, and substrate. Then, the electric properties of the InN sample are sometime affected by this buffer layer and it is difficult to separate the electric properties of InN crystalline layer from those of the buffer layer. Then, the infrared (IR) reflectance measurements have been investigated as one of the complementary measurement methods for obtaining the carrier concentration of the InN crystalline layer [1-3]. In this method, since the dielectric constant is described by the phonons and plasmon which are characterized by each layer, the reflectance spectra are analyzed by the fitting calculations based on the appropriate layer model.

In this paper, we extend the aim of those analyses to the electron mobility and demonstrate that the

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temperature dependence of the electron mobility can be discussed using the infrared reflectance spectra analyses.

2. Experiment

All InN samples were grown on the sapphire substrates by the MOVPE method [4]. The thicknesses of all samples are about 0.2 ~ 0.5 μm . The carrier concentration measurements by using the normal electric methods are carried out by the van der Pauw method [5,6].

The IR reflectance spectra and their temperature dependence have been measured at the IR beamline (BL6B) of UVSOR-II (Institute for Molecular Science, Okazaki, Japan) by using both Martin-Puplett type Fourier Transform Far IR (FTFIR) and Michelson type FTIR spectrometers. The wavenumber range and resolution are 50 ~ 1500 cm^{-1} and 1 ~ 4 cm^{-1} , respectively, and the temperature range is 17 ~ 300 K.

The IR spectra have been analyzed by the numerical simulations. The fitting calculations are carried out by using Eq. (1) to (3) [7]. The complex dielectric constant in the IR region is usually described as the LO and TO phonons frequencies (ω_L , ω_T) and their damping factor Γ . However, the contribution of the electric carriers must be considered when the plasma frequency is located in the IR region. In such a case, the dielectric constant including the coupling between plasmon and phonons are represented by Eq. (1),

$$\varepsilon(\omega) = \varepsilon_\infty \left[1 + \frac{\omega_L^2 - \omega_T^2}{\omega_T^2 - \omega^2 - i\omega\Gamma} - \frac{\omega_p^2}{\omega(\omega + i\gamma)} \right] \quad (1)$$

where ω_p and γ are plasma frequency and damping factor of the plasmon, respectively. The ω_p and γ are also described by Eq. (2) and (3),

$$\omega_p^2 = \frac{4\pi n e^2}{m^* \varepsilon_\infty} \quad (2)$$

$$\gamma = \frac{e}{m^* \mu} \quad (3)$$

where m^* , n and μ are effective mass, carrier concentration, and mobility of the carriers, respectively. The dielectric constant of the sapphire

substrate is simply represented by the contributions of the phonons. Then, by using the transfer matrix method, the theoretical total reflectance including the multiple reflection based on any layer model can be described, and the fitting analyses to the observed reflectance spectra are available using the physical values ω_L , ω_T , ω_p , γ , Γ , ε_∞ as the fitting parameters.

3. Result and discussion

Fig. 1 shows the typical reflectance spectrum of the InN thin film at 300K and the fitting curve. The carrier concentration measured by Van der Pauw method is $9.8 \times 10^{18} \text{cm}^{-3}$. The fitted curve shown in Fig. 1 is based on the two layers model which means no consideration of the buffer layer in the optical point of view and in agreement with the observed spectrum. The fitting parameters are $\varepsilon_\infty = 8.3$, $\omega_L = 581 \text{cm}^{-1}$, $\omega_T = 478 \text{cm}^{-1}$, $\omega_p = 836 \text{cm}^{-1}$, $\gamma = 265$, $\Gamma = 2$ and h (thickness of crystalline InN layer) = 200 nm. Since the effective mass is still unknown for InN, we suppose it is $0.1m_0 \sim 0.15m_0$ according to the theoretical prediction [8], where m_0 is the static mass of the electron. The carrier concentration derived from IR reflectance spectrum is estimated to $6.5 \sim 9.7 \times 10^{18} \text{cm}^{-3}$, which is a little smaller than Van der Pauw result. However, from this result, it is concluded that the electric properties of this InN sample are not affected by those of the buffer layer.

From Eq. (3), the electron mobility can be derived from the IR reflectance measurement by using the damping factor γ of the plasmon which is one of the

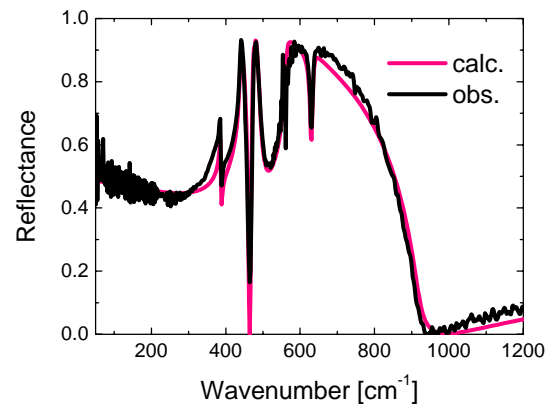


Fig. 1: Typical IR reflectance spectrum of InN thin film ($9.8 \times 10^{18} \text{cm}^{-3}$) and fitting curve.

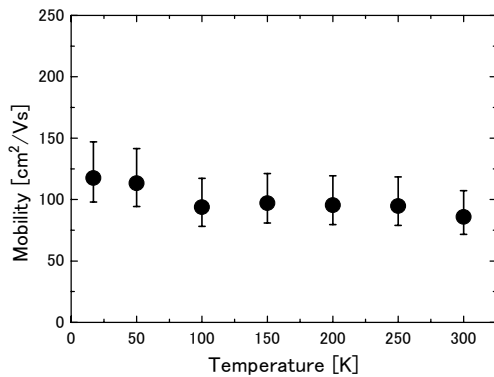


Fig. 2: The electron mobility of the InN crystalline layer derived from the infrared reflectance measurements as a function of the temperature.

fitting parameters in Eq. 1 and the effective mass m^* . Fig. 2 shows the electron mobility of the InN crystalline layer derived from the IR reflectance measurements as a function of the temperature. The electric mobility of InN is almost constant in the whole temperature range, and is in agreement with the results of the Hall effect measurement. The inverse of the total electric mobility is described as the summation of the inverse mobilities of the scattering processes, and it is known that both the polar phonon scattering and the deformation potential scattering have $T^{-2/3}$ temperature dependences, and the ionized impurity scattering has $T^{3/2}$ dependence [9,10]. However, it is difficult to explain such a constant behavior in the wide temperature range. It seems that the additional contributions, for example, the space charge scattering should be considered. This analysis is still under investigation.

4. Conclusion

We have measured the infrared reflectance spectra of the InN thin films at the temperature range of 11 to 300 K. Since the damping factor of the plasmon which is one of the fitting parameters of the infrared reflectance spectrum analysis is reciprocal proportion to the electron mobility, we demonstrate that the electron mobilities of the InN crystalline layers and their temperature dependence can be discussed using the infrared reflectance spectra analyses. The electron mobilities of the InN crystalline layers are almost independent on the temperature in the range of 11 to

300 K, and it is difficult to explain this flatness by using both phonon and impurity scattering processes.

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References

- [1] Y. Ishitani, K. Xu, W. Terashima, N. Hashimoto, M. Yoshitani, T. Hara, and A. Yoshikawa, *phys. stat. sol. (c)* **0**, 2838 (2003).
- [2] J. S. Thakur, G. W. Auner, D. B. Haddad, R. Naik, and V. M. Naik, *J. Appl. Phys.* **75**, 4795 (2004).
- [3] K. Fukui, Y. Kugumiya, N. Nakagawa, and A. Yamamoto, *phys. stat. sol. (c)* **3**, 1879 (2006).
- [4] A. Yamamoto, K. Sugita, H. Takatsuka, A. Hashimoto, and V. Yu. Davydov, *J. Cryst. Growth* **261**, 275 (2004).
- [5] L. J. van der Pauw, *Philips Res. Repts.* **13**, 1 (1958).
- [6] L. J. van der Pauw, *Philips Tech. Rev.* **20**, 220 (1958).
- [7] M. Kuball, *Surf. Interface. Anal.* **31**, 98 (2001).
- [8] J. Wu and W. Walukiewicz., *Superlattice. Microst.* **34**, 63 (2003).
- [9] J. Bardeen, W. Shockley, *Phys. Rev.* **80**, 72 (1950).
- [10] W.A. Harrison, *Phys. Rev.* **100**, 255 (1956).