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Effect of coarse particle volume fraction on the yield stress and thixotropy of cementitious materials

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Abstract

In order to help modelling the yield stress of fresh concrete, we study the behavior of suspensions of coarse particles in a thixotropic cement paste. Our aim is to relate the yield stress of these mixtures to the yield stress of the suspending cement paste, to the time passed at rest, and to the coarse particle volume fraction. We present here procedures that allow for (i) studying an homogeneous and isotropic suspension, (ii) comparing the yield stress of a given cement paste to that of the same cement paste added with particles, (iii) accounting for the thixotropy of the cement paste. We observe that the yield stress of these suspensions of cement paste with coarse particles follows the very simple Chateau-Ovarlez-Trung model [1], consistently with the experimental results of Mahaut et al. [2] obtained with many different particles and suspending yield stress fluids. This consistency between the results obtained in various yield stress fluids shows that the yield stress of the suspension does not depend on the physicochemical properties of the suspending yield stress fluid; it only

depends on its yield stress value. This shows that studies of suspensions in model yield stress fluids can be used as a general tool to infer the behavior of fresh concrete. Moreover, we show that the thixotropic structuration rate of the interstitial paste (its static yield stress increase rate in time) is not affected by the presence of the particles. As a consequence, it is sufficient to measure the thixotropic properties of the constitutive cement paste in order to predict the thixotropic structuration rate of a given fresh concrete. This structuration rate is predicted to have the same dependence on the coarse particle volume fraction as the yield stress.

Key words: A. Fresh Concrete, A. Rheology, D. Aggregate, D. Cement Paste, E. Modeling

1 1 Introduction

- 2 Knowing and predicting the flow properties of fresh concrete is a major issue of
- 3 concrete casting and concrete mix-design. Basically, fresh concretes exhibit a
- yield stress [3] and have a solid viscoelastic behavior below this yield stress [4];
- 5 above the yield stress they behave as liquids, and their steady flow behavior
- 6 is usually well represented by a Bingham or a Herschel-Bulkley model [3,5].
- ⁷ However, fresh concrete is also known for its evolving rheological behavior.
- 8 Even, if its steady state flow may be described by the above models, the
- o characteristic time to reach this steady state flow may be rather long [6–9] and,
- after a long time of rest, the stress that has to be applied to induce a flow may
- be one or two orders higher than the dynamic yield stress measured when the

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material stops flowing i.e. it is thixotropic [10–12]. The static yield stress and its increase rate at rest are actually the most important rheological quantities 13 in terms of potential applications in the case of SCC [13]: it has been shown recently that they determine the formwork pressure [9,14–17], the stability 15 vs. sedimentation of the coarsest particles in SCC [18] and the occurrence 16 of distinct layer casting [19]. As a consequence it is of high importance to understand the role of the various components of a given concrete on this yield 18 stress and its evolution at rest. Moreover, measuring directly the rheological 10 properties of fresh concrete is very difficult [20]; any model providing the yield stress of concrete as a function of the suspending cement paste properties and 21 the properties and the volume fraction of sand and aggregates would then 22 prove to be very useful.

The link between concrete mix-design and its flow properties in the fresh state may be studied in the more general framework of suspensions rheophysics [12]. Actually, fresh concretes belong to the wide family of dense suspensions, which often involve a broad range of particle sizes [21] and can be found in 27 many industrial processes (drilling muds, foodstuff transport...) and natural phenomena (debris-flows, lava flows...). All these materials share the same complex features, which originate from the great variety of interactions between the particles (colloidal, hydrodynamic, frictional, collisional...) and of 31 physical properties of the particles (volume fraction, sensitivity to thermal 32 agitation, shape...) affecting the material behavior [22,12]. Basically, in the absence of a contact network of noncollodial particles (i.e. for moderate non-34 collodial particles volume fraction), the yielding behavior originates from the 35 colloidal interactions which create a jammed network of interacting particles [5,12]. Structuration at rest (which has nothing to do with setting) is observed

in many aggregating suspensions and colloidal glasses [12]: the evolution of the behavior of aggregating suspensions at rest may be explained by a progres-39 sive and reversible formation of a solid structure by flocculation. Within this frame, the problem of the influence of coarse particles on the behavior of fresh 41 concrete may be seen more generally as the problem of the influence of non-42 colloidal particles on the properties of yield stress fluids. It is thus of high importance to clarify the cases where the rheological properties of a suspension of coarse particles in a yield stress fluid depend only on the rheological properties of the suspending fluid and on the coarse particle volume fraction and size distribution. This should provide results applicable to any particles in any yield stress fluid, in particular to sand and aggregates suspended in a 48 cement paste. It would allow the use of results obtained in studies performed e.g. with noncolloidal particles in clay dispersions to predict the behavior of a mortar or a concrete. On the other hand, any departure from generic results 51 would be the result of specific physicochemical interactions in the suspensions 52 (or specific slippage at the paste/particle interface), as e.g. the adsorption of a fraction of the superplasticizer of the cement paste on the fine aggregates 54 in SCC [23], and would justify for each material a specific study with the particular particles and particular paste involved. In this paper, we test the idea of fresh concrete being a suspension of particles in a yield stress fluid. We compare the results obtained when suspending particles in a cement paste to those recently obtained in a broad range of materials (suspensions of various particles in various yield stress fluids) by Mahaut et al. [2].

The influence of the aggregates on the rheological properties of fresh concrete has been studied theoretically and experimentally by de Larrard [24], de Larrard and Sedran [25], Geiker *et al.* [26], Erdogan [27] and Toutou and Roussel

[28]. De Larrard [24] has proposed a model in which concrete is looked as a granular mix in a water suspension. Then, the overall yield stress is the macroscopic counterpart of the friction between solid particles and is interpreted as the stress one needs to apply in order to overcome the intergranular 67 contact forces. The overall yield stress can be estimated from the value of the solid volume fraction and close packing density of the different components of the granular mixture. However, if this model may help understanding the properties of fresh concrete displaying "ordinary" rheology, it is unadapted to 71 the description of modern fluid concrete which contains less coarse particles and where friction between the grains is negligible [29]. Geiker et al. [26] have 73 studied experimentally the effect of coarse particle volume fraction on the rhe-74 ological properties of SCC. They have measured the steady-state flow curves of various materials thanks to the procedure developed in [8]; the dynamic yield stress was then extracted from a fit of the flow curve with a Bingham model. It was found to increase strongly with the coarse particle volume fraction. To model the behavior, they assume that the effect of aggregates on concrete rheological properties can be studied by looking to concrete as a suspension of 80 coarse particles in the mortar seen as a continuum medium. Their experimen-81 tal data are compared to a model proposed by Nielsen [30] which provides the yield stress of a suspension of ellipsoids as a function of the volume fraction of particles and of the aspect ratio. This model rests on heuristic rules which are not rigorously justified. Nevertheless, the theory can be calibrated in order to accurately describe the data of Geiker et al. [26]. Erdogan [27] have studied the effect of aggregate particle shape and surface texture on rheological 87 properties of fresh concrete. Artificial aggregate particles of regular geometric shapes (spheres, cubes and rectangular prisms) with similar centimeter size and volume were prepared. A Couette-vane rheometer (ICAR) was used to

measure the dynamic yield stress as the low shear rate limit of a flow curve. In addition, slump tests were performed. Erdogan has observed that the yield 92 stress increases slightly when the coarse particle volume fraction increases. This trend is confirmed by slump experiments: the slump value was clearly a 94 decreasing function of the coarse particle content, whatever the shape of the particles is. Toutou and Roussel [28] have studied the flow behavior of mortars, considered as suspensions of sand in a cement paste, and the flow behavior of concretes, considered as suspensions of gravel in a mortar. In both cases, the influence of the coarsest inclusions volume fraction on the suspending paste properties was investigated. The dynamic yield stress was extrapolated from 100 the measured flow-curves. The yield stress of the mortar was found to increase 101 with the sand volume fraction. However, at low volume fraction (below 20%) 102 yield stresses of mortars were found to be lower than the yield stress of the 103 suspending cement paste. Toutou and Roussel [28] attributed this feature to 104 the increased deflocculation of the cement paste due to the presence of the 105 inclusions during mixing of the suspension, in agreement with Williams et al. 106 [31]. The yield stress of concrete was also found to increase with the gravel 107 volume fraction. However, Toutou and Roussel [28] found that adding gravel 108 at a given volume fraction to a mortar yields a much larger increase of the 109 yield stress than adding sand at the same volume fraction to a cement paste. 110

The influence of coarse particles on the rheological properties of other yield stress fluids has been studied by Coussot [32] and Ancey and Jorrot [33]. Ancey and Jorrot [33] have suspended coarse particles within a clay dispersion.

They measured the yield stress of the suspension by means of a slump test.

They showed that for well-graded particles, the suspension yield stress does not depend on the particle characteristics (diameter, material) and that the yield

stress diverges when the solid volume fraction value tends toward the maxi-117 mum packing fraction. Of course, when the coarse particles are polydisperse, 118 the value of the maximum packing fraction depends on the size distribution of 119 the particles, and the yield stress diverges for values of the solid volume frac-120 tion depending on this distribution. They observed sometimes that, for low 121 reduced solid volume fraction, the yield stress can be a decreasing function of 122 the solid volume fraction of the coarse particle. This effect was ascribed to a 123 depletion phenomena: the clay particles are supposed to be expelled from the 124 suspending fluid in the neighborhood of the coarse particles which are then 125 embedded in a shell of pure water. Then, they cannot contribute to the over-126 all yield stress: they behave as voids. Note that this depletion mechanism is 127 specific to the suspending yield stress fluid studied by Ancey and Jorrot; thus, 128 it cannot be used to predict what happens when the particles are suspended 129 in another yield stress fluid. 130

The few existing experimental studies provide very different results; e.g., when 131 particles having the same shape (spheres) are embedded at a volume fraction 132 ϕ corresponding to 70% of the maximum packing fraction ϕ_m in a paste, 133 Geiker et al. [26] find that the yield stress of the paste is increased by a factor 134 50 when the paste is a mortar, whereas Erdogan [27] finds that it is increased 135 by only a factor 1.3 when the paste is also a mortar, and Ancey and Jorrot [33] 136 find that, when the paste is a clay dispersion, the yield stress is increased by a 137 factor 2. Other surprising discrepancies are shown by Toutou and Roussel [28]: they find that for sand suspended at 70% of ϕ_m in a cement paste the yield 139 stress is increased by a factor 8 whereas for gravel (of comparable shape and 140 dispersity) suspended at 70% of ϕ_m in a mortar, it is increased by a factor 25. 141 As pointed out above, if rigid noncolloidal particles of a given shape and dis-

persity were to interact only rheologically with the suspending paste, we would 143 expect all the results to be roughly consistent as they should not depend on the 144 paste physicochemical nature. However, the discrepancy between the results 145 of Geiker et al. [26], Erdogan [27], and Ancey and Jorrot [33], and between the 146 mortar and the concrete case in the work of Toutou and Roussel [28], does not 147 necessarily imply that there are specific physicochemical interactions between 148 the different particles and the different pastes involved in these studies, and 149 that we would fail describing these materials as suspensions of rigid particles 150 in yield stress fluids. Such discrepancy may indeed find its origin in differences 151 and shortcomings in the experimental procedures used. Actually, the experi-152 ments of Geiker et al. [26], Erdogan [27] and Toutou and Roussel [28] involve 153 a flow of the material. It is then well known that shear-induced migration 154 of particles towards low shear zones (the external cylinder in coaxial cylin-155 ders geometries) is likely to occur [34–36], whatever the care that is taken; 156 this would cause the material to be heterogeneous inside the measurement 157 cell, and the measurement to be non-representative of the homogeneous mate-158 rial. This is particularly true at high concentrations (above 50% for spherical 159 monodisperse particles) where it has been shown by Ovarlez et al. [36] that 160 radial migration occurs as an almost instantaneous and unavoidable process in 161 a Couette geometry. In this case, all the measurements performed in time are 162 likely to be performed on the same stationary heterogeneous structure: testing 163 the material at the same rotational velocity at two different times [8] may then 164 wrongly lead to conclude that there is no shear-induced migration while the 165 only correct conclusion is that the structure is stationary. Moreover, Geiker et 166 al. [26], Erdogan [27] and Toutou and Roussel [28] use a Herschel-Bulkley (or 167 Bingham) fit of the flow curve to extrapolate the value of the yield stress 168 instead of a direct measurement. Chateau et al. [1] have shown that such

an extrapolation generally provides an overestimation of the yield stress of 170 the suspension, and that this overestimation is more dramatic as the particle 171 concentration increases. The reason is that the suspension departs from the 172 Herschel-Bulkley (or Bingham) model at very low shear rate (unaccessible to 173 most concrete rheometers) and has a lower yield stress than the one extrap-174 olated from the measurable flow curve [1]. On the other hand, as the strain 175 involved in this test is small, there should be no migration, nor extrapolation 176 problems, in the slump test used by Ancey and Jorrot [33], as long as the yield 177 stress is high enough to avoid spreading of the material and the correlation 178 between measured slump and yield stress is suitable to their experiments [37]. 179 Another difference between the procedures is that the particle distribution 180 after a flow is anisotropic [38–40], whereas the particle distribution is hardly 181 changed by the slump flow and is thus isotropic in the experiments of An-182 cey and Jorrot [33]; as a consequence, the results of Ancey and Jorrot [33] are 183 not related to the same state of the suspension as the one of Erdogan [27] and 184 Geiker et al. [26]. Finally, note that Ancey and Jorrot [33] and Toutou and 185 Roussel [28] found in some cases that the suspension yield stress can be lower 186 than the suspending paste yield stress; as pointed out by Chateau et al. [1], 187 this should not occur if the noncolloidal particle interact only mechanically 188 with the paste, i.e. these results are likely to apply only to their systems. 189

Finally, it is therefore of high importance to clarify the cases where suspensions can actually be considered as particles in a yield stress fluid, i.e. the cases
where the rheological properties of the suspension depend only on the rheological properties of the suspending fluid and on the coarse particle volume
fraction, shape and size distribution. With the aim of providing such generic
results, Mahaut et al. [2] have recently performed an experimental study on

a broad range of materials. They have suspended beads of various sizes and 196 made of various materials in very different pastes whose common point is to 197 exhibit a yield stress, and they sought consistency between the results. More-198 over, they had a careful look at all the steps of the measurement procedure to 199 ensure that an homogeneous and isotropic material is studied in all cases. They 200 showed that the dimensionless elastic modulus $G'(\phi)/G'(0)$ and the dimension-201 less yield stress $\tau_c(\phi)/\tau_c(0)$ of such monodisperse suspensions depend on the 202 bead volume fraction ϕ only (as expected for systems free from specific physico-203 chemical interactions or specific slippage at the paste/particle interface). They 204 found that the elastic modulus/concentration relationship is well fitted to a 205 Krieger-Dougherty model $(1 - \phi/\phi_m)^{-2.5\phi_m}$ with $\phi_m = 0.57$ for monodisperse 206 isotropic suspensions. They showed that the yield stress/concentration rela-207 tionship is related to the elastic modulus/concentration relationship through 208 a very simple law $\tau_c(\phi)/\tau_c(0) = \sqrt{(1-\phi)G'(\phi)/G'(0)}$ in agreement with the 209 micromechanical analysis of Chateau et al. [1], yielding the Chateau-Ovarlez-210 Trung model $\tau_c(\phi)/\tau_c(0) = \sqrt{(1-\phi)(1-\phi/\phi_m)^{-2.5\phi_m}}$ for the yield stress of 211 suspensions of monodisperse beads in a yield stress fluid. 212

In this paper, we study suspensions of coarse spherical particles in a thixotropic cement paste. We measure the static yield stress of the suspensions as a function of the resting time and of the particle volume fraction. We design new procedures that allow for comparing the yield stress of a given cement paste to that of the same cement paste added with particles. We also take care of designing a procedure that allows for properly accounting for thixotropy of the paste, independently of any irreversible change in the paste behaviour. In Sec. 2, we present the materials and the experimental setup. In Sec. 3, we present the procedure we developed to ensure comparing properly the yield

stress of the suspensions to the yield stress of the suspending cement paste, as a function of the resting time. We present the results in Sec. 4 and compare the yield stress obtained with particles suspended in a cement paste with this procedure to the ones obtained on model materials by Mahaut *et al.* [2], and to the Chateau-Ovarlez-Trung model [1].

27 Materials and methods

2.1 Pastes and particles

We performed our experiments with a thixotropic cement paste. White Cement CEM I/52.5 N CE CP2 NF "SB" from Gargenville Calcia was used 230 to prepare all the cement pastes. Its specific gravity is 3.01. Its compressive 231 strength is 62 MPa at 28 days according to NF EN 196-1 test. The size dis-232 tribution was measured in water using a laser granulometer (according to NF 233 ISO 13320-1 test) for different amount of superplasticizer and is given in Fig. 1. 234 The specific area determined using a BLAINE permeameter, according to NF 235 EN 196-6 test, is 4117 cm2/g. The cement chemical constituents are summa-236 rized in Tab. 1. The Water to Cement ratio W/C studied here was 0.35. A 237 Superplasticizer (Glenium 27) and a nanosilica slurry (Rhoximat CS 60 SL, 238 Rhodia) were added to the mixture with a Superplasticizer to cement mass 239 ratio of 1% and a nanosilica slurry to cement mass ratio of 2%. The fluids (water + superplasticizer + nanosilica) were first mixed together to obtain an 241 homogeneous suspension, and then added to the cement powder before a 5 242 minutes mixing phase in a planetary Controlab mixer: the velocity was first 243 set to 140rpm during 2min, and then to 285rpm during 3min. All the experiments were performed on the fresh cement paste, less than 75min since the constituents were mixed together. Before any measurement, the cement paste was presheared again in the mixer at 285rpm during 2min in order to always start the experiments on a paste in an initially destructured state.

The particles suspended in the cement paste are spherical monodisperse glass
beads of 2 mm diameter. This ensures that the particle size is much larger
than the paste microstructure, so that the particles may "see" the cement
paste as a continuum medium.

We chose to compare the results obtained with the suspensions of particles in a cement paste to the one obtained by Mahaut et al [2] where particles are 254 suspended in various other yield stress fluids: emulsions, colloidal suspensions, 255 and a physical gel (see Mahaut et al. [2] for details on the preparation of these 256 materials). The emulsions are water in oil emulsions, in which the origin of 257 the yield stress is the surface tension between the droplets [5]. The colloidal 258 suspensions are bentonite suspensions, made of clay particles of length of or-259 der 1μ m and thickness 10nm. The yield stress then originates from colloidal 260 interactions between the particles. The physical gel is a Carbopol dispersion. 261 Basically, the polymers arrange in roughly spherical blobs which are squeezed 262 together [41,42]; this yields a yield stress. The particles used in the Mahaut et 263 al. study are spherical monodisperse beads. They are either polystyrene beads 264 of density 1.05, or glass beads of density 2.5., of various particle diameters: 80, 265 $140, 315\mu m$ in the case of the polystyrene beads, and 140, 330 and $2000\mu m$ in the case of the glass beads. The beads are washed in an ultrasound bath 267 during 30 minutes and then dried. This is particularly important for experi-268 ments performed in Carbopol gels: when the unwashed beads are embedded 269 into a Carbopol gel, it actually results in a lower yield stress than when the

washed beads are suspended, indicating residual surface effects [2]; such residual surface effects may be due to colloidal impurities at the particle surface 272 (or residual surfactant at the particle surface when polystyrene particles are 273 used [2]). A single washing is enough to ensure a reproducible state. All ma-274 terials were prepared (i) to ensure that the particle size is much larger than 275 the paste microstructure size, (ii) to check that the results depend only on 276 the mechanical properties of the paste i.e. that they are independent of the 277 physicochemical origin of the yield stress, (iii) to check that the results are 278 independent of the noncolloidal particles size (when the particles are monodis-279 perse and have constant shape and surface texture), (iv) to check that there 280 are neither particle/particle nor particle/paste physicochemical interactions. 281 Moreover, by varying the suspending paste yield stress, it was checked that the 282 dimensionless yield stress depends only on the particle volume fraction (when 283 the particle are monodisperse). If we obtain the same behavior with suspen-284 sions prepared with all materials, including the cement pastes, and whatever 285 the particle size, this ensures that there is no contribution from specific parti-286 cles/material physicochemical interactions and that the results we obtain can 287 be applied to the case of any other particles in any other yield stress fluid (in 288 particular to any cement paste formulation). 289

The insertion of air is unavoidable. The effect of air on the yield stress is not negligible [2], it should thus be checked that its content is negligible: it changes not only the continuous phase mechanical properties [5] but also the effective bead volume fraction, which is a sensitive parameter at high volume fractions. However, methods such as centrifugation to remove the bubbles cannot be used if we want to ensure that the materials remain homogeneous as explained in Sec. 3. We thus chose to work with a constant volume of material in order to

check that the air content is always lower than 1%.

All the measurements we present in this paper were performed on suspensions of coarse particles embedded in pastes at a volume fraction ϕ ranging between 0 and 55%, with an air content lower than 1%.

2.2 Rheological tools

Most rheometric experiments are performed within a vane in cup geometry 302 (inner radius $R_i = 22.5$ mm, outer cylinder radius $R_e = 45$ mm, height H = 10303 45mm) on a commercial rheometer (Bohlin C-VOR 200) that imposes either 304 the torque or the rotational velocity (with a torque feedback). In order to 305 avoid wall slip [43,12], we use a six-blade vane as an inner tool, and we glue 306 sandpaper on the outer cylinder wall. For the small particles in model yield 307 stress fluids, we use another six-blade vane in cup geometry (inner radius 308 $R_i = 12.5$ mm, outer cylinder radius $R_e = 18$ mm, height H = 45mm). Working 309 within these wide-gap geometries allows for studying easily coarse particles and to ensure that, for all the materials studied, there are enough particles in 311 the gap to consider that we measure the properties of a continuum medium 312 (the suspension). 313

We measure the yield stress $\tau_c(\phi)$ of the paste as a function of the volume fraction ϕ of coarse particles embedded in the pastes. In a wide gap geometry, the shear stress τ continuously decreases within the gap: the shear stress at a radius R is $\tau(R) = \frac{T}{2\pi H R^2}$. Therefore, one has to choose a definition of the shear stress that is measured in a given rheological experiment. Here, we want to perform yield stress measurements; whatever the measurement method we choose, yield first occurs where the stress is maximal i.e. along the inner virtual cylinder. As consequence, we define the shear stress measurement as $\tau(R_i) = \frac{T}{2\pi H R_i^2}$, so that the yield stress τ_c is correctly measured (any other definition of the shear stress would provide an underestimation of the yield stress). Anyway, we will focus on the evolution of the dimensionless yield stress $\tau_c(\phi)/\tau_c(0)$ with the bead volume fraction ϕ , which should be independent of the definition of τ .

27 3 Experimental procedure

In this section, we present the procedure aiming at showing the influence of the inclusion of coarse particles on the yield stress of cement pastes. We first show that the choice of the sample preparation and of the yield stress measurement procedure is critical to know how the particles are distributed in the suspension. We then establish a new procedure to ensure a good knowledge of the interstitial paste properties in the suspension.

334 3.1 Preparation and yield stress measurement

First, we need to define precisely the state of the materials we want study.

Three points are actually important: (i) we want to perform our yield stress

measurement on a homogeneous suspension, otherwise the measurement would

have no meaning, (ii) we want to control the microstructure of the suspensions (i.e. the distribution of the neighbors of the coarse particles) to ensure

that all measurements deal with the same state of the suspension, and can

be compared and modelled, (iii) we need the interstitial cement paste to be

initially destructured in order to study thixotropy.

These three points impose severe restrictions about the preparation and yield 343 stress measurement procedures, as shown by Mahaut et al. [2]. They showed 344 that measurements involving an important flow of the material (a large strain) 345 pose several problems. First, a flow causes particle migration towards the low 346 shear zones (the outer cylinder in coaxial cylinder geometries) i.e. creation of 347 a heterogeneous structure. This migration phenomenon is well documented 348 for suspensions of noncolloidal particles in Newtonian fluids [34–36] but is still badly known in yield stress fluids. As it needs a large strain to occur 350 for moderate volume fraction [34,35], it may be avoided in these cases by 351 performing only short duration experiments. However, for volume fractions 352 of the order of 50% and more, migration is a critical phenomenon: it seems 353 unavoidable as it is almost instantaneous as shown by Ovarlez et al. [36]. 354 Another problem when suspensions flow is that an anisotropic microstructure 355 of the particles is created by the flow, as observed in suspensions of particles 356 in Newtonian fluids [38–40]. It is also a critical phenomenon: Mahaut et al. [2] 357 showed that suspensions of isotropic and anisotropic microstructure have very 358 different rheological properties. 359

These problems imply that we cannot preshear our materials with the rheometer and that we cannot use a yield stress measurement method based on a shear flow such as shear rate [8] or shear stress ramps [44] and creep tests [45]; we then have to measure the static yield stress. On the other hand, as the static yield stress of thixotropic materials depends on the time passed at rest in the solid state [46], the measurements have to be performed on a well defined state of the paste, i.e. the material needs to be first strongly presheared to get a destructured initial state. However, as pointed out above, we cannot apply

a controlled preshear with the rheometer to the system after its preparation. 368 That is why, before loading the material in the measurement cup, we first pres-369 heared the cement paste alone during 2 minutes with the mixer at 285rpm; 370 this ensures that the cement paste is initially in a destructured state. Then, 371 the particles and the paste are mixed together in the measurement cylinder, 372 and the loaded suspension is strongly stirred by hand in random directions to 373 disperse the particles; this random stirring should ensure keeping the material 374 in a destructured state while avoiding particle migration and anisotropy. Af-375 terwards, the vane tool is inserted in the cup, and we perform our yield stress 376 measurement after a given resting time with the vane method [47,48]: a small 377 rotational velocity, corresponding to a shear rate of $0.01s^{-1}$ is imposed to the 378 vane tool. Note that we checked that we observe the same effect of the parti-379 cles on the yield stress whatever the low velocity that is chosen to drive the 380 vane tool. Fig. 2 shows the shear stress vs. strain for yield stress measurement 381 experiments performed in a cement paste. There is an overshoot, followed by 382 a slow decrease of the shear stress: the peak defines the static yield stress, the 383 decrease corresponds to destructuration of the material under the shear flow; 384 the suspension structure at yield should then be isotropic and homogeneous. 385 Then, any new yield stress measurement requires a new sample preparation 386 or a new random manual preshear in the cup: it has been shown by Mahaut et387 al. [2] that the small strain of order 1 induced by the whole measurement 388 procedure is sufficient to change the material state (it is sufficient to change 389 the suspension microstructure or to induce migration): the suspension states before and after the yield stress measurements are characterized by different 391 rheological properties.

As we are interested in the influence of the particles on the yield stress of cement pastes, we will need to compare the suspension yield stress and the cement paste yield stress. It is thus important to ensure that we have a good knowledge of the properties of the interstitial paste state in the suspension.

The procedure developed to ensure this measurement is presented in detail in the Appendix A. We present here the main steps.

First, we have to note that it is very difficult to achieve a good reproducibility
of a cement paste mechanical behavior (see Appendix A). That is why we
chose to work on the same batch for the measurement of the properties of the
paste alone and for the suspension.

Then, for a given cement paste batch, we observe that yield stress measure-404 ments performed in the same conditions as regards thixotropic effects (i.e. for 405 a 2 minutes resting time after a strong stirring of the paste) provide values 406 that depend on the time t_{age} elapsed since the constituents of the cement paste 407 were mixed together (Appendix A). This means that one cannot know what 408 is the yield stress of the interstitial paste in a suspension if the yield stress of 409 the cement paste alone is not measured at exactly the same time after mixing 410 as the yield stress of the suspension. That is why we chose to measure simul-411 taneously the yield stress of the suspension and the yield stress of the cement paste alone in exactly the same conditions (same age t_{age} after mixing the 413 constituents of the cement paste, same time t_{rest} after the end of the strong 414 stirring), with the help of 2 rheometers that perform their measurements in 415 parallel.

We have also shown that when the same suspension sample, after a first resting period and a first measurement, is stirred again in the measurement cup, its interstitial cement paste is not in the same state of destructuration as the cement paste alone stirred with the same procedure (Appendix A). This means that the suspension and the cement paste cannot be compared anymore. A solution to this problem is to perform only a single measurement on a suspension, for a given resting time after its preparation.

A key point of the comparison between the suspension and the cement paste is actually that the cement paste is initially strongly presheared in the mixer 425 for both samples: this defines an initial destructured state of the paste that is 426 the same both for the interstitial cement paste and for the cement paste alone. After this preshear, the cement paste is loaded alone in one measurement cup, 428 and with the particles in another cup. Both samples are then strongly stirred 429 by hand during 30s in random directions: this ensures an homogeneous dis-430 persion of the particles in the suspension, while keeping the cement paste in 431 a destructured state in both samples. Then, the stirring is stopped simultane-432 ously for both samples: this defines the beginning of the resting period. With 433 this procedure, we have shown that the paste alone and the interstitial paste 434 have the same history and thus the same behavior (see Appendix A). 435

Finally, as the cement paste is thixotropic, its static yield stress increases as a function of the time t_{rest} elapsed since the end of the stirring. However, we showed that the yield stress value also depends on the time t_{age} elapsed since the constituents were mixed together, even at short times. This would mean that a characterization of thixotropy would only have a meaning for this age t_{age} , and it would make the study of the impact of the coarse particles on this thixotropy difficult. Nevertheless, we have shown that the irreversible phenomena can be separated from the reversible phenomena. The thixotropic (reversible) increase of the yield stress is actually the same whatever the paste age t_{age} : it depends only on the time t_{rest} elapsed since the end of a preshear (Appendix A). The increase of the yield stress of our cement paste due to thixotropy is basically linear in t_{rest} : it reads $\tau_c(t_{rest}) = A_{thix} t_{rest}$ with an increase rate $A_{thix} = 12$ Pa/min.

$_{ ext{\tiny 449}}$ 3.3 Summary

As a summary we present in Fig. 3 a sketch of the whole procedure used to study the influence of coarse particles on the yield stress of cement pastes.

This procedure ensures (i) that an homogeneous material is studied; (ii) that
we study a well defined state of the material: we chose to study the case of
isotropic distributions of particles; (iii) that the interstitial cement paste is well
characterized; (iv) that the initial destructured state of the interstitial cement
paste is well defined; (v) that thixotropy is accounted for and separated from
irreversible phenomena; (vi) that the results obtained with cement pastes can
be compared to measurements performed in other yield stress fluid.

459 4 Experimental results

In this section, we summarize the results of the yield stress measurements performed on the suspensions with the procedure presented above. We compare the results obtained with the cement pastes to the results obtained by Mahaut et al. [2] with various yield stress fluids, and to the Chateau-Ovarlez-Trung model [1]. In Fig. 4 we plot the dimensionless yield stress $\tau_c(\phi)/\tau_c(0)$ vs. the volume fraction ϕ of coarse particles embedded in the cement paste, when the yield stresses are measured with the procedure developed in Sec. 3 for various times t_{rest} after the end of a strong stirring.

We first observe that the yield stress increases when the coarse particle volume fraction is increased. This increase is quite limited for volume fraction lower than 45%: in this case, the yield stress is increased by a factor less than 3. However, the yield stress is found to increase sharply at the approach of a 60% volume fraction. E.g., the yield stress of a suspension of 55% particles is 20 times higher than the yield stress of the interstitial cement paste.

We also observe in Fig. 4 that the same evolution of the yield stress with the particle volume fraction is found whatever the time t_{rest} passed at rest before the measurement. This means that the yield stress of suspensions of coarse particles embedded at a volume fraction ϕ in a thixotropic cement paste of time-dependent yield stress $\tau_c(0,t)$ reads

$$\tau_c(\phi, t) = \tau_c(0, t)g(\phi) \tag{1}$$

This feature is expected if the coarse (i.e. noncolloidal) particles have only a mechanical interaction with the cement paste [2]: in this case, they should not interfere with the physical process at the origin of thixotropy. Then, at time t the interstitial paste has naturally the same yield stress $\tau_c(0,t)$ as if it had not been in contact with the coarse particles. Finally, as the relative increase

of the yield stress due to the monodisperse particles should be a function of their volume fraction ϕ only, independently of the value of the interstitial fluid yield stress, the yield stress of the suspension at time t is expected to be equal to $\tau_c(0,t)$ multiplied by some function $g(\phi)$ whatever $\tau_c(0,t)$, as observed experimentally.

Eq. 1 has an interesting consequence: it means that it is sufficient to know how the interstitial cement paste evolves in time to predict the suspension evolution at rest. This is important for fresh concrete as their behavior is hard to measure: our results show that the knowledge of the cement paste structuration rate at rest is sufficient to predict the fresh concrete structuration rate. As found on the cement paste we studied (see Sec. 3.2) the yield stress evolution at rest after a preshear of a cement paste usually reads [19]:

$$\tau_c(0,t) = \tau_c(0) + A_{thix}t \tag{2}$$

where A_{thix} is the structuration rate of the paste. In this case, Eq. 1 reads:

$$\tau_c(\phi, t) = \tau_c(0)g(\phi) + A_{thix}g(\phi)t \tag{3}$$

As a consequence, if the mechanical impact of the coarse particles is to increase the yield stress by a factor $g(\phi)$, then their impact on the structuration rate of the paste is to increase it also by a factor $g(\phi)$. It is thus sufficient to measure the cement paste yield stress evolution in time (i.e. A_{thix}) and to measure the increase of the yield stress with the volume fraction (i.e. $g(\phi)$) for a single resting time t_{rest} to infer the value $A_{thix} g(\phi)$ of the structuration rate of the suspension (and more generally of fresh concrete). In Fig. 5, we plot a summary of the dimensionless yield stress measurements $\tau_c(\phi)/\tau_c(0)$ performed on all the materials by Mahaut *et al.* [2], together with the results obtained with cement pastes.

We find that all the results are consistent: the dimensionless yield stress $\tau_c(\phi)/\tau_c(0)$ is independent of the physicochemical origin of the material yield 511 stress, of the bead material and of the bead size, and of the paste yield stress; 512 it is a function of the volume fraction only. This means that the particles have 513 a purely mechanical contribution to the paste behavior, which is indepen-514 dent of the physicochemical properties of the materials: the only important 515 matter is the value of the yield stress of the paste. This also validates our 516 approach: as long as the coarse particle size is much larger than the cement 517 paste microstructure, a suspension of coarse particles in a cement paste can 518 be considered more generally as a suspension of rigid noncolloidal particles in 519 a yield stress fluid.

This result helps proposing a method than can be applied to obtain quickly 521 the effect of particles of any kind (any shape, any size distribution) on the yield 522 stress of a cement paste. Actually, preparing a model yield stress fluid of stable 523 and reproducible rheological properties, showing no setting nor thixotropic 524 effects (e.g. an emulsion), is quite easy, and measurements are much easier to perform on these materials. Then a great amount of accurate experiments can 526 be performed to measure the properties of suspensions of particles in this yield 527 stress fluid. Finally, the result of the measurement of the dimensionless yield 528 stress $\tau_c(\phi)/\tau_c(0)=g(\phi)$ as a function of the volume fraction ϕ of particles

in this yield stress fluid should hold if the interstitial paste is a cement paste. Moreover, we have shown that the knowledge of the structuration rate A_{thix} of 531 a cement paste is sufficient to infer the structuration rate of the suspension of 532 particles in this cement paste (it is equal to $A_{thix}g(\phi)$). A measurement of the 533 cement paste structuration at rest plus the measurement $\tau_c(\phi)/\tau_c(0)=g(\phi)$ in 534 a model yield stress fluid then provides everything that is needed to infer the 535 behavior of mortars or concretes. Note however that these results apply only 536 as long as the particle size is much larger than the cement paste microstructure 537 typical size so that the particles see the yield stress fluid as an homogeneous 538 material. This should not be true otherwise: if the particles were to be sensitive 539 to the cement paste microstructure, then the behavior should depend on the 540 exact details of the specific microstructure of each paste. E.g., in the case of 541 particles suspended in a foam, Cohen-Addad et al. [49] found that the behavior of the suspension depends on the particle size for particles of size lower than 543 5 times the bubble size in the foam. Note finally that another important 544 requirement is that the fraction of superplasticizer adsorbed at the surface 545 of the aggregates suspended in the paste is negligible. The study of Hammer 546 and Wallevik [23] suggests that in some cases (it may depend strongly on the 547 cement paste composition) this may be true only if the aggregates are larger 548 than 0.25 to 0.5mm; in such cases, our approach would then be valid for SCC only if the suspending yield stress fluid includes the fine aggregates (of size 550 lower than 0.25 to 0.5mm in the study of Hammer and Wallevik). 551

Proposing a theoretical value for the dimensionless yield stress is challenging. However, it has been shown by Chateau et al. [1] that it is possible to give 554 a general relationship between the linear response of the materials (e.g. its 555 dimensionless elastic modulus $G'(\phi)/G'(0)$ as probed under the yield stress) 556 and the dimensionless yield stress $\tau_c(\phi)/\tau_c(0)$ of a suspension of rigid parti-557 cles in a yield stress fluid. This estimate is based on the following hypotheses: 558 the particles are rigid and noncolloidal; there are no physicochemical interac-559 tions between the particles and the paste; the distribution of the particles is 560 isotropic. This is what we have managed to perform experimentally, therefore, 561 our experiments are fitted to provide a test of these theoretical predictions. 562 Chateau et al. [1] find 563

$$\tau_c(\phi)/\tau_c(0) = \sqrt{(1-\phi)G'(\phi)/G'(0)}$$
 (4)

Mahaut et al. [2] have measured the elastic modulus of all the suspensions studied above, and found a Krieger-Dougherty model to apply $G'(\phi)/G'(0) = (1-\phi/\phi_m)^{-2.5\phi_m}$ for the dimensionless elastic modulus. Combining this equation and the theoretical expression Eq. 4 thus yields for the yield stress the Chateau-Ovarlez-Trung model [1]

$$\frac{\tau_c(\phi)}{\tau_c(0)} = \sqrt{\frac{1 - \phi}{(1 - \phi/\phi_m)^{2.5\phi_m}}}$$
 (5)

which should be valid for any isotropic suspension of rigid spherical noncolloidal particles in yield stress fluids with no physicochemical interactions between the particles and the paste. Our experimental data are compared to Eq. 5 on Fig. 5. We find a remarkable agreement between our data and this model with a best fit for $\phi_m = 0.56$; note that this value of 0.56 is valid only for the case of monodisperse spherical particles we studied.

Note however that Eq. 5 can a priori be easily modified to account for polydis-576 persity and for complex shapes of the particles when studying more complex 577 suspensions. Actually, Eq. 4 should hold in all cases. It is then sufficient to 578 know what is the linear response of a suspension made with the studied parti-579 cles to infer the yield stress value. This linear response can be measured with 580 the method presented in this paper (it is the dimensionless elastic modulus 581 $G'(\phi)/G'(0)$; it can also be inferred from the huge amount of dimension-582 less viscosity data from the literature dealing with suspensions of particles in 583 Newtonian fluids (with the same particles): the problem of the elasticity of 584 a suspension of rigid particles in a linear elastic material is actually formally 585 similar to the problem of the viscosity a suspension of rigid particles in a 586 Newtonian (thus linear) material.

4.4 Comments

Our results are naturally close to the Ancey and Jorrot [33] ones, as they have chosen to measure the yield stress of the suspension by means of a slump test which ensures avoiding migration of particles and anisotropy of the material. On the other hand, we find very different values from Geiker *et al.* [26], Erdogan [27], and Toutou and Roussel [28]. As pointed out in Sec. 1 and Sec. 3, this is due to the shortcomings of their experimental procedure which is based on a flow and an extrapolation of the dynamic yield stress from a flow curve. Their

materials are then likely to be heterogeneous and anisotropic. Moreover, an extrapolation from a flow curve provides an overestimation of the yield stress of the suspension [1] because the suspension departs from a Herschel-Bulkley (or Bingham) model at very low shear rate (unaccessible to most concrete rheometers) and has a lower yield stress than the one extrapolated from the accessible flow curve, even if the suspending yield stress fluid has a Herschel-Bulkley (or Bingham) behavior.

Finally, note that in most papers the results are presented vs. ϕ/ϕ_{max} where ϕ_{max} is the maximum packing fraction (taken at about 0.65 for monodisperse 604 particles) and the yield stress divergence is expected to occur for $\phi/\phi_{max}=1$. 605 This is not correct: the maximum volume fraction ϕ_m for the yield stress sharp 606 increase should not be taken as the maximum volume fraction one can reach 607 by packing particles together (which is the definition of the maximum pack-608 ing fraction ϕ_{max}). The maximum volume fraction for the yield stress sharp 609 increase is rather the one at which direct contacts become important, which 610 is the limit of application of models including only hydrodynamic interactions 611 between the particles, and also the limit between SCC and ordinary rheology 612 concretes [29]. This explains why we find the yield stress to diverge at around 613 56% while the maximum packing fraction is of about 65% for spheres. 614

5 Conclusion

We have studied the behavior of suspensions of coarse particles in a thixotropic cement paste. We managed to design procedures that allow for (i) studying an homogeneous and isotropic suspension, (ii) comparing the yield stress of a given cement paste to that of the same cement paste added with particles,

(iii) accounting properly for the thixotropy of the cement paste. We observed 620 that the yield stress of these pastes follows the very simple Chateau-Ovarlez-621 Trung model [1] $\tau_c(\phi)/\tau_c(0) = \sqrt{(1-\phi)(1-\phi/\phi_m)^{-2.5\phi_m}}$, with $\phi_m = 0.56$ for 622 monodisperse spherical particles, consistently with the experimental results 623 of Mahaut et al. [2] obtained with many different suspensions. This supports 624 the fact that the yield stress of the suspension is independent of the physic-625 ochemical properties of the yield stress fluid, and depends only on its yield 626 stress value. This shows that studies of suspensions in model yield stress fluids 627 can be used as a general tool to infer the behavior of fresh concrete. More-628 over, we showed that the thixotropic structuration rate of these pastes (their 629 static yield stress increase rate in time) is not changed by the presence of the 630 particles. This shows that it is sufficient to measure the cement paste yield 631 stress evolution in time and to measure the increase of the yield stress with 632 the volume fraction of coarse particles for a single resting time to predict the 633 value of the structuration rate of fresh concrete. For a linear increase of the 634 cement paste yield stress with a rate A_{thix} , we predict a linear increase of the 635 suspension with a rate $A_{thix}\sqrt{(1-\phi)(1-\phi/\phi_m)^{-2.5\phi_m}}$. 636

637 A Characterization of the interstitial paste

In this appendix, we detail the arguments that have led to develop the procedure presented in Sec. 3. This new procedure is built to ensure a good knowledge of the mechanical properties and of the state of structuration of the interstitial cement paste in the suspension.

First, we have to note that it is very difficult to achieve a good reproducibility of a fresh cement paste mechanical behavior. In Fig. A.1a we show the result

of the yield stress measurements performed (apparently) exactly in the same conditions in 2 cement pastes having the same composition. We observe that 645 the uncertainty on the yield stress of the cement paste we get is of order 25%. 646 This means that, if we want to measure accurately the ratio of the suspension 647 yield stress to the interstitial cement paste yield stress, we cannot compare the 648 properties of suspensions of particles in a cement paste to the properties of a cement paste having the same composition but being from a different batch. 650 That is why we chose to work on the same batch for the measurement of the 651 properties of the paste alone and for the suspension. 652

Then, for a given cement paste batch, we could propose to first measure the 653 cement paste yield stress and then the suspension yield stress. For the results 654 to be comparable, one would then just have to perform the experiment in the 655 same conditions as regards thixotropy (i.e. for the same resting time after a 656 strong preshear). In order to check the validity of this method, we performed 657 yield stress measurements several times in the same conditions (i.e. for a 2 658 minutes resting time after a strong stirring of the paste) on a single cement 659 paste batch. The results of this experiment are depicted in Fig. A.2 as a 660 function of the time t_{age} elapsed since the constituents of the cement paste 661 were mixed together. We observe that due to various irreversible chemical 662 interactions in the material, the cement paste yield stress, measured in the 663 same conditions as regards thixotropic effects, evolves (non-monotonously) as 664 a function of the time t_{age} elapsed since the constituents of the cement paste were mixed together. This means that one cannot know what is the yield 666 stress of the interstitial paste in a suspension if the yield stress of the cement 667 paste alone is not measured at exactly the same time after mixing as the yield 668 stress of the suspension. That is why we chose to measure simultaneously the yield stress of the suspension and the yield stress of the cement paste alone in exactly the same conditions (same age t_{age} after mixing the constituents of the cement paste, same time t_{rest} after the end of the strong stirring), with the help of 2 rheometers that perform their measurements in parallel. We show in Fig. A.1b that, as expected, this method yields a very low uncertainty when the measurements are performed on the same cement paste (without particles).

Now, by performing these simultaneous measurements of the suspension yield 677 stress $\tau_c(\phi)$ and of the cement paste yield stress $\tau_c(0)$ several times, at various 678 ages t_{age} after mixing the constituents of the cement paste, we should observe 679 the same effect of the particles on the yield stress whatever the age of the 680 cement paste. In Fig. A.3 we present the dimensionless yield stress $\tau_c(\phi)/\tau_c(0)$ 681 as a function of the time t_{age} elapsed since the constituents of the cement paste 682 were mixed together. We observe that $\tau_c(\phi)/\tau_c(0)$ is not constant in time. This 683 means that the interstitial paste is not in the same mechanical state as the 684 paste alone although they have apparently the same history. The only differ-685 ence stands in the preshear procedure: before performing each measurement, 686 the paste and the suspension are presheared to ensure a reproducible destruc-687 tured initial state. As pointed out above, the preshear has to be manual to 688 avoid migration and anisotropy. Our results show that this preshear is not as 689 efficient in the suspension as in the cement paste. It is harder to shear the sus-690 pension, thus an experimentalist cannot shear the suspension the same way as the paste alone. It can be noted that, in the case of a strong mechanical pres-692 hear in a mixer, an opposite result has been obtained by Toutou and Roussel 693 [28] due to the mixing effect of the particles. As a result of these imperfect 694 and perturbing preshears, the differences between the structuration state of

the paste alone and of the interstitial paste in the suspension increases with 696 time. The suspension and the cement paste cannot be compared anymore, and 697 the function $\tau_c(\phi)/\tau_c(0)$ is no more correctly measured by this means. A solu-698 tion to this problem is to perform only a single measurement on a suspension, 699 for a given resting time after its preparation. As pointed out in Sec. 3.2, a 700 key point of the comparison between the suspension and the cement paste is 701 then that the cement paste alone is first initially strongly presheared in the 702 mixer for both samples before being loaded (and eventually mixed with the 703 particles) in the measurement cups: this defines an initial destructured state 704 of the paste that is the same both for the interstitial cement paste and for 705 the cement paste alone. The manual stirring in the measurement cup then 706 ensures an homogeneous dispersion of the particles in the suspension, while 707 keeping the cement paste in a destructured state in both samples. With this 708 procedure, one ensures that the paste alone and the interstitial paste have the 709 same history and thus the same behavior. We show actually in Fig. A.3 that, 710 in these conditions, the same value of $\tau_c(\phi)/\tau_c(0)$ is found within the measure-711 ment uncertainty whatever the time t_{age} elapsed since the constituents of the 712 cement paste were mixed together.

Finally, as the cement paste is thixotropic, its static yield stress increases as a function of the time t_{rest} elapsed since the end of the stirring. However, if we want to account properly for the (reversible) thixotropic behavior of the cement paste, and to check what the influence of the particles on this thixotropic behavior is, we face a problem. We showed that the value of the yield stress measured 2 minutes after a strong stirring evolves as a function of the time t_{age} elapsed since the constituents were mixed together, even at short times. This would mean that in order to characterize the increase of

the yield stress of cement pastes due to structuration at rest, as a function of the resting time t_{rest} after a strong stirring, we would need to perform 723 all the yield stress measurements only at a same given age t_{age} after mixing 724 the constituents of the cement paste. And this characterization of thixotropy 725 would only have a meaning for this age t_{age} . However, we show in the following 726 that the thixotropic increase of the yield stress is actually the same whatever 727 the paste age. In Fig. A.4a, we plot the yield stress of a cement paste as a 728 function of the age t_{age} of the paste for 3 different times t_{rest} after a strong 729 stirring; note that as we have only 2 rheometers, these measurements had to 730 be performed on 2 batches, so that the uncertainties may be rather large (as 731 in Fig. A.1a). 732

We observe the same evolution of the paste behavior as a function of t_{age} whatever t_{rest} . An important consequence is that the irreversible effects can be separated from the reversible effects by writing

$$\tau_c(t_{rest}, t_{age}) = \tau_c(t_{age}) + \tau_c(t_{rest})$$
(A.1)

where $\tau_c(t_{age})$ is the yield stress that would be measured just after a preshear, which depends on the time t_{age} elapsed since the constituents were mixed together, and $\tau_c(t_{rest})$ represents the increase of the yield stress due to thixotropic effects, which depends only on the time elapsed since the end of a preshear. This is shown in Fig. A.4b: all data are superposed when shifted by a constant value that depends only on t_{rest} . From the superposition of data in Fig. A.4b, we find that $\tau_c(t_{rest} = 4 \text{min}) - \tau_c(t_{rest} = 2 \text{min}) = 24 \text{Pa}$ and $\tau_c(t_{rest} = 6 \text{min}) - \tau_c(t_{rest} = 2 \text{min}) = 48 \text{Pa}$. This is consistent with the simple law proposed by Roussel [19] i.e. the increase of the yield stress due to thixotropy is basically linear in t_{rest} : it reads $\tau_c(t_{rest}) = A_{thix} t_{rest}$ with an increase rate $A_{thix} = 12 \text{Pa/min}$. Finally, as the absolute increase of the yield stress due to thixotropic effects is the same at any time t_{age} (lower than 90 min) since the constituents were mixed together, this shows that studies of thixotropy and of the effect of the coarse particles on this thixotropy performed at different times t_{age} can be compared together and provide relevant information on the thixotropy of the suspensions.

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Constituents	% by mass
SiO_2	20.95
$\mathrm{Al_2O_3}$	4.08
${ m TiO_2}$	0.14
Fe_2O_3	0.22
CaO	65.55
MgO	0.49
Na ₂ O	0.12
$ m K_2O$	0.20
SO_3	2.60
RI	1.47
PAF	3.36

Table 1
Cement chemical constituents.

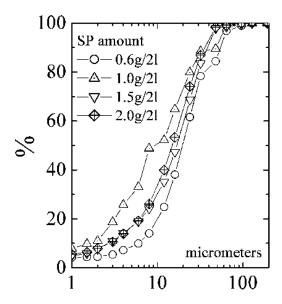


Fig. 1. Cement size distribution curve for various superplasticizer (SP) amount.

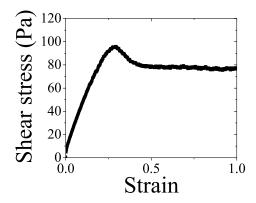


Fig. 2. Shear stress vs. strain when slowly shearing a cement paste at 10^{-2} s⁻¹ 2 minutes after a strong stirring of the paste.

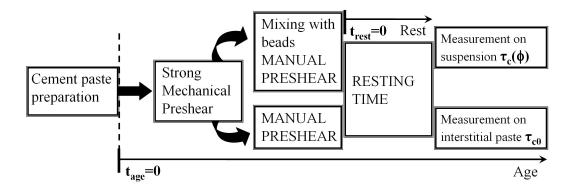


Fig. 3. Sketch of the procedure designed to study the evolution of the dimensionless yield stress $\tau_c(\phi)/\tau_c(0)$ with the volume fraction ϕ of particles in the suspension.

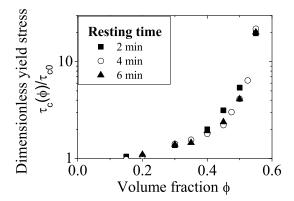


Fig. 4. Dimensionless yield stress $\tau_c(\phi)/\tau_c(0)$ vs. the bead volume fraction ϕ for suspensions of 2mm glass beads in a cement paste, measured with the procedure developed in Sec. 3 for various times t_{rest} after a strong stirring of the suspension.

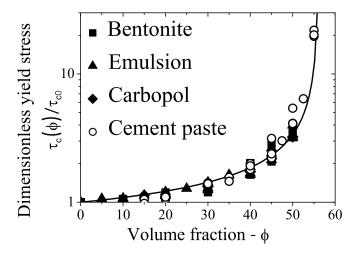


Fig. 5. Dimensionless yield stress $\tau_c(\phi)/\tau_c(0)$ vs. the bead volume fraction ϕ for suspensions of 80, 140, and 315 μ m polystyrene beads and 140 μ m, 330 μ m and 2mm glass beads in various bentonite suspensions, emulsions and Carbopol gels (results from Mahaut *et al.* [2]), and for 2mm glass beads suspended in a cement paste. The solid line is the Chateau-Ovarlez-Trung model $\sqrt{(1-\phi)\times(1-\phi/\phi_m)^{-2.5\phi_m}}$ with $\phi_m=0.56$.

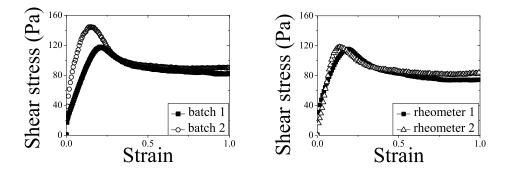


Fig. A.1. a) Shear stress vs. strain when slowly shearing two batches of a cement paste at 10^{-2} s⁻¹ 2 minutes after a strong stirring of the paste. b) Shear stress vs. strain when slowly shearing simultaneously on 2 rheometers a cement paste from a single batch at 10^{-2} s⁻¹ 2 minutes after a strong stirring of the paste.

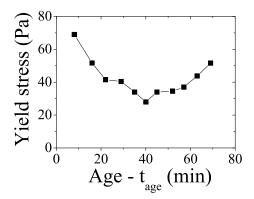


Fig. A.2. Yield stress of a cement paste measured 2 minutes after a strong stirring of the paste vs. the time t_{age} elapsed since the constituents of the cement paste were mixed together.

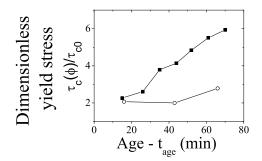


Fig. A.3. Dimensionless yield stress $\tau_c(\phi)/\tau_c(0)$ measured 2 minutes after a strong stirring of the suspension vs. the time t_{age} elapsed since the constituents of the cement paste were mixed together (with a volume fraction of coarse particles $\phi = 40\%$), in two cases: when the same suspension of particles is used for all measurements (squares); when the particles are mixed with the cement paste just before each measurement (open circles).

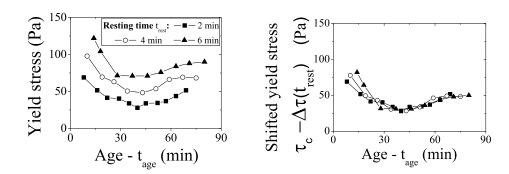


Fig. A.4. a) Yield stress of a cement paste measured 2, 4 and 6 minutes after a strong stirring of the paste vs. time t_{age} elapsed since the constituents of the cement paste were mixed together. b) Data of Fig. A.4a rescaled by shifting the yield stress values by a function $\Delta \tau_c(t_{rest})$ of the time t_{rest} elapsed since the end of the strong stirring.