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SrSnO$_3$:N – nitridation and evaluation of photocatalytic activity

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ABSTRACT

The SrSnO$_3$ was prepared in powders form by a conventional method of solid state reaction and submitted to thermal treatment under NH$_3$ in order to obtain the oxynitride and then promote photocatalytic activity at the visible region. Three different nitridation conditions were tested. The results showed that when the SrSnO$_3$ is subjected to nitridation at moderate conditions (600 °C and 650 °C) oxynitride is formed preserving the perovskite structure. On the other hand, when the nitridation occurs at more severe conditions (750 °C), it is possible to observe the decomposition of the structure instead of the incorporation of N. Samples in which oxynitride is formed, SrSnO$_3$:N, showed high potential photocatalysis under visible light, showing to be about 4 times more active than SrSnO$_3$ or TiO$_2$ at the same conditions.

Keywords: Strontium stannate, Oxynitride, Nitridation, Visible range, Photocatalysis, Methylen Blue.

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1. Introduction

Recently, perovskites photocatalyst have attracted wild attention in all kinds of photocatalytic applications [1-3]. SrSnO$_3$ is a distorted perovskite with an orthorhombic structure (space group Pbnm) due to tilted octahedra [4]. It has been extensively studied due to its potential as thermally stable capacitors, high-quality humidity sensors and photoluminescent and photocatalytic materials [5-8]. More recently, photocatalytic properties of SrSnO$_3$ have been reported indicating that it is a promising photocatalyst due to its suitable valence band and conduction band edge positions [9-12]. For instance, chemically prepared SrSnO$_3$ nanostructures showed high activity for water splitting [13,14] and methylene blue discoloration [12,15]. Nevertheless, the band gap is too large to be used with solar radiation 4.1 eV [13]).

Nitridation is a well known strategy to shorten oxide’s band gap [16,17]. Comparing to metal doping, it presents the advantages of being superficial (not bulk) and disfavors fast electron-hole recombination [18]. For nitrided materials, band gap reduction can be explained by a decrease in electronegativity (3.50 for oxygen and 3.07 for nitrogen) which results in a higher degree of covalency in the metal-anion bonding [19]. Moreover, there is an overlap between N$_{2p}$ and O$_{2p}$ orbitals which increases the valence band maximum (VBM) [13,20,21].

The literature provides some examples in which N incorporation resulted in higher photocatalytic activity in the visible region. For instance, TiO$_x$N$_y$ [16], TaO$_x$N$_y$ [22,23], (Zn$_x$Ga$_{1-x}$)(O$_{2-x}$N$_x$) [24] are some systems that have showed very attractive results. Some perovskites, such as SrTiO$_3$:N and BaTiO$_3$:N [19], have also showed promising results, although the photocatalytic reaction needs to be conducted in the presence of H$_2$O$_2$ when BaTiO$_3$:N is used. No experimental study concerning nitridation of stannates was found. However, Chen et al. [11] performed density functional theory calculations in order to evaluate the effect of cationic (Cr$^{3+}$, Fe$^{3+}$, and Rh$^{3+}$) and anionic (N$^{3-}$, N$^{2-}$, and S$^{2-}$) doping in SrSnO$_3$ and concluded that N$^{3-}$ is the most effective dopant to reduce the band gap. In this work, we discuss the nitridation of SrSnO$_3$ as a strategy to shift its photoactivity into the visible region.

2. Experimental procedure

2.1 Synthesis by solid state reaction

The SrSnO$_3$ was synthesized by solid state reaction. SrCO$_3$ (Sigma-Aldrich, 99.9 % purity) and SnO$_2$ (Merck, 99.9 % purity) powders were thoroughly mixed in an agate mortar. The mixture was initially treated at 800 °C for 6 h in air. After cooling, the powder was further ground, pressed in pellets and sintered at 1100 °C for 24 h. After cooling, the pellets were ground again.
2.2 Nitridation

SrSnO₃ powder was nitrided in a horizontal tubular furnace under gas flow of NH₃ (20 L.h⁻¹) varying the temperature and reaction time according to Table 1. The heating rate was 10 °C.min⁻¹ and cooling was inertial. A N₂ flowing was introduced during cooling (20 L.h⁻¹).

<table>
<thead>
<tr>
<th>Nitrided samples</th>
<th>Temperature/°C</th>
<th>Time/h</th>
</tr>
</thead>
<tbody>
<tr>
<td>B</td>
<td>600</td>
<td>24</td>
</tr>
<tr>
<td>C</td>
<td>650</td>
<td>21</td>
</tr>
<tr>
<td>D</td>
<td>750</td>
<td>8</td>
</tr>
</tbody>
</table>

2.3 Photocatalysis tests

The photocatalytic activity of oxide and oxynitride was investigated by means of the discolouration of an aqueous solution of methylene blue (MB) (5.0 mg.L⁻¹). For this purpose, we used a batch type reactor containing two high pressure Hg lamps for UV tests (120 W; λ = 254 nm) and two tungsten lamps (100 W) for testing in the visible. In all assays, 50.0 mL of MB solution and 25.0 mg of catalyst were used. Every 30 minutes, aliquots (5.0 mL) were collected, the catalyst was separated by centrifugation and dye quantitation was performed by UV-Vis spectroscopy (Varian Cary 5000 UV-Vis NIR) spectroscopy, measuring the absorbance maximum at 660 nm.

Some adsorption tests were performed in the dark under the same conditions of the photocatalytic tests. Other tests were also performed in the absence of catalyst to check photolysis of MB.

2.4 Characterization

All of the samples were characterized by X-ray diffraction (XRD) (Philips PW 3710 diffractometer operating with -CuKα radiation of λ = 0.15418 nm for 5°– 90° (2θ)). The quantity of nitrogen and oxygen were measured in the synthesized powders with a TC-600 LECO® Analyzer. The specific surface area was estimated by BET isotherm (Micromeritics FlowSorb II 2300). UV-Vis spectra (Varian Cary 5000 UV-Vis NIR) were obtained by the reflectance method using BaSO₄ as reference. The Kubelka-Munk function [33] was applied in order to bring the energy of the band gap. The Raman spectra were recorded with a Renishaw spectrophotometer (inVia Raman Microscope) with Ar laser source, operating at 20 mW and 514.5 nm (1
scan, exposure time 10000s). The images were obtained with Scanning Electron Microscope (SEM) images and Energy Dispersive Spectroscopy (EDS) analyses were performed with a JEOL – JSM 7001-F equipped with an Oxford Instruments EDS analyzer. All samples were coated with gold in order to improve conductivity.

3. Results and discussion

Crystalline single phase and carbonate free SrSnO<sub>3</sub> was obtained through solid-state reaction at 1100 °C. All of the SrSnO<sub>3</sub> diffraction peaks were indexed according to the JCPDS 77-1798, confirming the formation of an orthorhombic perovskite structure (space group Pbnm - Fig. 1a). After the different nitridation treatments, no peak shift was observed but the perovskite structure was maintained except for the sample treated at the highest temperature (750 °C/8h), for which we observed secondary phases such as SrSn(OH)<sub>6</sub> (Fig. 1d).

The Raman spectra of SrSnO<sub>3</sub> and of nitrided samples are shown in Fig. 2. It can be seen five main bands related to the typical vibrational modes of perovskite. The main bands were assigned according to the literature [28-30]: bands at 115, 146 and 170 cm<sup>-1</sup> correspond to the Sr-SnO<sub>3</sub> network mode; the highest intensity band at 222 cm<sup>-1</sup> is attributed to A<sub>6</sub> mode which corresponds to the scissors movement of the Sn-O-Sn group along the c axis. The band at 258 cm<sup>-1</sup> is attributed to the angle variation of the O-Sn-O group at the ab plan and to the scissors movement at Sn-O-Sn perpendicular to the c axis. Additional lower intensity bands are observed at 304, 398 e 401 cm<sup>-1</sup> and correspond to the torsion mode of SnO<sub>3</sub><sup>2-</sup>.

After nitridation, it can be observed that the bands become broader and shift to a lower frequency region indicating a local symmetry breaking and an increase of the short-range disorder. Yoon et al. [26] observed a similar effect to the SrTiO<sub>3</sub>N system. The formation of short-range less orderly structures can be good since it is expected that SnO<sub>2</sub> octahedra distortions favor the photocatalytic activity [31].

The surface area (S<sub>BET</sub>) was estimated by N<sub>2</sub> adsorption isotherm (BET). As expected for powders prepared by solid state reaction, the oxide SrSnO<sub>3</sub> precursor presented a low value of S<sub>BET</sub> (Table 2). After thermal treatment under NH<sub>3</sub>, the value of S<sub>BET</sub> increased by 4 to 5 times. The increase of the superficial area is typical of the nitridation process and more relevant to samples prepared through the solid state reaction, it means, when the precursor oxide presents low superficial area [32]. Besides the increase of the superficial area, there was a darkening of the samples indicating the loss of O and/or N inclusion [19,27].

The content of N and O in the samples were measured by LECO® TC-600 analyzer and the Sr/Sn atomic ratio was estimated by EDS (Table 2). It is very difficult to calculate the composition of the nitrided samples because Sn<sup>4+</sup> reduction can occurs. Nevertheless, it is possible to observe that, considering only the content of N and O, the samples would present an excess of positive charge. Thus it is necessary to imagine a charge
compensation mechanism. Ebbinghaus et al. [19] suggest that the charge compensation mechanism for nitrided perovskites can occur through (i) the generation of vacancy of oxygen, (ii) anionic and cationic co-substitution, (iii) oxidation of the cation B, or (iv) reduction of the quantity of anions. EDS data show that SrSnO$_3$ is deficient in Sr and this deficiency tend to be enhanced upon nitridation. Moreover, XRD data do not show the presence of any secondary phase containing reduced tin (ex. SnO or Sn$^0$). Considering this, it is possible to assume that Sr$^{+2}$ vacancy is a possible mechanism to charge compensation; nevertheless Sn reduction can not be disregarded.

Moreover, LECO® TC-600 analysis showed very low content of N. Nonetheless, nitride samples showed a different photocatalytic behavior when compared to SrSnO$_3$ (which will be discussed hereafter), besides enhancement in surface area and darkening. So, it is assumed that nitridation indeed occurred but it probably took place mainly at the surface of the sample. Thus, it is reasonable to suppose that there may be a region where the N concentration is much greater (surface) and another region where the concentration is insignificant (bulk).

Table 2. Specific Surface Area, Oxygen and Nitrogen content (% weight) (measured by LECO®), Optical Band Gap and Color of samples.

<table>
<thead>
<tr>
<th>Nitridation Condition</th>
<th>Area/m$^2$g$^{-1}$</th>
<th>O %</th>
<th>N %</th>
<th>Sr/Sn atomic ratio</th>
<th>Band gap/eV</th>
<th>Color</th>
</tr>
</thead>
<tbody>
<tr>
<td>650 °C/21 h</td>
<td>10.0</td>
<td>17.2</td>
<td>0.114</td>
<td>0.899</td>
<td>4.0</td>
<td></td>
</tr>
<tr>
<td>600 °C/24 h</td>
<td>14.7</td>
<td>18.3</td>
<td>0.065</td>
<td>0.885</td>
<td>3.9</td>
<td></td>
</tr>
<tr>
<td>SrSnO$_3$</td>
<td>2.8</td>
<td>-</td>
<td>-</td>
<td>0.924</td>
<td>4.0</td>
<td></td>
</tr>
</tbody>
</table>

The band gap value of the samples was estimated from analysis by means of spectroscopy at the UV-Vis region (Fig. 3 and Table 2). An insignificant variation on the band gap value due to the nitridation treatment was observed. On the other hand, it is noted that there is a formation of a shoulder between 2.5 eV 3.5 eV (415 and 620 nm, respectively) due to the treatment with NH$_3$. This shoulder can be associated to the creation of permitted levels inside of the band gap, which can be related either to the incorporation of N or to the loss of reticular oxygen. These levels may come from the richer surface region in N. Yoon et al. [26] also describe the formation of a shoulder between 400 and 500 nm for the SrTiO$_3$:N.
The photocatalytic activity of the samples was evaluated through tests conducted under UV (254 nm) and visible light, Fig. 4. The photocatalytic activity of the SrSnO$_3$ is already known [12-15], but the commercial TiO$_2$ (P25) was applied in similar conditions for comparison effect.

When UV radiation is used, it can be observed a small activity of the SrSnO$_3$ and a performance about 30 % superior of the nitrided samples. Nevertheless, when the visible radiation is employed, it is observed a much more intense activity for the nitride sample - approximately 4 times greater than the one observed for the SrSnO$_3$ and 2 times greater than the one observed for the TiO$_2$ in the same conditions. The results of the photocatalytic tests suggest that the reduction of the band gap at the surface can be considerably greater than the one evidenced at the UV-Vis spectrum (which represents an average between surface and bulk). This happens because the nitridation occurs only at the superficial region and proves that even though only subtle modifications were observed in the structure of the SrSnO$_3$, they are efficient enough for the promotion of the photocatalytic activity at the visible region.

4. Conclusions

Nitridation of SrSnO$_3$ powders was performed in different conditions. Samples of SrSnO$_3$ submitted to nitridation in moderate conditions (600-650 °C) presented a significant increase in the photocatalytic activity at the visible region. The results arise from a superficial incorporation of N, even in small proportion. In more severe conditions (750 °C), the nitridation leads to the degradation of the structure.

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References


Figure

Fig. 1. X-ray diffraction patterns of a) SrSnO$_3$; b) SrSnO$_3$ treated under NH$_3$ flow at 600 °C/24 h; c) SrSnO$_3$ treated under NH$_3$ flow at 650 °C/21 h and d) SrSnO$_3$ treated under NH$_3$ flow at 750 °C/8 h. Magnification of X-ray diffraction peaks evidencing peak shift and broadening as function of nitridation treatment.

Fig. 2. Raman spectra of a) SrSnO$_3$; b) SrSnO$_3$ treated under NH$_3$ flow at 600 °C/24 h and c) SrSnO$_3$ treated under NH$_3$ flow at 650 °C/21 h.

Fig. 3. UV-visible spectra of a) SrSnO$_3$; b) SrSnO$_3$ treated under NH$_3$ flow at 600 °C/24 h; and c) SrSnO$_3$ treated under NH$_3$ flow at 650 °C/21 h.

Fig. 4. Discoloration of MB dye upon (A) UV and (B) visible light: a) SrSnO$_3$; b) SrSnO$_3$ treated under NH$_3$ flow at 600 °C/24 h; c) SrSnO$_3$ treated under NH$_3$ flow at 650 °C/21 h; and d) TiO$_2$. 
The image shows a series of X-ray diffraction patterns for SnSr(OH)$_6$.

- **a**: Wavenumber/ cm$^{-1}$
  - (110), (200), (220), (312), (400), (332)

- **b**: Diffraction pattern with peaks at different positions.

- **c**: Similar diffraction pattern as in a, but with additional peaks.

- **d**: Highest intensity peaks, likely corresponding to different crystallographic planes.

The graph on the left indicates the intensity in arbitrary units (a.u.).
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Highlights

- SrSnO$_3$:N presents a great potential to be used as photocatalyst under visible light.
- Very low amount of N increase significantly the photoactivity of SrSnO$_3$:N under visible light.
- Nitridation of SrSnO$_3$ was succeed at mild conditions and the perovskite structure was preserved.