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# SIMULATION OF SPACE RADIATION EFFECTS ON POLYIMIDE FILM MATERIALS FOR HIGH **TEMPERATURE APPLICATIONS**

**Final Report** 

 $by:$ 

Lawrence B. Fogdall **Sheridan S. Cannaday** 

Submitted to:

The Jet Propulsion Laboratory Pasadena, California

In Completion of Contract 954701

November 1977

**Boeing Aerospace Company** Seattle, Washington 98124



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#### **ABSTRACT**

Experiments have been conducted in simulation of vacuum and space radiation. The purpose of these experiments was to determine space environment effects on candidate materials for the solar sail film. Polymers, including metallized polyimides that might be suitable solar radiation receivers, were exposed to combined proton and solar electromagnetic radiation. Each test sample was weighted, to simulate the tension on the polymer when it is stretched into near-planar shape while receiving solar radiation. Exposure rates up to 16 times that expected in Earth orbit were employed, to simulate near-sun solar sailing conditions. Sample temperatures were controlled in one test; in another, they were allowed to reach equilibrium values under the influence of irradiation rate, solar absorptance and thermal emittance coefficients. Sample appearance, elongation, and shrinkage were monitored, noted, and documented in situ.

The relative performance of several polyimides was evaluated from the test results. Thermosetting polyimides tended to perform better than thermoplastics. Tha. is, the former showed less degradation or visual change in appearance than the latter. In the first test (at controlled temperature), Upjohn 2080 and Ciba-Geigy B100 thermoplastic polyimides had more discoloration and greater length changes than Kapton (thermosetting). In the second experiment (free-standing samples), Ciba-Geigy thermoplastic polyimide samples underwent as much as 30% shrinkage, whereas shrinkage of Kapton samples was limited to 8% or less. This effort was performed under Contract 954701 between the Jet Propulsion Laboratory and the Boeing Aerospace Company.

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### LIST OF FIGURES

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![](_page_5_Picture_4.jpeg)

#### LIST OF TABLES

![](_page_6_Picture_26.jpeg)

#### **ACKNOWLEDGMENTS**

The authors are grateful for the contributions of several individuals who were key to the successful completion of this work. Wallace Rowe and William Carroll at JPL contributed to many helpful discussions regarding experimental rationale. Paul W. Olson provided the in situ mechanical test apparatus, while Richard W. Earle was active in electrical test systems and all phases of the program through final documentation.

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#### Section 1

#### INTRODUCTION AND SUMMARY

Two experiments have been conducted at Poeing's Radiation Effects Laboratory to simulate the radiation environment for a solar sail in interplanetary space, and to ascertain some of the effects of that environment on samples of materials considered for possible application on such a sail. A solar sail would likely be constructed of very thin polymer sheet (e.g., 2.0 to 2.5 micrometers thick)<br>in order to minimize mass per unit area while maintaining adequate strength. The k in order to minimize mass per unit area while maintaining are parts to suggest polymer would be metallized to a high-reflectance, to maximize momentum oransier from solar radiation pressure to the sail and its payload (Reference 1). Propulsive force diminishes as the square of the distance from sun to spacecraft, so many solar sail applications are expected to be in the near-sun radiation environment for a significant portion of their mission life. The intensity of solar radiation could easily be 16 times that at Earth's distance from the sun, raising material temperatures in accordance with their absorbing and radiating properties. Section 4 of this report discusses the relationships of solar radiation, simulated radiation sources in the laboratory, and temperatures of test materials at equilibrium.

> In Section 2 of this report there is discussed the first experiment Boeing performed, in which unmetallized polymers were placed in the vacuum chamber and irradiated with 1.3-keV protons and UV, visible, and IR radiation at an ll-sun rate. The purpose of this first set of tests was to ascertain the effects of such an environment on unprotected film, in the event there were areas in the sail where metallization might be missing because of problems occurring during fabrication, handling, stowage, deployment, etc. This represented a worst case analysis. Sixteen polyimide samples were "draped" over temperature-controlled cylindrical sections and weighted with  $\sim$ l-gram masses, resulting in  $\sim$ lO0-psi loading throughout the 300-hour irradiation period. Relatively small changes in sample length were documented using photographs of the weights' positions as a function of time. Simultaneously the irradiated sample faces were photographed to document changes in appearance. Portions of the samples' irradiated zones which were in contact with the 300°C copper cylinders were chemically altered and embrittled during exposure. More detailed presentation of test results is included in Section 2.

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Section 3 of this report describes the second experiment or "second test stage" of the program, in which a more selective group of ten aluminized polyimide samples were irradiated with 1.3-keV protons and UV, visible, and IR radiation at a 16-sun rate. Test conditions differed from the first test set in that the ten samples were suspended so that their irradiated zones were "free-standing," i.e., not in contact with any temperature control mechanism, nor any portion of the vacuum chamber or other apparatus. Eacn sample acquired a steady-state temperature as determined by (1) its absorption and emission coefficients, (2) the proton and "UV" exposure rate, and (3) the rate of receiving secondary radiation from nearby surfaces. Again, each sample was weighted with a 1-gram mass, resulting in ~100-psi loading throughout the test, which lasted some 1100 hours. Appreciable reduction occurred in the lengths of 4 out of 5 samples made of Ciba Geigy polyimide. Lesser reduction in length occurred in all five DuPont Kapton polyimide samples. Detailed test findings are presented in Section 3.

Much high-quality photographic documentation of the condition of each sample took place before, during (in situ), and after each "test stage". Most of the transparency slides and enlarged photos for this documentation effort were delivered to JPL periodically during the program. Illustrations in this report summarize the test results and serve as examples of the types of in situ photo documentation undertaken. The printing of photos in this report is necessarily lower in quality than the original photographs, however.

#### Section 2

#### UNMETALLIZED FILMS EXPERIMENT

Both for this and for the other experiments conducted under this program, specialized combined radiation effects test chamber (CRETC) facilities at the Boeing Radiation Effects Laboratory (BREL) were modified to meet the test requirements. An "In Situ Mechanical Property Test Apparatus" was designed and fabricated to fit the CRETC II sample exposure chamber so that polymer test samples would face the incoming, combined beams from existing radiation sources placed around CRETC II. This sample apparatus is shown in Figure 1 before integration with the CRETC II vacuum chamber. Strips of flexible polymer film materials (Kapton and other polyimides described in Tables I and II) for subsequent mechanical property tests were

FILM MATERIAL	<b>SOURCE</b>	<b>JPL</b> LOG	MEASURED <sup>(3)</sup> <b>THICKNESS</b>	<b>TEST SAMPLE</b> NO./TYPE	
Plasma-etched commercial Kapton polyimide, etched side up(1)	NASA-Ames	141	0.165 mils		
Solvent-cast Kapton-type polyimide, glass side up	NASA- Langley	66	0.06	$\overline{2}$	
Commercial Kapton, NR-150-B2G adhesive bond, pristine (unetched) side up	Dupont/ <b>JPL</b>	118		$\overline{3}$	
Alkaline-etched commercial Kapton, 24-hr/600°F post cure, unetched side up	Dupont/ <b>JPL</b>	82	0.07	4	
Electro-cast polyimide, smooth side $up()$	<b>TRW</b>	135	0.055	5	
Ciba-Geigy solvent-cast B100X polyimide, glass side $up(2)$	JPL/ Ciba Geigy	140	0.300	6	
Upjohn #2080 solvent cast polyimide, glass side up(2)	JPL/ Upjohn	85	0.017	$\overline{\mathcal{L}}$	
Commercial Kapton, no post cure, alkaline-etched side up	Dupont/ <b>JPL</b>	106	0.147	8	
(1) Thermosetting polyimide. (2) Thermoplastic polyimide. 31 mil=25.4 micrometers.					

SOLAR SAIL FILM MATERIALS IRRADIATED TABLE I. IN BOEING CRETC II TESTS

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![](_page_10_Picture_0.jpeg)

![](_page_10_Figure_1.jpeg)

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SAMPLE <b>TYPE</b>	<b>MEASURED</b> THICKNESS	<b>SAMPLE</b> <b>WIDTH</b>	CROSS-SECTION AREA	SAMPLE LOADING
	$0.165$ mils $(1.65 \times 10^{-4} \text{ inch})$	0.219 inch	3.61 x $10^{-5}$ in <sup>2</sup>	91.5 $1bs/1n^2$
$\overline{2}$	$0.06$ mils	0.219	$1.31 \times 10^{-5}$	168 $1bs/in2$
$\overline{3}$	(Joined sections - no single thickness)			
$\overline{4}$	0.07	0.219	$1.53 \times 10^{-5}$	144 $1bs/in2$
5	0.055	0.219	$1.20 \times 10^{-5}$	147 $1bs/in2$
$6\phantom{a}$	0.300	0.219	6.56 $\times$ 10 <sup>-5</sup>	33.6 $1bs/in2$
$\overline{7}$	0.17	0.219	3.72 $\times$ 10 <sup>-5</sup>	59.2 $1bs/in2$
8	0.147	0.219	3.22 $\times$ 10 <sup>-5</sup>	68.4 $1bs/in^2$

TABLE II. TEST LOADS IN SOLAR SAIL SAMPLES BEFORE RADIATION EXPOSURE

placed in contact with the curved sectors of two cylinders shown in Figure 1. These cylindrical bodies were heated to and controlled at a temperature of +300°C throughout the test period. Each sample was made longer than the zone visible in Figure 1 so that  $\sim$ 1-gram masses could be suspended at the bottom of each sample to provide non-variable tensile loading of ~100 psi. Differences between projected sample thicknesses and actual measured values of film thickness account for sample loading less than and greater than 100 psi in Table II. The 16 masses with their pointers were visible from behind the test apparatus (see Figure 2) throughout the 300-hour exposure period of this "first test stage".

The CRETC II facility into which the sample apparatus was integrated for this program is shown in Figure 3.

Besides the 16 "strip" samples that were loaded by means of weights at their lower ends, 16 smaller ESCA test samples were placed in intermediate rows for exposure and chemical analysis later elsewhere. Figure 4 is a closeup photo showing all four rows, for a total of 32 test samples.

The basic plan involved shielding the top two rows of samples during the first 13 days of proton and "UV" exposure and then raising the shield for an additional day of exposure. The movable shield is shown "open" or "up" at the top of Figure 1. Thus, the top two rows of samples were exposed for one day and the

![](_page_12_Picture_0.jpeg)

Figure 2. Weights Providing Tensile Loading of 16 Unmetallized Polyimide Films

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![](_page_13_Picture_0.jpeg)

Figure 3. Boeing Combined Radiation Effects Test Chamber II and<br>X-25 Solar Simulator

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![](_page_14_Picture_0.jpeg)

Figure 4. Array of Tensile and ESCA Test Samples Prepared From Unmetallized Polyimides.

bottom two rows received up to 14 days of irradiation.. (Some of the sample area in the bottom two rows was partially out of the UV beam during the 14th day, as explained below further. The exact UV coverage of the bottom row of samples during the 14th day may be seen most precisely in photographs and slides previously delivered to JPL.) After this all samples were removed from the CRETC II facility for further analysis. A Spectrolab Spectrosun X-25 solar simulator was modified to produce ll-sun electromagnetic radiation, including the solar UV continuum (<sub>00.25</sub> to C.40 micrometers, or 5 to 3 eV) from the simulator's zenon arc-discharge source. Figure 5 is a uniformity map of the simulator's output beam as measured during final calibration. The positions of the bottom two rows of samples during their<br>exposure period of 13 days are shown as overlays on the simulator's output beam<br>pattern in Figure 5. The "UV" beam was later shifted to expose pattern in Figure 5. The "UV" beam was later shifted to expose the upper two rows

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![](_page_15_Figure_0.jpeg)

![](_page_15_Figure_1.jpeg)

![](_page_15_Figure_2.jpeg)

![](_page_15_Figure_3.jpeg)

of samples, which if overlaid on Figure 5 would occupy an equal area.

The 1.3-keV proton beam from the CRETC II proton source was checked out relative to the sample array size. Figure 6 indicates the proton beam uniformity along the two Faraday cup tracks. The sweep of these tracks across the sample array is apparent from Figure 1. The "100%" proton radiation rate used during the "first test stage" was 4 x 10<sup>9</sup> protons/cm<sup>2</sup>-sec. As seen in Figure 6, it is possible that the top two rows received somewhat more intense proton radiation.

acuum levels during the experiment were limited only by sample outgassing haracteristics. Vacuum as measured by an ionization gauge ranged from lo" so lo t**o**rr during**pe**ak**o**utga**ss**in**g**periods.

Table I gives the type number (1 through 8) assigned by JPL to each prepared film. As stated before, the lower row of "strip" or weighted samples and the lower row of small samples -- both called rows B -- were exposed first while the upper two rows A were shielded.

A side view of the sample block would shows that  $\frac{1}{2}$  to both  $\frac{1}{2}$  to  $\frac{1}{2}$  is  $\frac{1}{2}$  in  $\frac{1}{2}$  is  $\frac{1}{2}$ "strip"(weighted)samples;two rowsof IA t**o 8A** I; (**l**st1**3 d**ays **small**, framed samples intended for later **with the steading of 14** and the test) ,' **te**s**t**) chemicalanalysis,b**e**hindw**h**ich are the ia ! weights and reference lines for the upper  $ESCA A \nightharpoonup \$ row of strip samples; and the lower row of **ESCA** BILITY , **EXCAPT THE Radiation**<br>weighted samples. This sample arrangement is sketched at the right. Inside the uppermost and lowermost curved contact surfaces are electrical heaters, to pro- **Samples**<br>vide the elevated test temperatures for **18 to 8** the weighted samples. A thermocouple **the electer** measures the temperatures of each row of<br>weighted samples. A third thermocouple w**e**ightedsam**p**les. **A** third thermocouple m**e**as**u**resth**e** t**e**mp**e**ratureof th**e** blockon L**J** which the two rowsof ESCA sample**s**are mounte**d**.

![](_page_16_Figure_5.jpeg)

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The lower two rows of samples (rows B) were first irradiated with UV and protonsfor 1**3** day**s**,while the upper two rowsof sample**s** (A an**d** ESCAA) were shielded from direct radiation exposure. However, the contact surfaces for all<br>**EXIGINAL PAGE** 10

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strip samples (1A to 8A, and 1B to 8B) were at a temperature of 290°C to 300°C during this time. The ESCA samples' thermocouple indicated a temperature of approximately 290°C. During the first two hours of exposure, the weight attached to sample 7B (Upjohn 2080) rose approximately 70 mils. This was first observed by telescope sightings after 1.8 and 2.2 hours exposure. The weight for sample 6B (Ciba-Geigy B100) rose approximately 30 mils during this time period. The weight positions for all other samples in row B (the row being exposed) were unchanged. Visual and photographic observations of the front (exposed) side of samples 1B through 8B during this time interval indicate that material 7B darkened very quickly compared to all others, and that 6B's exposed-side visual appearance was unchanged until later in the test.

It is difficult to determine the effective sample length over which shrinkage (or stretching) would apply. By the end of the two-week test numerous locations in the exposed zones were observable at which polyimide material bonded to its heated copper substrate (the curved contact base). Thus, it is impossible to state with certainty the amounts of shrinkage (weight position change) as percents of sample length. The total shrinkage might or might not be uniform over the sample length from top to bottom. In some cases (see below, where exposed-side photographs are discussed) sample length change may be concentrated at the chemically altered "burn spots", "puckers", or "gathered" sites. The origin of these spots is postulated to be locations where there was lack of good contact to the heat-dissipating copper block.

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It is possible to approximate the length of various samples suspended below their respective exposed zones. The group of reference lines below each row of weighted samples spans 16 times 40 mils, or 0.64 inch. Thus, the entire suspended length of the strip samples (below the exposure zone) is on the order of one inch. From this consideration we see that the shrinkage noted above is less than 10 percent of the sample's suspended length.

Further, smaller changes in sample length were observed as the test exposure period continued. Telescope sightings after 48 hours indicated: 7B shrank 10 mils more (total, 80 mils); 6B shortened 40 mils more (70 mils total); and other Row-B strip samples changed in length less than 10 mils. Furthermore, observation of the samples in Row A (heated but not exposed to UV or protons prior to opening their shield) revealed that sample 5A (TRW electro-cast) stretched 20 mils, as determined by its weight position, and the weight for sample 6A rose 30 mils.

 $\mathbf{1}$ 

Subsequent photographs and telescope sightings confirmed all these observations except the stretching of sample 5A, indicating (if it was a valid observation) it was transient whereas the others were permanent changes. Telescope sightings after approximately 90 hours and 130 hours exposure of Row-B samples indicated no further changes. The visual appearance of Row B samples as photographed after 13 days of irradiation by protons and "UV" is shown in Figure 7.

![](_page_18_Picture_1.jpeg)

**F**i**g**ure**7**. **E**ff**e**c**tso**f C**o**mbi**n**e**dU**V/Pr**o**t**onExpos**ur**eo**f 1**6 Un**m**e**talli**zed** Polyimide Film Samples for 13 Days (11-Sun Rate)

## E\_.\_p**o**sureof Row-A Sam**ples**

The 16 polyimide samples in Rows A (8 tensile-test strips plus 8 ESCA) were irradiated by protons and UV at an 11-sun rate for 24 hours. Photographs of the irradiated film surfaces in contact with the cylindrical copper surfaces were taken frequently, to document the rate of thermal degradation. As with all other portions of the program, high quality color enlargements of the changes in these samples were forwarded to JPL. Figure 8 shows the appearance of these samples just after the beginning of exposure. Figure 9 is a photograph taken almost 24 hours later, just before the end of the l-day irradiation.

![](_page_19_Picture_0.jpeg)

![](_page_20_Picture_0.jpeg)

Condition of Unmetallized Polyimide Films After 1-Day Proton/UV Exposure. Figure 9. į

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Regarding movement of weights below samples 1A through 8A, no significant position changes were observed 1.2 hours after beginning the combined UV/proton exposure. Within three hours after starting the irradiation, however, the weight below sample 3A (the bond joint) had risen approximately 40 mils and shifted slightly to the left as well. This weight continued to rise throughout the 24hour exposure, whereas the weights below the other seven samples remained relatively stable in their positions. Table III summarizes the position changes for the weight below this bonded Kapton film sample. Figure 10 shows the location of all 16 weights at the conclusion of the "first test stage."

EXPOSURE TIME	RISE IN WEIGHT POSITION
0 to 1.2 hours 3 5 7 8 101/2 $14$ $1/2$ 18 1/2	<b>Baseline</b> Up 40 mils Up 50 Up 60 Up 70 <b>Up 80</b> <b>Up 90</b> Up 100 Up 110
23	

TABLE III. WEIGHT POSITIONS OF TENSILE SAMPLE 3A

The darkening of various test samples during irradiation is thought to result from "carbonizing" of the materials during "thermal runaway." That is, as material temperature limits were exceeded they became more absorptive, leading to yet higher temperatures in the affected films. Thermal runaway was a major factor in enhancing the damage and rate of damage in affected samples.

![](_page_22_Picture_0.jpeg)

Figure 10. Positions of Weights Below 16 Tensile Samples After<br>Unmetallized Films Experiment.

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#### Section 3

#### **H**ETALLIZED FILMS EXPERIMENT

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This experiment plus related effort to be discussed in Section 4 constituted the "second test stage" of this program. Boeing irradiated 10 metallized polyimide film samples with "UV" (i.e., simulated solar electromagnetic radiation including ultraviolet radiation) and with protons for nearly llO0 hours. The metallization on the samples consisted of vapor deposited coatings of  $\sim$ 125 A of chromium (emitter) on one surface and \_lO00 A of aluminum (reflect**o**r) on the side facing the radiation. Unlike the previous experiment, these "strip" samples' entire lengths made no contac**t** with other temperature-controllingsurfaces. Rather, they exchanged radia**t**ive energy with a water-cooled plate behind the exposure zone, and acquired equilibrium temperatures dependent on their solar absorptance (a<sub>S</sub>) and thermal emitted (c) processes the contract (c) Processes and the contract (c) Processes and the contract of the contract of the contract of the contract of the c coefficients and the exposure rate. During all but the first 22 hours of this experiment, irradiation took place at a 16-total-sun rate, as determined by thermopile and calorimetric measurements. The proton exposure rate was also proportionally greater than in the earlier experiment, namely 6 x  $10^9$  protons/cm $^2$ -sec. This value is approximately 16 "solar winds."

The metallized film materials are listed in Table IV. The numbers 9 through  $\blacksquare$ metallized films (Section 2). The face of the Kapton and Ciba Geigy polymers to be irradiated had been aluminized by vapor deposition to a thickness of lO0 nm except for specific gap areas on six of the specimens that allowed irradiation of<br>the base polymer. The gaps were transverse to the lengths of the samples. Most<br>samples had been cut to exposure size along (not transverse t B: t**he** ba**se po**lym**e**r. **T**h**e** ga**p**s were tra**n**sverse to the lengths of the samples. Most <sup>P</sup> **s**ample**s h**ad **b**een cut t**o** exp**o**sure size along (not transverse to) characteristic I **p**r**o**c**ess**i**ng** mark**s**. **T**h**e e**ntire back **s**ur**f**ace of each sample had been vapor-deposited with chromium lO nm thick t**o** i**n**c**r**ease emittance. This coating system (**o**f these thickne**s**ses) is referred to as the "benchmark" coating.

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**F**igure II is a closeup photograph of the lO metallized 1\_Ims prior t**o** irradiation. The photograph is taken from the side to be exposed to UV and protons. The gaps that were intenti**o**nally put in the aluminum overcoating by selective masks during vapor deposition are visible in some samples. Views of the reverse side of the sample apparatus as modified for this experiment (Figure 12) show the l-gram mass below each sampl**e** and show horizontal grid markings that allowed sample length changes to be measured in situ during exposure.

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I. t**o** b**e** m**e**as**u**red in situ during exposure.

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## TABLE IV. ALUMINIZED SAMPLES FOR SECOND TEST STAGE

![](_page_24_Figure_1.jpeg)

![](_page_24_Picture_39.jpeg)

Benchmark coating consisted of 125 Å of Cr on the emitting side of film and 1000 A of Al on the reflecting (sun) side.

<sup>2</sup>This sample (#18) was cut for irradiation testing in a direction 90 $^{\circ}$  from the direction of cut of the other Ciba-Geigy samples. The others were cut with the long direction parallel to the machine direction. This s verse direction. This sample was dimensionally much more stable than the other Ciba-Geigy samples. (See Figure 11).

![](_page_24_Picture_5.jpeg)

![](_page_25_Picture_0.jpeg)

Figure 11. Metallized Polyimide Film Samples Before 16-Sun UV/Proton Irradiation.

Many of the details involved in setting "UV" exposure intensity and attempting the temperature calibration for the chamber are left for the discussion in Section 4 of this report. A principal objective was to expose the temperature-sensitive films to no more than 16 total suns. During preirradiation photography of the samples' initial conditions it was determined that the Ciba Geigy polymers that were "longitudinally cut" had already shrunk appreciably under illumination by the photo lighting source. Irradiation by the solar simulator continued the length shrinkage in these samples and to a lesser extent in the Kapton films. Data on sample length changes, as reduced from periodic photographs of sample weight positions in situ before, during, and after the 1100-hour exposure period, is presented for Kapton films in Figure 13 and for Ciba Geigy films in Figure 14. Since the current through the UV source lamp must periodically be increased to

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![](_page_26_Picture_0.jpeg)

Figure 12. Weight Positions Below Suspended Metallized Polyimide Films Prior to Exposure

c**o**mpensate for lamp aging, an**d** since optical surfaces between the source lamp and the samples must periodically be cleaned, Figure 13 includes a charting of these adjustments to show the degree of correlation between small intensity adjustments and sample lengths. The main coincidence of sample length changes and solar simulator intensity increases is seen in Figure 13 to be after 22 hours of exposure when, as described in Section 4, the rationale for setting "UV" exposure intensity was modified by JPL. Figure 14 compares the stability of sample length among Kapton samples as a group, individual Ciba-Geigy samples cut longitudinally, and Ciba-Geigy sample No. 18 cut transversely.

The irradiated faces of all 10 samples were affected by proton/UV exposure, as shown in Figures 15, 16, and 17. The photouraph in Figure 15 was taken at the beginning of the 1100-hour irradiation period, before "necking" or "dog-boning"

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![](_page_27_Figure_0.jpeg)

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**REAL ALL REAL** 

![](_page_28_Figure_0.jpeg)

EXACT CHANGE IN SAMPLE LENGTH (mils)

22

![](_page_29_Picture_0.jpeg)

**F**i**g**ur**e 15**. **Appe**arance**o**f Metalli**z**ed**F**ilm**s** Just **A**fter Start **o**f llOO-Hour Test (9-Sun Intersity).

sample shaping began in the exposure zone. After the first 20 hours of exposure approximately 80% of the Ciba Geigy samples' irradiated lengths had narrowed, as (presumably) their transition temperature was exceeded. (Nevertheless, note these had undergone shrinkage, not elongation.) The transverse-cut Ciba Geigy sample (no. 18) appears in Figure 16 to be an exception to this statement about narrowing, although it did neck down soon after the exposure intensity was increased to 16 suns. The shape of the "transition line" (dashed lines in Figure 16) is not the shape of the solar simulator beam. (The shape of the solar simulator beam is indicated in Figure 18, and the uniformity of the proton beam is shown in Figure 19.) Toward the end of the experiment, extensive curling, warping, and twisting has affected both the Kapton and the Ciba Geigy materials, as shown in Figure 17.

![](_page_29_Picture_3.jpeg)

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![](_page_30_Picture_0.jpeg)

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Figure 16. Metallized Films After 20 Hours Exposure.

![](_page_30_Picture_2.jpeg)

Figure 17. Metallized Films After 1000 Hours Exposure

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![](_page_31_Figure_0.jpeg)

![](_page_31_Figure_1.jpeg)

![](_page_31_Picture_2.jpeg)

Combined narrowing plus shortening of a sample implies increase in material thickness (which was n**o**t measured) and/**o**r loss of v**o**latile molecules through **o**utgassing in the vacuum test chamber. Vacu**u**m gauge indications confirm the latter.

Thermal conditi**o**ning **o**f the Ciba Geigy test sampl**e**s before **d**elivery to B**o**eing for irradiation may have been a factor in their greater instability. The Ciba Geigy test sampIes were "soaked" at 250"C in va**c**uum for 5 minutes. JPL believe**s** that a l**o**nger c**o**nditi**o**ning perio**d** might have improved the **d**imensional stability **o**f the Ciba Geigy material tested.

### Section 4 IRRADIATION CALIBRATION EXPERIMENT

as designed by JPL and Boeing to compare solar illumination ratory with the intensity of radiation anticipated for a A series of experiments was eventually performed, by which doing UV dosimetry in the laboratory were checked against a  $\bullet$  have solar absorptance ( $\alpha_{\mathbf{c}}$ ) and thermal emittance ( $\epsilon$ ) those of the candidate solar sail film materials. The rred to Boeing and installed inside the CRETC II sample in front of the test film sample plane (Figure 20).

![](_page_33_Picture_2.jpeg)

Aluminized Kapton Calorimeter Exposed to 16-Sun Solar Simulator Beam.

A solar sail film in a plane, receiving radiation on its front side and emitting radiation on both sides, would equilibrate at a temperature T such that

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T^4 = \frac{\alpha_S (SC) (SR)}{\sigma(\epsilon_f + \epsilon_b)}
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where SC is the solar constant, SR is the sun rate, and  $\sigma$  is the Stefan-Boltzmann constant. It can be shown that solar sailing under near-sun conditions (namely, SR = 16 times the solar intensity at Earth's orbit) results in an equilibrium temperature of approximately +250°C for aluminized Kapton that has an emittanceenhancing coating on the back side. On this basis a calorimeter closely simulating a solar sail film in a and a characteristics could be used in a vacuum chamber to set the intensity of radiation from a solar simulator, provided secondary (bounce") radiation in the chamber were comparable to view factors (spacecraft geometry) in space.

The calorimeter discussed above is a circular disk with a diameter of two inches and thickness of 0.1 inch. Hence it approximates a solar sail film in a plane. It was fabricated by TRW with 2 chromel-alumel thermocouples on the unirradiated side, one at the center and the second 0.707 inch from the center  $-$  the radius which divides the calorimeter's total area into halves. Assuming edge losses of heat are small around the perimeter of the calorimeter, its thermocouple(s) will indicate the same equilibrium temperature as stated above for the case of irradiation and partial absorption on its front side and thermal emission from both sides (front and unirradiated back).

The irradiation source must be discussed from the standpoint of its spectral content, since absorption on the front face of a solar sail or the calorimeter varies with the wavelength. It is presently impractical to obtain total irradiation levels like 16 suns (to simulate near-sun solar sailing trajectories) along with "close filtering" spectral matches using a simulator mobile enough to be combined with an ultra-high-vacuum chamber and charged particle accelerator. (That is, the nearly immobile X-75 and X-200 style solar simulators that have enough energy to trade for close spectral filtering are not currently available, and the more mobile, X-25type power limitations must be settled for.) As a result, the relatively large emission from a xenon arc in the near infrared (0.8 to 1.2 micrometers) is utilized along with the desired continuum across the visible and near-ultraviolet wavelength regions. Table V shows the relative spectral energy applicable to the start of

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exposure of metallized film samples (the second principal experiment for this program, described in Section 3 of this report).

<b>BANDWIDTH</b> (micrometers)		<b>SCALED</b> <b>DATA</b>	DATA MATCHED TO I SOLAR CONST. <u>(watts/m2)</u>	<b>ENGINEERING</b> <b>STANDARD</b> <u>(watts/m<sup>2</sup>)</u>	$\frac{\partial J}{\partial \theta}$ DEVIATION
0.25	0.35	4.7	27.3	58.5	$-53.3$
0.35	0.40	06.7	38.9	56.9	$-31.6$
0.40	0.45	8.5	51.3	86.8	$-40.9$
0.45	0.50	10.7	64.6	100.9	$-36.0$
0.50	0.60	22.8	137.7	177.0	$-22.2$
0.60	0.70	22.4	135.3	151.5	$-10.7$
0.70	0.80	19.5	117.8	123.6	$-4.7$
0.80	0.90	33.0	199.3	99.3	$+100.7$
0.90	1.00	32.2	194.5	82.6	$+135.5$
1.00	1.20	22.2	134.1	120.7	$+11.1$
1.20	1.50	18.1	109.3	111.8	$-2.2$
1.50	1.80	10.5	63.4	66.9	$-5.2$
1.80	2.20	8.6	51.9	43.8	$+18.5$
2.20	2.50	4.1	24.8	19.9	$+24.6$

TABLE V. X-25 SOLAR SIMULATOR RELATIVE OUTPUT

The three bands with the longest wavelengths partially represent emission from the incandescent electrodes of the xenon arc source lamp. This we eliminate by insertion of a water window between the solar simulator and the sample exposure chamber, since pure water absorbs wavelengths longer than ~1.4 micrometers and transmits shorter wavelengths down to approximately 0.2 micrometers. "Scaled data" represents energy arriving at the detector on a modified Beckman DK-1/spectroradiometer measuring system. This data is then matched to the spectral shape of the solar constant in space (air mass zero) in accordance with NASA/IES engineering standards. The deviations of spectral power obtainable, compared with the ideal represented by latest measurements of actual solar output, are included in Table V.

Between the first and second test stages for this program, Boeing modified the in situ mechanical property test apparatus to provide for insertion of the disk calorimeter in front of the sample plane, and removal at any time using an in situ remote manipulator. As the metallized film experiment was about to begin, the calorimeter disk was moved into the solar simulator's "UV" beam. The group of 10

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test samples intervened between the calorimeter's emitting back surface and the water-cooled sample block, modifying the effective emittance of the calorimeter an unknownamount. It wa**s** determinedthat the solarsimu**l**at**o**rnee**d**e**d**to be **s**et at an output level considerably lower than that indicated as 16 suns by a pyrheliometer, if a calorimeter equilibrium temperature of +250°C or below were to be maintained.A pyrheliometerc**o**mm**on**ly**used**f**o**r this ty**p**e**o**f **dos**im**e**trythen in**d**icated that approximately nine total suns and seven UV suns were incident on the sample (or calorimeter)plane underthe**s**e c**o**ndition**s**.(See Figure1**3**).

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After 22 hours of exposure the decision was made to return to the usual pyrheliometer do**s**imetrya**s** a basis f**o**r **s**ettings**o**lar simulator**o**utput intensity. Thi**s** decision acknowledged that the presence of an array of test samples between the calorimeter and the water-cooled sample block effectively altered the  $\epsilon$  properties **o**f the cal**o**rimeterso that its indicatedtemperatureshouldn**o**t be usedas a basi**s** for setting the solar simulator's output level. The solar simulator output was accordinglyincreasedto an intensityof 16 total sunsas determine**d**by **p**yrheli**o**meter dosimetry readings. This change is shown in the top portion of Figure 13. The calorimeter was not reintroduced into the "UV" beam at this time, because it was known that the solar simulator intensity was now great enough to degrade the calorimeter materials thermally. That is, previous short-time insertions of the calorimeter into such an intense beam had shown by the slope of the response curve on a strip chart recorder that an equilibrium temperature substantially above +**3**30**°**Cwould be reached.

At the end of the metallized films (Test Set *!*:o. 2) experiment the solar simulat**o**r**o**utputlevel,whichwa**s** still beingmaintainedat 16 t**o**tal suns u**s**ing pyrheliometer dosimetry, was again compared with indicated calorimeter temperature. A**s F**igure13 sh**o**w**s**,the outputlevelof the solar simulatorhad been a**d**justedand the optics cleaned from time to time throughout the ll00-hour exposure period of the "second test stage." The final output dial settings, not easily relatable to the output settings at 22 hours, understandably also resulted in calorimeter temperature indicationstoo high t**o** be sustainedsafely. The arrayof samplesat test en**d** was **q**uite twistedand irregularin shape,as shown in Figure17. Porti**o**ns**o**f the back surfaces of some test samples can be seen facing the radiation sources. In such areas greater absorption occurred, due to the existence of the emissive coating. This led to higher temperatures than planned for the specimens. The twisting modified the calorimeter's effective emittance to an unknown and different degree than at the start of the experiment. It was determined that the only precise data that could be

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obtained would be a comparison of the calorimeter's temperatures for cases of samples present and samples absent (for a certain reduced simulator output level). Subsequently an equilibrium temperature of +334°C was obtained from the calorimeter with samples still present and a certain solar simulator dial setting. After the ten test samples had been removed a calorimeter thermocouple temperature of +330°C was obtained with the same dial setting. All temperature readings noted here were obtained using the thermocouple at the center of the calorimeter's unirradiated side. Readings from the thermocouple closer to the unirradiated side's edge were consistently 3°C less.

With the ten test samples removed, an outgassed film was clearly observable on the water-cooled sample block just behind the test sample plane. The patterns of outgassed material were typical of thin film interference coatings. As with all other phases of the program, high quality color photographs were taken to document the condition. Color enlargements were subsequently forwarded to JPL. Figure 21

![](_page_37_Picture_2.jpeg)

Outgassed Thin Film Pattern Behind Metallized Figure 21. Polyimide Films Irradiated for 1100 hours.

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is a r**e**pr**o**duction of t**h**e patterns of outgassed molecul**e**s. Clos**e e**xamination I of the original photos shows the patt**e**rns can b**e** relat**e**d to polyimid**e** sampl**e** placement and subsequent twisting during the llO0-h**o**ur irradiati**o**n per**i**od. Ev**e**n th**e e**levat**e**d p**o**sitions of the l-gram w**e**ights below the sh**o**rtest (Ciba G**e**igy) metalli**z**ed **s**amples can be discerned from a close examination of the ph**o**tos.

The emittance of the thin film pattern of outgassed molecular matter was n**o**t mea**s**ure**d**. It has been estimated, however, as being substantially greater than the emittance of the nickel-plated, water-coole**d** sample block. The small 4° C difference in calorimeter equilibrium temperatures when the ca**s**es of **s**amples present and samples ab**s**ent are compared shows that the array of narrow, twi**s**ted te**s**t **s**amples f**o**llowing llO0 hours of irra**d**iation had net radiative exchange characteristics n**o**t unlike the **o**utgassed thin film pattern.

From the test chamber standpoint, further control over the reflective and radiative properties of surfaces surrounding test specimens should be exercised during future experiments of the type performed for this program. Boeing, in fact, is already developing improved test techniques and configurations, including use of wider samples and chamber surfaces with controlled reflectance (Reference 3).

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