# NASA Contractor Report 3475

NASA CR 3475 c.1



Radiation Testing of Composite Materials. In Situ Versus Ex Situ Effects

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Richard M. Kurland, James F. Thomasson, and William C. Beggs

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# Radiation Testing of Composite Materials, In Situ Versus Ex Situ Effects

Richard M. Kurland, James F. Thomasson, and William C. Beggs
TRW Defense and Space Systems Group
Redondo Beach, California

Prepared for Langley Research Center under Contract NAS1-15848



Scientific and Technical Information Branch

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#### **FOREWORD**

The work reported herein was performed by TRW Defense and Space Systems Group (DSSG), Redondo Beach, California, under NAS1-15848 and was sponsored by the National Aeronautics and Space Administration, Langley Research Center (NASA/LaRC), Hampton, Virginia. Mr. W.S. Slemp, Environmental Effects Branch, Materials Division, served as the NASA/LaRC technical representative.

The program at TRW DSSG was performed under the auspices of the Mechanical Engineering Laboratory (MEL) of the Space Systems Division, Mr. M.E. White, MEL Manager. TRW Program Manager was Mr. R.M. Kurland. Principal Investigator was Mr. J.F. Thomasson. Responsible engineer for High Energy Radiation Test Operations was Mr. W.C. Beggs. Several other engineers and laboratory support personnel contributed to the success of this program. Their contributions are acknowledged below:

M.D. Cawley	Vacuum Chamber Design and Tensile Property Measurements
G.I. Fukumoto	Tensile Property Measurements
N. Harvey, Jr.	Van de Graaff Calibration and Operation
P. Guilfoyle	Van de Graaff Operation
J.C. McKeegan	Van de Graaff Operation

Certain commercial materials and products are identified herein in order to specify adequately which materials and products were investigated in the research effort. In no case does such identification imply recommendation or endorsement of the product by NASA, nor does it imply that the materials and products are necessarily the only ones or the best ones available for the purposes. In many cases equivalent materials and products are available and could produce equivalent results.

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#### NOMENCLATURE

Below is a list of the most frequently used symbols and acronyms in the report. Except for radiation flux and fluence notations, all analysis and test data were developed using the English system of units and later converted into SI units for reporting purposes (with English units in parentheses).

A cross-sectional area of test specimen; ampere

 $A_s$  frontal area of electron scatterer

AMFL Air Force Materials Laboratory

C coulomb

C6000/P1700 graphite/polysulfone composite material

D dose

 ${\bf D_R}$  dose, fully backed specimen

 $D_{o}$  dose, unbacked specimen

DVM digital voltmeter

E modulus of elasticity; electron energy level

E on-target average electron energy level

 $E_{o}$  VDG emerging electron energy level

 ${\sf E}_{\sf p}$  most probable emerging electron energy level

EA exit aperture

 $F_{TU}$  ultimate tensile strength

 $F_{TY}$  yield tensile strength

FUV far ultraviolet

ISTTC in situ tensile test chamber

L gage length

LaRC Langley Research Center

LN<sub>2</sub> liquid nitrogen

LVDT linear variable differential transducer

#### NOMENCLATURE (Continued)

MeV million electron volts

NASA National Aeronautics and Space Administration

NUV Near ultraviolet

P applied tensile load

Py yield tensile load (at 0.01 strain)

 $P_{ij}$  ultimate tensile load

Q<sub>FC</sub> Faraday cup charge (current)

 $Q_{FA}$  exit aperture charge (current)

S standard deviation (small population)

SESL Space Environmental Simulation Laboratory

TIGER Monte Carlo electron transport computer code

TLD thermoluminescent dosimeter

T300/934 graphite/epoxy composite material

T300/5208 graphite/epoxy composite material

VDG Van de Graaff generator

WPAFB Wright Patterson Air Force Base

 $\overline{X}$  average value of population of  $X_i$ 's

d distance

e electron

keV thousand electron volts

 $\Delta\Omega$  solid angle

δ deflection; change-in-length

ε strain or elongation

 $\epsilon_{_{11}}$  ultimate strain or elongation

σ stress

#### 1.0 SUMMARY

Three  $[\pm 45/\mp 45]$  laminate composite material systems, T300/5208, T300/934, and C6000/P1700, were irradiated up to a bulk dose of 1 x  $10^{10}$  rads using a mono-energetic fluence of 700 keV electrons from a Van de Graaff accelerator. Irradiations were performed in vacuum at 20°C (68°F) and 120°C (248°F). Post-irradiation tensile testing was conducted in situ (while being irradiated), in vacuo, and ex situ in air.

Because of the exploratory nature of the study, only a limited number of specimens per material and test condition were evaluated; thus, engineering design data was not generated. Nevertheless, the data were representative of material behavior. The results were useful in establishing trends and in developing general conclusions about the significance of post-irradiation test environments and the suitability of the materials to withstand long-term space radiation.

The radiation-induced changes to the tensile properties (modulus of elasticity, ultimate strength, yield strength, ultimate elongation) for all materials were small. There was indication that radiation improved some tensile properties in that there was a slight increase in modulus and strength. There was, however, a decrease in ultimate elongation.

Because radiation damage thresholds may not have been reached during the study or because radiation-induced changes were small for these materials, the ability to realistically evaluate the importance of post-irradiation test environments was compromised. The differences among in situ, in vacuo, and ex situ test results were small.

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#### 2.0 INTRODUCTION

#### 2.1 BACKGROUND

Future space systems will use increasing amounts of composite materials in the primary structural system. Composites are a natural candidate material due to their unique combination of good strength, high stiffness, low density, high dimensional stability, and because their properties can be tailored for specific applications. However, the reliable performance of composites in the long-term space radiation environment is presently unknown. Since advanced composite materials may be susceptible to radiation-induced changes in their mechanical properties, it is important that the radiation effects on critical properties be well characterized to ensure that only relatively stable materials are used.

The significant space radiation environment components which may cause changes in both surface and/or bulk properties are solar ultraviolet radiation and charged particles (electrons and protons). The other environmental components such as cosmic rays, neutrons, alpha particles and X-rays are considered to have negligible probability for creating measurable effects in the materials because of the relatively less severe energy levels or fluences. The charged particle radiation is of special importance because of its ability to penetrate deep into the bulk of the material and thus affect an appreciable percentage of the material cross-section.

Organic materials, like polymer films, adhesives and resin systems, because of the dependence of their bulk physical properties on molecular weight and the integrity of their carbon-to-carbon and other covalent linkages, are more susceptible to radiation-induced changes than metals and inorganic materials. For organic materials, changes in material properties due to ionization of atoms dominates those changes that may result from displacement of atoms. Thus, for most composite material systems, the resin and resin/fiber interface will be more vulnerable than the reinforcement fibers.

The types of reactions that take place in organic materials exposed to space radiation are chain scission and cross-linking. Both processes are induced by free radical formation and interaction which result in changes

within or between adjacent molecular chains. Typical manifestations of this energy interaction and deposition within the material can include outgassing, shrinkage, cracking, crazing, pitting, embrittlement and discoloration. These, in turn, can cause changes in strength, stiffness, thermal expansion, and thermophysical and optical characteristics.

Unlike conventional material property testing under ambient conditions or at temperature, space radiation effects testing is complex and expensive. Conducting real-time tests in space is not practical for obvious reasons. Performing simulation tests in earth-based laboratories with total fidelity to the natural space environment is neither possible, necessary nor economically feasible. Compromises in testing scenarios/techniques are necessary to obtain reliable material properties data in a reasonable time period for a reasonable cost. The question is to determine which compromises are acceptable.

To date there is little hard data to provide guidance in the selection of acceptable/required simulation techniques. The issues of accelerated testing effects, post-radiation test environments, ionization equivalence of different ionizing sources, synergistic effects of combined radiation, and serial versus simultaneous radiation, to name a few, have yet to be fully investigated. Unfortunately, the resolution of many of these issues may be a function of the material system and the material property in question and may not be possible without the development of better test facilities.

#### 2.2 OBJECTIVE

NASA/LaRC and WPAFB/AFML (Reference 1) have recently completed studies for the evaluation of radiation effects on composite materials. All of the post-irradiation testing, however, on these programs was done ex situ; that is, the material specimens were tested in air following irradiation because of the limitations of the testing equipment. One of the major uncertainties about the above approach was whether the post-irradiation exposure to air could mask or alter important radiation effects.

The objective of this program was to investigate the effect of the post-irradiation test environment on the tensile properties of representative advanced composite material systems. It was not the purpose of the tests to obtain engineering design data because only a limited number of specimens per material and test condition could be evaluated under this exploratory program. However, the data would be useful: (1) to draw initial conclusions about the importance of the test conditions, and (2) to provide some indications about the radiation hardness of the materials tested.

#### 2.3 TECHNICAL APPROACH

Test specimens were fabricated by NASA/LaRC and supplied to TRW for preconditioning prior to testing. The radiation test facility had a vacuum chamber in which a miniature tensile testing apparatus was incorporated. This permitted undersize specimens –  $2.54~\rm cm$  (1 in.) gage length by  $0.95~\rm cm$  (0.375 in.) wide by  $0.635~\rm mm$  (0.025 in.) thick – to be irradiated and evaluated under different post-irradiation test environments without removing them from the chamber.

Tensile stress-strain tests were performed on control specimens under ambient conditions using a regular testing machine and in the chamber in air and in vacuum to obtain reference-level material properties. Test specimens were then exposed in vacuum to a radiation environment up to 1 x  $10^{10}$  rads consisting of a mono-energetic fluence of 700 keV electrons from a Van de Graaff accelerator. The bulk dose was nearly uniform throughout the material. The dose levels were selected to represent that expected from long-term exposure to critical trapped radiation belt environments.

In present conventional testing techniques, (1) interruptions in irradiation are permitted, (2) time delays between cessation of irradiation and post-irradiation testing are allowed, and (3) post-irradiation testing is performed in air or an inert environment. These compromises are accepted because of the complexity, expense and constraints associated with alternative test methods which attempt to replicate the natural environment more realistically. Under this study, test programs were designed to compare more exact techniques with the more conventional techniques.

Post-irradiation testing was performed while the specimens were being irradiated (termed in situ testing), in vacuum immediately after cessation of irradition (termed in vacuo testing), and after varying periods of air exposure (termed ex situ testing). Room temperature and elevated temperature effects were evaluated. Effects of intermittent versus continuous irradiation were also investigated to evaluate the presence of vacuum annealing of the radiation-induced changes to the tensile properties.

#### 3.0 TEST SPECIMENS

#### 3.1 MATERIAL DESCRIPTION

Three composite material systems were evaluated during the course of the study. Table 3-1 lists the materials and associated identification symbols. During the Phase I Test Series, a laminate of Union Carbide Thornel 300 PAN fiber impregnated with Narmco 5208 epoxy resin (T300/5208) was tested. In Phase II, two materials were tested: a laminate of Union Carbide Thornel 300 PAN fiber impregnated with Fiberite 934 epoxy resin (T300/934); and Celanese Celion 6000 fiber impregnated with Union Carbide Udel P1700 polysulfone thermoplastic resin (C6000/P1700).

The chemistry of two of the composites was determined to assist in the calculation of absorbed dose-depth profiles. Analysis of the polysulfone laminate was provided by NASA/LaRC. The T300/934 analysis was performed at TRW. It was assumed that the T300/5208 chemistry was similar to the T300/934 material for absorbed dose-depth profile calculations. Table 3-2 summarizes the chemical analysis.

#### 3.2 CONFIGURATION

Figure 3-1 shows a typical test specimen. An undersize specimen was selected to maximize the number of specimens that could be accommodated at one time within the target plane radiation zone, and to be compatible with the stiffness of the tensile testing apparatus/support structure within the vacuum chamber. Figure 3-2 presents the critical dimensions. The specimen consisted of the composite gage length section with oversized fiberglass reinforcement tapered tabs bonded on each end. Overall specimen length was 50.8 mm (2.0 in.). Gage length was 25.4 mm (1.0 in.). Specimen width was 9.53 mm (0.375 in.). Nominal thickness for the specimens was 0.635 mm (0.025 in.). A 2.54 mm (0.10 in.) diameter hole was located in each end to accommodate small pins in the target plane fixturing for positive grip control.

All of the composites tested consisted of a 4-ply balanced layup [+45, -45, -45, +45 degrees]. The balanced layup provided a flat laminate, resulting in more precise tensile strain measurements. The 45-degree orientation allowed modulus measurements to be obtained at relatively low

Table 3-1. Test Materials

No.	Symbol Symbol	Fiber/Resin	Source .
1	T300/5208	Union Carbide Thornel 300 PAN Fiber/Narmco 5208 Epoxy Resin	NASA/LaRC
2	T300/934	Union Carbide Thornel 300 PAN Fiber/Fiberite 934 Epoxy Resin	NASA/LaRC
3	C6000/P1700	Celanese Celium 6000 Fiber/ Union Carbide Udel P1700 Polysulphone Resin	NASA/LaRC

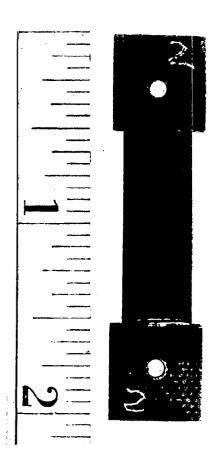
Table 3-2. Test Material Chemistry

Element	Percent by Weight			
Erement	T300/5208*	T300/934	C6000/P1700	
Carbon	85.7	85.7	81.6	
Hydrogen	1.7	1.7	2.1	
Nitrogen	5.0	5.0	3.6	
Sulphur	0.1	0.1	2.8	
Phosphorus	-	-	1.3	
Oxygen	7.5	7.5	8.6	
Density (gm/cm <sup>3</sup> )	1.58	1.58	1.51	

\*Assumed similar to T300/934

loadings, reducing the effect of support and specimen holder structural deflections on measured strain values. In addition, the 45-degree layup would be more sensitive than a unidirectional layup to changes in resin or resin-fiber interface properties resulting from irradiation.

Each specimen was numbered with a scribe on the fiberglass tabs on each end of the specimen. The numbers identified the location of each specimen on the target plane. In addition, the outer end load fittings were also stamped with similar numbers.





(a) Front View

(b) Side View

Figure 3-1. Typical Test Specimen (2x).

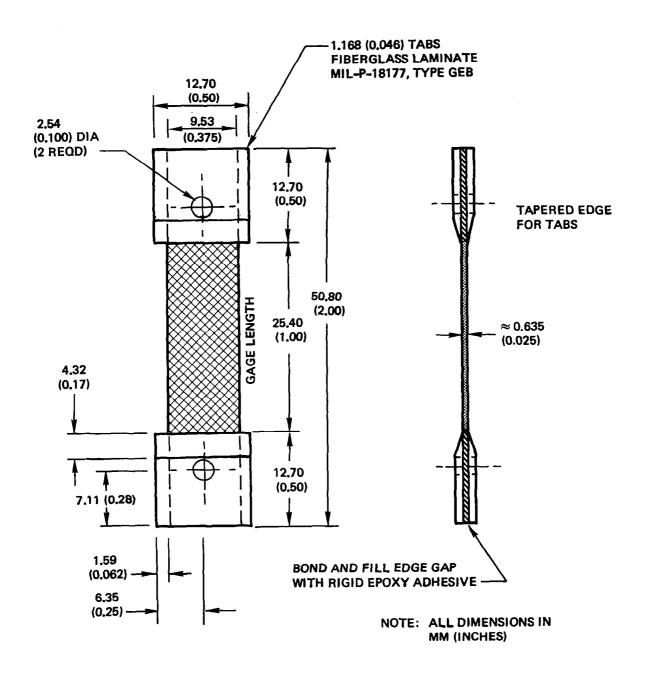


Figure 3-2. Test Specimen Dimensions.

#### 3.3 ENVIRONMENTAL PRECONDITIONING

All specimens were preconditioned prior to radiation or baseline tensile testing by exposure to a temperature of 65 °C  $\pm 5$  °C (150 °F  $\pm 9$  °F) for 72 hours in a vacuum. Heating was accomplished in a Brew vacuum furnace which utilizes a diffusion pumping system. Vacuum was maintained below  $5 \times 10^{-5}$  torr.

Upon removal from the preconditioning environment, the specimens were immediately mounted on the target plane and installed in the vacuum chamber. Preconditioned specimens were under vacuum or tested to failure (in the case of control specimens) within 3 hours of exposure to ambient conditions. Ambient conditions did not exceed 60 percent relative humidity or a temperature of  $20^{\circ}\text{C}$   $\pm 5^{\circ}\text{C}$  ( $68^{\circ}\text{F}$   $\pm 9^{\circ}\text{F}$ ).

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#### 4.0 TEST ENVIRONMENT DEFINITION

No attempt was made to accurately represent natural space radiation conditions. Radiation was limited to a mono-energetic dose of high-energy electrons at an energy level and fluence sufficient to provide a bulk dose in the material representative of that to be experienced under long-term geosynchronous orbit conditions. No other type of radiation (i.e., ultraviolet, protons) were provided. Irradiation was performed in vacuum at room and elevated temperatures.

#### 4.1 HIGH ENERGY ELECTRON RADIATION

The specific radiation environment required that the mono-energetic electrons be of sufficient energy level to keep the dose through the material uniform within a factor of two. For Phase I tests, specimens were irradiated to an average bulk dose of 1 x  $10^9$  rads and 2 x  $10^9$  rads. This was increased to 1 x  $10^{10}$  rads for Phase II tests.

A preliminary estimate of the required electron energy was made by comparing the range from Berger-Seltzer tables (Reference 2) to obtain the energy required to get a range 2.5 to 3 times the average specimen thickness of 0.635 mm (25 mils) and density of 1.55 gm/cm<sup>3</sup>. From this comparison, 700 keV was determined to be an appropriate energy. Initial calculations of 500 keV, 600 keV, 700 keV and 1 MeV were then made using TIGER (Reference 3), a one-dimensional Monte Carlo electron transport computer code. For these calculations, 2000 history calculations were made, since the statistics (-10 percent batch) were adequate for scoping purposes. Figures 4-1 and 4-2 present the results of these initial calculations for two of the three materials. From these calculations, 700 keV was determined to adequately meet the criteria previously stated for the energy.

To better define the dose profiles and energy deposited, 8000 history calculations (-5 percent batch statistics) were then made for the two materials. The results (dose profiles and average dose) are presented in Figures 4-3 and 4-4. The average dose is the total energy deposited in the material divided by the material mass per square centimeter (or thickness). Table 4-1 presents these average doses and their associated fluence levels for a dose of 1 x  $10^9$  rads. For 2 x  $10^9$  rads and 1 x  $10^{10}$  rads, the levels are 2X and 10X those shown, respectively. As a comparison, the

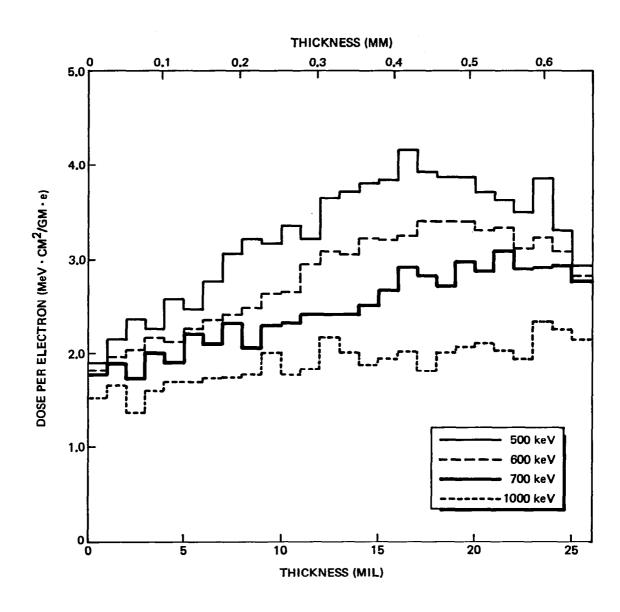


Figure 4-1. Unit Dose as a Function of Thickness for a Unit Electron Fluence Normally Incident on Material C6000/P1700.

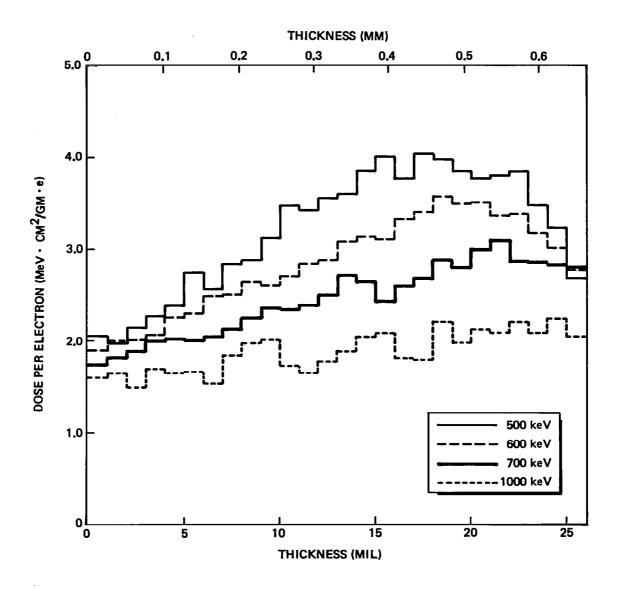


Figure 4-2. Unit Dose as a Function of Thickness for a Unit Electron Fluence Normally Incident on Material T300/934.

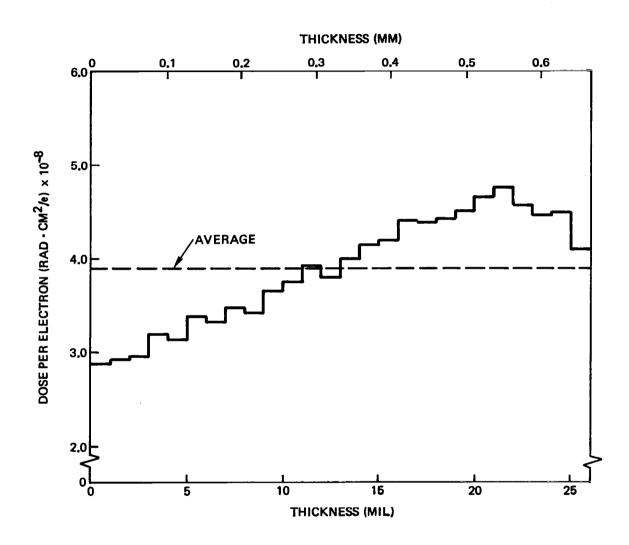


Figure 4-3. Unit Dose as a Function of Thickness for a 700 keV Electron Fluence Normally Incident on Material C6000/P1700.

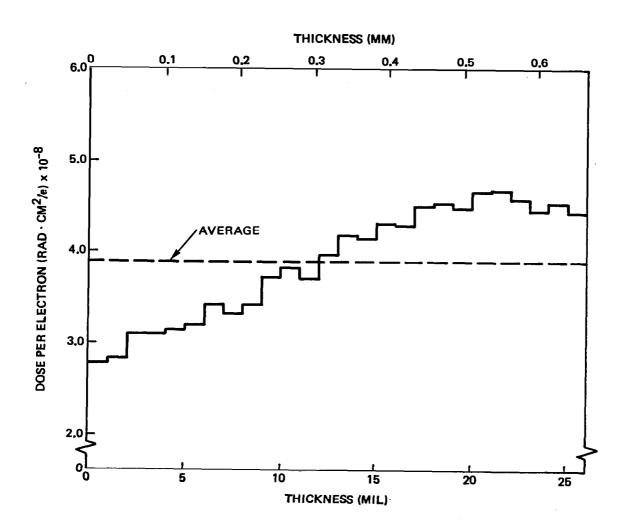


Figure 4-4. Unit Dose as a Function of Thickness for a 700 keV Electron Fluence Normally Incident on Material T300/934.

Table 4-1. Energy and Fluence for 1 x 10<sup>9</sup> rads Average
Dose from a Mono-Energenic Electron Beam
Normally Incident on Specimens

Energy (keV)	Material	Average Unit Dose (rad·cm²/e)	Fluence (e/cm <sup>2</sup> )
(b)	C6000/P1700	3.89 x 10 <sup>-8</sup>	2.57 x 10 <sup>16</sup>
700	T300/934 <sup>(a)</sup>	3.89 x 10 <sup>-8</sup>	2.57 x 10 <sup>16</sup>
(c)	C6000/P1700	5.19 x 10 <sup>-8</sup>	1.93 x 10 <sup>16</sup>
500	T300/934 <sup>(a)</sup>	5.15 x 10 <sup>-8</sup>	1.94 x 10 <sup>16</sup>
(c)	C6000/P1700	4.47 x 10 <sup>-8</sup>	2.24 x 10 <sup>16</sup>
600	T300/934 <sup>(a)</sup>	4.52 x 10 <sup>-8</sup>	2.24 x 10 <sup>16</sup>
(c)	C6000/P1700	3.02 x 10 <sup>-8</sup>	3.31 x 10 <sup>16</sup>
1000	T300/934 <sup>(a)</sup>	3.00 x 10 <sup>-8</sup>	3.34 x 10 <sup>16</sup>

- (a) Assume results applicable to T300/5208 material
- (b) 8000 History Monte Carlo analysis
- (c) 2000 History Monte Carlo analysis

table includes the 500 keV, 600 keV and 1 MeV results from the 2000 history calculations. The absorbed dose-depth profiles for the materials are almost identical because of the similar chemistry and density properties.

The results listed in Table 4-1 were based on one-dimensional analyses and assumed no influence from the surrounding metal structure in the immediate vicinity of the test specimens. For portions of the Phase I testing, only the rear and sides of the stainless steel vacuum chamber were available for backscattering of electrons. For other Phase I tests and all Phase II tests, a resistance wire heating coil and reflector plate were placed in close proximity behind the specimens.

Appendix A presents additional calculations used to quantify any back-scattering dose effects from surrounding hardware in the vacuum/irradiation chamber. The results indicate: (1) the backscattering from the chamber walls is negligible; thus, the fluence levels listed in Table 4-1 are acceptable; and (2) the effect of the heating unit behind the specimens

increased the average bulk dose by approximately 18 percent; thus, the 700 keV electron fluence required to produce a given average bulk dose listed in Table 4-1 must be reduced appropriately.

#### 4.2 VACUUM

Irradiation and all in situ and in vacuo post-irradiation tensile testing were performed in a pressure environment below 1 x  $10^{-6}$  torr. Vacuum was maintained by a 400  $\ell$ /sec ion pump. No losses in vacuum were incurred during any radiation period or tensile test.

#### 4.3 TEMPERATURE

Control specimens were maintained at a nominal temperature of  $20\,^{\circ}\text{C}$   $\pm 5\,^{\circ}\text{C}$  ( $68\,^{\circ}\text{F}$   $\pm 9\,^{\circ}\text{F}$ ). For those radiation sequences that required the specimens to be near room temperature, the temperature was held to  $25\,^{\circ}\text{C}$   $\pm 5\,^{\circ}\text{C}$  ( $77\,^{\circ}\text{F}$   $\pm 9\,^{\circ}\text{F}$ ). For those radiation sequences that required the specimens to be at an elevated temperature, the temperature was held to a nominal level of  $120\,^{\circ}\text{C}$   $\pm 5\,^{\circ}\text{C}$  ( $248\,^{\circ}\text{F}$   $\pm 9\,^{\circ}\text{F}$ ), using a combination of heating from the electron irradiation and the resistance wire heater located behind the specimens. During periods when the elevated temperature test specimens were receiving no radiation (i.e., overnight shutdowns), a low heater power was maintained to keep the specimen temperatures at approximately  $50\,^{\circ}\text{C}$   $\pm 5\,^{\circ}\text{C}$  ( $122\,^{\circ}\text{F}$   $\pm 9\,^{\circ}\text{F}$ ). All post-irradiation tensile testing was done at a temperature of  $30\,^{\circ}\text{C}$   $\pm 5\,^{\circ}\text{C}$  ( $86\,^{\circ}\text{F}$   $\pm 9\,^{\circ}\text{F}$ ).

#### 5.0 TEST FACILITIES

#### 5.1 LABORATORY OVERVIEW

Irradiation of the materials and subsequent tensile testing was performed in TRW's Space Environmental Simulation Laboratory (SESL). SESL is a state-of-the-art facility for exposure of materials and small space-craft components to a simulated natural space radiation environment. Key features include: (1) combined simulation of critical components of the space environment; (2) simultaneous irradiation of many test specimens; (3) in situ and ex situ evaluation of radiation-induced effects; and (4) evaluation of mechanical, optical, thermophysical, and electrical properties.

Figure 5-1 shows a layout of the overall laboratory. The facility consists of three vacuum/irradiation chambers integrated to a 1 MeV Van de Graaff (VDG) electron accelerator. Additional radiation sources (i.e., ultraviolet, low-energy electrons and protons) are individually connected to the appropriate chambers. Table 5-1 summarizes the capabilities and functions of each chamber. While the chambers have the inherent capabilities shown in Table 5-1, they can be readily adapted to handle other types of specimens and measurements. Each chamber is stainless steel and approximately 46 cm (18 in.) in diameter by 76 cm (30 in.) long. Two of the three chambers have all-metal seals and a full door on each end (except Chamber No. 2), along with various ports extending from the sides and ends to accommodate radiation sources, view windows, vacuum pump, instrumentation leads, and measurement devices.

Table 5-2 indicates how the important natural space radiation environmental components can be simulated and lists the corresponding source characteristics. Radiation from each source covers a 15 cm (6 in.) diameter target plane at an off-normal angle of incidence ranging from 0 to 22 degrees (depending on the specific vacuum chamber, radiation sources and type of test). Flux variation over the target plane is typically  $\pm 10$  to  $\pm 15$  percent for the VDG and electron flood gun sources and  $\pm 20$  to  $\pm 25$  percent for the proton source and ultraviolet sources.

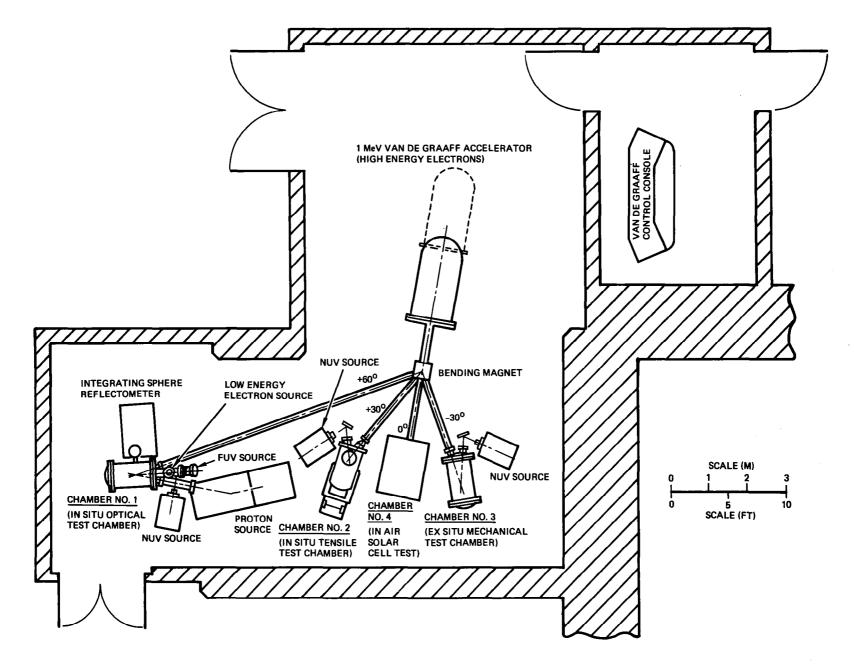


Figure 5-1. Space Environmental Simulation Laboratory (SESL) Layout.

Table 5-1. Vacuum/Irradiation Chamber Functions

Chamber	Number of Samples	Combined Environmental Components	Baseline Measurements*
		Near UV	In Situ Spectural Reflectance
No. 1	28	Far UV High Energy Electrons	In Situ Diffuse Reflectance In Situ Bidirectional Reflectance
(In Situ	(1 x 2 cm)	Low Energy Electrons	
Optical)		Low Energy Protons High Vacuum	
No. 2	18	Near UY High Energy Electrons	In Situ Tensile Stress - Strain Properties
(In Situ Tensile)	(2.54 cm Gage Length)	J 35	
No. 3	40	Near UV High Energy Electrons	Ex Situ Mechanical Properties
(Ex Situ Mech)	(2.54 cm Gage Length)	High Vacuum	

<sup>\*</sup>All chambers can be adapted for ex situ mechanical, electrical or chemical property tests

Table 5-2. Environmental Simulation Description

Radiation Component	Simulation Radiation Source	Simulation Source Characteristics
Near Ultraviolet	3-KW Short Arc Xenon Lamp	0.18 to 0.40 mm Up to 5X Sun Intensity
Far Ultraviolet	Electrodeless Krypton Gas Lamp	0.10 to 0.18 mm Up to 5X Sun Intensity
Radiation Belt Electrons	Van de Graaff Accelerator	70 keV to 1.1 MeV 10 <sup>7</sup> to 10 <sup>11</sup> e/cm <sup>2</sup> sec
Plasma Sheet Electrons	Electron Flood Gun	0.5 to 10 keV Up to 10 <sup>11</sup> e/cm <sup>2</sup> sec
Radiation Belt Protons	Ionization Equivalent Electrons from Van de Graaff Accelerator	70 keV to 1.1 MeV $10^7$ to $10^{11}$ e/cm $^2$ sec
Solar Flare Protons	Ionization Equivalent Electrons from Van de Graaff Accelerator	70 keV to 1.1 MeV 107 to 1011 e/cm <sup>2</sup> sec
Plasma Sheet Protons	Hydrogen Ion Plasma Generator	Up to 30 keV Up to 10 <sup>11</sup> p/cm <sup>2</sup> sec
Vacuum	GN Aspiration, Cryosorption, and 400 P/sec Ion Pumping	10 <sup>-6</sup> to 10 <sup>-8</sup> Torr

Specimen temperature control is achieved (1) by conductive and radiative cooling techniques within each chamber using  $LN_2$  or chilled water in shrouds and coldplates, (2) by filtering the infrared energy from the near ultraviolet (NUV) source beams, and/or (3) by heater plates.

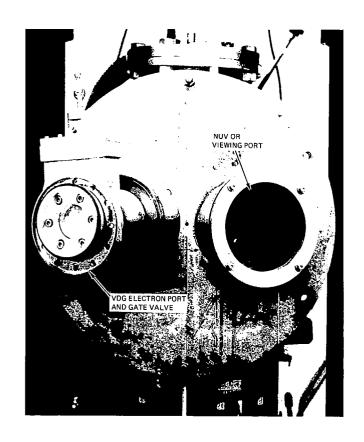
### 5.2 TEST CHAMBER DESCRIPTION

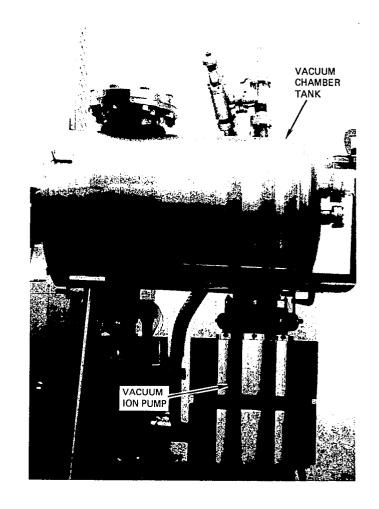
Chamber No. 2 (In Situ Tensile Test Chamber - ISTTC) was used to perform the radiation and tensile testing. The ISTTC is a 46 cm (18 in.) diameter by 76 cm (30 in.) long stainless steel cylindrical chamber with multiple ports (Figure 5-2). All of the ports are metal sealed, except for two ports which house the tensile testing apparatus, the front door, and the manipulator shaft that rotates the carousel target plane. These are sealed with Viton gaskets.

Two ports on the front door are approximately 23 degrees off normal to the specimen carousel target plane. One of these ports is a high purity silica window which allows for viewing or accommodation of the near ultraviolet radiation source. The other port accommodates the high-energy electron beam and incorporates a gate valve for isolation of the chamber from the VDG system. High vacuum is achieved in the chamber through carbon vane pump aspiration, cryosorption and ion pumping with a 400 £/sec pump.

For this test program, since NUV radiation was not being used, the target plane was oriented normal to the incident electron beam. This resulted in a quasi-symmetric electron flux distribution on the target plane, with all specimens receiving about the same fluence.

Figures 5-3 and 5-4 show schematic representations of the components of the chamber that permit in situ, in vacuo or ex situ tensile testing of the specimens. A 2224 N (500 lb) miniature load cell is mounted in the interior of the chamber on top of a rigid support platform which is rigidly attached to the cover of the bottom port. The bottom port incorporates a stainless steel bellows and an aligned feedthrough rod. The internal end of the rod has a clevis-type fitting which is used to grip a compatible fitting on the end of each tensile specimen. The external end of the rod is attached to a ball screw drive through a mechanical coupler structure. The ball screw drive is activated by a variable speed motor. The external mechanism structure acts like the movable head of a tensile testing machine, with a maximum head travel rate of 0.09 cm/min (0.036 in/min).





(a) Front view

(b) Side view

Figure 5-2. Vacuum/Irradiation Chamber No. 2.

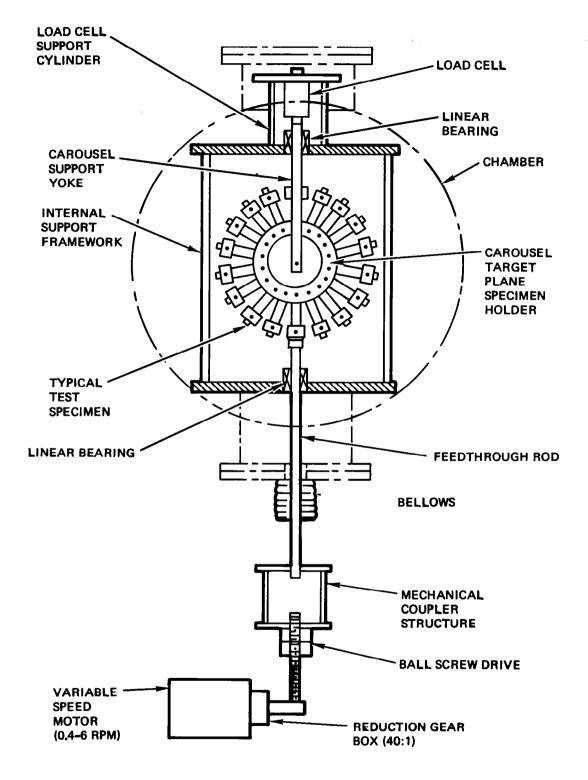


Figure 5-3. Schematic Representation of Tensile Testing Apparatus for the In Situ Tensile Test Chamber.

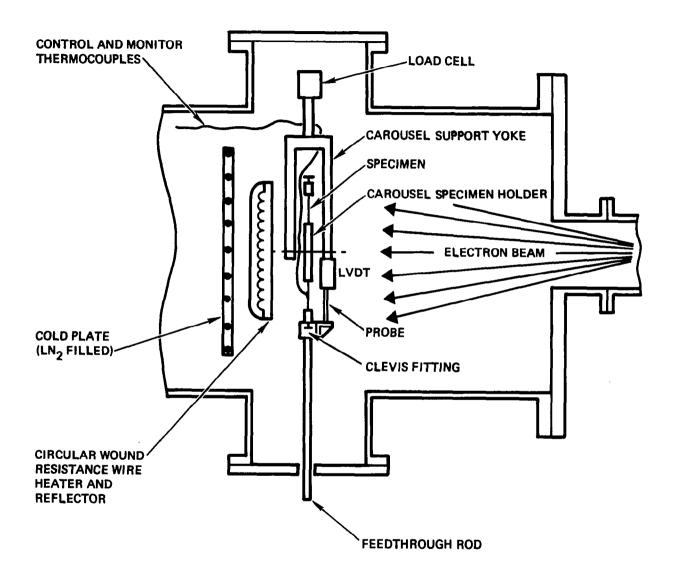


Figure 5-4. Schematic Representation of LVDT Specimen Deflection/ Strain Monitoring Transducer, Heater and Coldplate Units.

The 15 cm (6 in.) diameter target plane (shown in Figure 5-5) is in the form of a carousel which is attached through a yoke directly to the load cell. The carousel is designed to accommodate 18 specimens, 25.4 mm (1 in.) gage length by up to 12.7 mm (0.5 in.) wide, placed in a radial manner. The inward end of the test specimen is firmly clamped in the carousel. The outer end is clamped in a fitting that mates with the clevis fitting on the upper end of the bellows feedthrough rod. The carousel is rotated by means of an external manipulator to index each specimen to the bellows feedthrough rod. It should be noted that the specimens are not directly backed up by any metal plate (like a heat sink plate). Instead they are free standing. Behind the carousel target plane are located heater and coldplate units to maintain proper control over specimen temperature. The heater unit was only installed for elevated temperature irradiation tests.

Axial motion of the feedthrough rod is monitored with a LVDT mounted on the carousel target plane yoke support and indexed to a rigid angle standoff attached to the clevis fitting (shown in Figure 5-6). The LVDT was shielded from direct electron beam exposure by a small lead plate. The LVDT output and load cell output are charted on an X-Y plotter to provide an instantaneous load-deflection (or load-strain) diagram.

Calibration of the chamber load-deflection apparatus with a relatively stiff steel specimen indicated negligible distortion or play in the "load chain" over the load range of interest [0.025 mm (0.001 in.) deflection at 444.8 N (100 lb) load for a 25.4 mm (1 in.) gage length specimen].

### 5.3 HIGH ENERGY ELECTRON SOURCE

The high energy electrons for the irradiations are produced by a High Voltage Engineering Corporation Model JS 1000 Van de Graaff accelerator (VDG). This generator operates by feeding charge onto a moving belt inside the pressurized VDG tank. This charge is removed from the belt at a terminal in the center of the tank, which then assumes the high voltage required to accelerate electrons from an electron source located in the terminal through an evacuated, many-electrode accelerator tube. The electron beam emerges from the accelerator tube at ground potential with an energy corresponding to the potential on the terminal. The electron beam

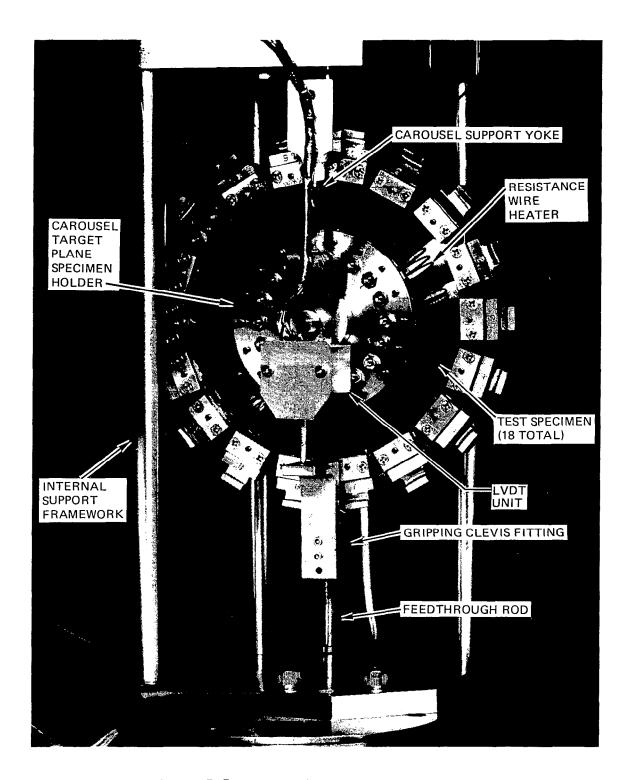


Figure 5-5. View of Internal Components.

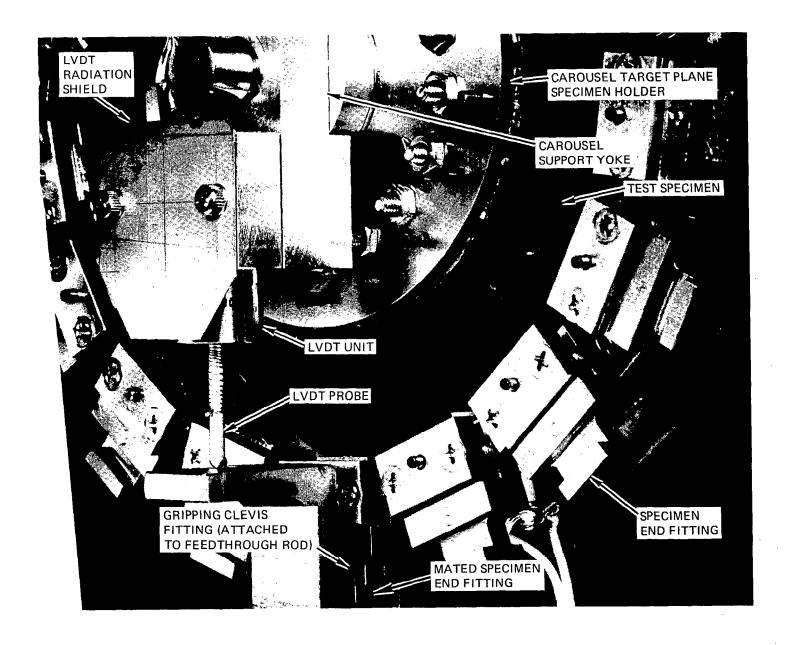


Figure 5-6. Detail of LVDT Installation.

is a dc beam and is controlled in energy and current by electronic stabilizers. The system is capable of operating continuously with beam currents from 20 nA to 50  $\mu$ A at electron beam energies from 0.070 to 1.1 MeV.

Figure 5-7 shows the beam line between the VDG and the vacuum/ irradiation chamber used for the subject program. The electron beam produced is initially focused by a solenoidal magnet and then passed through vacuum piping to a switching/bending magnet which can direct the beam into any one of four beam lines. After passing through the switching magnet, the beam passes through a second solenoidal magnet which re-focuses the beam upon scattering foils located in the beam scattering chamber. The beam plumbing is enclosed by Conetic (high permeability nickel steel) shielding to keep the earth's magnetic field and other stray fields from altering the beam path. With the use of the solenoidal magnets and shielding, over 85 percent of the electron beam passes through a 1.27 cm (0.5 in.) diameter aperture at the end of the beam leg into the beam scattering chamber.

This aperture is positioned at the entrance to a scattering chamber (Figure 5-8) where the beam passes through a series of thin aluminum foil diaphragms before passage into the vacuum/irradiation chamber. The last foil acts to isolate the test chamber vacuum from the VDG accelerator vacuum and to disperse the electron beam over the carousel target plane. At the 700 keV energy level, the full dispersion can not be achieved by one foil; hence, additional foils are interposed in the electron beam by means of a remotely controlled rotator. The beam, in passing through the scattering foils, loses about 4 percent of its energy. This is compensated for by an appropriate increase in the beam energy level leaving the VDG.

The scattered electron beam is monitored on the downstream annulus/ exit aperture which also defines the extent of the beam that can freely expand into the test chamber. The current impinging on this annulus, biased against externally arriving secondary electrons, is fed to a current integrator to determine accumulated target plane fluence.

Calibration of the VDG system is presented in Appendix B. The appendix details the calibration of the annular current against the on-target electron flux (using a Faraday cup), target plane flux mapping (using thermoluminescent dosimeters), and beam energy calculations.

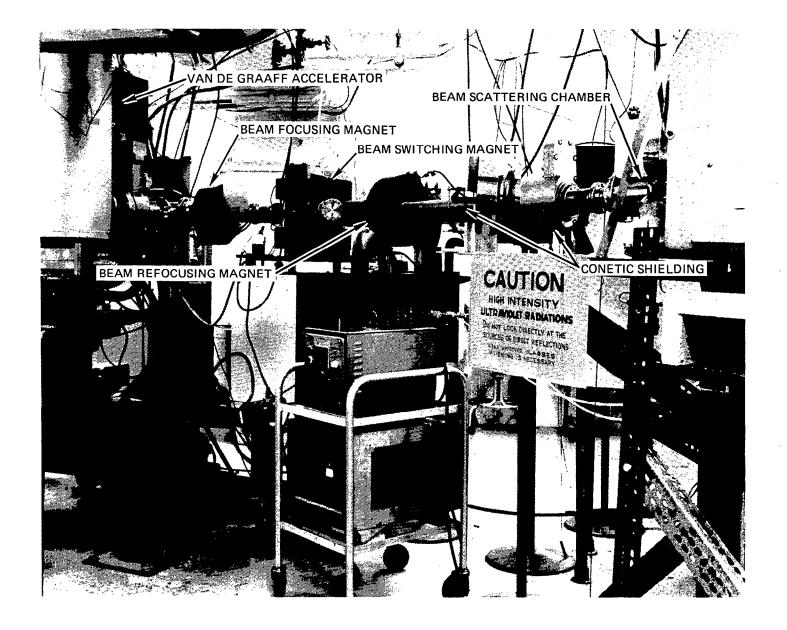


Figure 5-7. High-Energy Electron Beam Distribution System.

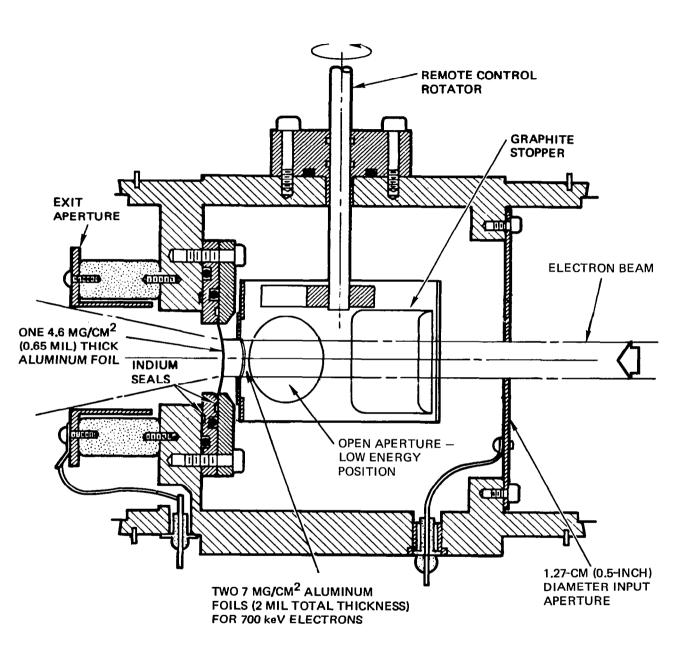


Figure 5-8. VDG Electron Beam Scattering Chamber.

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### 6.0 TEST RESULTS

### 6.1 TEST OVERVIEW

Tensile testing consisted of baseline measurements on unexposed specimens and measurement of property changes on irradiated specimens. Baseline data were obtained ex situ (in the ambient environment) using an Instron testing machine, and in vacuo and in air in the vacuum/irradiation chamber using the chamber load-deflection testing apparatus.

Testing of irradiated specimens was done exclusively in the vacuum/irradiation chamber using the chamber load-deflection apparatus. Tests were performed on irradiated specimens in situ (while the specimens were being irradiated), in vacuo (in vacuum after cessation of irradiation), and ex situ (in air after cessation of irradiation).

Testing was broken into two phases, with a number of individual test sequences for each phase. Phase I tests evaluated the effects of radiation and post-irradiation test environments on one composite material system (T300/5208). Radiation levels were nominally 1 to 2 x  $10^9$  rads at room and elevated temperature (120°C). Post-irradiation test environments included in vacuo, in situ, and ex situ.

Based on the negligible changes in material properties obtained during the Phase I tests, the Phase II tests were designed to obtain radiation damage thresholds on the test materials. Two composite material systems (T300/934, C6000/P1700) were subjected up to 1 x  $10^{10}$  rads at elevated temperature (120°C). Post-irradiation test environments included in vacuo and ex situ.

### 6.2 PHASE I TEST RESULTS

### 6.2.1 Test Plan

Table 6-1 presents the Phase I Test Series. The series consisted of five test sequences on T300/5208 [ $\pm 45/\mp 45$ ] laminate graphite/epoxy material. Control specimens were tested during Sequences 1 and 2 to obtain baseline material properties. Tests on irradiated specimens were conducted during Sequences 3, 4 and 5 to evaluate the differences among in situ, in vacuo, and ex situ effects on tensile properties.

Test Sequence	Radiation Parameters	Radiation Time (Hrs)	Specimen Temperature °C (°F)	Test Machine	Tensile Test Atmosphere	Material	Number of Specimens Tested	Remarks
1	No radiation	0	20 (68)	Instron	In air	T300/5208 [ <u>+</u> 45/+45]	9	Control specimen tests
2a 2b	No radiation	0	20 (68)	 Chamber	In vacuo In air	T300/ <u>5</u> 208 [+45/+45]	9	Control specimen tests comparing the effects of air and vacuum on tensile properties and comparing chamber results with Instron results.
3a 3b 3c	1 x 10 <sup>9</sup> rads; 700 keV; 2.1 x 10 <sup>11</sup> e/cm <sup>2</sup> sec; 10 <sup>-6</sup> torr	34 hrs total at ≈8 hrs per day for 4.5 days	20 (68)	Chamber	In vacuo In air In air	T300/ <u>5</u> 208 [ <u>+</u> 45/+45]	6 6 6	18 specimens on target plane. 6 specimens tested to failure in vacuum immediately after termination of radiation. 6 specimens tested to failure in air 64 to 72 hours after chamber is backfilled with dry air.
4a 4b 4c 4d 4e 4f 4g 4h 4i	1 x 10 <sup>9</sup> rads; 700 keV; 2.1 x 10 <sup>11</sup> e/cm <sup>2</sup> sec; 10 <sup>-6</sup> torr	34 hrs continuous 34 34.5 34.5 35 35 35 35.5 36 36	20 (68)	Chamber	In situ In vacuo In situ	T300/ <u>52</u> 08 [ <u>+</u> 45/ <del>+</del> 45]	1 2 1 2 1 2 1 2 2	18 specimens on target plane. 1 specimen tested to failure just prior to termination of the initial 34 hours of irradiation. 2 specimens tested to failure in vacuum immediately after ter- mination of the initial 34 hours of irradiation. Remaining specimens are irradiated for another 30 minutes. 1 specimen tested to failure just prior to termination of the 30-minute radiation. 2 specimens tested to failure in vacuum immediately after termination of the 30-minute radiation. This sequence of 30-minute radiations with 1 specimen tested prior to and 2 specimens tested immedi- ately after additional irradia-
4k 41		36.5 36.5			In situ In vacuo		1 2	tion is repeated.
5a 5b 5c	2 x 10 <sup>9</sup> rads; 700 keV; 2.1 x 10 <sup>11</sup> e/cm <sup>2</sup> sec; 10 <sup>-6</sup> torr	68 hrs total at ≈8 hrs per day for 9 days	120 (248) (Irradiated) 20 (68) (Tensile Tested)	Châmber 	In vacuo In air In air	T300/ <u>5</u> 208 [ <u>+</u> 45/+45]	6 6	See remarks for test sequence 3.

The objectives of Sequences 1 and 2 were as follows: (1) to obtain baseline data, (2) to determine any differences in test data caused by differences between the load-deflection apparatus of the Instron and vacuum chamber equipment, and (3) to evaluate whether dry air or vacuum has any effect on the baseline results. For Sequence 1, nine unirradiated control specimens were tensile tested to failure under ambient conditions using a standard Instron testing machine. For Sequence 2, eighteen unirradiated control specimens were tensile tested to failure in the vacuum/irradiation chamber. Nine specimens were tested while under vacuum, and the remaining nine specimens were tested with dry air in the chamber.

The objective of Sequence 3 was to provide data on in vacuo versus ex situ effects. Eighteen specimens were subjected in vacuum to a nominal dose of 1 x  $10^9$  rads. Radiation occurred over a 4.5 day period ( $\approx$ 8 hours/day), with specimens at a nominal temperature of  $20^{\circ}$ C ( $68^{\circ}$ F). Six specimens were tensile tested to failure immediately (within 30 minutes) after completion of the full term dose (in vacuo tests). The chamber was then backfilled with dry air and another six specimens were immediately (within 45 minutes) tested to failure (ex situ tests). The remaining six specimens were exposed to dry air for approximately three days in the chamber before being tested to failure (ex situ tests).

Sequence 4 tests explored the differences between in situ and in vacuo effects. Eighteen specimens were subjected in vacuum to a nominal dose of 1 x  $10^9$  rads. Full term dose was applied continuously (24 hours/day) over a 34-hour period with specimens at a nominal temperature of  $20\,^{\circ}\text{C}$  (68°F). One specimen was tensile tested to failure while being irradiated just prior to completion of full term dose (in situ test). Immediately after completing irradiation, two specimens were tensile tested to failure in vacuum (in vacuo tests). The remaining fifteen specimens were irradiated for an additional 30 minutes. One specimen was tested to failure while being irradiated just prior to completion of the 30-minute added radiation, and two specimens were tested to failure immediately after completion of the added radiation. This sequence of 30-minute added radiation periods with tensile testing before and immediately after cessation of irradiation

was repeated with the remaining specimens until all were tested.

This resulted in six specimens being tested in situ and twelve specimens being tested in vacuo.

Sequence 5 was a repeat of Sequence 3, except that full term dose was nominally 2 x  $10^9$  rads with specimen temperatures at 120°C (248°F). Radiation occurred over a nine-day period ( $\approx 8$  hours/day). Tensile testing was performed at near room temperature.

The only major deviations from the plan described above were as follows: (1) temperature of the specimens during irradiation for Sequences 3 and 4 was  $7^{\circ}$ C ( $45^{\circ}$ F); (2) temperature of the specimens during irradiation for Sequence 5 was 110 to  $118^{\circ}$ C (230 to  $245^{\circ}$ F); (3) the temperature of the specimens during tensile testing for Sequences 3, 4, and 5 was above and below the  $20^{\circ}$ C ( $68^{\circ}$ F) temperature desired; (4) heater malfunction during Sequence 5 resulted in specimen temperatures of  $46^{\circ}$ C ( $115^{\circ}$ F) during an 8-hour period at the beginning of irradiation and during a 12-hour period at the end of irradiation; and (5) due to radiation backscatter from the heater unit installed for Sequence 5 tests, the total dose was  $2.4 \times 10^9$  rads.

## 6.2.2 Test Procedures

Specimens received from NASA/LaRC were subjected to a general inspection to ensure that they were straight, free of rough edges, and had properly installed/aligned end-tabs. Some specimens were rejected over the course of the Phase I test series because of flaws. Only those specimens that passed the inspection were preconditioned. After preconditioning, another inspection was made to ensure none of the specimens had warped.

Specimens were preconditioned prior to radiation or testing by exposure to a temperature of  $65\,^\circ$ F ( $150\,^\circ$ F) for 72 hours in a vacuum of  $10^{-5}$  torr. Upon completion of the preconditioning sequence, specimens were placed in a vacuum desiccator and evacuated with a mechanical pump for transportation to the laboratory or for storing prior to installation on the carousel target plane or testing in the Instron machine. In most instances the specimens, once preconditioned, were immediately tested to failure (Instron tests) or installed on the carousel target plane and placed in the vacuum/irradiation test chamber. Thus, preconditioned specimens were tested to

failure or under vacuum in the test chamber within 3 hours of exposure to ambient conditions.

The specimens to be tested in the vacuum/irradiation chamber, once full vacuum was reached, were permitted to sit 48 to 72 hours before the start of irradiation or before the start of in vacuo baseline tests.

Except for one test sequence (Sequence No. 4), irradiation was applied approximately 8 hours/day in a nearly continuous manner with only minor interruptions during that 8-hour period. There was an approximate 16-hour stand-down period between daily irradiations. Irradiations were performed on consecutive days with no weekend or holiday interruptions. For Sequence 4, irradiation was applied in a nearly continuous manner 24 hours/day, with only minor interruptions during the total period (~34 hours), to obtain full term dose.

At the conclusion of full term dose or at intermediate dose levels, specimens were tensile tested to failure using the in situ loading apparatus described in Section 5.2 and in the sequences described in Section 6.2.1.

After completion of tensile testing on each batch of specimens (18 specimens for radiation test sequences), the specimens or remnants thereof were removed from the vacuum chamber and placed in a plastic container for storage. Microscopic examination of each specimen was performed and photomicrographs were taken to illustrate typical damage and failure modes.

Following completion of post-test examinations at TRW, specimens from each test sequence were returned to NASA/Langley for customer examination. Results of any customer-conducted examinations/tests were not available for inclusion in this report.

# 6.2.3 Tensile Properties

The Phase I Test Series results are summarized in Tables 6-2 through 6-6. These tables present the calculated values of modulus of elasticity (E), ultimate tensile strength ( $F_{TU}$ ), yield strength ( $F_{TY}$ ) and ultimate elongation ( $\varepsilon_{u}$ ) for the T300/5208 [ $\pm$ 45/ $\mp$ 45] laminate composite material for Test Sequences 1 through 5. Refer to Appendix C for a definition of these

Table 6-2. Phase I Test Series, Sequence 1 Test Results, T300/5208 [ $\pm 45/\mp 45$ ], Baseline, Instron Machine, No Radiation

S/N	Irradiation Temperature °C (°F)		Tempe	le Test rature (°F)	Tensile Test Environment	Ten	E sile ulus (Msi)	F <sub>TU</sub> Ultim Tensi Stren MPa	ate le	Deflection Measurement Method
1						10.3	(1.5)	144.8	(21.0)	
2		ľ				10.3	(1.5)	153.1	(22.2)	
3	N,	/A	20	(68)	Ambient	11.0	(1.6)	144.8	(21.0)	Head Travel
4						11.0	(1.6)	151.7	(22.0)	of Machine*
5						9.7	(1.4)	142.7	(20.7)	
6						17.2	(2.5)	160.6	(23.3)	
7						19.6	(2.7)	148.9	(21.6)	
8						13.8	(2.0)	151.0	(21.9)	
9		}				13.1	(1.9)	130.3	(18.9)	Extensiometer
10	N,	/A	20	(68)	Ambient	18.6	(2.7)	160.0	(23.2)	(2.54 cm G.L.)
11						15.1	(2.2)	151.0	(21.9)	
12						15.1	(2.2)	140.0	(20.3)	
13						14.5	(2.1)	142.7	(20.7)	
14						13.1	(1.9)	139.3	(20.2)	

<sup>\*</sup>Specimen gripped in machine with special grips and pin through specimen end tabs

E = 10.5 GPa (1.52 Msi) (Head Travel)

E = 15.5 GPa (2.24 Msi) (Extensiometer)

 $\overline{F}_{TU}$  = 147.4 MPa (21.38 ksi) (Head Travel)

 $\overline{F_{TU}}$  = 147.1 MPa (21.33 ksi) (Extensiometer)

Table 6-3. Phase I Test Series, Sequence 2 Test Results,  $T300/5208 [\pm 45/\mp 45]$ , Baseline Properties, In the Vacuum Chamber, No Radiation

<b>.</b>	Irradiation Temperature	Tensile Test Temperature °C (°F)	ture Test	Modulus		FTU Ultimate Tensile Strength MPa (ksi)		F <sub>T</sub> y Yield Tensile Strength* MPa (ksi)		ε <sub>u</sub> Ultimate Elongation (Percent)	
S/N	°Ć (°F)	°C (°	F) Environment	GPa	(MS1)	мРа 	(KS1)	MPa	. (KS1)	(Percent)	
1	1	1	In vacuo	10.9	(1.58)	138.4	(20.069)	87.8	(12,738)	1.83	
2				11.2	(1.62)	139.8	(20.275)	92.0	(13.344)	1.93	
3				11.4	(1.66)	146.5	(21.251)	89.0	(12.902)	2.18	
4			•	11.8	(1.71)	150.9	(21.892)	100.4	(14.558)	2.01	
5				11.1	(1.61)	140.7	(20.408)	86.6	(12.559)	2.20	
6				11.2	(1.63)	147.2	(21.347)	91.7	(13.304)	2.11	
7			In vacuo	12.8	(1.85)	145.9	(21.167)	95.2	(13.803)	2.00	
8			In air	11.3	(1.64)	154.5	(22.409)	92.6	(13.424)	2.12	
9	N/A	20 (58)		11.3	(1.64)	152.0	(22.045)	94.5	(13.714)	2.17	
10				11.4	(1.66)	143.1	(20.762)	95.7	(13.876)	1.85	
11				13.7	(1.98)	147.8	(21.435)	107.2	(15.547)	1.79	
12		İ		12.6	(1.83)	149.8	(21.726)	97.2	(14.101	1.90	
13				12.8	(1.86)	152.6	(22.141)	91.7	(13.306)	2.22	
14				12.6	(1.83)	149.1	(21.628)	105.0	(15.233)	1.85	
15				12.1	(1.75)	148.7	(21.568)	99.5	(14.425)	1.91	
16				12.6	(1.83)	154.1	(22.349)	92.1	(13.358)	2.11	
17				12.6	(1.82)	147.0	(21.315)	95.3	(13.825)	2:01	
18			In air	12.3	(1.79)	148.3	(21.504)	97.1	(14.089)	1.88	

<sup>\*</sup>Strength at 1 percent (0.010 in/in) elongation

= 11.5 GPa (1.67 Msi) (In vacuo)

= 12.3 GPa (1.78 Msi) (In air)

 $F_{TII}$  = 144.2 MPa (20.916 ksi) (In vacuo)

 $F_{TII}$  = 149.7 MPa (21.716 ksi) (In air)

 $\overline{F}_{TY}$  = 91.8 MPa (13.315 ksi) (In vacuo)

 $\overline{F}_{TY}$  = 98.4 MPa (14.082 ksi) (In air)  $\overline{\epsilon}_{u}$  = 2.04 percent (In vacuo)

= 1.98 percent (In air)

Table 6-4. Phase I Test Series, Sequence 3 Test Results, T300/5208 [ $\pm$ 45/ $\mp$ 45], 1 x 109 Rads Dose ( $\simeq$ 8 Hrs/Day, 4-1/2 Days Irradiation)

S/N	Irradiation Temperature °C (°F)	Tensile Test Temperature °C (°F)	Tensile Test Environment		sile ulus (Msi)	Tens	mate	F <sub>TY</sub> Yiel Tens Stren MPa	d sile <sub>/a</sub> )	ε <sub>u</sub> Ultimate Elongation (Percent)
1		18.5 (65)	In vacuo	9.6	(1.39)	158.4	(23.041)	80.5	(11.673)	2.81
2	Ì	18.9 (66)		12.5	(1.81)	164.0	(23.785)	91.3	(13.237)	3.04
3		17.5 (63)	(b)	11.2	(1.62)	152.7	(22.150)	84.7	(12.283)	2.59
4		13.5 (56)		11.8	(1.71)	154.4	(22.399)	94.0	(13.639)	2.22
5		12.2 (54)		12.8	(1.85)	151.9	(22.033)	98.1	(14.223)	2.40
6		11.1 (52)	In vacuo	10.4	(1.51)	152.5	(22.116)	87.4	(12.681)	2.48
7		17.5 (63)	In air	12.1	(1.75)	154.7	(22.432)	91.0	(13.195)	2.46
8		15.0 (59)	}	12.1	(1.75)	152.5	(22.116)	93.0	(13.493)	2.36
9	7 (45)	15.5 (60)	(c)	12.2	(1.77)	157.8	(22.883)	94.1	(13.648)	2.53
10		15.5 (60)		12.6	(1.82)	156.2	(22,662)	96.8	(14.039)	2.47
11		15.5 (60)		12.9	(1.87)	156.7	(22.732)	99.3	(14.408)	2.24
12		15.5 (60)	In air	13.5	(1.96)	166.1	(24:092)	94.9	(13.759)	2.74
13			In air	12.9	(1.87)	155.7	(22.584)	101.0	(14.648)	2.63
14	ĺ			12.6	(1.82)	160.8	(23.324)	94.6	(13.720)	2.89
15		18.5 (65)	(d)	13.3	(1.93)	150.0	(21.753)	100.0	(14.502)	2.48
16				12.6	(1.82)	168.5	(24.440)	99.0	(14.358)	3.54
17				12.4	(1.80)	169.2	(24.538)	95.6	(13.860)	3.70
18 <sup>(e</sup>	)		In air	10.3	(1.50)	90.0	(13.051)	90.0	(13.052)	1.04

<sup>(</sup>a) Strength at 1 percent (0.010 in/in) elongation
(b) Within 3 to 30 min. after termination of radiation
(c) Within 3 to 46 min. after exposure to air
(d) 72 hrs. after exposure to air
(e) Thermocouple covered back of specimen; results not included in averages

Ē	= 11.4 GPa (1.65 Msi) (In vacuo)	$\overline{F}_{TY}$ = 89.3 MPa (12.956 ksi) (In vacuo)
Ē	= 12.6 GPa (1.82 Msi) (In air, ≃25 min)	$\overline{F}_{TY}$ = 94.8 MPa (13.757 ksi) (In air, $\approx$ 25 min)
Ē	= 12.8 GPa (1.85 Msi) (In air, 72 hrs)	$\overline{F}_{TY}$ = 98.0 MPa (14.218 ksi) (In air, 72 hrs)
$F_{TU}$	= 155.7 MPa (22.587 ksi) (In vacuo)	$\overline{\varepsilon}_{u}$ = 2.59 percent (In vacuo)
FTU	= 157.3 MPa (22.820 ksi) (In air, ≃25 min)	$\overline{\epsilon}_{\rm u}^{-}$ = 2.47 percent (In air, 25 min)
$\overline{F}_TU$	= 160.8 MPa (23.328 ksi) (In air, 72 hrs)	$\overline{\epsilon}_{u}$ = 3.05 percent (In air, 72 hrs)

Table 6-5. Phase I Test Series, Sequence 4 Test Results, T300/5208 [ $\pm$ 45/ $\mp$ 45], 1 x 10<sup>9</sup> Rads Dose (24 Hrs/Day for 1-1/2 Days Continuous)

S/N	Irradiation Temperature °C (°F)	Tensile Test Temperature °C (°F)	Tensile Test Environment	E Tens Modi GPa	sile	Tens	mate	F <sub>TY</sub> Yiel Tens Stren MPa	d ile <sub>(a)</sub>	€ <sub>U</sub> Ultimate Elongation (Percent)
1,,	,	15.5 (60)	In situ	11.2	(1.62)	149.6	(21.701)	82.9	(12.028)	2.98
2 <sup>(b</sup>	'	1.7 (35)	In vacuo	10.8	(1.56)	105.4	(15.292)	90.0	(13.064)	1.41
3		-1.7 (29)	In vacuo	11.0	(1.59)	155.6	(22.574)	86.4	(12.530)	2.83
4		12.2 (54)	In situ	10.9	(1.58)	163.7	(23.752)	82.6	(11.980)	3.61
5	i	12.8 (55)	In vacuo	11.5	(1.67)	151.3	(21.948)	82.1	(11.906)	2.83
6		12.8 (55)	In vacuo	11.1	(1.61)	153.7	(22.294)	82.9	(12.020)	2.89
7		18.5 (65)	In situ	10.5	(1.53)	142.8	(20.717)	78.1	(11.330)	2.83
8		22.8 (73)	In vacuo	11.0	(1.60)	154.9	(22.470)	85.1	(12.344)	3.08
9	7 (45)	23.9 (75)	In vacuo	11.4	(1.66)	152.8	(22.160)	88.1	(12.780)	2.97
10		30.6 (87)	In situ	11.0	(1.59)	142.8	(20.715)	80.2	(11.633)	2.81
11		34.4 (94)	In vacuo	12.8	(1.85)	159.4	(23.118)	84.9	(12.310)	2.98
12		35.0 (95)	In vacuo	11.6	(1.68)	150.2	(21.790)	86.9	(12.610)	2.76
13		35.6 (96)	In situ	11.1	(1.61)	157.6	(22.857)	85.8	(12.444)	3.54
14		34.4 (94)	In vacuo	13.0	(1.88)	168.2	(24.392)	90.4	(13.106)	3.27
15		29.4 (85)	In vacuo	11.2	(1.62)	148.0	(21.464)	80.1	(11.622)	2.99
16		35.0 (95)	In situ	10.9	(1.58)	145.0	(21.027)	80.2	(11.626)	3.08
17		35.6 (96)	In vacuo	11.0	(1.59)	149.6	(21.704)	84.3	(12.227)	2.90
18	ł	40.0 (104)	In vacuo	11.2	(1.62)	153.0	(22.190)	84.6	(12.272)	3.27

<sup>(</sup>a) Strength at 1 percent (0.010 in/in) elongation

 $\begin{array}{lll} E & = 11.0 \; \text{GPa} \; (1.59 \; \text{Msi}) \; (\text{In situ}) & & F_{\text{TY}} & = 82.7 \; \text{MPa} \; (11.840 \; \text{ksi}) \; (\text{In situ}) \\ E & = 11.5 \; \text{GPa} \; (1.67 \; \text{Msi}) \; (\text{In vacuo}) & & F_{\text{TY}} & = 85.1 \; \text{MPa} \; (12.339 \; \text{ksi}) \; (\text{In vacuo}) \\ F_{\text{TU}} & = 150.3 \; \text{MPa} \; (21.795 \; \text{ksi}) \; (\text{In situ}) & & \overline{\epsilon}_{\text{U}} & = 3.14 \; \text{percent} \; (\text{In situ}) \\ \hline F_{\text{TU}} & = 161.1 \; \text{MPa} \; (23.373 \; \text{ksi}) \; (\text{In vacuo}) & & \overline{\epsilon}_{\text{U}} & = 2.98 \; \text{percent} \; (\text{In vacuo}) \\ \end{array}$ 

<sup>(</sup>b) Specimen failed in grips; results not included in averages

Table 6-6. Phase I Test Series, Sequence 5 Test Results, T300/5208 [ $\pm$ 45/ $\mp$ 45], 2.4 x 10<sup>9</sup> Rads Dose ( $\approx$ 8 Hrs/Day, 9 Days Irradiation)

S/N	Irradiation Temperature °C (°F)	Tensile Test Temperature °C (°F)	Tensile Test Environment	E Tens Modi GPa	ile	F <sub>T</sub> Ulti Tens Stre MPa	mate	F <sub>TY</sub> Yiel Tens Stren MPa	d ile(a) gth (ksi)	ε <sub>u</sub> Ultimate Elongation (Percent)
12	1	29.4 (85)	In vacuo	11.7	(1.69)	154.4	(22.394)	99.6	(14.441)	3.00
13		[	i ideas	11.0	(1.59)	154.7	(22.436)	89.2	(12.938)	3.00
14	ļ	ļ <sub>i</sub>	(b)	14.7	(2.13)	174.9	(25.365)	99.3	(14.402)	3.45
15	ĺ		1.27	12.1	(1.76)	147.4	(21.385)	93.8	(13.604)	2.56
16				11.7	(1.70)	154.6	(22.426)	89.6	(12.994)	3.20
17		29.4 (85)	 In vacuo	11.3	(1.64)	153.4	(22.256)	98.5	(14.286)	2.50
6		21.1 (70)	In air	14.3	(2.07)	162.2	(23.523)	111.1	(16.116)	2.55
7	110-118 <sup>(e)</sup>	T .		13.2	(1.92)	150.3	(21.800)	94.4	(13.693)	2.65
8	(230-245)		(c)	13.2	(1.91)	147.7	(21.421)	94.1	(13.649)	2.80
9				13.1	(1.90)	152.9	(22.177)	86.8	(12.595)	3.25
10				15.0	(2.17)	153.6	(22.277)	97.1	(14.082)	3.03
11			In air	12.3	(1.78)	151.4	(21.959)	94.2	(13.664)	2.95
1			In air	12.9	(1.87)	159.0	(23.069)	89.7	(13.005)	3.50
2	}		1	13.0	(1.89)	146.8	(21.300)	90.6	(13.137)	3.25
3			(d)	12.2	(1.77)	150.7	(21.853)	88.5	(12.837)	3.00
4	1		[,	12.7	(1.84)	151.3	(21.941)	91.7	(13.301)	3.25
5				14.4	(2.09)	151.8	(22,017)	95.4	(13.832)	2.75
18		21.1 (70)	In air	15.1	(2.19)	168.7	(24.464)	101.7	(14.745)	3.50

<sup>(</sup>e) First 8 hours and last 12 hours were at 46°C (115°F); otherwise, the remaining 48 hours were at 110-118°C (230-245°F)

Ē	= 12.1 GPa (1.75 Msi) (In vacuo)	$\overline{F}_{TY}$ = 95.0 MPa (13.778 ksi) (In vacuo)
E	= 13.5 GPa (1.96 Msi) (In air, ≃25 min)	F <sub>TY</sub> = 96.3 MPa (13.967 ksi) (In air, ≃25 min)
Ē	= 13.4 GPa (1.94 Msi) (In air, 72 hrs)	$\overline{F}_{TY}$ = 92.9 MPa (13.476 ksi) (1n air, 72 hrs)
FTU	= 156.6 MPa (22.710 ksi) (In vacuo)	$\overline{\varepsilon}_{u}$ = 2.97 percent (In vacuo)
FTU	= 153.0 MPa (22.193 ksi) (In air,≈25 min)	$\overline{\epsilon}_{u}$ = 2.87 percent (In air, 25 min)
FTU	= 154.7 MPa (22.441 ksi) (In air, 72 hrs)	ε <sub>u</sub> = 3.21 percent (In air, 72 hrs)

<sup>(</sup>a) Strength at 1 percent (0.010 in/in) elongation (b) Within 3 to 30 min after termination of radiation

<sup>(</sup>c) Within 3 to 45 min after exposure to air

<sup>(</sup>d) 72 hrs after exposure to air

properties and how they were calculated from the raw data (load-deflection curves). At the bottom of the tables are listed average values for each test condition and material property.

Almost all specimens were well behaved and failed in the center of the gage length region (see Figure 6-1). No problems were experienced with the tensile testing mechanism incorporated into the vacuum/irradiation chamber. The load-deflection curves were all of a similar nature - having a fairly linear portion over the first part of the curve, then departing from linearity and rising to a maximum load value, then experiencing a noticeable load dropoff due to initiation of failure, followed by total loss of load due to complete failure (separation) of the specimen. In most instances initiation of failure was at a strain level very close to that where maximum loading occurred.

Review of the baseline data (Tables 6-2 and 6-3) indicates that for modulus values, the data is comparable for the Instron and the test vacuum chamber when using similar deflection monitoring techniques (i.e., head travel). The Instron modulus data do indicate that the use of an extensiometer does provide a truer measurement of modulus of elasticity because the localized deflections on the grips and specimen end tabs are not recorded. Ultimate strength values are not affected by the test equipment used, and the effects on ultimate elongation would be expected to be small. Furthermore, the load-deflection data does not appear to be affected by the presence of air or vacuum in the chamber.

As this was an exploratory program, greater emphasis was placed on the results obtained from testing in the vacuum chamber because: (1) the presence of trends was considered more important than absolute values; and (2) the "boundary conditions" assocated with the tensile testing were identical for Test Sequences 2 through 5.

Table 6-7 presents a statistical analysis summary of the Phase I Test Series data obtained from vacuum chamber testing. The data is arranged by groups of material properties and includes the average values  $(\overline{X})$  from the previous tables, plus standard deviation (S) and coefficient of variation (S/ $\overline{X}$ ). The scatter within the individual groups of data in Tables 6-3 through 6-6 is small, with a coefficient of variation being less than 10 percent in almost all instances.



Figure 6-1. Typical Failure Mode (T300/5208, Specimen No. 4,  $2 \times 10^9$  Rads, Tested in Air).

Table 6-7. Phase I Test Series Data Summary, T300/5208 [±45/∓45] Laminate Composite Material

			TEST CONDI	TIONS							
Test Sequence	Radiation Dose <sub>g</sub> Level (x10 rads)	Radiation Dose Application	Irradiation Temperature °C (°F)	Tensile Test Temperature "C ("F)	Tensile Test Environment	Elapsed Time After Irradiation or Exposure to Air	Material Property	No. of Specimens N	Average Yalue T	Standard Deviation S	Coefficient of Variation S/X
2	0	W/A	N/A	20 (68)	In vacuo	N/A		7	11.5 (1.666)	0.63 (0.091)	0.055
2	0	N/A	N/A	20 (68)	In air	N/A		11	12.3 (1.785)	0.72 (0.105)	0.059
3	ì	8 hr/day, 4 1/2 days	7 (45)	15 (59)	In vacuo	15 min		6	11.4 (1.648)	1.22 (0.177)	0.108
3	1	8 hr/day, 4 1/2 days	7 (45)	15.6 (60)	In air	25 min	Ε	6	12.6 (1.820)	0.57 (0.083)	0.046
3	1	8 hr/day, 4 1/2 days	7 (45)	18.3 (65)	In air	72 hrs	Modulus of	5	12.7 (1.848)	0.37 (0.053)	0.028
4	1	24 hr/day, 1 1/2 days	7 (45)	24.4 (76)	In situ	During irradiation	Elasticity	6	10.9 (1.585)	0.21 (0.031)	0.020
4	1	24 hr/day, 1 1/2 days	7 (45)	25.6 (78)	ln vacuo	5 min	GPa	11	11.5 (1.670)	0.70 (0.101)	0.061
5	2.4	8 hr/day, 9 days	116 (240)	29.4 (85)	ln vacuo	15 min	(Ms1)	6	12.1 (1.750)	1.34 (0.194)	0.111
5	2.4	8 hr/day, 9 days	116 (240)	29.4 (85)	In air	25 min		6	13.5 (1.960)	0.96 (0.139)	0.171
5	2.4	8 hr/day, 9 days	116 (240)	21.1 (70)	In air	72 hrs		6	13.4 (1.940)	1.12 (0.162)	0.083
2	0	N/A	N/A	20 (68)	In vacuo	N/A		7	144.2 (20.916)	4.6 (0.671)	0.032
2	0	N/A	W/A	20 (68)	In air	N/A		11	149.7 (21.717)	3.4 (0.489)	0.023
3	1	8 hr/day, 4 1/2 days	7 (45)	15 (59)	In vacuo	15 min	F <sub>TU</sub>	6	155.7 (22.587)	4.8 (0.692)	0.031
3	1	8 hr/day, 4 1/2 days	7 (45)	15.6 (60)	In air	25 min	Ultimate	6	157.3 (22.820)	4.7 (0.678)	0.030
3	1	8 hr/day, 4 1/2 days	7 (45)	18.3 (65)	In air	72 hrs	Tensile	5	160.8 (23.328)	8.3 (1.197)	0.051
4	1	24 hr/day, 1 1/2 days	7 (45)	24.4 (76)	in situ	During irradiation	Strength	6	150.3 (21.795)	8.7 (1.256)	0.058
4	1	24 hr/day, 1 1/2 days	7 (45)	25.6 (78)	In vacuo	5 min	MPa	11	154.2 (22.373)	5.6 (0.810)	0.036
5	2.4	8 hr/day, 9 days	116 (240)	29.4 (85)	In vacuo	15 min	(ks1)	6	156.6 (22.710)	9.4 (1.361)	0.060
5	2.4	8 hr/day, 9 days	116 (240)	29.4 (85)	în air	25 min		6	153.0 (22.193)	5.0 (0.719)	0.032
5	2.4	8 hr/day, 9 days	116 (240)	21.1 (70)	In air	72 hrs		6	154.7 (22.441)	7.9 (1.146)	0.051
2	0	N/A	N/A	20 (68)	In vacuo	N/A		7	91.8 (13.315)	4.8 (0.690)	0.051
2	0	N/A	N/A	20 (68)	In air	N/A		11	97.1 (14.062)	5.1 (0.734)	0.052
3	1	8 hr/day, 4 1/2 days	7 (45)	15 (59)	In Ascho	15 min	FTY	6	89.3 (12.956)	6.4 (0.930)	0.072
3	1	8 hr/day, 4 1/2 days	7 (45)	15.6 (60)	In air	25 min	Yield	6	94.8 (13.757)	2.9 (0.424)	0.031
3	1	8 hr/day, 4 1/2 days	7 (45)	18.3 (65)	In air	72 hrs	Tensile	5	97.9 (14.218)	2.8 (0.407)	0.029
4	1	24 hr/day, 1 1/2 days	7 (45)	24.4 (76)	In situ	During irradiation	Strength	6	81.6 (11.840)	2.7 (0.392)	0.033
4	1	24 hr/day, 1 1/2 days	7 (45)	25.6 (78)	In vacuo	5 min	MPa	11	85.1 (12.339)	2.9 (0.414)	0.034
5	2.4	8 hr/day, 9 days	116 (240)	29.4 (85)	ln vacuo	15 min	(ksi)	6	95.0 (13.778)	4.8 (0.698)	0.051
5	2.4	8 hr/day, 9 days	116 (240)	29.4 (85)	In air	25 min		6	96.3 (13.967)	8.0 (1.164)	0.083
5	2.4	8 hr/day, 9 days	116 (240)	21.1 (70)	in air	72 hrs		6	92.8 (13.467)	4.9 (0.709)	0.053
2	0	N/A	N/A	20 (68)	In vacuo	H/A		7	2.04	0.135	0.066
2	0	N/A	N/A	20 (68)	In air	N/A		11	1.98	0.149	0.075
3	1	8 hr/day, 4 1/2 days	7 (45)	15 (59)	in vacuo	15 min		6	2.59	0.295	0.114
3	1	8 hr/day, 4 1/2 days	7 (45)	15.6 (60)	In air	25 min	E <sub>U</sub>	6	2.47	0.168	0.068
3	1	8 hr/day, 4 1/2 days	7 (45)	18.3 (65)	In air	72 hrs	Ultimate	5	3.05	0.545	0.179
4	1	24 hr/day, 1 1/2 days	7 (45)	24.4 (76)	In situ	During irradiation	Elongation	6	3.14	0.351	0.112
4	1	24 hr/day, 1 1/2 days	7 (45)	25.6 (78)	ln vacuo	5 min	(Percent)	11	2.98	0.169	0.057
5	2.4	8 hr/day, 9 days	116 (240)	29.4 (85)	în vacuo	15 min		6	2.97	0.349	0.118
5	2.4	8 hr/day, 9 days	116 (240)	29.4 (85)	In air	25 min		6	2.87	0.258	0.089
5	2.4	8 hr/day, 9 days	116 (240)	21.1 (70)	ln air	72 hrs	i	6	3.21	0.292	0.091

There doesn't appear to be trends in the data to indicate any appreciable change in tensile properties caused by radiation (or radiation at elevated temperature), nor any significant differences as the result of in situ, in vacuo or ex situ testing, or as the result of continuous versus intermittent irradiation procedures. Through null hypothesis statistical analysis of the average values and standard deviations, comparing various post-irradiation test environments and radiation methods, the results indicate that the differences in average values among the groups of data is statistically insignificant except in a few isolated cases.

### 6.2.4 Post-Test Examination

All specimens were microscopically examined at 20x for possible differences in failure mechanisms or other evidence of changes or degradation. No indication of any physical change was observed as the result of irradiation of 1 or 2 x  $10^9$  rads dose.

Typical specimens at each test condition were photographed at approximately 2x for record purposes. These same specimens were then examined at high magnification (typically 1000x) using a scanning electron microscope. Again, the failed surfaces appeared to be very similar and no differences in failure mechanism were discernable.

### 6.3 PHASE II TEST RESULTS

### 6.3.1 Test Plan

Results from the Phase I Test Series on T300/5208 graphite epoxy laminate material indicated that, for dose levels up to 2 x  $10^9$  rads and temperatures up to  $120\,^{\circ}\text{C}$  ( $248\,^{\circ}\text{F}$ ), the resulting changes in tensile properties were small. The data also showed that the post-irradiation test environments had negligible effect on the results. One reason postulated for the latter result was that radiation damage thresholds were not achieved with the applied dose levels. Therefore, it was concluded that the dose levels would have to be substantially increased in an attempt to obtain radiation damage thresholds. If damage thresholds were achieved, then the effects of post-irradiation test environment could be more realistically evaluated.

Table 6-8 presents the Phase II Test Series. The series consisted of six test sequences. Two composite materials were tested: T300/934 [±45/∓45] laminate graphite/expoxy and C6000/P1700 [±45/∓45] laminate graphite/polysulfone. Control specimens were tested during Sequences 1, 2, 3 and 4 to obtain baseline material properties. Tests on irradiated specimens were conducted during Sequences 5 and 6 to evaluate the differences between in vacuo and ex situ effects on tensile properties.

The objectives of Sequences 1 and 2 (3 and 4) were as follows: (1) to obtain baseline data, (2) to determine any differences in test data caused by differences between the load-deflection apparatus of the Instron and vacuum chamber equipment, and (3) to evaluate whether dry air or vacuum has any effect on the baseline results. For Sequences 1 and 3, six unirradiated control specimens of each material were tensile tested to failure under ambient conditions using a standard Instron testing machine. For Sequences 2 and 4, twelve unirradiated control specimens of each material were tensile tested to failure in the vacuum/irradiation chamber. For each material, six specimens were tested under vacuum and six specimens were tested with dry air in the chamber.

The objectives of Sequence 5 were to determine the radiation damage threshold for T300/934 material and to evaluate in vacuo and ex situ effects on tensile properties. Eighteen specimens were subjected in vacuum to a nominal dose of 1 x  $10^{10}$  rads. Radiation occurred over a consecutive 33-day period ( $\approx$ 8 hours/day), with specimens at a nominal temperature of  $120\,^{\circ}\text{C}$  ( $248\,^{\circ}\text{F}$ ). At incremental dose levels of 3 x  $10^{9}$  rads and 6 x  $10^{9}$  rads, four specimens were tensile tested to failure immediately after cessation of irradiation (in vacuo tests). At the end of the full term dose of 1 x  $10^{10}$  rads, five specimens were immediately tensile tested to failure in vacuum (in vacuo test). The chamber was then back-filled with dry air and the remaining five specimens were tensile tested to failure after 72 hours exposure in air.

The objectives of Sequence 6 tests were identical to those of Sequence 5, except for C6000/P1700 material. The radiation levels, temperatures and post-irradiation test environments were the same as those used in Sequence 5.

Table 6-8. Phase II Test Series

Test Sequence	Radiation Parameters	Radiation Time (Hours)	Specimen Temperature °C (°F)	Test Machine	Tensile Test Atmosphere	Material	Number of Specimens Tested	Remarks
1	No radiation	0	20 (68)	Instron	In air	T300/ <u>9</u> 34 [ <u>+</u> 45/+45]	6	Control specimen tests.
2a 2b	No radiation	0	20 (68) 20 (68)	 Chamber	In vacuo In air	1 T300/ <u>9</u> 34 [ <u>+</u> 45/+45] 	6	Control specimen tests com- paring the effects of air and vacuum on tensile pro- perties and comparing chamber results with Instron results.
3		Identical	to Test Sequence	1 except u	sing C6000/P1	700 material.		
4a 4b		Identical to	Test Sequences 2a	, 2b excep	t using C6000	/P1700 materia	il.	
5a 5b 5c 5d	$3 \times 10^{9}$ , $6 \times 10^{9}$ and $1 \times 10^{10}$ rads; 700 keV electrons; $2.9 \times 10^{11}$ e/cm <sup>2</sup> sec; Up to $2.6 \times 10^{17}$ e/cm <sup>2</sup> ; $10^{-6}$ torr	250 hrs total at =8 hrs per day for 33 days	120 (248) (Irradiated) 20 (68) (Tensile tested)	Chamber	In vacuo In vacuo In vacuo In air	T300/ <u>9</u> 34 [ <u>+</u> 45/+45]	4 4 5 5	18 specimens on target plane. 4 specimens tested to failure in vacuum immediately after reaching 3 x 10 <sup>9</sup> rads dose. Radiation is continued on remaining specimens up to 6 x 10 <sup>9</sup> rads dose. 4 specimens tested to failure in vacuum immediately after reaching 6 x 10 <sup>9</sup> rads. Radiation is continued on remaining specimens up to 1 x 10 <sup>10</sup> rads dose. 5 specimens tested to failure in vacuum immediately after reaching 1 x 10 <sup>10</sup> rads. 5 remaining specimens tested to failure in air 72 hours after chamber is backfilled with dry air.
6a 6b 6c 6d		Identical to Tes	t Sequence 5a, 5b,	5c 5d exc	ept using C60	00/P1700 mater	rial.	

### 6.3.2 Test Procedures

Test procedures were identical to those used for the Phase I Test Series (see Section 6.2.2) with respect to: pretest inspection of specimens, specimen preconditioning and handling, daily radiation schedules (i.e.,  $\approx 8$  hours/day for consecutive days until incremental or full term dose was achieved), post-irradiation testing techniques, and post-test specimen examination.

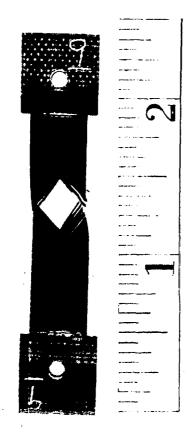
## 6.3.3 Tensile Properties

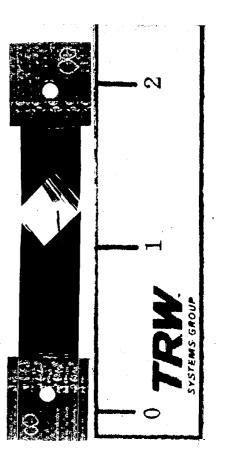
The Phase II Test Series results are summarized in Tables 6-9 though 6-14. These tables present the calculated values of modulus of elasticity (E), ultimate tensile strength ( $F_{TU}$ ), yield strength ( $F_{TY}$ ), and ultimate elongation ( $\varepsilon_{u}$ ) for the T300/934 and C6000/P1700 [ $\pm$ 45/ $\mp$ 45] laminate composite materials for Test Sequences 1 through 6. Refer to Appendix C for a definition of these properties and how they were calculated from the raw data (load-deflection curves). At the bottom of the tables are listed average values for each test condition and material property.

Almost all specimens were well behaved and failed in the center of the gage length region (see Figure 6-2). One irradiated T300/934 specimen and four irradiated C6000/P1700 specimens warped sufficiently as the result of radiation/elevated temperature to render them unusable. No problems were experienced with the tensile testing mechanism incorporated into the vacuum/irradiation chamber. The shape of the load-deflection curves were similar to those obtained from the Phase I Test Series.

Because of the type of end gripping used for the Instron tests it was impossible to compare baseline results between Instron and vacuum chamber testing for the T300/934 material. However, there was good correlation for the C6000/P1700 baseline tests, although consistently higher values for ultimate tensile strength were obtained in the chamber tests. For both materials, there was a greater difference between the in vacuo and in air chamber test results for modulus of elasticity (in vacuo gave lower results) than experienced in the Phase I baseline chamber tests.

Tables 6-15 and 6-16 present a statistical analysis summary of the Phase II Test Series data obtained from the vacuum chamber testing for T300/934 and C6000/P1700, respectively. The data is arranged by groups of





(a) T300/934, Specimen No. 9,  $1 \times 10^{10}$  Rads, Tested In Vacuo

(b) C6000/P1700, Specimen No. 8,  $1 \times 10^{10}$  Rads, Tested In Vacuo

Figure 6-2. Typical Failure Mode.

Table 6-9. Phase II Test Series, Sequence 1 Test Results T300/934 [ $\pm 45/\mp 45$ ], Baseline, Instron Machine, No Radiation

S/N	Irradiation Temperature °C (°F)	Tensile Test Temperature °C (°F)	Tensile Test Environment	Ten	E sile ulus (Msi)	F <sub>TU</sub> Ultim Tensi Stren MPa	ate 1e	Deflection Measurement Method
1				7.6	(1.1)	153.7	(22.3)	<u> </u>
2				8.3	(1.2)	162.7	(23.6)	
3	N/A	20 (68)	Ambient	8.3	(1.2)	155.1	(22.5)	Head Travel*
4	1			6.9	(1.0)	158.6	(23.0)	of Machine
5				7.6	(1.1)	138.6	(20.1)	
6				20.0	(2.9)	159.2	(23.1)	
7				20.0	(2.9)	151.0	(21.9)	
8	N/A	20 (68)	Ambient	17.9	(2.6)	158.6	(23.0)	Extensiometer
9	1		1	20.0	(2.9)	149.6	(21.7)	(2.54 cm G.L.)
10				20.7	(3.0)	153.7	(22.3)	

\*Specimen gripped in machine only by pins through end specimen tabs

E = 7.7 GPa (1.12 Msi) (Head Travel)

E = 19.7 GPa (2.86 Msi) (Extensiometer)

 $\overline{F}_{TU}$  = 153.7 MPa (22.3 ksi) (Head Travel)

 $\overline{F_{TU}}$  = 154.4 MPa (22.4 ksi) (Extensiometer)

Table 6-10. Phase II Test Series, Sequence 2 Test Results, T300/934 [ $\pm$ 45/ $\mp$ 45], Baseline, in the Vacuum Chamber, No Radiation

S/N	Irradiation Temperature °C (°F)	Tensile Test Temperature °C (°F)	Tensile Test Environment	Tens Modu GPa	sile	FT Ulti Tens Stre MPa	mate	FTY Yiel Tens Stren MPa	d ile(a) gth(a) (ksi)	ε <sub>u</sub> Ultimate Elongation (Percent)
1		1	In air	11.8	(1.71)	119.7	(17.358)	87.5	(12.696)	1.91
2 <sup>(b)</sup>	)									
3			,	14.2	(2.06)	151.8	(22.015)	105.2	(15.264)	2.01
4	ļ	1		11.0	(1.60)	121.2	(17.578)	84.4	(12.247)	2.02
5	}	}	1	10.1	(1.47)	121.6	(17.638)	82.2	(11.917)	2.02
6			In air	9.2	(1.34)	127.6	(18.512)	83.8	(12.154)	2.10
7	N/A	20 (68)	In vacuo	9.6	(1.39)	137.4	(19.933)	81.3	(11.797)	2.30
8 9(b	,			9.3	(1.35)	128.8	(18.681)	78.1	(11.322)	2.23
•		Ì								
10		1	ļ	9.4	(1.36)	129.1	(18.720)	75.5	(10.956)	2.00
11		ļ	ļ	9.4	(1.37)	132.5	(19.221)	83.6	(12.129)	1.96
12	[		In vacuo	9.7	(1.40)	131.8	(19.112)	85.8	(12.447)	2.01

(a) Strength at 1 percent (0.010 in/in) elongation

(b) Specimens were not preconditioned; test results not included

= 11.4 GPa (1.63 Msi) (In air)

= 9.4 GPa (1.37 Msi) (In vacuo)

= 128.4 MPa (18.620 ksi) (In air)

= 131.9 MPa (19.133 ksi) (In vacuo)

 $F_{TY} = 88.6 \text{ MPa } (12.856 \text{ ksi}) \text{ (In air)}$ 

 $\overline{F}_{TY}$  = 80.9 MPa (11.730 ksi) (In vacuo)  $\overline{\varepsilon}_{u}$  = 2.01 percent (In air)

= 2.10 percent (In vacuo)

Table 6-11. Phase II Test Series, Sequence 5 Test Results, T300/934 [ $\pm$ 45/ $\mp$ 45], 1 x 10<sup>10</sup> Rads Dose ( $\approx$ 8 Hours/Day, 34 Days)

S/N	Irradiation Temperature °C (°F)	Tensile Test Temperature °C (°F)	Tensile Test Environment	Dose (x10 <sup>10</sup> rads)		E nsile dulus (Msi)	Ult: Ten:	TU imate sile ength (ksi)	F <sub>T</sub> Yie Ten Stre MPa	Y ld sile ngth (ksi)	ε <sub>u</sub> Ultimate Elongation (Percent)
1		20.0 (68)	In air	1.0	11.0	(1.59)	156.0	(22.632)	92.1	(13.353)	3.57
2	1	20.0 (68)			11.7	(1.70)	164.2	(23.821)	88.0	(12.768)	4.15
3 <sup>(P)</sup>			(c)								
4		20.0 (68)			15.6	(2.26)	185.1	(26.849)	121.0	(17.546)	3.42
5	1	20.0 (68)	In air		11.2	(1.63)	157.7	(22.876)	91.0	(13.202)	3.23
6		23.9 (75)	In vacuo		10.3	(1.50)	140.0	(20.302)	82.0	(11.898)	2.97
7	1	26.1 (79)			13.0	(1.88)	151.0	(21.896)	89.6	(13.001)	2.36
8		29.4 (85)			13.9	(2.01)	157.5	(22.839)	103.7	(15.048)	2.39
9		29.4 (85)	İ		11.8	(1.71)	161.8	(23.473)	88.3	(12.813)	2.95
10	120 (248)	37.8 (100)		1.0	11.4	(1.65)	174.2	(25.270)	84.8	(12.295)	4.82
11		32.2 (90)	(d)	0.67	8.3	(1.21)	146.2	(21.210)	76.0	(11.029)	3.08
12		29.4 (85)	ļ	0.67	10.3	(1.49)	170.0	(24.652)	86.3	(12.523)	5.50
13		30.0 (86)		0.67	11.3	(1.64)	157.7	(22.875)	89.9	(13.043)	3.44
14		32.8 (91)		0.67	9.8	(1.42)	156.5	(22.698)	84.4	(12.245)	5.47
15		33.9 (93)		0.33	11.5	(1.67)	157.9	(22.896)	86.5	(12.553)	4.12
16		36.7 (98)		0.33	11.5	(1.67)	145.4	(21.086)	95.1	(13.794)	3.19
17		38.9 (102)		0.33	12.2	(1.77)	145.4	(19.470)	95.9	(13.907)	2.13
18		43.3 (110)	In vacuo	0.33	13.5	(1.96)	173.0	(25.099)	106.6	(15.455)	3.68

<sup>(</sup>a) Strength at 1 percent (0.010 in/in) elongation(b) Specimen warped, not tested

<sup>(</sup>d) Within 3 to 30 minutes after termination of radiation

Ē	= 12.2 GPa (1.77 Msi) (0.33 x 10 <sup>10</sup> rads, vac)
E	= 9.9 GPa (1.44 Msi) (0.67 x $10^{10}$ rads, vac)
E	= 12.0 GPa (1.75 Msi) (1 x 10 <sup>10</sup> rads, vac)
E	= 12.4 GPa (1.80 Msi) (1 x 10 <sup>10</sup> rads, air)
F <sub>Tli</sub>	= 152.6 MPa (22.138 ksi) (0.33 x 10 <sup>10</sup> rads, vac)
FTU	= 157.6 MPa (22.859 ksi) $(0.67 \times 10^{10} \text{ rads, vac})$
FTII	= 156.9 MPa (22.756 ksi) (1 x 10 <sup>10</sup> rads, vac)
F <sub>TU</sub> F <sub>TU</sub> F <sub>TU</sub> F <sub>TU</sub>	= 165.8 MPa (24.044 ksi) (1 x 10 <sup>10</sup> rads, air)

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F_{TY} = 96.0 MPa (13.927 ksi) (0.33 x 10<sup>10</sup> rads, vac)

F_{TY} = 84.2 MPa (12.210 ksi) (0.67 x 10<sup>10</sup> rads, vac)

F_{TY} = 89.7 MPa (13.011 ksi) (1 x 10<sup>10</sup> rads, vac)
  F_{TY} = 98.0 \text{ MPa} (14.217 \text{ ksi}) (1 \times 10^{10} \text{ rads, vac})
\overline{\epsilon}_{\rm u} = 3.28 percent (0.33 x 10<sup>10</sup> rads, vac)

\overline{\epsilon}_{\rm u} = 4.37 percent (0.67 x 10<sup>10</sup> rads, vac)

\overline{\epsilon}_{\rm u} = 3.10 percent (1 x 10<sup>10</sup> rads, vac)

\overline{\epsilon}_{\rm u} = 3.59 percent (1 x 10<sup>10</sup> rads, air)
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<sup>(</sup>c) 72 hours after exposure to dry air

Table 6-12. Phase II Test Series, Sequence 3 Test Results, C6000/P1700 [ $\pm$ 45/ $\mp$ 45], Baseline, Instron Machine, No Radiation

S/N	Irradiation Temperature °C (°F)	Tensile Test Temperature °C (°F)	Tensile Test Environment	Ten	E sile ulus (Msi)	F <sub>T</sub> Ulti Tens Stre MPa	mate ile	Deflection Measurement Method	
1				11.7	(1.7)	56.5	(8.2)		
2				8.3	(1.2)	47.6	(6.9)		
3	N/A	20 (68)	Ambient	8.3	(1.2)	57.2	(8.3)	Head Travel*	
4		1		9.7	(1.4)	61.4	(8.9)	of Machine	
5				11.0	(1.6)	73.1	(10.6)		
6				11.7	(1.7)	68.9	(10.0)		

\*Specimen gripped in machine with special grips and pin through specimen end tabs

 $\overline{E}$  = 10.1 GPa (1.47 Msi)

 $\overline{F}_{TU} = 61.6 \text{ MPa} (8.82 \text{ ksi})$ 

Table 6-13. Phase II Test Series, Sequence 4 Test Results, C6000/P1700 [±45/∓45], Baseline, In the Vacuum Chamber, No Radiation

S/N	Irradiation Temperature °C (°F)	Tensile Test Temperature °C (°F)	Tensile Test Environment	Mod	E sile ulus (Msi)	U1 t Ten	TU imate sile ength (ksi)	F <sub>TY</sub> Yiel Tens Stren MPa	d ile	Eu Ultimate Elongation (Percent)
1			In air	9.7	(1.40)	64.8	(9.400)	52.9	(7.673)	1.78
2			ļ	11.6	(1.68)	53.6	(7.778)	44.7	(6.481)	1.69
3				9.7	(1.41)	76.0	(11.026)	55.7	(8.076)	3.32
4				13.7	(1.99)	91.3	(13.248)	63.2	(9.172)	4.23
5				10.1	(1.47)	76.0	(11.025)	56.3	(8.162)	3.49
6	N/A	20 (68)	In air	11.3	(1.64)	61.5	(8.924)	56.1	(8.143)	1.50
7			In vacuo	10.7	(1.55)	80.2	(11.630)	56.5	(8.194)	2.37
8				8.2	(1.28)	88.1	(12.772)	47.6	(6.899)	6.15
9				11.2	(1.63)	74.7	(10.840)	60.8	(8.824)	2.02
10				7.9	(1.15)	70.1	(10.172)	49.3	(7.210)	2.26
11				8.3	(1.20)	87.5	(12.697)	57.1	(8.280)	5.44
12			In vacuo	8.4	(1.22)	82.9	(12.026)	55.1	(7.996)	5.40

<sup>\*</sup>Strength at 1 percent (0.010 in/in) elongation

= 11.0 GPa (1.60 Msi) (In air)

= 9.2 GPa (1.34 Msi) (In vacuo)

= 70.6 MPa (10.234 ksi) (In air)

= 80.6 MPa (11.690 ksi) (In vacuo)

 $F_{TY} = 54.8 \text{ MPa} (7.951 \text{ ksi}) (In air)$ 

 $\overline{F}_{TY}$  = 54.5 MPa (7.901 ksi) (In vacuo)  $\overline{\epsilon}_{u}$  = 2.67 percent (In air)  $\overline{\epsilon}_{u}$  = 3.94 percent (In vacuo)

Table 6-14. Phase II Test Series, Sequence 6 Test Results, C6000/P1700 [±45/∓45], 1 x  $10^{10}$  Rads Dose ( $\approx 8$  Hours/Day, 34 Days)

S/N	Irradiation Temperature °C (°F)	Tensile Test Temperature °C (°F)	Tensile Test Environment	Dose (x10 <sup>10</sup> rads)		E nsile dulus (Msi)	Ult Ten:	TU imate sile ength (ksi)	FT Yie Tens Strei MPa		E <sub>u</sub> Ultimate Elongation (Percent)
1 <sup>(b)</sup>					ura	(1151)	rira	(K51)	a	(K51)	(rercent)
2(p)			In air	1.0							
2,5,				1							
3		18.3 (65)	(c)		13.7	(1.99)	75.8	(10.989)	65.4	(9.490)	2.21
4		18.3 (65)			13.3	(1.93)	84.8	(12.298)	62.6	(9.073)	4.36
5,		18.3 (65)			11.5	(1.67)	72.5	(10.522)	63.1	(9.151)	1.65
6 <sup>(e)</sup>			In air								
7		17.8 (64)	In vacuo		8.9	(1.29)	80.7	(11.703)	55.8	(8.088)	2.65
8		17.8 (64)			10.1	(1.46)	79.2	(11.491)	57.8	(8.385)	4.55
9	20 (248)	17.8 (64)			11.1	(1.61)	79.3	(11.503)	65.2	(9.451)	2.11
10 <sup>(e)</sup>	1			1.0							
11		17.8 (64)		0.67	9.8	(1.42)	70.4	(10.218)	55.4	(8.037)	2.41
12		17.8 (64)	(d)	0.67	12.6	(1.82)	75.8	(10.991)	63.6	(9.226)	1.95
13		18.3 (65)	İ	0.67	11.4	(1.66)	77.0	(11.174)	52.3	(7.591)	2.54
14		20.0 (68)		0.67	8.7	(1.26)	69.1	(10.021)	55.2	(8.012)	1.83
15		29.4 (85)		0.33	8.8	(1.28)	64.3	(9.326)	51.4	(7.461)	1.76
16		33.3 (92)		0.33	9.4	(1.36)	83.7	(12.140)	57.0	(8.272)	6.03
17		37.2 (99)		0.33	10.7	(1.55)	55.1	(7.987)	55.1	(7.987)	0.99
18		40.0 (104)	In vacuo	0.33	9.1	(1.32)	67.7	(9.819)	55.2	(8.004)	4.01

<sup>(</sup>a) Strength at 1 percent (0.010 in/in) elongation(b) Specimen warped, not tested(c) 72 hours after exposure to dry air

 $F_{TY}$  = 54.7 MPa (7.931 ksi) (0.33 x 10<sup>10</sup> rads, vac)  $F_{TY}$  = 56.7 MPa (8.217 ksi) (0.67 x 10<sup>10</sup> rads, vac)  $F_{TY}$  = 59.6 MPa (8.641 ksi) (1 x 10<sup>10</sup> rads, vac)  $F_{TY} = 63.7 \text{ MPa } (9.238 \text{ ksi}) (1 \times 10^{10} \text{ rads, air})$  $\overline{\epsilon}_{II}$  = 3.28 percent (0.33 x 10<sup>10</sup> rads, vac) = 2.18 percent (0.67 x 10<sup>10</sup> rads, vac) = 3.10 percent (1 x  $10^{10}$  rads, vac) = 2.74 percent  $(1 \times 10^{10} \text{ rads, air})$ 

<sup>(</sup>d) Within 3 to 30 minutes after termination of radiation (e) Slight bend in specimen; results not included in averages

 $<sup>= 9.5 \</sup>text{ GPa} (1.38 \text{ Msi}) (0.33 \times 10^{10} \text{ rads. vac})$ = 10.6 GPa (1.54 Msi) (0.67 x  $10^{10}$  rads, vac) = 10.0 GPa (1.45 Msi) (1 x  $10^{10}$  rads, vac) = 12.8 GPa (1.86 Msi) (1 x  $10^{10}$  rads, air) = 67.7 MPa (9.818 ksi) (0.33 x 10<sup>10</sup> rads, vac) = 73.2 MPa (10.611 ksi) (0.67 x 10<sup>10</sup> rads, vac)  $F_{TU}$  = 79.7 MPa (11.565 ksi) (1 x 10<sup>10</sup> rads, vac)  $F_{TU}$  = 77.7 MPa (11.271 ksi) (1 x 10<sup>10</sup> rads, air)

Table 6-15. Phase II Test Series Data Summary, T300/934 [ $\pm$ 45/ $\mp$ 45] Laminate Composite Material

			Test Condi	i 1	No. of	Average		Stan	dard	Coefficient			
Test Sequence	Radiation Dose Level (x10 <sup>10</sup> Rads)	Radiation Dose Application	Irradiation Temperature °C (°F)	Tensil Test Temperat	Tensile Test ure Environment	Elapsed Time After Irradiation or Exposure to Air	ter Irradiation Material or Exposure Property	5 Specimens  N  6 5 4 4 5 4			Deviation S		of Variation S/X
2 2 5 5 5 5	0 0.33 0.67 1.0	N/A N/A 8 Hr/Day, 11 Days 8 Hr/Day, 22 Days 8 Hr/Day, 34 Days 8 Hr/Day, 34 Days	N/A N/A 120 (248) 120 (248) 120 (248) 120 (248)	20 (1 38 (1 31 (29 (	68) In vacuo 68) In air 00) In vacuo 88) In vacuo 85) In vacuo 68) In air	N/A =15 Min =15 Min =15 Min			9.4 11.4 12.2 9.9 12.0 12.4	(1.37) (1.63) (1.77) (1.44) (1.75) (1.80)	0.14 1.89 0.94 1.23 1.37 2.16	(0.021) (0.275) (0.137) (0.179) (0.199) (0.313)	0.015 0.168 0.077 0.124 0.114 0.174
2 2 5 5 5 5	0 0.33 0.67 1.0	N/A H/A 8 Hr/Day, 11 Days 8 Hr/Day, 22 Days 8 Hr/Day, 34 Days 8 Hr/Day, 34 Days	N/A N/A 120 (248) 120 (248) 120 (248) 120 (248)	20 ( 38 (1 31 ( 29 (	68) In vacuo 68) In air 00) In vacuo 88) In vacuo 85) In vacuo 68) In air	H/A H/A +15 Min +15 Min +15 Min 72 Hrs	FTU Ultimate Tensile Strength MPa (ksi)	6 5 4 4 5	128.4 152.6 157.6 156.9	(19.133) (18.620) (22.138) (22.859) (22.756) (24.044)	3.48 13.43 16.68 9.72 12.71 13.37	(0.506) (1.948) (2.420) (1.410) (1.844) (1.939)	0.026 0.104 0.109 0.062 0.081 0.081
2 2 5 5 5 5	0 0.33 0.67 1.0	M/A M/A 8 Hr/Day, 11 Days 8 Hr/Day, 22 Days 8 Hr/Day, 34 Days 8 Hr/Day, 34 Days	N/A N/A 120 (248) 120 (248) 120 (248) 120 (248)	20 ( 38 (1 31 ( 29 (	68) In vacuo 68) In air 00) In vacuo 88) In vacuo 85) In vacuo 68) In air	N/A N/A *15 Min *15 Min *15 Min 72 Hrs	F <sub>TY</sub> Yield Tensile Strength MPa (ksi)	6 5 4 4 5	88.6 96.0 84.2 89.7	(11.730) (12.856) (13.927) (12.210) (13.011) (14.217)	4.14 9.48 8.19 5.88 8.40 15.39	(0.601) (1.376) (1.189) (0.854) (1.219) (2.233)	0.051 0.107 0.085 0.070 0.094 0.157
2 2 5 5 5 5	0 0 0.33 0.67 1.0	M/A H/A 8 Hr/Day, 11 Days 8 Hr/Day, 22 Days 8 Hr/Day, 34 Days 8 Hr/Day, 34 Days	N/A N/A 120 (248) 120 (248) 120 (248) 120 (248)	20 ( 38 (1 31 ( 29 (	68) In vacuo 68) In air 00) In vacuo 88) In vacuo 85) In vacuo 68) In air	N/A N/A *15 Min *15 Min *15 Min 72 Hrs	Ultimate Elongation (%)	6 5 4 4 5	2.1 2.0 3.2 4.3 3.1 3.5	1  8  7  0	0. 0. 1 1	.154 .068 .856 .293 .006 .397	0.073 0.034 0.260 0.296 0.325 0.111

Table 6-16. Phase II Test Series Data Summary, C6000/P1700 [ $\pm$ 45/ $\mp$ 45] Laminate Composite Material

Test Conditions						]	No. of	Average	Standard	Coefficient	
Test Sequence	Radiation Dose Level (x10 <sup>10</sup> Rads)	Radiation Dose Application	Irradiation Temperature °C (°F)	Tensile Test Temperatur °C (°F)	Tensile Test Environment	Elapsed Time After Irradiation or Exposure to Air	Material Property	Specimens N	¥a1ue ▼	Deviation S	of Variation S/X
4 4 6 6 6 6	0 0 0.33 0.67 1.0	N/A N/A 8 Hr/Day, 11 Days 8 Hr/Day, 22 Days 8 Hr/Day, 34 Days 8 Hr/Day, 34 Days	N/A N/A 120 (248) 120 (248) 120 (248) 120 (248)	20 (68 20 (68 35 (95 18.3 (65 17.8 (64 18.3 (65	In air In vacuo In vacuo In vacuo	N/A N/A =15 Min =15 Min =15 Min 72 Hrs	E Modulus of Elasticity GPa (Ms1)	6 6 4 4 3 3	9.2 (1.34) 11.0 (1.60) 9.5 (1.38) 10.6 (1.54) 10.0 (1.45) 12.8 (1.86)	1.39 (0.201) 1.55 (0.225) 0.83 (0.120) 1.72 (0.249) 1.10 (0.160) 1.17 (0.170)	0.150 0.141 0.087 0.162 0.110 0.091
4 4 6 6 6 6	0 0 0.33 0.67 1.0	N/A N/A 8 Hr/Day, 11 Days 8 Hr/Day, 22 Days 8 Hr/Day, 34 Days 8 Hr/Day, 34 Days	N/A N/A 120 (248) 120 (248) 120 (248) 120 (248)	20 (68 20 (68 35 (95 18.3 (65 17.8 (64 18.3 (65	In air In vacuo In vacuo In vacuo	N/A N/A #15 Min #15 Min #15 Min 72 Hrs	F <sub>TU</sub> Ultimate Tensile Strength MPa (ksi)	6 6 4 4 3 3	80.6 (11.690) 70.6 (10.234) 67.7 (9.818) 73.2 (10.611) 79.7 (11.565) 77.7 (11.271)	7.11 (1.032) 13.36 (1.938) 11.91 (1.727) 3.91 (0.567) 0.82 (0.119) 6.35 (0.921)	0.088 0.189 0.176 0.053 0.010 0.082
4 4 6 6 6	0 0.33 0.67 1.0	N/A N/A 8 Hr/Day, 11 Days 8 Hr/Day, 22 Days 8 Hr/Day, 34 Days 8 Hr/Day, 34 Days	N/A N/A 120 (248) 120 (248) 120 (248) 120 (248)	20 (68 20 (68 35 (95 18.3 (65 17.8 (64 18.3 (65	In air In vacuo In vacuo In vacuo	N/A N/A =15 Min =15 Min =15 Nin 72 Hrs	FTY Yield Tensile Strength MPa (ksi)	6 4 4 3 3	54.5 (7.901) 54.8 (7.951) 54.7 (7.931) 56.7 (8.217) 59.6 (8.641) 63.7 (9.238)	4.94 (0.717) 6.03 (0.875) 2.34 (0.339) 4.85 (0.703) 4.94 (0.717) 1.53 (0.222)	0.091 0.110 0.043 0.083 0.083 0.024
4 4 6 6 6	0 0 0.33 0.67 1.0	M/A N/A 8 Hr/Day, 11 Days 8 Hr/Day, 22 Days 8 Hr/Day, 34 Days 8 Hr/Day, 34 Days	N/A N/A 120 (248) 120 (248) 120 (248) 120 (248)	20 (68 20 (68 35 (95 18.3 (65 17.8 (64 18.3 (65	) In air ) In vacuo ) In vacuo ) In vacuo	N/A N/A *15 Min *15 Min *15 Min 72 Hrs	Ultimate Elongation (Percent)	6 6 4 4 3 3	3.94 2.67 3.28 2.18 3.10 2.74	1.910 1.153 2.198 0.345 1.282 1.431	0.485 0.432 0.670 0.158 0.413 0.522

material properties and includes average values  $(\overline{X})$  from the previous tables plus standard deviation (S) and coefficient of variation (S/ $\overline{X}$ ). The scatter within the individual groups of data for both materials in Tables 6-10, 6-11, 6-13, and 6-14 is larger than that experienced for the Phase I tests, especially for ultimate elongation. Even so, except for a few instances in the modulus and strength data, the coefficient of variation is less than 15 percent. There appears to be more scatter in the C6000/P1700 data than in the T300/934 data. Through null hypothesis statistical analysis of the average values and standard deviations, comparing various post-irradiation test environments and dose effects, the results indicate that the differences in the average values among the groups of data is statistically insignificant, except in a few isolated cases.

Figures 6-3 through 6-6 plot the average values of the tensile properties evaluated as a function of dose level for the T300/934 and C6000/P1700 [ $\pm$ 45/ $\mp$ 45] laminate composite materials. Each data point shown represents the average of 3 to 6 individual test points. Only data obtained from tensile testing in the vacuum/irradiation chamber is shown. Ex situ results (in air baseline and after full term dose) and in vacuo results are included, with lines connecting the in vacuo data. There appears to be no appreciable radiation-induced changes to the tensile properties for both materials. It is difficult to identify substantial trends in the data although some may exist. Comparing the in vacuo baseline data to the full term dose in vacuo data indicates the following:

- 1) Slight stiffening of the materials (10 to 20 percent increase in modulus of elasticity)
- Slight strengthening of the materials (10 to 20 percent increase in ultimate tensile strength, with a lesser effect on yield strength)
- 3) Because of the large scatter in the ultimate elongation data, it is very difficult to reach any conclusion; however, there may be a slight increase in ultimate elongation for T300/934 and a slight decrease for C6000/P1700.

Since the absolute changes in the tensile properties were small, the existence of a post-irradiation test environment effect is difficult to determine. There appears to be a trend indicating that ex situ testing results in slightly higher test values for irradiated material, although

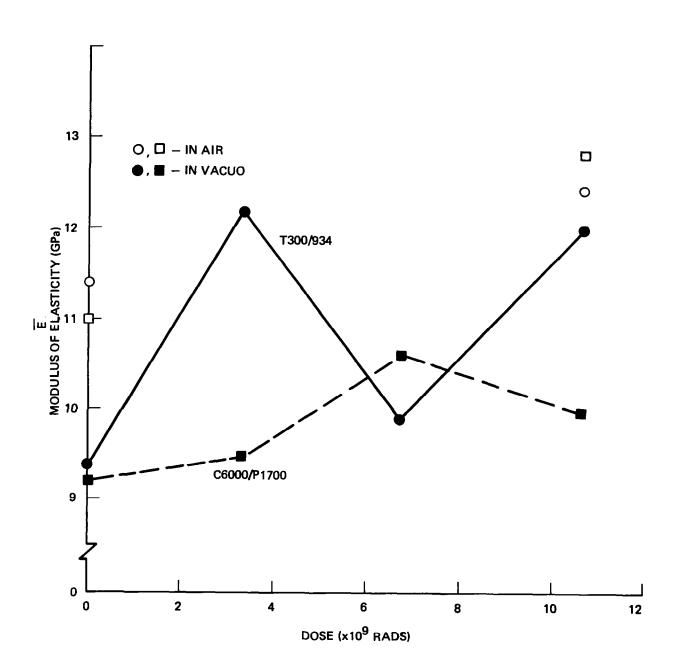


Figure 6-3. Effect of Radiation on Modulus of Elasticity, T300/934 and C6000/P1700 [ $\pm$ 45/ $\mp$ 45] Laminate Composite Materials.

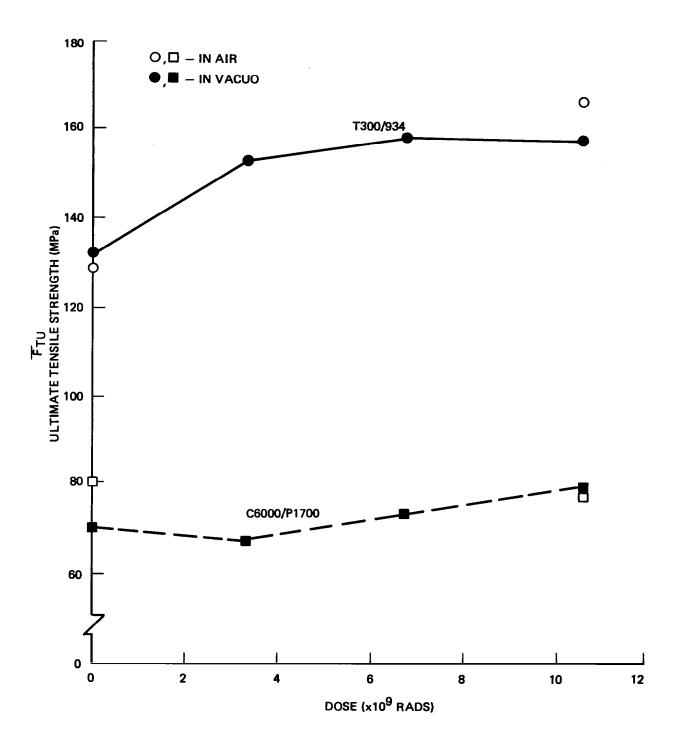


Figure 6-4. Effect of Radiation on Ultimate Tensile Strength, T300/934 and C6000/P1700 [ $\pm45/\mp45$ ] Laminate Composite Materials.

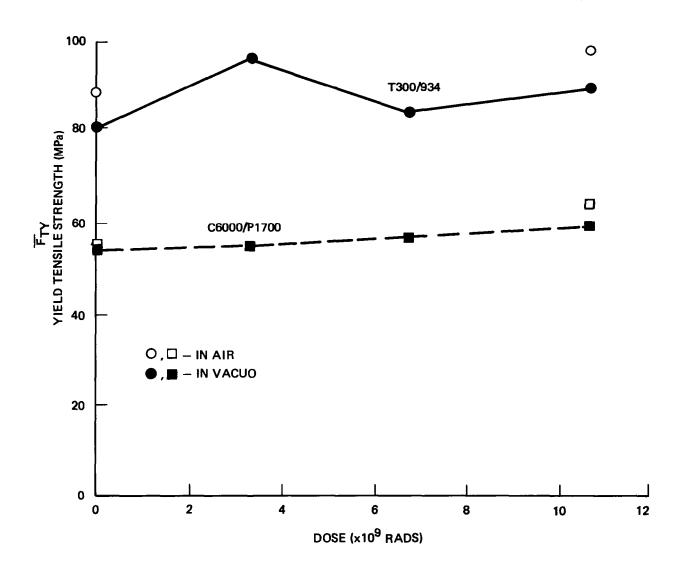


Figure 6-5. Effect of Radiation on Yield Tensile Strength, T300/934 and C6000/P1700 [ $\pm 45/\mp 45$ ] Laminate Composite Materials.

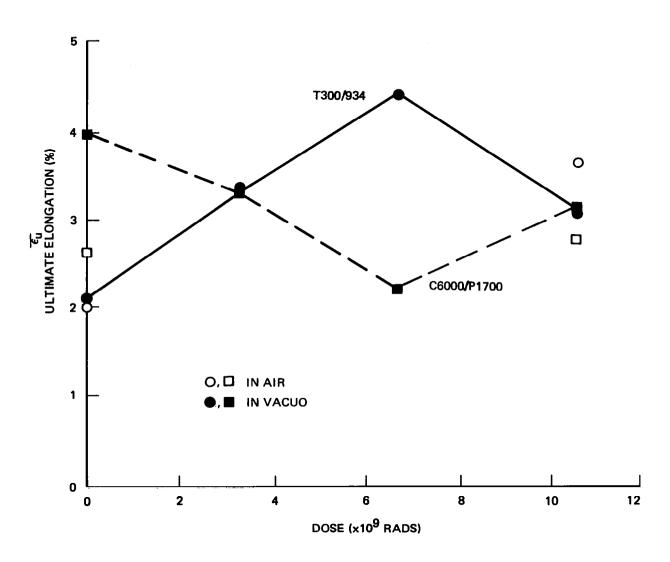


Figure 6-6. Effect of Radiation on Ultimate Elongation, T300/934 and C6000/P1700 [ $\pm 45/\mp 45$ ] Laminate Composite Materials.

not in all instances. On the other hand, this apparant effect may be due to small differences in the behavior of the in situ tensile testing apparatus, since the baseline chamber tensile testing results indicated that in air data is larger than in vacuo data, in most instances.

Because of the somewhat surprising lack of radiation-induced changes at the relatively high dose level used in Phase II, a review of the test operation was made to insure that indeed the test specimens were exposed to the electron fluences generated. The data collected from the downstream annulus in the VDG beam scattering chamber, coupled with the Faraday cup calibration results, clearly indicated that the specimens were exposed to the full electron beam. As to whether the electrons penetrated the material or were somehow impeded/repulsed by the formation of an "electron cloud" at the surface of the specimens, there was no evidence to indicate this to be the case. The materials are electrically semiconductive. Any charge built up on the surface or within the material can bleed off to ground through the carousel structure. The material breakdown voltage capabilities are low relative to the 700 keV energy level. Thus, the ability to deflect the impinging beam with any surface charge buildup is small. There was no evidence of discharging marks on the surface of the specimens. Hence, it must be concluded that the specimens received and absorbed the fluences and dose levels intended.

# 6.3.4 Post-Test Examination

All specimens were microscopically examined at 20x for possible differences in failure mechanisms or other evidence of changes or degradation. No indication of any obvious physical change was noted as the result of the  $1 \times 10^{10}$  rads dose. Examination of the specimen surface revealed no discolorations or marks from any discharge phenomena.

Typical specimens were photographed at approximately 2x for record purposes. These same specimens were examined at higher magnifications (typically 1000x) using a scanning electron microscope. Figures 6-7 through 6-9 represent typical photomicrographs for control and irradiated specimens. The failed surfaces appeared very similar. No obvious differences in failure mechanisms were noticeable. Figure 6-10 shows one of the typical warped specimens. This warpage occurred more frequently on the  $\frac{10000}{1000}$  material and was noticed after  $\frac{10000}{1000}$  rads dose.

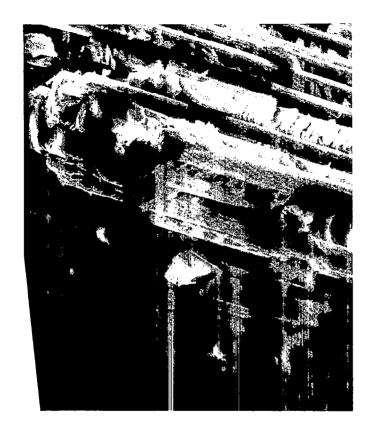


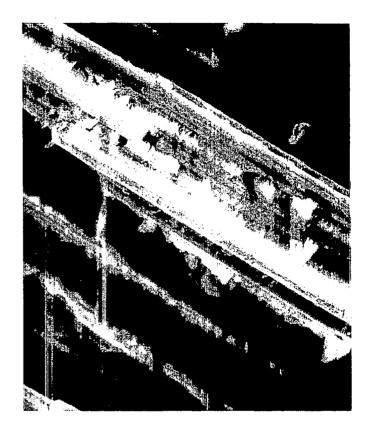
(a) Specimen No. 2, Baseline, Tested In Air



(b) Specimen No. 8, 1 x 10<sup>10</sup> Rads Dose at 120°C, Tested In Vacuo

Figure 6-7. Photomicrographs (1000x) of T300/934 Specimens.





- (a) Specimen No. 2, Baseline, Tested In Air (b) Specimen No. 5, 1 x  $10^{10}$  Rads Dose at  $120^{\circ}$ C, Tested In Air

Figure 6-8. Photomicrographs (1000x) of T300/934 Specimens.







Figure 6-9. Photomicrographs (1000x) of C6000/P1700 Specimens.

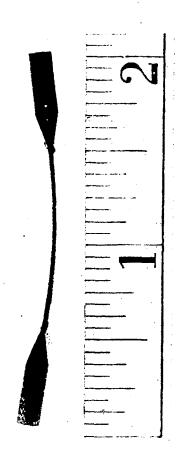


Figure 6-10. Photograph (2x) of Warped C6000/P1700 Specimen No. 2, 1 x  $10^{10}$  Rads Dose at 120° C.

#### 7.0 CONCLUSIONS AND RECOMMENDATIONS

#### 7.1 CONCLUSTONS

The ability to realistically evaluate the significance of post-irradiation test environments required: 1) there be well-established trends in the radiation-induced changes to the materials, and 2) the magnitude of the changes be large. This was not the case for the T300/5208, T300/934, and C6000/P1700 composite materials tested under this program, at least in terms of the tensile properties measured.

Using a  $[\pm 45/\mp 45]$  laminate specimen provided a good opportunity to measure the radiation-induced effects on the resin matrix, as well as the resin-fiber interface, since it was assumed that the radiation damage to the reinforcement fibers would be negligible. Nevertheless, exposures up to 1 x  $10^{10}$  rads dose at  $120^{\circ}$ C ( $248^{\circ}$ F) caused only small changes to modulus of elasticity, ultimate tensile strength, yield strength, and ultimate elongation. The change to most properties was less than 20 percent. Differences between in vacuo test data and ex situ test data were small.

The trends obtained from the data indicated: (1) radiation slightly increased the stiffness and strength of the materials, and (2) radiation decreased the ultimate elongation of the materials.

## 7.2 RECOMMENDATIONS

Before concluding that the requirement for in vacuo or in situ testing of radiation-exposed composite materials is not warranted, additional studies are required to evaluate other significant material properties that were not measured. These properties include glass transition temperature, coefficient of thermal expansion, compression and shear strength. Larger sample populations are required to reduce the statistical effects and to handle the variability in properties generic to composite materials.

The simulation of the space radiation environment should be enlarged to include high energy protons, low energy charged particles and ultraviolet which can cause large absorbed doses in the surface plies. This may cause more dramatic changes in material properties not monitored under this program.

The apparent improvements in some of the tensile properties observed in this study as the result of "radiation curing" is an interesting byproduct that needs further investigation.

#### APPENDIX A

## BACKSCATTER EFFECTS ON TEST DOSE LEVELS

The dose-depth calculations shown in Figures 4-3 and 4-4 and in Table 4-1 of Section 4 assumed that there was no backing behind the test specimens that could affect the dose levels. In actuality, for some tests the only close-in backing was the rear and sides of the stainless steel vacuum chamber. For other tests, a heater unit with a stainless steel reflector plate was located in close proximity behind the specimens.

As a worst-case assumption, the previous unbacked calculations presented in Section 4.1 are repeated with a stainless steel surface being flush with the back of the test specimen. Figures A-1 and A-2 present the dose-depth profiles for the fully backed condition for 700 keV electrons, using a 2000 history case TIGER code Monte Carlo solution. Also shown is a comparison to the unbacked solutions. Table A-1 lists the average unit dose and fluence required for 1 x  $10^9$  rads dose. The results indicate that the fully backed specimen dose increases more rapidly as the thickness is traversed, such that the back-to-front dose ratio is greater than two. The average bulk dose is 33.5 percent greater than the unbacked specimen dose.

Table A-1. Average Unit Dose From a 700 keV Electron Beam Normally Incident on Specimens (Comparison of Unbacked Versus a Flush Backing of Stainless Steel)\*

Materia	1	Average Un	Fluence for 1 x 10 <sup>9</sup> rads	
		(MeV·cm <sup>2</sup> /gm·e)	(rad.cm <sup>2</sup> /e)	(e/cm <sup>2</sup> )
C6000/P1700	Unbacked	2.46	$3.94 \times 10^{-8}$	2.54 x 10 <sup>16</sup>
(0000771700	Backed	3.29	5.26 x 10 <sup>-8</sup>	1.90 x 10 <sup>16</sup>
T300/934	Unbacked	2.45	3.92 x 10 <sup>-8</sup>	2.55 x 10 <sup>16</sup>
1300/334	Backed	3.27	5.24 x 10 <sup>-8</sup>	1.91 x 10 <sup>16</sup>

<sup>\*2000</sup> History Monte Carlo Analysis

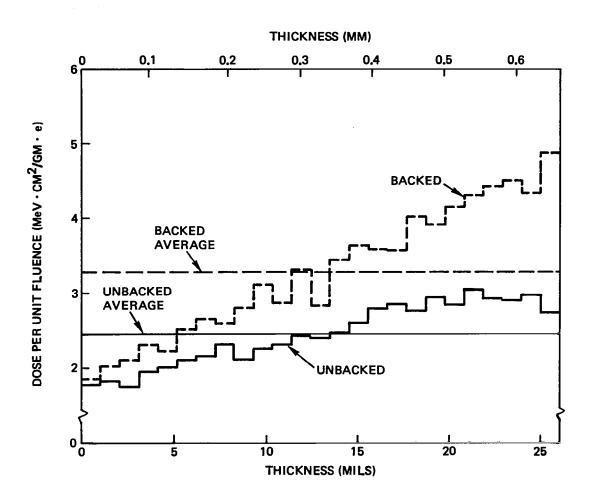


Figure A-1. Unit Dose as a Function of Thickness for a 700 keV Electron Fluence Normally Incident on Material C6000/P1700 Unbacked and Backed by Stainless Steel.

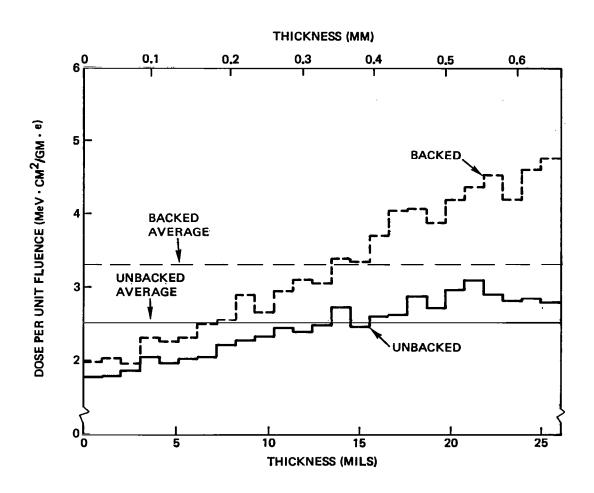


Figure A-2. Unit Dose as a Function of Thickness for a 700 keV Election Fluence Normally Incident on Material T300/934 Unbacked and Backed by Stainless Steel.

In reality, the actual conditions in the test chamber lie somewhere between the two extremes presented. For the condition where only the rear and side surfaces of the vacuum chamber can back-scatter the electrons, the backscattering relative to that produced by a flush mounted scatterer can be estimated assuming that: (1) the backscatter effect is isotropic in the electron direction, (2) forward scatter from the specimen is isotropic, and (3) there is little if any direct primary beam illumination of the backscatterer. The first and second assumptions are reasonably justified by noting that the angular distributions appearing in the TIGER code calculations are roughly isotropic. The last assumption is not entirely correct. The electron beam diameter at the target plane is limited to a radius only slightly greater than the carousel radius. The target plane geometry further restricts the illuminating beam to an annulus from 10 cm to 15 cm (4 in. to 6 in.) diameter of which roughly one-half is blocked by the specimen area. The open spaces in this annular area allow the primary beam to further expand onto the chamber walls in a non-isotropic illumination. To a first approximation, it is correct to assume that the generator of forward scattering has the area of the entire annulus. Under these circumstances it can be shown that the dose, D, delivered to the specimens is approximated by:

$$\frac{D}{D_{0}} = 1 + \left(\frac{D_{B}}{D_{0}} - 1\right) \frac{A_{S}}{d^{2}} \cdot \frac{\Delta \Omega}{2\pi},$$
 (A1)

where

 $\mathbf{D}_{\mathbf{O}}$  is the bulk dose calculated for the unbacked specimen,

 $\mathbf{D}_{\mathbf{R}}$  is the bulk dose calculated for the intimately backed specimen,

 $\mathbf{A}_{\mathbf{s}}$  is the area of the forward scatterer,

d is the "average" distance of the backscatterer from the specimen,

 $\Delta\Omega$  is the solid angle subtended by the backscatterer at the specimen.

As noted before, for the case of backscatter from the relatively distant chamber walls,  $A_{\rm S}$  is taken as the B.3full annulus area, where:

$$A_s = \frac{\pi}{4}(15^2 - 10^2) = 98.1 \text{ cm}^2(15.2 \text{ in.}^2)$$
 (A2)

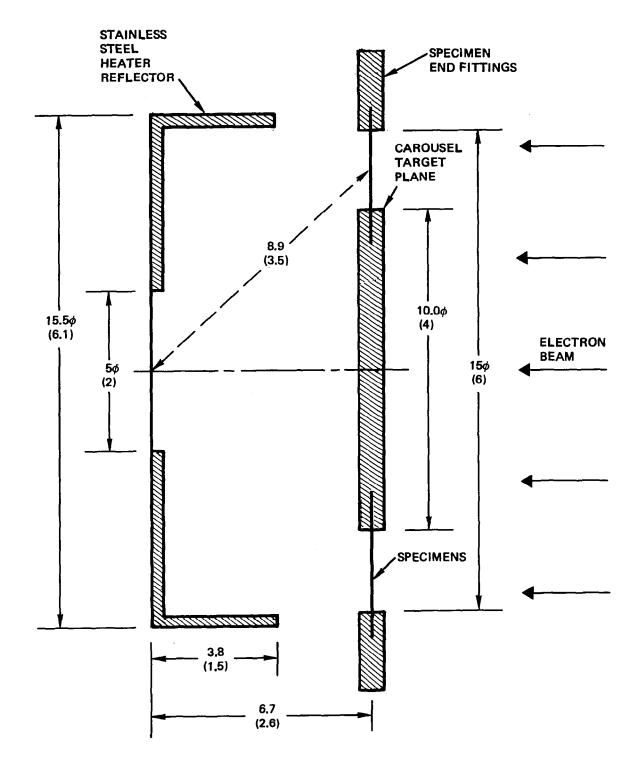
The chamber rear wall is roughly (45.7 cm) (18 in.) from the target plane and the chamber diameter is (61.0 cm) (24 in.), giving an average distance of about 38.1 cm (15 in.) from the specimens to the chamber walls. The backscattering solid angle ( $\Delta\Omega$ ) from the walls is essentially in  $2\pi$  geometry. Therefore,

$$\frac{D}{D_{0}} = 1 + \left(\frac{D_{B}}{D_{0}} - 1\right) \frac{98.1}{1451.6} \left(\frac{2\pi}{2\pi}\right)$$

$$= 1 + 0.068 \left(\frac{D_{B}}{D_{0}} - 1\right).$$
(A3)

Since  $D_B/D_0$  is 1.335 for the fully backed condition for the specimen materials,  $D/D_0=1.023$ . Therefore, the backscattering effect from the chamber walls is negligible for Phase I testing, Sequences 3 and 4.

A heating coil/heat reflector plate was installed directly behind the specimens for the Phase I, Sequence 5 tests and all subsequent Phase II radiation tests as shown in Figure A-3. The scatterer (reflector plate) does not subtend a full  $2\pi$  solid angle at the specimens. The solid angle,  $\Delta\Omega$ , is estimated to be approximately  $1.2\pi$  steradians. The actual illumination for the heat reflector backscatterer is about 50 percent from the primary election beam normal to the reflector and about 50 percent from the forward isotropic scatter from the specimens. The first component will not produce backscatter as effectively because of the higher energy and non-isotropic impingement. Hence, the effective  $A_{\rm S}/d^2$  is reduced. The average distance (d) can be taken as that to the centerline of the reflector and equals 8.9 cm (3.5 in.). A value of 2.2 is used as a correction factor to convert from omnidirectional to isotropic reflection off the heater reflector plate.



NOTE: ALL DIMENSIONS IN CM (IN.)

Figure A-3. Phase II Heater Backscattering Geometry.

Therefore,

$$\left(\frac{A_s}{d^2}\right)_{eff} = \frac{A_s/2 + A_s/(2)(2.2)}{d^2}$$

$$= \frac{0.73 A_s}{d^2}$$

$$= 0.73 (98.1)/(8.9)^2$$

$$= 0.90 .$$
(A4)

Thus,

$$\frac{D}{D_0} = 1 + (1.335 - 1)(0.90) \left(\frac{1.2\pi}{2\pi}\right)$$

$$= 1.18 .$$
(A5)

The effective dose is approximately 18 percent greater than the unbacked specimen dose due to the presence of the heater unit. Therefore, the 700 keV electron fluence required to produce an average dose in the specimens of 1  $\times$  10<sup>9</sup> rads (refer to Table 4-1) for those tests having the installed heater unit is reduced to approximately

$$\frac{2.57 \times 10^{16} \text{ e/cm}^2}{1.18} = 2.2 \times 10^{16} \text{ e/cm}^2. \tag{A6}$$

The conversion from flux to dose is increased to

$$\frac{3.89 \times 10^{-8} \text{ rad cm}^2/\text{e}}{1.18} = 4.59 \times 10^{-8} \text{ rad cm}^2/\text{e}. \tag{A7}$$

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#### APPENDIX B

### CALIBRATION OF VAN DE GRAAFF HIGH-ENERGY ELECTRON SOURCE

This Appendix covers calibrations of electron beam energy, target plane flux and flux distribution, and the computations used in setting up irradiation schedules. An evaluation of the accuracies and uncertainties involved is also included.

## **B.1 REQUIREMENTS**

The electron energy requirement was specified to be that which would allow a mono-energetic electron dose through the material to be uniform within a factor of two, within the constraints of the system energy capability and irradiation time. An energy of 700 keV was determined to be an appropriate level (see Section 4.1).

Depending on the specific test, the average absorbed dose in the test material was  $1 \times 10^9$  rads,  $2 \times 10^9$  rads, or  $1 \times 10^{10}$  rads. Section 4.1 and Appendix A summarize calculations made to arrive at a dose-to-fluence conversion factor for the case where no backing material was in the vicinity of the test specimens and for the case where a backscattering material was in the vicinity of the test specimens.

The desired goal on beam fluence variation over the specimens was to be less than  $\pm 10$  percent. Since the target plane was normal to the center line of the electron beam and all specimens were at approximately the same angle with respect to this centerline, the only substantial specimen-to-specimen fluence variations could be due to the beam being off-center, to shadowing, or to localized scatterers.

Table B-1 lists the <u>desired</u> irradiation levels for the Phase I and Phase II tests that required electron exposure. The fluxes listed in the table were determined from machine calibrations discussed in Section B.3. In Section B.5 the <u>actual</u> flux, fluence and dose levels achieved during the program are listed.

Table B-1. Electron Irradiation Requirements

Test Sequence No.	Nominal Average Dose (Rads)	Nominal Fluence (e/cm <sup>2</sup> )	Nominal Flux (e/cm <sup>2</sup> sec)	Nominal Radiation Time
Phase I				
3	1 x 10 <sup>9</sup>	$2.6 \times 10^{16}$	$2.1 \times 10^{11}$ (a)	34 hr at ≃8 hr/day for 4.5 consecutive days
4	1 x 10 <sup>9</sup>	$2.6 \times 10^{16}$	$2.1 \times 10^{11}$ (a)	34 hr at 24 hr/day for 1.5 days
5	2 x 10 <sup>9</sup>	$5.2 \times 10^{16}$	2.1 x 10 <sup>11</sup> (a) (c)	68 hr at ≃8 hr/day for 9 consecutive days
Phase II				
5	$1 \times 10^{10}$	2.2 x 10 <sup>17</sup>	$2.4 \times 10^{11}$ (b)	250 hr at ≃8 hr/day for 33 consecutive days
6	$1 \times 10^{10}$	$2.2 \times 10^{17}$	$2.4 \times 10^{11}$ (b)	250 hr at ~8 hr/day for 33 consecutive days

<sup>(</sup>a) No backscattering, machine operating at  $\simeq 45~\mu A$  beam current

<sup>(</sup>b) 15 percent dose addition assumed due to backscattering at  ${\approx}60~\mu\text{A}$  beam current

<sup>(</sup>c) Assumed rate before backscatter of heater unit was recognized

#### **B.2 ELECTRON ENERGY**

The Van de Graaff (VDG) accelerator is equipped with a generating voltmeter which samples the electrostatic field existing between the high voltage terminal and the tank wall. The instrument tracks over the voltage range of the accelerator to within one percent. Its output is coupled to a digital voltmeter (DVM) located in the VDG control console. On a previous program (Reference 4) calibrations were made against the current through the column resistors and against the attenuation of the beam in aluminum layers. It was found that 102.5 mV ( $\pm 3$  percent) on the digital voltmeter  $\equiv 100$  kV terminal potential. Since the electrons are accelerated from the terminal potential to ground, a 100 kV terminal potential will accelerate electrons to 100 keV energy.

In addition to diffusing the electron beam as it passes through the scattering foils, the electron beam loses some energy. Previous programs in the SESL facility have utilized Monte Carlo calculations by Seltzer and Berger (Reference 5) in determining this energy loss; however, the published calculations do not extend above 400 keV incident energy except for one curve given for a silicon scatterer at energies up to 1 MeV. This curve plots the most probable energy after passing through the foil, which is somewhat higher than the desired average energy from which the dose distribtuion is determined. Reference 5 includes determinations within the 100-400 keV range of electron energies of both the average emergent energy and the most probable emergent energy in terms of the incident energy. These data were utilized to determine the average on-target energy for this study on the assumption that the mechanism for electron scattering is nearly the same at 700 keV as at 100-400 keV. Comparisons were made with values obtained using data from References 4 and 6.

A foil thickness was chosen for the program which would maximize the flux on the carousel. A foil that was too thin would not disperse the beam out to the proper scattering angle. If the foil were too thick, the beam would be dispersed too much. To the normal 0.0165 mm (0.65 mil) vacuum separator foil was added two thicknesses of 0.0254 mm (1 mil) aluminum foil on the rotator, giving a total thickness of 0.0673 mm (2.65 mil) of

aluminum. Using a machine setting of 746 mV on the DVM resulted in an on-target average energy level of approximately 700 keV. This was determined from the following relationships:

- a) From the VDG machine calibration, 1 keV incident energy ( $E_0$ ) on the foil system  $\equiv 1.025$  mV on the DVM
- b) For a total foil thickness of 0.0673 mm (2.65 mil), the ratio of most probable emerging energy (E $_{\rm p}$ ) to incident energy on the foil (E $_{\rm o}$ ) was calculated to be 0.968
- c) The ratio of on-target average energy ( $\bar{E}$ ) to the most probable energy ( $E_{\rm p}$ ) was calculated to be 0.988.

Therefore,

$$E_0 = (\frac{DVM}{1.025}) = \frac{746}{1.025} = 728 \text{ keV}$$
 (B1)

and

$$\overline{E} = E_0 \left( \frac{E_p}{E_0} \right) \left( \frac{\overline{E}}{E_p} \right)$$

$$= 728(0.968)(0.988)$$

$$= 697 \text{ keV (to an uncertainty of ±3 percent).}$$
(B2)

#### **B.3 FLUX CALIBRATION**

Electron Flux calibrations were made using a small three-element graphite Faraday cup with an entrance aperture area of  $0.503 \, \mathrm{cm}^2$  ( $0.078 \, \mathrm{in.}^2$ ). The carousel was removed for the calibration to allow the Faraday cup to be positioned in the same plane as the carousel, which was oriented normal to the electron beam centerline. The Faraday cup was oriented with its centerline normal to the beam direction but displaced with respect to the axis of the carousel by  $6.25 \, \mathrm{cm}$  ( $2.46 \, \mathrm{in.}$ ), which is the radial distance to the center of the specimen locations. The Faraday cup centerline was positioned in azimuth in the approximate specimen position no. 5.

The electrical leads from the Faraday cup were protected from primary and secondary electron impingement by 0.20 cm (80 mil) wall thickness

copper tubing. The outer shield was grounded at the chamber. The aperture and cup bias were varied while under beam loading. No changes were noted, indicating that no secondary electron transport was taking place.

The Faraday current was monitored on a current integrator as was the current from the exit aperture. The beam energy was set a DVM reading of 746 mV (corresponding to an on-target average energy of approximately 700 keV), and three runs were made for simultaneous integration periods on each current integrator. The results are shown in Table B-2. The average current ratio is  $1.69 \times 10^{-3}$ . Using the Faraday cup area of  $0.503 \text{ cm}^2$ , for the charge collected on the exit aperture, this translates into a calibration constant of:

$$\frac{1.69 \times 10^{-3}}{1.602 \times 10^{-19} \times 0.503} = 2.10 \times 10^{10} \text{ e/cm}^2 \text{sec/}_{\mu}\text{A}.$$
 (B3)

From Section 4.1 and Appendix A, for Phase I Test Sequences 3 and 4, the dose conversion factor was calculated to be 2.57 x  $10^7$  e/cm<sup>2</sup>rad for the materials. For Phase I Test Sequence 5 and all Phase II Test Sequences the dose conversion factor was 2.2 x  $10^7$  e/cm<sup>2</sup>rad.

Table B-2. Exit Aperture Target Plane Flux Calibration at 746 mV DVM Electron Energy (≡ 697 keV)

	Beam	Exit Aperture (EA)		Faraday C		
Run	Current (µA)	Current (µA)	Charge (µC)	Current (µA)	Charge (µC)	Q <sub>FC</sub> /Q <sub>EA</sub>
1	49.7	12	3000	0.020	5.08	1.69 x 10 <sup>-3</sup>
2	49.7	12	1500	0.020	2.53	$1.69 \times 10^{-3}$
3	49.7	12	1500	0.020	2.52	$1.68 \times 10^{-3}$

From these conversion values, the experimental values determined in Table B-2, and the requirements in Table B-1, irradiation schedules were drawn up for each sequence test run. Typical schedules for Phase I (Sequence 3) and Phase II (Sequence 5) are presented in Tables B-3 and B-4, respectively. Sequence I-4 is the same as Sequence I-3 except that the

# Table B-3. Irradiation Schedule (Phase I, Test Sequence No. 3, +30 Degree Beam Leg)

# Van De Graaff Settings:

Voltage: 746 mV on DVM Current:  ${\approx}45~\mu\text{A}$  nominal Shorting Bar: Not used

# Scattering Foils:

0.0165 mm (0.65 mil) alum normal and two pc at 0.0254 mm (1 mil) each Alum on 1.27 cm (0.5 in.) aperture rotator position

Exit Aperture Current: ~10 μA

Total Dose Delivered:  $1 \times 10^9$  rads = 2.57  $\times 10^{16}$  e/cm<sup>2</sup> at 700 keV

Exit Aperture Conversion Factor:  $2.10 \times 10^{10} \text{ e/cm}^2 \text{µC}$ 

Total Charge Required:  $\frac{2.57}{2.10} \times 10^6 \, \mu \text{C} = 1.224 \, \text{C}$ 

Delivery Program: 4 1/2 days consecutive at ≃8 hr/day

Charge/Day:  $1.224/4.5 = 0.273 \text{ C/day} = 2.73 \text{ x } 10^5 \text{ } \mu\text{C/day}$ 

Current Integrator Settings:

Current Range: 30  $\mu A$ 

Count Per Day:  $\frac{2.73 \times 10^5}{30} = 9,100$  counts

### Schedule:

First 4 days: 9,100 counts each day

Last day: 4,550 counts

Approximate Irradiating Time:

7.5 hr. (first 4 days)

3.7 hr. (last day)

# Table B-4. Irradiation Schedule (Phase II, Test Sequence No. 5, +30 Degree Beam Leg)

# Van de Graaff Settings:

Voltage: 746 mV on DVM Current:  $\approx 55 \mu A$  nominal Shorting Bar: Not used

# Scattering Foil(s):

0.0165 mm (0.65 mil) alum normal and two pc at 0.0254 mm (1 mil) each on 1.27 cm (0.5 in.) aperture rotator position

Exit Aperture Current:  $\approx 12 \mu A$ 

Total Dose Delivered: 1 x 10<sup>10</sup> rads

Flux to Dose Conversion Factor:  $2.23 \times 10^{-7} \text{ e/cm}^2 \text{ rad at } 700 \text{ keV}$ 

Exit Aperture Conversion Factor:  $2.10 \times 10^{10} \text{ e/cm}^2 \mu\text{C}$ 

# Total Charge Required:

3.19 C at 3  $\times$  10<sup>9</sup> rads 6.39 C at 6  $\times$  10<sup>9</sup> rads 10.64 C at 1  $\times$  10<sup>10</sup> rads

Delivery Program: 34.5 days consecutive =7 hr/day

Charge/Day:  $\frac{10.64}{34.5} = 0.310 \text{ C/day} = 3.1 \times 10^5 \, \mu\text{C/day}$ 

# Current Integrator Settings:

Current Range: 30  $\mu A$ 

Count Per Day:  $\frac{3.1 \times 10^5}{30} = 10,333$  counts

## Total Count:

103,000 counts at 3 x  $10^9$  rads 206,660 counts at 6 x  $10^9$  rads 354,670 counts at 1 x  $10^{10}$  rads delivery was continuous over a 34-hour period. Sequence I-5 is the same as Sequence I-3 except that the dose was increased by a factor of two to 2 x  $10^9$  rads; thus, the irradiation period and total count were increased proportionately (8 1/2 days and 81,900 count). Sequence II-6 was originally planned to be terminated at 6 x  $10^9$  rad but was replanned to continue until 1 x  $10^{10}$  rads was reached. Hence, the schedule for Sequence II-6 was identical to Sequence II-5.

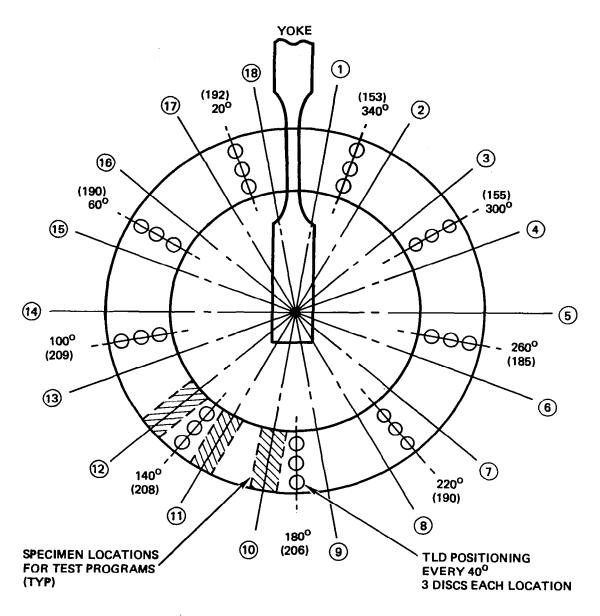
#### **B.4 TARGET PLANE FLUX VARIATION**

The target plane was mapped with small disc thermoluminescent dosimeters (TLD) to determine the variation in dose at the specimen carousel positions. The central part of the carousel, yoke and upright support bars were all in place during the calibration irradiation test. Three dosimeters were mounted at each location on a thin piece of insulating board, and carefully positioned with respect to the yoke and specimen positions every 40 degrees around the carousel. After irradiation at a DVM setting of 746 mV ( 700 keV), the dosimeters were removed and read out on a calibrated reader. The results are displayed in Figure B-1. The average of three readings is given in parentheses. In general, the outside TLD readings were lower than the inside TLD readings.

The average of the nine locations is 187, with high and low extremes of 209 and 153 respectively. A flux variation of  $\pm 11$  percent,  $\pm 18$  percent results. If the two upper right hand locations are not considered, the average is 197, resulting in a flux variation over three-fourths of the specimen locations of  $\pm 6$  percent. The measured variation is believed due to (1) an offset of the real beam axis from the center of the carousel, and (2) the use of a foil system which maximized the flux on the specimens at the expense of having to operate in the wings of the angular scattering distribution where flux falloff is rapid with angle.

# B.5 SUMMARY OF ACTUAL DOSE LEVELS

Table B-5 lists the actual doses received by the test specimen for each radiation test sequence. These have been calculated from conversion factors derived in Section B.3. The dose levels shown in Table B-5 are for specimen position no. 5 (which was closest to the Faraday cup position).



X SPECIMEN POSITION NUMBER
(xxx) RELATIVE TLD DOSE, AVERAGE OF 3 DOSIMETERS

Figure B-1. Relative Electron Dose Map at 700 keV Electron Energy.

Table B-5. Summary of 700 keV Electron Irradiation Run Data

Phase	Test Sequence	Planned Dose (x 10 <sup>9</sup> rad)	Flux (x 10 <sup>11</sup> e/cm <sup>2</sup> sec)	Charge Collected, Exit Aperture (C)	Dose Delivered (x 10 <sup>9</sup> rads)	Delivery Schedule
	3	1.0	2.2	1.230	1.0 <sup>(a)</sup>	≃8 hr/day, 4 1/2 day
I	4	1.0	2.1	1.224	1.0 <sup>(a)</sup>	34 hr continuous
	5	2.0	2.4	2.457	2.4 <sup>(b)</sup>	≃8 hr/day, 8 1/2 day
		3.0	2.7	3.11	3.6	≃7 hr/day, 10 days
ΙΙ	<sub>5</sub> (c)	6.0	2.7	6.21	6.9	≃7 hr/day, 20 days
		10.0	2.7	10.39	10.0	≃7 hr/day, 33 1/2 days
		3.0	2.7	3.39	3.3	≃7 hr/day, 11 days
H	<sub>6</sub> (c)	6.0	2.7	6.66	6.4	≃7 hr/day, 21 1/2 days
		10.0	2.7	11.00	10.6	≃7 hr/day, 34 days

<sup>(</sup>a) Backscatter effect from vacuum chamber walls is negligible (b) Includes backscatter effect from heater unit (c) Electron fluence adjusted to offset backscatter effect

The doses delivered to the rest of the specimens can be obtained by using the relative dose factors shown in Figure B-1, and comparing the value at position no. 5 to the relative dose at the specimen of interest.

The high dose of 2.4  $\times$  10<sup>9</sup> rads obtained (2  $\times$  10<sup>9</sup> rads planned) in Phase I, Sequence 5 reflects the additional dose due to the backscatter from the heating unit installed in the chamber at that time. No allowance was made to the irradiation program at the time of the test to account for this additional dose. The Phase II testing program incorporated the backscatter effect (Appendix A) in determining the required test fluences to achieve the desired dose levels.

#### B.6 DOSE UNCERTAINTY/ACCURACY

The current integrators are kept in calibration by reference to N.B.S. Standards at six-month intervals and are quoted as being accurate to within  $\pm 2$  percent of integrated count. The Faraday cup calibration is believed to be accurate to  $\pm 5$  percent. The actual fluences experienced at each specimen position, if corrected using Figure B-1 data, would include an rms uncertainty of 3 percent for the averaged dosimeter readouts. The rms'd uncertainty in delivered electron fluence to each specimen is then about 6 percent at one sigma.

The dose calculations performed in Section 4.1 and Appendix A are subject to uncertainties caused by the statistical process of the Monte Carlo method, as well as the inaccuracies associated with the basic assump tions employed. Some estimate of the combined uncertainties can be obtained by comparison with experiment. This has been done in Reference 2 which uses most of the codes employed by TIGER. The number of histories employed in that reference is 20,000 in contrast to the 8,000 employed here. The comparison with experiment for roughly comparable electron energy and material appears to show agreement within about 5 percent. For the lowered statistics of this experiment, the uncertainty might be expected to rise to perhaps as much as 8 percent. Combined with the fluence uncertainty of 6 percent, this indicates that the average deposited dose can be assigned an uncertainty of approximately 10 percent rms.

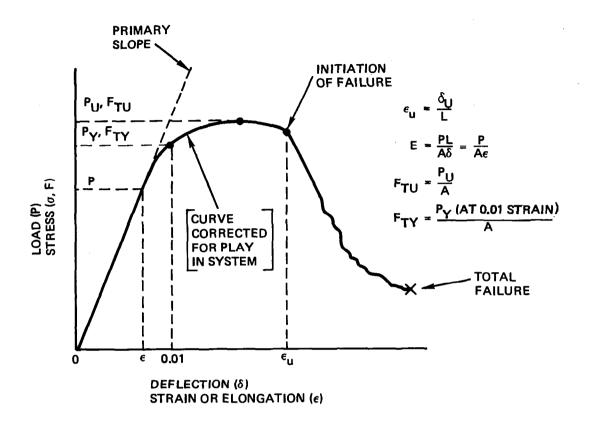
#### APPENDIX C

## DEFINITION OF TENSILE STRESS-STRAIN PROPERTIES

The raw data consisted of load-deflection charts that were automatically plotted by the Instron test machine (for ex situ baseline measurements) or by the vacuum/irradiation chamber loading mechanism (for baseline and all irradiated measurements). All raw data was taken in the English system of units (load in pounds, deflection and length in inches, area in square inches, and strain in inches/inch). Reduction of the raw data to determine key material tensile properties was initially done in English units and then converted to appropriate SI values (gigapascal for modulus of elasticity, megapascal for strength); strain (elongation) values are identical in both system of units.

The key tensile properties determined from the raw data included: modulus of elasticity, ultimate strength, yield strength, and ultimate elongation. Figure C-1 illustrates a hypothetical load-deflection diagram and defines the key relationships. Strain ( $\varepsilon$ ) is obtained by dividing the gage length (L) into the specimen change in length ( $\delta$ ). When the gage length is 2.54 cm (1 in.) – as was the case for most of the testing – specimen change-in-length is equivalent to strain. Elongation is merely strain expressed in percent. Stress ( $\sigma$ ) is obtained by dividing applied load (P) by the specimen average cross-sectional area. The ultimate stress ( $F_{TV}$ ) is the stress at maximum load. The yield stress ( $F_{TV}$ ) is defined as the stress at 0.01 strain (1 percent elongation). The ultimate strain or elongation ( $\varepsilon_{U}$ ) is determined at the initiation of failure, rather than at total separation of the specimen.

Modulus of elasticity (E) is the initial slope of the stress-strain curve. The initial portion of almost every test curve was fairly straight; thus it was fairly easy to draw a primary slope line. Modulus of elasticity was calculated from the primary slope line by dividing the applied load (P) at a given strain by the average cross-sectional area of the specimen (A) and the strain ( $\epsilon$ ) at that applied load level.



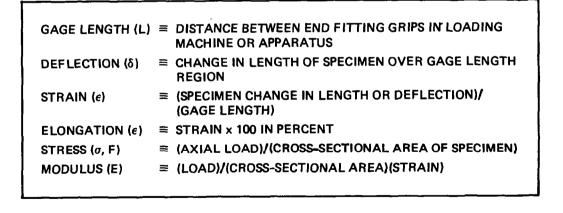


Figure C-1. Definition of Stress-Strain Parameters.

Calibration of the vacuum chamber load-deflection mechanism using a steel specimen indicated that the "play" in the system resulted in a 0.0254 mm (0.001 in.) deflection under a 444.8 N (100 lb) loading. Before calculating the key tensile properties, the deflection values were reduced a proportional amount (proportional to load) to correct for the "play" in the system (see Figure C-2).

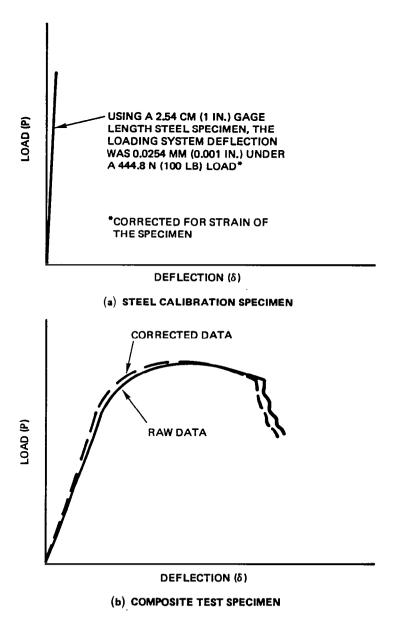


Figure C-2. Technique for Correcting Raw Data to Account for Play in Loading System.

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1. Report No.	2. Government Acces	sion No.	3. Reci	pient's Catalog No.			
NASA CR-3475		·		ort Date			
4. Title and Subtitle	DACTON ALAGORDA AT C	TN CTOU	1 .	5. Report Date November 1981			
RADIATION TESTING OF COM VERSUS EX SITU EFFECTS	PUSITE MATERIALS,	IN SITU		orming Organization Code			
7. Author(s) R.M. Kurland, J.F. Thoma	sson, and W.C. Beg	gs	3517	orming Organization Report No. 70–6028–UT–01			
9. Performing Organization Name and Addr	ess		10. Worl	k Unit No.			
TRW Defense and Space Sy	stems Group		11. Cont	tract or Grant No.			
One Space Park			İ	NAS1-15848			
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15. Supplementary Notes		<u> </u>					
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A study was conducted to investigate the effect of post-irradiation test environments on tensile properties of representative advanced composite materials (T300/5208, T300/934, C6000/P1700). Four-ply [ $\pm45/\mp45$ ] laminate tensile specimen were exposed in vacuum up to a bulk dose of 1 x $10^{10}$ rads using a mono-energetic fluence of 700 keV electrons from a Van de Graaff accelerator. Post-irradiation testing was performed while specimens were being irradiated (in situ data), in vacuum after cessation of irradiation (in vacuo data), and after exposure to a (ex situ data). Room temperature and elevated temperature effects were evaluated The radiation-induced changes to the tensile properties were small. Since the absolute changes in tensile properties were small, the existence of a post-irradiation test environment effect was indeterminate.							
17. Key Words (Suggested by Author(s))	'			18. Distribution Statement			
Mechanical properties Radiation testing techni	Radiation testing techniques			ited			
Space radiation effects		Subject Category 2					
19. Security Classif. (of this report)	20. Security Classif. (of this	page)	21. No. of Pages	22. Price			
Unclassified	Unclassified		108	A06			