# STRAIN ENERGY RELEASE RATE AS A FUNCTION OF TEMPERATURE AND PRELOADING HISTORY UTILIZING THE EDGE DELAMINATION FATIGUE TEST METHOD 

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# COMPOSITE MATERIALS RESEARCH GROUP <br> DEPARTMENT OF MECHANICAL ENGINEERING <br> UNIVERSITY OF WYOMING <br> LARAMIE, WYOMING 82071-3295 

## PREFACE

This final report presents the results of a research program, initiated in July 1986, sponsored by the National Aeronautics and Space Administration under Grant No. NAG-1-674 (University of Wyoming Project No. 5-32474). The NASA Technical Monitor was Dr. T. Kevin O'Brien, Fatigue and Fracture Branch, Materials Division, NASA-Langley Research Center.

All work was performed by the Composite Materials Research Group (CMRG) within the Department of Mechanical Engineering at the University of Wyoming. Co-principal investigators were Mr. Richard S. Zimmerman, Staff Engineer, and Dr. Donald F. Adams, Professor. Making significant contributions to this program were Mr. Michael Borgman and Ms. Janice Atkins, undergraduate students in Mechanical Engineering and members of the CMRG.

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## SECTION 1

## INTRODUCTION

The edge delamination test (EDT) is a mixed-mode fracture test method comprised of components of a tensile mode and a sliding or interlaminar shear mode of delamination (Mode $I$ and Mode II). It was initially developed by Pagano and Pipes to characterize interlaminar fracture toughness in composite laminates and provide a method to determine their relative damage tolerance and toughness [1,2]. The EDT has been further developed by O'Brien and most recently has been proposed as a standard test method by the American Society for Testing and Materials (ASTM) [3,4].

The edge delamination specimen develops interlaminar normal stresses at the free edges when loaded in tension. These stresses, caused by large Poisson's ratio mismatches between plies, cause the specimen to delaminate at the ply interface with the highest interlaminar normal stress. The delamination is controlled by a combination of Mode $I$ and Mode II. Several different laminate types have been used to perform this test, with varying percentages of Mode I and Mode II components based on the layup and material system [2,3]. O'Brien has also developed a finite element computer program to quantify the contribution of each mode in the EDT for the purpose of predicting material behavior related to damage tolerance and fracture toughness [3].

The EDT method is normally performed under static tensile loading but can be performed under tension-tension fatigue loading. Fatigue loading allows the determination of dynamic fracture toughness
properties due to the cyclic loading. The purpose of this study was to measure effects of temperature and preloading history on critical strain energy release rate $\left(G_{C}\right)$ determined using the edge delamination test method (EDT) in tension-tension fatigue. Tension-tension fatigue testing at two test temperatures was performed on IM7/8551-7 carbon fiber-reinforced epoxy composite specimens laid up in the $[ \pm 35 / 0 / 90]_{s}$ orientation. Some edge delamination specimens were fatigue tested with no preconditioning while others were preconditioned using one of two methods, i.e., a high mean load for 1000 cycles or a high spike load. Static tests to determine in-plane lamina properties and static edge delamination tests were also performed on IM7/8551-7 material.

All test specimens were fabricated and tested by the Composite Materials Research Group (CMRG) from cured panels supplied by NASA-Langley. The complete test matrix is given in Tables 1 and 2.

Static lamina testing was performed to generate in-plane material properties necessary to calculate the critical strain energy release rate $\left(G_{C}\right)$ values measured in edge delamination tests (EDT). Static EDT were conducted to allow comparisons with dynamic EDT results. Transverse coefficient of thermal expansion (CTE) tests were also performed. Tension-tension fatigue testing was performed using the EDT method with three types of preconditioning of the test specimens.

Average axial and transverse tensile, edge delamination, and in-plane Iosipescu shear results are presented in Figures 1 through 4 as a function of test temperature. Average tabulated data are given in Section 4 and individual specimen data and stress-strain plots are presented in Appendix A.

Figure 1 shows the axial tensile test results for the IM7/8551-7 carbon fiber-reinforced/epoxy material. Figures la through lc show that there were minimal differences in the longitudinal material properties between the two test temperatures. Figure 2 presents the transverse tensile material properties measured. Slightly more variation was seen in the transverse tensile material properties because of specimen configuration and material variations at the two test temperatures. These differences are discussed in detail in Section 4. Critical strain energy release rate $\left(G_{C}\right)$ calculated from the static EDT results

TABLE 1
Test Matrix for Static Tests


## STATIC TENSILE STRENGTH <br> 

a) Static Axial Tensile Strength

## STATIC TENSILE MODULUS


b) Static Axial Tensile Modulus

c) Static Axial Tensile Ultimate Strain

Figure 1. Average Static Axial Tensile Results for the IM7/8551-7 Carbon/Epoxy Unidirectional Composite as a Function of Temperature.

## static tensile streng th


a) Static Transverse Tensile Strength

## STATIC TENSILE MODULUS


b) Static Transverse Tensile Modulus

## ULTimate tensile strain


c) Static Transverse Tensile Ultimate Strain

Figure 2. Average Static Transverse Tensile Results for the IM7/8551-7 Carbon/Epoxy Unidirectional Composite as a Function of Temperature.
decreased slightly at the higher test temperature, as seen in Figure 3. In-plane Iosipescu shear properties are shown in Figure 4. Shear strength and modulus values decreased slightly at the higher test temperature. Ultimate shear strain variation with temperature could not be determined due to saturation of the strain gage rosettes at both test temperatures. Transverse CTE was measured for the IM7/8551-7 material.

Six complete fatigue curves were generated using EDT specimens. Three fatigue curves were generated at room temperature and three at $180^{\circ} \mathrm{F}\left(82^{\circ} \mathrm{C}\right)$, all at 10 Hz , using a sinusoidal waveform and a load ratio $R$ equal to 0.1. Three preconditioning load histories (described in Section 3) were used to determine their effects on $G_{C}$. A laminated plate computer program (AC3) was used to calculate the sub-laminate modulus required for the calculation of the stiffness contribution in the strain energy release rate equation [12]. Cycles to delamination were reduced significantly after specimens had been preconditioned with the 1000 cycle high mean loading. Less effect on number of cycles to delamination was measured when a high spike loading precondition was performed prior to the fatigue loading. Values of $G_{C}$ were higher at the elevated test temperature when the same precondition history fatigue curves were compared. This effect on $G_{C}$ was the reverse of that seen in the static $G_{C}$ data.

Section 3 discusses the test procedures and test apparatuses for all testing performed. Section 4 presents average test results for static and fatigue testing performed. Conclusions are given in Section 5. Individual test results are given in Appendices $A$ and $B$.

## StRain energy release rate



Figure 3. Average Static Critical Strain Energy Release Rate for IM7/8551-7 Carbon/Epoxy Unidirectional Composite as a Function of Temperature.

a) Static In-plane Iosipescu Shear Strength

SHEAR MOUULUS

b) Static In-plane Iosipescu Shear Modulus

SHEAR StRAIN

c. Static In-plane Iosipescu Shear Strain

Figure 4. Average In-plane Iosipescu Shear Results for IM7/8551-7 Carbon/Epoxy Unidirectional Composite as a Function of Temperature.

## SECTION 3

## SPECIMEN FABRICATION AND TEST PROCEDURES

### 3.1 Material

The material system chosen by NASA-Langley for this study was IM7/8551-7 carbon fiber-reinforced epoxy. It is a relatively new material system manufactured by Hercules Aerospace, Salt Lake City, Utah. The IM7 is a high strength, medium modulus carbon fiber ( 683 ksi (4.71 GPa), 41.0 Msi (283 GPa)) [4]. The $8551-7$ is a rubber-toughened epoxy with good $G_{I C}$ toughness properties (5.5 in-lb/in ${ }^{2}\left(958 \mathrm{~J} / \mathrm{m}^{2}\right)$ ) [5]. All materials used in this program were supplied by NASA-Langley in flat plate form. Six inch ( 152.4 mm ) square panels were supplied in the edge delamination test layup $\left([ \pm 35 / 0 / 90]_{S}\right)$ and three different thickness unidirectional layups $\left([0]_{8},[0]_{12}\right.$, and $\left.[0]_{20}\right)$. Enough material was supplied to complete six edge delamination fatigue curves and all required static tests.

### 3.2 Static Test Procedures

All specimens were cut from the supplied panels utilizing an abrasive cut-off wheel mounted on a surface grinder. Water cooling was used to ensure that the material did not overheat during the cutting process.

Acid digestion fiber volume determinations were performed on all panels. The procedure given in ASTM Standard Test Method D3171-76 was followed, where a $70 \%$ nitric acid was used to dissolve the matrix from the fibers using a hot plate to heat the samples to approximately $170^{\circ} \mathrm{F}$ $\left(75^{\circ} \mathrm{C}\right)$ to speed up the reaction time [6]. The $8551-7$ toughened epoxy
dissolved at a slower rate than usually seen in this procedure on previously studied epoxies.

All static tests were conducted using a computer-controlled Instron Model 1125 electromechanical testing machine. A BEMCO Model FTU 3.0 environmental chamber was used to achieve the $180^{\circ} \mathrm{F}\left(82^{\circ} \mathrm{C}\right)$ test temperature. A crosshead rate of $0.08 \mathrm{in} / \mathrm{min} \mathrm{(2} \mathrm{~mm} / \mathrm{min}$ ) was used for all static testing except the edge delamination testing, which was performed at $0.04 \mathrm{in} / \mathrm{min}(1 \mathrm{~mm} / \mathrm{min})$.

### 3.2.1 Static Tension

Guidelines in ASTM Standard Test Procedure D3039-76 were followed for all static tension testing [7]. Conventional wedge action grips were utilized to load specimens for all static tension testing. Static axial tension test specimens were 6 in. ( 152 mm ) long, 0.5 inch ( 12.7 mm) wide, and approximately 0.04 in. ( 1.0 mm ) thick. Longer specimens are normally used for axial tension (9 in. ( 229 mm )) but the supplied panels limited the specimen length to the smaller dimension. Glass/epoxy circuit board material end tabs $1.5 \mathrm{in}. \mathrm{( } 38 \mathrm{~mm}$ ) long were bonded to each specimen using a two-part epoxy adhesive. The adhesive was Techkits A-12 (Techkits, Inc., Demarest, New Jersey). It is used extensively for bonding tabbing material to composite specimens and has good material properties up to $350^{\circ} \mathrm{F}\left(177^{\circ} \mathrm{C}\right)$. Axial strains were measured using a 2 inch ( 50.8 mm ) gage length Instron extensometer, allowing complete stress-strain curves to be generated. Lateral strains were measured with a 0.5 inch ( 12.7 mm ) gage length extensometer on the axial tension test specimens, allowing for the calculation of Poisson's ratio.

Transverse tension test specimens were 6 in. ( 152 mm ) long, 0.75 in. (19 mm) wide, and 0.115 in. ( 2.9 mm ) or 0.040 in. ( 1.0 mm ) thick. Not all transverse tension specimens could be cut from the same thickness panel due to the small panel dimensions. None of the supplied panels were large enough to accommodate all the required ten transverse tension specimens. The five room temperature test specimens were cut from the 0.115 in. ( 2.9 mm ) thick $[0]_{20}$ panel while the five elevated temperature test specimens were cut from a 0.04 inch ( 1.0 mm ) thick [0]s panel. End tabs were not used with these specimens. Axial strains were measured using a 2 inch ( 50.8 mm ) gage length Instron extensometer, allowing complete stress-strain curves to be generated. Transverse strains were not measured on the transverse tension specimens.

Edge delamination test (EDT) specimens were 6 in. ( 152 mm ) long, 0.5 inch ( 12.7 mm ) wide, and approximately 0.04 in. ( 1.0 mm ) thick. This specimen geometry was used for previous edge delamination testing for NASA-Langley at the University of Wyoming [10]. Other geometries can be used for this test method as described by o'Brien and Carlsson and Pipes in references [3,14]. The smaller specimen configuration was used in this program to allow more specimens to be fabricated. Approximately 1.0 inch ( 25.4 mm ) of each end of the EDT specimens was held in the grip area, leaving 4 in. (102 mm) between the grips. Axial strains were measured using a 2 inch ( 50.8 mm ) gage length Instron extensometer, allowing complete stress-strain curves to be generated. Lateral strains were measured with a 0.5 inch ( 12.7 mm ) gage length extensometer, allowing for the calculation of Poisson's ratio for the static EDT tests.

### 3.2.2 In-Plane Iosipescu Shear

In-plane Iosipescu shear test specimens were $3 \mathrm{in} .(76.2 \mathrm{~mm})$ long, 0.75 in. ( 19.1 mm ) wide, and approximately $0.06 \mathrm{in} .(1.6 \mathrm{~mm})$ thick. Opposing $90^{\circ}$ notches were cut on each edge of the specimens to a depth of 0.15 in. ( 3.8 mm ). The notches were cut using a silicon-carbide grinding wheel dressed to the $90^{\circ}$ angle with a $0.05 \mathrm{in} .(1.3 \mathrm{~mm})$ radius at the bottom of the notch.

Loads were applied to all in-plane shear specimens using a Wyoming Iosipescu shear test fixture. Shear strains were measured with a Measurements Group No. EP-13-062TH-120 two-element rosette strain gage mounted between the notches of the specimens. Complete shear stressshear strain curves were generated for all Iosipescu shear specimens.

### 3.2.3 Transverse Coefficient of Thermal Expansion (CTE)

The transverse CTE test specimens were 5 in. ( 127 mm ) long, 0.375 in. (9.5 mm) wide, and 0.115 in. ( 2.9 mm ) thick. No axial coefficient of thermal expansion specimens were tested in this program.

Transverse CTE tests were performed using a microprocessorcontrolled quartz-tube dilatometer in conjunction with a linear-variable differential transformer (LVDT). The specimens were exposed to thermal excursions between $-40^{\circ} \mathrm{F}\left(-40^{\circ} \mathrm{C}\right)$ and $250^{\circ} \mathrm{F}\left(120^{\circ} \mathrm{C}\right)$. Data were acquired on the heat-up portion of the thermal cycles only. Two thermal cycles each were completed on the three specimens tested.

### 3.3 Tension-Tension Edge Delamination Fatigue Procedure

All fatigue tests were performed on an Instron Model 1321 biaxial servo-hydraulic testing machine using a 10 Hz sinusoidal excitation
waveform and a load ratio $R$ (ratio of minimum to maximum applied cyclic loading) approximately equal to 0.1. Axial strains were measured using a 2 inch ( 50.8 mm ) gage length Instron extensometer. The extensometer knife edges were bonded to the EDT specimens using Devcon 5-minute two-part epoxy to ensure that the measured strains were not affected by slippage of the knife edges on the specimens' surface during the fatigue test.

Model 647.02S hydraulic grips manufactured by mTS, Inc., Minneapolis, Minnesota, were used to grip specimens for all edge delamination fatigue testing. The grips were equipped with special high temperature seals and actuating fluid system to allow usage up to $350^{\circ} \mathrm{F}$ $\left(177^{\circ} \mathrm{C}\right)$. They have a load capacity of 5500 pounds ( 25 kN ), which was sufficient to perform all EDT fatigue testing.

An Applied Test Systems, Inc. Series 2911 environmental chamber was used to achieve the required $180^{\circ} \mathrm{F}\left(82^{\circ} \mathrm{C}\right)$ test temperature. Chamber temperature was measured using a Type "E", Chromel-Constantan, thermocouple placed near the test specimen. Figure 5 shows the tension-tension fatigue test configuration used with the environmental chamber in place.

Six complete fatigue curves were generated using the EDT method while monitoring the dynamic modulus of the specimens. Three fatigue curves were generated at room temperature and three at elevated temperature, $180^{\circ} \mathrm{F}\left(82^{\circ} \mathrm{C}\right)$. The three fatigue curves at each temperature were generated using different preloading schemes prior to the normal sinusoidal fatigue test. One curve at each test temperature consisted of a normal sinusoidal loading to specimen delamination with no preload history (normal baseline curve). One curve at room temperature was

ORIGINAL PAGE IS OF POOR QUALITY

a) Overall View of EDT Fatigue Test Setup

b) Close-up View of EDT Fatigue Test Setup

Figure 5. Edge Delamination Fatigue Test Configuration Showing Extensometer Mounted on EDT Specimen, Hydraulic Grips, Environmental Chamber, Instron Model 1321 Testing Machine, and Controls Cabinet.
generated with normal sinusoidal loading to delamination after the specimens had been subjected to a high cyclic load to 70 percent of ultimate load for 1000 cycles (high mean load curve). The elevated temperature fatigue curve with a high mean load for 1000 cycles was performed to only 60 percent of ultimate load because the EDT specimens delaminated before the 1000 cycles were completed at 70 percent of ultimate load. This precluded the possibility for further fatigue loading. The third fatigue curves at both test temperatures were generated after the EDT specimens had been subjected to a maximum load of 70 percent of ultimate load for one cycle (spike load curve).

The fatigue test procedure for all testing was quite complicated and required a great deal of time and effort to perfect. The application computer programs necessary to calibrate and perform the testing were written by the CMRG in Fortran 77 computer language on a PDP $11 / 24$ minicomputer. The application programs interfaced with an Instron Machine Driver (IMD) to perform the required subroutine calls to control the test machine and to acquire the data during each fatigue test. The Instron Machine Driver is written in the Macro-11 computer language and is supported by Instron Corporation, Canton, Massachusetts. It consists of numerous machine control subroutines and data acquisition subroutines interfaced with the Instron testing machine hardware. Data recorded during each fatigue test were load, strain, stroke, and cycle count.

The application computer programs also performed many calculations and decisions based on the status of the fatigue test, such as to continue with the test if the specimen was intact, or stop the test if the specimen had delaminated causing a loss of stiffness. The decision
to continue or suspend testing was based on the dynamic modulus calculated during the test. The dynamic modulus was calculated using a linear regression curve-fit technique. The lowest stress level during a cycle ( $\sigma_{\min }$ ) was found and the slope of the stress-strain curve was calculated over the next 20 data stress-strain pairs. If the calculated dynamic modulus had decreased by 5 percent or more from the initial calculated modulus, signifying delamination, then the computer would stop the fatigue test. Storage of data was accomplished using a logarithmic scheme similar to that used in previous modulus decay fatigue testing for NASA-Langley $[8,9]$. Table 5 gives the data storage progression used in this program.

There were five major steps required to prepare for and perform an EDT fatigue test. Step 1 was to calibrate the load, strain, and stroke

## Table 5

## DATA STORAGE PROGRESSION

Cycle Range
Cycle Increment Between Disk Storage
1 to $100 \quad 1$
101 to 100010
1001 to 10,000100
10,001 to $100,000 \quad 1,000$
100,101 to $1,000,00010,000$
$1,000,001$ to $10,000,000100,000$
transducers to the appropriate values required. This step was performed periodically during the testing to verify that calibration values had not changed for the three transducers and their respective amplifiers.

Step 2 was to calibrate the computer with the three transducer outputs. This step was performed to ensure the computer had stored the current transducer calibrations before each fatigue test.

Step 3 was to perform a preliminary fatigue test using a square wave excitation on a specimen with similar stiffness to the actual test specimen. A square wave excitation was used to set the load transducer gain level for optimum control and response of the Instron test machine at each load level as described in the Instron Machine Operation Manual [11]. Step 3 was critical because of the limited load control response of the Instron 1321 at the 10 Hz cycling frequency. After setting the gain level to the optimum level, the computer program calculated the appropriate amount of computer control overprogramming necessary to ensure the loads transmitted to the specimen were close to the desired values. The mass of the torsional actuator below the linear actuator was very detrimental to the performance of the Instron Model 1321 test machine at the 10 Hz cycling frequency, but was fully compensated for by this load command overprogramming in the application computer program. The stiffness of the composite material specimens being tested was low enough and the inertia of the load frame actuator high enough to require a command overprogramming of 10 to 20 percent for all fatigue tests.

Step 4 was performed on only those specimens subjected to a pretest load history. The high mean load EDT specimens were tension-tension fatigue tested for 1000 cycles at a maximum peak load equal to 70 percent of ultimate load in the room temperature case, and 60 percent of
ultimate load in the $180^{\circ} \mathrm{F}\left(82^{\circ} \mathrm{C}\right)$ case. The lower peak load precondition used for the elevated temperature specimens was necessary to prevent the delamination of the specimens prior to completing the full 1000 cycles. The spike load EDT specimens were cycled once to a peak load of 70 percent of ultimate prior to being subjected to the normal sinusoidal loading. It was not necessary to reduce the spike peak load level for the elevated temperature specimens because they did not delaminate during the one cycle at the 70 percent peak load level.

Performance of the actual tension-tension fatigue test on the edge delamination specimen at a particular stress level was Step 5. This step included performing the fatigue test between two predetermined stress values until delamination occurred and the specimen calculated dynamic modulus had decreased by at least 5 percent from the initial calculated value. Figure 6 shows a typical modulus decay curve. All modulus decay curves are included in Appendix $B$. When the dynamic modulus decreased to less than 95 percent of the original calculated value, the computer suspended the fatigue test and ramped to zero load. As Figure 6 shows, the drop in modulus was easily determined from the data. The critical strain value $\left(\varepsilon_{C}\right)$ used in the $G_{C}$ calculation was determined by looking at these curves and picking the location of the onset of delamination as the point where the modulus started to decrease to the minimum value for that test. At least one specimen for each fatigue curve was tested to $10^{7}$ cycles to investigate material behavior at longer than normal fatigue test runout (typically $10^{6}$ cycles). Optical inspection of test specimens, after the computer had suspended the tests, always revealed they had delaminated on one or both free edges.


Specimen data files were then transferred from the PDP11/24 to a VAX $11 / 750$ minicomputer for reduction and plotting after test completion. Complete test results are presented in Section 4.

### 3.4 Dye-Enhanced X-Ray and Optical Photography

Dye-enhanced x-ray and optical photographs were taken of some of the specimens, to document the extent of the delaminations. A PANTEK Model HF75 Industrial X-ray unit was used to take all radiographs. This unit was designed especially for low density materials such as composites. The dye was the same used for previous work for NASA-Langley [10]. The mixture formula used is as follows:

```
60 gm ZnI2 (Zinc Iodide)
10 ml Isopropyl Alcohol
10 ml Deionized Water
    1 \mathrm { ml } \text { Kodak "Photo-Flo", Wetting Agent}
```

Polaroid Type 55 sheet film was used for all x-ray photographs. Settings on the $x$-ray controls were adjusted to result in good contrast for better interpretation of the photographs. The control settings were 17 kilovolts ( kV ), 12 milliamps ( mA ), 20 seconds exposure time, and 17 inch (432 mm) film focal distance (FFD). Two specimens from each EDT fatigue curve were radiographed. These radiographs are presented in Section 4.

### 3.5 Temperature and Relative Humidity Measurement

Temperature and relative humidity measurements were recorded during most of the fatigue testing phase in the test laboratory using a Cole-Parmer Model 8368-50 hygrothermograph. Charts were changed periodically during the testing phase. Relative humidity ranged from 12 to 20 percent while laboratory temperature ranged from $68^{\circ}$ to $86^{\circ} \mathrm{F}\left(20^{\circ}\right.$ to $30^{\circ} \mathrm{C}$ ).

## SECTION 4

## TEST RESULTS

### 4.1 Fiber and Void Volume Results

All IM7/8551-7 panels supplied by NASA-Langley had a fiber volume of approximately 60 percent. The $[0]_{8}$ and $[ \pm 35 / 0 / 90]_{s}$ panels had slightly higher fiber volumes than the $[0]_{12}$ and $[0]_{20}$ panels (62 percent versus 59 percent). Void volumes were typically less than 1 percent except for the EDT panels where 1.2 to 4.7 percent voids were measured. Individual tabulated fiber and void volume data are given in Appendix A.

### 4.2 Static Tension Test Results

All average static test results are given in Table 4. Individual static test results are given in Appendix A. Static tensile test results for the two test temperatures indicate the moderate elevated temperature, i.e., $180^{\circ} \mathrm{F}\left(82^{\circ} \mathrm{C}\right)$, had some effect on the material behavior.

A slight decrease in axial tensile strength, and axial Young's modulus, and an increase in ultimate axial tensile strain and major Poisson's ratio were measured at the elevated test temperature, possibly due to the higher test temperature causing slight softening of the matrix material.

Transverse tensile test results indicate that the IM7/8551-7 material had lower strength and stiffness properties at the elevated test temperature. Because the specimens tested at the two different
table 4 Average Static Test Results for IM7/8551-7
Carbon Fiber-Reinforced Epoxy Composite Material

| Test Method | Test Temperature $\left({ }^{\circ} \mathrm{F}\right)$ | Ultimate Strength |  | Modulus |  | Ultimate Strain (percent) | Poisson's Ratio | Stress at Delamination (ksi) (MPa) |  | Strain at Delamination (percent) | Strain Release $\frac{(\mathrm{in}-\mathrm{lb})}{\left(\mathrm{in}^{2}\right)}$ | Energy Rate $\frac{(\mathrm{J})}{\left(\mathrm{m}^{2}\right)}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| Axial Tension | 75 | 363 | 2505 | 23.3 | 160.7 | 1.19 | 0.18 |  |  |  |  |  |
|  | 180 | 359 | 2475 | 21.8 | 150.3 | 1.54 | 0.36 |  |  |  |  |  |
| Transverse Tension | e 75 | 7.3 | 51 | 1.20 | 8.2 | 0.62 |  |  |  |  |  |  |
|  | 180 | 5.9 | 41 | 1.08 | 7.4 | 0.56 |  |  |  |  |  |  |
| In-Plane Iosipescu Shear | 75 | 15.0 | 104 | 0.83 | 5.7 | >11.50 |  |  |  |  |  |  |
|  | 180 | 12.7 | 88 | 0.75 | 5.2 | >11.30 |  |  |  |  |  |  |
| Edge Delam ination [ $\pm 35 / 0 / 90$ ] | $\text { m- } \quad 75$ | 106.6 | 735 | 9.85 | 67.9 | 1.16 | 0.20 | 81 | 559 | 0.85 | 2.84 | 497 |
|  | $]_{s} 180$ | 109.4 | 754 | 9.99 | 68.9 | 1.45 | 0.45 | 74 | 509 | 0.76 | 2.32 | 406 |

temperatures were cut from different panels and were of different thicknesses, an independent temperature effect could not be determined. The overall effect of specimen configuration and test temperature was seen as a decrease in transverse tensile strength and modulus and ultimate transverse tensile strain values at the higher test temperature.

Static EDT results indicate that the IM7/8551-7 material was almost 20 percent tougher at room temperature than at the elevated test temperature. Critical strain energy release rate ( $\mathrm{G}_{\mathrm{C}}$ ) values were slightly lower at the elevated temperature due to the lower measured critical strain at delamination ( $\varepsilon_{c}$ ) and the lower stiffness values for the laminates. Equation (1) was used to calculate $G_{C}$ :

$$
\begin{equation*}
\left.G_{C}=\frac{1}{2} \varepsilon_{C}^{2} t(E)-E^{*}\right) \tag{1}
\end{equation*}
$$

where $G_{C}=$ critical strain energy release rate
$\varepsilon_{c}=$ axial strain at delamination onset
t = laminate thickness
$\mathrm{E}_{\text {lam }}=$ initial laminate modulus
$E^{*}=$ laminate modulus if completely delaminated along one or more interfaces

Strain at delamination onset ( $\varepsilon_{C}$ ) was determined from the EDT stressstrain curves at the point where the curve began to deviate from linear behavior. Elam was determined by calculating the initial tangent modulus for the static EDT specimens. Values for $\varepsilon_{C}$ and $E_{l a m}$ are given in Table 4. The delaminated modulus ( $E^{*}$ ) was calculated using Equation (2):

$$
\begin{equation*}
\mathrm{E}^{\star}=\frac{8 \mathrm{E}_{[ \pm 35 / 0] s}+2 \mathrm{E}_{22}}{8} \tag{2}
\end{equation*}
$$

No $[ \pm 35 / 0]_{s}$ sub-laminate material was supplied to measure the stiffness values required to calculate the $G_{C}$ values. Laminated Plate Theory Program AC3 was used to calculate the stiffness values for the sub-laminate used in the $E^{*}$ calculations [12]. The $E_{22}$ value was determined by calculating the initial tangent modulus from the transverse tensile tests. Appropriate values for each test temperature were used in all calculations.

### 4.3 In-Plane Iosipescu Shear Test Results

Average in-plane shear test results are presented in Table 4. Individual tabulated results and plotted results are given in Appendix A. Significant differences were seen in shear test results at the two test temperatures. Shear strength decreased by 25 percent and shear modulus decreased by 10 percent at the higher test temperature. Ultimate shear strains could not be determined due to saturation of the strain gage rosettes used to measure shear strain on the Iosipescu shear test specimens. Shear strains measured were quite nonlinear for this material system at both test temperatures.

### 4.4 Transverse Coefficient of Thermal Expansion (CTE) Results

Average transverse CTE results are presented here. Individual tabulated and plotted transverse CTE results are presented in Appendix A. The average transverse CTE value for the IM7/8551-7 material was $19.4 \mu \varepsilon /{ }^{\circ} \mathrm{F}\left(34.8 \mu \varepsilon /{ }^{\circ} \mathrm{C}\right)$. This value is quite typical for most carbon fiber/epoxy material systems. No unusual behavior was seen
in the thermal expansion testing results. No axial thermal expansion testing was conducted in this program due to the dilatometer equipment not being able to adequately measure the extremely small displacements in the axial direction for carbon fiber/epoxy composites.

### 4.5 Tension-Tension Edge Delamination Fatigue Results

Six complete fatigue curves were generated using the EDT method subjected to sinusoidal excitation loading at $10 \mathrm{~Hz}, \mathrm{R}=0.1$, at various stress levels. Three of the fatigue curves were generated at room temperature and three at elevated temperature, $180^{\circ} \mathrm{F}\left(82^{\circ} \mathrm{C}\right)$. Three different specimen preconditioning methods were used to prepare the EDT test specimens for cyclic fatigue loading. One curve at each test temperature was generated using a normal sinusoidal loading to specimen delamination with no pretest load history (baseline curves). One curve at room temperature was generated with normal sinusoidal loading after the specimen had been subjected to a high cyclic load to 70 percent of ultimate load for 1000 cycles (high mean load curves). The elevated temperature precondition, with the high mean load for 1000 cycles, was performed to only 60 percent of ultimate load. The EDT specimens delaminated before the 1000 cycles were completed at 70 percent, precluding the need for further fatigue loading. The third fatigue curve at both test temperatures was generated after the EDT specimens had been subjected to 70 percent of ultimate load for one cycle only (spike load).

Figures 7 through 9 illustrate the temperature effect on $G_{C}$ for each of the three load preconditions. Figure 7 shows the two normal fatigue curves generated at different temperatures. Figures 8 and 9

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Figure 8. Critical Strain Energy Release Rate for IM7/8551-7 [ $+35 / 0 / 90$ ]s Laminates with High
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Figure 9. Critical Strain Energy Release Rate for IM7/8551-7 [ $+35 / 0 / 90$ ] Laminates with High Mean Load for 1000 Cycles Precondition as a Function of Temperature.
show the high spike and high mean load precondition fatigue curves at the two test temperatures, respectively. The elevated temperature curve in all three of these figures indicates that higher dynamic $G_{C}$ values were calculated independent of precondition method, for comparable cycles to delamination.

Comparisons of room temperature fatigue $G_{C}$ values at low numbers of cycles with static $G_{C}$ data (Table 4) indicate that the dynamic $G_{C}$ values are much lower (a factor of two) than those generated under static loading at room temperature. Elevated temperature dynamic $G_{C}$ values at a low number of cycles to delamination indicate much more consistent values compared with elevated temperature static $G_{C}$ values.

The normal, high spike load, and high mean load curves at the room test temperature and elevated test temperature are plotted in Figures 10 and 11, respectively. Figure 10 indicates that there are significant effects due to the preloading history at room temperature. The two precondition loading curves indicate that damage incurred by the specimen is worse for the high mean load precondition than for the high spike precondition. At room temperature, nearly three decades fewer cycles were seen in the high mean precondition curve than the normal curve data. The spike precondition curve at room temperature indicates that approximately two decades fewer cycles to delamination were measured compared to the normal fatigue curve. Precondition comparisons in Figure 11 show that the effects were reduced significantly at the higher test temperature. Little difference could be seen between the three fatigue curves at the $180^{\circ} \mathrm{F}\left(82^{\circ} \mathrm{C}\right)$ test temperature after passing 10-100 cycles.

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### 4.6 X-ray and Optical Photograph Results

Dye-penetrant enhanced x-rays and optical photographs were taken of some of the failed EDT specimens to document the extent of the delaminations. The x-ray opaque dye was composed of the materials listed in Section 3.5 at the given mixing ratios. Figures 12 through 17 are dye-penetrant enhanced x-rays of two specimens from each fatigue curve. At least two specimens from each fatigue curve were x-rayed to see if there were any differences in delamination zones due to temperature or the load history variations used for the fatigue testing. One specimen with less than 100 cycles and one with more than 10,000 cycles were x-rayed from each fatigue curve. The delamination zones appear as darkened areas in the radiographs. Number of cycles to delamination, temperature, and load precondition did not seem to affect the delamination zone size or appearance. Darkened areas at the points of extensometer attachment were due to the glue lines absorbing the dye and are not caused by actual damage to the specimens.

Optical photographs were taken of specimen edges to document the delamination crack at the free edge. Figures 18 and 19 are optical photographs of two specimen edges showing typical edge delamination cracks. The delamination occurred at the $0 / 90$ interfaces in all cases, and wandered back and forth between the $0^{\circ}$ plies within the $90^{\circ}$ plies along the length of the specimens. This behavior was as expected for the $[ \pm 35 / 0 / 90]_{s}$ layup used in this program. No variations in delaminations were seen between the different test temperatures or preloading history specimens to indicate any visible effect on the delamination crack path.

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Figure 12. Dye-enhanced X-ray Photograph of Two EDT Fatigue Failed Specimens at Room Temperature, No Precondition.

Top: Specimen No. LFLESA - 36 Cycles to Delamination. Bottom: Specimen No. LFLLSD - 950 Cycles to Delamination.

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Figure 13. Dye-enhanced X-ray Photograph of Two EDT Fatigue Failed Specimens at Room Temperature, High Spike Load Precondition.

Top: Specimen No. LFLDPA - 1 Cycle to Delamination. Bottom: Specimen No. LFLNPB - 1000 Cycles to Delamination.

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Figure 14. Dye-enhanced X-ray Photograph of Two EDT Fatigue Failed Specimens at Room Temperature, High Mean Load Precondition.

Top: Specimen No. LFLDMB - 66 Cycles to Delamination. Bottom: Specimen No. LFLLMA - 3300 Cycles to Delamination.

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Figure 15. Dye-enhanced X-ray Photograph of Two EDT Fatigue Failed Specimens at $180^{\circ} \mathrm{F}\left(82^{\circ} \mathrm{C}\right)$, No Precondition.

Top: Specimen No. LFHESA - 51 Cycles to Delamination. Bottom: Specimen No. LFHMSB - 210,000 Cycles to Delamination.


Figure 16. Dye-enhanced X-ray Photograph of Two EDT Fatigue Failed Specimens at $180^{\circ} \mathrm{F}\left(82^{\circ} \mathrm{C}\right)$, High Spike Load Precondition.

Top: Specimen No. LFHEPA - 1 Cycle to Delamination. Bottom: Specimen No. LFHMPA - 70,000 Cycles to Delamination.

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Figure 17. Dye-enhanced X-ray Photograph of Two EDT Fatigue Failed Specimens at $180^{\circ} \mathrm{F}\left(82^{\circ} \mathrm{C}\right)$, High Mean Load Precondition.

Top: Specimen No. LFHFMA - 1 Cycle to Delamination. Bottom: Specimen No. LFHMMA - 890,000 Cycles to Delamination.

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Figure 18. Optical Photograph of Failed EDT Fatigue Specimen Showing a
Crack at the Free Edge in the $90^{\circ}$ Plies.

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Figure 19. Optical Photograph of Failed EDT Fatigue Specimen Showing a Crack on the Free Edge in the $90^{\circ}$ Plies.

## SECTION 5

## CONCLUSIONS

Static lamina, static edge delamination, and tension-tension fatigue edge delamination testing was performed on IM7/8551-7 carbon fiber-reinforced/epoxy. Static lamina test results indicated some reduction in material properties from the elevated test temperature on this material system. Matrix-dependent material properties (shear, transverse tension, and EDT) measured at the elevated test temperature were typically lower than those measured at room temperature. Higher test temperatures could be used for future testing to identify an upper use temperature for this material system. The IM7/8551-7 performed well at the $180^{\circ} \mathrm{F}\left(82^{\circ} \mathrm{C}\right)$ test temperature. Static test results indicated that the $G_{C}$ of this material system was quite good compared to other material systems previously tested $[5,6]$. Results from this test program should allow the finite element analysis used by o'Brien to predict the contributions of $G_{I c}$ and $G_{I I c}$ to $G_{C}$ for this material [3].

At room temperature, EDT fatigue results indicated lower $G_{C}$ values (half of static values) were measured compared to static $G_{C}$ values. All three preconditions resulted in lower $G_{C}$ at room temperature. EDT fatigue testing at the $180^{\circ} \mathrm{F}\left(82^{\circ} \mathrm{C}\right)$ test temperature indicated that comparable values of $G_{C}$ were measured compared to static EDT values.

Effects on $G_{C}$ at elevated temperature due to preconditioning test specimens were minimal. Only at the high stress level/low number of cycles portion of the fatigue curves was there much variation in $G_{C}$ between the different load preconditioning data. The high mean load and high spike load curves resulted in $G_{C}$ values lower than the no
precondition curve values up to about 100 cycles. Elevated temperature data from the three precondition curves approached the same $G_{C}$ value at runout ( $10^{7}$ cycles).

The room temperature $G_{C}$ values at equal number of cycles from the three precondition curves were quite different after 10,000 cycles. The three room temperature curves showed a strong influence of the precondition method used with the high mean load having the lowest $G_{C}$ values and the no precondition $G_{C}$ values being the highest values at greater than 10,000 cycles. Number of cycles to delamination was reduced by three to four decades compared with the no precondition case after specimens had been preconditioned at room temperature by the high mean level method. Two to three fewer decades of cycles to delamination were measured after preconditioning with the single high spike loading compared with the no precondition case.

Optical photographs showed that the delamination crack on the free edge of specimens did not vary due to test temperature or precondition. Dye-enhanced radiographs taken of delaminated specimens showed that no discernible differences could be attributed to the two different test temperatures or three preconditions.

## REFERENCES

1. R.B. Pipes and N.J. Pagano, "Interlaminar Stresses in Composite Laminates under Uniform Axial Extension," Journal of Composite Materials, Vol. 4, 1970, pp. 538-548.
2. N.J. Pagano and R.B. Pipes, "Some Observations on the Interlaminar Strength of Composite Laminates," International Journal of Mechanical Sciences, Vol. 15, 1973, pp. 679-688.
3. T.K. O'Brien, "Characterization of Delamination Onset and Growth in a Composite Laminate," Damage in Composite Materials, ASTM STP 775 , American Society for Testing and Materials, Philadelphia, Pennsylvania, 1982, pp. 140-167.
4. T.K. O'Brien and N.J. Johnston, "A Simple Test for the Interlaminar Fracture Toughness of Graphite/Epoxy Laminates," Proceedings of the 27th National SAMPE Symposium, May 1982, Reno, Nevada, p. 401-415.
5. "Magnamite Graphite Fiber, Type IM7," Data Sheet 868, Hercules, Inc., Wilmington, Delaware, 1988.
6. "Hercules Prepreg Tape Materials Characterization Data Package," Hercules Composite Products Group, Magna, Utah, November 1988.
7. "Standard Test Method for Fiber Content of Resin-Matrix Composites by Matrix Digestion," ASTM Standard Test Method D-3171-76 (Reapproved 1982), 1988 ASTM Annual Book of Standards, Section 15, Volume 15.03, American Society for Testing and Materials, Philadelphia, Pennsylvania, 1988, pp. 122-124.
8. "Standard Test Method for Tensile Properties of Fiber-Reinforced Composites," ASTM Standard Test Method D-3039-76 (Reapproved 1982), 1988 ASTM Annual Book of Standards, Section 15, Volume 15.03, American Society for Testing and Materials, Philadelphia, Pennsylvania, 1988, pp. 117-121.
9. E.M. Odom and D.F. Adams, "Stiffness Reductions During Tensile Fatigue Testing of Graphite/Epoxy Angle-Ply Laminates," NASA Contractor Report 166019, National Aeronautics and Space Administration, Washington, D.C. , November 1982.
10. R.S. Zimmerman, D.F. Adams, and E.M. Odom, "Load Ratio and Frequency Effects on Strain Energy Release Rate During Tensile Fatigue Testing Utilizing the Edge Delamination Test," Report UWME-DR-401-109-1, Department of Mechanical Engineering, University of Wyoming, December 1984.
11. Operation Manual for 2150 Controllers, Manual No. 11-1-12, Instron Corporation, Canton, Massachusetts, 1977, pp. 2.4-2.6.
12. Standard Tests for Toughened Resin Composites, NASA Reference Publication 1092, Revised Edition, Compiled by ACEE Composites Project Office, NASA-Langley Research Center, Hampton, Virginia, 1983, pp. 7-14.
13. Advanced Composites Design Guide, Vol. II-Analysis, Air Force Flight Dynamics Laboratory, Dayton, Ohio, January 1973, pp. 2.B.1.1-2.B.1.29.
14. L.A. Carlsson, and R.B. Pipes, Experimental Characterization of Advanced Composite Materials, Prentice-Hall, Inc., Englewood Cliffs, New Jersey, 1987.

## APPENDIX A

INDIVIDUAL STATIC TEST RESULTS

Table A1

## Individual Fiber and Void Volume Results for IM7/8551-7 Composites

$\left.\begin{array}{lcc}\text { Plate Layup } & \begin{array}{c}\text { Fiber Volume } \\ \left(\mathrm{V}_{\mathrm{f}}\right) \\ (\text { Percent })\end{array} & \begin{array}{c}\text { Void Volume } \\ \left(\mathrm{V}_{\mathrm{v}}\right)\end{array} \\ \text { (Percent) }\end{array}\right)$

Table A2

Individual Static Axial Tensile Results for IM7/8551-7 Composites

| Specimen Name | Test Temperature$\left({ }^{\circ} \mathrm{F}\right)$$\qquad$ | Tensile Strength |  | Tensile Modulus |  | Ultimate Strain (Percent) | $\begin{gathered} \text { Poisson's } \\ \text { Ratio } \end{gathered}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  | (ksi) | (GPa) | (Msi) | (GPa) |  |  |
| NLOTD2 | 75 | 372 | 2.56 | 23.9 | 165 | 1.5 | 0.168 |
| 3 |  | 383 | 2.64 | 23.8 | 164 | 0.6* | 0.186 |
| 4 |  | 341 | 2.35 | 21.7 | 150 | 1.9 | 0.174 |
| 5 |  | 231* | 1.59* | 24.7 | 170 | 0.8* | 0.200 |
| 6 |  | 358 | $\underline{2.47}$ | 22.3 | 154 | 1.3 | 0.189 |
| Average |  | 364 | 2.51 | 23.3 | 161 | 1.5 | 0.183 |
| Std. Dev. |  | 18 | 0.12 | 1.2 | 8 | 0.3 | 0.013 |
| NLFSH1 | 180 | 321 | 2.21 | 22.1 | 152 | 1.2* | 0.419* |
| 2 |  | 364 | 2.51 | 21.4 | 148 | 1.6 | 0.404 |
| 3 |  | 367 | 2.53 | 21.3 | 147 | 1.5 | 0.345 |
| 4 |  | 369 | 2.54 | 21.8 | 150 | 1.5 | 0.345 |
| 5 |  | $\underline{372}$ | $\underline{2.56}$ | 22.5 | 155 | 1.5 | 0.337 |
| Average |  | 359 | 2.47 | 21.8 | 150 | 1.5 | 0.358 |
| Std. Dev. |  | 21 | 0.15 | 0.5 | 3 | 0.1 | 0.031 |

*Not included in Average or Standard Deviation

Individual Static Transverse Tensile Results for IM7/8551-7 Composites

| Specimen Name | Test Temperature$\left({ }^{\circ} \mathrm{F}\right)$ | Tensile <br> Strength |  | Tensile <br> Modulus |  | Ultimate Strain (Percent) |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  | (ksi) | (GPa) | (Msi) | (GPa) |  |
| NLTTDO | 75 | 6.84 | 47.8 | 1.19 | 8.2 | 0.58 |
| 1 |  | 7.56 | 52.1 | 1.15 | 7.9 | 0.67 |
| 2 |  | 7.61 | 52.5 | 1.21 | 8.3 | 0.64 |
| 3 |  | 2.80* | 19.3* | 1.15 | 7.9 | 0.22* |
| 4 |  | 7.40 | 51.0 | 1.21 | 8.3 | 0.62 |
| 5 |  | 7.30 | 50.3 | 1.27 | 8.8 | 0.59 |
| Average |  | 7.34 | 50.7 | 1.20 | 8.2 | 0.62 |
| Std. Dev. |  | 0.03 | 1.86 | 0.05 | 0.3 | 0.04 |
| NTTD21 | 180 | 3.84* | 26.5* | 1.07 | 7.4 | 0.37* |
| 22 |  | 4.72 | 32.5 | 1.05 | 7.2 | 0.46 |
| 23 |  | 5.05 | 34.8 | 1.05 | 7.2 | 0.53 |
| 24 |  | 6.20 | 42.7 | 1.08 | 7.4 | 0.60 |
| 25 |  | 6.50 | 44.8 | 1.11 | 7.6 | 0.62 |
| 26 |  | 6.63 | 45.7 | 1.09 | 7.5 | 0.65* |
| 27 |  | 6.19 | 42.7 | 1.09 | 7.5 | 0.60 |
| Average |  | 5.88 | 40.5 | 1.08 | 7.4 | 0.56 |
| Std. Dev. |  | 0.80 | 5.5 | 0.02 | 0.2 | 0.07 |

*Not included in Average and Standard Deviation

Table A4

Individual In-Plane Iosipescu Shear Results for IM7/8551-7 Composites

| Specimen Name | Test Temperature$\left({ }^{\circ} \mathrm{F}\right)$ | Shear Strength |  | Shear <br> Modulus |  | Ultimate Strain (Percent) |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  | (ksi) | (GPa) | (Msi) | (GPa) |  |
| NFIRD1 | 75 | 16.1 | 111 | 0.86 | 6.0 | 11.9 |
| 2 |  | 14.9 | 103 | 0.87 | 6.0 | 11.9 |
| 3 |  | 15.8 | 109 | 0.83 | 5.7 | 11.9 |
| 4 |  | 15.5 | 107 | 0.83 | 5.7 | 10.3 |
| 5 |  | 12.8 | 88 | 0.76 | 5.2 | 5.4* |
| Average |  | 15.0 | 104 | 0.83 | 5.7 | 11.5 |
| Std. Dev. |  | 1.3 | 9 | 0.05 | 0.3 | 0.8 |
| NFIHD1 | 180 | 12.4 | 85 | 0.81 | 5.6 | 11.4 |
| 2 |  | 10.1* | 70* | 0.77 | 5.3 | 5.9* |
| 3 |  | 13.3 | 91 | 0.76 | 5.3 | 11.3 |
| 4 |  | 13.1 | 90 | 0.78 | 5.4 | 11.3 |
| 5 |  | 12.3 | 85 | 0.62 | 4.2 | 11.3 |
| Average |  | 12.7 | 88 | 0.75 | 5.2 | 11.3 |
| Std. Dev. |  | 0.5 | 3 | 0.08 | 0.6 | 0.1 |

*Not included in Average or Standard Deviation
Table A5
Individual Static Edge Delamination Test Results
for $\operatorname{m7} / 8551-7$ Composites

| $\begin{aligned} & \text { Specimen } \\ & \text { Name } \end{aligned}$ | Test Temperature ( ${ }^{\circ} \mathrm{F}$ ) | Thickness(in) | Ultimate Strength |  | Modulus Strength |  | Strength Delamination |  | Critical Strain Energy Release Rate |  | Strain at Delamination <br> (percent) | $\begin{aligned} & \text { Poisson's } \\ & \text { Ratio } \end{aligned}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  | (ksi) | ( MPa ) | (Msi) | ( MPa ) | (ksi) | (MPa) | ( in-lbf/in ${ }^{2}$ ) | $\left(\mathrm{J} / \mathrm{m}^{2}\right)$ |  |  |
| NLDTD1 | 75 | 0.042 | 108 | 745 | 9.0 | 62.0 | 93* | 641* | 3.95* | 692* | $0.99 *$ | 0.196 |
| 2 |  | 0.043 | 99 | 683 | 9.6 | 66.2 | 93* | 641* | 3.85* | 675* | 0.97 | $0.138 *$ |
| 3 |  | 0.044 | 99 | 683 | 9.7 | 66.9 | 79 | 545 | 2.62 | 459 | 0.79 | 0.168 |
| 4 |  | 0.044 | 120 | 827 | 10.1 | 69.6 | 92 | 634 | 3.59 | 628 | 0.93 | 0.211 |
| 5 |  | 0.044 | 94 | 648 | 10.2 | 70.3 | 81 | 558 | 2.91 | 509 | 0.83 | 0.183 |
| 6 |  | 0.042 | 115 | 793 | 9.6 | 66.2 | 75 | 517 | 2.37 | 416 | 0.77 | 0.258* |
| 7 |  | 0.043 | 108 | 745 | 9.9 | 68.3 | 78 | 538 | 2.70 | 473 | 0.81 | 0.232 |
| 8 |  | 0.044 | 111 | 765 | 10.8 | 74.5 | 62* | 428 ${ }^{\text {* }}$ | 1.38* | 241* | 0.57* | 0.267* |
| Average |  |  | 107 | 736 | 9.9 | 68.0 | 81 | 558 | 2.84 | 497 | 0.85 | 0.198 |
| Std. Dev. |  |  | 9 | 61 | 0.5 | 3.7 | 7 | 45 | 0.46 | 80 | 0.08 | 0.025 |
| NLTV21 | 180 | 0.045 | 112 | 772 | 9.9 | 68.3 | 72 | 496 | 2.32 | 406 | 0.73 | 0.388 |
| 22 |  | 0.044 | 109 | 751 | 10.8 | 74.5 | 70 | 483 | 1.94 | 339 | 0.67 | $0.564 *$ |
| 23 |  | 0.045 | 108 | 745 | 10.1 | 69.6 | 81 | 558 | 2.98 | 521 | 0.83 | 0.471 |
| 24 |  | 0.045 | 108 | 745 | 9.4 | 64.8 | 65 | 448 | 2.03 | 355 | 0.68 | 0.421 |
| 25 |  | 0.045 | 110 | 758 | 9.7 | 66.9 | 81 | 558 | 3.27* | 573* | 0.87 | 0.415 |
| Average |  |  | 109 | 754 | 10.0 | 68.8 | 74 | 509 | 2.32 | 405 | 0.76 | 0.424 |
| Std. Dev. |  |  | 2 | 11 | 0.5 | 3.6 | 7 | 48 | 0.47 | 82 | 0.09 | 0.035 |

*Not included in Average or Standard Deviation

TABLE A6
Lamina and Laminate Material Properties for IM7/8551-7 Composites

Test Temperature
$75^{\circ} \mathrm{F}\left(25^{\circ} \mathrm{C}\right) \quad 180^{\circ} \mathrm{F}\left(82^{\circ} \mathrm{C}\right)$
$[0]_{8}$ (Input Properties From Static Tests Used in AC3)

| $\mathrm{E}_{11}=$ | $23.3 \mathrm{Msi}(160.7 \mathrm{GPa})$ | $21.1 \mathrm{Msi}(145.5 \mathrm{GPa})$ |
| :--- | :--- | :--- |
| $\mathrm{E}_{22}=$ | $1.2 \mathrm{Msi}(8.3 \mathrm{GPa})$ | $1.1 \mathrm{Msi}(7.6 \mathrm{GPa})$ |
| $\mathrm{G}_{12}=$ | $0.83 \mathrm{Msi}(5.7 \mathrm{GPa})$ | $0.75 \mathrm{Msi}(5.2 \mathrm{GPa})$ |
| $\nu_{12}=$ | 0.180 | 0.370 |

$\underline{[ \pm 35 / 0 / 90]_{s}}$ (Predicted Properties From AC3)

| E | $=$ | $11.1 \mathrm{Msi}(76.5 \mathrm{GPa})$ | $10.3 \mathrm{Msi}(71.0 \mathrm{GPa})$ |
| :--- | :--- | ---: | ---: |
| $\mathrm{E}_{\mathrm{X}}=$ | $7.6 \mathrm{Msi}(52.4 \mathrm{GPa})$ | $7.1 \mathrm{Msi}(49.0 \mathrm{GPa})$ |  |
| $\mathrm{G}_{\mathrm{XY}}=$ | $3.1 \mathrm{Msi}(21.4 \mathrm{GPa})$ | $2.9 \mathrm{Msi}(20.0 \mathrm{GPa})$ |  |
| $\nu_{\mathrm{XY}}=$ | 0.311 | 0.332 |  |

$[ \pm 35 / 0]_{s}$ (Predicted Properties From AC3)

| E | $=11.8 \mathrm{Msi}(81.4 \mathrm{GPa})$ | $10.9 \mathrm{Msi}(75.2 \mathrm{GPa})$ |  |  |
| :--- | :--- | ---: | ---: | :---: |
| $\mathrm{E}_{\mathrm{X}}=$ | $2.3 \mathrm{Msi}(15.9 \mathrm{GPa})$ | $2.1 \mathrm{Msi}(14.5 \mathrm{GPa})$ |  |  |
| $\mathrm{G}_{\mathrm{XY}}^{\mathrm{Y}}=$ | $3.9 \mathrm{Msi}(26.9 \mathrm{GPa})$ | $3.6 \mathrm{Msi}(24.8 \mathrm{GPa})$ |  |  |
| $\nu_{\mathrm{XY}}=$ | 1.093 |  | 1.140 |  |

$\mathrm{E}^{*}=\quad 9.18 \mathrm{Msi}(63.3 \mathrm{GPa}) \quad 8.41 \mathrm{Msi}(58.0 \mathrm{GPa})$
(Calculated Using AC3 and Equation 2)
$\mathrm{E}_{\text {lam }}=9.85 \mathrm{Msi}(67.9 \mathrm{GPa}) \quad 9.99 \mathrm{Msi}$ ( 68.9 GPa )
(Measured Static EDT Initial Tangent Modulus)

## AXIAL TENSION 23 DEG C



## AXIAL TENSION 23 DEG C



AXIAL TENSION 82 DEG C


## AXIAL TENSION 82 DEG C



TRANSVERSE TENSION 23 DEG C


TRANSVERSE TENSION 82 DEG C


## IOSIPESCU SHEAR 23 DEG C



## IOSIPESCU SHEAR 23 DEG C



IOSIPESCU SHEAR 82 DEG C


## IOSIPESCU SHEAR 82 DEG C



## EDGE DELAMINATION 23 DEG C



EDGE DELAMINATION 23 DEG C


EDGE DELAMINATION 82 DEG C


EDGE DELAMINATION 82 DEG C



NLOTD2. TEN TEMP = 23.0 DEG. $\mathrm{C} \quad \mathrm{NU}=0.168$


NLOTD3. TEN


NLOTD3. TEN
ULT. STRESS $=382.500 \mathrm{ksi}$ TEMP = 23.0 DEG. $\mathrm{C} \quad \mathrm{NU}=0.186$



NLOTD4. TEN
$\begin{array}{cccc}\text { ULT. STRESS }= & 340.700 \mathrm{ksi} & \\ \text { TEMP }=23.0 \text { DEG. } \mathrm{C} & \mathrm{NU}= & 0.174\end{array}$


NLOTD5. TEN
ULT. $\operatorname{STRESS}=231.200 \mathrm{ksi}$
TEMP $=23.0$ DEG. $C \quad$ MOD $=24.672 \mathrm{Msi}$



NLOTDG. TEN
ULT. STRESS $=358.300 \mathrm{ksi}$
TEMP $=23.0$ DEG. $C \quad$ MOD $=22.318 \mathrm{Msi}$


NLOTDG. TEN
ULT. STRESS $=358.300 \mathrm{ksi}$
TEMP = 23.0 DEG. $\mathrm{C} \quad \mathrm{NU}=0.189$





NLFSH2.TEN
ULT. STRESS $=364.300 \mathrm{ksi}$
TEMP $=82.0$ DEG. $C \quad N U=0.404$





NLFSH4.TEN
ULT. STRESS $=368.500 \mathrm{ksi}$
TEMP = 82.0 DEG. $\mathrm{C} \quad \mathrm{NU}=0.345$


NLFSH5.TEN
ULT. STRESS $=372.300 \mathrm{ksi}$
TEMP $=82.0$ DEG. $C \quad$ MOD $=\quad 22.448 \mathrm{Msi}$





NLTTD2.TEN
ULT. STRESS $=7.607 \mathrm{ksi}$
TEMP $=23.0$ DEG. $C \quad$ MOD $=1.205 \mathrm{Msi}$


NLTTD3.TEN
ULT. STRESS $=2.801 \mathrm{ksi}$
TEMP $=23.0$ DEG. $\mathrm{C} \quad$ MOD $=1.152 \mathrm{Msi}$


NLTTD4. TEN




NTTD22.TEN TEMP $=\begin{array}{ll}\text { ULT. STRESS }= & 4.722 \mathrm{ksi} \\ \text { 日2.0 DEG. } \mathrm{C} & \mathrm{MOD}=\end{array} \quad 1.051 \mathrm{Msi}$








NFIRD1.IOS


NFIRD2.IOS


NFIRD2.IOS



NFIRD3.IOS



NFIRD4.IOS



NFIRD5.IOS



NFIHD1.IOS


NFIHD2.IOS
ULT. STRESS $=10.110 \mathrm{ksi}$
TEMP $=82.0$ DEG. $C \quad$ MOD $=0.772 \mathrm{Msi}$


NFIHD2.IOS



NFIHD3.IOS



NFIHD4.IOS



NFIHD5.IOS




NLDTD2.EDT


NLDTD2.EDT
ULT. STRESS $=98.660 \mathrm{ksi}$ TEMP $=23.0$ DEG. $\mathrm{C} \quad \mathrm{NU}=0.138$




NLDTD4.EDT
TEMP $=\begin{aligned} & \text { ULT. STRESS }=120.000 \mathrm{ksi} \\ & 23.0 \text { DEG. } \mathrm{C} \quad \mathrm{MOD}=10.061 \mathrm{Msi}\end{aligned}$




NLDTD5.EDT
ULT. STRESS $=93.990 \mathrm{ksi}$
TEMP = 23.0 DEG. $C \quad N U=\quad 0.183$








NLTD21.EDT
ULT. STRESS $=112.400 \mathrm{ksi}$ TEMP $=82.0$ DEG. $C \quad$ MOD $=9.882 \mathrm{Msi}$

NLTD21.EDT
ULT. STRESS $=112.400 \mathrm{ksi}$
TEMP $=82.0$ DEG. $\mathrm{C} \quad \mathrm{NU}=0.388$


NLTD22.EDT
ULT. STRESS $=108.800 \mathrm{ksi}$
TEMP $=82.0$ DEG. $\mathrm{C} \quad \mathrm{MOD}=10.789 \mathrm{Msi}$


NLTD22.EDT
ULT. STRESS $=108.800 \mathrm{ksi}$
TEMP $=82.0 \quad$ DEG. $\mathrm{C} \quad \mathrm{NU}=\quad 0.564$



NLTD23.EDT
ULT. STRESS $=107.900 \mathrm{ksi}$
TEMP = 82.0 DEG. $C \quad N U=\quad 0.471$



NLTD24.EDT
ULT. STRESS $=107.700 \mathrm{ksi}$
TEMP = 82.0 DEG. $\mathrm{C} \quad \mathrm{NU}=\quad 0.421$




IM7/8551-7 90 DEG NO. 1
ALPHA $=+3.618 \mathrm{E}-05 / \mathrm{C}$


$$
\begin{gathered}
\text { IM7/8551-7 } 90 \text { DEG NO. } 2 \\
\text { ALPHA }=+3.471 \mathrm{E}-05 / \mathrm{C}
\end{gathered}
$$



IM7/8551-7 90 DEG NO. 3
ALPHA $=+3.365 \mathrm{E}-05 / \mathrm{C}$


## APPENDIX B

# Naming Convention for Edge Delamination Fatigue Specimens 

Specimen name $=123456$. FTG - six characters with . FTG extension

Character Meaning:

1. L - NASA-Langley
2. F - Fatigue Specimen
3. L or H - Test Temperature
$\mathrm{L}=\mathrm{Low} 75^{\circ} \mathrm{F}\left(25^{\circ} \mathrm{C}\right)$
$\mathrm{H}=\mathrm{High} 180^{\circ} \mathrm{F}\left(82^{\circ} \mathrm{C}\right)$
4. Maximum Load Value where:

| $\mathrm{A}=95 \%$ of Ultimate Load | $\mathrm{J}=50 \%$ of Ultimate Load |
| :--- | :--- |
| $\mathrm{B}=90 \%$ | $\mathrm{~K}=45 \%$ |
| $\mathrm{C}=85 \%$ | $\mathrm{~L}=40 \%$ |
| $\mathrm{D}=80 \%$ | $\mathrm{M}=35 \%$ |
| $\mathrm{E}=75 \%$ | $\mathrm{~N}=30 \%$ |
| $\mathrm{~F}=70 \%$ | $\mathrm{O}=25 \%$ |
| $\mathrm{G}=65 \%$ | $\mathrm{P}=20 \%$ |
| $\mathrm{H}=60 \%$ | $\mathrm{Q}=15 \%$ |
| $\mathrm{I}=55 \%$ | $\mathrm{R}=10 \%$ |

5.     - Precondition Type
$S=$ No precondition (Normal)
$M=$ High Mean Load (1000 cycles)
$P=$ Spike Load (One cycle)
6.     - Specimen Number 0-9, A-Z

Example - LFLESA.FTG

```
Explanation - LF = NASA-Langley Fatigue Program
    L = Low temperature }7\mp@subsup{5}{}{\circ}\textrm{F}(2\mp@subsup{5}{}{\circ}\textrm{C}
    E = 75% of Ultimate Load
    S = No Precondition
    A = Specimen No. A
```


## Table B1

Individual Tension-Tension Edge Delamination Fatigue Results for
IM7/8551-7 Composites at Room Temperature No Precondition

| Specimen Name | Thickness |  | Peak Stress |  | Initial Modulus |  | Final Modulus |  | $\begin{gathered} \mathrm{R} \\ \text { Ratio } \end{gathered}$ | Cycles to 5\% Modulus Decay | Cycles to Delamination | $\begin{gathered} \text { Delami- } \\ \text { nation } \\ \text { Strain } \\ \text { (percent) } \end{gathered}$ | Gc$\left(\frac{1 b_{f} \text { in }}{i n^{2}}\right)$ |  |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  | (in) | (mm) | (ksi) | (MPa) | (Msi) | (GPa) | (Msi) | (GPa) |  |  |  |  |  | $\left(\mathrm{J} / \mathrm{m}^{2}\right)$ |
| LFSASA 95\% | 0.044 | 1.12 | 79.1 | 545 | 9.34 | 64.35 | 8.86 | 61.05 | 0.113 | 66 | 1 | 0.87 | 1.11 | 193.58 |
| LFLASC 95\% | 0.044 | 1.12 | 81.8 | 564 | 9.37 | 64.56 | 8.26 | 56.91 | 0.151 | 6 | 1 | 0.81 | 0.970 | 169.77 |
| LFLCSA 85\% | 0.044 | 1.12 | 73.6 | 507 | 10.12 | 69.73 | 9.55 | 65.80 | 0.139 | 56 | 1 | 0.66 | 0.642 | 112.44 |
| LFLDSB 80\% | 0.044 | 1.12 | 70.8 | 488 | 9.82 | 67.66 | 9.31 | 64.15 | 0.119 | 390 | 170 | 0.56 | 0.454 | 79.51 |
| LFLESA 75\% | 0.044 | 1.12 | 62.5 | 431 | 10.16 | 70.00 | 9.65 | 66.49 | 0.112 | 950 | 36 | 0.61 | 0.545 | 95.42 |
| LFLFSA 70\% | 0.045 | 1.14 | 55.2 | 380 | 8.96 | 61.73 | 8.51 | 58.63 | 0.110 | 3,90 | 150 | 0.64 | 0.619 | 108.47 |
| LFLGSA 65\% | 0.044 | 1.12 | 50.8 | 350 | 9.30 | 64.08 | 8.83 | 60.84 | 0.087 | 37,000 | 3,900 | 0.57 | 0.484 | 84.75 |
| LFLISC 55\% | 0.043 | 1.09 | 42.6 | 294 | 9.27 | 63.87 | 8.87 | 61.11 | 0.127 | 240,000 | 8,200 | 0.47 | 0.322 | 56.44 |
| LFLISH 55\% | 0.045 | 1.14 | 47.4 | 327 | 8.82 | 60.77 | 8.37 | 57.67 | 0.124 | 31,000 | 2,400 | 0.53 | 0.422 | 73.88 |
| LFLISK 55\% | 0.047 | 1.19 | 38.5 | 265 | 8.23 | 56.70 | 7.79 | 53.67 | 0.112 | 240,000 | 150 | 0.45 | 0.317 | 55.59 |
| LFLJSE 50\% | 0.044 | 1.12 | 39.7 | 274 | 9.01 | 62.08 | 8.56 | 58.98 | 0.087 | 52,000 | 4,000 | 0.44 | 0.283 | 49.52 |
| LFLKSA 45\% | 0.044 | 1.12 | 34.3 | 236 | 8.93 | 61.53 | 8.48 | 58.43 | 0.099 | 300,000 | 9,600 | 0.40 | 0.231 | 40.48 |
| LFLKSE 45\% | 0.044 | 1.12 | 35.2 | 243 | 9.41 | 64.83 | 8.88 | 61.18 | 0.107 | 370,000 | 8,800 | 0.39 | 0.220 | 38.46 |
| LFLLSA 40\% | 0.044 | 1.12 | 33.7 | 232 | 9.52 | 65.59 | 8.99 | 61.94 | 0.112 | 8,600 | 1,900 | 0.36 | 0.190 | 33.27 |
| LFLLSB 40\% | 0.043 | 1.09 | 32.9 | 227 | 9.68 | 66.70 | 9.17 | 63.18 | 0.096 | 39,000 | 37,000 | 0.34 | 0.166 | 28.99 |
| LFLLSC $40 \%$ | 0.045 | 1.14 | 30.2 | 208 | 9.10 | 62.70 | 8.63 | 59.46 | 0.126 | +83,003 | 7,500 | 0.38 | 0.212 | 37.12 |
| LFLMSA 35\% | 0.045 | 1.14 | 30.2 | 208 | 9.03 | 62.22 | 8.65 | 59.60 | 0.116 | 9,500,000 | 4,300,000 | 0.34 | 0.179 | 31.42 |

Table B2
Individual Tension-Tension Edge Delamination Fatigue Results for
IM7/8551-7 Composites at High Temperature No Precondition

| Specimen Name | Thickness |  | Peak Stress |  | Initial Modulus |  | Final Modulus |  | $\stackrel{\mathrm{R}}{\text { Ratio }}$ | Cycles to 5\% Modulus Decay | Cycles to Delamination | Delami- <br> nation <br> Strain <br> (percent) | $\begin{gathered} \left.\frac{\mathrm{Gc}}{\left(\mathrm{lb}_{\mathrm{f}} \mathrm{in}\right.}\right) \\ \mathrm{in}^{2} \end{gathered}$ |  |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  | (ksi) | (MPa) | (Msi) | (GPa) | (Msi) | (GPa) |  |  |  |  |  | $\left(\mathrm{J} / \mathrm{m}^{2}\right)$ |
| LFLBPA 90\% | 0.044 | 1.12 | 74.2 | 511 | 9.14 | 62.97 | 8.68 | 59.81 | 0.138 | 330 | 11 | 0.82 | 1.00 | 175.69 |
| LFLBPB 90\% | 0.045 | 1.14 | 73.9 | 509 | 9.50 | 65.46 | 9.01 | 62.08 | 0.109 | 36 | 1 | 0.76 | 0.864 | 151.28 |
| LFLCPA 85\% | 0.044 | 1.12 | 70.6 | 486 | 10.22 | 70.42 | 9.71 | 66.90 | 0.144 | 86 | 6 | 0.68 | 0.690 | 120.76 |
| LFLDPA 80\% | 0.044 | 1.12 | 65.6 | 452 | 9.04 | 62.29 | 8.57 | 59.05 | 0.133 | 490 | 1 | 0.70 | 0.733 | 128.29 |
| LFLEPA 75\% | 0.043 | 1.09 | 66.0 | 455 | 10.23 | 70.48 | 9.71 | 66.90 | 0.105 | 320 | 1 | 0.61 | 0.541 | 94.79 |
| LFLFPA 70\% | 0.045 | 1.14 | 59.1 | 407 | 9.07 | 62.49 | 8.59 | 59.19 | 0.103 | 490 | 1 | 0.64 | 0.612 | 107.12 |
| LFLFPB 70\% | 0.045 | 1.14 | 54.7 | 377 | 9.13 | 62.91 | 8.66 | 59.67 | 0.109 | 3,800 | 750 | 0.60 | 0.546 | 95.67 |
| LFLGPA 65\% | 0.045 | 1.14 | 53.7 | 370 | 8.87 | 61.11 | 8.37 | 57.67 | 0.118 | 15.000 | 9,900 | 0.57 | 0.493 | 86.37 |
| LFLHPA 60\% | 0.043 | 1.09 | 50.9 | 351 | 9.16 | 63.11 | 8.70 | 59.94 | 0.104 | 5,800 | 9,400 | 0.55 | 0.437 | 76.59 |
| LFLIPA 55\% | 0.045 | 1.14 | 42.0 | 289 | 8.91 | 61.39 | 8.45 | 58.22 | 0.114 | 200,000 | 160,000 | 0.45 | 0.301 | 52.75 |
| LFLIPB 55\% | 0.044 | 1.12 | 46.9 | 323 | 10.18 | 70.14 | 9.63 | 66.35 | 0.099 | 24,000 | 3.200 | 0.45 | 0.299 | 52.27 |
| LFLJPA 50\% | 0.044 | 1.12 | 41.7 | 287 | 8.97 | 61.80 | 8.52 | 58.70 | 0.106 | 140,000 | 6,700 | 0.45 | 0.299 | 52.27 |
| LFFLKPA 45\% | 0.044 | 1.12 | 36.2 | 249 | 8.88 | 61.18 | 8.42 | 58.01 | 0.118 | 41,000 | 5,500 | 0.41 | 0.247 | 43.18 |
| LFLLPA 40\% | 0.044 | 1.12 | 31.8 | 219 | 9.90 | 68.21 | 9.40 | 64.77 | 0.104 | 7.900 | 1 | 0.31 | 0.145 | 25.45 |
| LFLLPC 40\% | 0.045 | 1.14 | 32.8 | 226 | 8.96 | 61.73 | 8.51 | 58.63 | 0.118 | 960,000 | 5,800 | 0.37 | 0.208 | 36.34 |
| LFLNPB 30\% | 0.044 | 1.12 | 24.5 | 169 | 8.86 | 61.05 | 8.40 | 57.88 | 0.110 | 19,000 | 1,000 | 0.24 | 0.088 | 15.49 |
| LFLQPB 15\% | 0.044 | 1.12 | 12.9 | 89 | 9.55 | 65.80 | 9.45 | 65.11 | 0.119 | *11,700,000 | 10,788,200 | 0.11 | 0.018 | 3.18 |

Table B3
Individual Tension-Tension Edge Delamination Fatigue Results for

| Specimen Name | Thickness |  | Peak Stress |  | Initial <br> Modulus |  | Final Modulus |  | $\begin{gathered} \mathrm{R} \\ \text { Ratio } \end{gathered}$ | Cycles to 5\% Modulus Decay | Cycles to Delamination | Delamination Strain (percent) | Gc$\left(\frac{\mathrm{lb}_{\mathrm{f}}^{\mathrm{in}}}{i n^{2}}\right)$ | Gc$\left(\mathrm{J} / \mathrm{m}^{2}\right)$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  | (in) | (mm) | (ksi) | (MPa) | (Msi) | (GPa) | (Msi) | (GPa) |  |  |  |  |  |  |
| LFLAMB 95\% | 0.043 | 1.09 | 80.1 | 552 | 9.40 | 64.77 | 8.91 | 61.39 | 0.102 | 170 | 1 | 0.83 | 0.985 | 172.53 |
| LFLAMC 95\% | 0.044 | 1.12 | 77.1 | 531 | 9.14 | 62.97 | 8.65 | 59.60 | 0.122 | 190 | 1 | 0.80 | 0.953 | 166.86 |
| LFLBMA 90\% | 0.043 | 1.09 | 77.0 | 531 | 9.80 | 67.52 | 9.25 | 63.73 | 1.177 | 11 | 1 | 0.80 | 0.917 | 160.64 |
| LFLBMB 90\% | 0.044 | 1.12 | 75.3 | 519 | 10.22 | 70.42 | 9.65 | 66.49 | 0.091 | 31 | 1 | 0.70 | 0.716 | 125.40 |
| LFLBMC 90\% | 0.045 | 1.14 | 73.4 | 506 | 9.76 | 67.25 | 9.26 | 63.80 | 0.114 | 6 | 1 | 0.72 | 0.775 | 135.71 |
| LFLDMA 80\% | 0.043 | 1.09 | 68.1 | 469 | 9.78 | 67.38 | 9.29 | 64.01 | 0.111 | 250 | 1 | 0.67 | 0.649 | 113.58 |
| LFLDMB 80\% | 0.044 | 1.12 | 66.3 | 457 | 9.73 | 67.04 | 9.23 | 63.59 | 0.119 | 460 | 66 | 0.69 | 0.700 | 122.54 |
| LFLFMA 70\% | 0.045 | 1.14 | 57.9 | 399 | 9.66 | 66.56 | 9.16 | 63.11 | 0.099 | 4,800 | 510 | 0.62 | 0.585 | 102.46 |
| LFLFMB 70\% | 0.045 | 1.14 | 58.6 | 404 | 9.48 | 65.32 | 8.99 | 61.94 | 0.101 | 900 | 100 | 0.60 | 0.534 | 93.46 |
| LFLHMA 60\% | 0.045 | 1.12 | 50.2 | 346 | 9.60 | 66.14 | 9.10 | 62.70 | 0.102 | 27,000 | 320 | 0.53 | 0.428 | 75.00 |
| LFLHMB 60\% | 0.044 | 1.14 | 51.0 | 351 | 9.62 | 66.28 | 9.13 | 62.91 | 0.101 | 8.400 | 960 | 0.51 | 0.382 | 66.88 |
| LFL.JMB 50\% | 0.045 | 1.12 | 43.5 | 300 | 8.49 | 58.50 | 8.06 | 55.53 | 0.089 | 74,000 | 540 | 0.50 | 0.378 | 66.26 |
| LFLKMA 45\% | 0.044 | 1.12 | 37.3 | 257 | 9.40 | 64.77 | 8.92 | 61.46 | 0.107 | 20,000 | 3,300 | 0.40 | 0.230 | 40.27 |
| LFLKMB 45\% | 0.044 | 1.14 | 37.5 | 258 | 9.85 | 67.87 | 9.32 | 64.21 | 0.092 | 25,000 | 1,100 | 0.38 | 0.213 | 37.27 |
| LFLKMC 45\% | 0.045 | 1.12 | 37.2 | 256 | 9.44 | 65.04 | 8.94 | 61.60 | 0.097 | 98,000 | 3,400 | 0.39 | 0.229 | 40.15 |
| LFLLMA 40\% | 0.044 | 1.09 | 33.0 | 227 | 8.81 | 60.70 | 8.35 | 57.53 | 0.132 | 36,000 | 3,300 | 0.37 | 0.197 | 34.58 |
| LFLMMA | 0.043 | 1.09 | 29.6 | 204 | 9.02 | 62.15 | 8.55 | 58.91 | 0.090 | 290,000 | 860 | 0.31 | 0.138 | 24.24 |
| LFLOMA | 0.043 | 1.09 | 21.4 | 147 | 9.71 | 66.90 | 9.21 | 63.46 | 0.096 | 13,000 | 3,400 | 0.20 | 0.058 | 10.19 |
| LFLQMA | 0.044 | 1.12 | 12.6 | 87 | 9.57 | 65.94 | 9.12 | 62.84 | 0.102 | 2,000,000 | 8,800 | 0.13 | 0.025 | 4.30 |

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Table B4
Individual Tension-Tension Edge Delamination Fatigue Results for

| Specimen Name | Thickness |  | Peak Stress |  | Initial Modulus |  | Final Modulus |  | $\begin{gathered} \mathrm{R} \\ \text { Ratio } \end{gathered}$ | Cycles to 5\% Modulus Decay | Cycles to Delamination | Delami- <br> nation <br> Strain <br> (percent) | $\begin{aligned} & \frac{\mathrm{Gc}}{\mathrm{lb}_{\mathrm{f}} \mathrm{in}} \\ & \mathrm{in}^{2} \end{aligned}$ | Gc$\left(\mathrm{J} / \mathrm{m}^{2}\right)$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  | (mm) | (ksi) | (MPa) | (Msi) | (GPa) | (Msi) | (GPa) |  |  |  |  |  |  |
| LFHASC 95\% | 0.049 | 1.24 | 77.8 | 535 | 7.60 | 52.36 | 7.21 | 49.68 | 0.102 | 320 | 1 | 0.98 | 3.74 | 655.03 |
| LFHASD 95\% | 0.044 | 1.12 | 78.8 | 543 | 8.97 | 61.80 | 7.94 | 54.71 | 0.113 | 6 | 1 | 0.87 | 2.64 | 462.85 |
| LFHCSA 85\% | 0.044 | 1.12 | 71.5 | 493 | 8.82 | 60.77 | 8.23 | 56.70 | 0.123 | 16 | 1 | 0.82 | 2.31 | 404.32 |
| LFHCSB 85\% | 0.045 | 1.14 | 71.4 | 492 | 9.80 | 67.52 | 9.30 | 64.08 | 0.103 | 26 | 1 | 0.72 | 1.83 | 320.04 |
| LFHDSA 80\% | 0.049 | 1.24 | 67.5 | 465 | 8.18 | 56.36 | 7.73 | 53.26 | 0.125 | 31 | 1 | 0.83 | 2.69 | 471.50 |
| LFHESA 75\% | 0.044 | 1.12 | 64.4 | 444 | 9.15 | 63.04 | 8.66 | 59.67 | 0.106 | 590 | 51 | 0.72 | 1.78 | 312.06 |
| LFHFSA 70\% | 0.051 | 1.30 | 59.0 | 407 | 8.09 | 55.74 | 7.66 | 52.78 | 0.116 | 220 | 1 | 0.71 | 2.03 | 354.67 |
| LFHFSB 70\% | 0.043 | 1.09 | 59.1 | 407 | 10.37 | 71.45 | 9.84 | 67.80 | 0.101 | 690 | 86 | 0.58 | 1.12 | 196.68 |
| LFHGSA 65\% | 0.043 | 1.09 | 55.1 | 380 | 10.41 | 71.72 | 9.87 | 68.00 | 0.100 | 1,400 | 110 | 0.53 | 0.954 | 167.10 |
| LFHHSB 60\% | 0.042 | 1.07 | 50.3 | 347 | 9.32 | 64.21 | 8.83 | 60.84 | 0.107 | 6,300 | 390 | 0.54 | 0.953 | 166.93 |
| LFHISA 55\% | 0.043 | 1.09 | 46.8 | 322 | 9.25 | 63.73 | 8.72 | 60.08 | 0.102 | 6,800 | 1,300 | 0.50 | 0.832 | 145.76 |
| LFHJSA 50\% | 0.043 | 1.09 | 42.3 | 291 | 9.57 | 65.94 | 9.08 | 62.56 | 0.145 | 50,000 | 1,400 | 0.42 | 0.611 | 106.94 |
| LFHJSB 50\% | 0.044 | 1.12 | 42.4 | 292 | 8.92 | 61.46 | 8.48 | 58.43 | 0.125 | 23,000 | 2,600 | 0.48 | 0.791 | 138.50 |
| LFHKSA 45\% | 0.044 | 1.12 | 37.8 | 260 | 9.73 | 67.04 | 9.23 | 63.59 | 0.106 | 48,000 | 7,300 | 0.39 | 0.540 | 94.49 |
| LFHMSB 35\% | 0.044 | 1.12 | 29.2 | 201 | 9.09 | 62.63 | 8.36 | 57.60 | 0.111 | 1,150,000 | 210,000 | 0.33 | 0.388 | 67.91 |
| LFHQSB 15\% | 0.045 | 1.14 | 11.8 | 81 | 8.88 | 61.18 | 8.81 | 60.70 | 0.106 | 10,000,000 | 9,700,000 | 0.08 | 0.026 | 4.50 |

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Table B5
Individual Tension-Tension Edge Delamination Fatigue Results for

| Specimen Name | Thickness |  | Peak Stress |  | Initial <br> Modulus |  | Final Modulus |  | $\begin{gathered} \mathbf{R} \\ \text { Ratio } \end{gathered}$ | Cycles to $5 \%$ Modulus Decay | Cycles to Delamination | Delami- <br> nation <br> Strain <br> (percent) | Gc$\left(\frac{l b_{f} i n}{i n^{2}}\right)$ | Gc |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  | (in) | (mm) | (ksi) | (MPa) | (Msi) | (GPa) | (Msi) | (GPa) |  |  |  |  |  | ( $\mathrm{J} / \mathrm{m}^{2}$ ) |
| LFHAPB 95\% | 0.044 | 1.12 | 79.5 | 548 | 9.15 | 63.04 | 8.64 | 59.53 | 0.143 | 140 | 21 | 0.86 | 2.60 | 455.45 |
| LFHBPA 90\% | 0.043 | 1.09 | 75.6 | 521 | 10.28 | 70.83 | 9.73 | 67.04 | 0.128 | 11 | 1 | 0.70 | 1.67 | 292.32 |
| LFHCPA 85\% | 0.045 | 1.14 | 72.8 | 502 | 8.91 | 61.39 | 8.39 | 57.81 | 0.091 | 16 | 1 | 0.76 | 2.03 | 355.81 |
| LFHDPA 80\% | 0.043 | 1.09 | 67.7 | 466 | 10.33 | 71.17 | 9.79 | 67.45 | 0.119 | 170 | 1 | 0.61 | 1.25 | 219.18 |
| LFHEPA 75\% | 0.043 | 1.09 | 64.6 | 445 | 10.46 | 72.07 | 9.94 | 63.49 | 0.115 | 520 | 1 | 0.59 | 1.18 | 206.37 |
| LFHFPA 70\% | 0.044 | 1.12 | 60.7 | 418 | 9.87 | 68.00 | 9.34 | 64.35 | 0.090 | 500 | 1 | 0.59 | 1.22 | 213.33 |
| LFHGPA 65\% | 0.043 | 1.09 | 54.3 | 374 | 10.21 | 70.35 | 9.70 | 66.83 | 0.125 | 2,600 | 41 | 0.51 | 0.891 | 155.94 |
| LFHHPA 60\% | 0.044 | 1.12 | 52.6 | 362 | 9.91 | 68.28 | 9.37 | 64.56 | 0.111 | 48 C | 1 | 0.51 | 0.908 | 158.95 |
| LFHHPB 60\% | 0.043 | 1.09 | 52.1 | 359 | 9.40 | 64.77 | 8.92 | 61.46 | 0.106 | 9,900 | 1,200 | 0.55 | 1.04 | 181.26 |
| LFHIPA 55\% | 0.044 | 1.12 | 47.9 | 330 | 9.45 | 65.11 | 8.96 | 61.73 | 0.097 | 7,500 | 1.200 | 0.50 | 0.859 | 150.36 |
| LFHJPA 50\% | 0.045 | 1.14 | 42.8 | 295 | 9.69 | 66.76 | 9.17 | 63.18 | 0.090 | 41,000 | 61.400 | 0.43 | 0.660 | 115.64 |
| LFHKPA 45\% | 0.043 | 1.09 | 39.4 | 271 | 8.90 | 61.32 | 8.40 | 57.88 | 0.097 | 130,000 | 19,000 | 0.45 | 0.688 | 120.46 |
| LFHLPA 40\% | 0.044 | 1.12 | 33.6 | 232 | 8.86 | 61.04 | 8.37 | 57.67 | 0.107 | 150,000 | 34,000 | 0.40 | 0.548 | 95.94 |
| LFHMPA 35\% | 0.044 | 1.12 | 30.0 | 207 | 9.55 | 65.80 | 9.05 | 62.35 | 0.099 | 370,000 | 70,000 | 0.32 | 0.347 | 60.78 |
| LFHQPB 15\% | 0.043 | 1.09 | 11.9 | 82 | 10.11 | 69.66 | 10.48 | 72.21 | 0.093 | *11,000,000 | 11,000,000 | 0.10 | 0.035 | 6.19 |

Table B6
Individual Tension-Tension Edge Delamination Fatigue Results for

| SpecimenName | Thickness |  | Peak Stress |  | Initial Modulus |  | Final <br> Modulus |  | $\begin{gathered} \mathrm{R} \\ \text { Ratio } \end{gathered}$ | Cycles to 5\% Modulus Decay | Cycles to Delamination | Delami- <br> nation <br> Strain <br> (Fercent) | $\left(\frac{\mathrm{Gc}}{i b_{f} \mathrm{in}}\right)$ | Gc$\left(\mathrm{J} / \mathrm{m}^{2}\right)$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  | (in) | (mm) | (ksi) | (MPa) | (Msi) | (GPa) | (Msi) | (GPa) |  |  |  |  |  |  |
| LFHAMB 95\% | 0.045 | 1.14 | 81.7 | 563 | 9.02 | 62.15 | 8.50 | 58.57 | 0.089 | 41 | 1 | 0.84 | 2.51 | 439.27 |
| LFHBMA 90\% | 0.044 | 1.12 | 76.2 | 525 | 9.31 | 64.21 | 8.66 | 59.67 | 0.089 | 16 | 1 | 0.76 | 1.99 | 348.82 |
| LFHCMA 85\% | 0.044 | 1.12 | 72.7 | 501 | 9.07 | 62.49 | 8.60 | 59.25 | 0.093 | 140 | 1 | 0.74 | 1.92 | 336.94 |
| LFHDMA 80\% | 0.044 | 1.12 | 69.5 | 479 | 9.07 | 62.49 | 8.61 | 59.32 | 0.081 | i 30 | 21 | 0.76 | 2.03 | 355.30 |
| LFHEMA 75\% | 0.042 | 1.07 | 63.1 | 435 | 9.77 | 67.32 | 9.28 | 63.94 | 0.134 | 390 | 21 | 0.64 | 1.36 | 237.25 |
| LFHFMA 70\% | 0.044 | 1.12 | 58.7 | 404 | 10.01 | 68.97 | 9.50 | 65.46 | 0.117 | 340 | 1 | 0.54 | 1.00 | 175.53 |
| LFHFMB 70\% | -0.044 | 1.12 | 58.9 | 406 | 9.51 | 65.52 | 9.01 | 62.08 | 0.130 | 790 | 36 | 0.63 | 1.37 | 239.30 |
| LFHGMA 65\% | 0.045 | 1.14 | 54.4 | 375 | 9.30 | 64.08 | 8.81 | 60.70 | 0.120 | 1.700 | 1 | 0.53 | 0.991 | 173.56 |
| LFHHMA 60\% | 0.045 | 1.14 | 50.9 | 351 | 9.52 | 65.59 | 9.03 | 62.22 | 0.144 | 2,700 | 11 | 0.51 | 0.925 | 161.92 |
| LFHIMA 55\% | 0.045 | 1.14 | 47.1 | 325 | 9.17 | 63.18 | 8.69 | 59.87 | 0.134 | 2,600 | 150 | 0.50 | 0.882 | 154.39 |
| LFHIMC 55\% | 0.044 | 1.12 | 46.8 | 322 | 9.83 | 67.73 | 9.27 | 63.87 | 0.127 | 9,800 | 1,300 | 0.44 | 0.667 | 116.78 |
| LFHJMA 50\% | 0.044 | 1.12 | 42.5 | 293 | 9.80 | 67.52 | 9.28 | 63.94 | 0.120 | 17,000 | 940 | 0.38 | 0.507 | 88.83 |
| LFHJMB 50\% | 0.042 | 1.07 | 41.7 | 287 | 9.63 | 66.35 | 9.04 | 62.29 | 0.111 | 180.000 | 2,600 | 0.44 | 0.628 | 109.95 |
| LFHKMA 45\% | 0.045 | 1.14 | 38.2 | 263 | 9.28 | 63.94 | 8.81 | 60.70 | 0.142 | 44,000 | 9,100 | 0.42 | 0.621 | 108.77 |
| LFHLMA 40\% | 0.044 | 1.12 | 32.9 | 227 | 8.89 | 61.25 | 8.45 | 58.22 | 0.139 | 160,000 | 4,500 | 0.36 | 0.446 | 78.01 |
| LFHMMA 35\% | 0.044 | 1.12 | 29.5 | 203 | 9.15 | 63.04 | 8.87 | 61.11 | 0.119 | 2,370,000 | 890,000 | 0.33 | 0.369 | 64.69 |
| LFHMMC 35\% | 0.042 | 1.07 | 27.5 | 189 | 8.88 | 61.18 | 8.34 | 57.46 | 0.142 | 7,400,000 | 2,000,000 | 0.31 | 0.321 | 56.20 |






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