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New Directions in InP Solar Cell Research

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ABSTRACT

Recent research efforts, representing new directions in InP solar cell research, are reviewed. These include heteroepitaxial growth on silicon and gallium arsenide substrates, V-grooved cells, large area high efficiency cells and surface passivation. Improvements in heteroepitaxial cell efficiency are described together with processing of 19.1%, 4 cm² cells. Recommendations are made for improvements in processing leading to increased efficiencies.

1. INTRODUCTION

Indium phosphide solar cells are known to be more radiation resistant than either gallium arsenide or silicon [1]. Furthermore, AMO efficiencies approaching 22% are predicted, 19.1% being the highest efficiency reported to date [2,3]. In addition, InP solar cells are currently being used to power a Japanese lunar orbiting satellite [4]. Presently, a small 4 cell module, on board the LIPS III satellite, shows no degradation after almost three years in orbit [5]. All of these efforts have been concerned with monolithic n/p solar cells processed by either organometallic chemical vapor deposition (OMCVD) or diffusion into InP substrates. One disadvantage of these approaches is the high cost of the InP wafers used as starting substrates. Current research efforts, aimed at cost reduction, include heteroepitaxial growth on cheaper more durable substrates and processing aimed at multiple substrate reuse. In addition to the cost reduction efforts, a research program aimed at increased efficiencies using a V-grooved front surface has produced encouraging preliminary results. A third effort, aimed at the processing of large area high efficiency cells has achieved notable success. These efforts, which represent new directions in InP solar cell research, are reviewed in the present paper.

2. LOW COST APPROACHES

From the viewpoint of reduced cost and increased strength, it would be advantageous to use silicon as a substrate in producing a heteroepitaxial InP solar cell. However, the 7.5% and 42% mismatches in lattice constant and expansion coefficient, respectively, introduces difficulties in the use of silicon substrates. In this respect, both germanium and gallium arsenide are preferrable. However, the significant cost reduction arising from the use of silicon and the relative ease of substrate thinning makes use of this substrate most desireable. In any case, intermediate layers are essential in reducing the dislocation densities resulting from the aforementioned mismatches.

Yamaguchi, et al, in their pioneering efforts, used OMCVD to deposit InP directly onto a silicon substrate. This resulted in a cell whose AMO efficiency was 4% [6]. On the other hand, in a feasibility study, sponsored by NASA Lewis, Spire, using OMCVD, produced InP cells on GaAs substrates (InP/GaAs) and on a silicon substrate with an intermediate GaAs layer (InP/GaAs/Si) [7]. In each case, the InP layer thickness was 4 micrometers. Cell parameters are shown in Table I, together with values obtained from the theoretical predictions of Yamaguchi, et al [8]. Paren-thetically, it is noted that all measurements reported in this paper were performed at NASA lewis at 25 °C using an InP flight cali-Lewis, at 25 °C, using an InP flight calibrated standard cell and an AMO solar intensity of 137.2 mW cm $^{-2}$. The low efficiencies are due to the high density of dislocations which in turn result in extremely small values of diffusion lengths [8]. The lower efficiency of the InP/GaAs/Si cell is attributed to doping of the GaAs by Si [7]. Since deposition of the GaAs layer occurs at 1000 $^{\circ}$ C, silicon, which is an n-dopant in InP, diffuses into the GaAs layer, creating a reverse diode. To circumvent this effect, the GaAs layer was appropriately

shorted out [7]. This, however, creates a high resistance path for holes arriving at the back surface, leading to increased series resis-tance. This is illustrated in figure 1 where the effects of series resistance are clearly evident in the I-V curve of the InP/GaAs/Si cell. The radiation resistance of the heteroepitaxial cells is shown to be significantly higher than that of the usual homoepitaxial InP cells after 10 MeV electron irradiation [9]. However, this is attributable to the high dislocation densities which dominate the present heteroepitaxial cell performance. In fact, the present cells have dislocation densities in the 10^8 cm⁻² range. Hence a reduction by several orders of magnitude is required to reach efficiencies over 18%. Obviously additional transition layers are needed to reduce the effects of mismatch between cell and substrate. This approach has been pursued by Wanlass at SERI (10). The initial result, using a GaAs substrate, was a heteroepitaxial InP cell whose AMO efficiency was 13.7% [10]. A major difference between this latter cell and both of the Spire heteroepitaxial cells is the use of a $Ga_XIn_{1-X}As$ transition layer. In fact this latter semi-conductor with x=0.47 is lattice matched to InP, resulting in an order of magnitude decrease in dislocation density. This pre-liminary result is highly encouraging and cell performance should improve with additional effort. Although a GaAs substrate was used, future efforts will be directed towards use of the more desireable silicon substrate. There still remains the problem of autodoping by the silicon. However, it is felt that appropri-ately heavy doping of the p⁺ buffer layer, next to the silicon substrate, should overcome this difficulty.

The preceding approaches use OMCVD to deposit the cell structure and transition layers on the relatively thick substrate. In another approach, liquid phase epitaxy (LPE) is used to deposit the thin, single crystal, films on a silicon substrate [11]. The potential of this technique has been demonstrated by fabricating LED's from InP films grown on silicon with transition layers grading from GaP to $In_xGa_{1-x}P$. The future near term goals of this research include the processing of solar cells using other types of transition layers. However, at the time of this writing, no cell fabrication results have been reported.

Another approach towards cost reduction makes use of processes which allow for multiple substrate reuse. In this regard, the CLEFT [12] and chemical lift-off techniques (13) are designed to produce thin cells which are lifted off reusable substrates. The substrates can then be reused to produce additional cells. The CLEFT process has produced GaAs cells whose efficiency, measured at NASA Lewis, was 20.5%. The chem lift off process has also been used to produce device quality GaAs films. In the expectation that the success with GaAs could be transferred to InP, the Lewis Research Center has initiated a program, at the Kopin Corporation, in which InP cells are to be produced by both processes. To date, InP films have been successfully lifted off from InP substrates using the chemical lift-off process. In this latter case, a thin pseudomorphic AlAs release layer was used with HF as the releasing agent. The CLEFT process has been demonstrated for an InP layer grown on an intermediate GaAs layer which was laterally overgrown on a GaAs sub-strate. The next step includes demonstration of lateral overgrowth of InP on an InP substrate with subsequent cleavage of the InP film. After successful demonstration of film separation, InP solar cells will be produced by both lift-off processes. Although in their embryo stages, it is expected that both processes will eventually result in lower cost, lightweight and highly efficient InP solar cells.

3. V-GROOVED INP SOLAR CELLS

Preliminary research on V-grooved InP solar cells has been conducted at NASA Lewis [14]. The cell geometry is shown schematically in figure 2. The grooves are formed by anisotropic etching, the etch rate being dependent on the orientation of the etched face [14, 15, 16]. Before etching, the cell surface is in the (100) direction. After etching, the slanted surface is in the (111) direction. The emitter of the n⁺p InP cell was formed by closed ampoule diffusion of sulphur using In₂S₃ as a diffusion source [14].

The main advantage of the V-grooved cell is lower total surface reflection. A disadvantage is the increased dark current due to the in-This latter effect creased junction area. leads to decreased cell voltage. Preliminary calculations indicate an expected increase of 10% in Isc and a 14 mV decrease in Voc. Although not spectacular, the initial results for a few unoptimized, low efficiency cells, are encouraging. As shown in figure 3, the quantum efficiency of the V-grooved cells are greater than that of the planar InP cell. In comparing three V-grooved cells to two planar cells the increase in Isc was 8% while the decrease in Voc was 37 millivolts. The much greater than expected reduction in Voc may be due to poor surface quality of the grooved cells, inadequate diffusion into the (111) face and striations and other imperfections in the Since the V-grooved cell grooved surface. processing is still in the experimental stage, additional effort is needed to effect improvements. In addition, the antireflection coating needs to be optimized before any meaningful comparision in cell efficiencies can be made. Eventually, the cells will be processed by OMCVD to attain the highest efficiencies.

LARGER AREA-HIGH EFFICIENCY CELLS

In previous years, the highest efficiency InP solar cell had an efficiency of 18.8% [17]. This cell, however, was of relatively small area (0.25 cm²). Since cells of this size are unsuited for practical use, it is essential to show that similar results can be attained with larger area cells. Most recently, the cell processing has been scaled up to produce 4 cm² cells whose efficiency was 19.1% [3]. The AMO parameters for this cell are shown in table II. The previous, smaller area, highest efficiency cell was processed by a combination of OMCYD and ion implantation the latter process being used to form the emitter region of the n^+pp^+ cell [17]. On the other hand, the 4 cm² n^+pp^+ cell was processed the 4 cm² n⁺pp⁺ cell was processed entirely by OMCVD [3]. The present high efficiency cell includes a graded emitter structure incorporating a doping of 3x10¹⁸ cm^{-3} near the main junction and a doping density of 1.5×10^{19} cm⁻³ at the front surface. In effect, this creates a front surface field which tends to reduce the flow of emitter minority carriers to the front surface. A surface recombination velocity of 4.5×10^6 cm sec⁻¹ was obtained when fitting the measured short wavelength quantum effi-ciency to calculated results (3). This indicates that surface passivation is necessary to achieve AMO efficiencies over 20%. Hence efforts to achieve the necessary passivation are underway in several laboratories.

Since efforts to achieve larger area, surface passivated, high efficiency InF cells have only begun to be seriously pursued during the past year, they represent a new direction in InP solar cell research. In fact, since areas of 8 cm² are the norm in GaAs cell production, further scale up of the present 4 cm², high efficiency InP cells is highly desireable.

5. CONCLUSION

When compared to previous homoepitaxial planar InP cell research, the efforts in heteroepitaxial growth, CLEFT, chemical lift off and V-grooved cells represent entirely new directions. Efforts aimed at larger area, surface passivated, high efficiency cells are not entirely new. However, because of their recent emergence, they represent new cirections in InP solar cell research. Although present efficiencies of the heteroepitaxial cells are admittedly low, the recent increase in efficiency from 9 to 13.7%, for cells based on GaAs substrates, demonstrates the progress which can be affected by the use of more appropriate transition layers. A similar demonstration, using silicon substrates, has yet to be achieved. However, assuming a solution of the problem presented by silicon autodoping, there appears to be no insurmountable obstacle toward achieving high efficiency cells using silicon as a substrate. InP cells produced by the lift-off processes have yet to be demonstrated. Successful efforts will result in lower cost, lightweight, high efficiency cells, qualities which make them attractive to the spacecraft array designer. The few V-grooved cells, produced to date, although demonstrating higher currents, are of relatively low efficiency. Further development is needed here in the area of cell processing. In fact, it appears necessary to use epitaxial growth rather than diffusion, into Czochralski substrates, to produce the cells. The drive toward larger area cells has shown that scale up is feasible. In fact, the 19.1%, 4 cm² cells produced to date are slightly better than the small area cells which demonstrated the previous record high efficiencies. Present efforts, directed at surface passivation, should result in efficiencies greater than 20%.

In summation, despite the relatively few laboratories engaged in InP solar cell research, the field is dynamic in its exploration of new concepts. Undoubtedly, high efficiencies comparable to GaAs will result from this activity. Furthermore, given sufficient time, one would expect the resultant cells to be low cost, lightweight, and with improved strength to weight ratios. Successful completion of this research effort will result in cells which are prime candidates for extensive use on satellites which experience degrading space radiation environments.

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Cell	Jsc	Voc	FF	Efficiency
	ma/cm ²	mv	%	2
InP/GaAs ^a	26.6±.7	667±4	69±1	8.9±.1
InP/GaAs/Si ^a	25.8±.3	642±4	57.7±1	6.95±.1
prodicted	29.6	600	77	9.9

Table II Parameters of High Efficiency 4 cm ² InP Cell ²						
Jsc		Voc	<u>FF</u>	Efficiency		
ma/cm ²		mY	r	2		
36.34		876	82.4	19.1		



Fig. 1. I-V curves of n/p homojunction and present heteroepitaxial cells



Fig. 2. Schematic of V-grocved cell

Fig. 3. Quantum efficiency-V-grooved and planar cell. (no antireflection coating)

600

PLANAR CELL

WAVELENGTH (NM)

800

1000

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