## Growth speed and thermal gradient dependence of primary dendrite trunk diameter in directionally solidified Al-Si alloys

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# Dendritic array morphology depends upon DS processing parameters: G<sub>1</sub>, R, C<sub>0</sub>, *Convection*



- Primary dendrite arm spacing (λ): Extensive literature (SCN/Metals)
- Secondary/tertiary arm spacing: Extensive-SCN/Metals
- Dendrite tip radius: SCN/ limited (Al-Cu, Pb-Au, Pb-Pd)
- 4. Primary dendrite trunk diameter (Φ): Limited (Esaka:Thesis-86, Grugel: 92/95)

## **Typical analysis of directionally solidified Al-7 wt% Si alloy samples** (Terrestrial: G<sub>I</sub>=41 Kcm<sup>-1</sup>, R=85 μm s<sup>-1</sup>, G<sub>m</sub>=51K cm<sup>-1</sup>)





Primary dendrite trunk diameter



Primary dendrite arm spacing???

## Primary dendrite trunk diameter ( $\phi$ )



D<sub>I</sub>G<sub>I</sub>k/(m<sub>I</sub>RC<sub>o</sub>(k-1))]: More branched dendritic morphologies will be located towards the left, and lessbranched/cellular towards the right side of the X-axis.

Esaka Thesis (1986): Trunk diameter increases rapidly near the tip till ~ 10 side-branch formations. He measured this initial trunk diameter ( $\phi_0$ ). Eighty DS experiments (four SCN-Acetone alloys grown with various R and G<sub>1</sub>)



(Initial trunk diameter ( $\phi_0$ )/tip radius) = 6.59±1.3

### Primary dendrite trunk diameter ( $\phi$ ) model

1. The trunk diameter ( $\phi$ ) increases rapidly near the tip till time,  $t_o = 22*r_t/R$ ), when  $\phi = \phi_o = 6.59 r_t$ (paraboloidal envelope near tip).



## Primary dendrite trunk diameter ( $\phi$ ) model

2. After  $t_o$  the trunk diameter increases via remelting of 4-side arms (r) and deposition of melted arm material on "trunk surface "over length h" =  $\phi$ .

#### **Assumptions:**

- 1. Kirkwood model (1985) of ripening applies.
- 2. Secondary arm melts back because of its curvature.
- 3. Mass of the melted arm deposits on trunk surface where there is negative curvature.



$$\pi \phi h \frac{d\phi}{2\,dt} = 4 * \pi r^2 \frac{dl}{dt} \qquad (2)$$

$$C_{l} = C_{o} + R G_{m} t/m_{l}$$
<sup>(3)</sup>

$$\phi^2 \frac{d\phi}{dt} = 32 \frac{D_l \Gamma}{m_l (1-k)(C_o + \frac{R Gm t}{m_l})} \quad (4)$$

## Primary dendrite trunk diameter (\*) model

$$\phi^{3} = 96 \frac{D_{l} \Gamma}{RGm \ (1-k)} \ln\{\frac{\left(1 + \frac{R \ Gm \ t}{m_{l} C_{o}}\right)}{\left(1 + \frac{R \ Gm \ t_{o}}{m_{l} C_{o}}\right)}\} + \Phi_{o}^{3}$$
(5)

### Mushy zone freezing time ~ $m_{I}(C_{E}-C_{o})/RG_{m}$

Use tip radius ( $r_t$ ) predicted from Trivedi (1980) or Hunt-Lu (1996) models to get the initial trunk diameter  $\phi_o = 6.59 r_t$  in order to predict the processing parameter dependence of "Primary dendrite trunk diameter" from above relationship.



Equation 4 has a reasonable fit with experimentally observed solute content and growth speed dependence ( whether we use  $r_t$  predictions from Trivedi or Hunt-Lu).



Equation 4 has a reasonably good fit with experimentally observed thermal gradient and growth speed dependence (whether we use  $r_t$  values from Trivedi or Hunt-Lu).

## Trunk diameters measured in quenched mushy-zone

(Al-Si alloys:  $G_1=150 \text{ K cm}^{-1}$ , velocity = 43  $\mu \text{m s}^{-1}$ ) Lines are predictions from Eq: 5 using  $\mathbf{r}_t$  (Trivedi)



Coarsening time, s

Trunk diameters in the mushy-zone are greater than those expected from the Trunk diameter model, especially near the array tips.

**Trunk diameter measured in quenched mushy-zone** (Al- Si alloys: G<sub>I</sub>=150 K cm<sup>-1</sup>, velocity = 156 μm s<sup>-1</sup>) Lines are predictions from Eq: 5 using **r**<sub>t</sub> (Trivedi)



Trunk diameters in the mushy-zone are greater than those expected from the Trunk diameter model.

## Trunk diameter measured in quenched mushy-zone

(Al-Si alloys:  $G_l=150 \text{ K cm}^{-1}$ , velocity = 43 µm s<sup>-1</sup>) Lines: Vary  $\mathbf{r}_t$  to obtain least-squared fit of the data to Eq: 5.



These r<sub>t</sub> values are larger than predicted from Trivedi/or Hunt-Lu models.

## Trunk diameter measured in quenched mushy-zone

Al-Si alloys: Growth speed 156  $\mu$ m s<sup>-1</sup> Lines: Vary **r**<sub>t</sub> to obtain least-squared fit of the data to Eq: 5.



These r<sub>t</sub> values are larger than predicted from Trivedi/or Hunt-Lu models.

The tip radii obtained by forcing a least squared fit of the observed trunk diameter vs. time data to the trunk-diameter coarsening equation are larger than the tip radii calculated from the Hunt-Lu or Trivedi models.

C <sub>o</sub> wt%	G <sub>l</sub> K/cm	G <sub>m</sub> K/cm	R µm/s	r <sub>t</sub> _HL μm	r <sub>t</sub> _Trivedi μm	r <sub>t</sub> from best fit least squared analysis μm
6	150	106.4	43	4.28	5.48	7.22
8	150	125	43	3.41	4.35	5.19
10	150	150	43	3.06	3.9	4.73
6	150	106.4	156	2.21	2.75	4.49
8	150	125	156	1.76	2.21	3.5
10	150	150	156	1.58	1.99	2.29

Does natural convection during terrestrial directional solidification increase dendrite trunk diameter (dendrite tip radius?)

## **Comparison of microstructures: Al-7% Si directionally** solidified on ground and on ISS (MICAST6)

**MICAST6 SEED 41** K cm<sup>-1</sup>, **22** μm s<sup>-1</sup>



### **Terrestrial DS:** 15 K cm<sup>-1</sup>

**Convection causes dendrite** clustering (steepling) at low thermal gradient and growth speeds during terrestrial DS.

 $5 \ \mu m \ s^{-1}$ **50 μm s<sup>-1</sup> 50 μm s<sup>-1</sup>** 15

MICAST6: 20 K cm<sup>-1</sup>

**5 μm s<sup>-1</sup>** 

# Comparison of microstructures: AI-7% Si directionally solidified on ground and on ISS (MICAST7)

MICAST7 SEED 41 K cm<sup>-1</sup>, 22 μm s<sup>-1</sup>



**21** µm s<sup>-1</sup>

MICAST7: 26 K cm  $^{-1}$  11  $\mu m$  s  $^{-1}$ 



Terrestrial DS: 24 K cm<sup>-1</sup> →



#### 23 µm s<sup>-1</sup>

**10 µm s<sup>-1</sup>** 16

## Primary dendrite trunk diameter as compared to



ISS-DS: Good agreement with predictions from the trunk-diameter model.
 Terrestrial DS ("Not steepled"): Good agreement with predictions from model.
 Terrestrial DS ("steepled"): Convection <u>increases</u> trunk diameter.

### ISS samples show better agreement with calculations from the models than terrestrial samples (primary dendrite arm spacing and trunk diameter)

	Trivedi					
	ISS-samples	Terrestrial (no steepling)	Terrestrial (steepling)			
Primary dendrite arm spacing/calculated from model	0.945± 0.0833	0.791± 0.0931	0.695± 0.223			
Primary dendrite trunk diameter/calculat ed from model	1.069± 0.0361	1.113±0.0890	1.513±0.560			

### Natural convection decreases primary dendrite arm spacing and increases primary dendrite trunk diameter in Al-26.5 % Cu

(M.D. Dupouy, D. Camel and J.J. Favier, Acta. Metall. Mater. Vol. 37, No. 4, pp. 1143-1157, 1989)

Al-26.5 wt% Cu, 30 K cm<sup>-1</sup>, **4.2** μm s<sup>-1</sup>



Al-26.5 wt% Cu, 25 K cm<sup>-1</sup>, **4.2** μm s<sup>-1</sup>



**Terrestrial: Solutally stable**, thermally stable mode

Trunk diameter 🗲 120 ± 18 μm

 $\sqrt{A/(N-1)}$ 

**122 ± 18 μm** 

Al-26.5 wt % Cu, 30 K cm<sup>-1</sup>,

**4.2** μm s<sup>-1</sup>

## Conclusions

- Primary dendrite trunk diameters in a range of Al-Si alloys directionally solidified under varying thermal gradients and growth speeds shows a reasonable fit with a simple analytical model (based on Kirkwood's approach) proposed here.
- Primary dendrite trunk diameters of Al-7 wt% Si alloy directionally solidified on the ISS show a very good fit with the analytical model.
- Natural convection which causes radial in-homogeneity (dendrite clustering) in these alloys appears to increase primary dendrite trunk diameter.
  - decreases primary dendrite arm spacing.

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## Microgravity Processing : Partially remelt and then DS from

terrestrially grown dendritic mono-crystal in µg.



ESA\_MSL Low Gradient Furnace

NASA\_MSSR-1 Flight Rack

## **Terrestrial processing**

#### Graphite crucible (~9 mm ID, ~19 mm OD), 10<sup>-4</sup> torr vacuum



## **Microgravity Processed Sample MICAST 7**



**Eutectic Melt Back** 

Alumina crucible

# MICAST6: ESA-Low Gradient Furnace (1-hr heat-up, 5-hr hold, $G_1$ ~ 20 K cm<sup>-1</sup>): 3.8 cm at 5 $\mu$ m s<sup>-1</sup>, 11.3 cm at 50 $\mu$ m s<sup>-1</sup>



# MICAST6: ESA-Low Gradient Furnace (1-hr heat-up, 5-hr hold, $G_1$ ~ 20 K cm<sup>-1</sup>): 3.8 cm at 5 $\mu$ m s<sup>-1</sup>, 11.3 cm at 50 $\mu$ m s<sup>-1</sup>

isotherm velocity vs. position along the Al-Si rod (note: both melting and solidification are shown)



## MICAST7: ESA-SQF (1-hr heat-up, 1-hr hold ( $G_{l} \sim 26$ K cm<sup>-1</sup>): 8.4 cm at 20 $\mu$ m s<sup>-1</sup>, 6.5 cm at 11 $\mu$ m s<sup>-1</sup>



## MICAST7: ESA-SQF (1-hr heat-up, 1-hr hold ( $G_{l} \sim 26 \text{ K cm}^{-1}$ ): 8.4 cm at 20 $\mu$ m s<sup>-1</sup>, 6.5 cm at 11 $\mu$ m s<sup>-1</sup>



## Growth conditions for MICAST6 and MICAST 7 transverse microstructures examined

Sample ID	G <sub>I</sub> , K cm <sup>-1</sup>	G <sub>m</sub> , K cm <sup>-1</sup>	R, μm s <sup>-1</sup>
MICAST6-1	19	18	52
MICAST6-11	20	18.5	47
MICAST6-9	21	19.3	34
MICAST6-7	22.8	20.4	5
MICAST7-3T	26	24	20
MICAST7-4T	26	24	11
MICAST7-5T	26	24	11