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Oxidation s

tate of iron

in hydrous phono-tephritic melts

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23   ABSTRACT

24

25   The oxidation state of iron in hydrous ultrapotassic (phono-tephritic) melts

26   coexisting with mixed H_2O-CO_2 fluids was experimentally studied at 1200 and 1250°C

27   and pressures from 50 to 500 MPa. The oxygen fugacity (f_{O_2}) varied from NNO–2.9 to

28   NNO+2.6 in $\log f_{O_2}$, relative to the Ni-NiO oxygen buffer (NNO), as imposed by

29   external redox conditions in experimental vessels and internal variations in water activity

30   from 0.05 to 1 inside the capsules. The iron redox state of the quenched melts was

31   determined by colorimetric wet-chemical analysis. This analytical method was optimized

32   to measure the $Fe^{2+}/\Sigma Fe$ ratio of mg-sized samples within ±0.03 (2σ). The accuracy and

33   precision was tested with international reference materials and with standards analyzed

34   by other methods. The $Fe^{2+}/\Sigma Fe$ ratio of the experimental glasses covered a range of 0.41

35   to 0.85. A small negative effect of dissolved water on $Fe^{2+}/\Sigma Fe$ at given f_{O_2} was found,

36   consistent with the thermodynamic model of Moretti (2005). No effect of pressure and

37   temperature on the redox state of iron was resolvable in the investigated P-T range.

38   Compared to hydrous ferrobasaltic melts that were studied previously under similar

39   conditions, systematically lower $Fe^{2+}/\Sigma Fe$ ratios were found for the phono-tephritic melts,

40   in particular at low oxygen fugacities. This effect is attributed to the much higher K_2O

41   contents of the phono-tephrite (7.5 compared to 0.3 wt%), but the difference in ΣFeO (7.8

42   wt% in the phono-tephrite and 12.9 wt% in the ferrobasalt) may have an influence as

43   well. Comparison of the experimentally obtained relationship between $\log f_{O_2}$ and

44   Fe^{3+}/Fe^{2+} for the studied hydrous ultrapotassic melts with commonly used empirical and

45   thermodynamic models suggest that these models can be successfully applied to phono-

46   tephritc melts, although such compositions were not implemented in the model

47   calibrations. Furthermore, the new data can be used to improve the models with respect to

48   the effects of compositional variables, such as H_2O or K_2O, on the redox state of iron in

49   silicate melts.

50   *Keywords: iron oxidation state, ferrous iron determination, oxygen fugacity, water*

51   *activity, phono-tephrite, ultrapotassic hydrous silicate melt, Alban Hills*

INTRODUCTION

52

53   The oxidation state of iron varies widely in natural magmas (e.g., Carmichael,

54   1991) and influences their physical and chemical properties as well as the phase

55   equilibria for iron-bearing minerals. The redox state of iron in the melt is related to the

56   oxygen fugacity via the reaction Fe(II)O $_{melt}$ + ¼ O$_{2\ gas}$ = Fe(III)O$_{1.5\ melt}$ and the

57   equilibrium constant of the reaction can be expressed as $K = a_{Fe(III)O_{1.5}} / (\, a_{Fe(II)O} \cdot (a_{O_2})^{\frac{1}{4}}\,)$

58   where a denotes the activity of the respective components in the melt. K depends on

59   temperature, pressure and melt composition. Knowledge of the Fe redox state in magmas

60   is a pre-requisite to understand the physical and chemical properties of magmas, to

61   constrain source regions of magmas and their redox states, and the processes occurring

62   during magma genesis and evolution. Specifically, the redox state of iron has influence

63   on the stability of iron bearing minerals that may crystallize during magma evolution, and

64   hence, control the iron content of the residual melt, and the stability and composition of

65   major silicate phases (e.g., Pichavant et al., 2002). Furthermore, the structural

66   incorporation of ferric and ferrous iron in silicate melts may affect the viscosity of a

67   magma (e.g., Liebske et al.; 2003; Vetere et al., 2006), which has strong influence on the

68   dynamics of volcanic eruptions. In particular, interaction between various species of

69   different elements may influence the partitioning of volatile elements between silicate

70   melts and coexisting gas phases, which strongly affects the degassing behavior of

71   ascending magmas (e.g., Moretti and Ottonello, 2003; Moretti and Papale, 2004;

72   Burgisser and Scaillet, 2007).

73   Several empirical relations have been proposed to quantify the effect of various

74   parameters on the Fe redox state in silicate melts and to predict the prevailing oxygen

75   fugacity in a magmatic system from Fe redox ratios of quenched melts (e.g., Sack et al.,

76   1980; Kilinc et al., 1983; Kress and Carmichael, 1988; Mysen, 1988; Borisov and

77   Shapkin, 1989; Kress and Carmichael, 1991; Nikolaev et al., 1996; Jayasuriya et al.,

78   2004). The early empirical models were calibrated over a relatively wide range of melt

79   compositions, temperatures and oxygen fugacities, but the data basis comprised dry

80   silicate melts only. Subsequent experimental investigations on silicate melts of different

81   chemical compositions and also on hydrous silicate melts revealed in part considerable

82   discrepancies between predicted Fe^{2+}/Fe^{3+} ratios of the melts and experimental findings

83   (Sisson and Grove, 1993; Moore et al., 1995; Baker and Rutherford, 1996; Gaillard et al.,

84   2001; Wilke et al., 2002; Gaillard et al., 2003; Partzsch et al., 2004; Botcharnikov et al.,

85   2005). The deviations might be either due to ignoring the component H_2O in the

86   empirical models or due to differences in anhydrous melt compositions studied in the

87   experiments compared to the compositions used to calibrate the models. As an

88   alternative, a thermodynamic model based on a polymeric approach was developed for

89   the prediction of the Fe redox state in dry silicate melts at atmospheric pressures by

90   Ottonello et al. (2001). Recently, this model was extended by Moretti (2005) to account

91   for the effects of dissolved water and pressure.

92   Here we use samples from a previous study on volatile solubility in phono-

93   tephritic melts (Misiti et al., this issue) to test the predictive power of commonly used

94   models for Fe^{2+}/Fe^{3+} ratios in ultrapotassic silicate melts because hydrous phono-tephritic

95   melts have not been used in the calibration of those models. A positive effect of

96   increasing K_2O on the $Fe^{2+}/\Sigma Fe$ has been proposed by Tangeman et al. (2001) for dry

97   iron-rich K_2O-FeO-Fe_2O_3-SiO_2 liquids at atmospheric pressure, whereas other studies

98 suggest a stabilization of tetrahedrally coordinated ferric iron by charge-balancing K_2O (

99 Sack et al., 1980; Kilinc et al., 1983; Dickenson and Hess, 1986; Kress and Carmichael,

100 1988). To date, no experimental data exists on hydrous K_2O-rich melts at elevated

101 pressures. The new data allow us to investigate the influence of water activity and oxygen

102 fugacity on the redox state of iron in the melts as well as to evaluate the compositional

103 effects of water, potassium and total iron content on the $Fe^{2+}/\Sigma Fe$ ratio. The results are

104 compared to the widely used empirical model of Kress and Carmichael (1991) and the

105 thermodynamic model of Moretti (2005).

106

107 <center>**EXPERIMENTAL METHODS**</center>

108   The experimental strategies and procedures are described in detail in Misiti et al.

109 (this issue) and are summarized briefly here. The starting material for the experiments

110 was a synthetic analogue of the phono-tephritic Mt. Mellone lava flow composition from

111 the Alban Hills Volcanic District in Central Italy (Marra et al., 2003; Gaeta et al., 2006).

112 For each experiment ~50 mg glass powder, 0 to 20 µL deionised water and 0 to 15 mg

113 silver oxalate ($Ag_2C_2O_4$) were sealed in $Au_{80}Pd_{20}$ capsules (~15 mm length, 2.6 mm inner

114 diameter, 0.2 mm wall thickness). To reduce Fe loss to the capsule walls in the

115 experiments under reducing conditions the capsules were pre-saturated with Fe as

116 described in Botcharnikov et al. (2005). The experiments were performed in internally

117 heated gas pressure vessels (IHPV) at temperatures of 1200 and 1250°C and pressures

118 between 50 to 500 MPa for 1.5 to 72 hours (Table 1). Uncertainties in temperature and

119 pressure were ±10°C and ±5 MPa, respectively. Samples were rapidly quenched at the

120 end of the experiments with an initial cooling rate of about 150°C/s (Berndt et al., 2002).

<center>5</center>

121   Most of the experiments were performed at intrinsic redox conditions of the IHPV

122   pressurized with Ar. The intrinsic oxygen fugacity in capsules with pure H_2O fluid (mole

123   fraction of water in the fluid $X^f_{H_2O} = 1$) in the IHPV used in this study was determined by

124   NiPd-solid sensors (Taylor et al., 1992) at 1200°C and 200 MPa. The obtained value of

125     $\log f_{O_2} = -7.5$ corresponds to NNO+2.6 (±0.5; 1σ from microprobe analyses of the NiPd

126   alloy) where NNO refers to the Ni-NiO buffer (Huebner and Sato, 1970). This f_{O_2} value is

127   about 0.9 log units lower than reported by Berndt et al. (2002) for a similar IHPV. The

128   difference reflects the uncertainty in f_{O_2} due to unbuffered hydrogen fugacity at intrinsic

129   conditions in the IHPV, i.e., the prevailing f_{H_2} depends on the specific components used

130   in the individual IHPV (furnace, sample holder, etc.).

131   Experiments at low f_{O_2} were performed in another IHPV pressurized with an Ar-

132   H_2 mixture. The IHPV is equipped with a Shaw-membrane to monitor the f_{H_2} at high

133   pressure and temperature (Berndt et al., 2002). The f_{H_2} controls the f_{O_2} in the capsule

134   through the equilibrium reaction $H_2 + \frac{1}{2} O_2 \leftrightarrow H_2O$. The accuracy of $\log f_{O_2}$ is estimated

135   to be ±0.2 log units for experiments with pure H_2O fluids. In experiments with mixed

136   H_2O-CO_2 fluids the fugacity of H_2O and hence the f_{O_2} is decreasing with increasing f_{CO_2}.

137   The prevailing f_{O_2} in the capsule was calculated from the fluid composition determined

138   after the experiment. Hence, differences in oxygen fugacity in capsules processed in the

139   same run are determined only by the differences in $X^f_{H_2O}$. The precision of the latter is

140   limited by the uncertainties associated with the weight-loss determination of H_2O and

141   CO_2 released from the capsules after the experiments. In this case the relative precision of

142   log f_{O_2} for experiments processed in the same run was estimated from error propagation

143   of weighing uncertainties to be approximately ±0.1 log units. However, taking the

144   uncertainty of the intrinsic redox condition in the IHPV into account, the error in absolute

145   log f_{O_2} values of experiments processed in different runs is higher (approximately ±0.5

146   log units).

147

148   **ANALYTICAL METHODS**

149   The composition of the fluid phase in equilibrium with the silicate melt (expressed

150   in mole fractions of H_2O and CO_2, $X^f_{H_2O}$ and $X^f_{CO_2}$, respectively) was determined by a

151   conventional weight-loss technique. H_2O and CO_2 concentrations in the glasses were

152   measured by FT-IR spectroscopy. Bulk H_2O contents of quenched melts were also

153   measured by Karl-Fischer titration (KFT). Analytical details and results of these

154   investigations are reported in Misiti et al. (this issue). Here, we focus on measurements

155   relevant for the oxidation state of iron in the melt. The chemical compositions of the post-

156   experimental glasses were determined by electron microprobe analysis (EMPA). The

157   redox state of iron ($Fe^{2+}/\Sigma Fe$) was analyzed using a wet-chemical technique, which is

158   based on the colorimetric method of Wilson (1960). Both techniques are described below.

159

160   **Electron microprobe analysis**

161   Glass fragments from representative samples were mounted in epoxy and polished

162   for electron microprobe analysis. Analytical conditions were 5 nA, 15 kV and a beam

163   diameter of 20 μm, with counting times of 8 s for Si, Al, Fe, Mg, Ca, Mn and Ti on the

164   peak and 4 s on the background and 4 s for K and Na on the peak and 2 s on the

165   background to minimize the loss of alkalis. Cameca supplied standards were used for

166   calibration and PAP matrix correction according to Pouchou and Pichoir (1991) was

167   applied. Between 8 and 20 spot analyses were made on each sample to check for

168   homogeneity of the glass compositions and analytical reproducibility. The results are

169   listed in Table 2.

170

171   **Ferrous iron analyses**

172   Different wet-chemical techniques have been developed to determine the redox

173   state of iron in geological materials, where most of them employ titration methods for

174   quantification. Conventional techniques usually involve the acid dissolution of 100 to 500

175   mg of powdered sample material and subsequent precise determination of the absolute

176   ferrous iron concentration. To obtain the $Fe^{2+}/\Sigma Fe$ ratio, the total iron concentration is

177   commonly determined by an additional method, i.e., electron microprobe analysis or

178   optical emission spectroscopy (ICP-OES). In experimental studies the amounts of

179   samples are often limited to <50 mg and this material is subjected to several different

180   analytical methods. This limits the amount of sample available for the wet-chemical

181   ferrous iron determination. Here, we follow the colorimetric method of Wilson (1960) to

182   measure the $Fe^{2+}/\Sigma Fe$ ratio in mg-sized samples after acid dissolution. The original

183   method was modified to minimize handling of toxic materials (i.e., beryllium sulfate was

184   replaced by boric acid, see below) and to improve the reproducibility. To asses the

185   accuracy and precision of this method, we have analyzed international reference materials

186   and in-house standards.

187   The critical point in ferrous iron analysis is to avoid an oxidation of Fe^{2+} during

188   the analytical procedure. Here, samples are decomposed in a $HF-H_2SO_4$ mixture in

189   presence of excess pentavalent vanadium, which oxidizes ferrous iron as soon as it is

190   released from the sample. According to the reaction

191                    $\overset{+5}{V}O_2^{+}+2H^{+}+Fe^{2+} \rightleftharpoons \overset{+4}{V}O^{2+}+H_2O+Fe^{3+}$ (1)
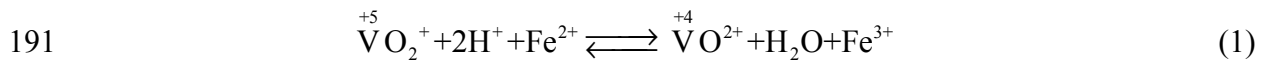
192   the amount of generated tetravalent vanadium, which is highly resistant to oxidation

193   compared to Fe^{2+}, is equivalent to the amount of Fe^{2+} in the sample. The equilibrium of

194   the reaction is shifted to the right hand side under the strongly acidic conditions during

195   sample dissolution (pH < 1). After complete sample dissolution ferrous iron is

196   regenerated from tetravalent vanadium by increasing the pH value to ~5, which shifts the

197   equilibrium of reaction (1) to the left hand side.

198   The analytical procedure is as follows (employed reagents are listed in

199   Appendix 1): The powdered sample (in case of references materials) or sub-millimeter-

200   sized glass fragments (from experimental products) were weighed into a 15-mL Savillex$^{®}$

201   Teflon beaker containing 1 mL of an ammonium vanadate solution dissolved in sulfuric

202   acid ($1M$ to $5M$ H_2SO_4). After addition of 1 mL HF (24 or 48%), the beakers were tightly

203   sealed and placed in an ultrasonic bath for about 15 minutes. Thereafter, the beakers were

204   left for 3 to 24 hours at temperatures between 20 to 100°C until complete sample

205   dissolution was attained. Acid concentrations, dissolution time and temperature were

206   systematically varied to test for potential effects of these parameters on the analytical

207   results (see discussion below). After sample dissolution, 5 mL saturated hot boric acid

208   solution (at ~80°C) was added, instead of beryllium sulfate as proposed by Wilson

209   (1960), to neutralize excess HF and to bring possibly formed fluorides back into solution.

210   Upon cooling to room temperature the content of the beaker was quantitatively

211   transferred into a 100-mL volumetric flask, containing 10 mL ammonium acetate

212   solution, 5 mL 2:2'bipyridyl solution, and the remaining volume was filled with distilled

213   water. The ammonium acetate buffer adjusted the pH value to ~5. The regenerated Fe^{2+}

214   forms a very stable complex with 2:2'bipyridyl in the solution which shows an intensive

215   absorption band in the visible spectrum (Fig. 1).

216   Measurements of ferrous Fe and total Fe were made on the same solution before

217   and after adding 5 to 10 mg solid hydroxylamine hydrochloride to an aliquot of about 10

218   mL. This reducing agent converts all ferric Fe into the ferrous state. Since both Fe^{2+} and

219   total Fe are measured in the same solution, the $Fe^{2+}/\Sigma Fe$ ratio can be directly calculated

220   by dividing the absorbances of the Fe^{2+} and total Fe aliquots. The advantage compared to

221   an absolute concentration measurement of ferrous iron and an additional total Fe

222   determination by another method (e.g., EMPA or ICP-OES) is that uncertainties in the

223    $Fe^{2+}/\Sigma Fe$ ratios arise mainly from the spectrometric measurements, whereas weighing and

224   dilution errors cancel out. Absolute concentrations were obtained as well after calibration

225   of the spectrometric technique using ferrous ammonium sulfate solutions of different

226   known Fe^{2+} concentrations. For all measurements 1 cm transmission cells and an UV/VIS

227   spectrometer (Zeiss Specord S10) was used. The sample solutions show a characteristic

228   absorption band of the Fe(II)-2:2'bipyridyl complex at about 523 nm (Fig. 1). The

229   maximum peak height was determined relative to a baseline measured at 700 nm. No

230   differences in the general appearance of the spectra and in the maximum peak position

231   were observed between samples of different matrices, i.e. basaltic to rhyolitic rocks or

232   pure Fe(II) solutions.

233   The results of the wet-chemical colorimetric iron analyses on international

234   reference materials and in-house standards are given in Table 3. To asses the accuracy of

235   the method, our results are compared to the recommended values obtained by other

236   studies (Govindaraju, 1994; Govindaraju, 1995; Liebske et al., 2003; Bertoldi et al.,

237   2007). There is good agreement between the recommended $Fe^{2+}/\Sigma Fe$ ratios and the values

238   obtained by this study (Fig. 2). Furthermore, the results for our in-house standard PU-3

239   are in good agreement with measurements reported by Liebske et al. (2003). They

240   analyzed a synthetic andesitic glass similar in composition and synthesis conditions to

241   our PU-3 sample (1600°C, air atmosphere). Noteworthy, the $Fe^{2+}/\Sigma Fe$ ratios obtained for

242   the granites GS-N and GA are significantly higher than the recommended values.

243   Furthermore, the ΣFeO values (ΣFeO refers to total iron expressed as wt% FeO) of those

244   samples are also systematically lower than the recommended values. The presence of

245   undissolved refractory minerals containing significant amounts of ferric iron may explain

246   the discrepancy for these particular samples. The experimental products analyzed in this

247   study consist primarily of glass, which is readily dissolved within a few hours at room

248   temperature. Thus, no attempts were made to optimize the method for analyses of highly

249   resistant minerals, although this would be principally possible given some minor

250   modifications and tests.

251   For the two reference materials, natural olivine and commercial ammonium

252   iron(II) sulfate hexahydrate, the expected $Fe^{2+}/\Sigma Fe$ ratios are close to unity (Table 3). The

253   measured values of 0.93±0.08 and 0.95±0.05 (2σ), respectively, are systematically lower,

254   while for the chlorite sample CA the reported value of 0.90±0.02 from Bertoldi et al.

255   (2007) is still well reproduced by our measurements (0.89±0.04). To test whether

11

256   exclusion of atmospheric oxygen improves the recovery yield, some samples were

257   dissolved under Ar atmosphere, but no difference in the $Fe^{2+}/\Sigma Fe$ ratios was found.

258   Whipple (1974) and later Yokoyama and Nakamura (2002) noted that variable

259   concentrations of sulphuric acid and hydrofluoric acid may affect the accuracy of the

260   measured $Fe^{2+}/\Sigma Fe$ ratio as well. We have varied acid concentrations in the range from

261   1M to 5M H_2SO_4 and 24% to 48% HF, respectively, but did not observe any systematic

262   bias of the results. The potential effect of sample decomposition temperature on the

263    $Fe^{2+}/\Sigma Fe$ ratio was also studied (Table 3), since the dissolution kinetics at room

264   temperature could be more sluggish for some rock samples. Results obtained from

265   samples dissolved at 25°C are indistinguishable from those at 100°C, except for the

266   granitic sample GS-N, as discussed above.

267   The precision of the method can be evaluated from replicate measurements (Table

268   3). Reproducibility was between ±0.01 and ±0.05 (2σ) in the $Fe^{2+}/\Sigma Fe$ ratio for different

269   rocks and minerals containing between 1 and 8 wt% ferrous FeO. The long term

270   reproducibility (over a time period of about one year) was assessed from n = 33 replicate

271   analyses of the synthetic andesitic glass PU-3. Based on these measurements the

272   uncertainty of the $Fe^{2+}/\Sigma Fe$ ratios is ±0.03 (2σ, external precision). Despite the low

273   sample mass used for wet-chemical analyses, the obtained ΣFeO agree well with electron

274   microprobe analyses for experimental products and recommended values for the

275   reference materials, most of them within <5% relative (Tables 1, 2 and 3). Thus, we

276   conclude that the $Fe^{2+}/\Sigma Fe$ ratio can be reliably determined in the range from 0.4 to 0.9

277   by our method. All $Fe^{2+}/\Sigma Fe$ ratios of our experimental run products fall within this

278   range. Procedural blanks were always below the detection limit of the method, i.e. <0.012

279 absorbance units (3σ of the background), which corresponds to less than 1 μg Fe. This

280 can be considered negligible relative to the processed amount of iron (100 to 500 μg Fe),

281 since the maximum bias in the measured $Fe^{2+}/\Sigma Fe$ would be <0.01.

282

283   **RESULTS**

284   Except for one sample (Alb1-1) with very low water content, which was probably

285 at the liquidus and thus partially crystallized, all experiments with less than 6 wt% of

286 dissolved water in the melt yielded only glass and a fluid phase as experimental products.

287 All melts with higher water contents contained crystals after quench. These crystals show

288 typical features of non-equilibrium growth and were probably formed during quenching.

289 The crystals were too small for a reliable quantitative microprobe analysis but semi-

290 quantitative results, obtained by energy dispersive x-ray analyses (EDX), indicate a

291 K-rich, Fe-bearing composition. X-ray powder diffraction on sample Alb1-5 gives

292 evidence that the quench phases are mica. In experimental studies investigating basaltic

293 melts under similar conditions and with similar experimental equipment no quench

294 crystals were observed (e.g., Berndt et al., 2002; Botcharnikov et al., 2005). This

295 discrepancy may be explained by the high potassium content of the phono-tephrite

296 facilitating K-rich mica crystallization from a H_2O-rich melt of relatively low viscosity

297 during cooling.

298   The results of electron microprobe analyses of representative post experimental

299 glasses are given in Table 2. Sample Alb1-5 consists of glass and numerous quench

300 crystals. Thus, the analyses were performed using a defocused electron beam (20 μm)

301 and represent the bulk composition. After normalizing to a sum of 100 wt% (i.e.,

13

302   anhydrous composition) most of the glass compositions are identical to the starting glass

303   (Alb1) and electron microprobe analyses show homogeneous ΣFeO concentrations in the

304   glasses. Furthermore, from these data no indication for extensive dissolution of cations

305   from the melt into the fluid phase is given. However, some samples (Alb1-1, Alb1-6a,

306   Alb1-6bis, Alb1-H45, Alb1-H47 to Alb1-H51, Alb1-H56) have significantly lower ΣFeO

307   (compare ΣFeO norm in Table 2 for samples with different water contents). These

308   experiments were carried out under the most reducing conditions and at the lowest a_{H_2O}.

309   At these conditions iron from the samples was partly dissolved as metallic Fe alloy in the

310   capsule walls. As a consequence of the reduction of ferric and ferrous iron from the melt

311   a small amount of oxygen was produced which reacted with hydrogen permeating from

312   the pressure medium into the capsule forming some additional H_2O. The generated H_2O

313   might have continuously increased the water activity and hence the oxygen fugacity

314   within the capsule. The largest iron loss was observed for sample Alb1-H51 with a final

315   ΣFeO of ~4.2 wt%. The corresponding increase in water content of the system is

316   expected to be ~0.5 mg. Considering the masses of glass, fluid and the partitioning of

317   H_2O between fluid and melt, f_{O_2} might have been increased by about half a log unit

318   during the experiment. The rate of iron reduction is controlled most likely by the sluggish

319   diffusion of Fe in the silicate melt. Fe diffusion is much slower than water and hydrogen

320   diffusion in the melt (Gaillard et al., 2002; Behrens et al., 2004; Watson and Baxter,

321   2007) and, therefore, we suggest that Fe^{2+}/ΣFe is in near-equilibrium with the oxygen

322   fugacity imposed by the fluid via the reaction $2\,Fe(II)O + H_2O = Fe(III)_2O_3 + H_2$.

323

324

325   **Redox state of iron**

326   The results of Fe redox analyses of the experimental run products are given in

327   Table l. The measured $Fe^{2+}/\Sigma Fe$ ratios range from 0.41 at 1250°C, 200 MPa, NNO+2.6

328   (Alb1-15) to 0.85 at 1200°C, 200 MPa, NNO-2.9. A comparison of run P and run Q at

329   1200°C, 200 MPa with durations of 5 and 1.5 hours, respectively, reveals consistent

330   results in terms of the redox state of iron. This finding and the homogeneous Fe

331   concentration in the run products (Table 3) suggest that the system is close to

332   equilibrium, in terms of both chemical and redox equilibrium, after 1.5 hours at 1200°C.

333   This is also consistent with a study on Fe redox kinetics in peralkaline hydrous rhyolitic

334   melts (Gaillard et al., 2002), revealing that redox kinetics are fast enough to equilibrate

335   the melt within 3 hours at 800°C, but slow enough to readily quench the $Fe^{2+}/\Sigma Fe$ ratio of

336   the melt in the experiments.

337   As mentioned above some samples are partially crystallized. Since the $Fe^{2+}/\Sigma Fe$

338   analyses were done upon complete dissolution of fragments of the experimental products

339   (including quench crystals) they represent bulk values for the quenched melts, assuming

340   that the bulk $Fe^{2+}/\Sigma Fe$ of the system is not significantly altered due to crystallization of

341   Fe bearing mica during cooling. Despite an almost instantaneous permeation of H_2 from

342   the pressure medium through the capsule walls and a transfer of Fe from the melt to the

343   crystals during cooling, at an initial quench rate of about 150°C/s only a few seconds

344   remain for a potential re-equilibration of the melt until the kinetics of the systems can be

345   considered as virtually frozen, i.e., at T <500°C. During this time interval no significant

346   change in the bulk $Fe^{2+}/\Sigma Fe$ ratio is expected from redox kinetics (Gaillard et al., 2002)

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347 and thus the measured Fe redox ratios are considered to represent close-to-equilibrium

348 values.

349   As shown in Figure 3, the $Fe^{2+}/\Sigma Fe$ ratio decreases nonlinearly with increasing

350 mole fraction of water in the coexisting fluid phase ($X^{f}_{H_2O}$). The change in $Fe^{2+}/\Sigma Fe$

351 with $X^{f}_{H_2O}$ is more marked at oxygen fugacities >NNO-0.2 compared to the series at

352 <NNO-0.2. At constant temperature, pressure, $X^{f}_{H_2O}$ and water content, the $Fe^{2+}/\Sigma Fe$

353 ratio increases with decreasing f_{O_2} (e.g., compare experiments at 200 MPa, 1200°C and

354 $a_{H_2O} = 1$, Table 1).

355

356   DISCUSSION

357 **Influence of oxygen fugacity and dissolved water on the speciation of iron**

358   The main factors controlling the oxygen fugacity in the system are the water

359 activity in the capsule and the hydrogen fugacity (f_{H_2}) in the IHPV at given P, T and

360 melt composition. Permeability of hydrogen through the capsule walls is high at our

361 experimental conditions, facilitating f_{H_2} equilibration between the capsule interior and

362 the vessel atmosphere. The time needed to permeate the amount of H_2 required for

363 reduction of all ferric iron in the starting glass (Alb1 $Fe^{2+}/\Sigma Fe = 0.29$) is less than two

364 minutes based on permeation data for Au tubes (Chou, 1986). The prevailing oxygen

365 fugacity within the capsule is mainly determined by the dissociation reaction of water

366 ($H_2O = H_2 + \frac{1}{2} O_2$) for which the equilibrium constant can be expressed as

367 $$K_w = f_{H_2O} \Big/ f_{H_2} \cdot \left(f_{O_2} \right)^{0.5} \tag{2}$$

16

368   and the logarithm of oxygen fugacity is given as
369
370                    $$\log f_{O_2} = 2\log f_{H_2O} - 2\log f_{H_2} - 2\log Kw \qquad (3)$$
371

372   K_w was derived from thermodynamic data of Robie et al. (1978). The water fugacity in

373   the capsule is the product of water activity (a_{H_2O}) and standard state water fugacity

374   ($f^0{}_{H_2O}$). If the experimental pressure is chosen as standard state, $f^0{}_{H_2O}$ equals the

375   fugacity of the pure H_2O fluid and the water activity is calculated as $a_{H_2O} = \gamma^f{}_{H_2O} \cdot X^f{}_{H_2O}$

376   where $\gamma^f{}_{H_2O}$ is the activity coefficient of H_2O in the fluid. Activity coefficients of H_2O for

377   mixed H_2O-CO_2 fluids were computed for given P, T, $X^f{}_{H_2O}$ after Aranovich and

378   Newton (1999) using molar volumes of pure H_2O and CO_2 from Pitzer and Sterner

379   (1994). These calculations are only valid assuming that H_2O and CO_2 are the dominant

380   species in the fluid, which is a reasonable assumption for most of our experimental

381   conditions (see Botcharnikov et al., 2006 and references therein). However, especially at

382   very reducing conditions other species (e.g., CO, H_2, CH_4) can become more abundant.

383   Except for sample Alb1-H56, no indication of an abrupt drop in CO_2 solubility in the

384   melts with decreasing f_{O_2} was observed (see Fig. 7 in Misiti et al., this issue), that would

385   indicate a change of the dominant carbon species in the fluid. The anomalously low CO_2

386   of sample Alb1-H56 concentration in the melt indicates a lower prevailing f_{CO_2} in the

387   capsule than calculated by mass balance. A value of ~0.38 is estimated for $X^f{}_{CO_2}$ from

388   the relation between CO_2 concentration in the melt and $X^f{}_{CO_2}$ in Fig. 7 of Misiti et al.

389   (this issue). Assuming that the additional fluid component is mainly CO (Holloway and

390   Blank, 1994), the fluid composition was recalculated to $X^f{}_{H_2O}$ ~0.04 and $X^f{}_{CO_2}$ ~0.58.

17

391   Based on the recalculated $X^f_{H_2O}$ the oxygen fugacity is NNO-2.9. For sample Alb1-1 the

392   fluid composition could not be determined and $X^f_{H_2O}$ was estimated from the H_2O-CO_2

393   saturation curve at 500 MPa and 1200°C (Misiti et al., this issue). It has to be noted that

394   the uncertainties in the so-derived molar fractions of the fluid components, and

395   consequently in the calculated oxygen fugacities, are probably higher compared to

396   samples where fluid compositions were directly measured by the weight-loss technique

397   (see Experimental Methods section).

398   The calculated oxygen fugacities are listed in Table 1. Additionally, the difference

399   relative to the Ni-NiO (ΔNNO) buffer is given (Huebner and Sato, 1970). This allows a

400   direct comparison of the experiments equilibrated at different temperatures and pressures

401   to evaluate the effect of f_{O_2} on the redox state of iron in the silicate melts. In Figure 4 the

402   iron redox state (expressed as Fe^{3+}/Fe^{2+} ratio) is plotted as a function of oxygen fugacity

403   (ΔNNO). The data show an almost linear trend. The scatter of the data is due to

404   superimposed variations in water content and experimental pressure as discussed below.

405   The experiment Alb1-1 deviates noticeably from the trend (Fig. 4). This sample with very

406   low water content (0.94 wt% H_2O) contains clinopyroxene crystals that may bias the

407   measured iron redox ratio. Therefore, this sample is not considered further on in the

408   systematics of the Fe redox state of the melt.

409   As shown in Figure 5, the $Fe^{2+}/\Sigma Fe$ ratios determined for the phono-tephritic

410   melts are in general agreement with the predictions of the models of Kress and

411   Carmichael (1991) and Moretti (2005). The dependence of the Fe^{3+}/Fe^{2+} ratio on log f_{O_2}

412   at pressures of 200 MPa and 500 MPa and temperatures of 1200°C and 1250°C is shown

413   in Figure 6. In such a plot, a slope of 0.25 is expected according to the reaction

414    $Fe(II)O_{melt} + ¼ O_{2\ gas} = Fe(III)O_{1.5\ melt}$. This slope is also implemented in the

415   thermodynamic model of Moretti (2005), whereas Kress and Carmichael obtained a slope

416   close to 0.2 for their emprical modelling. Considering the experiments done at relatively

417   oxidizing conditions (IHPV intrinsic; f_{H_2} ~0.6; Fig. 6a, b, d), the slope defined by the data

418   points at given P, T and f_{H_2} is 0.31 (1200°C, 500 MPa), 0.28 (1250°C, 500 MPa) and

419   0.37 (1200°C, 500 MPa), respectively, which is much larger than expected. This finding

420   is consistent with a positive dependence of the Fe^{3+}/Fe^{2+} ratio on the H_2O concentration

421   (Fig. 6d) in the melt as suggested by Moretti (2005). The data obtained at 1200°C, 200

422   MPa (Fig. 6c) define a much smaller slope of 0.16. Here, it has to be noted that due to

423   different f_{H_2} the oxygen fugacity is not directly correlated with the water fugacity over the

424   entire experimental f_{O_2} range.

425   A comparison of our data for phono-tephrite with the results of Botcharnikov et

426   al. (2005) for ferrobasalt studied at similar conditions can be used to evaluate the effect of

427   chemical composition on the redox state of iron in mafic melts. The phono-tephrite has a

428   much higher K_2O content (0.3 *vs.* 7.5 wt%) and a lower ΣFeO content (12.9 *vs.* 7.8 wt%)

429   than the ferrobasalt (Table 2). The comparison between our experimental dataset at

430   1200°C and 200 MPa and the data from Botcharnikov et al. (2005) for hydrous

431   ferrobasaltic melts obtained at the same P-T conditions (Fig. 6c) reveal slightly higher

432   Fe^{3+}/Fe^{2+} ratios for the ultrapotassic hydrous melts relative to the ferrobasaltic melts. This

433   effect is more pronounced at lower f_{O_2}. From linear regressions through each of the two

434   datasets the differences in $Fe^{2+}/\Sigma Fe$ ratios can be quantified. At a log f_{O_2} of -9 the

435    $Fe^{2+}/\Sigma Fe$ ratio of the phono-tephrite is 0.08 lower than that of the ferrobasalt. At more

436   oxidizing conditions of $\log f_{O_2} = -5$ this difference decreases to a value of 0.01. This trend

437   is consistent with the model of Kress and Carmichael (1991) which predicts a decrease in

438   the $Fe^{2+}/\Sigma Fe$ ratio with increasing K_2O. For an increase from 0.3 to 8 wt% K_2O

439   (ferrobasalt $vs.$ phono-tephrite) at 1200°C, 200 MPa and $\log f_{O_2} = -5$ a decrease in

440    $Fe^{2+}/\Sigma Fe$ by 0.05 is calculated. The corresponding increase at $\log f_{O_2} = -9$ is only 0.03.

441   Within this $\log f_{O_2}$ range and at water contents between 0 to 5 wt% in the melts, the

442   model of Moretti (2005) predicts no significant variation in $Fe^{2+}/\Sigma Fe$ with changing K_2O

443   (i.e., <0.003). If the ΣFeO content is reduced from 13 wt% to 8 wt% (ferrobasalt $vs.$

444   phono-tephrite), the decrease in the $Fe^{2+}/\Sigma Fe$ ratio calculated by the model of Kress and

445   Carmichael (1991) is 0.02 and 0.01 at $\log f_{O_2}$ of -9 and -5, respectively. Again, no

446   significant variation (<0.005) is predicted by the model of Moretti (2005). In conclusion,

447   both an increase of K_2O and a decrease of ΣFeO are predicted by Kress and Carmichael

448   (1991) to shift the Fe^{3+}/Fe^{2+} ratio in the same direction, whereas Moretti (2005) suggests

449   an insignificant change. It is difficult to clearly attribute the observed shift in Fe^{3+}/Fe^{2+} to

450   either K_2O or ΣFeO. A stabilization of tetrahedrally coordinated ferric iron by charge-

451   balancing K_2O has been suggested by various authors (Sack et al., 1980; Kilinc et al.,

452   1983; Dickenson and Hess, 1986; Kress and Carmichael, 1988) supporting that K_2O has a

453   positive impact on the ferric-ferrous ratio.

454   Noteworthy, in contrast to the model predictions and our findings, Tangeman et

455   al. (2001) proposed a negative effect of increasing K_2O and a positive effect of increasing

456   ΣFeO on the Fe^{3+}/Fe^{2+} ratio. However, these discrepancies may be due to significant

457   compositional differences, since these authors investigated anhydrous K_2O-FeO-Fe_2O_3-

458   SiO_2 liquids and their experiments were performed at atmospheric pressure.

459

460   **Effect of temperature and pressure on the redox state of iron**

461   The models of Kress and Carmichael (1991) and Moretti (2005) both predict a

462   small positive dependence of the $Fe^{2+}/\Sigma Fe$ ratio on temperature. At a given f_{O_2}, both

463   models calculate an increase of the $Fe^{2+}/\Sigma Fe$ ratio by 0.01 to 0.08 when the temperature is

464   raised from 1200 to 1250°C. This effect is more pronounced at lower pressure and higher

465   oxygen fugacity. The comparison of our experimental datasets obtained at 1200°C and

466   1250°C reveals no clearly resolvable systematic trend regarding experimental and

467   analytical uncertainties. However, predicted effects on the iron redox state caused by this

468   moderate temperature change are small and therefore difficult to resolve.

469   To evaluate the effect of pressure on the redox state of iron we performed

470   experiments at pressures from 50 to 500 MPa (Table 1) at intrinsic redox conditions in

471   the IHPV (NNO+2.6). All experiments were done with a pure H_2O fluid to obtain a water

472   saturated silicate melt. In Figure 7a, the $Fe^{2+}/\Sigma Fe$ ratios of the silicate glasses are shown

473   as a function of pressure. The data suggest a slight negative trend with increasing

474   pressure. However, this trend is basically defined by the 50 MPa experiments, whereas

475   most of the experimentally obtained $Fe^{2+}/\Sigma Fe$ ratios agree within uncertainties.

476   Furthermore, it is important to note that the water solubility in the melt increases with

477   increasing pressure (Fig. 7b) and hence for a comparison of the experiments a

478   superimposed effect of water content – as suggested by Moretti (2005) – has to be

479   considered as well.

480   Both models (Kress and Carmichael, 1991; Moretti, 2005) predict a nonlinear

481   positive dependence of the $Fe^{2+}/\Sigma Fe$ ratio on increasing pressure (Fig. 7a). According to

482   the model of Moretti (2005) the pressure effect is more prominent for water-rich than for

483   dry melts. Our experiments at $a_{H_2O} = 1$ cover a range from about 2.4 to 12 wt% H_2O and

484   the measured $Fe^{2+}/\Sigma Fe$ ratios are consistent with the range of $Fe^{2+}/\Sigma Fe$ calculated by the

485   model of Moretti (2005) for corresponding water contents. Thus, the dependence of the

486   Fe redox state on water contents (linked to the specific pressures) that is superimposed on

487   the pressure effect could explain the seeming negative trend mentioned above. No

488   analytically resolvable change in the Fe redox ratio is found within the pressure range

489   investigated in this study. An extension of the experimental dataset to higher pressures is

490   needed to better constrain the effect of pressure on $Fe^{2+}/\Sigma Fe$ in hydrous silicate melts and

491   to allow a comparison to anhydrous silicate melts, e.g. O'Neill et al. (2006), where a

492   pressure effect was observed. Such data are useful to retrieve precise information about

493   partial molar volumes of iron species for thermodynamic modeling.

494

495   **CONCLUDING REMARKS**

496   An experimental study was performed to investigate the dependence of the

497    $Fe^{2+}/\Sigma Fe$ ratio in phono-tephritic melts on oxygen fugacity. The redox conditions were

498   adjusted at prevailing f_{H_2} using mixed H_2O-CO_2 fluids which control the water activity in

499   the system. The experimentally obtained relationship between f_{O_2} and Fe^{3+}/Fe^{2+} for

500   hydrous ultrapotassic melts is in general agreement with predictions from the models of

501   Kress and Carmichael (1991) and Moretti (2005). This suggests that these models can be

502 applied to phono-tephritic melts as well, although such compositions were not

503 implemented in the model calibrations.

504   A small negative effect of dissolved water on $Fe^{2+}/\Sigma Fe$ at given f_{O_2} was found that

505 confirms the predictions of the thermodynamic model of Moretti (2005). On the other

506 hand, no effect of pressure and temperature on the redox state of iron was resolvable in

507 the investigated P-T range. Compared to hydrous ferrobasaltic melts systematically

508 higher $Fe^{2+}/\Sigma Fe$ ratios were found for the phono-tephrite in particular at low oxygen

509 fugacity. This effect is most likely due to the much higher K_2O contents of the phono-

510 tephrite (7.5 wt% compared to 0.3 wt%). However, the difference in ΣFeO (7.8 wt% in

511 the phono-tephrite and 12.9 wt% in the ferrobasalt) may additionally contribute to the

512 observed differences in $Fe^{2+}/\Sigma Fe$. The new data may be used to improve the computation

513 models, i.e., to calibrate the effects of H_2O, K_2O and FeO on the redox state or iron in

514 silicate melts. Furthermore, complementing other recent investigations (e.g., Freda et al.,

515 2006; Gaeta et al., 2006; Carapezza and Tarchini, 2007; Freda et al., in press), this study

516 contributes to a better understanding of the redox conditions of the Alban hills magmatic

517 system and of potassic magmatism, in general.

518

519

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532

533

534   **APPENDIX 1 -** Reagents used for wet-chemical colorimetric iron determination

535

536   All reagents were freshly prepared before each analytical session:

537   *Ammonium vanadate solution*. 0.165 g NH_4VO_3 (Alfa Aesar, 99.93% metals basis) were

538   dissolved in 100 mL distilled water and H_2SO_4 (Merck, p.a.). The final concentration of

539   H_2SO_4 was generally 1M, or in some cases up to 5M to test whether the sulphuric acid

540   concentration has an influence on the ferrous iron determination (see text). 1 mL of this

541   solution was added to each sample (corresponds to ~14.1 µmol NH_4VO_3), capable to

542   oxidize about 800 µg ferrous Fe. We always processed <500 µg Fe to ensure V^{5+} excess.

543   *Boric acid solution*. Approximately 20 g H_3BO_3 (Merck, p.a.) were dissolved in ~100 mL

544   distilled water at ~80°C to obtain a saturated solution. After sample dissolution 5 mL of

545   the hot solution was added to each sample.

546   *2:2'bipyridyl solution*. 0.15 g $C_{10}H_8N_2$ (Alfa Aesar, p.a.) were dissolved in 100 mL

547   distilled water. 5 mL of this solution (corresponds to ~9.6 µmol $C_{10}H_8N_2$) was added to

548   each sample. This amount can complex about 900 µg ferrous Fe to form the colored

549   chromophore. Less than <500 µg Fe were processed to ensure 2:2'bipyridyl excess.

550   *Ammonium acetate solution***:** Approximately 20 g CH_3COONH_4 were dissolved in 200

551   mL distilled water.

552   *Hydroxylamine hydrochloride*. Between 5 and 10 mg $NH_2OH \cdot HCl$ (Merck, p.a.) were

553   added to each 10-mL-sample-aliquot containing <50 μg ferric Fe, to ensure quantitative

554   reduction. At least 8 mg ferric Fe can be reduced by this amount (>72 μmol

555    $NH_2OH \cdot HCl$).

556   *Hydrofluoric acid* was 24% or 48% (v/v, Merck, p.a.) (see text).

557   *Ferrous ammonium sulphate.* $(NH_4)_2Fe(SO_4)_2 \cdot 6H_2O$ (Merck, p.a.). Standards with

558   concentrations between 1 and 8 μg/mL ferrous Fe were prepared in $1M$ H_2SO_4 for

559   quantitative calibration of the method.

560

561

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Table 1. Summary of experimental conditions, results of wet-chemical colorimetric iron analyses and $Fe^{2+}/\Sigma Fe$ ratios from model calculations after Moretti (2005) and Kress and Carmichael (1991).

									wet-chemical colorimetric analyses						model calculations	
					experimental conditions											
sample	run[a]	P	T	t	$X^f_{H_2O}$ [b]	a_{H_2O} [c]	$\log f_{O_2}$ [d]	ΔNNO [e]	sample weight	A_{523} Fe^{2+}	A_{523} ΣFe	ΣFeO		$Fe^{2+}/\Sigma Fe$	$Fe^{2+}/\Sigma Fe$	
		(MPa)	(°C)	(h)					(mg)			(wt%)	(2σ)	(2σ)	(Moretti)	(K&C)
Ar-IHPV (f_{H_2} intrinsic); H_2O fluid, $a_{H_2O} = 1$																
Alb1-20	A	50	1250	15	1.00	1.00	-4.39	2.60	4.65	0.2323	0.3880	7.07	0.24	0.60 0.02	0.40	0.48
									6.01	0.2999	0.5014	7.06	0.22	0.60 0.02	0.40	0.48
Alb1-21	A	50	1250	15	1.00	1.00	-4.39	2.60	5.76	0.2886	0.4852	7.12	0.22	0.59 0.02	0.39	0.48
Alb1-22	B	100	1250	20	1.00	1.00	-4.38	2.60	4.54	0.1953	0.3921	7.31	0.25	0.50 0.02	0.39	0.48
Alb1-23	B	100	1250	20	1.00	1.00	-4.38	2.60	4.57	0.1877	0.3775	6.99	0.24	0.50 0.02	0.39	0.48
Alb1-36	C	200	1200	20	1.00	1.00	-4.90	2.60	5.64	0.2307	0.4665	7.00	0.22	0.49 0.02	0.41	0.48
Alb1-37	C	200	1200	20	1.00	1.00	-4.90	2.60	3.50	0.1607	0.2994	7.25	0.28	0.54 0.03	0.42	0.48
Alb1-10	D	200	1250	20	1.00	1.00	-4.35	2.60	5.89	0.2176	0.4986	7.17	0.22	0.44 0.02	0.41	0.49
Alb1-24[e]	E	300	1250	14	1.00	1.00	-4.32	2.60	4.78	0.1877	0.3918	6.94	0.23	0.48 0.02	0.43	0.49
Alb1-25[e]	E	300	1250	14	1.00	1.00	-4.32	2.60	4.79	0.1870	0.3922	6.93	0.23	0.48 0.02	0.43	0.49
Alb1-35[e]	F	400	1200	72	1.00	1.00	-4.84	2.60	5.53	0.2148	0.4477	6.85	0.22	0.48 0.02	0.45	0.50
Alb1-26[e]	G	400	1250	48	1.00	1.00	-4.29	2.60	4.69	0.1848	0.3772	6.81	0.23	0.49 0.02	0.45	0.50
Alb1-27[e]	G	400	1250	48	1.00	1.00	-4.29	2.60	4.58	0.1764	0.3630	6.72	0.23	0.49 0.02	0.45	0.50
Alb1-5[e]	H	500	1200	20	1.00	1.00	-4.81	2.60	4.12	0.1500	0.3298	6.78	0.25	0.45 0.03	0.46	0.50
Alb1-15[e]	I	500	1250	20	1.00	1.00	-4.26	2.60	4.55	0.1544	0.3759	6.99	0.24	0.41 0.02	0.48	0.51
									4.98	0.1619	0.3934	6.69	0.22	0.41 0.02	0.48	0.51
Alb1-28[e]	L	500	1250	17	1.00	1.00	-4.26	2.60	1.95	0.0723	0.1579	6.84	0.40	0.46 0.06	0.47	0.51
Alb1-41	N	200	1200	24	1.00	1.00	-4.90	2.60	5.49	0.2366	0.4641	7.16	0.23	0.51 0.02	0.44	0.48
Ar-IHPV (f_{H_2} intrinsic); H_2O-CO_2 fluid																
Alb1-7	D	200	1250	18	0.33	0.40	-5.15	1.80	4.46	0.2322	0.3805	7.23	0.25	0.61 0.02	0.58	0.58
Alb1-8	D	200	1250	18	0.50	0.57	-4.84	2.10	4.49	0.2183	0.3946	7.44	0.25	0.55 0.02	0.52	0.55
Alb1-6bis	M	200	1250	20	0.10	0.13	-6.10	0.85	4.52	0.2555	0.3277	6.13	0.22	0.78 0.03	0.75	0.67
Alb1-6a	M	200	1250	20	0.18	0.23	-5.62	1.33	4.97	0.2649	0.3822	6.51	0.22	0.69 0.03	0.68	0.63
Alb1-1[e]	H	500	1200	18	n.a.	0.15[e]	-6.47	0.93	4.45	0.2658	0.3187	6.07	0.22	0.83 0.03	0.78	0.68
Alb1-2	H	500	1200	18	0.22	0.28	-5.91	1.50	3.99	0.2191	0.3425	7.27	0.26	0.64 0.03	0.69	0.63
Alb1-3	H	500	1200	18	0.60	0.66	-5.17	2.23	4.56	0.1952	0.3653	6.78	0.23	0.53 0.02	0.56	0.55
									2.45	0.1013	0.2005	6.92	0.34	0.51 0.04	0.56	0.55
Alb1-4[e]	H	500	1200	18	0.83	0.85	-4.95	2.46	4.18	0.1617	0.3537	7.17	0.25	0.46 0.02	0.50	0.46
Alb1-12	I	500	1250	18	0.32	0.39	-5.08	1.78	4.62	0.2167	0.4055	7.42	0.25	0.53 0.02	0.63	0.54
Alb1-13[e]	I	500	1250	18	0.75	0.78	-4.47	2.38	3.55	0.1368	0.3052	7.27	0.27	0.45 0.03	0.51	0.47
Alb1-14[e]	I	500	1250	18	0.79	0.81	-4.44	2.42	4.48	0.1714	0.3870	7.32	0.25	0.44 0.02	0.51	0.47

Table 1. - continued

sample	run[a]	P	T	t	$X^f_{H_2O}$ [b]	a_{H_2O} [c]	$\log f_{O_2}$ [d]	ΔNNO[e]	sample weight	A_{523} Fe^{2+}	A_{523} ΣFe	ΣFeO	(2σ)	Fe^{2+}/ΣFe	(2σ)	Fe^{2+}/ΣFe (Moretti)	(K&C)
		(MPa)	(°C)	(h)					(mg)			(wt%)					

Ar-H$_2$-IHPV (f_{H_2} varied); H$_2$O fluid or H$_2$O-CO$_2$ fluid

sample	run	P	T	t	$X^f_{H_2O}$	a_{H_2O}	$\log f_{O_2}$	ΔNNO	sample weight	A_{523} Fe^{2+}	A_{523} ΣFe	ΣFeO	(2σ)	Fe^{2+}/ΣFe	(2σ)	Moretti	K&C
Alb1-H42	O	200	1200	18	1.00	1.00	-5.64	1.87	5.56	0.2302	0.4640	7.06	0.22	0.50	0.02	0.51	0.57
Alb1-H43	O	200	1200	18	0.76	0.78	-5.86	1.65	5.48	0.2357	0.4611	7.12	0.23	0.51	0.02	0.55	0.53
Alb1-H44	O	200	1200	18	0.30	0.34	-6.57	0.93	5.59	0.2863	0.4658	7.05	0.22	0.61	0.02	0.69	0.61
									4.47	0.2251	0.3705	7.02	0.24	0.61	0.03	0.69	0.61
Alb1-H45	O	200	1200	18	0.07	0.09	-7.78	-0.28	5.27	0.2921	0.4083	6.55	0.22	0.72	0.02	0.84	0.73
									4.37	0.2379	0.3359	6.50	0.23	0.71	0.03	0.84	0.73
Alb1-H47	P	200	1200	5	1.00	1.00	-7.70	-0.20	5.11	0.2883	0.3646	6.04	0.21	0.79	0.03	0.77	0.76
Alb1-H48	P	200	1200	5	0.71	0.73	-7.97	-0.47	5.20	0.3039	0.3893	6.34	0.21	0.78	0.03	0.81	0.74
Alb1-H49	P	200	1200	5	0.39	0.43	-8.43	-0.93	5.70	0.3057	0.3720	5.52	0.19	0.82	0.03	0.87	0.78
Alb1-H50	P	200	1200	5	0.23	0.27	-8.85	-1.35	5.02	0.2371	0.2886	4.87	0.19	0.82	0.04	0.90	0.81
									4.07	0.1949	0.2338	4.86	0.21	0.83	0.04	0.90	0.81
Alb1-H51	P	200	1200	5	0.14	0.17	-9.26	-1.76	5.36	0.2084	0.2577	4.07	0.17	0.81	0.04	0.93	0.83
									3.82	0.1594	0.1976	4.37	0.21	0.81	0.05	0.93	0.86
Alb1-H52	Q	200	1200	1.5	1.00	1.00	-7.80	-0.29	5.26	0.3419	0.4567	7.34	0.23	0.75	0.02	0.79	0.72
Alb1-H54	Q	200	1200	1.5	0.50	0.53	-8.35	-0.85	6.24	0.4169	0.5276	7.16	0.22	0.79	0.02	0.86	0.76
Alb1-H56	Q	200	1200	1.5	0.04[f]	0.05	-10.40[f]	-2.90[f]	5.23	0.2842	0.3342	5.41	0.19	0.85	0.03	0.96	0.89

n.a. not available

a) Each letter refers to an individual run in the IHPV, containing one or more samples, i.e. capsules.

b) Mole fraction of water and CO$_2$ in the fluid phase measured by the weight loss method (see text).

c) Water activities for mixed H$_2$O-CO$_2$ fluids were calculated from $X^f_{H_2O}$ after Aranovich and Newton (1999) using molar volumes of pure H$_2$O and CO$_2$ from Pitzer and Sterner (1994).

d) Oxygen fugacity calculated from water activity: $\log f_{O_2} = \log f_{O_2}(\text{IHPV}_{apparent}) + 2 \log(a_{H_2O})$, where $f_{O_2}(\text{IHPV}_{apparent})$ is the oxygen fugacity in the capsule at $a_{H_2O} = 1$ imposed by the f_{H_2} in the IHPV (intrinsic of defined by Ar-H$_2$ mixtures). ΔNNO is $\log f_{O_2}$ expressed relative to the Ni-NiO buffer.

e) Experimental products which contain quench crystals formed during cooling at the end of the experiment. All other run products consist of glass and a fluid phase only; except sample Alb1-1, which additionally contains clinopyroxene crystals formed in equilibrium with the silicate melt at experimental conditions (see text).

f) Fluid composition in mole fractions ($X^f_{H_2O}$, $X^f_{CO_2}$, X^f_{CO}) for sample Alb1-H56 (that was used to calculate the oxygen fugacity) estimated from CO$_2$ solubility trends due to presence of other carbon species (beside CO$_2$) in the fluid (see text).

g) Water activity was calculated from an estimated $X^f_{H_2O}$ of 0.11 for Alb1-1 (see text).

Notes for wet-chemical colorimetric analyses: A_{523} values are the measured baseline-corrected absorbances at the Fe(II)-2'2-bipyridiyl absorbance maximum at about 523 nm. Given uncertainties calculated from error propagation of individual uncertainties in sample weight, dilution and absorbance measurements. The external reproducibility (2σ) of the $Fe^{2+}/\Sigma Fe$ ratio with this method is 0.03 (see text). Multiple analyses for the same sample represent full wet-chemical replicates including individual dissolution of fragments from the experimental products.

Table 2. Electron microprobe analyses of the starting phono-tephritic glass and the experimental glasses. Analysis of the ferrobasalt SC1 studied by Botcharnikov et al. (2005) is shown for comparison, to highlight compositional differences. Given are averages and the standard deviation s(1σ) of n replicate analyses reported in wt%.

sample	n	SiO_2	Al_2O_3	ΣFeO	MgO	CaO	MnO	TiO_2	Na_2O	K_2O	Total	ΣFeO norm[b]
phono-tephritic starting glass												
Alb1	31	49.89 ± 0.42	15.57 ± 0.21	7.82 ± 0.32	5.75 ± 0.18	11.40 ± 0.21	0.02 ± 0.09	0.89 ± 0.03	1.95 ± 0.18	7.52 ± 0.16	100.80 ± 0.67	7.75
ferrobasalt from Botcharnikov et. al (2005)												
SC1		48.34 ± 0.29	14.61 ± 0.13	12.91 ± 0.28	6.40 ± 0.11	10.87 ± 0.15	-	2.86 ± 0.05	2.60 ± 0.11	0.30 ± 0.03	98.89 ± 0.67	13.05
experimental products												
Alb1-1	8	48.31 ± 0.40	15.65 ± 0.26	6.09 ± 0.32	5.22 ± 0.17	10.53 ± 0.19	0.04 ± 0.08	0.91 ± 0.06	2.14 ± 0.12	7.47 ± 0.08	96.36 ± 0.83	6.32
Alb1-2	11	46.20 ± 0.27	14.56 ± 0.14	6.92 ± 0.22	5.31 ± 0.10	10.44 ± 0.24	0.01 ± 0.06	0.83 ± 0.06	1.97 ± 0.12	6.90 ± 0.14	93.14 ± 0.62	7.43
Alb1-5[a]	9	44.42 ± 0.68	13.94 ± 0.16	6.58 ± 0.73	4.93 ± 0.81	10.28 ± 1.56	0.05 ± 0.08	0.80 ± 0.04	1.78 ± 0.19	6.56 ± 0.53	89.35 ± 0.81	7.37
Alb1-8	11	46.82 ± 0.38	14.61 ± 0.18	7.01 ± 0.22	5.26 ± 0.11	10.38 ± 0.22	0.08 ± 0.08	0.83 ± 0.03	1.94 ± 0.12	7.04 ± 0.15	93.96 ± 0.67	7.46
Alb1-6a	11	48.08 ± 0.35	15.08 ± 0.16	6.74 ± 0.36	5.40 ± 0.10	10.91 ± 0.26	0.01 ± 0.08	0.87 ± 0.05	2.02 ± 0.09	7.14 ± 0.14	96.27 ± 0.63	7.00
Alb1-6bis	11	48.54 ± 0.49	15.29 ± 0.18	5.95 ± 0.26	5.50 ± 0.15	10.92 ± 0.25	0.03 ± 0.10	0.91 ± 0.04	2.06 ± 0.24	7.33 ± 0.11	96.55 ± 0.95	6.16
Alb1-14	11	44.56 ± 0.39	13.98 ± 0.16	6.77 ± 0.30	4.94 ± 0.19	9.96 ± 0.20	0.02 ± 0.05	0.79 ± 0.04	1.82 ± 0.10	6.74 ± 0.12	89.58 ± 0.61	7.56
Alb1-20	12	47.28 ± 0.49	14.83 ± 0.24	6.94 ± 0.23	5.44 ± 0.20	10.66 ± 0.26	0.01 ± 0.07	0.86 ± 0.02	2.01 ± 0.11	7.10 ± 0.14	95.14 ± 0.86	7.30
Alb1-23	12	47.06 ± 0.38	14.65 ± 0.15	7.04 ± 0.29	5.20 ± 0.11	10.49 ± 0.25	0.06 ± 0.06	0.84 ± 0.04	1.97 ± 0.11	6.93 ± 0.13	94.24 ± 0.54	7.47
Alb1-25	12	45.47 ± 0.35	14.06 ± 0.13	6.79 ± 0.22	5.00 ± 0.08	9.99 ± 0.16	0.06 ± 0.08	0.82 ± 0.03	1.81 ± 0.16	6.72 ± 0.16	90.73 ± 0.38	7.48
Alb1-26	12	44.77 ± 0.27	14.03 ± 0.19	6.69 ± 0.29	5.14 ± 0.16	9.89 ± 0.27	0.05 ± 0.07	0.79 ± 0.03	1.80 ± 0.14	6.62 ± 0.12	89.78 ± 0.61	7.45
Alb1-41	12	46.42 ± 0.41	14.56 ± 0.25	7.06 ± 0.33	5.27 ± 0.16	10.70 ± 0.24	0.01 ± 0.08	0.83 ± 0.04	1.81 ± 0.08	7.12 ± 0.13	93.78 ± 0.69	7.53
Alb1-H42	15	45.57 ± 0.35	14.18 ± 0.21	6.85 ± 0.25	5.03 ± 0.13	10.32 ± 0.28	0.02 ± 0.10	0.83 ± 0.03	1.77 ± 0.13	7.06 ± 0.11	91.63 ± 0.58	7.48
Alb1-H43	18	45.28 ± 0.35	14.22 ± 0.18	7.03 ± 0.33	5.21 ± 0.15	10.34 ± 0.27	0.06 ± 0.06	0.83 ± 0.05	1.87 ± 0.12	7.04 ± 0.15	91.86 ± 0.75	7.65
Alb1-H44	13	46.69 ± 0.41	14.63 ± 0.20	6.91 ± 0.39	5.36 ± 0.13	10.75 ± 0.21	0.01 ± 0.05	0.87 ± 0.06	1.91 ± 0.15	7.18 ± 0.17	94.30 ± 0.72	7.33
Alb1-H45	16	47.39 ± 0.83	14.98 ± 0.23	6.32 ± 0.35	5.29 ± 0.28	10.78 ± 0.31	0.03 ± 0.06	0.87 ± 0.05	1.87 ± 0.14	7.42 ± 0.19	94.94 ± 1.39	6.65
Alb1-H47	12	45.60 ± 0.27	14.18 ± 0.24	6.17 ± 0.24	5.14 ± 0.18	10.46 ± 0.24	0.03 ± 0.08	0.87 ± 0.02	1.79 ± 0.14	7.10 ± 0.12	91.33 ± 0.65	6.75
Alb1-H48	14	46.24 ± 0.45	14.46 ± 0.16	6.40 ± 0.16	5.23 ± 0.11	10.37 ± 0.19	0.01 ± 0.07	0.84 ± 0.03	1.90 ± 0.22	7.13 ± 0.15	92.58 ± 0.47	6.92
Alb1-H49	16	46.97 ± 0.56	14.80 ± 0.24	5.52 ± 0.35	5.32 ± 0.16	10.64 ± 0.29	0.03 ± 0.07	0.86 ± 0.04	1.87 ± 0.15	7.37 ± 0.16	93.39 ± 0.71	5.91
Alb1-H50	16	48.01 ± 0.39	15.13 ± 0.21	4.72 ± 0.32	5.41 ± 0.15	10.83 ± 0.18	0.02 ± 0.06	0.85 ± 0.04	1.95 ± 0.15	7.56 ± 0.17	94.47 ± 0.59	5.00
Alb1-H51	20	47.93 ± 0.97	15.22 ± 0.23	4.50 ± 0.34	5.57 ± 0.15	10.95 ± 0.25	0.01 ± 0.08	0.89 ± 0.05	1.91 ± 0.18	7.60 ± 0.17	94.58 ± 1.29	4.76

a) Analysis represents glass and quench crystals (see text).

b) ΣFeO concentration in the glasses recalculated to a water-free basis (normalized to a sum of 100).

Table 3. Results of wet-chemical colorimetric iron analyses on international reference materials and in-house standards.

sample	description	sample weight[a] (mg)	T^b (°C)	n^c	this study Fe^{2+}/ΣFe (2σ)	this study ΣFeO (wt%) (2σ)	this study ferrous FeO (wt%) (2σ)	recommended values Fe^{2+}/ΣFe (2SE) (2σ)	recommended values ΣFeO (wt%) (2SE) (2σ)	recommended values ferrous FeO (wt%) (2SE) (2σ)	ref.
international reference materials											
BIR-1	basalt	6-7	20	9	0.82 0.03	10.20 0.21	8.35 0.35	0.82 n.a. 0.03	10.17 n.a. 0.22	8.34 n.a. 0.20	(1)
BE-N	basalt	6-7	20	3	0.61 0.01	11.53 0.22	6.98 0.26	0.58 0.01 0.06	11.55 0.05 0.63	6.74 0.08 0.60	(2)
RGM-1	calc-alkaline rhyolite	19-22	20	15	0.76 0.04	1.63 0.05	1.25 0.06	0.76 n.a. 0.06	1.67 n.a. 0.05	1.27 n.a. 0.10	(1)
RGM-1	calc-alkaline rhyolite	19-23	100	3	0.78 0.02	1.63 0.04	1.28 0.07	0.76 n.a. 0.06	1.67 n.a. 0.05	1.27 n.a. 0.10	(1)
STM-1	peralkaline nepheline syenite	10-11	20	9	0.49 0.02	4.23 0.20	2.05 0.11	0.44 n.a. 0.02	4.70 n.a. 0.18	2.09 n.a. 0.06	(1)
STM-1	peralkaline nepheline syenite	10-12	100	3	0.46 0.02	4.63 0.10	2.15 0.05	0.44 n.a. 0.02	4.70 n.a. 0.18	2.09 n.a. 0.06	(1)
GS-N	granite	15-21	20	5	0.63 0.04	2.88 0.34	1.82 0.10	0.49 0.02 0.12	3.37 0.04 0.25	1.65 0.07 0.38	(2)
GS-N	granite	15-22	100	4	0.57 0.01	3.28 0.05	1.86 0.04	0.49 0.02 0.12	3.37 0.04 0.25	1.65 0.07 0.38	(2)
GA	granite	20-21	20	5	0.61 0.01	2.28 0.11	1.39 0.06	0.52 0.02 0.11	2.55 0.05 0.38	1.32 0.04 0.18	(2)
GA	granite	20-22	100	4	0.60 0.01	2.40 0.11	1.44 0.06	0.52 0.02 0.11	2.55 0.05 0.38	1.32 0.04 0.18	(2)
AC-E	granite	20-21	20	2	0.47 0.01	2.19 0.04	1.04 0.01	0.47 0.02 0.12	2.28 0.02 0.23	1.07 0.04 0.26	(2)
AC-E	granite	20-21	100	4	0.49 0.01	2.22 0.07	1.08 0.03	0.47 0.02 0.12	2.28 0.02 0.23	1.07 0.04 0.26	(2)
in-house standards											
PU-3[d]	andesite glass (syn.)	5-9	20	33	0.39 0.03	7.44 0.15	2.87 0.24	0.42 n.a. 0.03	7.49 0.15 0.34	3.15 n.a. 0.21	(3,4)
PU-3[d]	andesite glass (syn.)	5-9	100	4	0.40 0.02	7.20 0.44	2.86 0.28	0.42 n.a. 0.03	7.49 0.15 0.34	3.15 n.a. 0.21	(3,4)
CT-1[e]	basalt glass (syn.)	5-6	20	8	0.39 0.03	12.78 0.23	5.02 0.29	n.a. n.a. n.a.	12.85 0.19 0.44	n.a. n.a. n.a.	(4)
CA	Chlorite	1.5	20	3	0.89 0.04	39.96 1.35	35.8 1.67	0.90 n.a. 0.02	41.56 0.12 0.79	37.32 0.11 0.71	(5)
olivine	$Mg_{1.829}Fe^{2+}_{0.171}SiO_4$ (natural)	3-6	20	13	0.93 0.05	8.30 0.74	7.75 0.74	1.00 n.a. n.a.	8.38 0.06 0.07	8.38 0.06 0.07	(4,6)
ammonium iron(II) sulfate hexahydrate $(NH_4)_2Fe(SO_4)_2 \cdot 6H_2O$ (Merck, p.a.)		1-2	20	11	0.95 0.08	18.46 0.70	17.57 1.50	1.00 n.a. n.a.	18.42 n.a. n.a.	18.42 n.a. n.a.	(6)

n.a. not available. a) Range of sample weight used for analyses. b) Decomposition temperature. c) Number of replicate analyses.

d) Electron microprobe analysis of synthetic glass CT-1 (wt%): SiO_2 47.40, Al_2O_3 14.24, ΣFeO 12.85, MgO 6.25, CaO 10.79, Na_2O 2.68, K_2O 0.32, TiO_2 3.17

e) Electron microprobe analysis of synthetic glass PU-3 (wt%): SiO_2 54.72, Al_2O_3 16.80, ΣFeO 7.49, MgO 4.15, CaO 9.00, Na_2O 3.32, K_2O 1.54, TiO_2 0.79, MnO 0.12

References for recommended values: (1) Govindaraju (1994), (2) Govindaraju (1995), (3) Fe^{2+}/ΣFe value calculated from ferrous iron analyses of sample Unzen-A given by Liebske et al. (2003), (4) ΣFeO values were measured by electron microprobe analysis (this study); FeO value of PU-3 calculated from ΣFeO (this study) and Fe^{2+}/ΣFe from Liebske et al. (2003). (5) Bertoldi et al. (2007), (6) ΣFeO (= ferrous FeO) calculated from stoichiometry.

Figure captions

738 **Figure 1.** UV/VIS spectra of Fe(II)-2:2'-bipyridyl solutions. Ferrous iron concentrations are

739 indicated (in μg/mL Fe). Spectra recorded in 1 cm transmission cells.

740

741 **Figure 2.** Comparison of measured $Fe^{2+}/\Sigma Fe$ ratios from this study with other studies

742 (recommended values). Open symbols refer to incomplete sample dissolution (see text). Solid

743 line represents a 1:1 correlation. Dashed lines indicate the interval ±0.03.

744

745 **Figure 3.** Oxidation state of iron as a function of H_2O mole fraction in the fluid. Lines

746 represent second order polynomials for selected datasets to illustrate the non-linear

747 relationships. Oxygen fugacity is expressed relative to the Ni-NiO buffer (ΔNNO) and the

748 given values represent the maximum $\log f_{O_2}$ values in the runs ($X^f_{H_2O} = 1$), i.e., f_{O_2} in the

749 capsule is lower at $X^f_{H_2O} < 1$.

750

751 **Figure 4 .** The dependence of the Fe^{3+}/Fe^{2+} ratio on oxygen fugacity expressed relative to the

752 NNO buffer. Note the logarithmic scaling. Alb1-1 (in parentheses) is the only sample

753 containing clinopyroxene crystals (see text). Symbols as in Figure 3.

754

755 **Figure 5.** Comparison between measured and calculated $Fe^{2+}/\Sigma Fe$ ratios from the experiments

756 and models of (a) Moretti (2005) and (b) Kress and Carmichael (1991), respectively. The

757 solid line is a 1:1 correlation and the dashed lines represent an envelope of ±0.05. Symbols as

758 in Figure 3.

759

760 **Figure 6.** Fe^{3+}/Fe^{2+} ratio as a function of log f_{O_2} in comparison with the predictions of the

761 empirical model of Kress and Carmichael (1991) (dashed lines) and the thermodynamic

762 model of Moretti (2005) (grey areas) for the phono-tephritic melt at 200 and 500 MPa. All

763 data in (a), (b) and (d) represent constant hydrogen fugacity (f_{H_2} ~0.6 bar) while for data

764 shown in (c) the hydrogen fugacity varied from 0.6 to 16.7 bar (measured values). The model

765 of Moretti (2005) takes the effect of water contents into account and the lower and upper

766 limits of the grey areas comprise the range from 0 to 10 wt% H_2O at 500 MPa (a, b) and from

767 0 to 5 wt% H_2O at 200 MPa (c, d). This range covers the measured water concentrations in

768 the experimental glasses. For comparison the experimental data of Botcharnikov et al. (2005)

769 for hydrous ferrobasaltic melts are shown (c).

770
771
772 **Figure 7.** (a) Redox state of iron in the phono-tephritic melt as a function of pressure for

773 experiments with pure H_2O fluid at intrinsic redox conditions in the IHPV (NNO+2.6). For

774 comparison the pressure dependence of the $Fe^{2+}/\Sigma Fe$ ratio calculated at 1250°C after Kress

775 and Carmichael (1991) and Moretti (2005) at given H_2O contents is shown. The

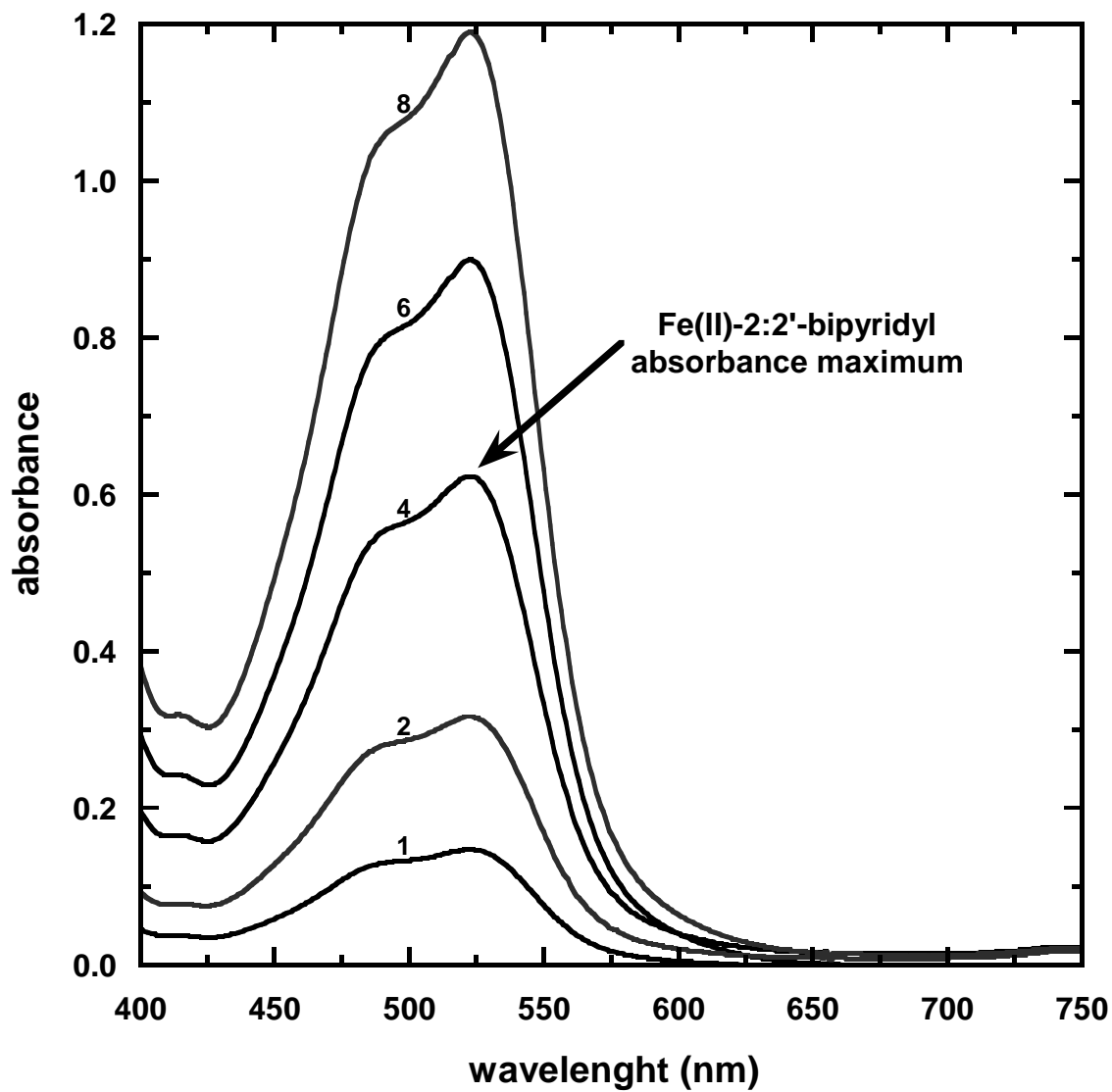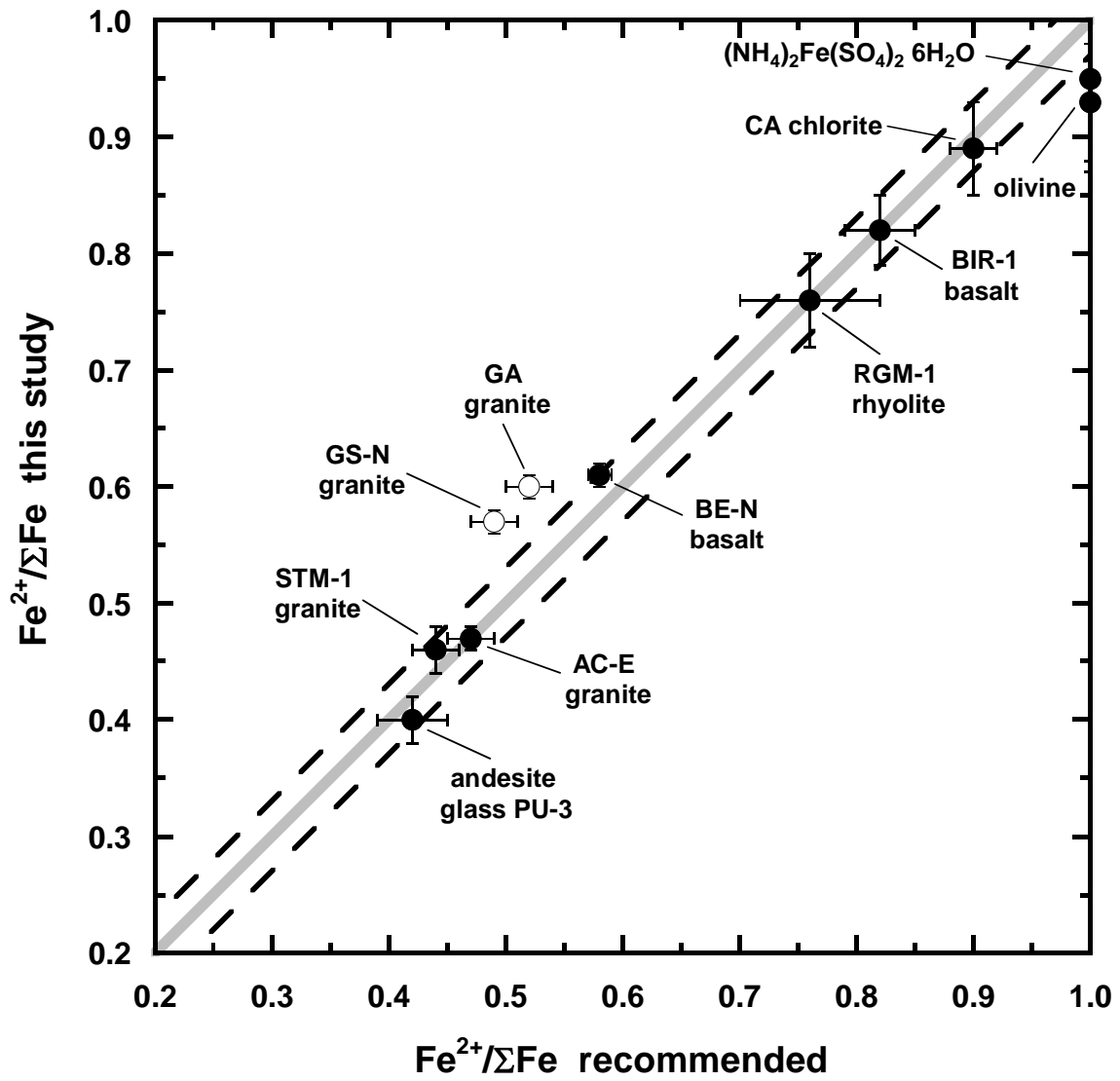776 corresponding water contents of the experimental samples are shown in (b).
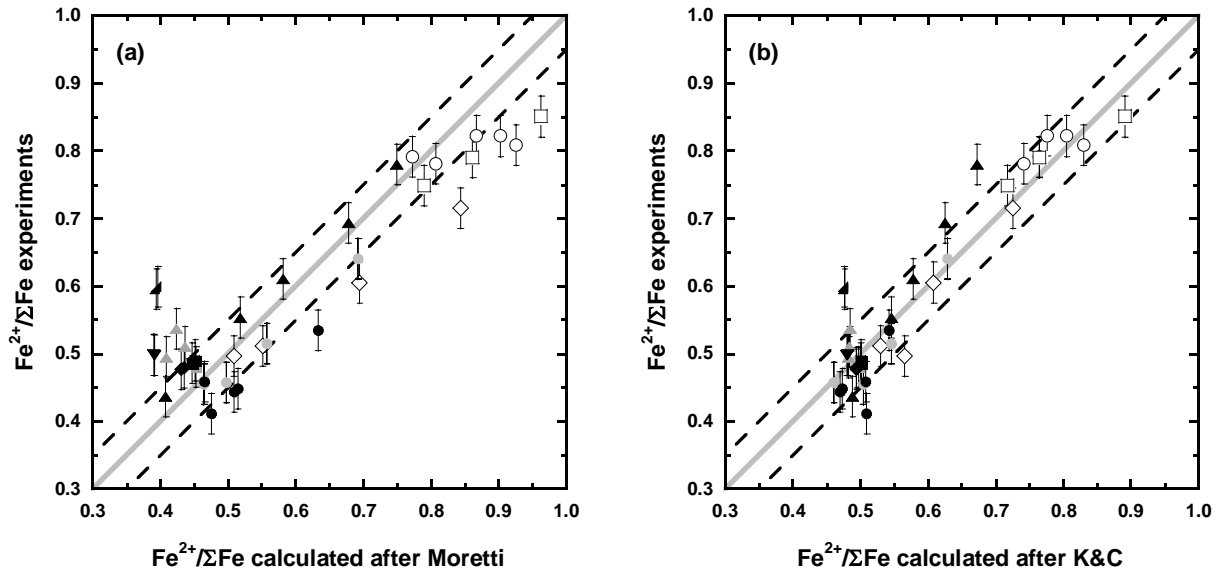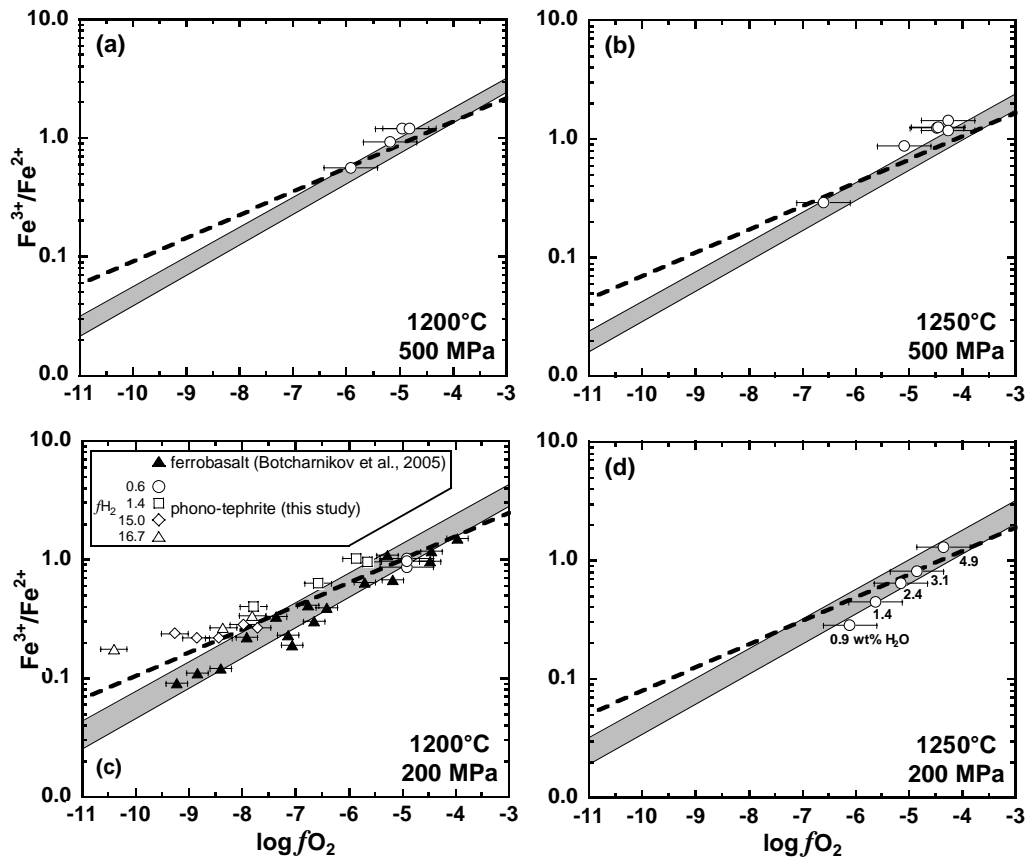
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Figure 1

Figure 2

Figure 3

Figure 4

Figure 5

Figure 6

Figure 7