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Selective etching of high- k Hf O 2 films over Si in hydrogen-added fluorocarbon (C F 4 /Ar/H 2 and C 4 F 8/Ar/H 2) plasmas

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Selective etching of high-*k* HfO₂ films over Si in hydrogen-added fluorocarbon ($CF_4/Ar/H_2$ and $C_4F_8/Ar/H_2$) plasmas

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Inductively coupled hydrogen-added fluorocarbon (CF₄/Ar/H₂ and C₄F₈/Ar/H₂) plasmas were used to etch HfO₂, which is a promising high-dielectric-constant material for the gate of complementary metal-oxide-semiconductor devices. The etch rates of HfO₂ and Si were drastically changed depending on the additive-H₂ flow rate in C₄F₈/Ar/H₂ plasmas. The highly selective etching of HfO₂ over Si was done in the condition with an additive-H₂ flow rate, where the Si surface was covered with the fluorocarbon polymer. The results of x-ray photoelectron spectroscopy indicated that the carbon content of the selectively etched HfO₂ surface was extremely low compared with the preetched surface contaminated by adventitious hydrocarbon in atmosphere. In the gas phase of the C₄F₈/Ar/H₂ plasmas, Hf hydrocarbide molecules such as metal-organic compounds and Hf hydrofluoride were detected by a quadrupole mass analyzer. These findings indicate that the fluorine species, carbon, and hydrogen can work to etch HfO₂ and that the carbon species also plays an important role in selective etching of HfO₂ over Si. © 2006 American Vacuum Society. [DOI: 10.1116/1.2187997]

I. INTRODUCTION

As dimensions of metal-oxide-semiconductor field-effect transistor (MOSFET) devices are scaled down in integrated circuits, the gate width will shrink to much less than 100 nm. The thickness of gate dielectrics should be reduced down to 2 nm or less for the present material, SiO_2 .¹ Then, the thickness reduction of SiO_2 brings a number of serious problems such as increased gate-leakage current and reduced oxide reliability. Therefore, it will be necessary to integrate the high-dielectric-constant (*k*) materials, which can give higher specific capacitance at a larger thickness than SiO_2 and which enable the reduction of gate-leakage current. Integration of high-*k* materials will be one of the important issues in scaling MOSFET devices at critical dimensions below 65 nm.

Recently, replacing SiO₂ with silicon oxynitrides of slightly higher dielectric constant has been tried. In the future, high-k (>20) dielectrics or metal oxides such as HfO₂,² ZrO₂,^{3,4} HfSi_xO_y,^{5,6} and ZrSi_xO_y (Refs. 6 and 7) will be developed to replace SiO₂. When integrating these materials into devices, these materials must be removed completely from the source and drain regions. Therefore, an understanding of the etch characteristics of high-k materials is required for the removal process.

Plasma etching of high-*k* materials has been studied recently for gate dielectric applications. Pelhos *et al.* reported on the etching of high-*k* gate dielectric $Zr_{1-x}Al_xO_y$ thin films with helical-resonator plasmas in Cl_2/BCl_3 .⁸ Sha *et al.* reported on the etching of ZrO₂ with electron-cyclotronresonance plasmas in Cl_2 and BCl_3/Cl_2 .^{9,10} Furthermore, Sha *et al.* also etched HfO₂ thin films in the chlorine chemistries.^{11,12} In their studies of HfO_2 etching, chlorinebased chemistries (not fluorine) were chosen because the HfO_2 was prevented from etching in the CHF₃ plasmas where Hf fluoride compound can be formed as the sidewall mask.^{13,14} Norasetthekul *et al.* reported on the etching of HfO₂ with inductively coupled plasmas in Cl₂/Ar, SF₆/Ar, and CH₄/H₂/Ar.¹⁵ Maeda *et al.* tried to integrate a MOSFET with a HfO₂ dielectric by using etching in CF₄ and Cl₂/HBr-based chemistries.¹⁶ Emphasis in these studies has been placed on etch chemistries giving the selectivity of more than 1 over the underlying Si substrate and on a better understanding of physics and chemistry for the etching.

The thickness of the gate dielectrics for next-generation MOSFET devices (in the 65-nm technology node and beyond) will be several nanometers. Therefore, selectivity to underlying layers or mask materials will be more important than etch rate in the gate process.¹⁷ From the point of view of HfO₂/Si selectivity, highly selective etching can be achieved in fluorocarbon plasmas. We found that HfO₂ can be etched by fluorine and that the selectivity of HfO₂/Si can be more than 5 in C₄F₈/Ar plasmas.¹⁸ In the plasmas, carbon species work as a surface inhibitor on Si not containing oxygen and contribute to obtaining the high selectivity.

To enhance selectivity of HfO_2/Si , HfO_2 etching should be enhanced and/or Si etching should be suppressed. In SiO₂/Si selective etching, fluorocarbon polymer deposited on the surface plays an important role in enhancing the selectivity. The H₂ addition has an effect on the polymer formation in fluorocarbon plasmas.^{19–21} Such chemistries may also be applied to HfO_2/Si selective etching. This article presents results of the etching of HfO_2 thin films on Si substrates in inductively coupled hydrogen-added fluorocarbon (CF₄/Ar/H₂ and C₄F₈/Ar/H₂) plasmas. We discuss the etch mechanism of HfO_2 affected by H_2 addition in the plasmas compared with that of SiO₂, which is well known in previous works.²²

II. EXPERIMENT

The samples for etching were 60-nm-thick HfO_2 films on Si substrates prepared by chemical vapor deposition, SiO₂ films formed by thermal oxidation, and bare Si. The samples were cleaved into 2-cm² pieces and attached to a 4-in.-diameter Si wafer, which was then clamped onto a wafer stage.

Etching experiments were performed in a low-pressure inductively coupled plasma (ICP) reactor supplied with 13.56-MHz rf power.¹⁸ The reactor consisted of a grounded stainless-steel chamber 25 cm in diameter and 25 cm in height. The rf power was coupled to the plasma via a threeturn planar rf induction coil 15 cm in outer diameter that was positioned on a quartz window 20 cm in diameter and 1.2 cm in thickness located at the top side of the chamber. The wafer stage was 13 cm in diameter and located at the bottom side of the chamber, where a close-fitting ground shield surrounded the stage. The distance from the bottom edge of the rf coupling window to the wafer stage was 5 cm. Gas mixtures of $CF_4/Ar(/H_2)$, $C_4F_8/Ar(/H_2)$ and CH_4/Ar , and pure Ar were introduced into the reactor, which was evacuated to a base pressure $<1 \times 10^{-6}$ Torr. The gas pressure and flow rate were maintained at 20 mTorr and 5-300 SCCM (SCCM denotes cubic centimeter per minute at STP), respectively.

The discharge was established at a nominal rf power of 280 or 300 W, corresponding to net powers of the π -type matching circuit driving the induction coil. The wafer stage was capacitively coupled to a separate 13.56-MHz rf power supply for additional biasing; the rf bias power was varied between 10 and 150 W (net power), resulting in a dc self-bias voltage on the stage down to between -40 and -160 V.

Sample pieces covered with masks of Si wafer were etched for several minutes. Steps appearing on the sample pieces were measured by stylus profilometry. The chemical composition of the surface was analyzed by x-ray photoelectron spectroscopy (XPS) using Mg $K\alpha$ x-ray radiation and a pass energy of 50 eV at a takeoff angle of 90°. The plasma parameters (ion density, electron temperature, and plasma potential) were determined by using a cylindrical Langmuir probe located at 2 cm above the wafer stage. The optical emissions from the F atom $(3s^4P_{5/2}-3p^4D_{7/2}^0, 685.6 \text{ nm})$, H atom $(2p^2P_{3/2}^0 - 3d^2D_{3/2}, 656.3 \text{ nm})$, HF molecule (486 nm),²³⁻²⁵ and Ar atom $(4s'\lfloor\frac{1}{2}\rfloor^0 - 4p'\lfloor\frac{1}{2}\rfloor, 750.4 \text{ nm})$ were observed to understand the chemical reactions in the gas phase. The etch products were detected by quadrupole mass spectrometry. A commercial quadrupole mass analyzer (QMA) was mounted on the chamber. Gas-phase species were introduced to the differentially pumped analyzer through a 100 μ m orifice. The orifice was placed 3 cm from the stage and 2 cm above the wafer surface.



FIG. 1. Etch rates of (a) HfO_2 and (b) SiO_2 in $CF_4/Ar/H_2$ plasmas plotted with that of Si, and etch selectivities of (a) HfO_2/Si and (b) SiO_2/Si as a function of additive-H₂ flow rate. The rf bias power was constant at 50 W.

III. RESULTS AND DISCUSSION

A. Selective etching of HfO₂ over Si

Figure 1 shows the etch rates of (a) HfO_2 and (b) SiO_2 in $CF_4/Ar/H_2$ plasmas as a function of the additive-H₂ flow rate at constant rf powers of 280 W (to the coil) and 50 W (for bias), together with that of Si, and the etch selectivities of (a) HfO₂/Si and (b) SiO₂/Si. In generating the plasmas, the gas flow rates of CF₄ and Ar were 2.5 and 247.5 SCCM, respectively. The pressure was maintained at 20 mTorr. Figure 2 shows the etch rates of (a) HfO_2 and (b) SiO_2 in $C_4F_8/Ar/H_2$ plasmas. The gas flow rate of C_4F_8 was 2.5 SCCM. The other parameters were the same as in the CF₄/Ar/H₂ plasmas. The etch depth was measured as a function of etch time up to several minutes and exhibited an approximately linear increase with time. Thus, the etch rate was calculated as the ratio of the depth to time. In the figure, error bars correspond to variance in the measurements, which are not extended to calculating selectivities.

In the CF₄/Ar/H₂ plasmas, the etch rates of HfO₂, Si, and SiO₂ were maintained to be almost constant in all the tested H₂ flow rates. On the other hand, the etch rates of HfO₂ and Si in the C₄F₈/Ar/H₂ plasmas were drastically changed depending on H₂ flow rate. The fluorocarbon polymer was deposited on the Si surface between 4 and 8 SCCM in the H₂ flow rate. When H₂ is added to fluorocarbon plasmas, fluorine is scavenged by hydrogen with the production of the HF molecule in the gas phase,^{26,27}

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FIG. 2. Etch rates of (a) HfO₂ and (b) SiO₂ in C₄F₈/Ar/H₂ plasmas plotted with that of Si, and etch selectivities of (a) HfO₂/Si and (b) SiO₂/Si as a function of additive-H2 flow rate. The rf bias power was constant at 50 W. At the conditions shown by the open triangle, complete selective etching of HfO₂ or SiO₂ was done.

$$C_r F_v + H \to C_r F_{v-1} + HF, \tag{1}$$

$$\mathbf{F} + \mathbf{H} \to \mathbf{HF}.\tag{2}$$

Involving these reactions, the C/F ratio in fluorocarbon species becomes higher. The carbon-rich species are likely to have high sticking probability and to deposit on the surface. If the flow rate of CF₄ is increased, much more carbon-rich species are produced. Even in CF₄/Ar/H₂ plasmas, the species abundant in the gas phase can form deposited films.

At 6 SCCM in H₂ flow rate, the polymer also appeared on the HfO_2 surface in the $C_4F_8/Ar/H_2$ plasmas. At 4 and 8 SCCM, however, HfO2 was etched selectively. In the case of a constant rf bias power, the self-bias voltage on the wafer stage was varied from -40 to -70 V with increasing additive-H₂ flow rate. The polymer formation on the surface can be affected by ion-bombarding energy changed with the self-bias voltage. Therefore, the etch characteristics were examined in the C₄F₈/Ar/H₂ plasmas at a constant self-bias voltage.

Figure 3 shows the etch rates in the plasmas where the self-bias voltage was maintained at a constant value of -90 V. The etch rates of HfO₂, Si, and SiO₂ were drastically changed between 0 and 6 SCCM in the H₂ flow rate and remained almost unchanged at flow rates more than 6 SCCM. The HfO₂ was etched selectively at 2 and 6 SCCM in the H₂ flow rate. Since the ion density was decreased with



FIG. 3. Etch rates of (a) HfO₂ and (b) SiO₂ in $C_4F_8/Ar/H_2$ plasmas plotted with that of Si, and etch selectivities of (a) HfO₂/Si and (b) SiO₂/Si as a function of additive-H2 flow rate. The self-bias voltage was maintained at -90 V.

increasing H₂ flow rate from 0 to 6 SCCM (Fig. 4), the ion flux to the surface also decreased. The etching reactions of Si and fluorocarbon polymer on the surfaces seemed to be suppressed with decrease of the ion flux.

Optical-emission intensities of F, H, and Ar atoms and HF molecules were measured. It may be crucial to estimate the density of emitting species from the emission intensities. The intensity depends on many factors, such as electron density, electron energy-distribution function, density of the emitting species, excitation cross section of the excited state, and so on. Usually, actinometry is employed to quantify the density



FIG. 4. Ion density and electron temperature determined in the Langmuir probe measurement as a function of additive-H2 flow rate.

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FIG. 5. Optical-emission intensities of F and H atoms and HF molecules as a function of additive- H_2 flow rate. These intensities were normalized by optical-emission intensity of Ar atoms.

of species by using noble gases.²⁸ In this work, all the optical-emission intensities were normalized by the emission intensity of Ar atoms at 750.4 nm. Figure 5 shows the normalized intensities of F and H atoms and HF molecules. Strictly speaking, since there is a difference in the electron excitation cross sections between the target species (F, H, and Ar atoms and HF molecules), the normalized intensities cannot represent quantitatively the density of the species. However, since the electron temperature (ranging between 2.8 and 3.0 eV in Fig. 4) was not changed significantly over the tested regime, the normalized intensities represent the qualitative trend in density for the species.

The density of H atoms increased with increasing H_2 flow rate. Especially, the density increased immediately between 6 and 8 SCCM in H_2 flow rate, where the reaction on the Si surface changed from deposition to etching. The F atoms were scavenged from fluorocarbon species and in the gas phase by H atoms from 2 to 6 SCCM in H_2 flow rate. The scavenging reaction reached saturation at 6 SCCM. The density of HF molecules increased with the increase of H atoms between 8 and 16 SCCM in the H_2 flow rate, where the H atom is abundant.

The density of F atoms between 2 and 6 SCCM in the H_2 flow rate became lower than that at 0 SCCM, since H atoms scavenged F atoms with the formation of HF molecules. Then, the density of F atoms was recovered between 8 and 16 SCCM, indicating that F atoms were produced by the electron-impact dissociation of HF molecules.^{29,30} Thus, the etch rate of Si was decreased between 0 and 6 SCCM in the H_2 flow rate, with decreasing density of F atoms being an etchant for Si. In this regime, the deposition species, which were produced by the scavenging reaction of F, also reduced the etch rate of Si. The etch rate of SiO₂ was maintained, which can be etched by fluorocarbon species (including deposition species on Si surface) as well as by F atoms. Regarding the dissociation reaction of HF molecules, there are two possible paths:²⁹



FIG. 6. Etch rates of HfO₂ and SiO₂ in C_4F_8/H_2 plasmas plotted with that of Si as a function of gas-mixture ratio of $[H_2]/[C_4F_8]$. The fluorocarbon gas flow rate, pressure, power to the coil, and self-bias voltage were maintained at 5 SCCM, 20 mTorr, 280 W, and -90 V, respectively.

$$HF + e \to H + F + e, \tag{3}$$

$$H + HF^* \to H_2 + F, \tag{4}$$

where HF^{*} means vibrationally excited HF molecules, which can be present in hydrogen-added fluorine-containing discharges.^{31–33}

Here we consider the role of H atoms in fluorocarbon plasmas. To elucidate the reaction of H atoms, the etch rates of HfO₂, Si, and SiO₂ were measured in C₄F₈/H₂ and CF₄/H₂ plasmas (shown in Figs. 6 and 7, respectively). The flow rates of C₄F₈ and CF₄ were 5 SCCM and that of H₂ was varied from 0 to 20 SCCM. The power to the coil and selfbias voltage were maintained at 280 W and -90 V, respectively. The pressure was set at 20 mTorr. In the C₄F₈/H₂ plasmas, the etching reaction occurred between 0% and 150% in the gas-mixture ratio of $[H_2]/[C_4F_8]$. The deposition film appeared at more than 150%. However, in CF₄/H₂ plasmas, the surface reactions on HfO₂ and Si were changed from etching to deposition with increasing gas-mixture ratio. Then the reactions were turned into etching again, since the



FIG. 7. Etch rates of HfO_2 and SiO_2 in CF_4/H_2 plasmas plotted with that of Si as a function of gas-mixture ratio of $[H_2]/[CF_4]$. The experimental parameters were the same as in Fig. 6.

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FIG. 8. XPS spectra of F_{1s} and C_{1s} on HfO_2 , Si, and SiO₂ surfaces etched in the $C_4F_8/Ar/H_2$ plasmas and on preetched surfaces (indicated by notation of "Ref."). The experimental conditions were the same as in Fig. 3. The values of additive- H_2 flow rate from 0 to 6 SCCM are shown in the graphs.

deposition film was made thinner with H₂ addition. Furthermore, when the C_4F_8 flow rate is less than 5 SCCM, the thickness of the deposition film is thinner and the etching reaction can occur at more than 400% of the gas-mixture ratio. Therefore, it is essential that the reaction on the Si surface changes from etching to deposition and from deposition to etching with the addition of H₂ atoms. These facts show that the excess H atoms, which do not contribute to the production of HF molecules or are produced by the dissociation of HF molecules, can etch the deposition film of the fluorocarbon polymer. One can understand the etching of fluorocarbon polymer by H atoms by the analogy of the etching of graphite by H atoms in the process of diamond synthesis.³⁴ In Fig. 3, the etching of HfO₂ and Si can proceed from 8 to 16 SCCM in the H₂ flow rate, since etchants (including F atoms) can reach the HfO₂, Si, and SiO₂ surfaces, as a result of the etching of excess H atoms and removing of the deposition film of the fluorocarbon polymer.

Figure 8 shows the XPS spectra of F_{1s} and C_{1s} on HfO₂, Si, and SiO₂ etched in the $C_4F_8/Ar/H_2$ plasmas, and on preetched surfaces (indicated by notation of "Ref."). The experimental conditions were the same as in Fig. 3. In the figure, the numbers from 0 to 6 correspond to the additive- H_2 flow rate. The carbon content on preetched surfaces can be detected from the adventitious hydrocarbon of atmospheric contaminants. On SiO₂ surfaces, intensities of F_{1s} and C_{1s} signals were weak compared with other samples. This means that SiO₂ can be etched by various fluorocarbon species with the formation of volatile products containing F and C atoms. On Si surfaces, the spectra of 2, 4, and 6 SCCM in the additive-H2 flow rate show the chemical composition of the deposition film of fluorocarbon polymer. The F content decreased and C content increased with the increased additive- H_2 flow rate, indicating that the deposition film became more



FIG. 9. Etch rates of HfO_2 in $C_4F_8/Ar/H_2$ (2.5/247.5/16 SCCM), pure Ar (250 SCCM), and CH_4/Ar (12.5/237.5 SCCM) plasmas as a function of ion energy. The pressure and power to the coil were maintained at 20 mTorr and 300 W, respectively.

carbon rich with the increasing additive-H₂ flow rate. On HfO_2 surfaces, especially, at 8 SCCM in the additive-H₂ flow rate, the intensity of the C_{1s} signal was extremely low, which was lower than that of the reference surface contaminated by adventitious hydrocarbon in atmosphere. Furthermore, the C/F ratio on the HfO₂ surface at 8 SCCM was lower than that of the fluorocarbon polymer film on Si surface. These facts imply that the etching of HfO₂ could proceed involving volatile etch products containing C atoms.

To understand species reactive with HfO₂, the etch characteristics of HfO₂ were examined in $C_4F_8/Ar/H_2$ (2.5/247.5/16 SCCM), pure Ar (250 SCCM), and CH₄/Ar (12.5/237.5 SCCM) plasmas. Figure 9 shows the etch rates in these plasmas. Here, the ion energy was defined by $|V_p|$ $-V_{\rm dc}$, where V_p and $V_{\rm dc}$ correspond to plasma potential measured by Langmuir probe and self-bias voltage, respectively. The power to the coil and pressure were maintained at 300 W and 20 mTorr, respectively. The ion densities in the $C_4F_8/Ar/H_2$, pure Ar, and CH_4/Ar plasmas were 1.5×10^{11} , 3.6×10^{11} , and 1.5×10^{11} cm⁻³, respectively. The etch rates in the pure Ar plasmas did not exceed those in the $C_4F_8/Ar/H_2$ plasmas, although the ion density in the pure Ar plasmas was two times higher than that in the $C_4F_8/Ar/H_2$ plasmas. Therefore, the etching of HfO₂ can proceed with involving chemical reactions related to C, F, and H species in the $C_4F_8/Ar/H_2$ plasmas. In addition, the etch rates in the CH_4/Ar plasmas did not exceed those in the pure Ar plasmas and is not dependent on the CH₄ flow rate. The deposition of carbon species suppressed etching of HfO₂.

B. Volatile products in HfO₂ etching

As mentioned above, at least the F species is necessary to etch HfO_2 . The carbon species may also play a role in the etching of HfO_2 , as implied by the results of XPS measurements. Understanding the etch mechanism is one of the most important issues in knowing the etchants of HfO_2 in fluorocarbon plasmas. In this study, a QMA with a mass range from 0.4 to 500 amu was used to observe the ionic species



FIG. 10. Mass spectra of ionic species in C_4F_8/Ar plasmas. The species were observed in the cases without (upper graph indicated by "With bias") and with bias power (lower graph indicated by "Non-bias"). The HfO₂ was etched in the condition with bias power, resulting in self-bias voltage of -90 V.

and the etch products in the gas phase. The ionic species were detected in the C_4F_8 /Ar plasmas, where the C_4F_8 flow rate, pressure, and power to the coil were maintained at 2.5 SCCM (1% of the total), 20 mTorr, and 280 W, respectively (Fig. 10). The various ionic species of fluorocarbon were observed in the plasmas, including those with mass higher than the parent molecule (≥ 200 amu). In the etching of HfO_2 with a self-bias voltage of -90 V, the ionic species as etch products were detected. The several peaks appearing in etching were assigned to Hf⁺, HfF⁺, HF⁺₂, and HF⁺₃ in the spectrum (Fig. 10), compared with the calculated mass patterns using the relative abundance of naturally occurring Hf isotopes, 35 i.e., 177 Hf (18.6%), 178 Hf (27.3%), 179 Hf (13.6%), and 180 Hf (35.1%). These Hf fluoride ions may be produced by the electron-impact dissociation of HfF₄. These cannot be identified to be the primary etch products, nor secondary or higher. It is certain that the Hf fluoride must be included in the etch products as volatile species.

In $C_4F_8/Ar/H_2$ plasmas, the etch products were also observed. The gas flow rates of C₄F₈, Ar, and H₂ were 2.5, 247.5, and 8 SCCM, respectively. The pressure, power to the coil, and self-bias voltage were set at 20 mTorr, 280 W, and -90 V, respectively. These experimental parameters correspond to the condition where HfO₂ can be selectively etched (Fig. 3) and where the etch products may contain C atoms, as implied by the XPS results (Fig. 8). Figure 11 shows the mass spectrum of ionic species with mass ranging from 189 to 201 amu. The spectrum was obtained by subtracting the spectrum in the nonbiased condition from that in the biased condition. Therefore, the spectrum indicates the contents of etch products only. The peaks of the contents can be assigned to $HfCH_{x}^{+}$ (x=0-4) and $HfH_{x}F^{+}$ (x=0-2). The peaks from 193 to 196 amu also correspond to HfO⁺. Since these peaks were not observed in CF₄/Ar and C₄F₈/Ar plasmas, the peaks should be assigned to HfCH₄⁺, whose molecular structure is unknown. Since the molecules of Hf carbide were detected as etch products, a carbon-poor surface prepared in $C_4F_8/Ar/H_2$ plasmas with an additive-H₂ flow rate of 8 SCCM was formed in etching reactions involving carbon species.



FIG. 11. Mass spectrum of ionic species as the etch products with mass range from 189 to 201 in $C_4F_8/Ar/H_2$ plasmas. The additive-H₂ flow rate, pressure, power to the coil, and self-bias voltage were set at 8 SCCM, 20 mTorr, 280 W, and -90 V (with biased condition). The spectrum was obtained by subtracting the spectrum in the nonbiased condition from that in the with biased condition where the HfO₂ could be selectively etched.

The HfCH_x⁺ and HfH_xF⁺ may be produced by the dissociation of molecules with the structure like metal-organic compounds. Although HfC⁺ was detected, the molecule may be produced from HfCH_x⁺, since the yield of C atoms is too low to etch HfO₂.³⁶ It may be natural that HfO₂ is etched with the production of metal-organic compounds as etch products, since such compounds are used for chemical vapor deposition of HfO₂.^{37–39} In addition, it may be possible to etch HfO₂ by CH₄ chemistries with optimized experimental parameters, although the results in CH₄/Ar plasmas (Fig. 9) could not prove this possibility in the present work.

IV. CONCLUSION

In the present study, the etch characteristics of HfO_2 were examined in $CF_4/Ar/H_2$ and $C_4F_8/Ar/H_2$ plasmas. When H_2 was added to the C_4F_8/Ar plasmas, the highly selective etching of HfO_2 over Si could be done. The HfO_2 was etched even in the condition where fluorocarbon polymer film was deposited on a Si surface.

On the HfO_2 surface etched selectively in the $C_4F_8/Ar/H_2$ plasmas, the carbon content was lower than the adventitious hydrocarbon of the atmospheric contaminant on the preetched surface. This implied that the carbon and/or hydrocarbon species may be etchants of HfO₂ and the etch products may contain Hf and carbon atoms. The sputtering rates of HfO₂ in pure Ar plasmas did not exceed the etch rates in the C₄F₈/Ar/H₂ plasmas and were higher than the etch rates in CH₄/Ar plasmas. Therefore, fluorine species are necessary to etch HfO₂ in our examples. In the gas phase, HfCH⁺_x (x=0-4) and HfH_xF⁺ (x=0-2) were detected by QMA. The Hf hydrocarbidelike metal-organic compound was determined to be one of the volatile etch products in the $C_4F_8/Ar/H_2$ plasmas. The formation of metal-organic compounds is an interesting topic for the etching and depositing of materials containing transition metals. Further analyses will be important for the materials introduced to nextgeneration devices.

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In the actual gate processes, the highly selective etching of HfO_2 over Si can be performed with precise control of H_2 addition in $C_4F_8/Ar/H_2$ plasmas. The polymer deposition brought by H_2 addition is effective for reducing the etch rate of Si. Furthermore, the polymer may even prevent etching of HfO_2 in narrow trenches of musks. Therefore, for practical use, the experimental parameters such as additive- H_2 flow rate should be optimized while observing the etch profiles.

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