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# Controlled chemistry of tailored graphene nanoribbons for electrochemistry: a rational approach to optimizing molecule detection

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This work describes a rationalization of the interactions between two fully characterized graphene nanoribbons (GNRs) and a set of significant target molecules. The GNRs were carefully synthesized by unzipping multi-walled carbon nanotubes (MWCNTs) to yield graphene oxide nanoribbons (GNRox) containing 44 wt.% oxygen. The GNRox were reduced to yield reduced graphene oxide nanoribbons (GNRred) containing 14 wt.%. Each material was characterized by atomic force microscopy, transmission electronic microscopy, Raman spectroscopy, X-ray diffraction, Fourier transform infrared spectroscopy, X-ray photoelectron spectroscopy and voltammetry techniques. Differential pulse voltammetry was used to assess the detection of two strategically selected groups of molecules, including benzenediols, hydroquinone, catechol, and resorcinol, as well as, L-dopa, ascorbic acid, uric acid, and L-tyrosine. The results showed that GNRs provided significantly better electrochemical responses compared to MWCNTs and the non-modified glassy carbon electrode. The chemistry of the few layers of graphene strongly influenced the electrochemical properties of the material. GNRox may be the material of choice for sensing molecules having high oxidation potentials. GNRred, on the other hand, yielded an excellent sensitivity for aromatic molecules in which  $\pi$ - $\pi$  interactions were dominant or the number of conjugated 1,2-diols present was high. GNRred combines the advantages of the high proportion of  $sp^2$ -carbon atoms with the presence of a few oxygen moieties remaining in the lattice after the reduction step. The primary interactions responsible for the shift in oxidation potentials were elucidated. This work presents new opportunities for tailoring graphene to a particular sensing application based on the specific chemistry of the molecule.

## INTRODUCTION

Graphene is a two-dimensional (2-D) sheet of carbon atoms connected by  $sp^2$  bonds. The graphene structure conveys extraordinary properties<sup>1-3</sup> to the material, such as a high surface area (theoretically 2630 m<sup>2</sup>/g for single-layer graphene) that is twice the surface area of single-walled carbon nanotubes (SWCNTs). Graphene also shows excellent thermal ( $k=5 \times 10^3$  Wm<sup>-1</sup>K<sup>-1</sup>) and electrical conductivities ( $\sigma=64$  mS cm<sup>-1</sup>). The physical properties of graphene include good optical transparency, a high mechanical strength (Young's modulus, ~1100 GPa), and a high elasticity. The high surface area, high electrical conductivity, and low production costs, in particular, are of interest for electrochemical applications.

Graphenes include graphitic structures that are dimensionally limited to a few to hundred nanometres along the basal plane of the graphene sheets in the x-y plane, such as graphene nanoribbons (GNRs), which can include single- (G-SL), few- (G-FL), or multilayer (G-ML) structures<sup>4</sup>. GNRs may be thought of as unzipped carbon nanotubes (CNTs) that have been created with structural control during the unzipping process. The synthesis of GNR<sup>5,6</sup> is carried out by applying plasma etching to

CNTs embedded in a polymer film<sup>7</sup>, by chemically oxidizing CNTs<sup>8,9</sup>, by ionic liquid-assisted splitting of CNTs under microwave radiation<sup>10</sup>, by the intercalation of metals<sup>11</sup> or nanoparticles<sup>12</sup> into CNTs, and the posterior exfoliation or unwrapping of MWCNTs using electrical currents and nanomanipulation<sup>13</sup>, or by bottom-up strategies<sup>14</sup> for producing GNRs of a desired width and length. The main applications<sup>15</sup> of GNRs occur in the fields of physics, nanoelectronics, spintronics, and nanoelectromechanical systems (NEMS)<sup>16</sup>, but GNRs are also relevant to sensing and biosensing applications.

Although the longitudinal unzipping techniques used to synthesize GNRs result in over-oxidation and a plethora of defect sites that do not in general benefit electronic applications<sup>17</sup>, these characteristics can actually be useful for certain electrochemical applications, as the authors have proved.

Electrochemical applications using graphene have been extensively discussed in different reviews reported in the literature<sup>18-21</sup>.

On the other hand, the corresponding applications of GNRs have not been extensively explored. The edge chemistries of chemically functionalized GNRs may offer certain advantages over the edge chemistries of non-functionalized graphene. Non-functionalized graphene presents an inert chemical surface. By

contrast, the functional moieties located at the edges of GNRs facilitate the adsorption of molecules by  $\pi$ - $\pi$  stacking, electrostatic, hydrogen bonding, and covalent interactions<sup>22</sup>.

A handful of studies have examined the use of GNRs in electrochemical sensing applications. The electrochemical sensing of model electroactive molecules in the presence of reduced GNR-modified screen-printed electrodes has been shown to display a higher sensitivity compared to the sensitivity of a bare screen-printed electrode<sup>17</sup>. Various GNR-based sensors and biosensors<sup>23-25</sup> have been developed for the detection of urea<sup>26</sup>, glucose<sup>27</sup>, 1-hydroxypyrene<sup>28</sup>, cysteine<sup>29</sup>, brevetoxin B<sup>30</sup>, and 2,4,6-trinitrotoluene<sup>31,32</sup>. These sensors displayed excellent electrochemical responses in terms of reproducibility, a low detection limit, and a high selectivity in all cases. In other approaches, GNRs have been incompletely unzipped to develop mixtures of GNR/MWCNTs. The utility of these mixtures for sensing electroactive molecules<sup>33</sup> has been examined, yielding good responses.

To explore these features mentioned above and elucidate the chemical interactions between target molecules and graphene in the context of electrochemical sensing, a set of analytically significant target molecules were evaluated using differential pulse voltammetry (DPV). We tested the detection of several target molecules, including three dihydroxybenzene isomers widely used in the chemical industries<sup>34</sup>: catechol (CT), resorcinol (RS), and hydroquinone (HQ); the neurotransmitter L-dopa (Levodopa/L-3,4 dihydroxyphenylalanine, LD); the amino acid L-tyrosine (L-Tyr); uric acid (UA) and ascorbic acid (AA), which are present in urine and blood serum<sup>35-37</sup>. These target molecules are usually oxidized at approximately the same potential; therefore, discrimination among these species in a mixture can be extremely difficult using most solid electrodes<sup>38</sup>.

In the bibliography interactions between molecules and graphene have been reported using density functional theory and quantum physics<sup>39, 40</sup>. The studies described here sought to (1) explore the electrochemical performances of fully characterized GNRs with respect to the detection of significant target molecules, and (2) elucidate the chemical interactions between graphenes and the target molecules. Since GNRox were obtained from MWCNTs and the GNRred were obtained from the chemical reduction of GNRox, a reliable and valuable comparison with the critical controls such as the GCE and MWCNTs could be made. Therefore, this traceability is of paramount importance for ascertaining the advantages of GNRs.

## MATERIALS AND METHODS

### Reagents, standards and samples

The sodium dihydrogen phosphate and disodium hydrogen phosphate used to prepare a phosphate buffered solution (PBS) and the potassium hexacyanoferrate (III) were purchased from Panreac, (Badalona, Spain). LD, UA, RS, HQ, and CT were purchased from Sigma-Aldrich (St. Louis, MO, USA). AA and L-Tyr were obtained from Fluka Chemika (Buchs, UK). The cosmetic sample (Pigmentasa formulation containing 4% (w/w) HQ) was acquired in a local market. Urine samples were

Standard solutions were prepared in 1 mM in 0.1 M phosphate

buffered solution (PBS) at pH=7.4, adjusted using NaOH. All working solutions were protected from light and prepared daily. A 0.1 g ( $\pm 0.0001$ ) sample of the Pigmentasa formulation was diluted in 10 mL PBS, sonicated in an ultrasonication bath for 10 min, filtered through a 0.2  $\mu$ m Nylon filter, and diluted 4-fold prior to analysis. Urine samples were diluted 25-fold prior to analysis.

All solutions were prepared with Milli-Q water produced in a Milli-Q system (Millipore, Bedford, MA, USA).

### Graphene nanoribbon samples

MWCNTs (0.2% oxygen content) were produced by the arc-discharge method<sup>41</sup> in a home-built electric arc-discharge apparatus under standard conditions<sup>42</sup>. The MWCNTs were characterized as being straight and highly graphitized.

GNRox, 44 wt.% oxygen, were synthesized from the MWCNTs via the longitudinal unzipping method in  $H_2SO_4/KMnO_4$ <sup>8</sup>. These oxidized nanoribbons were used as starting materials to produce GNRred, 14% oxygen, via chemical reduction with  $N_2H_4/NH_3$ <sup>43</sup>.

### Apparatus and measurements

Atomic force microscopy (AFM) images were recorded using a Nanoscope IIIa scanning probe microscope (Digital Instruments, USA) operated in the dynamic and contact modes. In both cases, silicon cantilevers (Veeco) with a force constant of  $\sim 40$  N/m and a nominal radius of 8 nm were employed. The samples used for the measurements were prepared by drop-casting 0.5  $\mu$ L of the graphene suspension (0.1 mg/mL) on the surface of a silicon wafer (in the case of the GNRox studies) or on a mica surface (in the case of the GNRred studies). The measurements were obtained in the surroundings of the wet drop where the concentration and aggregation of graphene sheets was lower. Transmission electron microscopy (TEM) images were collected on a JEOL microscope model 2000 FXII at an acceleration potential of 200 kV, which yielded a maximum resolution of 0.28 nm. Raman spectra were obtained on a Micro-Raman confocal spectrophotometer model Horiba Jobin Yvon HR800 UV using a green laser at 532 nm, which yielded a resolution of 0.4  $cm^{-1}$ . X-ray photoelectron spectroscopy (XPS) was carried out on an ESCAPlus Omicron outfitted with a Mg anode and operated at 1253.6 eV with a power of 150 W (14 mA, 10 kV). X-ray diffraction (XRD) measurements were obtained using an X-ray diffractometer Bruker D8 Advance Series. The oxygen content of the graphene samples was direct determined using a Flash 1112 analyzer from Thermo Fisher Scientific.

All electrochemical measurements were performed on an electrochemical station  $\mu$ -AUTOLAB type II (Ecochemie, Utrecht, Holland) using a conventional three-electrode system comprising a platinum wire as an auxiliary electrode, a silver/silver chloride, 3 M KCl (Ag/AgCl) reference electrode (CH Instrument, China), and a glassy carbon electrode (GCE) 3.0 mm in diameter (BAS Instrumental, Warwickshire, UK) as the working electrode. Electrochemical experiments were performed at room temperature.

### Procedures

## Preparation of the graphene-modified electrode

GNRox was dispersed in water by bath ultrasonication to form a 0.5 mg mL<sup>-1</sup> colloidal dispersion. GNRred was dispersed to obtain a 0.5 mg mL<sup>-1</sup> dispersion in water/NH<sub>3</sub> (1% v/v) by ultrasonication in a bath for 30 min, followed by tip sonication using a VCX130, (Sonics, Newtown, USA) for 2 minutes at 130 W. Interestingly, this last step was highly important for producing excellent dispersions of the GNRred (see Fig. S1). The MWCNTs were dispersed in dimethylformamide (DMF) by ultrasonication to form a 0.5 mg mL<sup>-1</sup> dispersion.

Prior to drop-casting deposition, the GCE was in turn polished using 0.1 and 0.05 μm alumina powders and sequentially sonicated in Milli-Q water and anhydrous ethanol. The GNR and MWCNT-modified electrodes were prepared by casting 10 μL of the GNR solutions (oxide or reduced) or MWCNTs dispersions on the GCE surface (see Table S1).

## Electrochemical procedures

The electrochemical effective surface areas of the bare and modified GCE were estimated based on the slope of the plot of  $Q$  vs.  $t^{1/2}$  obtained by chronocoulometry based on Equation 1, as described by Anson<sup>44, 45</sup>, using 0.45 mM K<sub>3</sub>[Fe(CN)<sub>6</sub>] in 0.1 M PBS (pH=7.4).

$$Q(t) = \frac{2nFAcD^{1/2}t^{1/2}}{\pi^{1/2}} + Q_{dl} + Q_{ads} \quad (\text{Equation 1})$$

In this equation,  $A$  is the effective electrochemical surface area of the working electrode (cm<sup>2</sup>),  $c$  is the concentration of the electroactive species (mol/cm<sup>3</sup>),  $n$  is the number of transfer electrons (that is, 1),  $F$  is the Faraday constant, and  $D$  is the diffusion coefficient,  $7.6 \times 10^{-6}$  cm<sup>2</sup>s<sup>-1</sup>.  $Q_{dl}$  is the double layer charge, which could be eliminated by background subtraction, and  $Q_{ads}$  is the Faradaic charge.

DPV was used for the voltammetry analysis with a pulse width of 0.05 V, a pulse frequency of 0.05 s, a pulse cycle of 0.2 s, a pulse interval of 0.004 V, and a standing time of 2 s.

## RESULTS AND DISCUSSION

### Characterization of the graphene nanoribbons

The structures and morphologies of the graphene samples were characterized using AFM, as shown in Fig. 1. This figure shows AFM images for a small GNRox sheet with an average thickness of about 0.8 nm and an area of 200 x 150 nm<sup>2</sup>. The theoretical thickness of a perfectly flat unoxidized sp<sup>2</sup>-carbon-atom network is predicted to be 0.4 nm<sup>1</sup>. The thickness of the GNRox sample measured here was consistent the thickness of two stacked graphene sheets; however, this thickness may also indicate a single layer graphene structure having oxygen functionalities on the surface<sup>47</sup>. Fig. 1 shows that the GNRred surface was rough due to the presence of stacked small fragments of the reduced graphene sheets. The average thickness of this graphene sample was 1.2 nm, suggesting that the stacked layers extended over an area of 120 x 50 nm<sup>2</sup>. The GNRred sample displayed a larger number of layers than the graphene oxide sample because the high proportion of sp<sup>2</sup>-carbons in GNRred increased the extent of π-π stacking interactions.

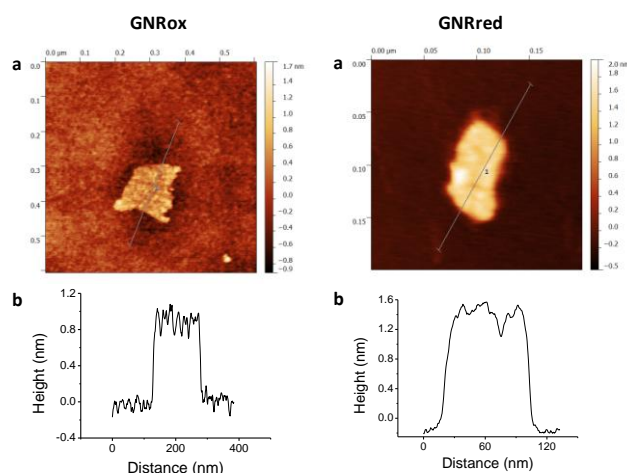


Fig. 1. (a) Tapping mode AFM images of GNRox (left) and GNRred (right) on silica and freshly cleaved mica substrates, respectively. (b) Height profiles along the dashed lines indicated in the panels (a).

The graphene morphologies were imaged using TEM, which also demonstrated the successful synthetic process based on the chemical unzipping of MWCNTs. Fig. 2 shows TEM images of a MWCNT sample (A), GNRox (B), and GNRred (C). Although the MWCNTs were more than 1 μm long and had an outer diameter of approximately 12 nm; GNRox displayed stacked layers and folds, and GNRred appeared to form thin layers with folds in the sheets. These micrographs could be used to visualize the openings of the MWCNTs used to generate the graphene layers, as well as the anisotropy of the GNR structures.

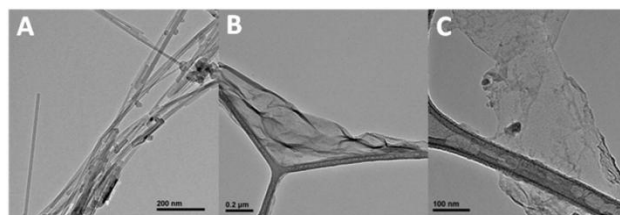


Fig. 2. TEM images of the MWCNTs (A), GNRox (B), and GNRred (C) samples.

X-Ray diffraction studies were performed to estimate the average distance between layers in the carbon allotropes. The crystalline structures of graphite, graphene, and carbon nanotubes permit the measurement of the inter-planar spacing and lattice parameters (see SI, Fig. S2). The XRD patterns obtained from the MWCNTs presented a peak at 26° corresponding to a basal plane of  $d_{002}=3.34$  Å. This peak matched the distance found in the graphite layer structure. The GNRox sample displayed a new peak at 10°, which was attributed to a plane at  $d_{001}=7.33$  Å. The separation between layers in the GNRox sample was high due to the presence of oxygen moieties in the lattice. The GNRred peak intensity at 10° was lower and the peak at 26° was higher than the corresponding peak intensities obtained from the GNRox diffractogram. The recovery of sp<sup>2</sup>-carbon in GNRred facilitated π-π stacking among the layers, and the distance between layers was smaller than the interlayer distance measured in the GNRox sample. XRD studies confirmed the change in the distance between these crystalline structures and the presence of a new graphitic structure.

The Raman spectra of graphite-derived materials usually display a D band at 1360 cm<sup>-1</sup> and a G band at 1590 cm<sup>-1</sup>, and an

overtone of the D band occurs at  $2650\text{ cm}^{-1}$  (the 2D or G' band)<sup>7,48</sup>. The D band arises from the out-of-plane vibrational modes and is indicative of the number of  $\text{sp}^3$  carbon atoms present, whereas the G band arises from the presence of in-plane  $\text{sp}^2$  vibrations. The intensity ratio of the D and G lines ( $I_{\text{D}}/I_{\text{G}}$  ratio), therefore, provides important information about the composition and domains in-plane giving valuable information regarding the average size of the  $\text{sp}^2$  carbon domains as well<sup>49, 50</sup>. **Fig. S3** illustrates the Raman spectra of the three materials, showing the D, G, and G' bands. The  $I_{\text{D}}/I_{\text{G}}$  ratio was calculated to be 0.075, 0.52, and 0.66 for the MWCNTs, GNROx, and GNRred samples, respectively. This increase also suggested a decrease in the average size of the  $\text{sp}^2$  graphitic domains, suggesting that the new graphitic domains created in the GNRred sample were smaller in size but more numerous than in the GNROx sample<sup>48</sup>. Because the  $I_{\text{D}}/I_{\text{G}}$  ratio is proportional to the average size of the  $\text{sp}^2$  carbon domains, a higher  $I_{\text{D}}/I_{\text{G}}$  was attributed to the presence of additional edges (more defects)<sup>51</sup> and shorter layers in the GNRred surface, consistent with the AFM results.

XPS was used to study the oxygen content and changes in the  $\text{sp}^2$ - $\text{sp}^3$  carbon structure in the graphene layers after chemical reduction of GNROx to obtain GNRred. The presence of  $\text{sp}^2$ -carbon atoms increased significantly and the presence of  $\text{sp}^3$ -carbon and oxygen moieties decreased correspondingly upon reduction of GNROx to GNRred. The presence of carboxyl, carbonyl, alcohol, epoxy, and ether moieties decreased to the same extent. The XPS results revealed the presence of C-N bonds in the as-synthesized GNRred sample as a result of the  $\text{N}_2\text{H}_4/\text{NH}_3$  reduction step (see **Fig. S4**). Considering that XPS measurements are sensitive to the elemental composition of the material surface, XPS offers an accurate measure of the oxygen content in the material. Direct determination of oxygen content was directly determined to be 0.2 wt.% in the MWCNTs, 44 wt.% in the GNROx, and 14 wt.% in the GNRred. Clearly, these data differed slightly from the XPS results because the XPS technique is not sensitive to the presence of oxygen groups below the surface and because some of the groups may be lost due to decomposition in the presence of the harsh XPS measurement conditions (X-ray irradiation under high-pressure conditions). The IR spectra were evaluated to corroborate the previous data (see **Fig. S5**).

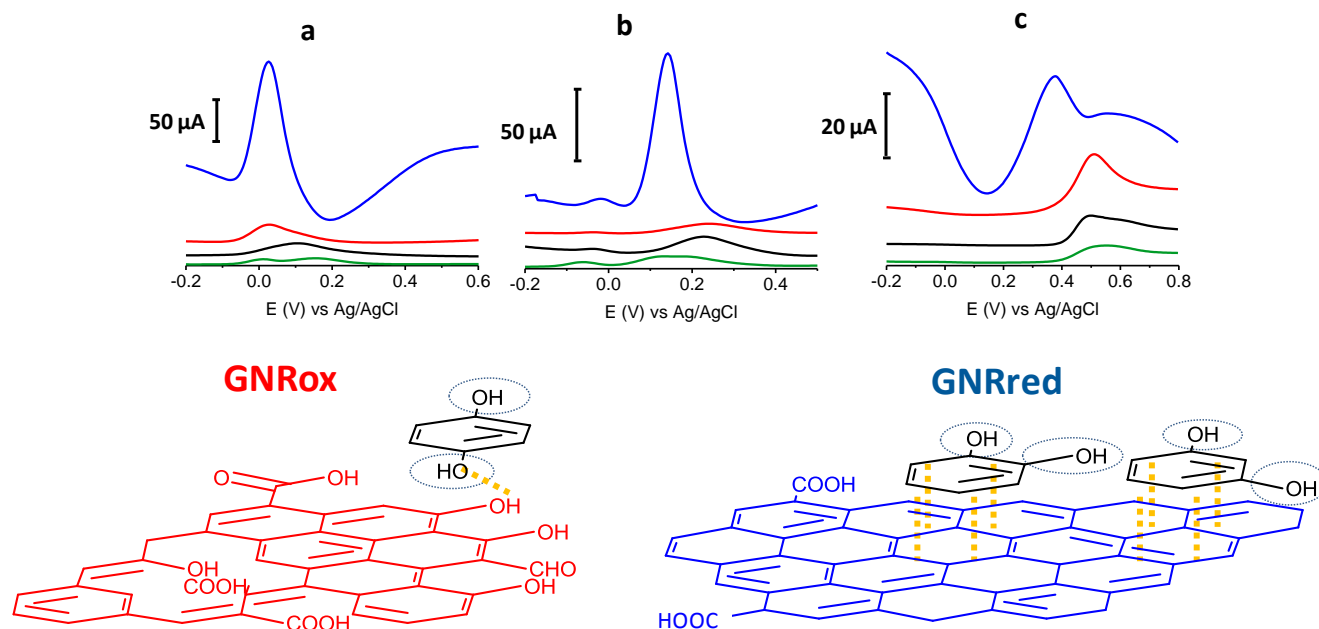
The dispersion of the graphene material was of paramount importance for obtaining these results (see **Fig. S1**). Stronger sonication conditions were required to prepare the GNRred dispersion. Because GNRred includes high  $\text{sp}^2$  content, the material readily stacks to form small piles on an electrode surface

that increase the resistivity and reduce the current. The strongest sonication conditions were not necessary to obtain a good GNROx dispersion because the layers did not stack as readily, the material was not found to accumulate, and the signals obtained with or without tip sonication were indistinguishable.

In the last analytical characterization step, the casting electrode was prepared using a 10  $\mu\text{L}$  drop of the GNR suspensions (see **Table S1**) and the effective electrochemical surface area was evaluated by chronocoulometry (see **Fig. S6**). The estimated areas were 0.030, 0.071, 0.113, and 0.267  $\text{cm}^2$  for the non-modified GCE, MWCNTs, GNROx, and GNRred electrodes, respectively. These results revealed that GNROx and GNRred significantly increased the electrochemical surface area of the electrode. This increase corresponded to a 4-fold increase, in the case of GNROx, and a 10-fold increase, in the case of GNRred, over the electrochemical surface area of the bare GCE. The surface areas were higher than the surface area of the MWCNTs by factors of 2 and 5 for the GNROx and GNRred samples, respectively. Good interelectrode precision ( $n=3$  electrodes) was obtained, with relative standard deviation (RSD) values of 2, 7, and 6% for the MWCNTs, GNROx and GNRred, respectively. These results indicated that the GNRs offered consistently high and reproducible electroactive areas, as predicted.

#### Interactions between the target molecules and the GNRs based on voltammetry studies

The electrochemical behaviors of the isomers HQ (1,4-diol), CT (1,2-diol), and RS (1,3-diol) were explored on bare GCE, and all the carbon materials (MWCNTs, GNROx, and GNRred) on the GCE, respectively. **Fig. 3 Top** illustrates the electrochemical responses of each electrode to the target molecules. The extraordinarily high oxidation peak currents on the graphene-modified electrodes, as compared to the bare electrode, in the presence of the three target molecules reflected the high conductivity of the GNRs. The high conductivity of the GNRred electrode was particularly remarkable as a result of the high  $\text{sp}^2$ -carbon content, which increased the electrical conductivity of this material. As an example, the electrochemical response of GNRred to catechol was one order of magnitude higher than the response to HQ, (see **Table S2**). The presence of incompletely unzipped MWCNTs, if any, was not expected to significantly affect the detection performance because the MWCNT electrode displayed a low signal level.



**Fig. 3 (Top)** Differential pulse voltammograms obtained from 1 mM HQ (a), CT (b), and RS (c). (—) Bare GCE, (—) MWCNTs, (—) GNRox, and (—) GNRred. (See the experimental section for a description of the working conditions). **(Bottom)** Schematic diagram of the interactions between the target molecules (black) and the graphene materials which showed the lower oxidation potential. The oxidation reaction center is indicated by the blue circles and the main interactions are indicated in yellow.

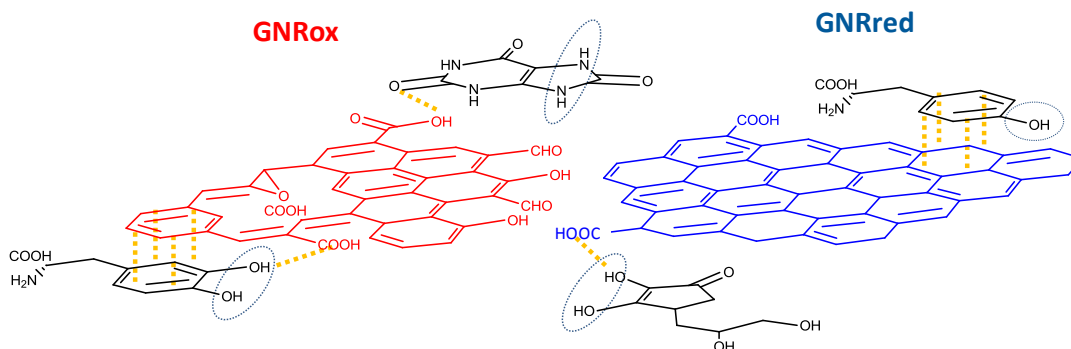
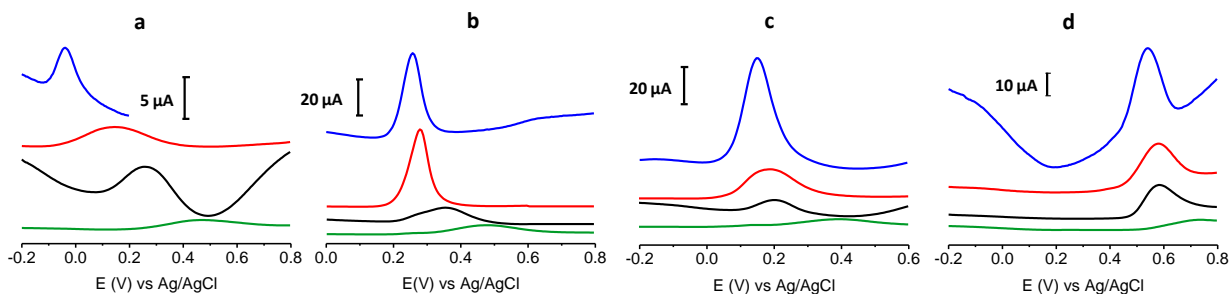
The chemistry underlying the shifts in oxidation potentials in the electrochemical detection of the target molecules may be understood as follows. 1,4-Benzenediol readily oxidizes at low potentials to produce 1,4-benzoquinones, whereas the oxidation of 1,3-diol is less favorable due to the lack of conjugation, thereby preventing the formation of the benzoquinone, as it occurs with 1,4 and 1,2-benzenediols. **Fig. 3 Bottom** illustrates the possible interaction processes between the target molecules and the graphene material. We hypothesized that the availability of more  $\pi$ - $\pi$  interactions in GNRred than in GNRox facilitated the electrocatalysis of CA and RS and, as a consequence, facilitated oxidation of both benzenediols with GNRred. By contrast, 1,4-benzenediol displayed similar oxidation potentials on both GNRs, suggesting that hydrogen bonds dominated the interactions between the oxide moieties on the edges of the GNR surfaces and the molecules, due to the similar potential of the HQ in the presence of GNRred and GNRox.

**Fig. 4 Top** presents the detection results of LD, AA, L-Tyr, and UA on GCE, MWCNTs, GNRox, and GNRred. A comparison of the graphene modified electrodes and the GCE and MWCNTs controls revealed that both graphene electrodes displayed significantly higher current peaks compared to the MWCNT control. The results obtained from GNRred were particularly spectacular. The exceptional electrocatalytic properties exhibited by this material enabled the detection of AA,

even below 0 V. Moreover, GNRred displayed analytical response from 5 to 10-fold higher compared to the response of the GCE.

**Table S3** summarizes the oxidation potentials and analytical signals obtained during the DPV measurements for the four molecules.

The enhanced sensitivity of the GNRred electrode was attributed to the good conductivity of the material. These results may be understood in terms of the previous results obtained in analytical characterization studies, which identified a high electrochemical surface area and the presence of high  $sp^2$ -carbon content. **Fig. 4 Bottom** provides a schematic diagram illustrating our rationalization of the interactions between the GNR and the target molecules. The oxygen groups of the GNRox layers and those oxygen functionalities remaining in the GNRred lattice (this latter material supposed to be in the form of carboxylic acids and carbonyls, as described in the model proposed by Lerf-Klinowski<sup>43,52,53</sup>) are expected to interact with the target molecules via hydrogen bonds. In the detection of AA, GNRred yielded the strongest electrocatalysis compared to UA, where similar electrocatalytic effect was found for both graphenes. This shift could be explained considering interactions with the oxidation center of AA and GNRred and without this oxidation center in the case of UA.



**Fig. 4. (Top)** Differential pulse voltammograms for 1 mM ascorbic acid (a), LD (b), UA (c) and L-Tyr (d). (—) Bare GCE, (—) MWCNTs, (—) GNRox and (—) GNRred. (See experimental section for working conditions). **(Bottom)** Schematic diagram illustrating the interactions between the analytes (black) and graphene surfaces which showed the lower oxidation potential. The oxidation reaction centers are indicated by the blue circle and the main interactions are indicated in yellow.

LD, which basic structure derived from CT, displayed indistinguishable behavior for GNRred and GNRox. Therefore, the predominant  $\pi$ - $\pi$  interactions plus hydro-gen interactions between the benzenediol group and the GNRred and GNRox surfaces appeared to be responsible for the similar responses.

The oxidation process of L-Tyr was more difficult due to the presence of only one hydroxyl group. For this reason the similar oxidation potentials observed on all carbon materials (MWCNTs, GNRox, and GNRred) could be explained in terms of weak  $\pi$ - $\pi$  interactions between the tyrosine and the carbon materials.

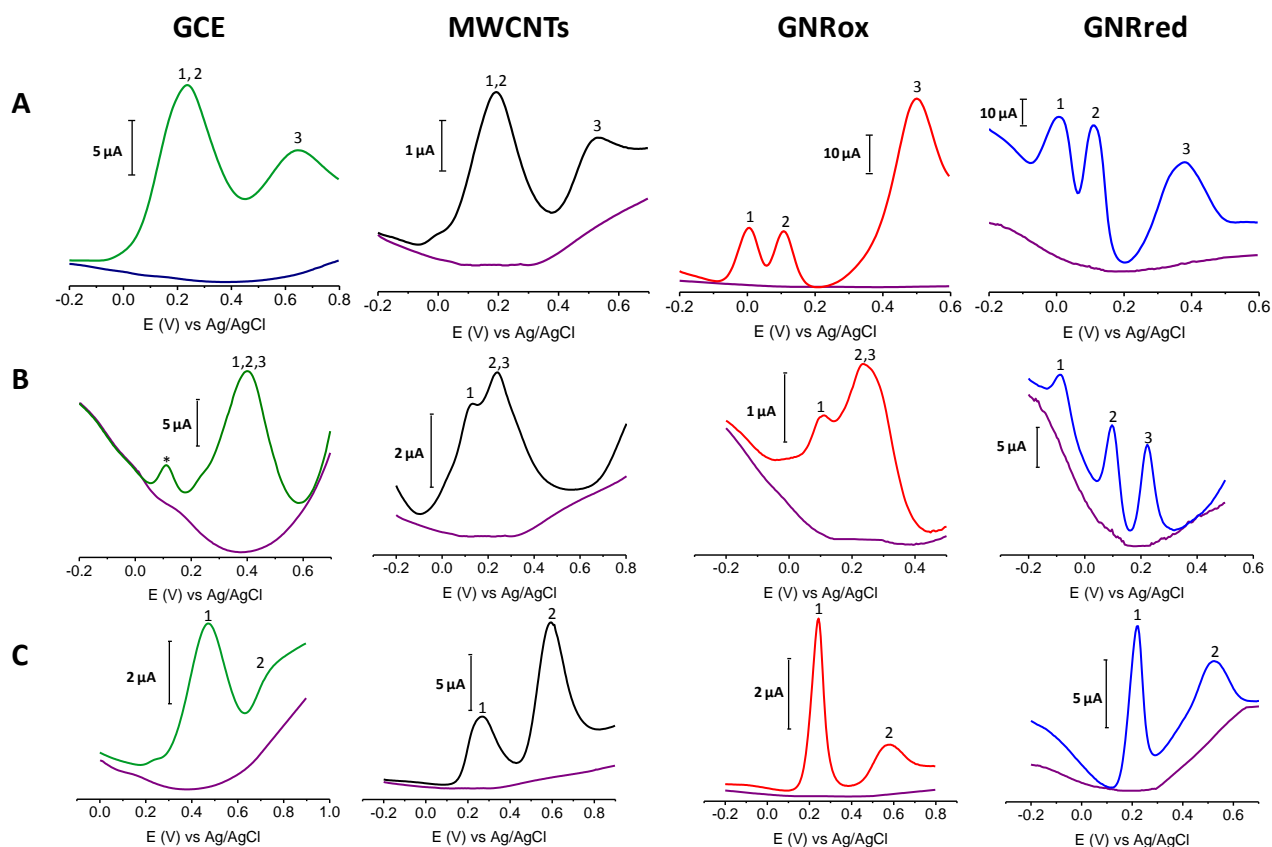
**Fig. 5A** shows the electrochemical responses of mix-tures of the three isomers: HQ, CT, and RS on GCE, MWCNTs, GNRox, and GNRred. The responses revealed that although the bare GCE did not permit the simultaneous detection of the benzene 1,2 and 1,4-diols, the electrocatalytic properties of GNRox and GNRred allowed the simultaneous and separate detection of the three target molecules. **Fig. 5B** shows that the selective detection of LD, AA, and UA could only be achieved using the GNRred electrode. **Fig. 5C** shows that the separate components of a mixture comprising tyrosine and UA could be identified using the GNRred electrode as well. The detection of individual target molecules in a mixture relies on the availability of distinct interactions between each nanomaterial and target molecule, and consequently, to the exceptional electrocatalytic properties

exhibited.

The strength of the voltammetric studies was examined by measuring the technique's repeatability and repro-ducibility. The repeatability (multiple experiments con-ducted on the same day) and the reproducibility (multiple experiments conducted on different days) of the oxidation peak positions was found to be excellent, with RSD values of <4% (n=10). Good inter-electrode precision was achieved, with RSD values of <6% (n=5, same day) and RSD<9% (n=5, different days), for the GNRred electrode. The precision was significantly better than the precision obtained from the GCE electrode (RSD<10% n=4 electrodes, same day).

It is worth noting that real samples were also tested. **Fig. S7** illustrates the electrochemical detection of HQ and UA in cosmetic and urine samples, respectively. Interestingly, in both cases, the GNRs displayed better intensity currents than the GCE. The excellent results in complex matrix suggest that this technique is suitable for the analysis of complex samples. Good precision was achieved for both samples. The cosmetic sample analysis was characterized by an RSD of <0.5% for the oxidation peaks and an RSD of <2% for the peak currents. The urine sample analysis was characterized by an RSD of <0.5% for the oxidation peaks and an RSD of <10% for the peak currents.





**Fig. 5.** (A) Voltammograms for the detection of: HQ 0.25 mM (peak 1); CA 0.25 mM (peak 2) and RS 1.0 mM (peak 3). (B) AA 0.5 mM (peak 1); LD 0.05 mM (peak 2), and UA 0.05 mM (peak 3). (C) UA 1.0 mM (peak 1) and L-Tyr 1.0 mM (peak 2). (—) Background signal, (—) Bare GCE, (—) MWCNTs, (—) GNRox, and (—) GNRred. Note: \* Pre-peak oxidation in LD. (See the experimental section for a description of the working conditions.)

## 5 Conclusions

Both graphenes displayed excellent electrochemical behaviour in the detection of the target molecules, being this behaviour rigorously attribute to graphene and to other graphitic materials. The interactions between the target molecules and the GNR materials present new opportunities in the field of electrochemistry. A suitable graphene material may potentially be tailored for a particular detection application with consideration for the relevant degree of oxidation and  $sp^2$  structure in the electrode that would be required to promote hydrogen bonding or  $\pi$ - $\pi$  interactions between the electrode and the target molecule. GNRox appeared to be ideal for the detection of molecules having high oxidation potentials, whereas GNRred displayed a better response to aromatic molecules, such as the conjugated 1,2-diols, which inter-acted with the electrode predominantly through  $\pi$ - $\pi$  interactions.

Although both graphenes exhibited excellent electro-chemical performance, GNRred became an exceptional material. The completely opened GNRred lattices, which displayed an average thickness of 1.2 nm and a high per-centage of  $sp^2$ -carbon were obtained through the syn-thetic methods described here. GNRred exhibited a 10-fold higher electrochemical surface area and a better analytical performance in the context of electrochemical sensing in comparison with the GCE. These GNRred features allowed for improved electrochemical sensing and suggested that this material was suitable for the electrochemical detection of

target molecules. The outstanding electrocatalytic effect performance relies on the presence of a restored  $sp^2$  structure that includes oxygen groups in the GNRred lattice. These combined features yielded excellent electrocatalytic properties due to the effects of both the  $\pi$ - $\pi$  and hydrogen bonding interactions between the molecules and the GNRs. These interactions enhanced electrocatalysis at the primary catalytic sites at the oxidation centers. The studies described here demonstrate that GNRred is a promising material for use in molecular sensing, with very rich chemistry and electrochemistry properties. GNRred combines the advantages derived from both the high proportion of  $sp^2$ -carbon atoms available within the surface layers (similar to the structure of exfoliated graphene, 0.4 nm corresponds to one layer) with the advantages derived from the remaining oxygen moieties present on the surface. The results presented here open new opportunities for electrochemical sensing applications and guide the process of tailoring a suitable graphene electrode material for use in a particular molecular detection application.

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- 1 K. S. Novoselov, A. K. Geim, S. V. Morozov, D. Jiang, Y. Zhang, S. V. Dubonos, I. V. Grigorieva and A. A. Firsov, *Science*, 2004, **306**, 666-669.
- 2 S. Park and R. S. Ruoff, *Nature Nanotechnology*, 2009, **4**, 217-224.
- 3 C. Lee, X. Wei, J. W. Kysar and J. Hone, *Science*, 2008, **321**, 385-388.
- 4 S. Stankovich, D. A. Dikin, G. H. B. Dommett, K. M. Kohlhaas, E. J. Zimney, E. A. Stach, R. D. Piner, S. T. Nguyen and R. S. Ruoff, *Nature*, 2006, **442**, 282-286.
- 5 M. Terrones, *Nature*, 2009, **458**, 845-846.
- 6 Liang Ma, Jinlan Wang and Feng Ding, *ChemPhysChem*, 2013, **14**, 47-54.
- 7 L. Jiao, L. Zhang, X. Wang, G. Diankov and H. Dai, *Nature*, 2009, **458**, 877-880.
- 8 D. V. Kosynkin, A. L. Higginbotham, A. Sinitskii, J. R. Lomeda, A. Dimiev, B. K. Price and J. M. Tour, *Nature*, 2009, **458**, 872-U5.
- 9 C. K. Chua, Z. Sofer and M. Pumera, *Chemistry-an Asian Journal*, 2012, **7**, 2367-2372.
- 10 S. Vadahanambi, J. Jung, R. Kumar, H. Kim and I. Oh, *Carbon*, 2013, **53**, 391-398.
- 11 A. G. Cano-Marquez, F. J. Rodriguez-Macias, J. Campos-Delgado, C. G. Espinosa-Gonzalez, F. Tristan-Lopez, D. Ramirez-Gonzalez, D. A. Cullen, D. J. Smith, M. Terrones and Y. I. Vega-Cantu, *Nano Letters*, 2009, **9**, 1527-1533.
- 12 L. Ci, Z. Xu, L. Wang, W. Gao, F. Ding, K. F. Kelly, B. I. Yakobson and P. M. Ajayan, *Nano Research*, 2008, **1**, 116-122.
- 13 K. Kim, A. Sussman and A. Zettl, *Acs Nano*, 2010, **4**, 1362-1366.
- 14 X. Yang, X. Dou, A. Rouhanipour, L. Zhi, H. J. Raeder and K. Muellen, *J. Am. Chem. Soc.*, 2008, **130**, 4216-4217.
- 15 M. Terrones, A. R. Botello-Mendez, J. Campos-Delgado, F. Lopez-Urias, Y. I. Vega-Cantu, F. J. Rodriguez-Macias, A. L. Elias, E. Munoz-Sandoval, A. G. Cano-Marquez, J. Charlier and H. Terrones, *Nano Today*, 2010, **5**, 351-372.
- 16 Y. Guo, W. Guo and C. Chen, *Appl. Phys. Lett.*, 2008, **92**, 243101.
- 17 D. B. Shinde, J. Debgupta, A. Kushwaha, M. Aslam and V. K. Pillai, *J. Am. Chem. Soc.*, 2011, **133**, 4168-4171.
- 18 M. Pumera, A. Ambrosi, A. Bonanni, E. L. K. Chng and H. L. Poh, *Trac-Trends in Analytical Chemistry*, 2010, **29**, 954-965.
- 19 D. A. C. Brownson and C. E. Banks, *Analyst*, 2010, **135**, 2768-2778.
- 20 Y. Shao, J. Wang, H. Wu, J. Liu, I. A. Aksay and Y. Lin, *Electroanalysis*, 2010, **22**, 1027-1036.
- 21 S. Wu, Q. He, C. Tan, Y. Wang and H. Zhang, *Small*, 2013, **9**, 1160-1172.
- 22 S. Zhang, S. Tang, J. Lei, H. Dong and H. Ju, *J Electroanal Chem*, 2011, **656**, 285-288.
- 23 F. Valentini, D. Romanazzo, M. Carbone and G. Palleschi, *Electroanalysis*, 2012, **24**, 872-881.
- 24 X. Dong, Q. Long, J. Wang, M. B. Chan-Park, Y. Huang, W. Huang and P. Chen, *Nanoscale*, 2011, **3**, 5156-5160.
- 25 F. Valentini, M. Carbone and G. Palleschi, *Analytical and Bioanalytical Chemistry*, 2013, **405**, 3449-3474.
- 26 Y. Yang, J. Zhou, H. Zhang, P. Gai, X. Zhang and J. Chen, *Talanta*, 2013, **106**, 206-211.
- 27 R. K. Srivastava, S. Srivastava, T. N. Narayanan, B. D. Mahlotra, R. Vajtai, P. M. Ajayan and A. Srivastava, *Acs Nano*, 2012, **6**, 168-175.
- 28 X. Shen, Y. Cui, Y. Pang and H. Qian, *Electrochim. Acta*, 2012, **59**, 91-99.
- 29 S. Wu, X. Lan, F. Huang, Z. Luo, H. Ju, C. Meng and C. Duan, *Biosens. Bioelectron.*, 2012, **32**, 293-296.
- 30 J. Tang, L. Hou, D. Tang, J. Zhou, Z. Wang, J. Li and G. Chen, *Biosens. Bioelectron.*, 2012, **38**, 86-93.
- 31 M. S. Goh and M. Pumera, *Analytical and Bioanalytical Chemistry*, 2011, **399**, 127-131.
- 32 S. M. Tan, C. K. Chua and M. Pumera, *Analyst*, 2013, **138**, 1700-1704.
- 33 C. Sun, C. Chang, H. Lee, J. Zhou, J. Wang, T. Sham and W. Pong, *Acs Nano*, 2011, **5**, 7788-7795.
- 34 S. Suresh, V. Srivastava and I. M. Mishra, *IJEEE*, 2012, **3**, 32.
- 35 J. M. Zen, C. T. Hsu, Y. L. Hsu, J. W. Sue and E. D. Conte, *Anal. Chem.*, 2004, **76**, 4251-4255.
- 36 A. Salimi, H. Mamkhezri and R. Hallaj, *Talanta*, 2006, **70**, 823-832.
- 37 K. Reddaiah, T. M. Reddy and P. Raghu, *J Electroanal Chem*, 2012, **682**, 164-171.
- 38 R. D. O'Neill, *Analyst*, 1994, **119**, 767-779.
- 39 Z. Wang, H. Hu, Y. Wei and Q. Huang, *Physica B-Condensed Matter*, 2010, **405**, 3895-3898.
- 40 M. Roos, D. Kuenzel, B. Uhl, H. Huang, O. B. Alves, H. E. Hoster, A. Gross and R. J. Behm, *J. Am. Chem. Soc.*, 2011, **133**, 9208-9211.
- 41 M. V. Antisari, R. Marazzi and R. Krsmanovic, *Carbon*, 2003, **41**, 2393-2401.
- 42 A. M. Benito, W. K. Maser and M. T. Martinez, *International Journal of Nanotechnology*, 2005, **2**, 71-89.
- 43 X. Gao, J. Jang and S. Nagase, *Journal of Physical Chemistry C*, 2010, **114**, 832-842.
- 44 F. C. Anson, *Anal. Chem.*, 1964, **36**, 932-934.
- 45 F. C. Anson and R. A. Osteryoung, *J. Chem. Educ.*, 1983, **60**, 293-296.
- 46 N. P. C. Stevens, M. B. Rooney, A. M. Bond and S. W. Feldberg, *Journal of Physical Chemistry a*, 2001, **105**, 9085-9093.
- 47 M. Zhou, Y. Zhai and S. Dong, *Anal. Chem.*, 2009, **81**, 5603-5613.
- 48 A. Ambrosi, A. Bonanni, Z. Sofer, J. S. Cross and M. Pumera, *Chemistry-a European Journal*, 2011, **17**, 10763-10770.
- 49 E. B. Barros, K. Sato, G. G. Samsonidze, A. G. Souza Filho, M. S. Dresselhaus and R. Saito, *Physical Review B*, 2011, **83**, 245435.
- 50 S. Stankovich, D. A. Dikin, R. D. Piner, K. A. Kohlhaas, A. Kleinhammes, Y. Jia, Y. Wu, S. T. Nguyen and R. S. Ruoff, *Carbon*, 2007, **45**, 1558-1565.
- 51 M. Cheng, R. Yang, L. Zhang, Z. Shi, W. Yang, D. Wang, G. Xie, D. Shi and G. Zhang, *Carbon*, 2012, **50**, 2581-2587.
- 52 A. Lerf, H. Y. He, M. Forster and J. Klinowski, *J Phys Chem B*, 1998, **102**, 4477-4482.
- 53 H. Y. He, J. Klinowski, M. Forster and A. Lerf, *Chemical Physics Letters*, 1998, **287**, 53-56.