

Recipes for enhanced molecular cooling

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Abstract

Molecular nanomagnets are considered valid candidates for magnetic refrigeration at low temperatures. Designing these materials for enhanced cooling requires the control and optimization of the quantum properties at the molecular level, in particular: spin ground state, magnetic anisotropy, and presence of low-lying excited spin states. Herein, we present the theoretical framework together with a critical review of recent results, and perspectives for future developments.

1 Introduction

The Magneto-Caloric Effect (MCE) is based on the change of the magnetic entropy upon application of a magnetic field. All magnetic materials intrinsically show MCE, although the intensity of the effect depends on the properties of each material. Besides the fundamental interest on related thermodynamical properties of novel materials, MCE is of great technological importance since it can be used for cooling applications¹ according to a process known as adiabatic demagnetization.² This energy-efficient and environmentally friendly technique is particularly promising for refrigeration in the ultra-low-temperature region, providing, *e.g.*, a valid alternative to the use of helium-3 which is quickly becoming rare and expensive.³

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The topic of magnetic refrigeration constitutes one of the potential applications envisioned for molecule-based materials.⁴ Recent studies have demonstrated that the MCE of selected molecular nanomagnets can be much larger than in the best inter-metallic and lanthanide alloys, and magnetic nanoparticles, conventionally studied and employed for low-temperature cooling applications. This undoubtedly represents an exciting prospect for these systems, maintaining them in the forefront of investigation in the context of nanoscience and nanotechnology. However in order to be competitive, the molecular magnets need first to be properly designed because characteristics like anisotropy and magnetic interactions among others, set the performance of these materials as coolers. The aim of the present work is to address the physical principles and synthetic strategies which can lead to an enhanced MCE, as well as to provide a brief overview of current activities in this field.

2 Magneto-caloric effect and refrigeration

In order to explain the origin of the magneto-caloric effect, we use thermodynamics which relates the magnetic variables (magnetization M and magnetic field H) to entropy S_E and temperature T . Let us recall⁵ that the definition of the entropy of a system having Ω accessible (and non-degenerate) states is $S_E = k_B \ln(\Omega)$. Since a magnetic moment of spin S has $2S + 1$ magnetic levels, the entropy content per mole of substance associated with the magnetic degrees of freedom between $T = 0$ and $T = \infty$ becomes

$$S_m = R \ln(2S + 1), \quad (1)$$

where $R = N_A k_B$ is the gas constant. The spin S should be considered as an effective spin describing the multiplicity of the states taking part in the magnetic process.

When a material is magnetized by the application of a magnetic field, the magnetic entropy is changed as the field changes the magnetic order of the material. The MCE and the associated principle of adiabatic demagnetization is readily understood looking at Figure 1. The system, assumed to be a paramagnetic material, is initially in state $A(T_i, H_i)$, at temperature T_i and field H_i . Under adiabatic conditions (*i.e.*, when the

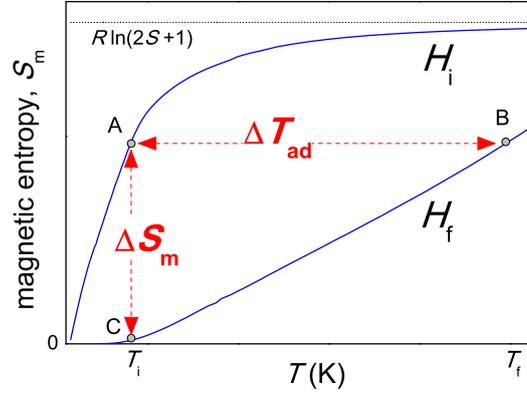


Figure 1: Molar magnetic entropy of a (super)paramagnet with spin S per formula unit, as a function of temperature, for magnetic field H_i and $H_f > H_i$. *AB process*: adiabatic magnetization providing ΔT_{ad} . *AC process*: isothermal magnetization providing ΔS_m .

total entropy of the system remains constant during the magnetic field change), the magnetic entropy change must be compensated for by an equal but opposite change of the entropy associated with the lattice, resulting in a change in temperature of the material. That is, the adiabatic field change $H_i \rightarrow H_f$ brings the system to state $B(T_f, H_f)$ with the temperature change $\Delta T_{\text{ad}} = T_f - T_i$ (horizontal arrow in Fig. 1). If otherwise the magnetic field is isothermally changed to H_f in a reversible process, the system goes to state $C(T_i, H_f)$ with the magnetic entropy change ΔS_m (vertical arrow in Fig. 1). Both ΔS_m and ΔT_{ad} represent the characteristic parameters of the MCE. It is easy to see that if the magnetic change ΔH reduces the entropy ($\Delta S_m < 0$), then ΔT_{ad} is positive, whereas if ΔH is such that $\Delta S_m > 0$, then $\Delta T_{\text{ad}} < 0$ (Fig. 1).

3 Evaluation of MCE from magnetization and heat capacity experiments

In order to establish the relationship between H , M and T to the MCE terms, ΔT_{ad} and ΔS_{m} , we consider the Maxwell equation for the magnetic entropy

$$\left(\frac{\partial S_{\text{m}}(T, H)}{\partial H}\right)_T = \left(\frac{\partial M(T, H)}{\partial T}\right)_H. \quad (2)$$

Integrating Eq. (2) for an isothermal process, we obtain

$$\Delta S_{\text{m}}(T, \Delta H) = \int_{H_i}^{H_f} \left(\frac{\partial M(T, H)}{\partial T}\right)_H dH. \quad (3)$$

This equation indicates that ΔS_{m} is proportional to both the derivative of magnetization with respect to temperature at constant field and to the field variation. The accuracy of ΔS_{m} calculated from magnetization experiments using Eq. (3) depends on the accuracy of the measurements of the magnetic moment, T and H . It is also affected by the fact that the exact differentials are replaced by the measured variations (ΔM , ΔT and ΔH).

Using the following thermodynamic relations:

$$\left(\frac{\partial T}{\partial H}\right)_{S_{\text{m}}} = - \left(\frac{\partial S_{\text{m}}}{\partial H}\right)_T \left(\frac{\partial T}{\partial S_{\text{m}}}\right)_H,$$

$$C = T \left(\frac{\partial S_{\text{m}}}{\partial T}\right)_H,$$

where C is the heat capacity at constant field, and taking into account Eq. (2), the adiabatic temperature change is given by

$$\Delta T_{\text{ad}}(T, \Delta H) = \int_{H_i}^{H_f} \left(\frac{T}{C(T, H)}\right)_H \left(\frac{\partial M(T, H)}{\partial T}\right)_H dH. \quad (4)$$

The measurement of the heat capacity as a function of temperature in constant magnetic field provides the most complete characterization of MCE in magnetic materials. From the experimental heat capacity, the temperature dependence of the magnetic entropy

$S_m(T)$ is obtained by integration, *i.e.*, using

$$S_m(T) = \int_0^T \frac{C_m(T)}{T} dT, \quad (5)$$

where $C_m(T)$ is the magnetic heat capacity as obtained by subtracting the lattice contribution to the total C measured. Hence, if $S_m(T)$ is known for H_i and H_f , both $\Delta T_{\text{ad}}(T, \Delta H)$ and $\Delta S_m(T, \Delta H)$ can be obtained. The accuracy in the evaluation of MCE using heat capacity data depends critically on the accuracy of the C measurements and data processing (*e.g.*, use of ΔT instead of dT). An additional source of uncertainty may eventually be given by the fact that, in order to carry out the integration of Eq. (5), one has to extrapolate the experimental magnetothermal data to $T = 0$ and to $T = \infty$.

4 Advantages of molecular magnets for MCE

Nanomagnets were proposed for magnetic refrigeration by Shull and co-workers⁶ in the 1990s. Nanoclustering the spin moments into non-interacting particles results in a net magnetic moment per particle, which is typically large and therefore yields large entropy according to Eq. (1). Superparamagnetic nanoparticles composed of rare-earth and transition-metal ions have attracted much the recent interest,^{7,8} but the presence of non-active solvent, interparticle interactions and particle size distributions, are all ingredients that negatively affect the performance of these nanostructured materials in terms of the MCE. Molecular magnets are in many ways superior to magnetic nanoparticles, because they exhibit ideal mono-dispersity in size, shape and magnetic moment. Furthermore, their molecular character opens up avenues for a fine tuning of their magnetic properties. As will be exemplified below, this last point is crucial for improving the performance in terms of MCE.

4.1 Molecular anisotropy

The splitting of the molecular energy levels gives rise to a characteristic heat capacity behaviour, that is the Schottky anomaly. Given the electronic configuration and so

the partition function Z of a system, the calculation of its Schottky contribution is straightforward by using the thermodynamic relationship

$$C = \frac{\partial}{\partial T} \left(RT^2 \frac{\partial \ln Z}{\partial T} \right).$$

The general formula of the Schottky anomaly for a set of energy levels E_i and corresponding degeneracies g_i is given by

$$C_{\text{Sch}} = \left(\frac{1}{k_B T} \right)^2 \frac{\sum_{i,j} g_i g_j (E_i^2 - E_i E_j) \exp[-(E_i + E_j)/k_B T]}{\sum_{i,j} g_i g_j \exp[-(E_i + E_j)/k_B T]}. \quad (6)$$

In a superparamagnetic molecular cluster (*a.k.a.* single-molecule magnet), the presence of an anisotropy barrier gives rise to a Schottky anomaly.⁹ The barrier is, indeed, the result of zero-field splitting brought about by crystal-field effects arising from the metal oxidation states and surrounding organic ligands, and sets a preferential direction for the molecular spins. The picture depicted in Figure 1 is still valid for a superparamagnet with S as net cluster moment, provided that it is at temperatures above the blocking temperature. The molecular magnetic anisotropy, which determines the blocking temperature, can be considered as a drawback in the MCE efficiency of superparamagnets. Let us demonstrate this last statement with the following example.

Consider three systems of non-interacting monodisperse magnetic clusters with spin $S = 10$ and *hypothetical* axial anisotropies $D = -0.5, -1.5$ and -3.0 K, respectively. We first calculate the corresponding Schottky heat capacities C_{Sch} according to Eq. (6) obtaining results akin to the ones depicted in the top panel of Figure 2 for the case of $D = -0.5$ K and different applied fields. From these, we calculate the corresponding magnetic entropies $S_m(T, H)$ by using Eq. (5). Finally, by knowing $S_m(T, H)$ we obtain the magnetic entropy changes $\Delta S_m(T, \Delta H) = [S_m(T, H_f) - S_m(T, H_i)]$ for the given field change of $\Delta H = (70 - 10)$ kG. The results are depicted in the bottom panel of Fig. 2, where it can be noticed that the increase of the anisotropy tends to lower the values of $-\Delta S_m$ (and similarly ΔT_{ad}) by shifting them towards higher temperatures. In other words, if the anisotropy is large, the polarization of the molecular spins is less sensitive to H , or (equivalently) higher fields are needed, thus yielding a lower MCE.

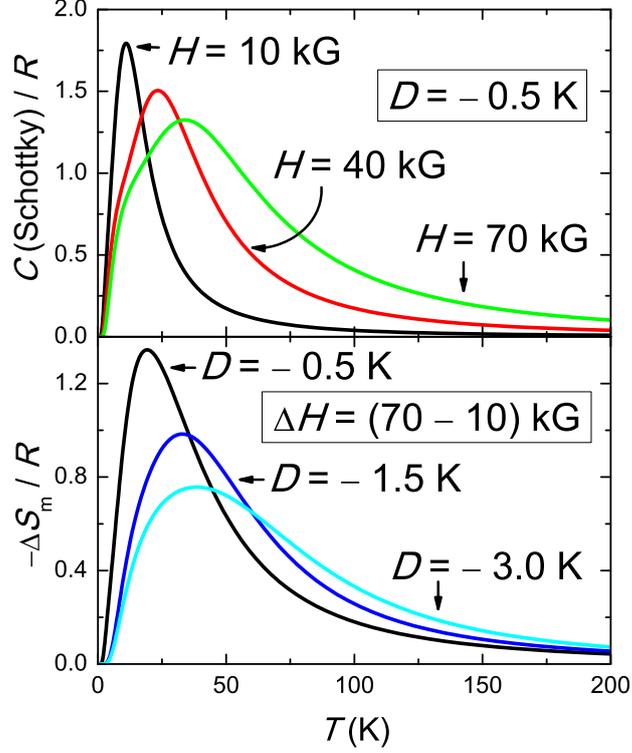


Figure 2: Top: Schottky heat capacities C_{Sch} of an isolated magnetic particle with $S = 10$ and axial anisotropy $D = -0.5$ K calculated for $H = 10, 40$ and 70 kG, respectively. Bottom: Magnetic entropy changes ΔS_m obtained from the top panel data and equivalent systems with $D = -1.0$ and 3.0 K following an applied field change of $\Delta H = (70 - 10)$ T.

Furthermore, it is important to consider that for temperatures below the blocking temperature, the spin-lattice relaxation slows down dramatically, *i.e.* the molecular spins tend to lose thermal contact with the lattice, as revealed again by heat capacity experiments.¹⁰ This off-equilibrium results in lower magnetic entropies and, consequently, lower MCE parameters. Therefore, if we target the highest performance as magnetic refrigerants, we should design the molecular magnets in such a way to have a *negligible anisotropy* which would permit an easy polarization of the net molecular spin, leading to a large magnetic entropy change. The above discussion also demonstrates that, in order to be successful, the applicability of the (isotropic) molecular coolers has to be

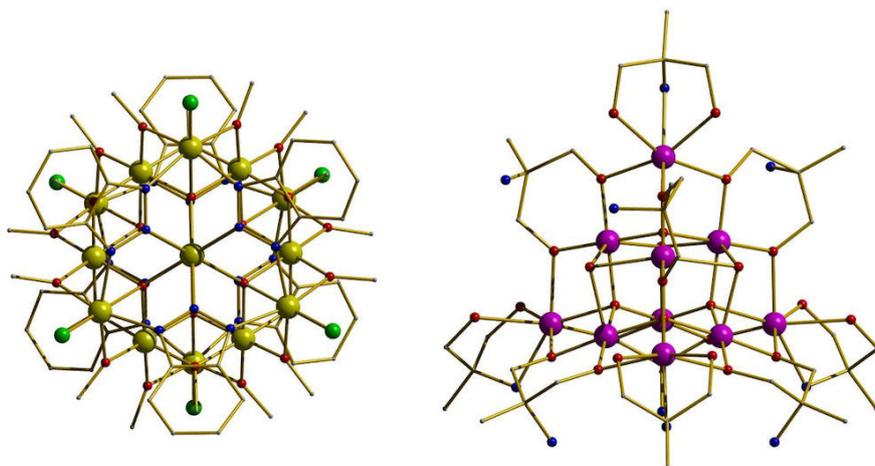


Figure 3: Molecular structures of Fe_{14} (left) and Mn_{10} (right). Yellow Fe, purple Mn, green Cl, red O, blue N. Hydrogen atoms are omitted for clarity.

in the very-low-temperature region.

The archetypal highly-anisotropic Mn_{12} and Fe_8 molecular magnets were the first to be investigated for magnetic refrigeration by Tejada and colleagues,¹¹ Zhang and colleagues,¹² and Tishin and colleagues.⁴ Owing to their well-defined spin ground state $S = 10$ at low temperatures, it is easy to show using Eq. (1) that Mn_{12} and Fe_8 cannot have values of $-\Delta S_m$ exceeding 12.5 and 11 $\text{Jkg}^{-1}\text{K}^{-1}$, respectively. Although these values are relatively large, they also are difficult to observe because the large anisotropy present in both systems freezes the orientation of the molecular spins once the temperature is lowered below (2–4) K. This limits their applicability as refrigerants. The search for isotropic molecular coolers led physicists in Modena together with chemists in Manchester to the heterometallic wheel Cr_7Cd ,¹³ whose limitation is in the low value of the spin $S = 3/2$. It is only very recently that a huge step forward has been accomplished via the synthesis and study of highly-symmetric molecules with high values of the spin ground state. The first was the Fe_{14} with a remarkable $S = 25$ ground state,^{14,15} and second the supertetrahedron Mn_{10} , a *ferromagnetic* cluster with $S = 22$ displaying practically zero anisotropy (Fig. 3).¹⁶ This fascinating combination results in MCE responses nearly three times greater than that of Cr_7Cd , and comparable to that of one of the best performing superparamagnetic nanoparticles, nominally

DyCo₂.⁸

4.2 Molecular spin degeneracy

A property favouring large MCE is the presence of degenerate or low-lying excited spin states, since the so-added degrees of freedom result into an extra magnetic entropy content according to Eq. (1). This situation is encountered, for instance, in magnetically-frustrated systems, as was recently pointed out by Schnack *et al.*¹⁷ and Zhitomirsky,¹⁸ who concluded that in terms of MCE the field-dependent efficiency of a geometrically frustrated magnet can exceed that of an ideal paramagnet with equivalent spin by more than an order of magnitude. In this regard recent results obtained for the magnetically-frustrated Fe₁₄ molecular cluster compound are indeed quite promising.¹⁵

Molecular magnets have recently proved that an alternative and simpler approach to frustration in promoting spin-degeneracy can be obtained by *designing very weak magnetic links between the single-ion spin centers.*^{19–21} Let us present the physics involved in the way in which intramolecular magnetic ordering into a total spin $S_{\text{tot}} = ns$ of a number n of spins s , which are part of the same molecule, can lead to a partial concentration of the total magnetic entropy change into a limited range of temperature. If a molecule is paramagnetic, with n non-interacting spins s , the magnetic entropy per mole is $S_{\text{m}}^{(\text{n-in})} = n R \ln(2s + 1)$ from Eq. (1). In the low-temperature range, however, where the n spins s couple into $S_{\text{tot}} = ns$, the entropy change to consider is $S_{\text{m}}^{(\text{in})} = R \ln(2S_{\text{tot}} + 1) = R \ln(2ns + 1)$, which is clearly different. Of course the total magnetic entropy gain that can be reached between zero and infinite temperature remains equal to $S_{\text{m}}^{(\text{n-in})}$, which is the maximum entropy gain. However, the way in which the magnetic entropy is released as a function of temperature is changed by the presence of the interactions from the smooth gradual temperature dependence of $S_{\text{m}}(T)$ in the paramagnetic case, into a dependence showing more steep behaviour in the temperature range where the interactions become important. This can be used advantageously to produce a large entropy change by means of a limited change in temperature and/or field, that is a much larger change than can be produced in the absence of such interactions, yielding an enhanced MCE.

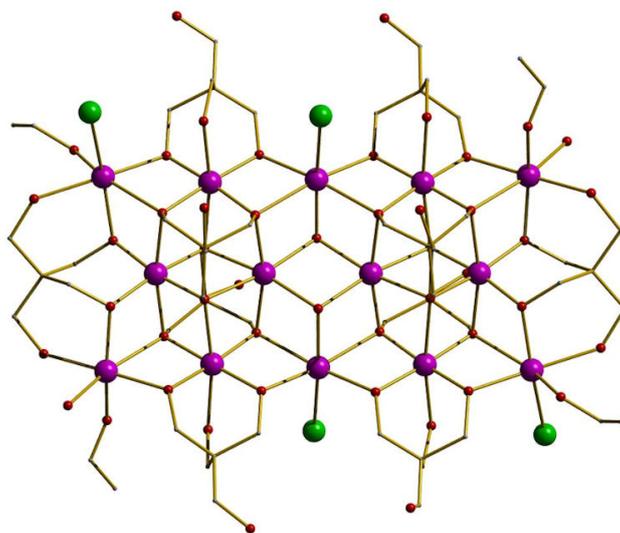


Figure 4: Molecular structures of Mn_{14} . Purple Mn, green I, red O, blue N. Hydrogen atoms are omitted for clarity.

The magnetic ordering in a bulk solid-state materials, *e.g.* pure gadolinium, is conceptually analogous to the aforementioned case of the cluster molecule. For a system of non-interacting Gd spins the temperature dependence of the magnetic entropy will show a smooth behaviour, gradually varying from zero at $T = 0$ to the maximum value, $R \ln(2S + 1)$ with $S = 7/2$ for Gd. When introducing the magnetic interactions, the resulting magnetic ordering at $T_C = 293$ K of the Gd spins will result in a steep fall of the entropy curve below T_C to near zero value in the magnetically ordered region. In principle one could then play similar games, *i.e.* producing large entropy changes by small changes in field or temperature. Unfortunately, the T_C value of Gd metal is too high for any practical purposes (here the magnetic contribution to the total heat capacity is in fact negligible compared to the lattice contribution). It is worth mentioning that the MCE driven by long-range magnetic order at high temperatures has already been investigated in molecule-based materials, nominally Prussian blue analogues.²²

The successful chemistry approach investigated so far for promoting weaker pairwise intramolecular exchanges has been that of making use of bridging hydroxides, Mn^{2+} or lanthanide ions.¹⁹ These strategies resulted in a truly enormous enhancement of the MCE: for instance, values of $-\Delta S_m$ as large as $20 \text{ J kg}^{-1} \text{ K}^{-1}$ for liquid-helium

temperatures and $\Delta H = 60$ kG have been reported in the case of the Mn_{14} molecular magnet (Fig. 4), that is almost a factor of two larger than that of the DyCo_2 nanoparticles.⁸

5 Final remarks and perspectives

While the concept of magnetic refrigeration has been around for a long time, it is with the recent introduction of molecular magnets that the search for ultra-low-temperature coolers has received a huge improvement. These promising materials are the finest refrigerants ever reported for low-temperature applications, that is in a temperature range of efficiency easily accessible with liquid- ^4He . The underlying physics together with the chemical modifications made feasible by their molecular character, led us to identify which specific aspects of the synthetic procedure may (positively) influence the final output. For instance, inducing spin-degeneracy is a major requirement for enhancing the performance in terms of the magneto-caloric effect. In this respect, the synthesis of new molecules based on Gd^{3+} ions, which promote weak intramolecular interactions and which possess the largest isotropic spin ($s = 7/2$) of any ion in the periodic table, would be a good idea.

A drawback in the use of molecular refrigerants resides in the low density of these materials. For practical purposes, this translates into a relatively large volume of the cooling apparatus. However, this shortcoming does not limit their applicability. Contrary to conventional bulk refrigerants, in which the main entropy changes are driven by phase transitions, here the effect takes place in the nanoscale range, *i.e.* at the molecular level. We expect that micron- and submicron-sized devices will soon be fabricated for exploiting the functionality of the molecular coolers. This goal requires the surface grafting of the molecules whilst preserving their magnetic properties, a task which cannot be accomplished if the refrigerant bases its efficiency on cooperative interactions. These mesoscopic devices will then find application as cooling platforms for all those instruments where local refrigeration down to very low temperatures is needed. This can include, although is not limited to, high-resolution X-ray and gamma-ray detectors

for, *e.g.*, astronomy, materials science, and security instrumentation.

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References

- [1] See, *e.g.*, (a) C. Zimm, A. Jastrab, A. Sternberg, V. K. Pecharsky, K. A. Gschneidner Jr., M. Osborne and I. Anderson, *Adv. Cryog. Eng.*, 1998, **43**, 1759; (b) V. K. Pecharsky and K. A. Gschneidner Jr., *J. Magn. Magn. Mater.*, 1999, **200**, 44; (c) K. A. Gschneidner Jr., A. O. Pecharsky and V. K. Pecharsky, in *Cryocoolers II*, ed. R. S. Ross Jr., Kluwer Academic/Plenum Press, New York, 2001, p. 433.
- [2] Originally described by (a) P. Debye, *Ann. Phys.*, 1926, **81**, 1154; (b) W. F. Giauque, *J. Am. Chem. Soc.*, 1927, **49**, 1864.
- [3] See, *e.g.*, T. Feder, *Physics Today*, 2009, **62**, 21.
- [4] Yu. I. Spichkin, A. K. Zvezdin, S. P. Gubin, A. S. Mischenko and A. M. Tishin, *J. Phys. D: Appl. Phys.*, 2001, **34**, 1162.
- [5] See, *e.g.*, C. Kittel, *Introduction to Solid State Physics*, John Wiley & Sons, Inc., 2005.
- [6] R. D. McMichael R. D. Shull, L. J. Swartzendruber, L. H. Bennett and R. E. Watson, *J. Magn. Magn. Mater.*, 1992, **111**, 29.
- [7] P. Poddar, S. Srinath, J. Gass, B. L. V. Prasad and H. Srikanth, *J. Phys. Chem. C*, 2007, **111**, 14060.

- [8] S. Ma, W. B. Cui, D. Li, N. K. Sun, D. Y. Geng, X. Jiang and Z. D. Zhang, *Appl. Phys. Lett.*, 2008, **92**, 173113.
- [9] See, e.g., M. Evangelisti, F. Luis, L. J. de Jongh and M. Affronte, *J. Mater. Chem.*, 2006, **16**, 2534, and references therein.
- [10] M. Evangelisti, F. Luis, F. L. Mettes, N. Aliaga, G. Aromí, J. J. Alonso, G. Christou and L. J. de Jongh, *Phys. Rev. Lett.*, 2004, **93**, 117202.
- [11] (a) F. Torres, J. M. Hernández, X. Bohigas and J. Tejada, *Appl. Phys. Lett.*, 2000, **77**, 3248; (b) F. Torres, X. Bohigas, J. M. Hernández and J. Tejada, *J. Phys.: Condens. Matter*, 2003, **15**, L119.
- [12] X. X. Zhang, H. L. Wei, Z. Q. Zhang and L. Zhang, *Phys. Rev. Lett.*, 2001, **87**, 157203.
- [13] M. Affronte, A. Ghirri, S. Carretta, G. Amoretti, S. Piligkos, G. A. Timco and R. E. P. Winpenny, *Appl. Phys. Lett.*, 2004, **84**, 3468.
- [14] M. Evangelisti, A. Candini, A. Ghirri, M. Affronte, E. K. Brechin and E.J.L. McInnes, *Appl. Phys. Lett.*, 2005, **87**, 072504.
- [15] R. Shaw, R. H. Laye, L. F. Jones, D. M. Low, C. Talbot-Eeckelaers, Q. Wei, C. J. Milios, S. Teat, M. Helliwell, J. Raftery, M. Evangelisti, M. Affronte, D. Collison, E. K. Brechin and E. J. L. McInnes, *Inorg. Chem.*, 2007, **46**, 4968.
- [16] M. Manoli, R. D. L. Johnstone, S. Parsons, M. Murrie, M. Affronte, M. Evangelisti and E. K. Brechin, *Angew. Chem. Int.-Ed.*, 2007, **46**, 4456.
- [17] J. Schnack, R. Schmidt and J. Richter, *Phys. Rev. B*, 2007, **76**, 054413.
- [18] M. E. Zhitomirsky, *Phys. Rev. B*, 2003, **67**, 104421.
- [19] M. Manoli, A. Collins, S. Parsons, A. Candini, M. Evangelisti and E. K. Brechin, *J. Am. Chem. Soc.*, 2008, **130**, 11129.
- [20] M. Evangelisti, A. Candini, M. Affronte, E. Pasca, L. J. de Jongh, R. T. W. Scott and E. K. Brechin, *Phys. Rev. B*, 2009, **79**, 104414.

- [21] G. Karotsis, M. Evangelisti, S. J. Dalgarno and E. K. Brechin, *Angew. Chem. Int.-Ed.*, 2009, **48**, 9928.
- [22] (a) E. Manuel, M. Evangelisti, M. Affronte, M. Okubo, C. Train and M. Verdaguer, *Phys. Rev. B*, 2006, **73**, 172406; (b) M. Evangelisti, E. Manuel, M. Affronte, M. Okubo, C. Train and M. Verdaguer, *J. Magn. Magn. Mater.*, 2007, **316**, e569.