



Ternary alloy nanocatalysts for hydrogen evolution reaction

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Abstract. Cu–Fe–Ni ternary alloys (size ~55–80 nm) with varying compositions viz. CuFeNi (A1), CuFe₂Ni (A2) and CuFeNi₂ (A3) were successfully synthesized using microemulsion. It is to be noted that synthesis of nanocrystalline ternary alloys with precise composition is a big challenge which can be overcome by choosing an appropriate microemulsion system. High electrocatalytic activity towards HER in alkaline medium was achieved by the formation of alloys of metals with low and high binding energies. A high value of current density (228 mA cm²) at an overpotential of 545 mV was obtained for CuFeNi (A1), which is significantly high as compared to the previously reported Ni₅₉Cu₄₁ alloy catalyst.

Keywords. Nanostructures; chemical synthesis; electrochemical measurements.

1. Introduction

Generation of hydrogen through water splitting is an important area of research. This reaction is not thermodynamically feasible and thus requires a catalyst and external stimuli like light or electricity. Splitting of water using electricity makes this reaction feasible, but requires a catalyst to overcome the kinetic barriers such as low faradaic efficiency and high overpotential associated with it. Thus, a need for efficient electrocatalyst that is an alternative to expensive Pt or Pt-based alloys is necessary. In this paper, we have successfully synthesized nanosized Cu–Fe–Ni ternary alloy with varying stoichiometry (1:1:1, 1:2:1 and 1:1:2) using microemulsion method. One of the key areas in which Cu–Fe–Ni alloy can be applied is the production of hydrogen by splitting water, which has been the prime focus of our group. The ternary system was chosen based on volcano plot. Ni is the most appropriate choice of low-cost catalyst for HER among non-noble metals in alkaline medium, due to low binding energy which can be observed from the volcano plot [1]. Although Ni use in acidic media is restricted due to corrosion, it can be used in the alkaline medium as the process of corrosion is insignificant. It has been well established that water splitting in the alkaline medium for the production of pure hydrogen has an advantage due to being environmentally benign, having low cost and being unaffected by poisoning [1,2], which is one of the greatest challenges associated with proton exchange membranes in water electrolysis [3–6]. Taking

these points into consideration, we have focussed on synthesizing nanostructured ternary alloys; based on Cu–Fe–Ni alloys which includes Ni as one of the constituent metal and alloyed with two metals, one with weak M–H binding energy (Fe) and the other having strong M–H binding energy (Cu) as observed from the volcano curve [7,8]. The aim is to investigate the synergistic effect of these metals on HER.

Synthesis of ternary alloys with desired composition using microemulsions is a big challenge that arises due to number of components taking part in the reaction. Microemulsion method is the most versatile method as there are several parameters associated with the formation of microemulsion viz., solvent, surfactant, co-surfactant and Wo (water to surfactant ratio) that can be optimized to yield particles with desired size and shape. This method involves coalescence of reverse micelles, i.e., exchange of materials between the micelles and de-coalescence of the reverse micelles containing the product. The detailed mechanism is discussed elsewhere [9]. Although there are several techniques that have been used earlier to fabricate ternary nanostructured alloy particles, such as mechanical alloying [10], electrodeposition [11], melt spinning [12], polyol processes [13] and decomposition of organometallic precursors [14], there are only few reports on the synthesis of Cu–Fe–Ni alloy nanoparticles. For instance, Giz *et al* [15] have fabricated Ni–Cu–Fe alloy using the electrodeposition method. Mechanical alloying method using high-energy ball mill under argon atmosphere was used to synthesize nanostructured copper–iron–nickel alloy with (Cu₆₀Fe₄₀)₇₀Ni₃₀ composition. In addition to the ternary alloy, we obtained BCC-Fe(Ni) solid solutions [16]. Cu–Fe–Ni alloy (20–200 nm) was synthesized by chemical reduction of Cu²⁺, Fe³⁺ and Ni²⁺ ions with sodium borohydride and

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subsequently treating the precursor at 300–900°C in an H₂ atmosphere [17].

To the best of our knowledge, there are no reports on the preparation of Cu–Fe–Ni alloy nanoparticles using the microemulsion method. This paper shows the advantage of microemulsion method of synthesis to yield alloy nanoparticles of precise composition which can be varied.

2. Experimental

Microemulsion method was employed for the synthesis of Cu–Fe–Ni alloys with varying compositions. We have used cationic surfactant CTAB (cetyltrimethyl ammonium bromide (Spectrochem, AR, 99%)) along with 1-butanol (Qualigens, 99.5%) as the co-surfactant and iso-octane (Spectrochem, 99%) as the oil phase for the formation of microemulsions. Five microemulsions containing 0.1 M Cu(NO₃)₂·3H₂O, 0.1 M Ni(NO₃)₂·6H₂O, 0.1 M Fe(NO₃)₂·9H₂O, 1 M N₂H₄·H₂O and 0.1 M NaOH were formed. The weight fraction of CTAB in the microemulsion was 16.76% with 13.9% 1-butanol, 59.29% iso-octane and 10.05% of aqueous solution. The microemulsions were mixed and stirred for 24 h. The brown-coloured product was separated by centrifugation and washed with a mixture of chloroform and methanol (1:1). The obtained product was dried in air and heated at 700°C for 5 h in H₂ atmosphere. Three different compositions were loaded for Cu–Fe–Ni in the ratio of 1:1:1, 1:2:1 and 1:1:2, referred as A1, A2 and A3, respectively.

To obtain structural information and phase purity of the synthesized nanoalloys, powder X-ray diffraction was carried out on a Bruker D8 Advance diffractometer with a Ni-filtered CuK α radiation ($\lambda = 1.5418 \text{ \AA}$). The data were recorded with 2θ values from 35 to 100° at a scanning rate of 0.02° s⁻¹. Refinement of lattice parameters was carried out with Powder Cell software [18].

Transmission electron microscopy (TEM) and bright field scanning transmission electron microscopy (BF-STEM) with a Technai G² 20 electron microscope operating at 200 kV were used. STEM-EDX measurements were carried out to analyse the distribution of Fe, Co and Ni atoms in the particle interior. The samples were prepared by dispersing the powder sample in absolute ethanol by ultrasonic treatment, adding a drop of this dispersion on the copper grid having porous carbon film support, and then drying in air. EDX studies were also carried out on a Hitachi TM 3000 SEM. Powder samples were mounted on a carbon tape attached on a circular metallic sample holder.

Cyclic voltammetric studies were carried out to study the hydrogen evolution reaction at room temperature with an electrochemical workstation (Autolab PGSTAT 302N). The cell consists of three electrode system Ag/AgCl/KCl (3 M KCl) as reference electrode (RE), Pt rod as a counter electrode (CE) and a glassy carbon electrode (GCE; Metrohm, 2 mm diameter, 0.031 cm²) modified with CuFeNi alloys as working electrode. The reported potentials in this paper

were vs. the reversible hydrogen electrode which were calculated using the following equation [19]:

$$\begin{aligned} E(\text{RHE}) &= E(\text{Ag/AgCl}(3 \text{ M KCl})) \\ &\quad + E^0(\text{Ag/AgCl}(3 \text{ M KCl})) + 0.059 \text{ pH} \\ &= E(\text{Ag/AgCl}(3 \text{ M KCl})) + 0.210 \text{ V} \\ &\quad + 0.059 \times 12.7 \text{ (pH at 0.5 KOH is } \sim 12.7) \\ &= E(\text{Ag/AgCl}(3 \text{ M KCl})) + 0.959 \text{ V.} \end{aligned}$$

The GCE before modification was polished with alumina powder (0.05 μm), ultrasonicated in distilled water followed by ethanol, and then dried in an oven. Analytical grade reagents were used for the preparation of solution from double distilled water. Cyclic voltammetry was carried out with blank GC electrode in 0.5 M KOH solution at a potential range from -1.5 to 0 V until a stable voltammogram was obtained. The solution was degassed by purging N₂ for half an hour. All experiments were conducted at 25°C. Modified electrodes were prepared from the dispersion of 2 mg of the alloy nanoparticles with 15 μl of isopropanol and 10 μl of Nafion in ethanol solution (5 wt% in lower aliphatic alcohols and water). Five microlitres of this dispersion was pasted on GC and the solvent was evaporated in an oven for half an hour. To remove unbound suspension that may interfere with the voltammograms during the measurement, the Cu–Fe–Ni/GC electrodes were washed again with distilled water. The cyclic voltammetry was recorded in the potential range from -1.5 to 0 V at a scan rate of 25 mV s⁻¹ (figure 4). The current density was normalized with respect to geometric electrode surface area [1].

3. Results and discussion

The precursor of the ternary alloy (Cu–Fe–Ni) obtained at room temperature was found to be amorphous by XRD. After annealing at 700°C for 5 h under hydrogen atmosphere, three sharp peaks were observed in the diffraction patterns (figure 1). Diffraction peaks were indexed to (111), (200) and (220) planes with respect to the fcc cubic lattice of Cu–Fe–Ni system. The lattice parameters were calculated as 3.573(4) \AA , 3.570(7) \AA and 3.559(3) \AA for A1, A2 and A3, respectively.

TEM studies were carried out to further investigate the shape and size of the formed alloy. Figure 2a–c showed the formation of nanoparticles with spherical shape having size of 70–80, 65–75 and 55–60 nm for A1, A2 and A3, respectively. EDX studies confirmed that the observed compositions were in accordance to the loaded ones (table 1). Electron diffraction studies indicated polycrystalline nature of the sample (A3) (figure 2d) and was found to match well with fcc structure of the alloy. Homogeneous distribution of Cu, Fe and Ni in the alloy was observed using the STEM mapping (figure 3a–d).

We have carried out HER studies in basic medium, in which the reaction was as follows:



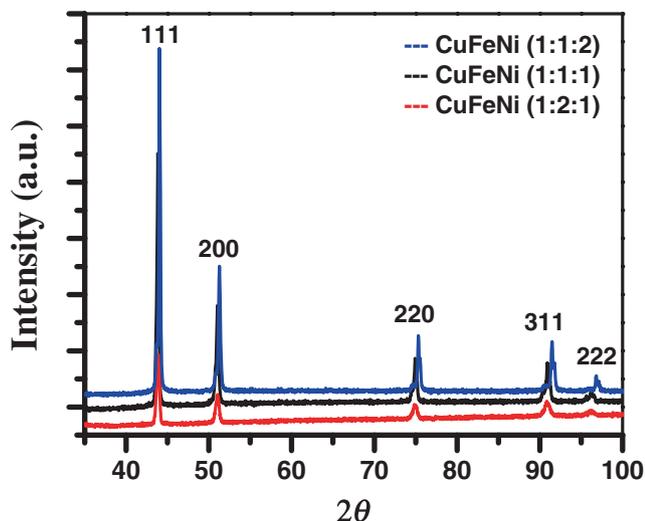


Figure 1. Powder X-ray diffraction patterns of (a) A1, (b) A2 and (c) A3 alloy nanoparticles.

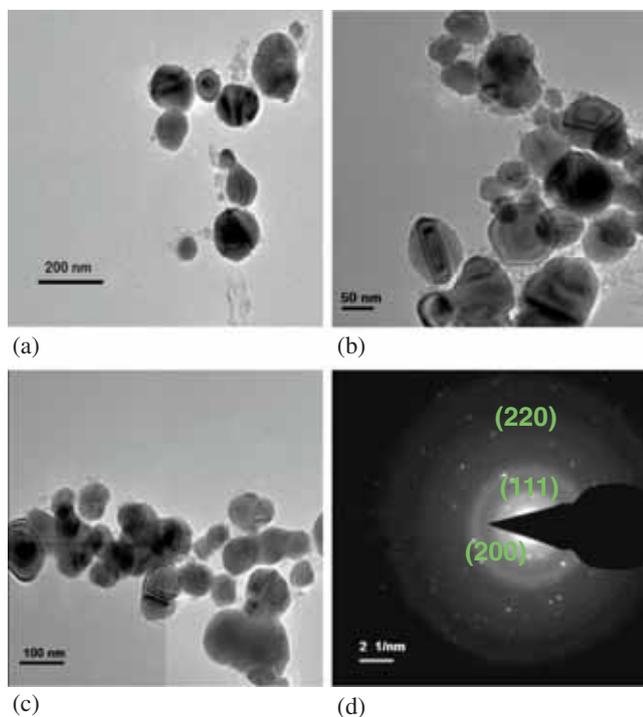


Figure 2. TEM micrographs of (a) A1, (b) A2, (c) A3 and (d) ED pattern of polycrystalline A3 alloy nanoparticles.

where M is the electrocatalyst. The overall reaction is



The current density for A1/GC, A2/GC and A3/GC electrodes was calculated by dividing observed current with geometric area of the electrode and found to be 228, 164 and 152 mA cm⁻² with an onset potential range of -0.25 to -0.20 V (vs. RHE) and also followed the same order as

Table 1. The EDS analysis of Cu–Fe–Ni alloy nanoparticles (error: 3–5%).

Sample	Loaded composition	Observed composition
A1	Cu ₃₃ Fe ₃₃ Ni ₃₃	Cu ₃₄ Fe ₃₀ Ni ₃₆
A2	Cu ₂₅ Fe ₅₀ Ni ₂₅	Cu ₂₂ Fe ₄₄ Ni ₃₄
A3	Cu ₂₅ Fe ₂₅ Ni ₅₀	Cu ₂₆ Fe ₂₆ Ni ₄₈

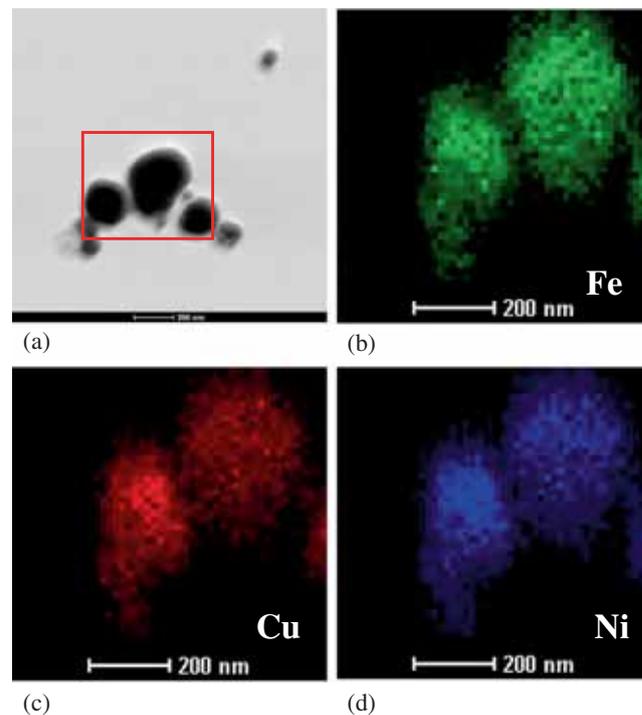


Figure 3. (a) STEM micrograph and (b–d) elemental mapping images of A1 nanoalloy.

A1/GC > A2/GC > A3/GC (figure 4). The normalized current density value obtained for Cu–Fe–Ni alloys is higher than the recent report for Ni₅₀Cu₄₁ (~170 mA cm⁻²) [1], Cu/CuNi (55 mA cm⁻²) [20], Fe–Co (~40 mA cm⁻²) [21] or Cu–Co (~15 mA cm⁻²) in 0.5 M KOH [22] and Ni–Fe (~120 mA cm⁻²) in 0.5 mol l⁻¹ NaCl [23].

For the reaction



the value of $E_e^0 = -0.83\text{V}$ (standard value). Since HER was carried out in 0.5 M KOH, using the Nernst equation, the theoretical potential at which the reaction is expected to occur was calculated to be -0.812 V. The onset potential for A1, A2 and A3 was observed at -0.24, -0.20 and -0.21 V, respectively. The overpotential was calculated using the following equation [24].

$$\eta = E - E_e^0 + \frac{RT}{nF} \ln c_{\text{OH}^-}^2.$$

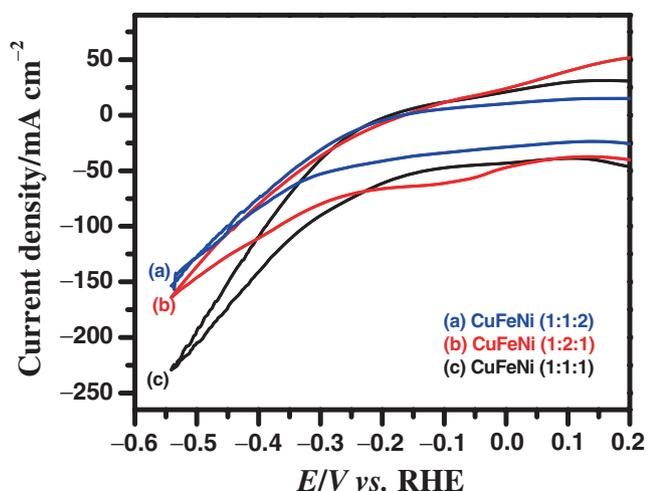


Figure 4. Cyclic voltammogram of hydrogen evolution reaction using A1/GC, A2/GC and A3/GC electrodes in 0.5 M KOH solution at a scan rate of 25 mV s⁻¹.

Thus, the overpotential was calculated to be 545, 565 and 573 mV for A1, A2 and A3, respectively, which is lower than the recent report for Ni₅₉Cu₄₁ catalyst (-1.6 V vs. SCE) in 6 M KOH [1].

4. Conclusions

Spherical nanoparticles of ternary alloys of Cu-Fe-Ni with varying composition were successfully synthesized using microemulsion method. The size of alloy nanoparticles varied from 60 to 80 nm. Synergistic effect of the metals (with low and high binding energies) by the formation of alloys resulted in the high electrocatalytic activity of the alloy towards HER in alkaline medium. Among the three compositions of Cu-Fe-Ni alloys, a notable high current density value of 228 mA cm⁻² with an onset potential of -0.267 V and an overpotential of 545 mV was observed for CuFeNi (1:1:1).

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References

- [1] Ahn S H, Park H Y, Choi I, Yoo S J, Hwang S J, Kim H J, Cho E, Yoon C W, Park H, Son H, Hernandez J M, Nam S W, Lim T H, Kim S K and Jang J H 2013 *Int. J. Hydrogen Energy* **38** 13493
- [2] Lu G, Evans P and Zangari G 2003 *J. Electrochem. Soc.* **150** A551
- [3] Santos D M F, Sequeira C A C, Macciò D, Saccone A and Figueiredo J L 2013 *Int. J. Hydrogen Energy* **38** 3137
- [4] Lubitz B and Tumas W 2007 *Chem. Rev.* **107** 3900
- [5] Barbaro C B P 2009 *Catalysis for Sustainable Energy Production* (Wiley-VCH: Weinheim)
- [6] Vielstich G W, Lamm A, Vielstich H W, Lamm A and Gasteiger H 2003 *Handbook of Fuel Cells: Fundamentals, Technology and Applications* (Chichester: Wiley)
- [7] Conway B E and Bockris J O 1957 *J. Chem. Phys.* **26** 532
- [8] Greeley J, Jaramillo T F, Bonde J, Chorkendorff I B and Nørskov J K 2006 *Nat. Mater.* **5** 909
- [9] Ganguli A K, Ganguly A and Vaidya S 2010 *Chem. Soc. Rev.* **39** 474
- [10] Mondal B N, Basumallick A and Chattopadhyay P P 2008 *J. Alloys Compd.* **457** 10
- [11] Wang C, Li W, Lu X, Xie S, Xiao F, Liu P and Tong Y 2012 *Int. J. Hydrogen Energy* **37** 18688
- [12] Curriotto S, Johnson E, Celegato F, Coisson M and Pryds N 2009 *J. Magn. Magn. Mater.* **321** 131
- [13] Toneguzzo P, Viau G, Acher O, Guillet F, Bruneton E, Fievet-Vincent F and Fievet F 2000 *J. Mater. Sci.* **35** 3767
- [14] Luo J, Wang L, Mott D, Njoki P N, Kariuki N, Zhong C-J and He T 2006 *J. Mater. Chem.* **16** 1665
- [15] Giz M J, Marengo M C, Ticianelli E A and Gonzalez E R 2003 *Eclat. Quim.* **28** 21
- [16] Slimi M, Azabou M, Escoda L, Suñol J J and Khitouni M 2014 *Powder Technol.* **266** 262
- [17] Stolk J and Manthiram A 1999 *Mater. Sci. Eng. B* **60** 112
- [18] Kraus W 2000 *Nolze G Version 2.4*, Berlin
- [19] Gao M-R, Cao X, Gao Q, Xu Y-F, Zheng Y-R, Jiang J and Yu S-H 2014 *ACS Nano* **8** 3970
- [20] Solmaz R, Doner A and Kardas G 2008 *Electrochem. Commun.* **10** 1909
- [21] Ahmed J, Kumar B, Mugweru A M, Trinh P, Ramanujachary K V, Lofland S E and Ganguli A K 2010 *J. Phys. Chem. C* **114** 18779
- [22] Ahmed J, Ganguly A, Saha S, Gupta G, Trinh P, Mugweru A M, Lofland S E, Ramanujachary K V and Ganguli A K 2011 *J. Phys. Chem. C* **115** 14526
- [23] Song Li-Jun and Meng H-M 2010 *Acta Phys.-Chim. Sin.* **26** 2375
- [24] Greef R, Peat R, Peter L, Pletcher D and Robinson J 1985 *Instrumental Methods in Electrochemistry* (Chichester: Ellis Horwood Ltd)