

Evolution of the electronic structure with size in II-VI semiconductor nanocrystals

Sameer Sapra and D. D. Sarma*

*Solid State and Structural Chemistry Unit,
Indian Institute of Science, Bangalore-560012, India*

Abstract

In order to provide a quantitatively accurate description of the band gap variation with sizes in various II-VI semiconductor nanocrystals, we make use of the recently reported tight-binding parametrization of the corresponding bulk systems. Using the same tight-binding scheme and parameters, we calculate the electronic structure of II-VI nanocrystals in real space with sizes ranging between 5 and 80 Å in diameter. A comparison with available experimental results from the literature shows an excellent agreement over the entire range of sizes.

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INTRODUCTION

It is now possible to grow a large variety of semiconductor nanocrystals and also control their sizes to obtain mono-dispersed particles. [1, 2] A large number of II-VI [3, 4, 5] and III-V [6, 7, 8] semiconductor nanocrystals have been prepared over the past two decades. These quantum dots are good candidates for electronic and optical devices [9, 10, 11, 12, 13] due to their reduced dimensions, enabling one to reduce the size of electronic circuitry. Also, due to the increased oscillator strengths in these nanocrystals as a result of quantum confinement, [14] these are expected to have higher quantum efficiencies in applications such as light emission. This is a direct consequence of a greater overlap between the electron and the hole wavefunctions upon size reduction. Moreover, one can tune these properties to suit a specific application by merely changing the size of the nanocrystals. For example, the band gap of CdSe can be varied from 1.9 eV to 2.7 eV by changing the size of the particle from 5.5 nm to 2.3 nm. [15] Alongwith the band gap of the particle, the photoluminescence can also be varied through the red to the blue region of the visible spectrum. [15] This quantum size effect can be explained qualitatively by considering a particle-in-a-box like situation where the energy separation between the levels increases as the dimensions of the box are reduced. Thus, one observes an increase in the band gap of the semiconductor with a decrease in the particle size

On a more quantitative footing, various different theoretical approaches have been employed to account for the variation in the electronic structure of nanocrystallites as a function of its size. The first explanation for the size-dependence of electronic properties in nanocrystals was given by Efros and Efros. [16] It is based on the effective masses of the electron (m_e^*) and the hole (m_h^*). Known as the Effective Mass Approximation (EMA), it is solved by taking various choices for the electron and hole wavefunctions and solving the effective mass equation variationally. In most EMA calculations, the confining potentials for the electron and the hole have been assumed infinite. [14, 16, 17, 18, 19, 20] Therefore, the electron and the hole wavefunctions vanish at and beyond the surface of the nanocrystal, without the possibility of any tunnelling. In the strong confinement regime, where R , the nanocrystal radius, is much smaller than a_B , the Bohr exciton radius, Brus proposed [17] the following expression for the bandgap of the finite sized system

$$E(R) = E_g + \frac{\hbar^2}{2} \left(\frac{1}{m_e^*} + \frac{1}{m_h^*} \right) \frac{\pi^2}{R^2} - 1.786 \frac{e^2}{\epsilon R} - 0.248 E_{Ry}^* \quad (1)$$

where, E_g is the bulk band gap. The second term is the kinetic energy term containing the effective masses, m_e^* and m_h^* , of the electron and the hole, respectively. The third term arises due to the Coulomb attraction between the electron and the hole, and the fourth term due to the spatial correlation between the electron and the hole which is generally small compared to the other two terms.

EMA calculations have also been reported where a finite confining potential was used to account for the passivating agents that coat the surface of the nanocrystals in order to arrest their growth. Finite potential calculations are shown to improve the description for CdS nanocrystals to a large extent. [21] Another improvement to the single band EMA is the inclusion of multiple bands for describing the hole effective mass. This is prompted by the fact that the top of the valence band for II-VI semiconductors is comprised of triply degenerate bands at the Γ point and thus is better defined using a multi-band theory. To account for this degeneracy, Einevoll [22] and Nair *et al.* [23] have used the effective bond-orbital model for the hole wavefunction, while the electron is described by a single-band EMA. Finite barrier heights and the electron-hole Coulomb attraction are included in the calculation and exciton energies are obtained variationally in an iterative Hartree scheme. The multi-band and finite potential EMA methods explain the experimental results reasonably well, but lack the predictive capabilities desirable of a theoretical model, since the finite potentials need to be adjusted to match the experimental results in each specific case. Pseudopotential calculations have also been carried out to study the variation of electronic structure with the nanocrystal size. [24, 25, 26] Recently, the semi-empirical pseudopotential method has been employed to calculate the electronic structure of Si, CdSe [25] and InP [26] nanocrystals. The atomic pseudopotentials are extracted from first principles LDA calculations on bulk solids. Thus, the wave functions are LDA-like while the band structures, effective masses, and deformation potentials are made to match experimental results. This method provides a reasonable description of the electronic structure of the nanocrystals. However, major computational efforts and difficulties do not allow one to calculate the properties of large sized nanocrystals.

The tight-binding (TB) scheme has been employed by a number of researchers over the past decade. [27, 28, 29, 30, 31, 32, 33, 34] This method enjoys several advantages over

the other methods discussed above, explaining its popularity. Compared to EMA, both pseudopotential method and the tight-binding approach provide a substantial improvement in the accuracy of the results. The tight-binding method has the further advantage of being significantly less demanding in terms of computational efforts, besides providing a simple physical picture in terms of the atomic orbitals and hopping interactions defined over a predetermined range. A detailed analysis of the first-principle electronic structure calculations can lead to a judicious tight-binding scheme that is minimal in terms of the dimension of the Hamiltonian matrix and yet is highly accurate due to the use of a physical and realistic basis. [35] The earliest such TB parametrization was provided by Vogl *et al.* [36] who used a TB model with the sp^3s^* orbital basis in order to describe the electronic structure of bulk semiconductors. The s^* orbital was employed in an *ad hoc* manner in addition to the sp^3 orbital basis in order to improve the TB-fit to the *ab-initio* band dispersions. Subsequently, this TB model was used by Lippens and Lannoo [27] to calculate the variations in the band gap for the corresponding semiconducting nanocrystals as a function of the size. Though their results are in better agreement compared to the infinite potential EMA, the sp^3s^* TB model tends to underestimate the band gap. The main problem with the sp^3s^* model appears to be a failure to reproduce even the lowest lying conduction band within that scheme. [36] Improvements in the nearest neighbor sp^3s^* model have been carried out by including the spin-orbit coupling and the electron-hole interaction. [28, 29] However, to account for the conduction bands, the inclusion of d orbitals becomes necessary. [30, 34] This has been shown in the case of InP [30] nanocrystals, a III-V semiconductor, that TB model with the sp^3d^5 orbital basis for the anion and the sp^3 basis for the cation with next nearest neighbor interactions, for both the anion and the cation, gives excellent agreement with the experimental data. In a recent work, we have shown that the sp^3d^5 orbital basis for both the cation and the anion and the inclusion of the next nearest neighbor interactions for the anions provide a very good description of the electronic structure of bulk II-VI semiconductors. [35] This model is shown to describe accurately the band gap and the band dispersions for both the valence and the conduction bands over the energy range of interest. Therefore, this improved model and the parametrization should provide a good starting point for calculating the electronic properties of corresponding nanocrystals, provided the model and parameters are transferable from the bulk to the cluster limit. *Ab initio* calculations for a CdS cluster of about 16 Å diameter [37] as well as results of Ref. [24] suggest that the

present scheme is of sufficient accuracy down to about 16 Å, though the applicability of this approach may be limited for still smaller sized clusters. In order to explore the possibility of utilizing it effectively, we have used this model for calculating the band gap variation over a wide range of sizes for A^{II}B^{VI} semiconductor nanocrystals, with A = Cd or Zn and B = S, Se or Te, comparing the calculated results with the experimental data from the literature. The present results show a good agreement with experimental results, where ever available.

THEORETICAL PROCEDURE

The appropriate minimal TB model for the bulk electronic structure of group II-VI semiconductors was developed in Ref. [35] by analyzing the atomic wave-function characters of the various bands. This established sp^3d^5 basis with the cation-anion and anion-anion interactions as the suitable model. The tight-binding electronic parameters, namely the orbital energies and the hopping strengths, were determined by fitting the *ab-initio* band dispersions to the band dispersions obtained from the tight-binding Hamiltonian, given by

$$H = \sum_{il_1\sigma} \epsilon_{l_1} a_{il_1\sigma}^\dagger a_{il_1\sigma} + \sum_{ij} \sum_{l_1 l_2 \sigma} (t_{ij}^{l_1 l_2} a_{il_1\sigma}^\dagger a_{jl_2\sigma} + h.c.) \quad (2)$$

where, the electron with spin σ is able to hop from the orbitals labelled l_1 with onsite energies equal to ϵ_{l_1} in the i^{th} unit cell to those labelled l_2 in the j^{th} unit cell, with a hopping strength $t_{ij}^{l_1 l_2}$; the summations l_1 and l_2 running over all the orbitals considered on the atoms in a unit cell, and i and j over all the unit cells in the solid. We use exactly the same model with the parameter strengths given in Ref. [35] to calculate the electronic structure of corresponding nanocrystals as a function of the size.

We build the cluster shell by shell, starting from a central atom. For the tetrahedrally coordinated compounds in the zinc-blende structure, the central atom, say the cation, is surrounded by a shell of four anions. In turn each of these anions is coordinated by four cations, one of them being the central cation. The other three cations form a part of the next shell. The clusters are generated in this manner by successive addition of shells. Assuming a spherical shape of the cluster, the diameter, d , is given by

$$d = a \left[\frac{3N}{4\pi} \right]^{\frac{1}{3}} \quad (3)$$

where, a is the lattice constant and N the number of atoms present in the nanocrystal. Table I lists the number of atoms present upto a given shell and the diameter of the nanocrystal for various $A^{II}B^{VI}$ compounds. The Hamiltonian matrix for any given sized cluster is obtained from Eq. 2 with the same atomic orbital basis and electronic parameter strengths as given in Ref. [35] and is diagonalized to obtain the eigenvalue spectra for the nanocrystal. Direct diagonalization methods are practical only for cluster sizes containing less than ~ 1500 atoms. For larger clusters, we use the Lanczos iterative method. [38]

The Lanczos algorithm uses a starting basis function $|\phi_0\rangle$ which can be a linear combination of the atomic orbitals, φ_i 's, *i.e.*

$$|\phi_0\rangle = \sum_i c_i |\varphi_i\rangle$$

Once the starting basis function has been generated, a new basis function $|\phi_1\rangle$ is generated by applying the Hamiltonian and then making the resulting function orthogonal to $|\phi_0\rangle$.

$$|\phi_1\rangle = H|\phi_0\rangle - \frac{\langle\phi_0|H|\phi_0\rangle}{\langle\phi_0|\phi_0\rangle}|\phi_0\rangle$$

Then onwards the subsequent basis functions can be generated by using the recursion formula

$$|\phi_{n+1}\rangle = H|\phi_n\rangle - a_n|\phi_n\rangle - b_n^2|\phi_{n-1}\rangle \quad n = 0, 1, 2, \dots$$

where, $a_n = \frac{\langle\phi_n|H|\phi_n\rangle}{\langle\phi_n|\phi_n\rangle}$, $b_n^2 = \frac{\langle\phi_n|\phi_n\rangle}{\langle\phi_{n-1}|\phi_{n-1}\rangle}$

with $b_0 = 0$ and $|\phi_{-1}\rangle = 0$. By construction, each basis function is orthogonal to the previously generated basis functions. Here the a_n 's are the diagonal elements, while b_n 's are the off-diagonal terms of the Hamiltonian matrix. Diagonalization of this tridiagonal matrix is less time consuming and gives the eigenvalue spectrum for the clusters. We choose the starting seed vector to be a particular orbital of an atom. The eigen spectrum thus obtained is composed of only those orbitals that couple with the seed vector. Thus, taking each orbital of every atom in the cluster we obtain the entire density of states. Due to the underlying symmetry in the nanocrystal we need not perform calculations for all atomic orbitals as seed vectors, but only those with distinct symmetries.

The band gap for a particular sized nanocrystal is then calculated by subtracting the energy of the top of the valence band (TVB) from that of the bottom of the conduction

band (BCB). However, the determinations of the TVB and the BCB become ambiguous due to the presence of dangling bonds at the surface of the nanocrystals. These non-bonded states lie in the band gap region of the nanocrystals. These surface states need to be either selectively disposed off [27] or passivated [31, 32] in order to remove the midgap states. Once the surface states are passivated, the band gap can be easily determined. In the present work, we have passivated the surfaces of the nanocrystals in order to remove the mid-gap states from the calculations.

RESULTS AND DISCUSSION

The various steps involved in the calculations for the variation of the band gap with size are quite similar for the different $A^{II}B^{VI}$ compounds studied here. We therefore use the case of CdS as an example to illustrate all the steps and various considerations, prior to presenting comprehensive results for all the systems together at the end.

Fig. 1a shows the first-principle results for the density of states (DOS) for CdS bulk obtained from the linearized muffin-tin orbital (LMTO) method with the atomic sphere approximation (ASA). It should be noted that the parameters appearing in the TB Hamiltonian (Eq. 1) were determined [35] by a least-squared-error approach in order to obtain dispersions at high symmetry points and a few other k -points along the symmetry directions in the Brillouin zone. Since we are eventually interested in the density of states which involves an integration over the entire momentum space, we have explicitly verified in each case that the DOS calculated within the TB approach is very similar to the one obtained from the LMTO-ASA method. We illustrate this point with the help of DOS calculated within the TB model for CdS with parameter strengths from Ref. [35]; this TB DOS is shown in Fig. 1b with the same energy scale as in Fig. 1a. We notice an excellent agreement of the TB DOS with the LMTO DOS over the entire range of the energy considered.

While the electronic structure of small sized nanocrystals is known to be pronouncedly dependent on the size, larger sized nanocrystals are expected to resemble the bulk in terms of their electronic structures; evidently in the limit of the large size, the electronic structure of the nanocrystal must smoothly evolve into that of the bulk. It is known that the quantum confinement effect is generally small for a nanocrystal with typical size larger than the excitonic radius. The excitonic diameter of CdS is about 58 Å. [39] We consider a CdS

cluster of 76 Å containing 9527 atoms that is considerably larger than the excitonic diameter. In Fig. 1c we show the DOS for this large CdS cluster. The DOS of the nanocrystal indeed resembles the bulk DOS closely, as is evident in Fig. 1, apart from the discrete nature of the DOS arising from the finite size of the nanocrystallite system.

As discussed in the previous section, the dangling orbitals on the surface atoms appear within the band gap region, complicating the identification of the band gap. Fig. 2a shows the normally obtained DOS for a 46 Å CdS nanocrystal; the corresponding inset shows an expanded view of the band gap region. As one can clearly see in the expanded view, there are many states spread out over an energy range appearing between the valence band and the conduction band due to the aforementioned dangling bonds within the band gap region. As already discussed, different authors approached the problem of dangling bonds or its removal from the DOS in different ways. For example, Lippens and Lannoo [27] got rid of the dangling bonds by removing the unconnected orbitals on the surface atoms in order to obtain the band gap free of the mid-gap states. In spirit, this approach is similar to the infinite potential barrier on the surface of the nanocrystal assumed in the infinite potential EMA. Akin to the finite potential EMA, we choose to passivate the surface with a layer of atoms, whose electronic parameters are so chosen that the hopping interactions between the surface atoms and the passivating atoms are stronger compared to those in the bulk of the nanocrystal. Specifically, we choose only the s orbital basis on the passivating atoms with the tight-binding hopping parameters about 2-3 times larger than that of A-B interactions. The corresponding DOS of the passivated nanocrystals of CdS is shown in Fig. 2b. In the main frame of the figures, the unpassivated case in the upper panel and the passivated case in the lower panel appear almost identical, suggesting that the intrinsic electronic structure of the nanocrystals remain largely unaffected by the passivation. In order to illustrate the effect of passivation on the midgap states, we show an expanded view of the band gap region between the TVB and the BCB in the inset to Fig. 2b. This inset shows that the surface passivation is indeed effective in removing the midgap states, present in the inset to Fig. 2a, illustrating the unpassivated case.

Most often, the total band gap variation as a function of the size of the nanocrystal is reported in the literature. [40] This is primarily motivated by the fact that this quantity, ΔE_g , is easily determined by experimental UV-visible absorption spectroscopy, that is a routine characterization tool. However, it is to be noted that the total change in the band

gap of any material is simultaneously contributed by shifts of the valence and the conduction band edges away from each other. In general, the shift of the top of the valence band is not the same as that of the bottom of the conduction band. Moreover, there are recent studies, though few in number [41, 42, 43, 44] that report the individual shifts in TVB and BCB as a function of the size employing various forms of high energy spectroscopies, such as the photoemission and the x-ray absorption spectroscopies. Thus, it is desirable to compute these shifts of the individual band edges with the size of the nanocrystallite. The variation of TVB (circles) and the BCB (squares) with respect to the bulk values are calculated for different sized passivated nanocrystals and shown in Fig. 3. As expected, the shifts of the band edges decrease smoothly to zero for large sized nanocrystals in every case. We find that the shift in the BCB is in general much larger compared to the shift in the TVB for any given size of the nanocrystal; this indicates that the shifts in the total band gap as a function of the nanocrystal size are always dominated by the shifts of the conduction band edge in these systems. A larger shift for the BCB is indeed expected in view of the fact that the band edge shifts are related inversely to the corresponding effective masses (see Eq. 1) and the effective mass of the electron is always much smaller than that of the hole in these II-VI semiconductors. For example, m_e^* and m_h^* in CdS are 0.18 and 0.53, respectively.

In the spirit of EMA, one can attempt to describe the shifts in the conduction and valence band edges, as

$$\Delta E_i^{edge} = \frac{a_i}{d^{b_i}} \quad (4)$$

where ΔE_i^{edge} is the variation in the band edge with diameter d ; $i = h$ for TVB and $i = e$ for BCB. Comparing with the EMA (Eq. 1), one expects the fitting parameter a_i to be inversely proportional to the electron (for BCB, $i = e$) or hole (for TVB, $i = h$) effective mass and b_i to equal 2. We have fitted the shifts in BCB and TVB as a function of d with Eq. 4 by varying the parameters a_i and b_i within a least-squared-error approach; the resulting best fits are shown in Fig. 3 by the solid lines overlapping the calculated data points. We find that the fits are reasonable, though not very good, in most cases. More importantly, these fits suggest a gross deviation from the EMA predictions; for example, the variations in TVB and BCB shown in Fig. 3 are far from the EMA-like d^{-2} dependence and instead the best exponent for d is in the range of 1.13-1.27 for TVB and 0.65-1.05 for BCB, as shown in Table II.

Fig. 4 shows the variation of the shift in the band gap (ΔE_g) for the A^{II}B^{VI} semiconductor

nanocrystals with A=Zn, Cd and B=S, Se and Te as a function of the nanocrystal size. ΔE_g is calculated in the present model after subtracting the Coulomb term (third term of Eq. 1) from the calculated difference between the TVB and the BCB to account for the excitonic binding energy, since the experimental data obtained from the UV-absorption include the contribution from the excitonic binding energy. The solid line passing through the calculated data points (small solid circles) is the best fit to the calculations. The best fit is obtained by using simple exponential functions relating ΔE_g to the diameter of the nanocrystallites as

$$\Delta E_g = a_1 e^{-d/b_1} + a_2 e^{-d/b_2} \quad (5)$$

While this expression is entirely phenomenological, it has the correct limiting behavior at large d . The advantage of such a best fit is that the ΔE_g for any given system can be readily calculated for any size of the nanocrystallites with the knowledge of the parameter values a_1 , b_1 , a_2 and b_2 , which are tabulated in Table III for all the systems investigated here. For comparison, we also show in the same panels the results obtained from the sp^3s^* nearest neighbor TB model (dashed line) [45] and the results from the EMA equation (dotted line) [46]. Experimental results available in the literature are also plotted as scattered points with different symbols for comparison with the calculated results. [4, 5, 47, 48, 49, 50, 51, 52, 53, 54, 55, 56, 57, 58, 59, 60, 61, 62] There is a plethora of experimental data for ZnS, CdS, CdSe and CdTe and we see that the present approach provides a better description of the experimental data in all these cases. The case of ZnSe, where the experimental results are limited, also exhibits good agreement between the experiment and the theory. In the case of ZnTe, the present as well as the earlier calculations give almost similar descriptions; unfortunately, the experimental data are limited and there are large uncertainties in the data, so it is difficult to compare the experimental results with our calculations. For most of the cases, the sp^3d^5 model with the next nearest neighbor interactions is in better agreement with the experiments compared to the nearest-neighbor-only sp^3s^* model. This is due to the fact that the sole s^* orbital does not account well for the unoccupied states. These can only be described by the inclusion of the empty anionic d orbitals and the anion-anion interactions which are of significance in the description of the bulk electronic structure. [35]

CONCLUSIONS

We have calculated the electronic structure as a function of the nanocrystallite size for $A^{II}B^{VI}$ semiconductors with $A=Zn$ and Cd and $B=S, Se$ and Te , using the tight binding method with the sp^3d^5 orbital basis set including the A-B and B-B interactions. It is shown that the shift in the top of the valence band as well as that in the bottom of the conduction band are different from the predictions based on the effective mass approximation, not only in quantitative terms, but also qualitatively. The calculated variations in the band gaps over a wide range of sizes are compared with all experimental data published so far in the literature. This comparison shows a very good agreement in every case, suggesting the reliability and the predictive ability of the present approach.

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* Also at Jawaharlal Nehru Centre for Advanced Scientific Research, Bangalore and Centre for Condensed Matter Theory, IISc.; Electronic address: sarma@sscu.iisc.ernet.in

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TABLE I: The unit cell edge length for zinc blende phase (a), number of shells (n_s), number of atoms (N) in n_s , and the average diameter, d , for various A^{II}B^{VI} semiconductors studied.

		ZnS	ZnSe	ZnTe	CdS	CdSe	CdTe
a (Å)		5.41	5.67	6.10	5.82	6.05	6.48
n_s	N	d (Å)					
3	17	8.63	9.04	9.73	9.28	9.65	10.34
4	41	11.57	12.13	13.05	12.45	12.94	13.86
5	83	14.64	15.34	16.51	15.75	16.37	17.53
6	147	17.71	18.56	19.97	19.05	19.81	21.22
7	239	20.83	21.83	23.48	22.41	23.29	24.95
8	363	23.94	25.09	26.99	25.76	26.77	28.68
9	525	27.07	28.38	30.53	29.13	30.28	32.43
10	729	30.20	31.66	34.06	32.49	33.78	36.18
11	981	33.35	34.95	37.60	35.87	37.29	39.94
12	1285	36.49	38.24	41.14	39.25	40.80	43.70
13	1647	39.63	41.54	44.69	42.64	44.32	47.47
14	2071	42.78	44.83	48.23	46.02	47.84	51.24
15	2563	45.93	48.14	51.79	49.41	51.36	55.01
16	3127	49.08	51.44	55.34	52.80	54.88	58.78
17	3769	52.23	54.74	58.89	56.19	58.41	62.56
18	4493	55.38	58.04	62.44	59.58	61.93	66.33
19	5305	58.53	61.35	66.00	62.97	65.46	70.11
20	6209	61.68	64.65	69.55	66.36	68.98	73.88
21	7211	64.84	67.95	73.11	69.75	72.51	77.66
22	8315	67.99	71.26	76.66	73.14	76.03	81.44
23	9527	71.15	74.57	80.22	76.54	79.56	85.22

TABLE II: The values of the parameters a and b and used in Eq. 4 for all the A^{II}B^{VI} semiconductors studied.

	ZnS	ZnSe	ZnTe	CdS	CdSe	CdTe
a_e	15.72	13.71	8.23	24.47	24.43	16.38
b_e	1.01	0.91	0.65	1.05	1.05	0.92
a_h	-14.93	-13.31	-20.47	-7.76	-19.49	-19.03
b_h	1.18	1.15	1.13	1.27	1.19	1.13

TABLE III: The values of the parameters a and b and used in Eq. 5 for all the A^{II}B^{VI} semiconductors studied.

	ZnS	ZnSe	ZnTe	CdS	CdSe	CdTe
a_1	7.44	2.65	5.10	2.83	7.62	5.77
b_1	2.35	7.61	10.35	8.22	6.63	8.45
a_2	3.04	1.90	1.05	1.96	2.07	1.33
b_2	15.30	23.50	97.93	18.07	28.88	43.73

FIG. 1: Comparison of (a) LMTO DOS, (b) TB DOS, and (c) DOS of a 76.5 Å CdS nanocrystals.

FIG. 2: The DOS for 46 Å (a) unpassivated and (b) passivated CdS nanocrystals. The inset shows the expanded region encompassing the top of the valence band and the bottom of the conduction band, showing the removal of the midgap states when the nanocrystal is passivated.

FIG. 3: The variation of the TVB and the BCB with size for II-VI nanocrystals.

FIG. 4: The sp^3d^5 TB model with the cation-anion and anion-anion interactions (Ref. [35], filled circles) compared with the sp^3s^* TB nearest neighbor model (Ref. [45], dashed line) and the experimental data points: (a) ZnS: \square Ref. [47], \triangle Ref. [48], ∇ Ref. [49], \diamond Ref. [50], * Ref. [5]; (b) ZnSe: \square Ref. [51], \triangle Ref. [52]; (c) ZnTe: \square Ref. [53]. (d) CdS: \circ Ref. [4], \triangle Ref. [54], * Ref. [55], \square Ref. [56]; (e) CdSe: \square Ref. [57], \triangle Ref. [58], * Ref. [59]; (f) CdTe: \square Ref. [60], \triangle Ref. [61], * Ref. [62]. The solid line passing through the calculated filled circles is the best fit to the calculated points obtained using Eq. 5.

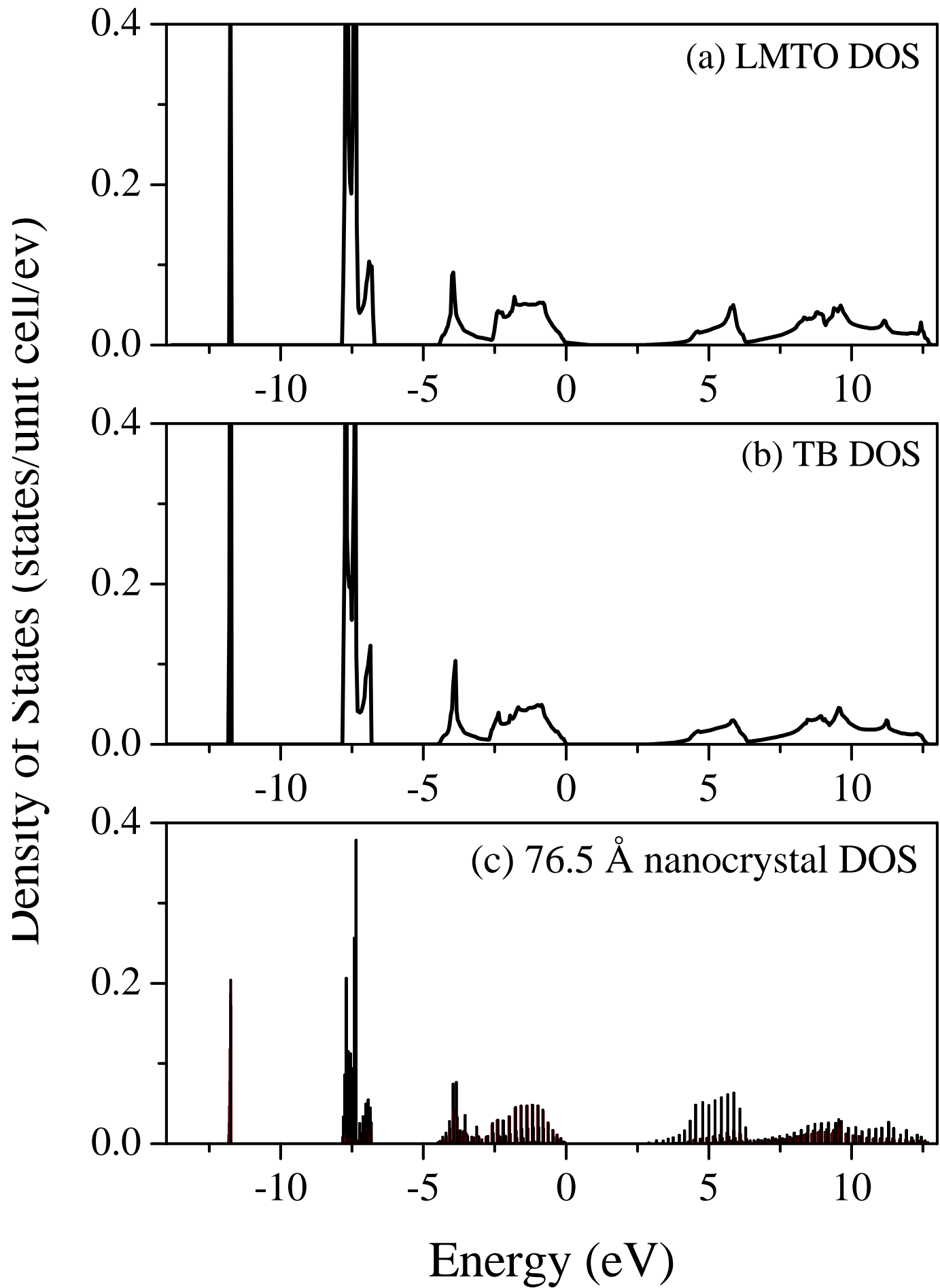


Fig. 1, Sapra and Sarma

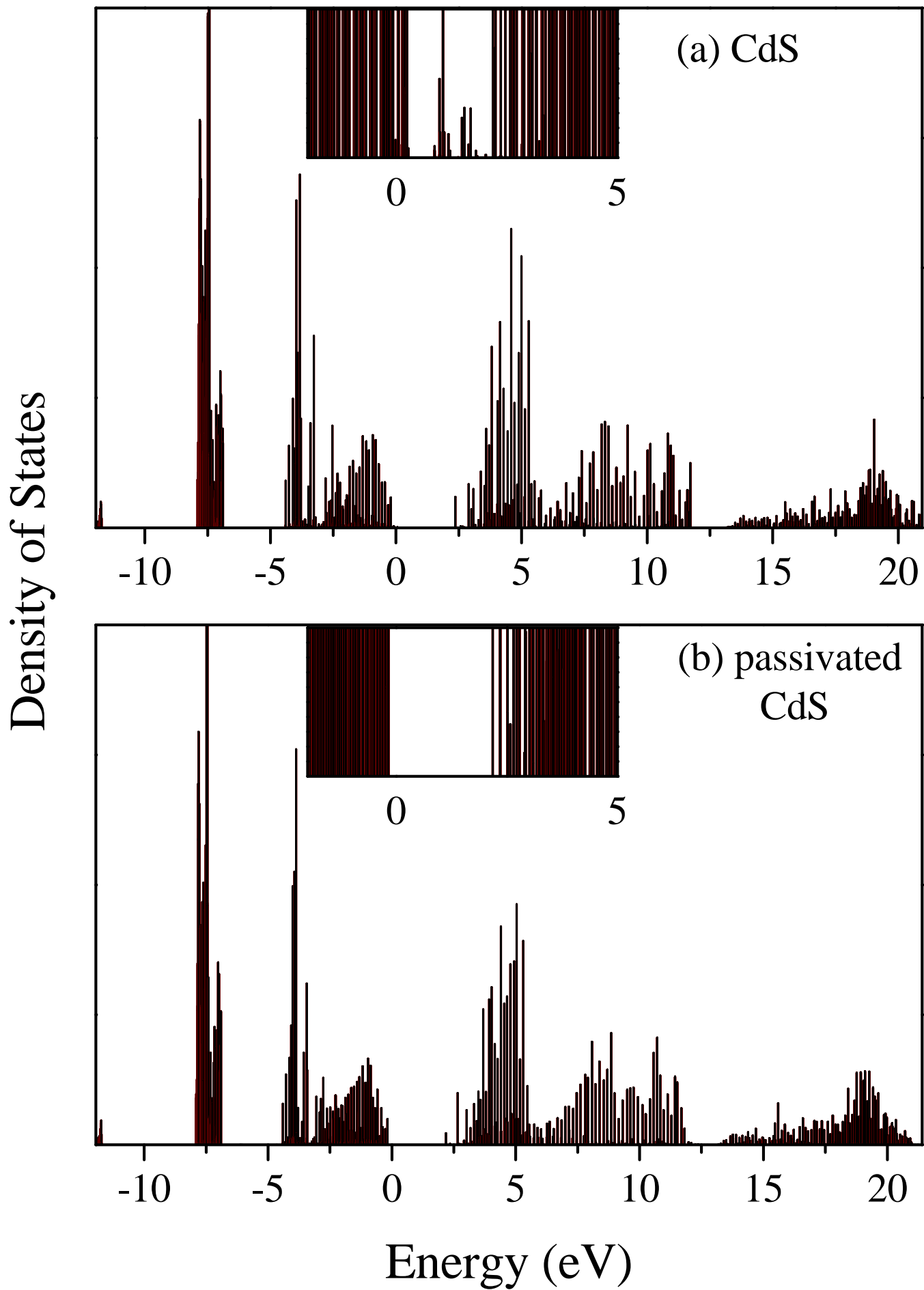


Fig. 2, Sapiro and Sarma

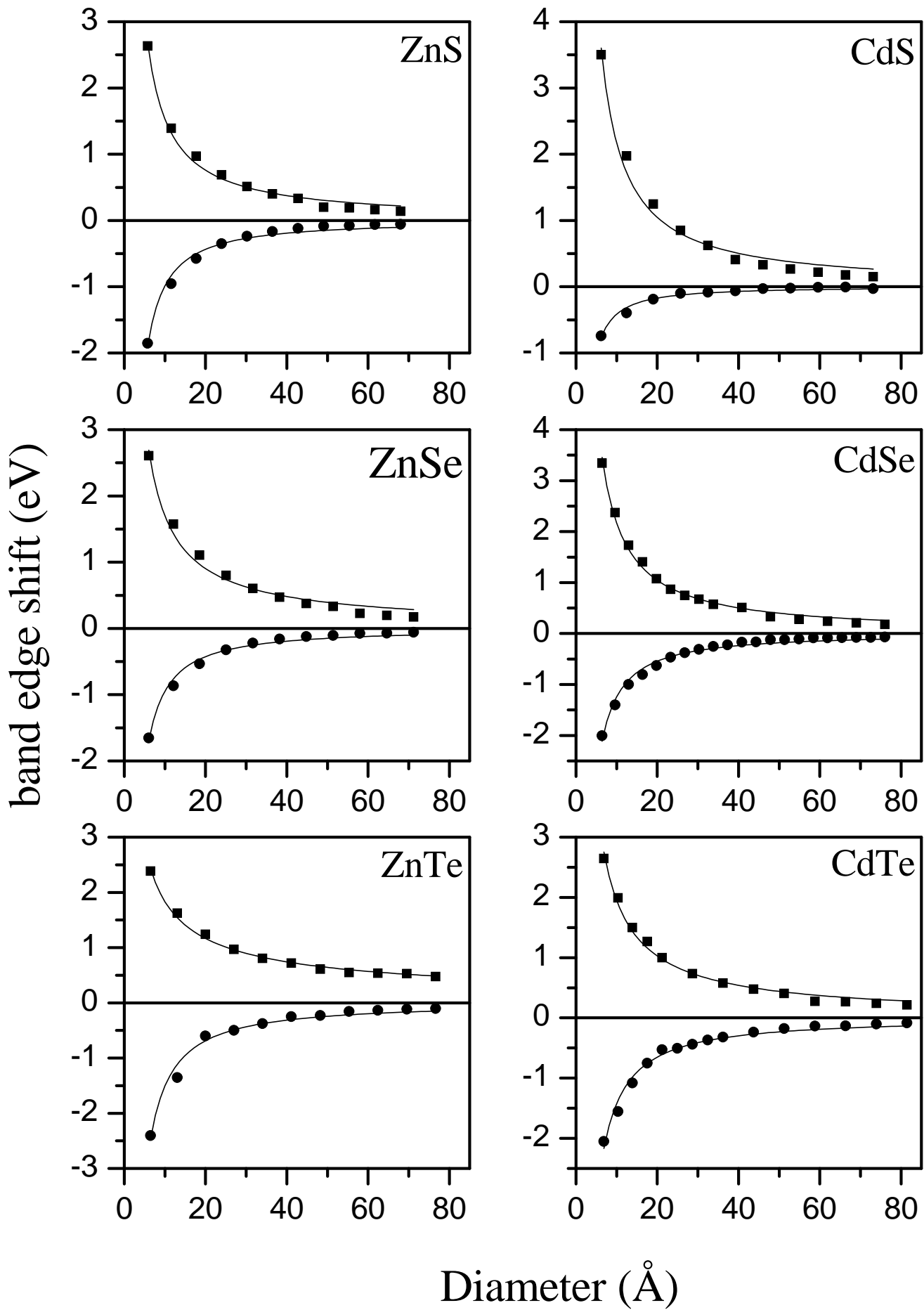


Fig. 3, Sapra and Sarma

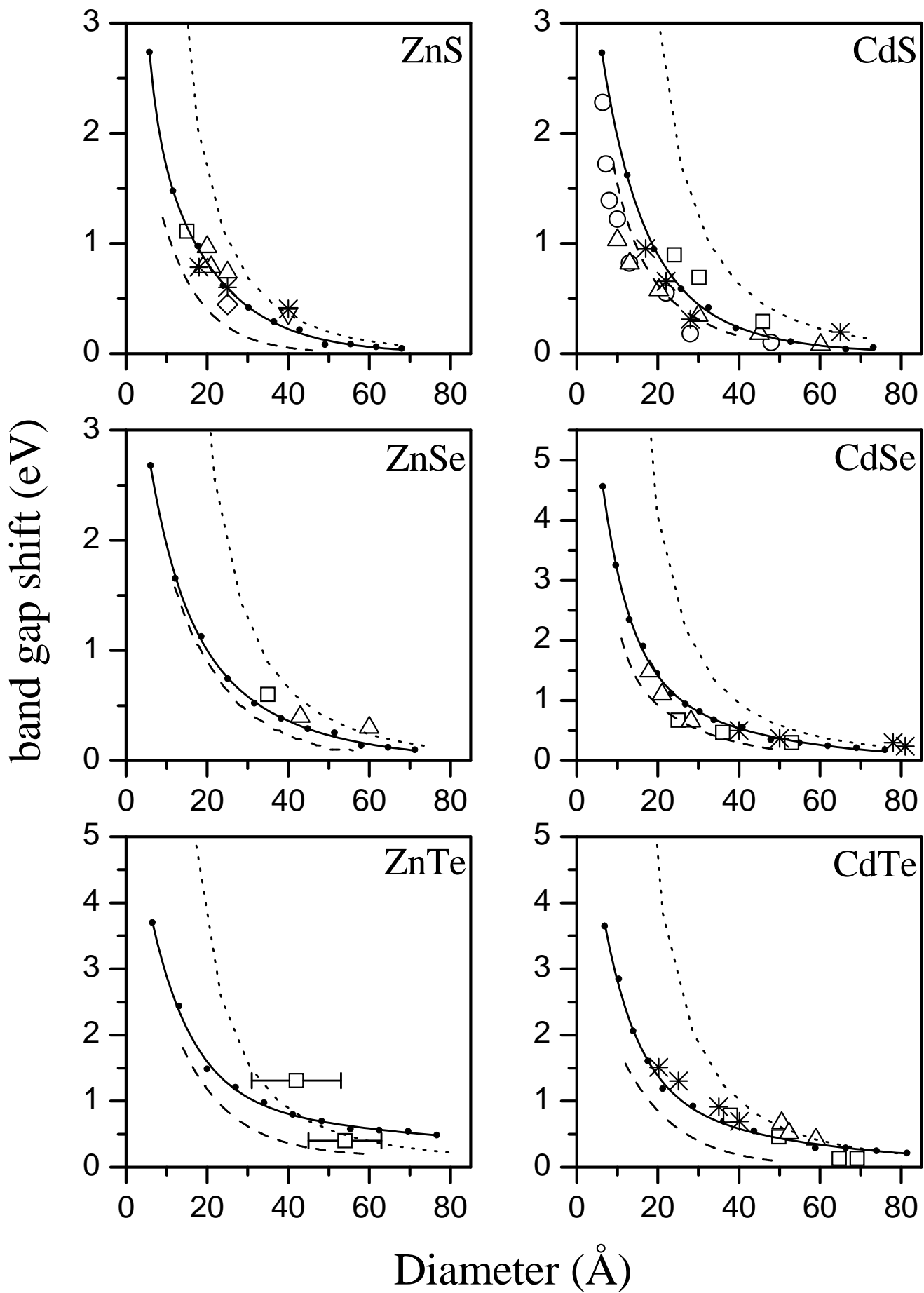


Fig. 4, Sapura and Sarma