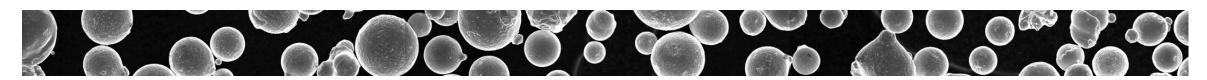
9th International Symposium on Superalloy 718 and Derivatives



Impact of Powder Variability on the Microstructure and Mechanical Behavior of Selective Laser Melted (SLM) Alloy 718



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Acknowledgements



NASA HEOMD / Space Launch System Liquid Engine Office / Additive Manufacturing Structural Integrity Initiative Project (FY16-FY18)

Powder Task:

NASA MSFC

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- William Tilson
- Richard Boothe
- Kenneth Cooper
- **Brian West**
- Douglas Wells
- Dr. Jonathan Woolley
- **AM Fabrication Facility**
- **Heat Treatment Facility**

NASA GRC

- **Robert Carter**
- Dr. Cheryl Bowman
- **Analytical Science Group**
- Mechanical Test Facility

GRC Student Interns

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- Paul Chao (CMU)
- Michael Kloesel (Cal Poly)
- Bethany Cooke (CWRU)
- Jonathan Healy (CWRU)

Space Launch System – Heavy Lift Launch Vehicle – Requires four RS-25 engines to lift core stage









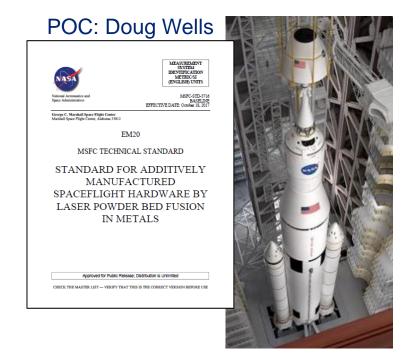
718 Powder Feedstock Variability Study

- Powders evaluated 18 powders from 8 suppliers (A-H)
 - ICP / LECO bulk powder chemistry measurements
 - Count basis particle size distributions (optical silhouettes)
 - Visual comparison of powders
- Processing and Testing Details
- Properties evaluated
 - Build quality and microstructure
 - Tensile behavior
 - High Cycle Fatigue (HCF) results
 - Crack initiation and failure mechanisms
- Summary and Concluding Remarks

Motivation



- Standardization is needed for consistent evaluation of AM processes and parts in critical applications.
- Data on powder feedstock variability in open literature are limited & inadequate
- Supported MSFC technical standard for <u>SLM 718</u> hardware by examining feedstock relationships to processing, homogeneity, durability & performance



Objectives

- Obtain comprehensive industry <u>supplier-to-supplier comparison</u> to understand and identify the feedstock controls important to SLM Alloy 718
- 5 unique powder lots (*B1*, *C1*, *G2*, *G3*, *H1*) have been down-selected for a larger-scale (300 lbs each) investigation underway to include reuse / recyclability study and more expansive mechanical testing





	Com	pare	powder	characteristics
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- > Screen mechanical behavior
- > Lot-to-lot variability
- \triangleright N₂-atomized: 3 of 16
- → 4 cuts same G supplier (separate out size effects)

> (*) 2nd builds allowed once reuse comparisons (SEE PAPER)

Unable to build G1, poor G4 builds

	GRC ID	Alloy 718 Powders	Powder Cut (µm)	Process	Gas
	A1	Supplier 1, Powder 1	15-45	GA	Ar
Reseller	A2	Supplier 1, Powder 2	10-45	GA	Ar
Vendor A	А3	Supplier 1, Powder 3	10-45	GA	Ar
Direct	B1	Supplier 2, Powder 1	15-45	Rotary	Ar
Suppliers	C1	Supplier 3, Powder 1	15-45	GA	Z
	D1	Supplier 4, Powder 1	16-45	GA	Ar
*	D2	Supplier 4, Powder 2	11-45	GA	Ar
	E1	Supplier 5, Powder 1	10-45	GA	N
*	E2	Supplier 5, Powder 2	10-45	GA	N
	F1	Supplier 6, Powder 1	15-45	GA	Ar
*	F2	Supplier 6, Powder 2	10-45	GA	Ar
	G1	Supplier 7, Powder 1	0-22	GA	Ar
	G2	Supplier 7, Powder 2	11-45	GA	Ar
	G3	Supplier 7, Powder 3	16-45	GA	Ar
	G4	Supplier 7, Powder 4	45-90	GA	Ar
	H1	Supplier 8, Powder 1	10-45	GA	Ar

Standard ~15-45 µm SLM cuts (6 powders)

Standard ~10-45 µm SLM cuts (8 powders)

Undersized / oversized cuts (2 powders)

<u>Approach</u>: Procure as many off-the-shelf Alloy 718 powders as possible for a comprehensive <u>supplier-to-supplier comparison</u>



Majority of powder compositions fall within a narrow range than AMS 5664 specification Ni-0.35-0.51 Al, 0-0.039 C, 18.1-19.2 Cr, 18.0-19.2 Fe, 2.9-3.1 Mo, 4.8-5.2 Nb, 0.8-1.0 Ti wt.% + trace impurities

Tight Nb range for primary strengthening by γ"precipitates

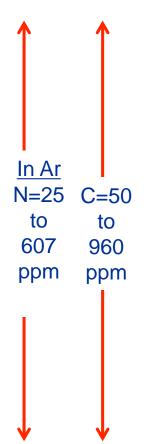
Variation in AI, Cr for secondary strengthening by γ'-precipitates

Will discuss impact of N

MC carbides (Nb, Ti)

E1 did not meet AMS 5664 rangesE2 higher in AI & C but within specC1 high in AI, low in Cr but within spec

									_
	GRC ID	Alloy 718 Powders	Powder Cut (µm)	Process	Gas	Al (wt.%)	Cr (wt.%)	C (wt.% ppm)	N (wt.% ppm)
	A1	Supplier 1, Powder 1	15-45	GA	Ar	0.395	18.82	350	325
Reseller	A2	Supplier 1, Powder 2	10-45	GA	Ar	0.505	18.94	240	90
Vendor A	А3	Supplier 1, Powder 3	10-45	GA	Ar	0.380	18.17	280	331
Direct	B1	Supplier 2, Powder 1	15-45	Rotary	Ar	0.465	19.00	50	25
Suppliers	C1	Supplier 3, Powder 1	15-45	GA	N	0.565	17.45	390	1395
	D1	Supplier 4, Powder 1	16-45	GA	Ar	0.480	19.02	330	122
*	D2	Supplier 4, Powder 2	11-45	GA	Ar	0.495	19.11	305	115
	E1	Supplier 5, Powder 1	10-45	GA	N	0.090	17.71	960	1220
*	E2	Supplier 5, Powder 2	10-45	GA	N	0.705	19.11	470	2770
	F1	Supplier 6, Powder 1	15-45	GA	Ar	0.345	18.25	330	607
*	F2	Supplier 6, Powder 2	10-45	GA	Ar	0.390	18.37	340	370
	G1	Supplier 7, Powder 1	0-22	GA	Ar	0.440	18.82	330	207
	G2	Supplier 7, Powder 2	11-45	GA	Ar	0.455	18.77	360	176
	G3	Supplier 7, Powder 3	16-45	GA	Ar	0.485	18.77	390	199
	G4	Supplier 7, Powder 4	45-90	GA	Ar	0.475	18.77	330	246
;	H1	Supplier 8, Powder 1	10-45	GA	Ar	0.355	18.52	215	562



Powders exhibit distinct particle size distributions



There is variation in average diameters, particle size distribution widths and modalities

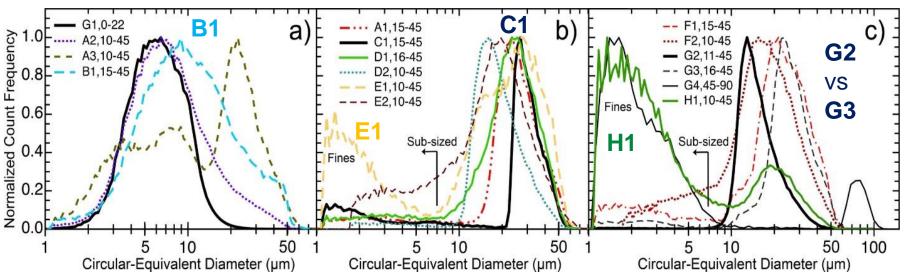
Malvern Morphologi G3SE
Silhouettes of a minimum
20,000 individual powder.

particles per scan

Undersized / Trimodal

Avg. particle diameter for primary peak for most powders is between 23-26 µm for most powders

Regular-sized powders mixture of unimodal and bimodal

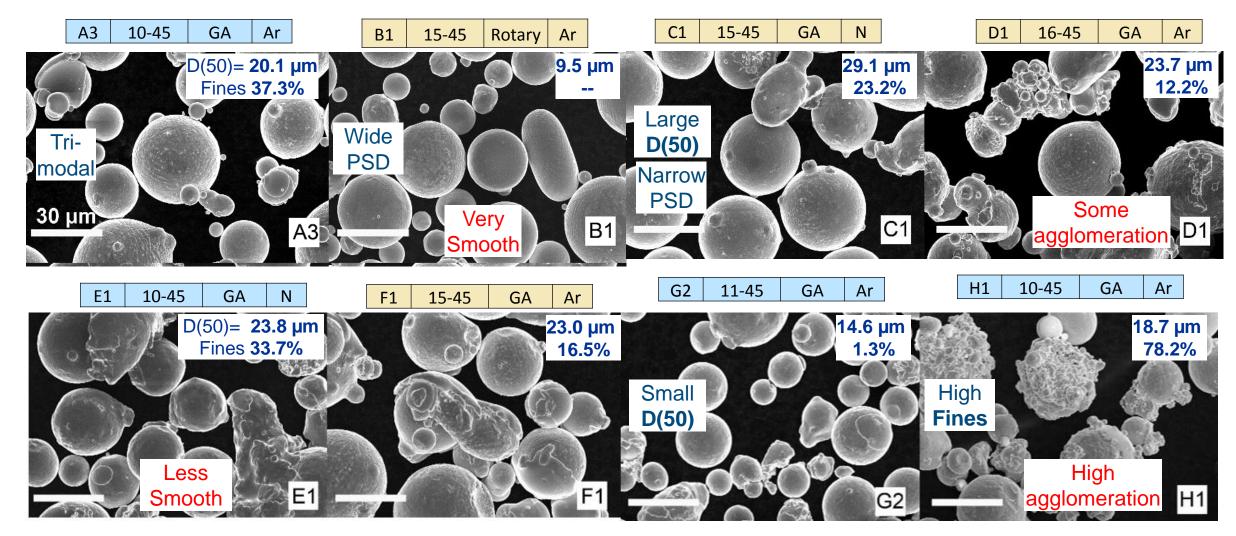


Number basis distributions are more sensitive to fines; Volume basis often reported.

Some suppliers are more successful at reducing fine content

Particles are all highly regular spheroids from all suppliers; Show distinct differences in roughness, fines, & agglomeration





Powders with higher percentage of fines and agglomeration more prone to unplanned stops

Processing and Testing Details

NASA MSFC Concept Laser M1 machine:

- Customized SLM 718 parameters for MSFC RS-25 projects
- Layer thickness: 30 µm
- Continuous scan strategy plus contours

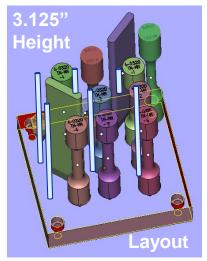
Visible refill lines



configuration requires start /stop to refill piston with powder

Green-state Planned restarts "met" bar

18 builds over 3 months at NASA MSFC



Taper Ends for Easy Snap Off

- 50 lbs of 718 powder procured from most suppliers
- Two microstructure bars
 - **Green-state bar** → inherent to the process
 - Fully heat treated (FHT) bar → post process response

MSFC Build 10 - 16 weeks To GRC

Stress Relief at MSFC

Hot Isostatic Pressing in batches at Outside Vendor

Solution and Aged to AMS 5664E in batches at MSFC

Machine

Reduce porosity, homogenize and remove as-built texture

Screen room temperature mechanical behavior

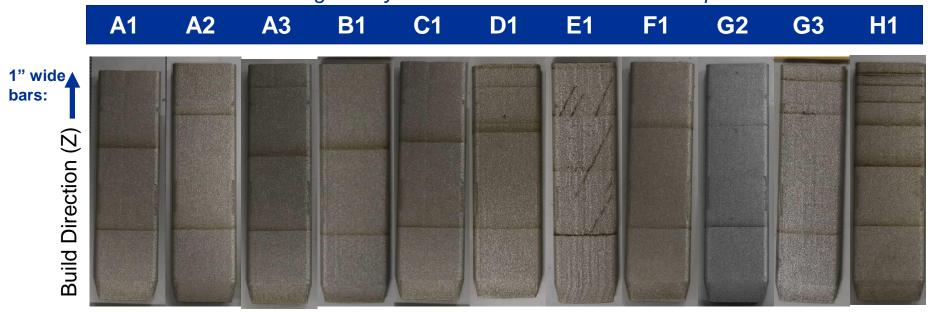
As-Fabricated (AF) vs. Low Stress-Ground (LSG) Surface Conditions

- A tensile test per surface condition
 - Strain control up to 2% then stroke control at equivalent strain rate
- 3 HCF tests per surface condition at 20 Hz and R(σ)= -1
 - Targeted 1 million cycle averages, Runouts above 10 million
 - Stress amplitudes of 271 MPa (40 ksi) for AF and 464 MPa (67 ksi) for LSG

Impact of Feedstock Variability on Build Quality



Green State Met Bars Threshold image analysis of 5 areas in 1 cm x 1 cm XZ piece from mid-section



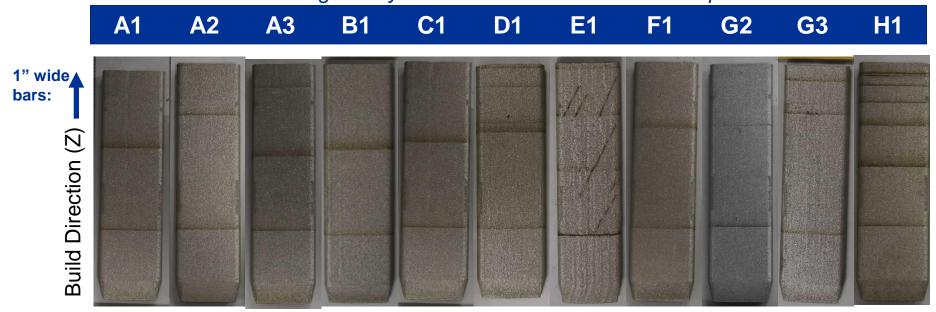
Optimized SLM parameters produces low porosity >

Green	0.19 ±	0.69 ±	0.19 ±	0.18 ±	0.14 ±	0.10 ±	0.46 ±	0.15 ±	0.14 ±	0.14 ±	0.19 ±
Porosity	0.09 %	0.23 %	0.15 %	0.09 %	0.07 %	0.07 %	0.32 %	0.09 %	0.09 %	0.07 %	0.11 %
Green	12.2 ±	22 ± 4	12 ± 3	11.5 ±	10.9 ±	9.6 ±	14.4 ±	9.5 ±	9.3 ±	10.0 ±	8.3 ±
Pore Size	3.0 μm	µm	μm	2.3 μm	2.3 μm	2.6 µm	3.0 µm	2.0 μm	1.8 μm	1.9 μm	1.5 µm
FHT Porosity											
FHT Pore Size											

Impact of Feedstock Variability on Build Quality



Green State Met Bars Threshold image analysis of 5 areas in 1 cm x 1 cm XZ piece from mid-section



Optimized SLM parameters produces low porosity \rightarrow excellent build quality that is further improved with HIP

Green	0.19 ±	0.69 ±	0.19 ±	0.18 ±	0.14 ±	0.10 ±	0.46 ±	0.15 ±	0.14 ±	0.14 ±	0.19 ±
Porosity	0.09 %	0.23 %	0.15 %	0.09 %	0.07 %	0.07 %	0.32 %	0.09 %	0.09 %	0.07 %	0.11 %
Green	12.2 ±	22 ± 4	12 ± 3	11.5 ±	10.9 ±	9.6 ±	14.4 ±	9.5 ±	9.3 ±	10.0 ±	8.3 ±
Pore Size	3.0 µm	µm	µm	2.3 µm	2.3 μm	2.6 µm	3.0 μm	2.0 μm	1.8 μm	1.9 μm	1.5 μm
FHT Porosity	< 0.02 %	< 0.02 %	< 0.02 %	< 0.02 %	0.04 ± 0.02 %	< 0.02 %	< 0.02 %	< 0.02 %	< 0.02 %	< 0.02 %	0.06 ± 0.04 %
FHT Pore	3.3 ± 0.4	3.3 ± 0.3	3.5 ± 0.6	3.4 ± 0.4	3.1 ± 0.6	5.1 ± 1.2	3.3 ± 0.4	3.3 ± 0.5	5.0 ± 0.6	4.5 ± 1.4	4.3 ± 0.6
Size	µm	µm	µm	µm	µm	µm	µm	µm	μm	μm	μm

Fine ~100 nm nitrides present in all builds where volume fraction is linked to N content. Select builds have large nitrides

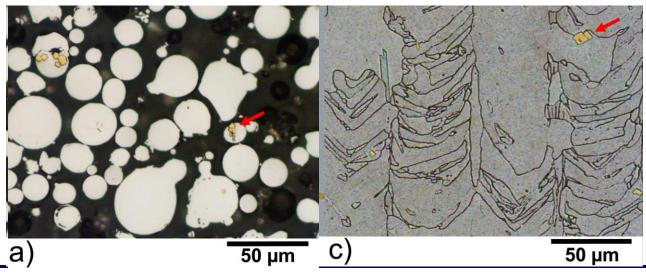


<u> </u>		_			
ID (Gas)	N (wt.% ppm)	Nitride V£** (%)	Nitride Mean Diam. ** (nm)	Carbide Vf(%)	Carbide Mean Diam. (µm)
A1	325	0.32	122 ± 18	0.038	0.56 ± 0.18
A2	90	0.23	106 ± 9	0.10	0.56 ± 0.09
A3	331	0.38	104 ± 12	0.023	0.47 ± 0.19
В1	25	0.17	97 ± 6	0.002	0.24 ± 0.14
C1 (N)	1395	0.54	127 ± 26	0.021	0.37 ± 0.11
D1	122	0.22	90 ± 3	0.09	0.56 ± 0.13
D2	115	0.26	94 ± 8	0.07	0.59 ± 0.17
E1 (N)	1220	0.49	80 ± 16	0.25	0.47 ± 0.05
E2 (N)	2770	0.87 (0.13)	141 ± 11 (7.8±1.0μm)	0.039	0.43 ± 0.05
F1	607	0.47	92 ± 9	0.012	0.40 ± 0.10
F2	370	0.35	110 ± 11	0.054	0.49 ± 0.10
G2	176	0.27	90 ± 5	0.058	0.59 ± 0.18
G3	199	0.34	105 ± 4	0.110	0.49 ± 0.08
G4	246	0.29	114 ± 14	0.058	0.59 ± 0.17
H1	562	0.42 (0.013)	112 ± 11 (6.9±0.4μm)	0.009	0.33 ± 0.10

Larger nitrides that are 6-8 µm in diameter may act as crack initiators

These large nitrides form during powder production

MC carbides are sub-micron in diameter and mostly uniformly distributed



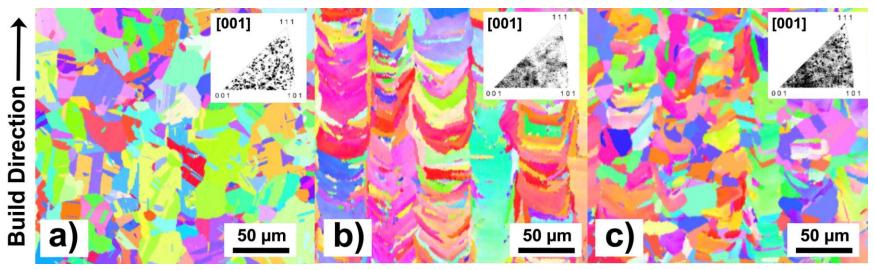
Three grain structure regimes observed after heat treat





Retained SLM structure for Partially Recrystallized lots atomized in nitrogen

for H1 lot



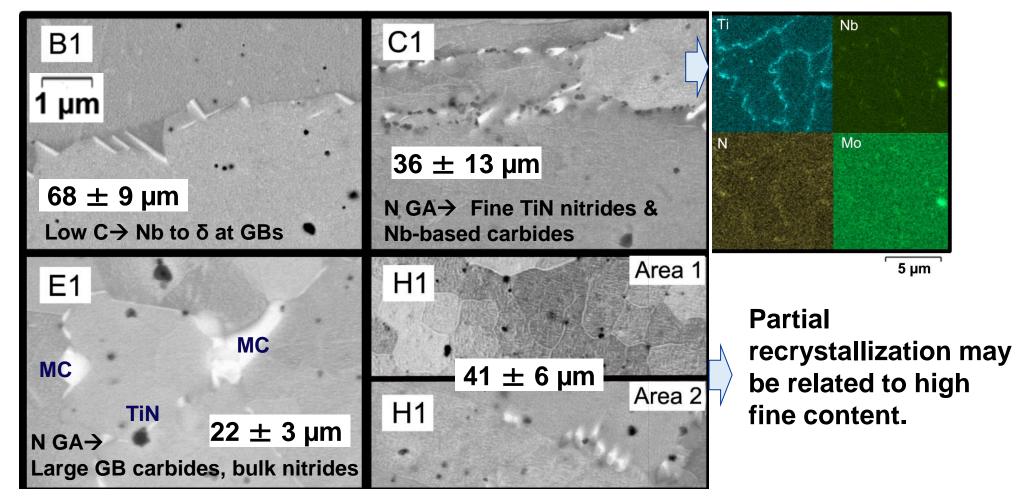
Recommend Ar-atomization and N content < 400 ppm for homogeneous grain distribution

Linear intercept	A 1	A2	А3	B1	C1 (N GA)	D1	D2	E1 (N GA)	E2 (N GA)	F1	F2	G2	G3	H1
Mean grain diameter (± 95% CI)	70 ± 5 μm	57 ± 4 μm	74 ± 12 μm	68 ± 9 μm	36 ± 5 μm	53 ± 4 μm		21.5 ± 1.3 µm	32 ± 3 μm	89 ± 12 μm	64 ± 18 μm	63 ± 6 µm	71 ± 6 μm	40.9 ± 2.3 μm
N content ppm	325	90	331	25	1395	122	115	1220	2770	607	370	176	199	562

Nitrides and carbides pin grain boundaries in N-atomized powders (C1, E1, E2), retains smaller (001)-oriented grain sizes from SLM fabrication post HIP.





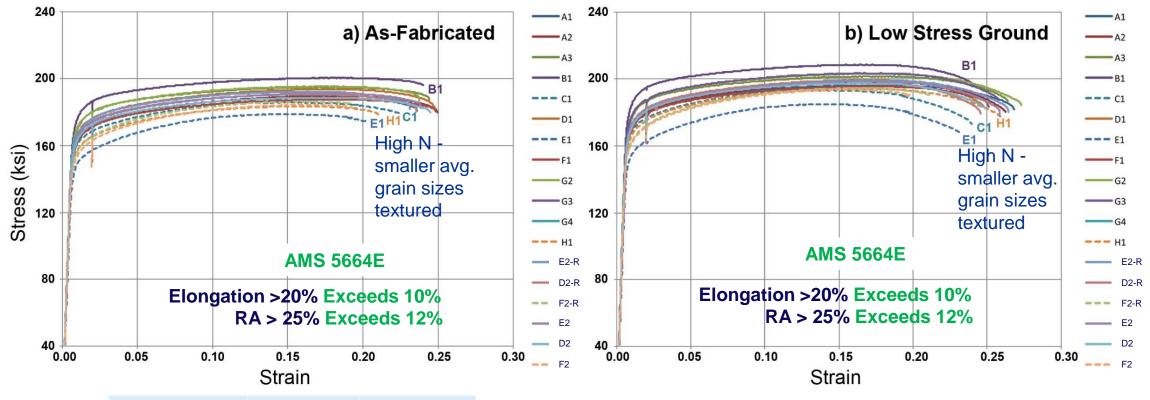


Majority builds show few minor phases at GBs: (N<500 ppm) & modest C

Room Temperature Tensile Testing

Heat Treated SLM 718 meets or exceeds minimum requirements for lots within chemistry specification





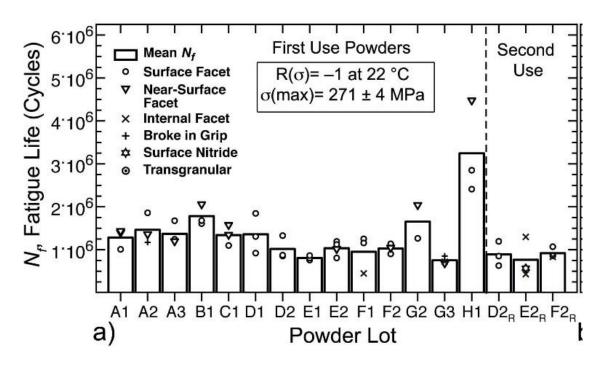
Solution and aged bars

As-fabricated	UTS (ksi)	0.2% YS Offset (ksi)
AMS 5664E	180.0	150.0
B1 (Low C)	200.5	171.1
Rest (H1 >>G2)	183.5-195.5	151.6-165.4
E1 (Off Spec)	178.8	144.9

Low Stress Ground	UTS (ksi)	0.2% YS Offset (ksi)
AMS 5664E	180.0	150.0
B1 (Low C)	208.8	179.3
Rest (H1 >>G3)	193.4-203.6	160.8-165.4
E1 (Off Spec)	185.0	150.6

HCF Response for As-fabricated surface condition





The surfaces of H1 test bars were more oxidized (SEE PAPER)

Overall low scatter in HCF response compared to the low stress ground

Predominant failure sites for was grain facets at or near the surface

Very few internal initiations

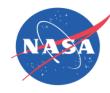


failures was significantly higher for AF surfaces due to stress concentrators associated with SLM surface asperities

Incidence of surface

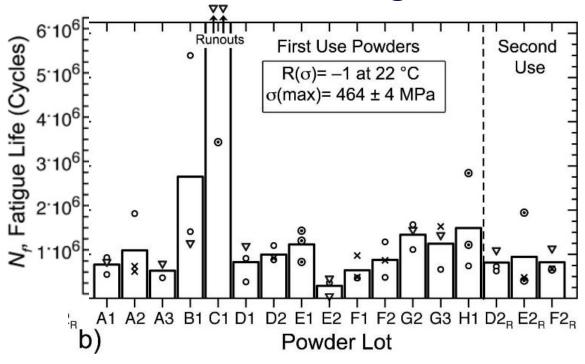
- **Surface facet**
- **Near-surface** facet (within ~200 µm)
- **Surface facet** with steps
- - Internal facet e) Large 50 µm nitride (dark contrast)

HCF Response for low-stress ground surface condition





- Surface Facet
- ▼ Near-Surface Facet
- × Internal Facet
- + Broke in Grip
- ☆ Surface Nitride
- Transgranular



Overall more scatter in HCF

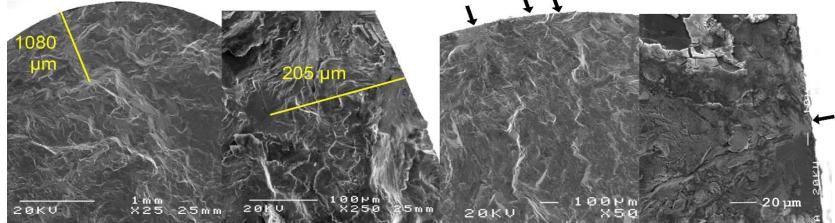
C1: N-atomized with refined grain size from pinned GBs

B1: highest strength, some GB pinning from delta

Predominant failure sites for was grain facets at or near the surface

More internal initiations

Transgranular crack initiation also observed



- a) Internal facet
- b) Near-surface c) facet (within ~200 μm)

d) Transgranular initiation

Summary and concluding remarks



- Powders evaluated are distinct similar in that particles are highly regular spheroids- show differences in Al, C, N; PSDs, degree of agglomeration and surface roughness
- Optimized SL M parameters for 718 yielded high quality builds with low porosity and acceptable tensile properties across many distinct powder lots
- Compositional differences has strongest impact on SLM 718 microstructure
 - → High N and C contents form TiN-nitrides and MC carbides on GBs that suppresses recrystallization during HT → 400 ppm N content a good rule of thumb cutoff to ensure equiaxed grain distribution
 - ➤ The **B1** alloy with very low in C led to **higher delta content leading to highest UTS**, while the **E1** alloy with very low in Al and high in C exhibited the lowest UTS
- Significant knock-down in room temp HCF response for as-built SLM surface condition;
 Stress concentrators at surface lead to higher incidence of surface crack initiation than observed in low stress ground condition
- For LSG surface condition, the best room temperature HCF was for N-atomized C1 with prior GB particles (TiN, Nb-based carbides) that persist through heat treatment

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