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# Research Article

# A Cost-Effective Solid-State Approach to Synthesize $g-C_3N_4$ Coated $TiO_2$ Nanocomposites with Enhanced Visible Light Photocatalytic Activity

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Novel graphitic carbon nitride  $(g-C_3N_4)$  coated  $TiO_2$  nanocomposites were prepared by a facile and cost-effective solid-state method by thermal treatment of the mixture of urea and commercial  $TiO_2$ . Because the  $C_3N_4$  was dispersed and coated on the  $TiO_2$  nanoparticles, the as-prepared  $g-C_3N_4/TiO_2$  nanocomposites showed enhanced absorption and photocatalytic properties in visible light region. The as-prepared  $g-C_3N_4$  coated  $TiO_2$  nanocomposites under  $450^\circ C$  exhibited efficient visible light photocatalytic activity for degradation of aqueous MB due to the increased visible light absorption and enhanced MB adsorption. The  $g-C_3N_4$  coated  $TiO_2$  nanocomposites would have wide applications in both environmental remediation and solar energy conversion.

#### 1. Introduction

Visible light photocatalysis has attracted the worldwide attention due to its potential application in environmental remediation and solar energy conversion [1–7]. The photocatalyst  ${\rm TiO_2}$ , however, can only utilize the ultraviolet light (about 5% of natural solar light) because of its wide band gap (ca. 3.2 eV for anatase  ${\rm TiO_2}$ ). During the past 40 years, many efforts have been devoted to enhance the visible light photocatalytic activity of  ${\rm TiO_2}$ , including metal doping [8–10], nonmetal doping [11–14], surface modification [15], and heterojunction construction [16–19].

In recent years, polymeric g-C<sub>3</sub>N<sub>4</sub> materials have attracted much attention because of their similarity to graphene. Zhang et al. reported that the polymeric g-C<sub>3</sub>N<sub>4</sub> semiconductors exhibit high photocatalytic performance for water splitting under visible light irradiation [20]. Dong and coworkes reported that polymeric g-C<sub>3</sub>N<sub>4</sub> layered materials as novel efficient visible light photocatalyst, which can be synthesized facilely by directly heating urea or thiourea [21, 22].

Very recently, Zhou et al. reported a g-C<sub>3</sub>N<sub>4</sub>/TiO<sub>2</sub> nanotube array heterojunction with excellent visible light

photocatalytic activity [17]. Zhao et al. reported g- $\mathrm{C_3N_4}/\mathrm{TiO_2}$  hybrids with wide absorption wavelength and effective photogenerated charge separation [18]. However, the precursors for g- $\mathrm{C_3N_4}$  (dicyandiamide and melamine) are poisonous and detrimental to the environment. The preparation processes were relatively tedious, which may prevent large-scale application [17, 18].

In the present work, g- $\mathrm{C_3N_4/TiO_2}$  nanocomposites were prepared by a facile and cost-effective solid-state method using urea and commercial  $\mathrm{TiO_2}$  as precursors. It was interesting to find that g- $\mathrm{C_3N_4}$  was in situ coated on the surface of  $\mathrm{TiO_2}$ . The precursors (urea and commercial  $\mathrm{TiO_2}$ ) are low cost and easily available. The as-prepared g- $\mathrm{C_3N_4}$  coated  $\mathrm{TiO_2}$  nanocomposites exhibited enhanced photocatalytic activity under visible light irradiation.

#### 2. Experimental

*2.1. Synthesis.* The g- $C_3N_4$  coated  $TiO_2$  nanocomposites were prepared by a facile and cost-effective solid-state method. In a typical synthesis, 2 g  $TiO_2$  and 6 g urea were immersed in 10 mL  $H_2O$  and dried at  $60^{\circ}C$  to completely remove the water.

The mixtures were put into an alumina crucible with a cover, and then heated to a certain temperature in the range of 400 and 600°C in a muffle furnace for 1h at a heating rate of 15°C min<sup>-1</sup>. The final samples were collected for use without further treatment.

2.2. Characterization. The crystal phases of the sample were analyzed by X-ray diffraction with Cu K $\alpha$  radiation (XRD: model D/max RA, Rigaku Co., Japan). The morphology and structure of the samples were examined by transmission electron microscopy (TEM: JEM-2010, Japan). The UV-vis diffuse reflection spectra were obtained for the dry-pressed disk samples using a Scan UV-Vis spectrophotometer (UV-Vis DRS: UV-2450, Shimadzu, Japan) equipped with an integrating sphere assembly, using BaSO<sub>4</sub> as reflectance sample. The spectra were recorded at room temperature in air range from 250 to 800 nm. X-ray photoelectron spectroscopy with Al K $\alpha$  X-rays ( $h\nu = 1486.6 \,\text{eV}$ ) radiation operated at 150 W (XPS: Thermo ESCALAB 250, USA) was used to investigate the surface properties. The shift of the binding energy due to relative surface charging was corrected using the C1s level at 284.8 eV as an internal standard. FT-IR spectra were recorded on a Nicolet Nexus spectrometer on samples embedded in KBr pellets. The nitrogen adsorption-desorption isotherms were determined by the BET method (BET-BJH: ASAP 2020, USA), from which the surface area, pore volume, and average pore diameter were calculated by using the BJH method. All the samples were degassed at 200°C prior to measurements.

2.3. Evaluation of Photocatalytic Activity. Photocatalytic activity of g-C<sub>3</sub>N<sub>4</sub>/TiO<sub>2</sub> for MB photodegradation was evaluated in a quartz glass reactor. 0.05 g of N-TiO2 was dispersed in MB aqueous solution (50 mL, 5 mg/L). The light irradiation system contains a 500 W Xe lamp with a jacket filled with flowing and thermostated aqueous NaNO2 solution (1 M) between the lamp and the reaction chamber as a filter to block UV light ( $\lambda$  < 400 nm) and eliminate the temperature effect. The suspension was first allowed to reach adsorptiondesorption equilibrium with continuous stirring for 60 min in the dark prior to irradiation. The degradation rate of MB was evaluated using the UV-Vis absorption spectra to measure the peak value of a maximum absorption of MB solution. During the irradiation, 5 mL of suspension was continually taken from the reaction cell at given time intervals for subsequent dye concentration analysis after centrifugation. The MB solution shows a similar pH value at 6.8, which does not affect the light absorption of MB. The maximum absorption of MB is at wavelength of 665 nm. The degradation rate  $\eta$  (%) can be calculated as

$$\eta(\%) = \frac{(C_0 - C)}{C_0} \times 100\%,$$
(1)

where  $C_0$  is the initial concentration of MB considering MB adsorption on the catalyst and C is the revised concentration after irradiation.

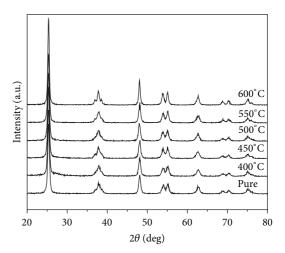


FIGURE 1: XRD patterns of the  $\rm C_3N_4/TiO_2$  nanocomposites obtained under different temperatures.

# 3. Results and Discussion

Figure 1 shows the XRD patterns of the as-prepared g- $C_3N_4$  coated  $TiO_2$  nanocomposites at different temperatures. The peaks of all the samples can be indexed to the anatase phase of  $TiO_2$  (JCPDS file No. 21-1272). It can be seen that the peak intensity increases gradually under higher treatment temperature, which indicates that the crystal sizes of  $TiO_2$  nanocomposites increase under higher treatment temperature. No typical peaks of  $g-C_3N_4$  can be found for all the samples due to the fact that  $g-C_3N_4$  with layered structures on the surface of  $TiO_2$  is ultrathin (Figure 2) and the crystallinity is low [22].

The morphology of pure  $TiO_2$  and  $g-C_3N_4/TiO_2$  nano-composites were observed by TEM. As shown in Figure 2, both samples contain a number of monodispersed nanoparticles of  $TiO_2$  with a size of about 11 nm. The intra-aggregation of particles could form the mesoporous structure [23]. It can be seen from Figure 2(b) that the ultrathin  $g-C_3N_4$  with layered structures are dispersed and coated on the surface of  $TiO_2$  particles, which is consistent with absence of the peaks of  $g-C_3N_4$  in XRD (Figure 1).

The FT-IR spectra of pure  $TiO_2$  and  $g-C_3N_4$  coated  $TiO_2$  nanocomposites are shown in Figure 3(a). The absorption band around  $400-800~\rm cm^{-1}$  is attributed to Ti-O bonds [23]. Several bands in the range of  $1100-1650~\rm cm^{-1}$  correspond to the typical stretching vibration of CN heterocycles in  $g-C_3N_4$ . The characteristic vibration mode of triazine units can also be found at  $801~\rm cm^{-1}$  [22]. The peak at  $1630~\rm cm^{-1}$  is associated with the stretching vibration of water molecules for both samples, including molecular water and hydroxyl groups [23]. The FT-IR spectra further confirm the existence of  $g-C_3N_4$  on the surface of  $TiO_2$ .

The TG and DSC thermograms (Figure 3(b)) show that there are several phase transformations during heating. An endothermal peak at 135°C is the melting point of urea. The peak at 242°C indicates the reaction of urea into melamine. The weight loss during the two stages decreases rapidly by 36.1%. The sharp peak at 367°C implies that the thermal

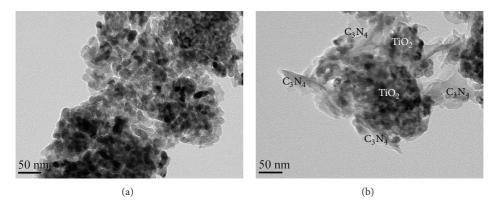


FIGURE 2: TEM images of pure TiO<sub>2</sub> (a) and g-C<sub>3</sub>N<sub>4</sub>/TiO<sub>2</sub> nanocomposites sample obtained under 450°C (b).

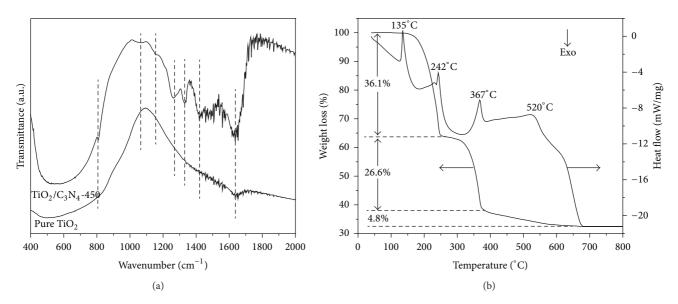


FIGURE 3: FTIR spectra of pure TiO<sub>2</sub> and g-C<sub>3</sub>N<sub>4</sub>/TiO<sub>2</sub> nanocomposites (a) and TG-DSC for heating the mixture of TiO<sub>2</sub> and urea (b).

condensation of melamine into g- $C_3N_4$  occurred in this temperature range. The weight loss in this stage is about 26.6%. The further weight loss of 4.8% with endothermal peak at 520°C can be attributed to the decomposition of g- $C_3N_4$ . The TG-DSC result implies that g- $C_3N_4$  can be *in situ* formed on the surface of  $TiO_2$ , which is consistent with Figure 2(b).

The CIs spectra in Figure 4(a) show that two main carbon species with binding energies of 284.9 and 288.1 eV, corresponding to C–C and C–N–C, respectively. Three binding energies in N1s region (Figure 4(b)) can be observed, which can be indexed to C–N–C (398.8 eV), N–(C)<sub>3</sub> (400.1 eV), and N–H groups (401.2 eV), respectively. The binding energy at 529.7 and 533.0 eV can be ascribed to Ti–O, surface hydroxyl groups, and adsorbed molecular water (Figures 4(c) and 4(d)) [22]. The XPS results are consistent with the FT-IR spectra. XPS results also indicate that no peak for Ti–C or Ti–N bond can be observed, which implies that there is no chemical bond connection between g-C<sub>3</sub>N<sub>4</sub> and TiO<sub>2</sub>.

The nitrogen adsorption-desorption isotherms of pure TiO<sub>2</sub> and g-C<sub>3</sub>N<sub>4</sub>/TiO<sub>2</sub> nanocomposites obtained under

450°C are shown in Figure 5(a). The two samples show a type IV adsorption isotherm with a  $\rm H_2$  hysteresis loop in the range  $(P/P_0)$  of 0.6–1.0, which indicates the presence of mesopores. The surface areas and pore volume of pure  $\rm TiO_2$  are 78 m²/g and 0.281 cm³/g, higher than those of g- $\rm C_3N_4/\rm TiO_2$  nanocomposites (48 m²/g and 0.216 cm³/g). The pore size distribution curve (Figure 5(b)) indicates that the large mesopores of pure  $\rm TiO_2$  and g- $\rm C_3N_4/\rm TiO_2$  nanocomposites are about 37 and 48 nm, respectively. The presence of large mesopores can be ascribed to the aggregation of  $\rm TiO_2$  particles. It can be observed that the g- $\rm C_3N_4/\rm TiO_2$  nanocomposites have small mesopores of around 13.6 nm (inset in Figure 5(b)), which originates from the presence of layered g- $\rm C_3N_4$  on the  $\rm TiO_2$  surface. The small mesopore is advantageous for enhancing the adsorption for reactant.

Figure 6 shows the UV-Vis DRS spectra of pure  ${\rm TiO_2}$  and the as-prepared  ${\rm g-C_3N_4/TiO_2}$  nanocomposites. It is clear that the visible light absorption of  ${\rm g-C_3N_4/TiO_2}$  nanocomposites is enhanced with increased treatment temperatures until 450°C. Then the visible light absorption decreases when the temperature is higher than 450°C. This fact implies

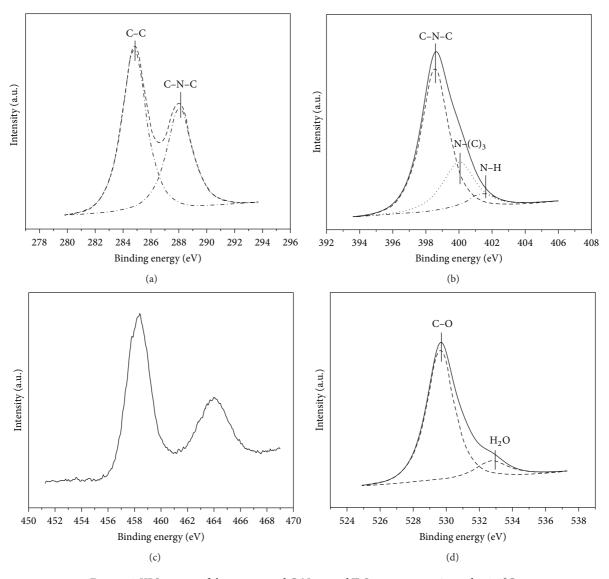


FIGURE 4: XPS spectra of the as-prepared C<sub>3</sub>N<sub>4</sub> coated TiO<sub>2</sub> nanocomposite under 450°C.

that the as-prepared g- $C_3N_4$  coated  $TiO_2$  nanocomposites under 450°C may exhibit excellent visible light photocatalytic activity. However, the decrease of visible light absorption intensity of coated  $TiO_2$  nanocomposites under higher treatment temperature can be attributed to the decomposition of g- $C_3N_4$ .

Figure 7 shows the adsorption and photocatalytic activity of pure  ${\rm TiO_2}$  and  ${\rm g-C_3N_4/TiO_2}$  nanocomposites for removal of MB. It can be seen that the  ${\rm g-C_3N_4/TiO_2}$  nanocomposites obtained under  ${\rm 450^{\circ}C}$  exhibit the highest adsorption capacity, which may be ascribed to presence of layered  ${\rm g-C_3N_4}$  and small mesopores of the nanocomposites sample. The photocatalytic activities of  ${\rm g-C_3N_4/TiO_2}$  nanocomposites first increase and then decrease with the increased treatment temperature. Pure  ${\rm TiO_2}$  shows low visible light activity due to its large band gap. The observed slight visible light activity for the pure  ${\rm TiO_2}$  sample can be ascribed to the photosensitization effect of the MB as MB can absorb visible light

[18]. During the visible light irradiation, the part of MB was self-decomposed due to the photosensitization. When TiO<sub>2</sub> was coated by g-C<sub>3</sub>N<sub>4</sub>, all the nanocomposite samples show decent visible light activity. Under visible light irradiation, g-C<sub>3</sub>N<sub>4</sub> with a band gap of 2.7 eV could be excited and the photogenerated electrons could transfer from the conduction band (CB) of g-C<sub>3</sub>N<sub>4</sub> to the CB of TiO<sub>2</sub> [17, 18, 24]. The holes in the valence band (VB) of g-C<sub>3</sub>N<sub>4</sub> and electrons on the CB of TiO<sub>2</sub> could initiate the following degradation reactions. The as-prepared g-C<sub>3</sub>N<sub>4</sub>/TiO<sub>2</sub> nanocomposites under 450°C exhibit the highest photocatalytic activity under visible light irradiation. Considering the fact that the surface area of g-C<sub>3</sub>N<sub>4</sub>/TiO<sub>2</sub> nanocomposites (48 m<sup>2</sup>/g) is lower than that of pure TiO<sub>2</sub> (78 m<sup>2</sup>/g), the surface area of g-C<sub>3</sub>N<sub>4</sub>/TiO<sub>2</sub> is not a positive factor. The enhanced visible light activity of g-C<sub>3</sub>N<sub>4</sub>/TiO<sub>2</sub> should be ascribed to the enhanced visible light adsorption because of the presence of g-C<sub>3</sub>N<sub>4</sub> (Figures 2(b) and 6) and the improved MB adsorption

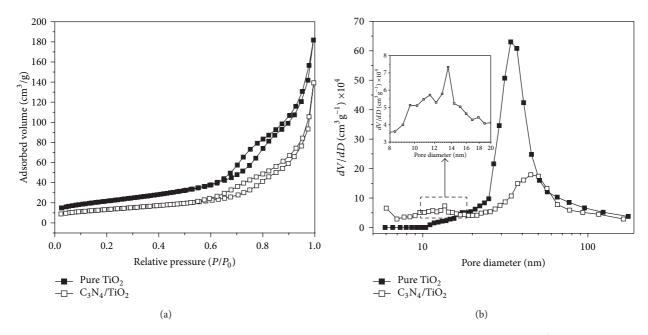


FIGURE 5: BET-BJH of the pure TiO<sub>2</sub> and C<sub>3</sub>N<sub>4</sub> coated TiO<sub>2</sub> nanocomposite obtained under 450°C.

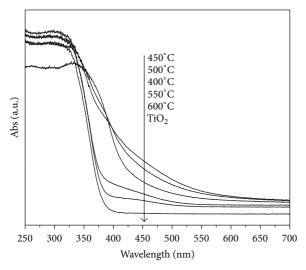


FIGURE 6: UV-Vis DRS of the pure  ${\rm TiO_2}$  and  ${\rm g\text{-}C_3N_4/TiO_2}$  samples obtained under different temperatures.

because of the small mesopores of the nanocomposites sample (Figure 5(b)). As the precursors (urea and commercial  ${\rm TiO_2}$ ) are cheap and preparation method is very simple, the as-prepared g-C<sub>3</sub>N<sub>4</sub> coated  ${\rm TiO_2}$  nanocomposites are ready for large-scale applications in environmental pollution control and solar energy conversion [25].

# 4. Conclusion

The  $g-C_3N_4/TiO_2$  nanocomposites were synthesized by a cost-effective solid-state approach by thermal treatment of the mixture of urea and commercial  $TiO_2$ . It was found that the surface of  $TiO_2$  particles was coated by the *in* 

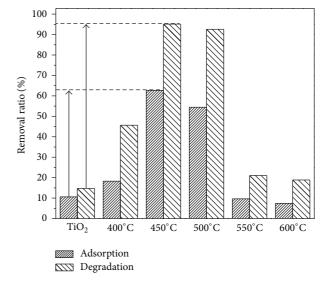


FIGURE 7: Adsorption and photocatalytic activity of the pure  ${\rm TiO_2}$  and  ${\rm g\text{-}C_3N_4/TiO_2}$  samples obtained under different temperatures for removal of MB.

situ formed thin layered g-C<sub>3</sub>N<sub>4</sub> from urea. The adsorption capacity and visible light photocatalytic activity were significantly enhanced. Under the optimized treatment temperature of 450°C, the g-C<sub>3</sub>N<sub>4</sub>/TiO<sub>2</sub> nanocomposites exhibited highest adsorption capacity and visible light photocatalytic activity toward removal of MB. The enhanced adsorption capacity can be ascribed to the presence of g-C<sub>3</sub>N<sub>4</sub> and small mesopores. The enhanced visible photocatalytic activity originated from the increased visible light adsorption and small mesopores of the nanocomposites sample. The novel

 $g-C_3N_4$  coated  $TiO_2$  nanocomposites prepared by the cost-effective solid-state approach would find wide application in environmental remediation.

#### **Conflict of Interests**

The authors declare no conflict of interests.

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