

# SURFACE PROPERTY CONTROL BY ELECTRON BEAM DEFLECTION DURING HARDENING

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## Resume

The usage of the high-energy electron beam source enables repeated surface quenching of chosen locations of an engineering part surface. Different techniques of electron beam deflection allow creating hardened layers of different shapes, hardness levels and thicknesses. Experiments were carried out with 42CrMo4 (1.7225) steel. The deflection modes tested were one-point, 6-point, line, field, and meander. The influence of process speed and defocusing of the electron beam was also taken into account. The electron beam surface quenching resulted in a very fine martensitic microstructure with a hardness of over 700 HV0.5. The thickness of the hardened layers depends on the deflection mode and is affected directly (except field deflection) by process speed. The maximum hardened depth (NCHD) was 1.49 mm. Electron beam defocusing affects the width of the hardened track and can cause extension of the trace up to 40%. The hardness values continuously decrease from the surface to the material core.

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## 1. Introduction

Electron beam (EB), together with laser beam, belongs to the advanced technologies that can be used for local surface heat treatment. Both methods have some similar characteristics; however, there are clear differences predetermining which of them will be chosen for a particular application. The fast beam deflection appears to be one of the typical features of EB processing that allows a different distribution of the supplied energy provided adequate programming of the hardening equipment is applied. An EB can be deflected both in a direction perpendicular to the component movement direction and also in a parallel direction [1 - 5].

The properties of the hardened layer can be directly controlled by process parameters. The total supplied power rate is controlled by a combination of the accelerating voltage

“ $U_{EB}$ ” and the electron current “ $I_{EB}$ ”. This energy is distributed to the component surface depending on the selected mode of EB deflection. The scanned area is determined by the dimensions “SWX” and “SWY” and is set together with the scanning frequency in the individual directions “FRQ”, “FRQ2” (Fig. 1). Usually some beam defocusing “Offset” is set up, which can be implemented by shifting of the focal plane above the quenched surface (a positive value) or below the surface (a negative value). The last very important parameter is the quenched component movement rate “ $v_s$ ” under the hardening beam “EB” [6 - 10].

The aim of the paper is to increase the knowledge of the influence of electron beam deflection modes on surface heat treatment processes, which is not discussed in the literature in comparison with electron

beam welding. The results can be efficiently used when constructing new EB devices and when optimizing and improving electron beam surface hardening processes.

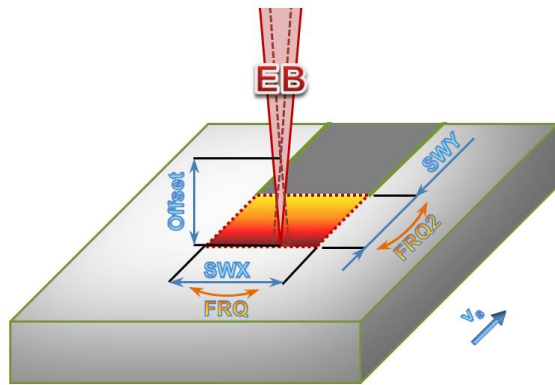


Fig. 1. Figure 1 - The EB surface quenching parameters scheme.  
(full colour version available online)

## 2. Experimental material and methods

Experiments were carried out on the high-grade 42CrMo4 (1.7225 or AISI 4140) steel with the following chemical composition of (in wt. %): C 0.41, Mn 0.69, Si 0.25, Cr 1.04, Mo 0.20, which is a suitable material for surface hardening. It finds application where elevated strength in combination with a defined and high level of toughness are the most important requirements – e. g. shafts or gears. The material to be tested in block shape (20 × 40 × 250 mm) was in a state after tempering at 600° C for 3 hours with a final fine sorbitic structure and an average hardness of 300 HV0.5.

The surface quenching was performed using PROBEAM K26 equipment adopting the electron beam technology with a maximum beam power of 15 kW and an accelerating voltage from 80 to 150 kV. The widths of the EB hardened traces were set to be SWX = 10 mm except for one-point deflection. The constant accelerating voltage  $U_{EB} = 80$  kV was used for the experiments and the electron beam current  $I_{EB}$  was subsequently optimized for each machine configuration. The EB modes tested were: one-point (stable

beam without deflection), 6-point, a line (consisting of 1.000 points distributed perpendicular to the “ $v_s$ ”), a field and a meander. Additional processing parameters such as the defocusing degree and the movement rate in each mode and their effect on the quality of the hardened layer were investigated. The common “Offset” values for each mode were 50, 100, 200 and 300 mA and the current sample-to-beam velocities “ $v_s$ ” were 5, 10, 15, 20 and 25 mm·s<sup>-1</sup>.

The field deflection mode was programmed to allow the local energy density to increase within a given area. This is used for intense heating on the surface of a treated material during the hardening. The rest of the area, with a lower beam intensity, contributes to the heating of the material deeper into the sample core. The length of the SWY field was determined for each movement rate “ $v_s$ ” based on the change in temperature across the affected area on the sample measured with a pyrometer.

The meander deflection technique differs from the other ones. The meandering pattern is a combination of the controlled component movement and of the electron beam deflection. The resulting EB trajectory on the specimen surface is shown in Fig. 2.

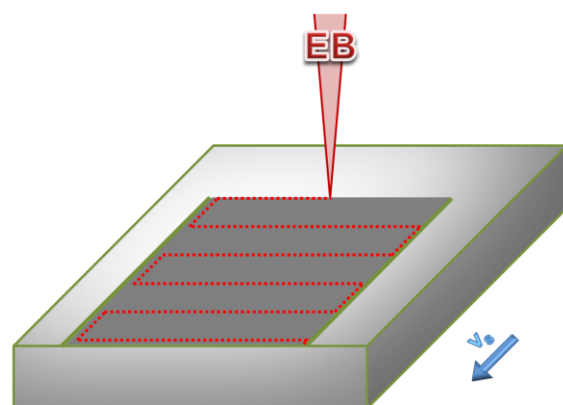


Fig. 2. The EB trajectory (red dotted line) on specimen surface at the meander deflection.  
(full colour version available online)

The metallographic specimens prepared by standard procedures were analysed by optical

and scanning electron microscopy. The LECO LM 247 AT microhardness tester was used to analyse the hardness HV0.5 profile from the surface to the sample core in the quenching trace axis. For the microstructural characterization, the scanning electron microscope (SEM) ULTRA PLUS, Carl Zeiss GmbH, Germany was used. For the surface analysis, the detector of secondary electrons (SE), type Everhart-Thornley, and the four-quadrant silicon detector of back scattered electrons (BSE) were used.

### 3. Results and discussion

Traces having a width of 10 mm were processed by surface quenching on the 42CrMo4 steel. Basic experiments were optimized from the point of view of the used electron beam current  $I_{EB}$ . The optimal energy density conditions were specified by the trial and error method based on the observation of the occurrence of molten areas on the specimen surface. The molten areas were brighter than the quenched ones. A slightly molten surface could not be identified by observing the microstructure because it was also formed by a fine martensitic structure similar to hardened layer. The maximum hardness of the hardened and the partially molten material was the same and therefore it could not be used to determine the optimal EB current for hardening.

The length of the field deflection was determined by pyrometric measurement of the temperature profile within the irradiated area. When too long, "SWY" caused a significant drop in temperature, while if too short, it did not exploit all the potential of the EB technology. "SWY" parameters optimized for an individual tested movement rate are given in Table 1.

From the macroscopic point of view, a constant width of traces was observed in the beam movement direction. A continuous hardening depth decrease to the trace edge was observed in the direction perpendicular to the beam movement (Fig. 3). The microstructure in the surface-hardened area of all the traces

consisted of fine martensite (Fig. 4). The finest martensite was obtained at the one-point deflection and coarsest martensite at the field mode. The meander-deflected martensite looks fine and very similar to the one-point deflection. No significant microstructure difference was observed when applying the 6-point and the line deflection. A continuous change of the fine martensitic structure to the basic material formed by a tempered martensitic structure with carbides was observed in the transition area (Fig. 5).

A comparison of the profiles of hardened layers made by different deflection modes shows that the lower number of deflected points forms a wider track - Fig. 6 (except the one-point mode). This effect is probably caused by an imperfection of the deflecting system. At the same frequency, the time of stay in a position is shorter at a higher point number and it is not possible to reach the physically programmed position by longer deflecting distances. For that reason, the adequate energy is not delivered to the edges and the hardened track is shorter. The track made by the field deflection is the deepest one; the shallowest ones are made by the one-point and the meander deflections. The material is heated for a longer time by the field mode in comparison to the one point resp. the meander modes. For this reason, the heated depth is higher. The one-point and the field deflections have a significant curvature in comparison to the other deflection modes, which are rather parallel to the surface. Different movement rates have a negligible influence on the trace profile. The "Offset" has a significant effect on the shape of the track. With increasing "Offset" value the trace is becoming wider while, on the contrary, very low "Offset" values lead to easier melting as well as to a significant deformation of the trace profile - Fig. 7. The "Offset" expands the electron affected area and changes points into circles with an indefinite radius. Too low value can cause an insufficient energy coverage of the treated areas and lead to a local decrease in the hardened depth.

Table 1

*Optimal length SWY of the field resulted from temperature profile.*

Movement rate	mm·s <sup>-1</sup>	5	10	15	20	25
Field length	mm	5	8	12	18	25



Fig. 3. The macrostructure of the surface hardened area in a perpendicular direction - field deflection. (full colour version available online)

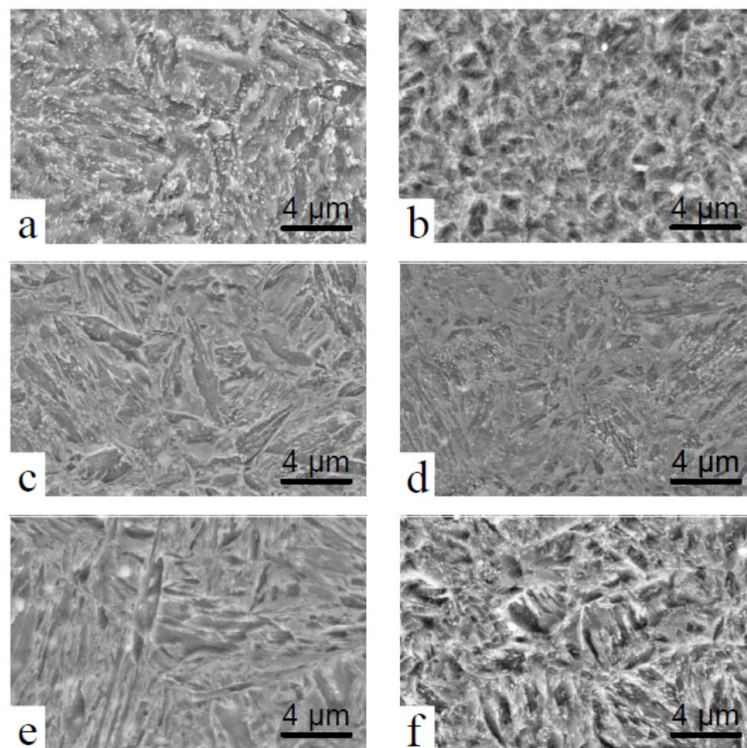
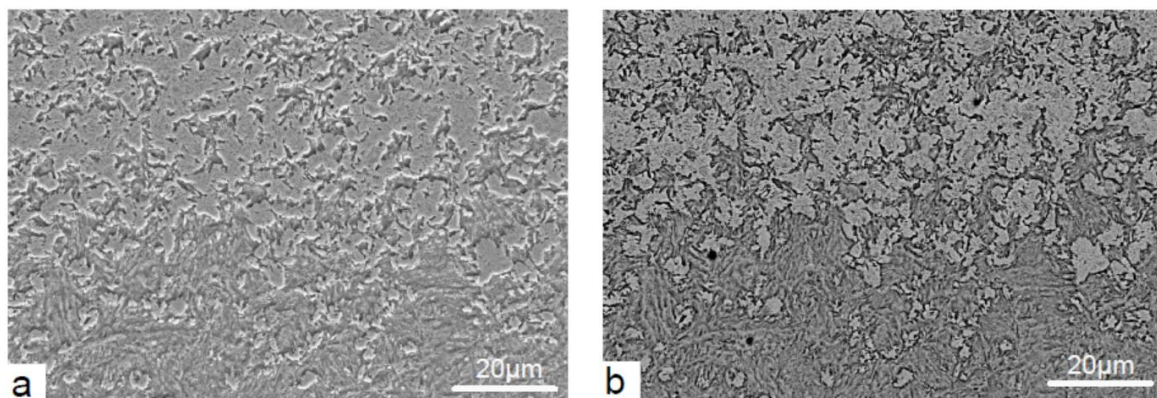


Fig. 4. The microstructure (SE) of (a) basic material, and hardened layers, (b) one-point, (c) 6-point, (d) line, (e) field and (f) meander deflection regimes respectively.



a) SEM – SE mode  
b) SEM – BSE mode  
Fig. 5. The microstructure of the transition area of field deflection sample.

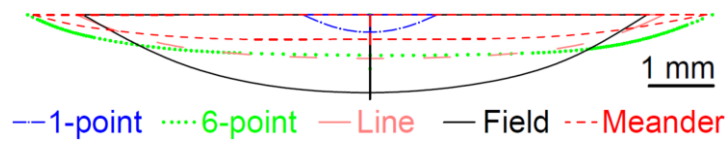


Fig. 6. Figure 6 - Comparison of the profiles of hardened layers. (full colour version available online)

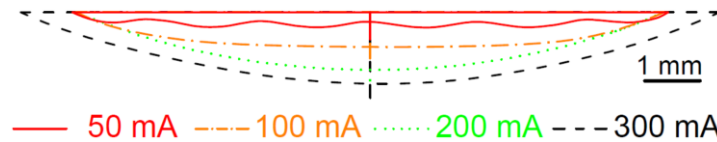


Fig. 7. The influence of Offset on the profiles of the hardened layers (6-point mode;  $v_s = 15 \text{ mm}\cdot\text{s}^{-1}$ ) (full colour version available online)

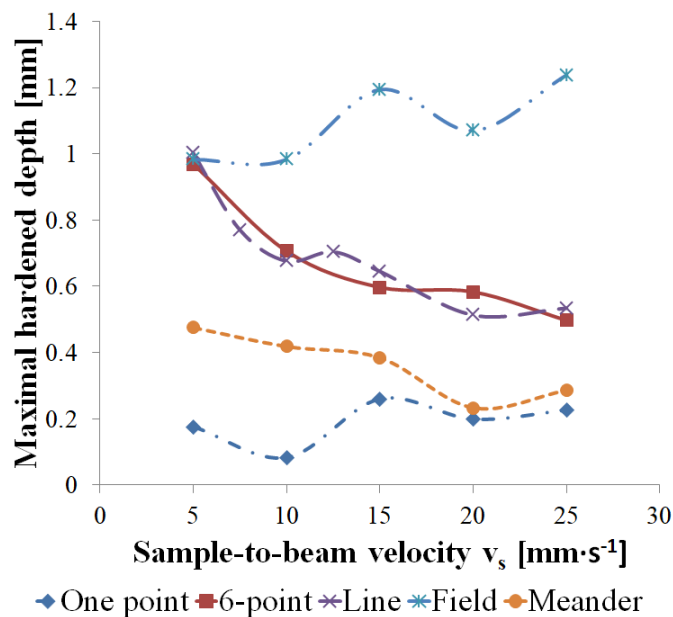


Fig. 8. The influence of sample-to-beam velocity  $v_s$  on maximal hardened depth. (full colour version available online)

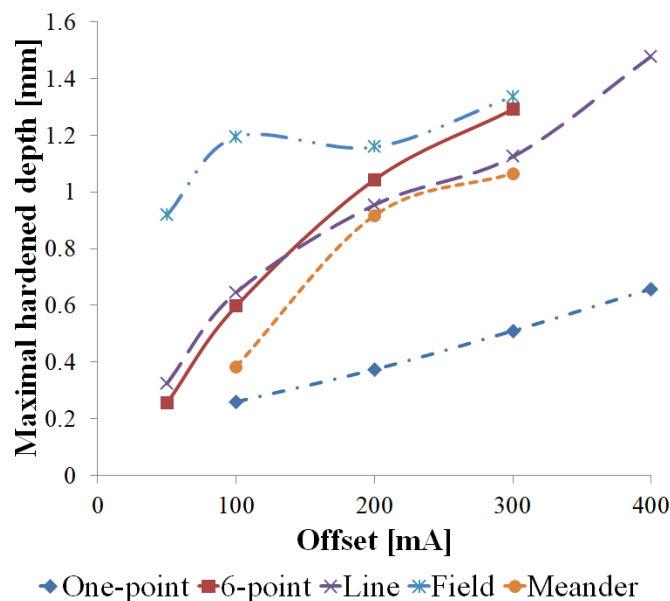


Fig. 9. The dependence of maximal hardened depth on the defocusing Offset. (full colour version available online)

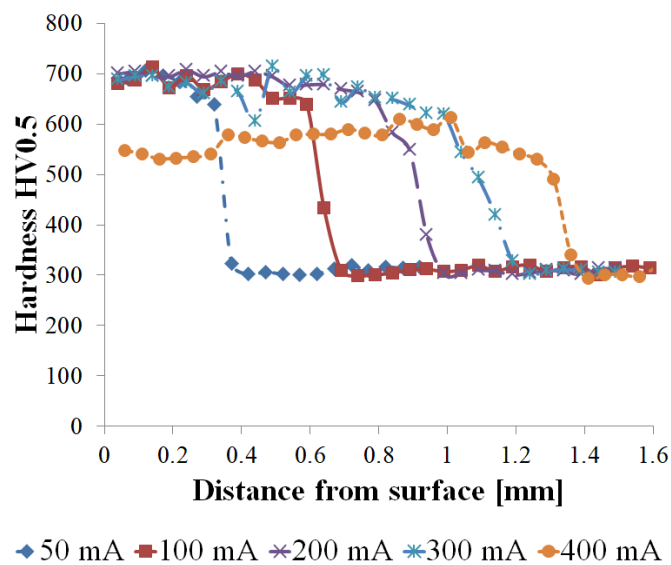


Fig. 10. The influence of the defocusing Offset on the hardness profiles of layers (full colour version available online)

The movement rate “ $v_s$ ” has only a little effect on the depth of the hardened layer, in particular when applying the one-point, the meander and the field deflection (Fig. 8). For the field deflection, it is the result of optimizing the field length SWY. The depth depends significantly on the movement rate for the 6-point and the line deflection and the dependence is nearly identical. The depth gradually increases with decreasing specimen to the electron beam speed and the greatest change can be seen between 5 and 10  $\text{mm}\cdot\text{s}^{-1}$ .

An increasing “Offset” leads to an increase in the hardened layer depth (Fig. 9) because this also extends the length of the scanned area, which means a longer heating time. It is interesting that at higher “Offset” values there is no significant difference between different deflection modes. The one-point and the field deflection differ from the other ones by a poorer response to the Offset change.

Maximal hardness values (up to 740 HV0.5) were reached with the one-point deflection because the heating and especially the cooling processes were very fast. Similar values were reached with the meander deflection. The experiments with the other deflection modes give hardness values between 600 and

700 HV0.5. The measured values decreased from the surface to the sample core. A continuous decrease in microhardness was observed on the interface between the quenched area and the basic material. No decrease in the microhardness of the basic material was observed in the vicinity of the hardened traces (Fig. 10). Hardness profiles were the same in the middle of the track as closer to the edges (except the different hardening depths).

It was not confirmed that the movement rate “ $v_s$ ” affected the final surface hardness. Very high EB defocusing causes a total reduction in the hardness in the entire layer. The Offset value 400 mA resulted in an average hardness of 570 HV0.5 of the hardened track and it represents a 20% decrease in comparison with a sharper beam (Fig. 10). This hardness decrease could be caused by sample dimensions and it is not certain whether this is really a technological limit of the electron beam hardening.

#### 4. Conclusion

The work was focused on a evaluation of the deflection mode (one-point, 6-point, line, field, meander) on the surfaces hardened

by the electron beam. The results showed that the deflection mode can affect a number of track parameters. The martensitic structure is the finest when the one-point and the meander deflection modes are applied. The coarsest structure is generated when applying the field deflection mode. The deflection mode affects the maximum hardness to only a little extent. The highest hardness 740 HV0.5 was observed for the one-point deflection mode. For the other regime modes, the maximum values are near 700 HV0.5.

The geometric profiles in the cross-sections of tracks are different for each of the applied deflection modes. The 6-point, the line and the meander deflection modes are parallel to the surface and the one-point mode together with the field-mode are significantly curved. The widths of the tracks were similar except for the one-point regime. The depths of hardened layers were in the range 0.1-1.5 mm. The lowest depth of tracks was produced by the one-point deflection mode and the deepest one by the field mode.

The sample-to-electron beam movement rate affects only the depth of the hardened layer. The depth slightly increases with the speed decreasing. The defocusing affects the depth more significantly. Moreover, an increasing "Offset" leads to wider tracks. If the "Offset" is too low, it can severely distort the profile of the hardened layer. Very high values, in turn, lead to an overall reduction in the hardness of the layer and eliminate the profile differences between the different types of deflection.

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