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Keith D. Boyer

J. W. Fisher

G. R. Irwin

R. Roberts

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Determination of Tolerable Flaw Sizes in Full Size Welded Bridge Details

FRACTURE ANALYSES OF FULL SIZE BEAMS

WITH WELDED LATERAL ATTACHMENTS

by
Keith D. Boyer
John W. Fisher
George R. Irwin
Richard Roberts
Gopala V. Krishna
Ulrich Morf
Robert E. Slockbower

This work has been carried out as part of an investigation sponsored by the Federal Highway Administration. The contents of this report reflect the views of the authors who are responsible for the facts and the accuracy of the data presented herein. The contents do not necessarily reflect the official views or policies of the Federal Highway Administration. This report does not constitute a standard, specification, or regulation.

Fritz Engineering Laboratory
Lehigh University
Bethlehem, Pennsylvania

April 1976

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1. INTRODUCTION

Recent fractures of steel bridges in the United States, along with the current trend of designing welded details with thick high-strength steel has prompted FHWA to sponsor this project.

Entitled "Determination of Tolerable Flaw Sizes in Full Size Bridge Weldments", the main objective is to correlate actual full size beam fractures with current material characterization tests. From these correlations, simple design guidelines and information are to be developed. Other objectives are to test present fracture toughness specifications and to develop guidelines for in-service bridge inspections.

A welded detail can be considered as a region of material with many small or microscopic flaws. Recent studies have revealed that these microscopic flaws can become macroscopic after repeated application of load. The major factors affecting crack initiation, crack growth and the eventual fatigue life of a welded bridge member are the stress range, the stress concentration, and the initial flaw condition^{1,2}.

The fabrication of a welded detail results in residual stresses. These residual stresses have large tensile components in or near the welds. This, in combination with the complex stress concentration and macroscopic fatigue flaws, can make welded details susceptible to rapid fracture. This is especially true of those details fabricated with thick high-strength steel.

This project consists of three parts. The first is the fatigue and fracture testing of 24 full size welded beam specimens with details which are commonly used in bridge design. The details were chosen from the AASHTO categories for fatigue design¹⁸. Two Category E details were chosen: the coverplate and the lateral attachment. The intermediate Category C detail was the transverse stiffener. The flange thickness transition provided the upper bound fatigue strength detail (Category B). Six beams were fabricated for each of the four detail categories. Each detail type was fabricated in three types of steel. A list of the details is shown in Table 1.

The second part of the study was a detailed material characterization. Materials from which these beams were fabricated were evaluated using several fracture toughness tests.

The third part is an analytical treatment of crack shapes which may be encountered during the beam tests. This has been completed in a report by Irwin and Tada³. The results of this study were used to estimate the critical stress intensity factor for the fractured beams.

This report contains the results and discussion of the six beam tests with lateral attachment details and a summary of part of the material characteristics. Also included is a description of the tests and testing procedures.

2. DESCRIPTION OF TESTS

2.1 Test Specimens

The six welded beam specimens were fabricated by the

Bethlehem Steel Corporation at their Bridge Division Fabrication

Plant in Pottstown, Pennsylvania. All specimens were fabricated using current fabrication and inspection techniques.

Each thickness of material was furnished from the same heat for each of the three types of steel. Chemical composition, as defined by the mill reports, is shown in Table 2. As beam components were flame cut from the larger rolled plates, a cutting schedule was maintained. Material testing samples were later cut from the same plate.

After the beam components were cut to size, the edges of the web plate were blast cleaned. The web and flange components were then assembled in a beam welder and the web to flange longitudinal fillet welds were then made by an automatic submerged-arc process. These welds were kept continuous. Any visible flaw such as excessive porosity was gouged out and rewelded.

The lateral attachment plates were connected after the cross section was completed. The groove weld lateral attachment plates were welded by a semi-automatic submerged arc process. The run-out tabs were then ground to an approximate radius of 0.75 in. (19.1 mm). The transverse fillet welds at the overlapped lateral attachment plate were made manually.

For each type of steel, ASTM A36, A588 Gr50, and A514, two beams were fabricated. A detailed drawing of beam specimen B4 is shown in Fig. 1. The measured beam dimensions are summarized in Table 3. Note that beams B2 and B2A have smaller flange dimensions which were necessary to satisfy the jack capacity.

2.2 Test Setup

All beam testing was done on the dynamic test bed in Fritz Engineering Laboratory, Lehigh University. The test span length was 21 feet (6.40 m). Two 110 kip (489.5 kN) Amsler jacks driven by a single pulsator were used for the 260 cpm (4.3 Hz) cyclic load. When needed to raise the level of maximum stress, a constant load jack was also used.

The latter jack was a 200 kip (890 kN) Parker-Hannifin jack loaded with an Amsler accumulator and maintained by a column of nitrogen. A schematic of the loading setup and geometry is shown in Fig. 2. Photographs of the setup are shown in Figs. 3 and 5.

2.3 Instrumentation

SR-4 strain gages were used extensively to control the strain during the fatigue and fracture tests. Also, electrical resistance temperature gages were used to monitor the beam's temperature.

Four electrical resistance strain gages were mounted on the tension flange and used as strain control when determining the beam

deflections and loads. Two gages on the compression flange were used as a lateral buckling indicator. Since the strain gages were mounted close to the section to be cooled, temperature compensation plates were used to counteract thermal effects. The position of these gages is shown in Fig. 4.

Initially, temperature gages were mounted directly on the steel beam at the critical section. After two fracture tests, it was found that the same surface temperature readings could be obtained by attaching the gages to steel plates, 1/16 in. x 1-1/2 in. x 1-1/2 in. (1.6 mm x 38.1 mm x 38.1 mm) and clamping these plates to the critical section of the beam. This procedure was very economical, since one gage could be reused for several tests. Usually three to five temperature gages were used on one beam section during a fracture test. The position of these gages is also shown in Fig. 4.

To eliminate air temperature effects, the outer surface of the plates was covered with a 1/2 in. (12.7 mm) thick styrofoam insulation. The gages were positioned to avoid direct liquid nitrogen contact to assure accurate surface temperature.

2.4 Cooling Apparatus and Enclosure

Each beam was cooled from room temperature to a desired temperature with liquid nitrogen. The section or sections of the beam to be cooled were completely enclosed in a styrofoam box. The boxes were made relatively leak-proof by the use of sealing compound and duct tape. Inside each box was a copper tubing network which sprayed the top and both sides of the beam with liquid nitrogen.

Since cold gaseous nitrogen is heavy, the cold gas had a tendency to settle to the bottom of the cooling box. Without convective flow, this would cause a sharp temperature gradient across the beam section. Therefore, the inlet for the nitrogen was placed at the top of the beam. Connected to this inlet was a pressurized dewar of liquid nitrogen. By regulating the pressure within this container, the temperature in the box could be controlled.

An attempt was made to achieve uniform temperature throughout the beam cross-section. Since most of the nitrogen still in its
liquid state remained in a tray at the bottom of the box, trays were
also placed in the upper section of the box. This device made temperatures noticeably more uniform across the section being cooled. A
sketch and photographs are shown in Figs. 2 and 5.

2.5 Design Stresses

In accordance with the 1974 Interim Specifications, the lateral attachment details were classified as Category E. The allowable stress range for this type of detail for two million design cycles is 8 ksi (55.2 MPa).

Each beam had two different lateral attachment details as illustrated in Fig. 1. One was an overlapped, 12 in. (305 mm) long attachment with transverse fillet welds on the inside of the tension

flange, and a longitudinal fillet weld along the beam flange-tip. The other was a 12 in. (305 mm) long, groove weld attachment welded to the flange-tip. The 1 in. (25.4 mm) thick plate was flush with the outer surface of the flanges. The groove welded attachment had a sharp radius of about 0.75 in. (19.1 mm) where the reinforcement was removed by grinding at the weld ends.

The maximum stress was governed by the outermost fiber of the tension flange. The stress range was set on the inside of the tension flange. This yielded a nominal applied maximum stress and stress range at the overlapped fillet weld detail of (0.889) x (0.55 $\sigma_{\rm Y}$) and 8 ksi (56.2 MPa) respectively. At the groove weld detail the maximum stress and stress range were 0.55 $\sigma_{\rm Y}$ and 9 ksi (62.1 MPa). These values were slightly different for beams B2 and B2A. Actual values are shown in a schematic for each steel type in Figs. 6a through 6c.

2.6 Load and Deflection Control

Deflection control was used during the fatigue testing at room temperature. The desired stresses were obtained by averaging the four strain gages mounted on the tension flange. For each stress, deflections were obtained from a pair of deflection gages placed on either surface of the tension flange. When the maximum and minimum stresses were set, an appropriate set of deflections was obtained. The beam was then loaded cyclically between these deflections. Therefore, load adjustments for inertia forces were not required. A tolerance of ±0.003 in. (0.8 mm) deflection was maintained.

The fracture test loading could not be deflection controlled since any small temperature gradient across the beam section may have caused misleading deflections. Therefore, the dynamic loads were noted during the fatigue testing and these loads were then used to control loading during the fracture tests. Dynamic stress measurements confirmed the adequacy of the procedure.

2.7 General Testing Procedure

The first beam tested, B4A, served as a pilot study. Initially 1.5 million cycles of load were applied at a stress range of 8 ksi (55.2 MPa) at the fillet weld detail and 9 ksi (62.1 MPa) at the groove weld detail. At this point the beam section containing the largest fatigue cracks was tested at -40° F (-40° C) for one-half hour. No fracture occurred and the beam was fatigue cycled for an additional 250,000 cycles, at which time another -40° F (-40° C) test was run. This fatigue and fracture test sequence was repeated until a fracture occurred.

Failure did not occur when the fatigue cracks were small and still in the stress concentration area. The fatigue cracks destroyed about 70% of the tension flange area before fracture occurred. This extended fatigue and fracture sequence took considerable time to complete as altogether eight test sequences were carried out. For these reasons the test procedure was modified on subsequent tests as follows.

Each subsequent beam was cyclically loaded for two million cycles or until the fatigue cracks became a possible critical size,

whichever occurred first. At this point each section of the beam containing the details was cooled to -40° F (-40° C). The beam was then cycled for at least one-half hour between a maximum stress of 0.55 $\sigma_{\rm Y}$ and a minimum stress of 0.55 $\sigma_{\rm Y}$ - $\sigma_{\rm r}$. If no visible fatigue cracks existed after two million cycles the fracture test was discontinued and further fatigue cycles applied at room temperature.

If there was a possible critical fatigue crack at the beginning of the first fracture test and no fracture occurred in the first one-half hour, either an extended test at -40° F (-40° C) was run or the temperature was dropped below -40° F (-40° C). This temperature drop was done slowly to obtain accurate surface temperature readings. This extended test was continued until fracture or until the liquid nitrogen supply was depleted. If there was no fracture, the beam was again fatigue cycled at room temperature to increase the crack size.

The next low temperature test was run on the detail with the largest fatigue crack after the crack had grown a predetermined amount. This fatigue and fracture test sequence was continued until a fracture occurred.

2.8 Fatigue Testing

The stress range used in the fatigue test was in accordance with the 1974 AASHTO allowable range of stress for two million cycles at the fillet welded attachment for a Category E detail. An allowable stress range of 8 ksi (55.2 MPa) is permitted for a Category E detail.

It was initially intended to fatigue cycle between the same minimum and maximum stress limits as in the fracture tests. However, this was discontinued after three tests for several reasons. First, operating the constant load jack under cyclic deflection for such extended periods caused excessive wear and heating which caused damage to the hydraulic ram. In addition, it appeared that fatigue cracking at room temperature at the limit of allowable stress could cause effects known as "warm prestressing" but in a greater apparent fracture resistant condition. The earlier studies by Fisher, et al. 1,2 have demonstrated that the level of maximum stress has no appreciable affect on fatigue. Hence, in subsequent tests, the cyclic stress range was applied at a lower level of maximum stress.

During the fatigue test period, frequent checks were made for visible fatigue cracks. Mainly, visual inspections were made with a 10X magnifying glass and a cleaner fluid. At times a magnetic particle probe was also used. Since the cycling was continued twenty-four hours a day, some of the cracks were 1 in. (25.4 mm) corner cracks before they were discovered.

An automatic shut-off switch was used to prevent extremely large edge cracks from occurring before the scheduled fracture tests. The switch was usually set for a 0.005 in. (.13 mm) deflection increase.

2.9 Fracture Testing

During the pilot study, the beam was tested at low temperatures after an initial 1.5 million cycles of loading. In subsequent tests, the initial fracture test was run after accumulating two million cycles of cyclic load, as it was apparent that no brittle fracture would occur at this stage of testing as the fatigue cracks were small.

In preparation for the fracture test, the moveable temperature gage plates were clamped to the beam at various points around both beam sections to be cooled as shown in Fig. 4. The gages used for test control were placed at the crack planes on the exterior surface of the tension flange. Actual temperature gage placement is noted in Table 4.

The cooling apparatus was then put in place and the styrofoam boxes were sealed. Most leakage was stopped during the initial cooling period. The temperature was monitored constantly and recorded every five minutes. When the temperature at the test control gages reached -40° F (-40° C), the liquid nitrogen flow was regulated to maintain the test temperature.

During the first fracture test, both beam sections containing the welded details were cooled simultaneously. By regulating the liquid nitrogen flow, the temperature in each box was kept relatively close, $\pm 5^{\circ}$ F ($\pm 2.8^{\circ}$ C).

When the temperatures at the critical details became stable, cyclic loads were applied. Prior to applying the maximum allowable

stress of 0.55 $\sigma_{_{\!Y}}$ and the full design stress range level, the crack tip was marked by applying cyclic stresses between the limits of 0.55 $\sigma_{_{\!Y}}$ - $\sigma_{_{\!r}}$ and 0.55 $\sigma_{_{\!Y}}$ - $\sigma_{_{\!r}}/2$. This cyclically applied stress was continued for approximately thirty minutes, after which the full stress range was applied to the maximum nominal stress of 0.55 $\sigma_{_{\!Y}}$. In most cases, the initial set of dynamic loads yielded a minimum stress of 0.55 $\sigma_{_{\!Y}}$ - $\sigma_{_{\!r}}$ and a maximum stress of 0.55 $\sigma_{_{\!Y}}$. A load history for each beam is shown in Tables 5 through 10.

During each low temperature test, one of the tension flange strain gages was monitored on a memory oscilloscope. This trace showed both the sinusoidal loading rate and the fracture point. Since the triggering at failure was manual, only one trace was obtained at fracture and is shown in Fig. 7.

A sinusoidal loading rate of 260 cpm (4.3 Hz) was provided by the Amsler pulsator. This resulted in a loading rate of about 0.12 sec. from the minimum stress to maximum stress level. The sinusoidal nature of the cyclic load yielded a maximum loading rate of 100 ksi/sec. (690 MPa/sec.). As can be seen in Fig. 7 the fracture occurred at a point approximately 95% of the maximum load. This was typical of subsequent tests as well. However, the nominal maximum load will be used for the fracture analysis.

3. MATERIALS CHARACTERIZATION

3.1 Test Plan

For the purposes of material characterization Standard Charpy V-Notch (CVN) and Dynamic and Static Fracture Toughness (K_C) tests were carried out on each plate thickness. Mill test data for each plate was also available. Initially it was desirable to determine the fracture toughness of the flange plates {2 in. (51 mm) - A36 steel; 2 in. (51 mm) - A588 steel; and 1-1/2 in. (38 mm) - A514 steel}. The chemical composition and mill test data are summarized in Tables 2a, b and c. These plates were used to fabricate the test beams described in this report. In this section, a brief description of the experimental procedure and the test results are presented.

3.2 Charpy V-Notch Impact Tests

In order to determine the macroscopic brittle-ductile transition behavior of the plate materials, conventional ASTM standard A370-68 Type A Charpy V-Notch specimens were prepared from each of the three plates. The specimens were all transverse (LT) with notch direction perpendicular to the rolling direction. The impact test data was analyzed using a least squares best fit sigmoidal computer program developed at Lehigh University.

3.3 Fracture Toughness Measurements

The Charpy V-Notch data was used to select a test temperature range so that valid fracture toughness data could be acquired for the

plates. Temperatures of 0° F (18° C), -40° F (-40° C) and -80° F (-62° C) were chosen for dynamic testing. A lower temperature range based on the transition temperature shift¹¹ was selected for the slow bend (intermediate loading rate) tests. Equation 1 was used to estimate where additional tests were conducted at other temperatures

$$T_{\text{shift}} = 215 - 1.5 \sigma_{\text{Y}}$$
 (1)

T_{shift} = transition temperature shift (° F)

 σ_{v} = room temperature static yield stress (ksi)

3.3.1 Drop Weight Test Apparatus

The dynamic K_C testing was carried out using the Lehigh drop weight test machine (see Fig. 8). The details of this apparatus are described in Ref. 6. The impact loading of the three-point bend specimen (Fig. 9) was achieved by means of a falling mass (400 lbs.) guided vertically along two parallel rails. An instrumented loading tup⁶ at the bottom of the mass was calibrated to act as a load-dynamometer. As the specimen was loaded the strain output from the tup was recorded. A typical load-time relationship is shown in Fig. 10. The drop weight mass in a given set of tests was chosen to minimize the test specimen inertia. In order to minimize the influence of the specimen inertia, 3/4 in. x 1/2 in. (19.1 mm x 12.7 mm) half-rounds were positioned on the test specimen. This cushioned the application of the load and increased the loading time to about one millisecond. The half round cushions were machined from unhardened drill rods. The test specimen

temperatures were controlled by a variety of means. All were held at the required test temperature for at least ten minutes prior to testing. A test was completed within ten seconds of the specimen's removal from the temperature bath.

3.3.2 Slow Bend Test Apparatus

Slow bend tests* were carried out on a standard 120 kip
Tinius-Olsen screw-type tensile testing machine. The cross head of
the machine could be moved at various speeds. The specimen was loaded
with the same tup used for the dynamic testing. A loading rate of
20 kips per second was selected for all slow bend tests. This resulted in a loading time of about 1 second. Load-time data was
recorded on x-y recorders. Fracture tests of the customary "static"
type, with a loading time to fracture of several minutes, were not
conducted.

3.3.3 K_c Specimen Preparation

The test specimen geometry for all $K_{\rm C}$ tests in this program is shown in Fig. 9. All specimens were saw cut from the original plate with their long dimension in the rolling direction. This resulted in the crack being perpendicular to the rolling direction.

^{*} Tests in which the fracture load occurs about 1 second after the start of loading are not "slow" in the customary usage of the term. Such tests are sometimes termed "intermediate speed" tests. However, for simplicity of language in this report, the 1 second loading time tests will be termed "slow bend".

After the individual specimens were saw cut from the plates the cut surfaces were shaped so as to be normal to the plate surfaces. The thickness of the A36 and A588 steel specimens was reduced to 1-1/2 in. (38 mm). A notch with a 45° chevron front was machined at the center of the specimens to help initiate crack growth during the precracking process. The cyclic-loading for precracking was done on a 10 ton Amsler Vibrafore using three-point bending. The fatigue crack was formed in two stages. During the first stage, the crack was grown as quickly as possible. The final 1/8 in. (3 mm) of the crack was grown slowly so that the average crack growth rate was equal or less than 1 microinch per cycle (25.4 nm per cycle). The maximum K during fatigue precracking was about 40 ksi $\sqrt{\text{in.}}$ (44 MPa $\sqrt{\text{m}}$).

3.3.4 Fracture Toughness Data Evaluation

The fracture toughness, $K_{_{\hbox{\scriptsize C}}}$, values were determined from the maximum load at the fracture of the three-point bend specimens 7 . K was determined from the relationship

$$Y = \frac{K B W^{2}}{1.5 \text{ PL } \sqrt{a'}} = 1.93 - 3.12 \left(\frac{a'}{W}\right) + 14.68 \left(\frac{a'}{W}\right)^{2} - 25.3 \left(\frac{a'}{W}\right)^{3} + 25.9 \left(\frac{a'}{W}\right)^{4}$$
(2)

where Y = dimensionless ratio

B = specimen width

W = specimen depth (3.0 in.)

P = applied load

L = span length (10.0 in.)

a = effective crack length

 r_{v} = plastic-zone size

a' = a +

The plastic-zone size, r_{γ} , was defined as

$$r_{y} = \frac{1}{2\pi} \left(\frac{K}{\sigma_{y}} \right)^{2} \tag{3}$$

where $\sigma_{_{\mathbf{Y}}}$ is yield stress

Equations 2 and 3 were solved by a simple iteration method 7 . The value of $\sigma_{\underline{Y}}$ corresponded to the temperature and loading speed of the test conditions. This was determined by the following equation 8 .

$$\sigma_{\text{Yd}} = \sigma_{\text{Y}}$$
 + $\frac{174,000}{(\text{T} + 459) \log (2 \times 10^{10} \text{t})} - 27.4$ (4)

where t = loading time to maximum load

 t_0 = time of load application for a static test (50 sec.)

T = testing temperature (°F)

 $\sigma_{\mathbf{v}}$ = yield stress (ksi)

 $\sigma_{Yd}^{}$ = elevated yield stress (ksi) at test conditions

3.4 Drop Tear Energy Measurements

A method of direct measurement of fracture energy was described in Ref. 6. After the specimen is fractured the drop weight is arrested by two cushions made from Type 1100-0 or 6061-0 electrical grade aluminum 1 in. (25.4 mm) diameter rods. Figure 8 shows the test setup. When the drop weight impacts the aluminum blocks, they are compressed inelastically and their difference in height is a measure of the energy absorbed. In addition, the drill rod cushions are subjected to permanent diamond shaped indentation during loading of the specimen. The length of the identation is also a function of the energy absorbed.

The initial potential energy in the system less the sum of the energies absorbed by the aluminum and drill rod cushions represents the net energy absorbed by the fractured specimen. This value divided by the fracture surface area yields the drop tear energy (DTE). Material behavior in terms of DTE as a function of temperature is obtained simultaneously with the K tests.

3.5 Results of Fracture Tests

3.5.1 Charpy V-Notch Tests

Figures 11 through 13 summarize the CVN test results in the form of standard Charpy V-Notch curves. For the three materials the energy absorption and the lateral expansion data, plotted against temperature, show a conventional form with relatively sharp transition behavior. The 15 ft.-1b. (20 joule) energy level and the 15 mil. (0.38 mm) lateral expansion transition temperatures are listed in Table 11 for each flange plate.

3.5.2 K_c Test Results

The dynamic and static fracture toughness for the flange plates are summarized in Figs. 14a, b, and c. Also shown is the limiting test validity requirement¹⁰.

$$B > 2.5 \left(\frac{K_c}{\sigma_Y}\right)^2 \tag{5}$$

where B = specimen thickness

K_c = fracture toughness value

 $\boldsymbol{\sigma}_{\boldsymbol{y}}$ = yield stress of the material at test conditions

In some cases, computed K_c values were obtained which did not satisfy the above ASTM thickness requirement. The trend curves for the limited test data were based on earlier results. Although from these curves it was possible to indicate the brittle-ductile transition temperatures, it appears that another independent method to evaluate fracture toughness values at these temperatures will be needed. The J-integral type tests with three-point bend specimens might provide the required data points to confirm the fracture behavior in the transition temperature range.

Barsom's temperature shift relationship (see Eq. 1) was used to determine the expected temperature shift caused by the change in loading rates between dynamic and static tests. These values are listed below for each steel.

	Temperature	Shift
	°F	<u>°C</u>
A36	149	65
A588	124	51
A514	32	.0

The actual temperature shifts are shown in the $K_{\rm C}$ vs. temperature plots (Figs. 14a, b, c) for the dynamic and intermediate loading rate tests used in this project. These actual values were in each case, larger than the shifts predicted by Barsom. Hence Eq. 1 is conservative.

The CVN and dynamic $K_{\mbox{\footnotesize{IC}}}$ results were compared by using the relationship proposed by \mbox{Barsom}^{11} for the transition temperature region of the CVN plots

$$K_{Id} = [5E (CVN)]^{\frac{1}{2}}$$
 (6)

E = modulus of elasticity (psi)

 K_{Id} = fracture toughness (psi $\sqrt{in.}$)

CVN = Charpy energy (ft.-1bs.)

These values are also plotted on the $K_{\rm c}$ vs. temperature plots in Figs. 14a, b, and c. There is a good correlation between the measured $K_{\rm Id}$ values and the plot given by Eq. 6 for A36 steel. However, the correlation is not as good for the A588 and A514 plates. Very conservative results were obtained for the A514 steel. Several unconservative points were obtained for the A588 steel.

3.5.3 Drop Tear Energy Test Results

The DTE data points were obtained simultaneously with the K_{Id} test data. A full DTE vs. temperature plot was not obtained. Most of the points were on the lower shelf or in the transition region. The DTE vs. temperature plots are presented in Figs. 15a, b, and c. Generally, the transition temperatures from these diagrams are higher and more conservative than the respective CVN transition temperature for the same plate.

4. BEAM TEST RESULTS AND ANALYSIS

4.1 Fatigue Cracks

The fatigue cracks at the groove weld lateral attachments were initially detected on the flange edge, at the sharp 0.75 in.

(19.1 mm) or less radius, as 1/4 in. (6 mm) elliptical surface cracks. These surface cracks soon became elliptical corner cracks and then edge cracks. All final fractures at this detail were precipitated from an edge crack.

On the overlapped fillet weld detail, fatigue cracks were initiated at the toe of the transverse fillet weld. Most of these cracks were initially detected as several 0.5 in. (13 mm) elliptical surface cracks which eventually connected to form one large elliptical surface crack. As with the groove weld detail, these cracks then became corner cracks and finally edge cracks. Beam B6 was the only specimen to fracture from this detail.

The size of the fatigue cracks at each critical detail can be found by referencing the small letters on the fracture surface drawings in Figs. 16-21 with the load history tables given in Tables 5-10.

Many additional fatigue cracks existed at other details on the beams. Figures 22 and 23 show these fatigue cracks at all details at two million cycles and prior to the last fracture test. The surface measurements of these cracks is shown adjacent to the crack. The crack shapes are merely estimates from these surface measurements.

4.2 Remaining Fatigue Life

The number of cyclic loads needed to propagate an edge crack from its fracture initiation point to an edge crack size of 75% of the flange width, b, was defined as the remaining useful fatigue life had brittle fracture not occurred. The following crack growth relationship determined from earlier studies on welded details was used 12.

$$\frac{da}{dN} = 2 \times 10^{-10} \Delta K^3$$
 (7)

As defined in Section 4.5.2 the stress intensity range can be found from the following relationship

$$\Delta K = \sqrt{\frac{2b}{\pi a'} \tan \frac{\pi a'}{2b}} \Delta \sigma \sqrt{\pi a'}$$
 (8)

Secondary stress intensity effects from residual stresses were neglected for this analysis. Also by this stage of growth the crack had grown out of the stress concentration zone. Through numerical integration of Eq. 8 the remaining fatigue life was estimated. The results for each beam are listed in Table 12.

Figure 24 shows the mean S-N curve and its confidence limits for Category E details. The data base used to develop this curve utilized tests on 12 to 14 inch (35 to 36 mm) deep beams with a maximum flange thickness of 1/2 in. (12.7 mm). The fatigue results for the lateral attachment beams, which had a maximum flange thickness of 2 in. (51 mm) are plotted on the same curve. The open figures represent the point at which the fatigue cracks were first observed and the

closed figures represent the point of fracture. There is a good correlation between the fracture points and the Category E fatigue-life relationship.

As can also be seen from Fig. 24 and the additional life estimated and tabulated in Table 12, an incremental addition to the fatigue life was small and would not have significantly altered the strength as all the points were well within the 95% confidence limits. Hence even if rapid fracture had not occurred very little residual life would have remained. Fatigue resistance design is therefore a major objective of any fracture control plan in the design of bridge girders.

4.3 Beam Fracture Tests

Beam B4A

Eight fracture tests were carried out on Beam B4A as the test procedure was developed. Three of these tests were on the over-lapped fillet weld detail while five were on the groove weld detail.

The first five fracture tests were run with fatigue cracks still in the stress concentration zone. After 1.5 million cycles the largest fatigue crack found was a 1 in. x 1/16 in. (25.4 mm x 1.6 mm) elliptical corner crack (see Fig. 22) at a transverse fillet weld. The first two fracture tests were on this detail. At two million cycles, a 3/8 in. x 1 in. (9.5 mm x 25.4 mm) elliptical corner crack was observed at a groove weld detail. The fracture tests were carried out at test temperatures between -40° F (-40° C) and -60° F (-51° C)

as can be seen in Table 8. No crack instability developed during any of these three tests.

A test was run on the fillet weld detail where a 1-7/8 in. x 9/16 in. (48 mm x 14 mm) elliptical corner crack existed. No fracture occurred there as well. With a 1-1/2 in. x 1-3/4 in. (38 mm x 44 mm) corner crack at the groove weld detail (test h) the next test reached a temperature of -170° F (112° C), however, no fracture occurred.

The cracks were extended by applying 250,000 cycles of fatigue loading at room temperature. The critical fatigue crack at the groove weld detail was grown to a $\sim 2-3/4$ in. (70 mm) edge crack during this cyclic loading. At this point a -70° F (-56.5° C) fracture test was run. The test lasted 2.67 hours. During this test, the fatigue crack grew very rapidly through the high tensile residual stress region of the web to flange fillet welds. Finally, the beam fractured with an average edge crack size of 4.8 in. (122 mm) and temperature of -96° F (-71° C). Fatigue crack extension of approximately 2 in. (51 mm) was experienced during this test prior to crack instability.

Beam B4

It was apparent from experience with Beam B4A that rapid fracture was not likely to occur at -40° F (-40° C) with small cracks in the stress concentration zone. Therefore, the beam was cycled at room temperature for two million cycles. At this point several large elliptical corner cracks existed as shown in Figs. 18 and 22. The

first fracture test lasted for one-half hour and both details were tested simultaneously. No fracture occurred.

The beam was then cycled at room temperature to extend the fatigue cracks. When the crack at the critical detail became a $\sim 2\text{-}3/8$ in. (60 mm) edge crack, a second fracture test was run. A temperature of -70° F (-56.5° C) was obtained before the cyclic load was applied. A stress range of 9 ksi (62.1 MPa) was applied for forty minutes. To speed the incipent fracture, the load range was increased to 9.8 ksi (67.6 MPa) while maintaining the same maximum stress. After one hour at this stress range and a nominal temperature of -70° F (-56.5° C) fracture occurred. At fracture, the temperature was -80° F (-62.0° C). A $\sim 3/4$ in. (19 mm) fatigue crack extension was experienced during this test. The fracture occurred when the crack tip was in the high tensile residual stress zone of the web to flange weld.

Beam B6

The first fracture test was run on both details simultaneously after two million fatigue cycles. Since very small fatigue cracks existed (see Fig. 22) no fracture occurred. After 800,000 cycles of additional fatigue load the elliptical surface crack at the critical fillet weld detail grew to a large 2-3/8 in. x 1-1/2 in. (60 mm x 38 mm) elliptical corner crack. At this point two consecutive five hour fracture tests were run (test d and e, see Fig. 20) on this detail. Fracture occurred after the elliptical fatigue crack became an edge crack. The fracture temperature was -53° F (-47.0° C). This was the only fracture to occur at a fillet weld detail.

During the fatigue cycling of this beam, the ram in the constant load jack overheated. This caused the maximum load to decrease during the fatigue cycling overnight. Although the maximum load decreased, the stress range remained the same. The actual drop in maximum stress was 4.5 ksi (31.03 MPa) for 400,000 cycles.

Beam B2A

Five fracture tests were run on this beam (see Fig. 17).

The first test at two million cycles was on both details. Both details contained large corner cracks at this point (see Fig. 22), however no fracture occurred at 40° F (-40° C). Since the elliptical corner crack at the groove weld detail grew quickly to a critical edge crack, the remainder of the fracture tests were conducted on this detail alone. During the last test, the temperature was maintained at -40° (-40° C) for 1½ hours. While the beam was still being cyclically loaded, the temperature was slowly dropped to -140° F (-95.5° C) in over 1½ hours. The -140° F (-95.5° C) temperature was maintained for another 1½ hours before fracture occurred at -144° F (-98° C).

About 1/4 in. (6.4 mm) fatigue crack extension was experienced during the test prior to crack instability.

Note that the beam was fatigue cycled at a lower maximum stress than that during the fracture test. The same stress range was maintained during both fatigue and fracture testing. See Table 6 for the actual stresses and stress ranges used.

Beam B6A

The first fracture test was run on both details (see Fig. 22) at -40° F (-40° C). No fracture occurred. After an additional 730,000 cycles of fatigue load at room temperature, a corner crack at the groove weld detail became a $\sim 1-1/4$ in. (32 mm) edge crack. The subsequent fracture test lasted 1.67 hours during which the temperature was slowly dropped from -40° F (-40° C) to -92° F (-69° C) at which point rapid fracture occurred. An average fatigue crack extension of 1/4 in. (6.4 mm) (see test d, Fig. 21) was experienced prior to fracture.

Beam B2

At two million cycles, a 1 in. (25 mm) edge crack existed at the groove weld detail while smaller elliptical corner cracks existed at the fillet weld detail (see Fig. 22). Both details were tested for forty minutes at -40° F (-40° C). At this time the cyclic load was stopped and the groove weld detail was cooled to -140° F (-95.5° C). After this temperature was obtained, the cyclic load was reapplied. After twenty minutes of cycling, fracture occurred at a temperature of -155° F (-104° C). A 1/4 in. (6.4 mm) fatigue crack extension was experienced during the last test (see test b, Fig. 16).

The beam was fatigue cycled at a lower maximum stress than that during the fracture test. The same stress range was maintained during both fatigue and fracture testing. See Table 5 for the actual stresses and stress ranges used.

4.4 Fracture Test Variables Affecting Fracture Toughness

Each fracture test had two major variables affecting the fracture resistance of the steel beam. These were the fatigue crack size and test temperature.

Since no beam fractured on the first cycle of load an effort was made to induce rapid fracture at -40° F (-40° C) by growing the fatigue crack to a critical size. As noted in Section 4.3, Beams B4, B4A, and B6 experienced average fatigue crack extensions of 0.65 in. (17 mm), 2.0 in. (51 mm), and 1.3 in. (33 mm), respectively, prior to brittle fracture. These large crack extensions took several hours to achieve.

Since time was a limiting factor, the test temperature was used as another variable. The slow cooling rate of approximately 1° F (.6° C) per minute was used. Temperature at the critical details are shown graphically in Figs. 25 to 27 for the final 60 minutes of the last fracture test. In every case the temperature was slowly decreasing when fracture occurred.

Although large temperature gradients existed around the critical beam section, as shown in Table 4, an effort was made to keep accurate account of the surface temperature at the critical welded detail. The temperature gages were positioned at the critical detail on the exterior of the tension flange, thus being out of direct contact with the liquid nitrogen.

4.5 Stress Intensity Estimates

4.5.1 Introduction

All the flange cracks in the lateral attachment details were large edge cracks at fracture. This tended to simplify the calculations of the stress intensity factor. However, since the plates were flame cut and the beams and details were welded a rather complex residual stress pattern was present at the detail cross-section. Therefore several steps were used to estimate the value of the stress intensity factor, K.

By the method of superposition the following contributions were used to determine the magnitude of K. The primary contribution was from the applied stresses at failure. A secondary contribution was from the residual stresses at the detail cross-section. The residual stresses at the cracked section resulted from two contributions. One contribution to K was from the residual stresses at a typical cross-section of the welded beam. These stresses were caused by the web-to-flange welds and the flame cut plate edges. The other contribution was due to the residual stresses caused by the local detail welds. In this draft these residual stresses were estimated from available information.

In one case, the flange edge crack grew through the web-to-flange welds. The fatigue crack growth continued in two directions, upward into the web and across the flange. Therefore, when estimating the stress intensity, the web interaction had to be considered as well. The web restrained the large flange crack from opening. Thus the

contribution of this web restraint to the stress intensity estimate was negative.

The actual value of K was found to be the sum of three or four terms as shown in Eq. 9

$$K = K_{AS} + K_{RS} + K_{LW} + K_{WR}$$
 (9)

The subscripts K_{ij} in Eq. 9 are the various contributions to the critical stress intensity. These include contributions from the applied stress, K_{AS} ; the residual stress caused by flame cut edges and web-to-flange welds, K_{RS} ; the residual stress caused by local detail welds, K_{LW} ; and the web restraint of the flange in B4A, K_{WR} .

Plastic-zone corrections were made by using the following plane stress relationship.

$$r_{y} = \frac{1}{2\pi} \left(\frac{K_{c}}{\sigma_{Yd}} \right)^{2} \tag{10}$$

Using an iterative process between Eqs. 9 and 10 values of K were obtained.

4.5.2 Contribution from the Applied Stress

To estimate the stress intensity from the applied stress for a flange edge crack, the following format was used. Generally,

$$K_{AS} = F (a') \sigma_{AS} \sqrt{\pi a'}$$
 (11)

where F (a') consists of four parts as discussed by Albrecht and $Yamada^{13}$

$$F(a') = F_E F_S F_W F_G$$

 $\mathbf{F}_{\mathbf{E}}$ = elliptical crack front correction

 F_c = free surface correction

 $\mathbf{F}_{\mathbf{G}}$ = stress concentration correction

 $\mathbf{F}_{\mathbf{W}}$ = finite width correction

For this study F_E was taken as 1.0 since the cracks were edge cracks. F_S was assumed to be \sim 1.0 because of the lateral restraint offered by the lateral attachment. F_G was also taken as 1.0 for the large edge cracks in this study. This correction affects only small elliptical surface and corner cracks and will be discussed in the next section.

The finite width correction, $F_{\overline{W}}$, was defined by Eq. 12 3 .

$$F_{W} = \sqrt{\frac{2b}{\pi a'} \tan \frac{\pi a'}{2b}}$$
 (12)

b = flange width

 $a^{\dagger} = a + r_{v}$

a = crack size

 $r_v = plastic-zone correction$

This finite width correction is exact for the model shown in Fig. 28a. This is not exactly the situation with the flange edge cracks adjacent to the lateral attachment details, however it is a good approximation. The web was assumed to prevent in-plane bending of the flange and the

lateral attachment plates were assumed to partially prevent Poisson contractions on the flange tip as shown in Fig. 28b. For these reasons the dimensions used are those shown in Fig. 28b.

In the actual beam fractures, the stresses were not uniform through the plate thicknesses nor were the edge crack fronts. For these reasons the critical stress intensity was estimated for 1/3 levels through the flange thickness. The average crack size and stress were used for the respective one-third thickness of the flange. The measured values of the critical crack size, a, for each beam are listed in Table 13. The estimated values of $K_{\overline{AS}}$ are listed in Table 14.

4.5.3 Contributions from Stress Concentration

The stress concentrations for the groove weld details were determined from a current study at Fritz Engineering Laboratory. In this study, similar details were modeled using a three-dimensionsal finite element analysis¹⁹. By comparing certain dimensional parameters, the stress concentration for the uncracked detail was determined to be 2.22 for the groove weld detail with a .75 in. (19 mm) radius transition at the 1.5 in. x 6 in. (38 mm x 152 mm) flange. Similarly, the stress concentration for the groove weld detail attached to the 2 in. x 7 in. (51 mm x 178 mm) flange was estimated as 2.19. These stress concentration factors are lower bound estimates. Examination of the fabricated details showed that for the critical details that cracked, the transition was irregular and not a

smooth radius (see Fig. 29a). These irregularities were modeled for the most severe case, a 45° angle reentrant corner with 3/4 in. (19 mm) legs (see Fig. 29b). A stress concentration factor of about 7.9 was estimated for this case.

The overlapped fillet weld detail had a comparable stress concentration at approximately 7.1 for the 1-1/2 in. x 6 in.

(38 mm x 156 mm) flange and 7.3 for the 2 in. x 7 in. (51 mm x 178 mm) flange. However, only one beam failed from this detail. There are at least two reasons for this. First, surface fabrication discontinuities at the radius elevated the apparent stress concentration. Second, the stress range at the groove weld detail was 12.5% higher than that at the fillet weld detail. The combination of these two differences made the groove weld detail more critical in all but one case.

The stress concentration, $K_{\rm T}$, decays as a crack initiates and grows at the detail. This decay is also being studied at Fritz Engineering Laboratory by Zettlemoyer¹⁹. The study matches the decay described by Albrecht and Yamada¹³, to an uncracked elliptical model. By varying the size of the ellipse in an infinite plate the effect of stress concentration decay can be matched. The purpose of this study is to develop a quick and inexpensive method to determine this decay for any detail and stress concentration situation. This analysis was used to model a groove weld detail for stress intensity variation with crack size.

The A514 steel groove weld detail on Beam B2A was examined for stress concentration effects on the stress intensity factor, K.

Results were obtained for two attachment-to-flange reentrant corner models: Case A was the smooth 3/4 in. (19 mm) radius transition (see Fig. 29a), Case B was the 3/4 in. (19 mm), 45° straight line transition shown in Fig. 29b. The stress concentration decay with crack size, $\mathbf{F_G}$, is shown in Fig. 29c for both cases. Since the stress concentration value, K, in Case B was much higher than that used in Case A, the decay of $\mathbf{K_T}$ with crack growth for Case B was more rapid than Case A. Because of this the maximum stress intensity obtained for Case B was lower than the value obtained for Case A (see Fig. 30). Hence, this elevated stress concentration (Case B) at these details did not appreciably magnify the stress concentration, when compared to Case A results, but did cause a more rapid crack initiation.

The variation of stress intensity and crack size is summarized in Fig. 30 for both cases. It was conservatively assumed that the small cracks began as small elliptical corner cracks. The variation of the semi-major and semi-minor axes was defined by Eq. 13

$$C = 1.465 \ a^{0.202} \tag{13}$$

where C = semi-major axis

a = semi-minor axis

This relationship was determined from crack size measurement data. As can be seen in Fig. 30, the maximum stress intensity obtained for elliptical corner cracks was 126 ksi $\sqrt{\text{in.}}$ (139 MPa $\sqrt{\text{m}}$) for a crack size of .35 in. (9 mm). This value was less than the critical stress

intensity of 145 ksi $\sqrt{\text{in.}}$ (159.5 MPa $\sqrt{\text{m}}$) for beam B2A. This value is also less than any slow bend material test result at -40° F (-40° C) (see Fig. 45).

4.5.4 Contribution From The Nominal Residual Stresses

 K_{RS} is either positive or negative depending upon the magnitude and distribution of the cross-section residual stresses and the crack size. When a crack grows through a tensile residual stress field there is an additional crack opening caused by the residual stresses which yields a positive K_{RS} . Similarly, when a crack grows through a compressive residual stress field there is crack closure and thus K_{RS} is negative. When a crack grows through both positive and negative residual stress fields, the residual stress condition near the crack tip, along the path of the crack, has an overriding effect.

The residual stress field through which the crack has grown can be approximated by superposition of small block stresses (see Fig. 31). $K_{\rm RS}$ can be obtained by using the following equation along with the method of superposition 14

$$K_{RS} = \frac{2}{\pi} \sigma_{rs} \sqrt{\pi a'} \sqrt{\frac{2b}{\pi a'} \tan \frac{\pi a'}{2b}} \sin^{-1} \left(\frac{\sin \frac{\pi c}{2b}}{\sin \frac{\pi a'}{2b}} \right)$$
(14)

a' = edge crack size + plastic zone correction (see Table 13)

c = dimension from the plate edge to the end or beginning of the approximated block of residual stress b = plate width

 σ_{rs} = magnitude of the residual stress block

To obtain a good approximation of K_{RS} , stress block widths of 0.02 in. (.5 mm) were used over the entire crack length. Results of K_{RS} for each beam fracture are listed in Table 14 and plotted as a function of crack size in Figs. 32 to 37.

Actual measured residual stresses of the nominal beam sections were used when available. If not available, the flange residual stresses were estimated from previous studies with similar plate thickness^{15,16}. Two assumptions were made in this estimation. First, the distribution of residual stresses through the plate thickness was assumed to be linear. Second, the residual stresses in the flange alone were assumed to be in equilibrium. The estimated residual stress distributions are shown in Figs. 39 through 41 for each steel.

4.5.5 Contribution from the Local Weld Residual Stresses

The local detail welds change the nominal section residual stress pattern over the entire cross-section at the detail. Ideally, there should be only one residual stress contribution from the actual residual stresses at this critical section. Since there was no available data on residual stress state at this section, a two step procedure was used to estimate the effect along with the principle of superposition.

After the nominal beam section residual stresses were estimated, an additional local residual stress was assumed to account for the detail welds. Both the nominal residual stresses and the local residual stresses are being measured. Pending completion of these measurements, the local welding effect was simulated in the following manner.

The residual stress distribution along the flange tip at the groove weld detail was assumed as is shown in Fig. 42a. The decay of the stress along the flange tip was assumed to be very rapid beyond the attachment edge. The stress at the location where most of the cracks initiated was assumed to be about $\sigma_{\rm Y}/4$. This stress was assumed to be distributed over 1/2 in. (12.7 mm) of the flange tip as shown in Fig. 42a. Equation 14 was again used to determine the contribution from local welding. These values are also listed in Table 14.

The fillet weld detail, top one-third analysis included a different local residual stress distribution because the detail had a fillet weld along the inside surface of the flange. It was assumed that the magnitude of the local residual stress, $\sigma_{\rm Y}/4$, at the flange tip decayed to $\sigma_{\rm Y}/8$ at the end of the transverse weld (see Fig. 42b). The middle and bottom third levels were treated similar to the groove weld details because there was also a longitudinal fillet weld made along the flange tip.

4.5.6 Contribution From The Web Restraint

Only beam B4A was observed to develop web restraint since the fatigue crack at fracture had grown as an edge crack through the web-to-flange welds and then became a two ended crack. This is shown

in Fig. 19. The analysis of the web restraint and the apparent reduction of the stress intensity is an iterative solution which is very involved. The actual analysis is discussed in detail in Appendix 1. The web restraint was predicted to decrease K by -12 ksi $\sqrt{\text{in}}$. (-13.2 MPa $\sqrt{\text{m}}$).

4.5.7 Summary and Discussion of the Various Contributions

The values of K_{AS} , K_{RS} , K_{LW} and K_{WR} are listed in Table 14 for each one-third level of the flange thickness for each critical fatigue crack. The critical value for each beam was taken as the maximum value. Some modification of these values will be made when actual residual stress measurements are available. Plots showing the variation of each K_{ij} parameter with crack size are presented in Figs. 32 to 37 for the critical one-third level of flange thickness.

The estimated residual stress diagrams shown in Figs. 39 to 41 were used to determine the average residual stress distribution for each one-third level of flange thickness. A linear distribution was assumed through the thickness. The upper one-third level had the greatest residual stress influence while the bottom one-third level had the least.

For crack growth less than approximately 1.1 in. (28 mm), the crack shape was an elliptical corner crack as described in Section 4.5.3 for the groove weld details. The local weld tensile residual stresses and the nominal section tensile residual stresses on

the flange tip both influenced the total stress intensity value in addition to the applied stress magnification by the stress concentration parameter, F_G . These variations with crack size, a, are shown in Fig. 30 for beam B2A. When the crack size for the elliptical corner cracks was approximately 1.1 in. (28 mm) the semi-major axis became equal to the flange thickness. At this point the crack rapidly became a 1.1 in. (28 mm) edge crack and the stress intensity suddenly increased. This discontinuity is shown in Fig. 30.

The residual stress effects on stress intensity for edge cracks can readily be seen in the K_{RS} vs. Edge Crack Size plots (see Figs. 32 to 37). As the edge crack grew a size of 1.1 in. (28 mm) into the negative residual stress zone there was a decrease in K_{RS} which extended over the next 2 in. of crack growth. In most cases this decrease in K_{RS} held the total stress intensity value, K_{RS} constant over this region.

Continued crack growth resulted in a rapid increase in K_{RS} as the fatigue crack grew into the high tensile residual stress region caused by the web-to-flange fillet welds. This also caused K to increase rapidly. This residual stress influence on K greatly affected the fracture of beam B4 (top one-third analysis, Fig. 34) and B6 (top one-third analysis, Fig. 36). Each beam fractured with a crack size at or near the peak K value caused by K_{RS} . The fracture of beam B6A, was caused by a rising K_{RS} and K, (see Fig. 37). The point of fracture is marked on each "K vs. a" plot.

 $K_{
m AS}$ increased at a near constant rate for edge crack growth. Since the applied stresses were very high in the A514 beams this parameter had an overriding effect on $K_{
m RS}$ and $K_{
m LW}$. This is shown in the bottom one-third analysis for beams B2 and B2A, Figs. 32 and 33 respectively.

 K_{LW} had its greatest influence on small elliptical corner cracks (see Fig. 30). For edge cracks at the groove weld details this contribution became constant and comparatively small. This contribution was slightly higher for the overlapped fillet weld detail.

The fracture of B4 was precipitated by the presence of the high tensile residual stress area at the web-to-flange welds. Beam B4A had a fatigue crack which grew through the same area during a fracture test and at a 6% higher applied stress but did not fail. This can only be explained by a difference in test temperatures when the fatigue cracks grew into this critical area. As can be seen from the material tests K vs. temperature plot for A36 steel (Fig. 43), a slight difference in the test temperatures would cause a large change in the critical stress intensity factor, $K_{\mathbb{C}}$. This was the case as the beam B4A test temperature $\{-70^{\circ}\ F\ (-57^{\circ}\ C)\}$ was warmer than the temperature of beam B4 $\{-80^{\circ}\ F\ (-62^{\circ}\ C)\}$ when the fatigue crack grew into this region.

As the fatigue crack in beam B4A grew through the web-to-flange welds K_{RS} was continually increasing. However, this was counter balanced by the flange crack opening restraint of the web. Only when the crack grew $\sim 1-1/4$ in. (~ 32 mm) past the web did

fracture occur. $K_{\mbox{RS}}$ had only a small effect on the estimated stress intensity since, at the time of fracture, the critical K was determined at the bottom one-third level of flange thickness.

5. COMPARISON OF BEAM K ESTIMATES AND MATERIAL K TESTS

5.1 Lateral Attachment Details

The beam fracture stress intensity estimates were correlated with the static and dynamic material toughness characterizations. Both the A36 and A588 beam fractures occurred at temperatures in the transition temperature region of the slow bend $K_{\rm IC}$ material tests. As can be seen in Figs. 43 to 45, there is a very good correlation between the beam K estimates and the slow bend material tests. The A514 beam fractures occurred at temperatures below the slow bend curve transition temperature region. The beam stress intensity estimates, however, were conservative since these points were above the $K_{\rm IC}$ value.

The good correlation between the beam stress intensity estimates and the slow bend $K_{\rm IC}$ material tests can be attributed to their similar loading rates. As discussed in Section 2.9, the beam fracture test loading rate was between 70 and 100 ksi/sec. and occurred as the crack front was being advanced under cyclic loading. The slow bend, three-point bend specimens were loaded at a rate of 20 kips/sec. which is 50 ksi/sec. at the crack tip. The dynamic $K_{\rm ID}$ specimens were fractured in approximately 4 x 10^{-4} sec. The beam tests demonstrated that the fracture resistance of these welded bridge details corresponded to the fracture toughness measurements which used a loading time of about one second.

Also plotted in Figs. 43 to 45 are the beam stress intensity estimates from the applied stress alone (K_{AS}). There is good correlation between K_{AS} for beams B2, B2A, B4A, and B6A and their respective slow bend material test results. This demonstrates that in these tests, the residual stresses from welding and flame cutting did not significantly alter the fracture resistance. However the contribution to the stress intensity estimate from the residual stress field, K_{RS} should be considered when the crack tip is in the high tensile residual stress region of the web-to-flange welds. This can readily be seen in Figs. 18 and 20 for beams B4 and B6, respectively. In both of these cases K_{RS} was nearly 50% of the total stress intensity estimate.

CONCLUSIONS

This report summarizes the fatigue and fracture resistance of full scale welded beams with lateral attachments. The fatigue test results were correlated with available test data obtained from smaller beams. The beam fracture resistance was correlated with fracture control tests made on the same material.

- 1. The stress intensity estimates from the beam fractures were best modeled by the slow bend $K_{\hbox{\scriptsize IC}}$ fracture toughness. The beam fracture tests and the slow bend $K_{\hbox{\scriptsize IC}}$ tests had similar loading rates. These tests have demonstrated the applicability of the one second loading time to measurements of fracture resistance of bridge beams.
- 2. For relatively large cracks, a good approximation of the critical stress intensity factor, K, for beam fractures can be estimated by only considering the applied stress. However if the edge crack tip has moved into the high tensile residual stress field near the web-to-flange welds, the residual stress contribution, K_{RS}, should be included. Fracture usually occurred when the crack tip was in this region. In one instance there was rapid fatigue crack growth through this region due to a rise in K, however, fracture did not occur until the fatigue crack was larger.

- 3. At the time of fracture, most of the fatigue life of the welded girders was exhausted. Hence, fatigue resistance design is a major objective of any fracture control plan for bridge girders.
- 4. Category E of the current AASHTO fatigue specifications was found to be applicable to the 12 in. flange attachment as was expected.
- 5. The stress concentration effects for small elliptical corner cracks at a groove weld detail was analyzed. The maximum stress concentration was at an elliptical corner crack with a simi-minor axis of 0.4 in. (10 mm). The predicted stress intensity factor was less than the estimated resistance at fracture. This value was also less than the predicted fracture toughness value from the slow bend material tests at a service temperature of -40° F $(-40^{\circ}$ C).
- 6. The Charpy V-notch data in the transition zone was converted to stress intensity values by Barsom's equation. Excellent correlation was found for the A36 steel. However, nonconservative values were predicted for the A588 steel and very conservative results were predicted for the A514 steel.
- 7. The measured loading rate temperature shift was always greater than the empirical approximation suggested by Barsom.

 Hence this approximation is a conservative estimate.

APPENDIX 1

The two-ended crack shown in Fig. 46 was analyzed by using a method similar to that proposed by Madison¹⁷. The crack openings of the flange and web crack at the web-to-flange junction are known to be equal. Therefore, to satisfy compatibility, a closing force was applied to the flange crack and an opening force is applied to the web crack.

The flange crack opening at the compatibility point is a function of the applied stress and the residual stress. Local weld effects can be neglected since the crack tip is distant from the detail welds.

$$v_f = v_{AS} + v_{fRS}$$
 (A1)

v_f was obtained from the formulation presented in Ref. 14 (see AS Fig. 46).

$$v_{f_{AS}} = \frac{4 \sigma a}{E} \quad V_{1} \quad \left(\frac{a}{b}\right) \tag{A2}$$

$$V_{1} \left(\frac{a}{b}\right) = \frac{1}{\left(\frac{\pi a}{2b}\right)} \left\{0.459 \left(\sin\frac{\pi a}{2b}\right) - 0.065 \left(\sin\frac{\pi a}{2b}\right)^{3} - 0.007 \left(\sin\frac{\pi a}{2b}\right)^{5} + \cosh^{-1} \left(\sec\frac{\pi a}{2b}\right)\right\}$$

 v_{f} was derived following the formulation presented by Madison¹⁷ for RS a partially loaded edge crack (see Fig. 48).

$$x < c: v_{RS} = \frac{4\sigma}{E\pi} \left(V_1 \left(\frac{a}{b} \right) \right) \left\{ \sqrt{a^2 - x^2} \sin^{-1} \frac{c}{a} + c \coth^{-1} \sqrt{\frac{a^2 - x^2}{a^2 - c^2}} - x \cot^{-1} \frac{c}{x} \sqrt{\frac{a^2 - x^2}{a^2 - c^2}} \right\}$$

$$x > c: V_{RS} = \frac{4\sigma}{E\pi} \left(V_1 \left(\frac{a}{b} \right) \right) \left\{ \sqrt{a^2 - x^2} \sin^{-1} \frac{c}{a} + c \tanh^{-1} \sqrt{\frac{a^2 - x^2}{a^2 - c^2}} - x \tanh^{-1} \frac{c}{x} \sqrt{\frac{a^2 - x^2}{a^2 - c^2}} \right\}$$
(A3)

x = 0

$$V_{RS} = \frac{4\sigma}{E\pi} \left(V_1 \left(\frac{a}{b} \right) \right) \left\{ a \sin^{-1} \frac{c}{a} + c \coth^{-1} \left(1 - \frac{c^2}{a^2} \right) \right\}$$
(A4)

The web crack opening at the compatibility point is also a function of the applied stress and the residual stress 14 .

$$v_w = v_{wAS} + v_{wRS}$$

v was estimated following the formulation presented in Ref. 14 for WAS the in-plane bending case (see Fig. 48 for the diagram)

$$v_{\text{WAS}} = \frac{4 \text{ o } a}{E} \quad V_{2} \quad \left(\frac{a}{b}\right)$$

$$V_2 = \left(\frac{a}{b}\right) = 0.8 - 1.7 = \left(\frac{a}{b}\right) + 2.4 = \left(\frac{a}{b}\right)^2 + \frac{0.66}{\left(1 - \frac{a}{b}\right)^2}$$
 (A5)

 $v_{
m w}$ was derived in a manner identical to $v_{
m f}_{
m RS}$

If $v_f < v_w$ there is no web restraint and the stress intensity can be computed by analyzing the flange edge crack alone. If $v_f > v_w$, there is a web restraining effect.

The difference between $v_{\mbox{\it f}}$ and $v_{\mbox{\it w}},\;\Delta v,\; has to be equal zero to meet the compatibility conditions$

$$\Delta v = v_f - v_w \tag{A6}$$

After defining an interaction area (see Fig. 44) a closing force was applied to the flange crack. Similarly, an opening force is applied to the web crack. This force must be defined as a stress acting over an interaction area since crack displacement at a point load is not defined. The flange closing, v_f , and the web opening, v_w , are defined by Eqs. Al and A5 as a function of stress σ_f and σ_w .

$$v_{f_c} = f(\sigma_f)$$

$$v_{w_o} = f(\sigma_w)$$
(A7)

Since the interaction area is assumed to be common to both the flange and web then

$$\left|\sigma_{\mathbf{f}}\right| = \left|-\sigma_{\mathbf{w}}\right| \tag{A8}$$

By the compatibility condition

$$v_{f_c} + v_{o} = \Delta v \tag{A9}$$

 σ_{f} or σ_{w} can be solved directly from Eqs. A7 and A9. From the stress in the flange σ_{f} , and the assumed interaction area a restraining value of K can be determined through Eq. 14 in Section 4.5.4.

Ideally this procedure should be an iterative one using the plane stress plastic zone correction

$$r_y = \frac{1}{2\pi} \left(\frac{K_c}{\sigma_Y} \right)^2$$

Since the fracture toughness, K_c , of the material from the material characterization is known, a first approximation of r_y can be obtained and thus a good estimate of K_{WR} . This was the case for analysis of beam B4A. Only one iteration was needed since the interaction area was in the top one-third of the flange thickness as shown in Fig. 47. The restraint was decreased linearly to the bottom one-third. Thus K_{WR} was -12, -6, 0 for top, middle and bottom levels of the flange thickness.

NOMENCLATURE

- a = edge crack size
- a' = a + r
- B = 3 point bend specimen width
- b = flange width
- c = dimension from the plate edge to the end or beginning of the approximated block of residual stress (see Fig. 4.17)
- E = Young's Modulus, 29000 ksi
- $F(a') = F_E F_G F_S F_W$
- $F_{\rm E}$ = elliptical crack front correction
- F_{C} = stress concentration correction
- F_S = free surface correction
- F_W = finite width correction
- K = linear elastic fracture mechanics stress intensity factor
 - $= K_{AS} + K_{RS} + K_{LW} + K_{WR}$
- K_{AS} = stress intensity contribution from the applied stress
- $K_{\mbox{RS}}$ = stress intensity contribution from the nominal section residual stresses
- K_{LW} = stress intensity contribution from the local weld residual
 stresses

 $K_{\overline{WR}}$ = stress intensity contribution from the web restraint

K_c = fracture toughness value

 $K_{\mbox{Id}}$ = fracture toughness value from the dynamic material test

 $K_{_{\rm T}}$ = stress concentration factor

P = applied load

r_v = plastic zone size

t = loading time to maximum load

t = time of load application for a static tensile test

T = testing temperature

t_f = flange thickness

v_f = flange crack opening, = v_f + v_f_{RS}

 $v_{f_{AC}}$ = flange crack opening from the applied stress

 $\mathbf{v}_{\mathbf{f}_{\mathbf{DC}}}$ = flange crack opening from the residual stress

v. = web crack opening

 $v_{M,Q}$ = web crack opening from the applied stress

 v_{pc} = web crack opening from the residual stress

 σ_{AS} = applied stress

 $\sigma_{\mathbf{v}}$ = yield stress

 $\sigma_{\rm Vd}$ = yield stress as a function of loading rate and temperature

 σ_r = stress range

 σ_{rs} = residual stress block stress

TABLES

TABLE 1 LIST OF TEST SPECIMENS

		Beam Numbers	
Detail Type Steel Type	A36	A588	A514
Lateral Attachment	В4	В6	В2
Category E	в4А	B6A	B2A
	•		
Cover Plate	В3*	B5*	B1
Category E	B3A*	B5A*	BlA
Transverse Stiffener	В9-	B11	В7
Category C	в9А	B11A	B7A
Flange Transition	В10	B12	В8
Category B	B10A	B12A	B8A

^{*} Rolled Beams

All Others Welded

²⁴ Beams - Total

TABLE 2a RESULTS OF MILL TESTS

Plate ť	Steel	Heat Number	Yield Pt. (ksi)	Tensile Strength (ksi)	Elong. Gage/%	С	M n	P	S	s	C _u	N _i	C _r	v	M _o	В
1/2"	A36	401P1041	44.10	66.20	8/31	.14	1.06	.013	.017	.19	1					
1"	A36	411P4511	40.70	61.40	8/32	.14	1.06	.014	.032	.19	-					
2"	A36	402P7031	44.00	70.00	2/34	.17	1.06	.013	.022	.21						
3"	A36	432N4711	45.00	72.00	2/32	.17	1.09	.015	.024	.21			,			
1/2"	A588	401N6061	57.20	74.70	8/26	.13	1.09	.019	.028	.28	.28	.37	.57	.038		
1/2"	A588	432N2461	53.50	74.60	8/27	.12	1.17	.011	.023	.25	.29	.34	.50	.031		
2"	A588	401P8161	56.50	78.50	2/33	.12	1.09	.013	.019	.24	.26	.32	.54	.033		
2"	A588	402P7731	61.00	80.00	8/33	.10	1.12	.011	.025	.28	.29	.28	.55	.030		
3"	A588	494N5681	57.50	79.50	2/30	.12	1.08	.010	.027	.29	.29	.31	.51	.028		
3/8"	A514/5	801P03810	113,53	118.50	2/24	.17	.61	.008	.023	.27					.57	.0025
1/2"	A514/5	801P03810	113.00	120.25	2/30	.17	.61	.008	.023	.27					.57	.0025
1"	A514/5	801P03810	114.55	121.80	2/32	.17	.61	.008	.023	.27				• :	.57	.0025
1-1/2"	A514/M	802P50780	125.10	134.15	2/31	.18	.61	.008	.023	.31		1.40			.52	.0028
1-1/2"	A514/ RQ1008	802N80660	117.00	129.50	2/21	.17	.59	.008	.021	.29		1.37			.49	.0022
2	A514/M	801N18640	110.00	122.25	2/19	.18	.66	.007	.023	.26		1.33			.50	.0036

TABLE 2b RESULTS OF MILL TESTS

Plate t	Steel	Heat Number	Yield Pt. (MPa)	Tensile Strength (MPa)	Elong. Gage/%	c	M n	P	S	S _i	C _u	N _i	C _r	V	Мо	В
1/2"	A36	401P1041	304	456	8/31	.14	1.06	.013	.017	.19						
1"	A36	411P4571	281	423	8/32	.14	1.06	.014	.032	.19						
2"	A36	402P7031	303	483	2/34	.17	1.06	.013	.022	.21						
3"	A36	432N4711	310	496	2/32	.17	1.09	.015	.024	.21						
1/2"	A588	401N6061	394	515	8/26	.13	1.09	.019	.028	.28	.28	37	.57	.038		
1/2"	A588	432N2461	369	514	8/27	.12	1.17	.011	.023	.25	.29	.34	.50	.031		
2"	A588	401P8161	390	541	2/33	.12	1.09	.013	.019	.24	.26	.32	.54	.033		
2"	A588	402P771	421	552	8/33	.10	1.12	.011	.025	.28	. 29	.28	.55	.030		
3"	A588	494N5681	396	548	2/30	.12	1.08	.010	.027	.29	.29	.31	.51	.028		
									•							
3/8"	A514/5	801P03810	783	817	2/24	.17	.61	.008	.023	.27					.57	.0025
1-1/2"	A514/51	801P03810	779	829	2/30	.17	.61	.008	.023	.27					.57	.0025
1"	A514/5	801P03810	790	840	2/32	.17	.61	.008	.023	.27					.57	.0025
1-1/2"	A514/M	802P50780	863	925	2/31	.18	.61	.008	.023	.31		1.40			.52	.0028
1-1/2"	A514/ RQ1008.	802N80660	807	893	2/21	.17	.59	.008	.021	.29		1.37			.49	.0022
2	A514/M	801N18640	758	843	2/19	.18	.66	.007	.023	.26		1.33			.50	.0036

TABLE 2c MILL TEST CVN RESULTS

.Plate	Steel	Heat Number		py Resu (Ft-1bs.		Test Temp. (°F)	Spec. Ft-1bs. @ °F		py Resu Joules) 2		Test Temp. (°C)	Spec. Joules @ °C
	Ç CCC1	2, 4110-02	-	_	J	(- /	-	•	-	J	(0)	
1/2"	A36	401P1041	157	170	163	40	15 @ 40	213	231	221	4.5	20 @ 4.5
1"	A36	411P4571	68	53	34	40	15 @ 40	92	72	46	4.5	20@4.5
211 .	A36	402P7031	39	54	53	40	15 @ 40	53	73	72	4.5	20@4.5
3"	A36	432N4711	74	. 75	60	40	15 @ 40	100	102	81	4.5	20 @ 4.5
												22 2 4 5
1/2"	A588	401N6061	52	46	49	40	15 @ 40	71	62	67	4.5	20 @ 4.5
1/2"	A588	432N2461	48	44	22	40	15 @ 40	65	- 60	30	4.5	20 @ 4.5
2"	A588	401P8161	82	65	83	40	15 @ 40	111	88	113	4.5	20@4.5
211	A588	402P7731	65	77	40	40	15 @ 40	88	105	54	4.5	20@4.5
3''	A588	494N5681	37	41	57	40	15 @ 40	50	56	77	4.5	20 @ 4.5
3/8"	A514/5	801P03810	28/39	20/34	19/28	0	25 @ 0	38/53	27/46	26/38	-18	34@-18
1/2"	A514/5	801P03810	32	32	34	0	25 @ 0	43	43	46	-18	34 @ -18
1"	A514/5	801P03810	62/26	56/26	47/26	0	25 @ 0	84/35	76/35	64/35	-18	34 @ -18
1-1/2"	A514/5	802P50780	55	56	49	. 0	25 @ 0	75	76	67	-18	34 @ -18
1-1/2"	A514/ RQ100B	802N80660	28	27	27	0	25 @ 0	38	37	37	-18	34 @ -18
2	A514/M	801N18610	64	62	60	0	25@0	87	84	81	-18	34@-18

TABLE 3a CROSS-SECTIONAL PROPERTIES OF TEST SPECIMENS

		Fla	ange	Web	Total	Nominal Moment of	Nominal Section
Beam		Width	Thickness	Thickness	Depth	Inertia	Modulus
Number	Steel	(in.)	(in.)	(in.)	(in.)	(in. ⁴)	(in. ³)
В2	A514	5.97	1.567	0.385	36.08	6482	360.1
B2A	A514	6.15	1.561	0.386	36.19	6482	360.1
В4	A36	6.97	2.019	0.375	35.98	9125	506.9
B4A	A36	7.00	2.016	0.375	35.91	9125	506.9
В6	A588	7.03	2.035	0.387	36.00	9125	506.9
в6А	A588	6.98	2.032	0.393	35.98	9125	506.9

TABLE 3b CROSS-SECTIONAL PROPERTIES OF TEST SPECIMENS

		F1a	ange	Web	Total	Nominal Moment of	Nominal Section
Beam Number	Steel	Width (mm)	Thickness (mm)	Thickness (mm)	Depth (mm)	Inertia (cm ⁴)	Modulus (cm³)
В2	A514	152	39.67	9.78	916	269 667	5901
B2A	A514	156	39.65	9.80	919	269 667	5901
в4	A36	177	51.28	9.53	914	379 623	8307
B4A	Λ36	178	51.21	9.53	912	379 623	8307
В6	. A588	179	51.69	9.83	914	379 623	8307
B6A	A588	177	51.61	9.98	914	379 623	8307

TABLE 4a CROSS-SECTION TEMPERATURES AT FRACTURE

		·		Te	emperatures a	t Fracture*	*		
Beam Number	Order of Test	Bottom Flange Tl (°F)	Web Stiff. T2 (°F)	Top Flange T3 (°F)	Bottom Flange T4 (°F)	Top Flange T5 (°F)	Bottom Flange T6 (°F)	Web Stiff. T7 (°F)	Top Flange T8 (°F)
В2	6	-155*	-106	-102	-171				 .
B2A	4	-61	-71		-144*	-67			
В4	2	-80*	- 59	-45					
B4A	1		-40		-105/-96*	-36			
В6	3		·			·	- 53*	-19	-08
в6А	5	-43	-77		-90/-94*	-68			

^{*} Denotes test control gage at critical detail

^{**} See Fig. 4 for gage locations

TABLE 4b CROSS-SECTION TEMPERATURES AT FRACTURE

			Temperatures at Fracture**											
Beam Number	Order of Test	Bottom Flange Tl (°C)	Web Stiff. T2 (°C)	Top Flange T3 (°C)	Bottom Flange T4 (°C)	Top Flange T5 (°C)	Bottom Flange T6 (°C)	Web Stiff. T7 (°C)	Top Flange T8 (°C)					
В2	6	-104*	-77	-74	-113									
B2A	4	- 52	-57	·	-98*	- 55								
В4	2	-62	-51	-43										
B4A	1		-40		-76/-71*	-38								
В6	3						-47*	-28	-22					
B6A	5	-42	-61 ·		-68/-70*	-56								

^{*} Denotes test control gage at critical detail

^{**} See Fig. 4 for gage locations

TABLE 5a LOAD HISTORY FOR BEAM B2 (A514)

Testing	ID	Subtotal	Cumm.]	Fracture	e Test D	ata		Fatigue Data		
Event	*	N	N	Detail Tested	No.	** Temp. °F	Fract. Temp. °F	or ksi	o max ksi	σ r ksi	σ max ksi	
Fatigue	a	2,009,100	2,009,100							G 8.7 F 8.0	26.0	
Fracture	b	10,000	2,019,100	F,G	1	-40		8.7 8.0	55.0			
	ъ	5,000	2,024,100	G	1	-130 to -155	-155	8.7	55.0			

^{*} See fracture surface sketches for banding identification

Steel type A514

F - Fillet welded detail

G - Groove welded detail

^{**} Temperatures at controling gages

TABLE 5b LOAD HISTORY FOR BEAM B2 (A514)

Testing Event	ID	Subtotal	Cumm.		F	ractur		Fatigue Data			
Event	*	N	N	Detail Tested	No.	** Temp. °C	Fract. Temp. °C	σ _r MPa	o max MPa	σ r MPa	o max MPa
Fatigue	а	2,009,100	2,009,100					,		G 60 F 55	179
Fracture	Ъ	10,000	2,019,100	F,G	1	-40		60 55	379		
	Ъ	5,000	2,024,100	G	1	-90 to -104	-104	60	379		

^{*} See fracture surface sketches for banding identification

Steel type A514

F - Fillet welded detail

G - Groove welded detail

^{**} Temperatures at controling gages

TABLE 6a LOAD HISTORY FOR BEAM B2A (A514)

Testing	ID	Subtotal	Cumm.				e Test I)ata		Fatigue	Data
Event	*	N	N	Detail Tested	No.	** Temp.	Fract. Temp. °F	σ r ksi	σ max ksi	σ _r ksi	σ max ksi
Fatigue	a	1,982,800	1,982,800					·		G 8.7 F 8.0	26.0 23.8
Fracture	b	15,000 ⁺ 35,000	2,017,800	G F G F	1 1 1	-40 -40 -40 -40		4.3 4.0 8.7 8.0	50.6 46.4 55.0 50.4		
: Fracture	С	13,800 ⁺ 55,000	2,072,800	G G	2 2	-40 -40		4.3 8.7	50.6 55.0		
Fatigue	đ	407,500	2,480,300							G 8.7 F 8.0	26.0 23.8
Fracture	е	12,500 ⁺ 48,750	2,529,050	G G	3 3	-40 -40		4.3 8.7	50.6 55.0		
Fracture	£	87,500	2,616,550	G	4	-40		8.7	55.0		
Fatigue	g	180,400	2,796,950				÷.			G 8.7 F 8.0	26.0 23.8
Fracture	h	68,750	2,865,700	G	5	-40 to -144	-144	8.7	55.0		

^{*} See fracture surface sketches for banding identification

G - Groove welded detail F - Fillet welded detail

^{**} Temperature at controlling gages

^{+ -} Cycles for marking crack front

Steel Type - A514

TABLE 6b LOAD HISTORY FOR BEAM B2A (A514)

Testing	ID	Subtotal	Cumm.				e Test I)ata		Fatigue	Data
Event	*	N	N	Detail Tested	No.	** Temp. °C	Fract. Temp. °C	or MPa	σ MPa	o r MPa	o max MPa
Fatigue	а	1,982,800	ĭ,982,800	,						G 60 F 55	179 164
Fracture	ъ	15,000 ⁺	2,017,800	G F G F	1 1 1	-40 -40 -40 -40		30 28 60 55	349 320 379 348		
Fracture	С	13,800 ⁺ 55,000	2,072,800	G G	2 2	-40 -40		30 60	349 379		; ;
Fatigue	d	407,500	2,480,300							G 60 F 55	179 164
Fracture	е	12,500 ⁺ 48,750	2,529,050	G G	3	-40 -40		30 60	349 379		
Fracture	f	87,500	2,616,550	G	4	-40		60	379		
Fatigue	g	180,400	2,796,950							G: 60 F 55	179 164
Fracture	h	68,750	2,865,700	G	5	-40 to -98	-98	60	379		

See fracture surface sketches for banding identification

+ - Cycles for marking crack front

G - Groove welded detail

F - Fillet welded detail ** Temperature at controlling gages Steel type - A514

TABLE 7a LOAD HISTORY OF BEAM B4 (A36)

Testing	ID	Subtotal	Cumm.		Fracture Test Data						Data
Event	*	N	N	Detail Tested	No.	** Nominal Temp. °F	Fract. Temp. °F	or ksi	σ max ksi	σ r ksi	σ max ksi
Fatigue	a	2,001,800	2,001,800							G 9.0 F 8.0	19.8 17.6
Fracture	ъ	10,000+		G F	1 1	-40 -40		4.5 4.0	15.3 13.6		
		7,500	2,009,300	G F	1 1	-40 -40		9.0 8.0	19.8 17.6		:
Fatigue	С	299,200	2,308,500						: •	G/F 9.0/8.0	19.8/17.6
Fatigue	d	36,700	2,345,200							G/F 6.0/5.3	15.0/13.3
Fracture	е	5,000 ⁺ 10,000 14,500	2,355,200 2,369,700		2 2 2	-55 -70 -70	-80	4.5 9.0 9.8	15.3 19.8 19.8		

^{*} See fracture surface sketches for banding identification

^{**} Temperature at controlling gages

G - Groove welded detail

F - Fillet welded detail

^{+ -} Cycles for marking crack front Steel type - A36

TABLE 7b LOAD HISTORY OF BEAM B4 (A36)

Testing	ID	Subtotal	Cumm.		Fra	cture Te	st Data	٠.		Fatigue Data		
Event	*	N	N	Detail Tested	No.	** Nominal Temp. °C	Fract. Temp. °C	or MPa	o max MPa	σ _r MPa	o max MPa	
Fatigue	а	2,001,800	2,001,800							G 62 F 55	137 121	
Fracture	Ъ	10,000+		G F	1 1	-40 -40		31 28	105 94		·	
		7,500	2,009,300	G F	1 1	-40 -40		62 55	137 121			
Fatigue	С	299,200	2,308,500							G/F 62/55	137/121	
Fatigue	đ	36,700	2,845,200							G/F 41/37	103/92	
Fracture	е	5,000 ⁺ 10,000 14,500	2,355,200 2,369,700	G G G	2 2 2	-48 -57 -57	-62	31 62 68	105 137 137	,	·	

^{*} See fracture surface sketches for banding identification

^{**} Temperature at controlling gages

G - Groove welded detail

F - Fillet welded detail

^{+ -} Cycles for marking crack front Steel type A36

TABLE 8a LOAD HISTORY OF BEAM B4A (A36)

Testing	ID	Subtotal	Cumm.		F	racture Test	Data			Fatigue	Data
Event	*		}			**	Fract.	Sr	σ	Sr	σ
				Detail		Temp.	Temp.	r	max	r	max
		N	N	Tested	No.	°F	°F	ksi	ksi	ksi	ksi
Fatigue	а	1,500,000	1,500,000							G/F 9.0/8.0	1.98/17.6
Fracture	a	7,500+		F	1	-40		4.0	13.6		
		7,500	1,507,500	F	1	-40		8.0	17.6		_
Fatigue	ā	250,000	1,757,500		<u> </u>			i	ł	G/F 9.0/8.0	19.8/17.6
Fracture	a	7,500+		F	2	-40		4.0	13.6		
		7,500	1,765,000	F	2	-40		8.0	17.6		_
Fatigue	а	250,000	2,015,000		ĺ					G/F 9.0/8.0	19.8/17.6
Fracture	Ъ	7,500+		G	3	-40		4.5	15.3		
		7,500	2,022,500	G	3	-40		9.0	19.8		_
Fatigue	С	250,000	2,272,500					1		G/F 9.0/8.0	19.8/17.6
Fracture	d	7,500+		G	4	-40		4.5	15.3		
		7,500	2,280,000	G	4	-40		9.0	19.8		
Fatigue	e	250,000	2,530,000							G/F 9.0/8.0	19.8/17.6
Fracture	f	7,500+	'	G	5	-60		4.5	15.3		
		18,750	2,548,750	G	5	- 60		9.0	19.8		
Fatigue	g	352,000	2,900,750	٠.	:	190	./			G/F 9.0/8.0	19.8/17.6
Fracture	g	7,500+		F	6	-40	<u> </u>	4.0	13.6		
		7,500	2,908,250	F	6	-40		8.0	17.6		
Fatigue	g	67,900	2,976,150					}		G/F 9.0/8.0	19.8/17.6
Fracture	h	7,500+		G	7	-40	ĺ	4.5	15.3		
		7,500	2,983,650	G	7	-40	}	9.0	19.8		
		5,000	2,988,650	G	7	-120 to - 170	į	4.5	15.3		
		27,500+		G	7	-170 to -100		9.0	19.8		
Fatigue	i	243,100	3,231,750							G/F 9.0/8.0	19.8/17.6
_		8,700+								G/F 4.5/4.0	15.3/13.6
Fracture	j	5,000	3,236,750	G	8	-70		4.5	15.3		
	L	40,000	3,276,750	G	8	-70 to - 96	- 96	9.0	19.8		

^{*} See fracture surface sketches for banding identification

G - Groove welded detail

^{**} Temperature at controlling gages

Steel type - A36

F - Fillet welded detail

^{+ -} Cycles for marking crack front

TABLE 8b LOAD HISTORY OF BEAM B4A (A36)

Testing	ID	Subtotal	Cumm.		I	racture Test				Fatigue I	Data
Event	*					**	Fract.	σ	α .	۲	۲
				Detail		Temp.	Temp.	σr	omax	$\sigma_{f r}$	σ max
		N	N	Tested	No.	°C	°C	MPa	MPa	MPa	MPa
Fatigue	a	1,500,000	1,500,00							G/F 62/55	137/121
Fracture	a	7,500 ⁺		F	1	-40		28	94		
ļ		7,500	1,507,500	F	1	-40		55	121	_	_
Fatigue	a	250,000	1,757,500							G/F 62/55	137/121
Fracture	a	7,500+		F	2	-40		28	94		
		7,500	1,765,000	F	2	-40		55	121		
Fatigue	a	250,000	2,015,000							G/F 62/55	137/121
Fracture	Ъ	7,500+		G	3	-40		31	105		
1		7,500	2,022,500	G	3	-40		62	137		
Fatigue	С	250,000	2,272,500							G/F 62/55	137/121
Fracture	đ	7,500+		G	4	-40		31	105		,
		7,500	2,280,000	G	4	-40		62	137		,
Fatigue	e	250,000	2,580,000							G/F 62/55	137/121
Fracture	f	7,500+		G	5	-51		31	105		
		18,750	2,548,750	G	5	-51		62	137		
Fatigue	g	352,000 7,500	2,900,750						•	G/F 62/55	137/121
Fracture	g	7,500 ^T		F	6	-40		28	94		
	-	7,500	2,908,250	F	6	40		55	121		
Fatigue	g	67,900	2,976,150							G/F 62/55	137/121
Fracture	h	7,500 ⁺		G	7	-40		31	105		
		7,500	2,983,650	G	7	-40		62	137		
		5,000	2,988,650	G	7	-84 to -112		31	105		
		27,500+		G	7	-112 to -73		62	137		_
Fatigue	i	243,100	3,231,750			,				G/F 62/55	137/121
		8,700 ⁺	-							G/F 31/28	105/94
Fracture	j	5,000	3,236,750	G	8	-57		31	105		
		40,000	3,276,750	G	8	-57 to -71	-71	62	137		

^{*} See fracture surface sketches for banding identification

G - Groove welded detail

^{**} Temperature at controlling gages

Steel type - A36

F - Fillet welded detail

^{+ -} Cycles for marking crack front

TABLE 9a LOAD HISTORY OF BEAM B6 (A588)

Testing	ID.	Subtotal	Cumm.			Fractui	e Test I	Data		Fatigue	e Data
Event	*	N	N	Detail Tested	No.	** Temp. °F	Fract. Temp. °F	σ r ksi	σ max ksi	σ r ksi	σ max ksi
Fatigue	а	1,999,800	1,999,800							G 9.0 F 8.0	27.5 24.4
Fracture	Ъ	5,000 ⁺ 7,500	2,007,300	G F G F	1 1 1	-30 -30 -40 -40		4.5 4.0 9.0 8.0	23.0 20.4 27.5 24.4		
Fatigue	c ·	797,400	2,804,700					-		G 9.0 F 8.0	27.5 ^x 24.4
Fracture	đ	18,750 ⁺ 75,000	2,879,700	F F	2 2	-40 -40		4.0 8.0	20.4 24.4	·	
Fracture	e	7,500 75,000	2,954,700	F F	3	-40 -40	-53	4.0 8.0	20.4 24.7 ^y		٠

^{*} See fracture surface sketches for banding identification

^{**} Temperature at controlling gages

G - Groove welded detail

F - Fillet welded detail

^{+ -} Cycles for marking crack front

x - Static jack dropped load maximum stress changed from 27.5 to ~23 for 400,000 cycles of load

y - Static jack increased load Steel type - A588

TABLE 9b LOAD HISTORY OF BEAM B6 (A588)

Testing	ID	Subtotal	Cumm.			Fractur	re Test I	ata		Fatigu	e Data
Event	*	N	N	Detail Tested	No.	** Temp. °C	Fract. Temp. °C	σ r MPa	σ _{max} MPa	o r MPa	o max MPa
Fatigue	а	1,999,800	1,999,800							G 62 F 55	190 168
Fracture	ъ	5,000 ⁺ 7,500	2,007,300	G F G F	1 1 1	-34 -34 -40 -40		31 28 62 55	159 141 190 168		
Fatigue	c	797,400	2,804,700		_					G 62 F 55	190 ^x 168
Fracture	đ	18,750 ⁺ 75,000	2,879,700	F F	2 2	-40 -40		28 55	141 168		
Fracture	е	7,500 75,000	2,954,700	F F	3 3	-40 -40	-47	28 55	141 170 ^y		

^{*} See fracture surface sketches for banding identification

Steel type - A588

^{**} Temperature at controlling gages

G - Groove welded detail

F - Fillet welded detail

^{+ -} Cycles for marking crack front

x - Static jack dropped load maximum stress changed from 27.5 to ~23 for 400,000 cycles of load

y - Static jack increased load

TABLE 10a LOAD HISTORY OF BEAM B6A (A588)

Testing	ID	Subtotal	Cumm.		F	racture T	est Data	1		Fatigue Data		
Event	*	. N	N	Detail Tested	No.	** Temp. °F	Fract. Temp. °F	or ksi	o max ksi	σ r ksi	o _{max} ksi	
Fatigue	a	2,042,600	2,042,600							G 9.0 F 8.0	19.0 16.9	
Fracture	ъ	22,500	2,065,100	G F	1	-40 -40		9.0 8.0	27.5 24.4			
Fatigue	С	732,400	2,797,500							G 9.0 F 8.0	19.0 16.9	
Fracture	đ	25,000	2,822,500	G	2	-40/-90	-92	9.0	28.3 ^x			

^{*} See fracture surface sketches for banding identification

^{**} Temperature at controlling gages

G - Groove welded detail

F - Fillet welded detail

x - Static jack increased load Steel type - A588

TABLE 10b LOAD HISTORY OF BEAM B6A (A588)

Testing	ID	Subtotal	Cumm.		F	racture T	est Data	L		Fatigu	ie Data
Event		N	N	Detail Tested	No.	** Temp. °C	Fract. Temp. °C	σ _r MPa	o _{max} MPa	σ _r MPa	o _{max} MPa
Fatigue	a	2,042,600	2,042,600							G 62 F 55	131 117
Fracture	b	22,500	2,065,100	G F	1 1	-40 -40		62 55	190 168	,	
Fatigue	С	732,400	2,797,500					,		G 62 F 55	131 117
Fracture	d	25,000	2,822,500	G	2	-40/-68	-69	62	195 ^x		

^{*} See fracture surface sketches for banding identification

^{**} Temperature at controlling gages

G - Groove welded detail

F - Fillet welded detail

x - Static jack increased load Steel type - A588

TABLE 11 TRANSITION TEMPERATURE DATA FOR FLANGE PLATES

	Transition Tempe	rature (°F)
Material	(15 ft1b.)	(15 mil)
A36 (2" P1)	-16	-26
A588 (2" P1)	-24	-15
A514 (1-1/2" P1)	-133	-102
	(a)	

	Transition Temp	erature (°C)
Material	(20 Joule)	(0.38 mm)
A36 (51 mm P1)	-26.5	-32
A588 (51 mm P1)	-31	-26
A514 (38 mm P1)	-91.5	-74.5
	(b)	•

TABLE 12 REMAINING FATIGUE LIFE

	Beam	Remaining Fatigue Life*						
Steel	Number	(Cycles)						
	В2	1,168,100						
A514	B2A	576,500						
	В4	175,200						
A36	в4А	9,800						
A588	В6 .	408,000						
	в6А	669,600						

* Fatigue failure defined at an edge crack size = $\frac{3}{4}$ flange width

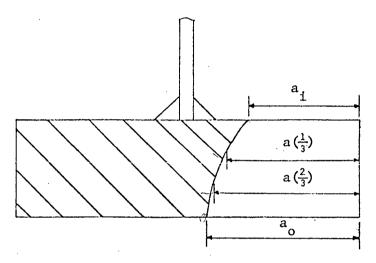


TABLE 13a CRACK SIZE MEASUREMENTS

	Measured Crack Sizes				Averag	r *			
	(1)	(2)	(3)	(4)	(1)+(2)+(3)+(4)	(1)+(2)	(2)+(3)	(3)+(4)	Correction
Beam	a i	$a'(\frac{1}{3})$	$a'(\frac{2}{3})$	a o	4	2	2	2	P1. Stress
Number	(in.)	(in.)	(in.)	(in.)	aave	a _T	a _M	В	(in.)
В2	0.60	0.90	1.17	1.26	0.98	0.75	1.04	1.21	0.09
B2A	1.37	1.64	1.78	1.80	1.65	1.51	1.71	1.79	0.14
В4	2.92	3.12	3.32	. 3.38	3.19	3.02	3.22	3.35	0.41
B4A	4.62	4.90	5.03	4.93	4.87	4.76	4.96	4.98	0.47
В6	2.97	2.85	2.58	2.19	2.65	2.93	2.72	2.39	1.27
ВбА	0.93	1.41	1.82	1.96	1.53	1.17	1.61	1.87	0.10

^{*} Correction used at critical flange $\frac{1}{3}$ thickness (see Table 14)

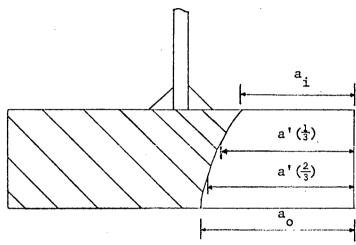


TABLE 13b CRACK SIZE MEASUREMENTS

	Measured Crack Sizes					Averaged Crack Sizes (mm)				
	(1)	(2)	(3)	(4)	(1)+(2)+(3)+(4)	(1)+(2)	(2)+(3)	(3)+(4)	Correction	
Beam	a	$a'(\frac{1}{3})$	$a'(\frac{2}{3})$	a _o	4	2	2	2	P1. Stress	
Number	(mm)	(mm)	(mm)	(mm)	a _I	a _T	a _M	В	(mm)	
B2	15	23	30	32	25	19	26	31	2	
B2A	35	42	45	46	42	38	43	45	4	
В4	74	79	84	86	81	77	82	85	10	
B4A	117	124	128	125	124	120	126	126	12	
В6	75	72	66	.56	67	74	69	61	32	
B6A	24	36	46	50-	39	30	41	48	3	

^{*} Correction used at critical flange $\frac{1}{3}$ thickness (see Table 14)

TABLE 14a STRESS INTENSITY ESTIMATES

Beam No./ Flange Thickness Level		a+r y	(1) K _{AS} (ksi√in)		(3) K _{LW} (ksi√in)	K _{WR}	(1)+(2)+ (3)+(4) K (ksi√in)
B2 (o _{yd} =15	5.6 ksi)			 			
TOD	46.5	0.78	74	-28	21	NA	67
MID		1.10	92	-16	17	NA	93
*BOT		1.30	101	-3	16	NA	114
B2A(o _{yd} =15	3.5 ksi)	· · · · · · · · · · · · · · · · · · ·					
TOP	51.2	1.56	118	50	15	NA	83
MID		1.80	130	-29	14	NA	115
*BOT	54.2	1.93	139	- 9	14	NA	144
B4 (o _{yd} =65	.5 ksi)						
*TOP		3.43	62	39	4	NA	105
MID	17.5	3.54	- 66	. 23	4	NA	93
BOT	18.2	3.57	69	4	4	NA	77
B4A (σ _{yd} =67.	7 ksi)						
TOP		5.20	103	15	5	-12	112
MID	18.7	5.43	116	0	5	-6	115
*BOT	19.4	5.45	102	9	5	0	116
B6 (o _{yd} =79.	3 ksi)						
*TOP		4.20	110	83	30	NA	223
MID		2.92	85	-3	6	NA	88
вот	27.0	2.54	81	-8	6	NA	79
B6A (σ _{yd} =84.	l ksi)						
TOP	25.0	1.18	49	-38	8	NA ·	19
MID	26.0	1.64	61	-30	7	ŃА	38
*BOT	27.0	1.99	70	-11	7	NA	66

^{*} Denotes critical flange thickness level σ_{yd} = Yield stress at test temperature and loading rate (Eq. 4)

TABLE 14b STRESS INTENSITY ESTIMATES

Beam No./ Flange Thickness Level		Crack Size a+r y (mm)	KAS	(2) KRS (MPa√m)	K _{LW}	K _{WR}	(1)+(2)+ (3)+(4) K (MPa√m)			
B2 (o _{yd} =10	73 MPa)									
TOP	321	20	81	-31	23	NA	73			
MID	330	30	101	-18	19	NA	102			
*BOT	339	33	111	-3	18	NA	126			
B2A (o _{yd} =1	.058 MPa)			·						
TOP	353	40	130	-55	17	NA	92			
MID	363	46		-32		NA	126			
*BOT	374	49	153	-10	15	NA	158			
B4 (σ _{yd} =	452 MPa)									
*TOP	116	87	68	43	4	NA	115			
MID	121	90	. 73	25	4	NA	102			
BOT	125	91	76	4	4	NA	84			
B4A (o _{yd} =	= 467 MPa)									
TOP	124	132		17	6	-13	123			
MID	129	138	128	0	6	-7	127			
*BOT	134	138	112	10	6	0	128			
B6 (σ _{yd} =	547 MPa)									
*TOP	172	91	121	91	33	NA	245			
MID	179	74	94	-3	7	NA	98			
вот	186	65	89	- 9	7	NA	87			
B6A (ogd =	$B6A (\sigma_{vd} = 580 MPa)$									
TOP	172	30	54	-42	9	NA	21			
MID	179	42	67	-33	8	· NA	42			
*BOT	186	51	77	-12	8	NA	73			

^{*} Denotes Critical Flange Thickness Level

 $[\]sigma_{yd}$ = Yield stress at test temperature and loading rate (Eq. 4)

FIGURES

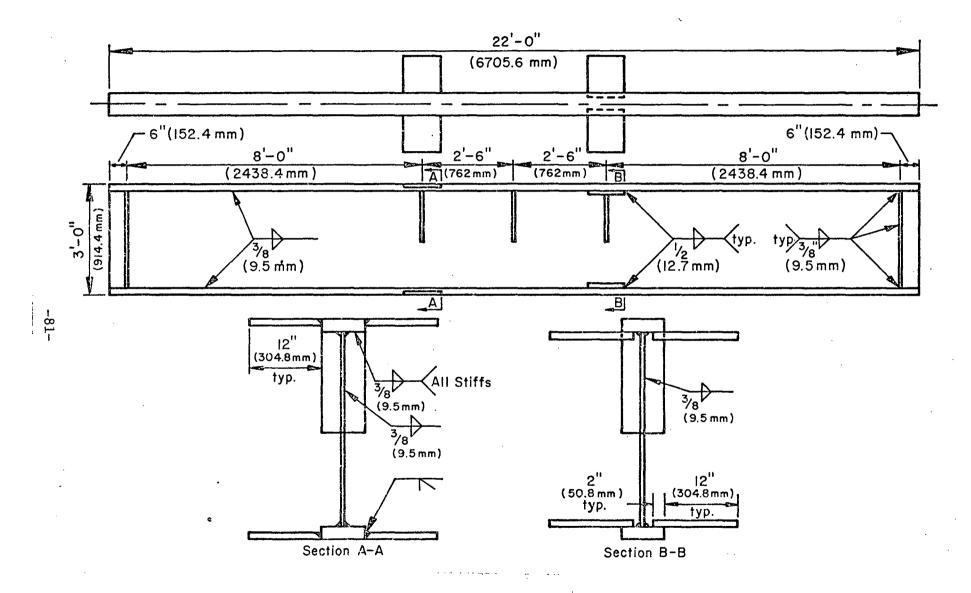


Fig. 1. Beam B4

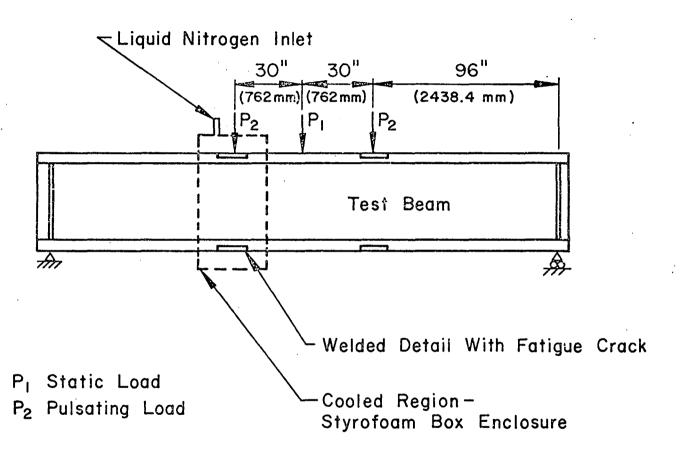


Fig. 2 Schematic of Fracture Test Setup

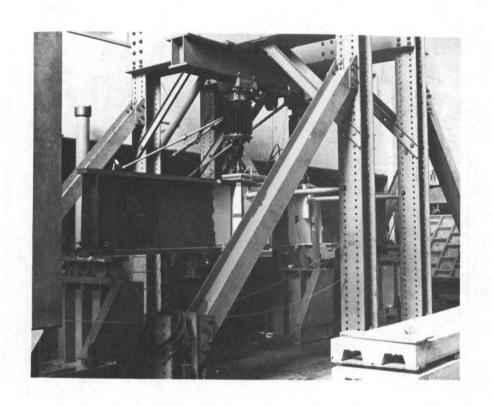
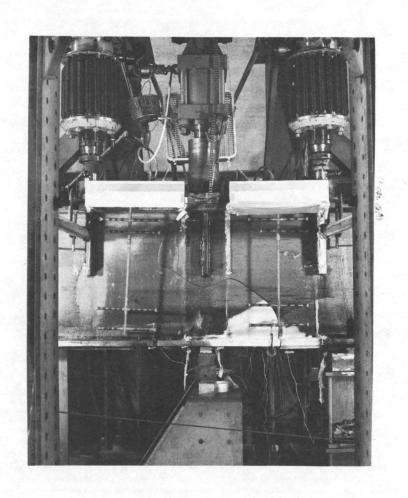


Fig. 3 Photograph of Test Set-Up

Fig. 4 Positions of Temperature and Strain Gages



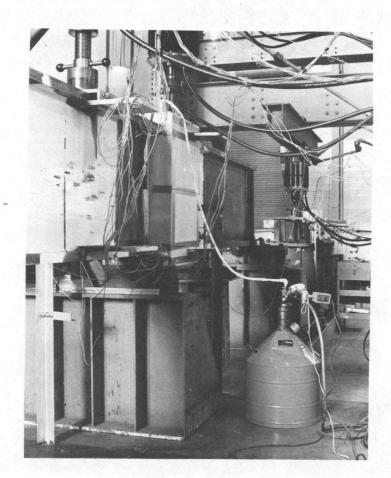


Fig. 5 Photographs of Cooling Network

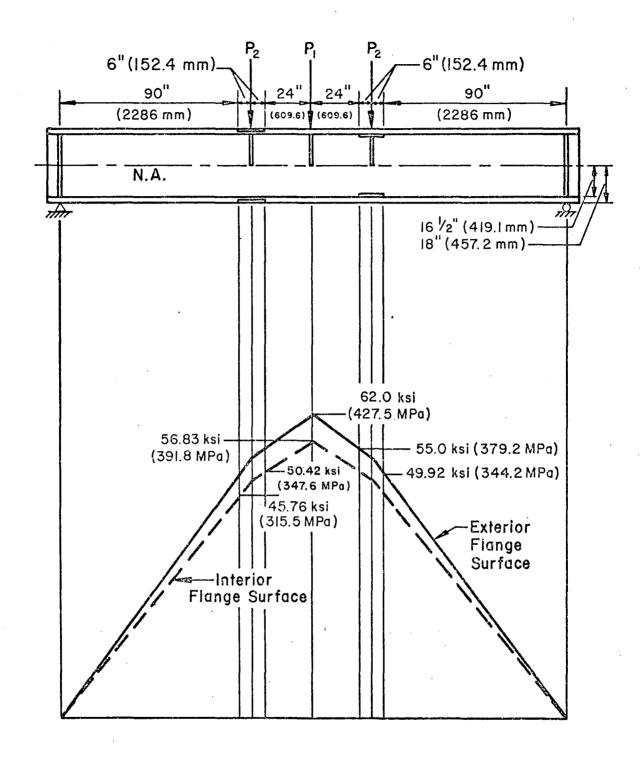


Fig. 6a Maximum Stress on Interior and Exterior Flange Surfaces of Beams B2 and B2A (A514)

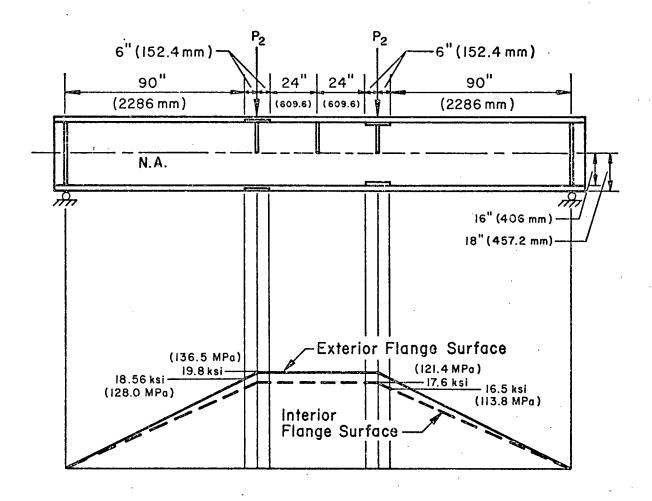


Fig. 6b Maximum Stress on Interior and Exterior Flange Surfaces of Beams B4 and B4A (A36)

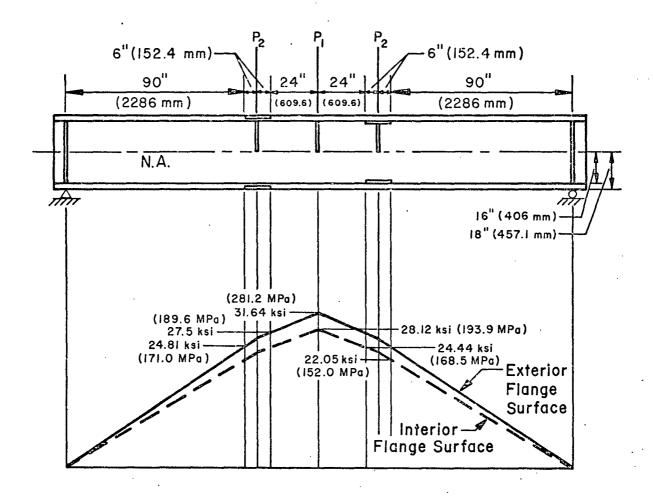
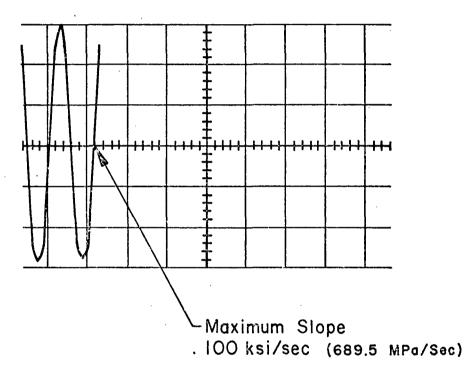


Fig. 6c Maximum Stress on Interior and Exterior Flange Surfaces of Beams B6 and B6A (A588)



Note: Horizontal Scale: 0.2 sec/cm

Fig. 7 Strain Rate Trace at Fracture

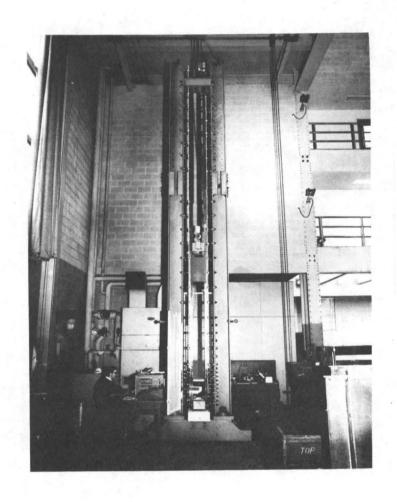
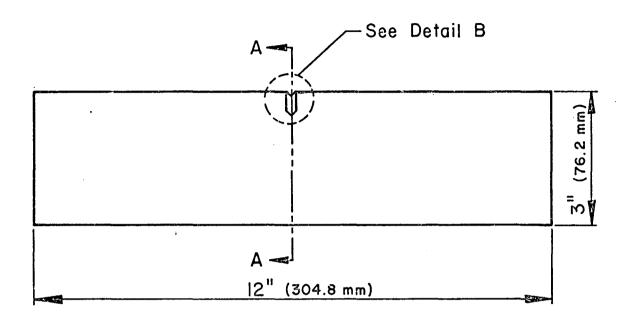


Fig. 8 Photograph of Lehigh Drop Weight Test Machine



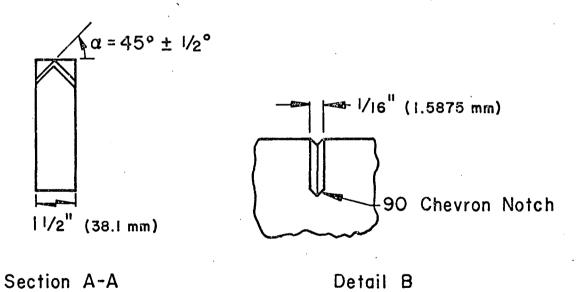
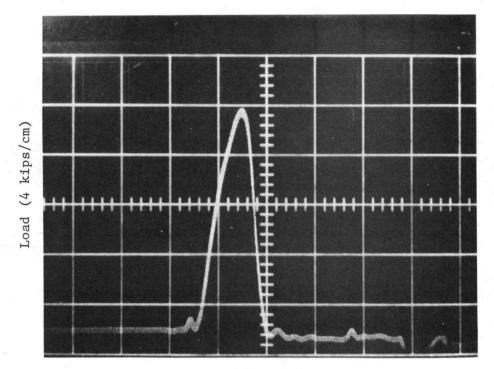
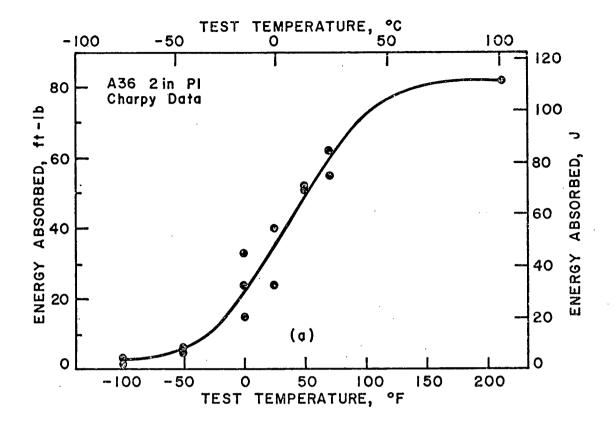


Fig. 9 Three-Point Bend Specimen



Time (0.5 ms/cm)

Fig. 10 Typical Load Time Relationship



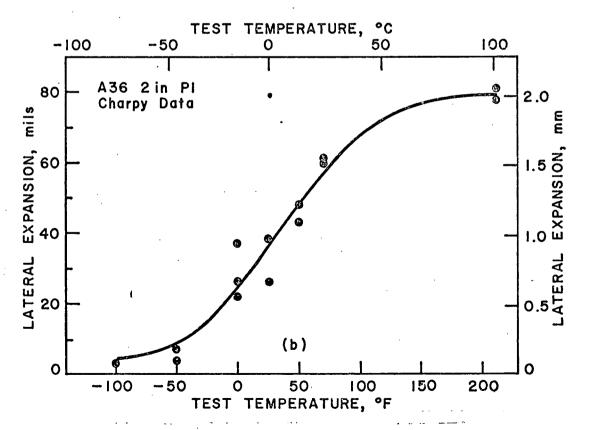


Fig. 11 Plotted CVN Data (A36/2 in. P1)

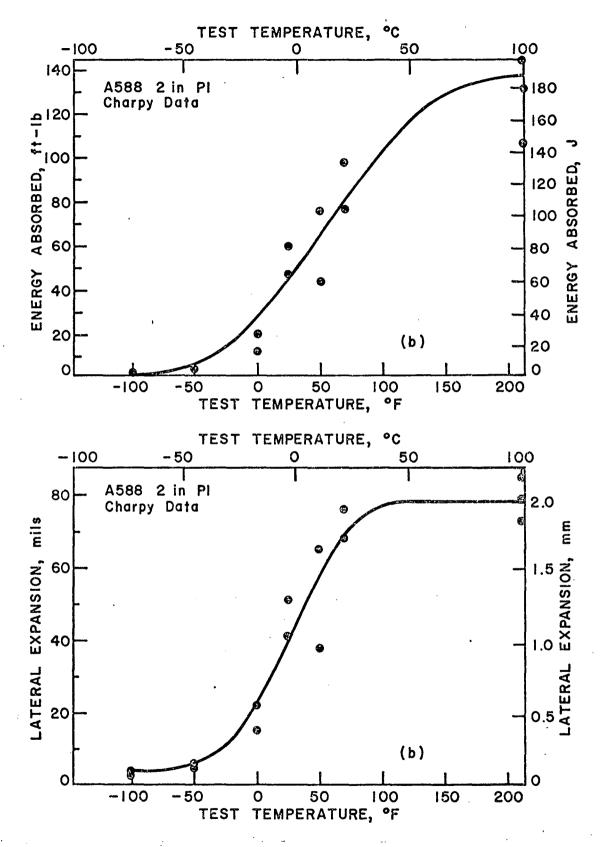


Fig. 12 Plotted CVN Data (A588/2 in. Pl)

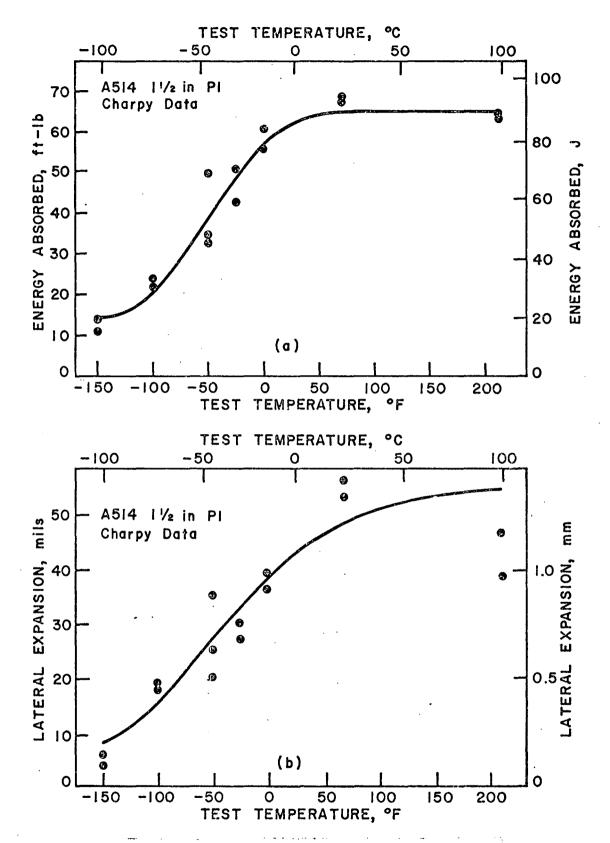


Fig. 13 Plotted CVN Data (A514/ $1\frac{1}{2}$ in. P1)

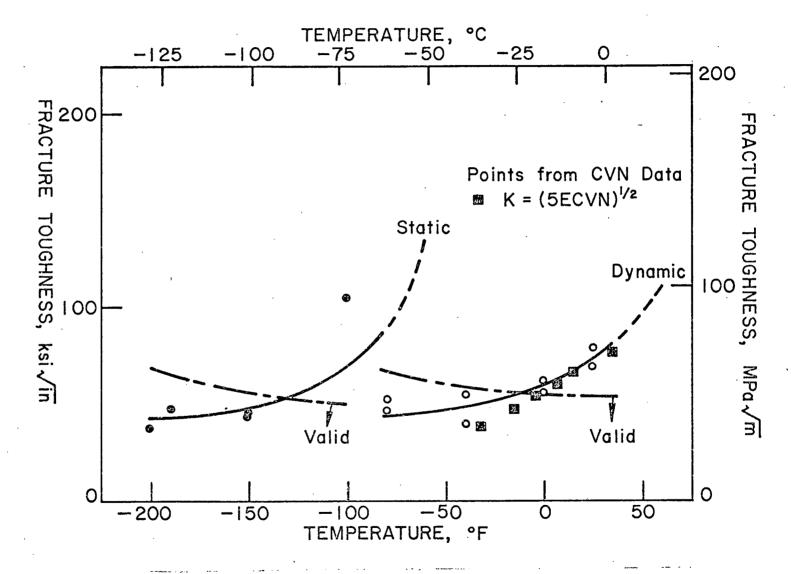


Fig. 14a Fracture Toughness vs. Temperature (A36/2 in. Pl)

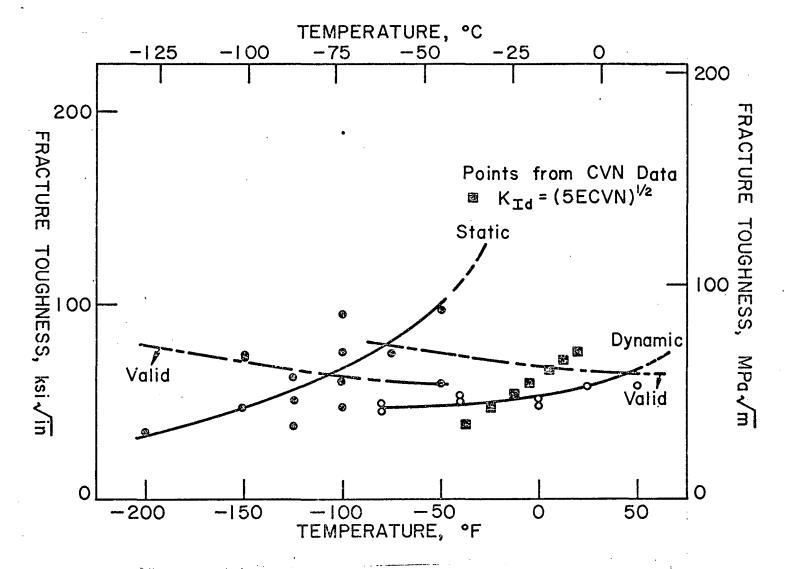


Fig. 14b Fracture Toughness vs. Temperature (A588/2 in. Pl)

Fig. 14c Fracture Toughness vs. Temperature (A514/12 in. Pl)

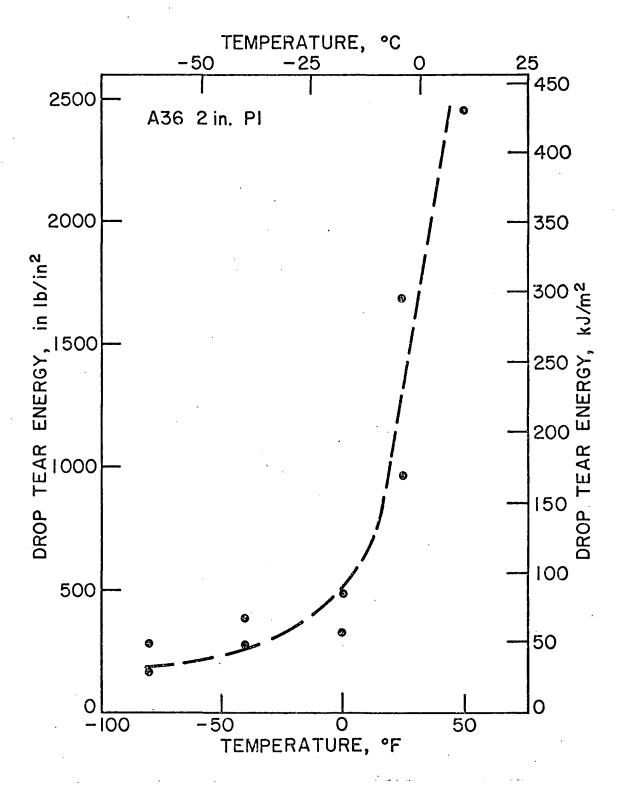


Fig. 15a Drop Tear Energy vs. Temperature (A36/2 in. Pl)

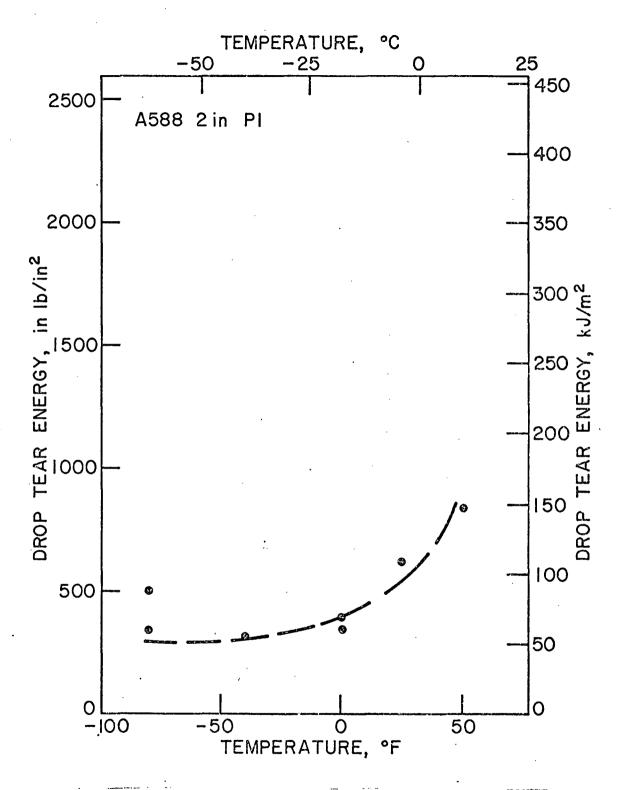


Fig. 15b Drop Tear Energy vs. Temperature (A588/2 in. P1)

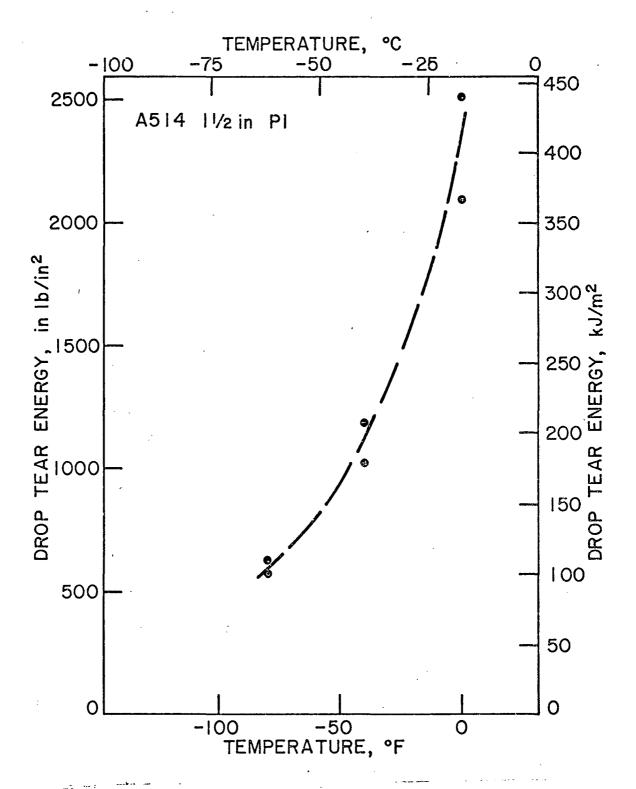


Fig. 15c Drop Tear Energy vs. Temperature (A514/ $1\frac{1}{2}$ in. P1)

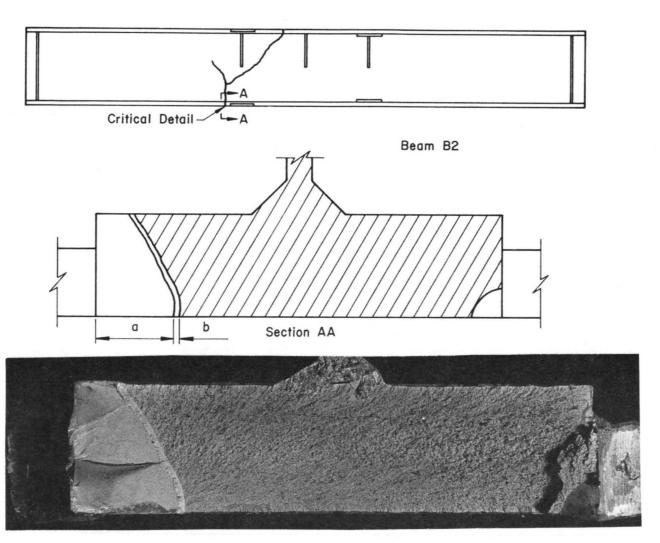


Fig. 16 Fatigue and Fracture Surface, B2 (A514)

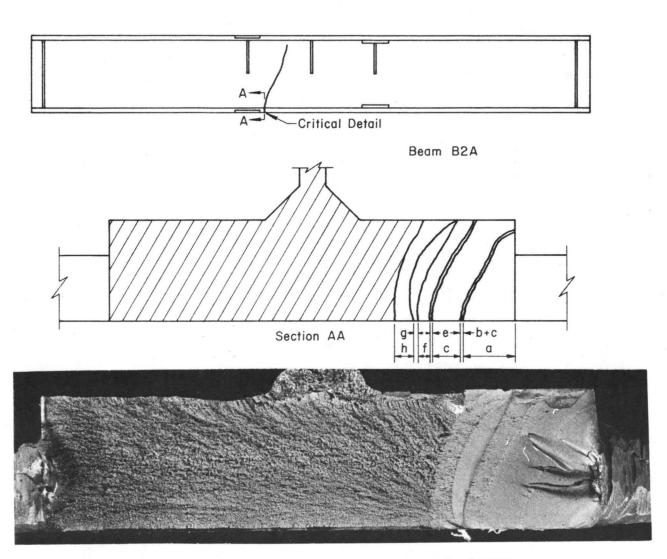


Fig. 17 Fatigue and Fracture Surface, B2A (A514)

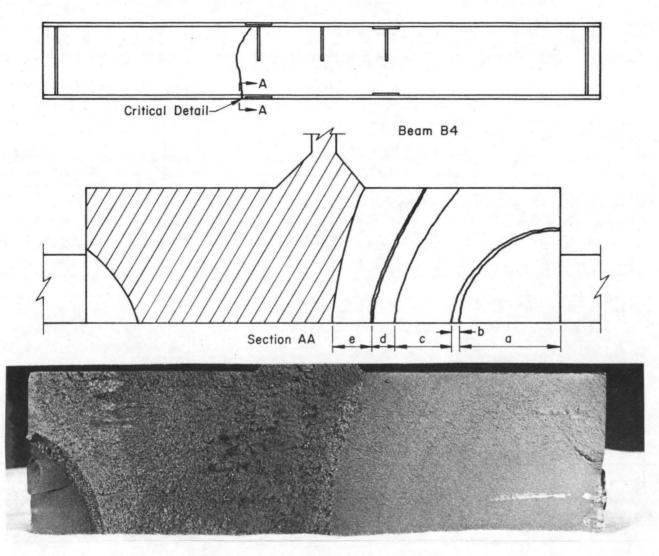


Fig. 18 Fatigue and Fracture Surface, B4 (A36)

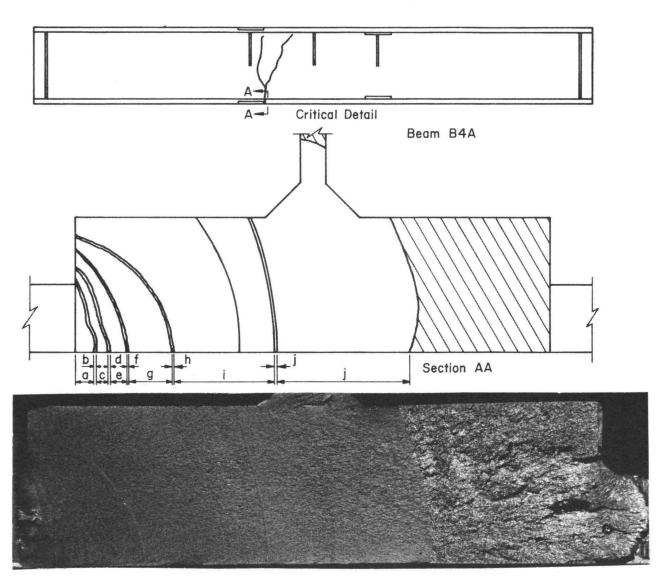


Fig. 19 Fatigue and Fracture Surface, B4A (A36)

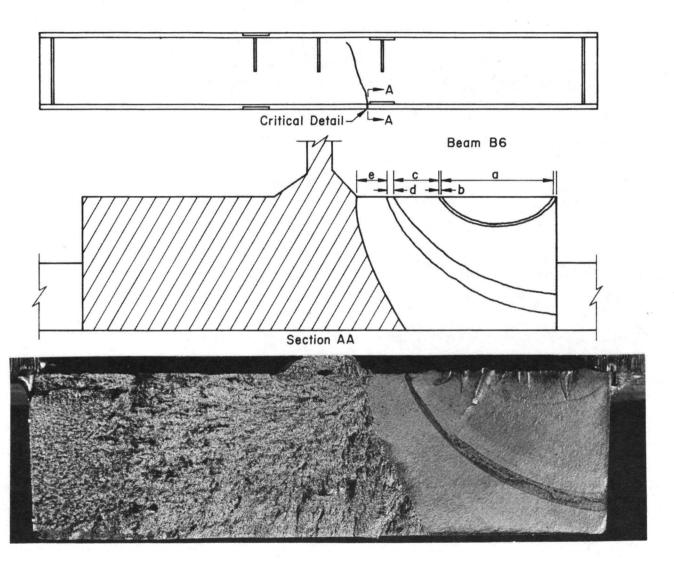


Fig. 20 Fatigue and Fracture Surface, B6 (A588)

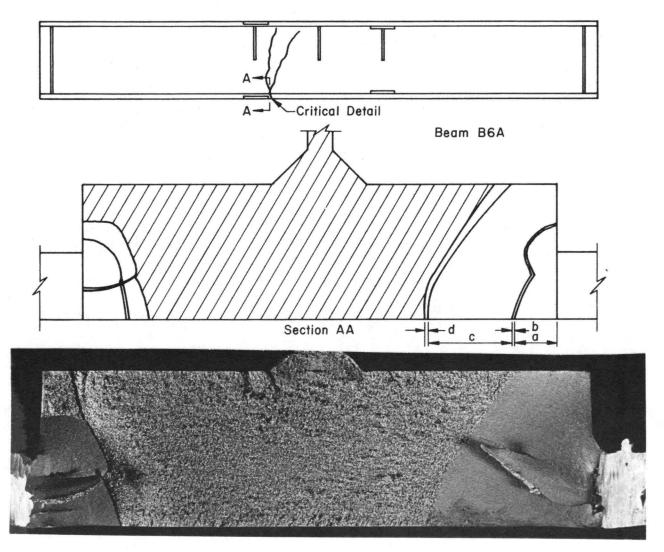
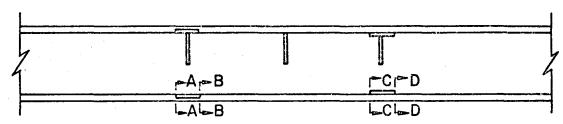


Fig. 21 Fatigue and Fracture Surface, B6A (A588)

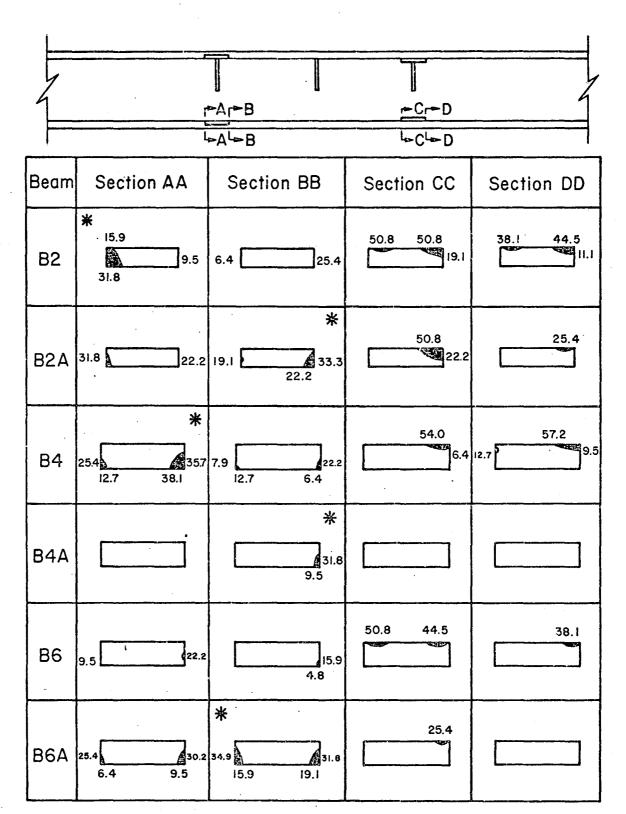


Beam	Section AA	Section BB	Section CC	Section DD
B2	* 5/8" 3/8" 1 1/4"	3/4"	2" 2" " ³ / ₄ "	½" ¾" ¾"
B2A	1½" 7/8" 7/8"	* 3/4"	2" 7/8"	<u> </u> "
B4	* 1" 1 ¹³ / ₃₂ 1 ² / ₂ "	5/ ₁₆	2 1/8"	½" 2 ½" 3%
В4А		* 1½4 3/8		
В6	3," 7,"	3,"8 3,"8	2" 3/4" *	11/2"
В6А	"	* 13/8 11/4 5/8" 3/4"	- 111	

Table Scale | 1/2"=1'-0"

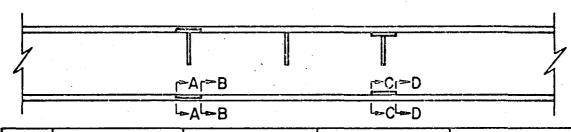
* Beam Fractured At This Section

Fig. 22a Fatigue Cracks at Two Million Cycles



** Beam Fractured At This Section Crack Measurements In mm

Fig. 22b Fatigue Cracks at Two Million Cycles

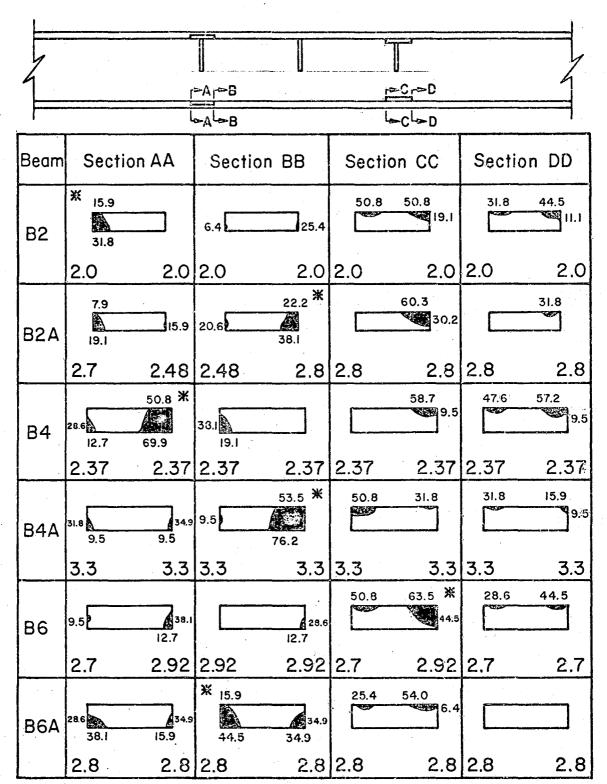


Beam	Section AA	Section BB	Section CC	Section DD
B2	* 5/8" [] 11/4"	1" 1" 14"	2" 2" 3/4"	11/4" 13/4" 7/16"
	2.0 2.0			2.0 2.0
B2A	⁵ /16" 5/8" 3/4"	7 ₈ " * 13/16	2¾ ₈ "	11/4"
		2.48 2.8		
B4	2" * // ₈		2 ⁵ / ₁₆ "	17/8" 21/4"
	2.37 2.37	2.37 2.37	2.37 2.37	
В4А	\frac{1}{4}\frac{1}{3}\frac{1}{8}\frac{3}{8}\frac{1}{8}\frac{3}{8}\frac{1}{8	√2" ※ 3/8"	2" 1 ¹ / ₄ "	1 ¹ /4" 5/8" 3/8
	3.3 3.3	3.3 3.3	3.3 3.3	3.3 3.3
В6	3/8" 11/2 5/8"	¹ / ₈ "	2" 2½" *	1%" 134"
	2.7 2.92	2.92 2.92	2.7 2.92	2.7 2.7
В6А	11/2" 5/8"	13/4" 13/8"	I" 2 ¹ /8"	
	2.8 2.8	2.8 2.8	2.8 2.8	2.8 2.8

Table Scale 11/2=1'-0"
Measurements Taken At Cycles Listed At Bottom
Of Cross Section (in millions of cycles)

** Beam Fractured At This Section

Fig. 23a Fatigue Cracks Prior to Last Fracture Test



Measurements Taken At Cycles Listed At Bottom Of Cross Section (In millions of cycles)

** Beam Fractured At This Section

Crack Measurements In mm

Fig. 23h Fatigue Cracks Prior to Last Fracture Test

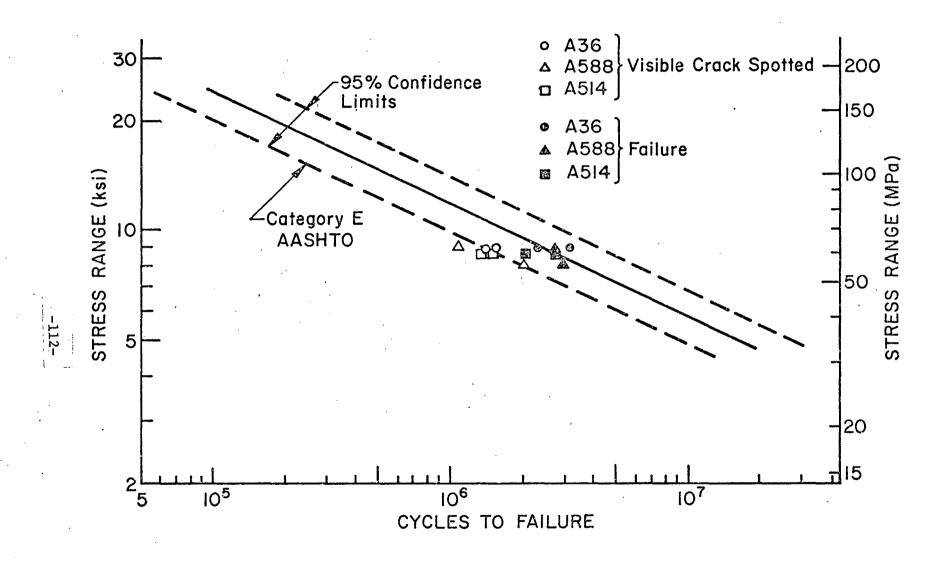


Fig. 24 Category E S-N Plot (Lateral Attachment Details)

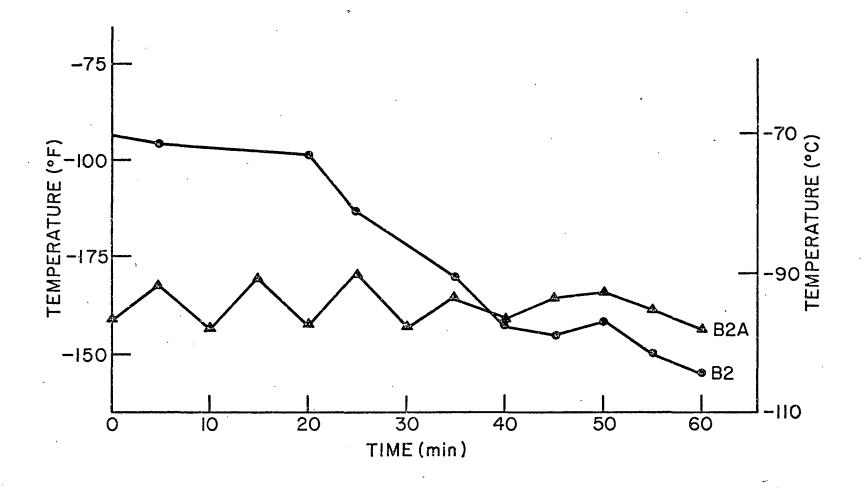


Fig. 25 Critical Detail Temperature/60 Minutes Prior to Fracture (A514)

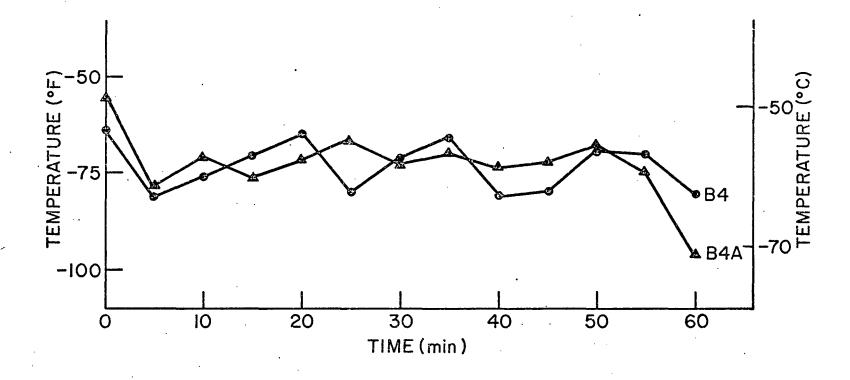


Fig. 26 Critical Detail Temperature/60 Minutes Prior to Fracture (A36)

Fig. 27 Critical Detail Temperature/60 Minutes Prior to Fracture (A588)

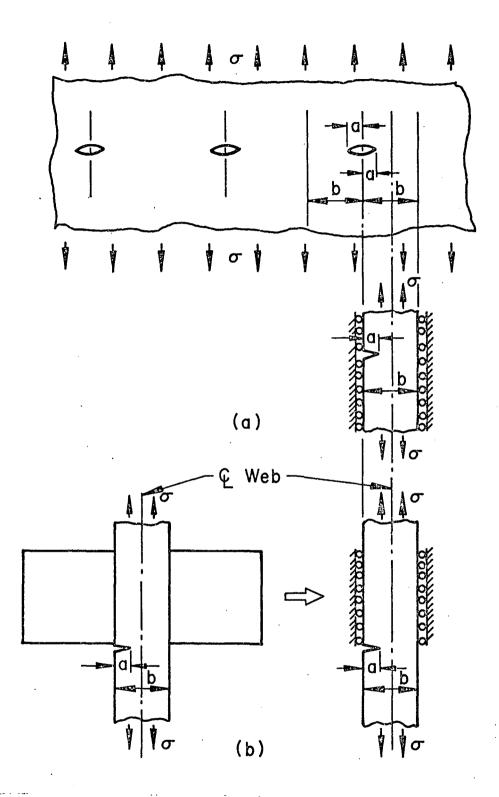


Fig. 28 Stress Intensity Model for Flange Edge Crack for Applied Stress

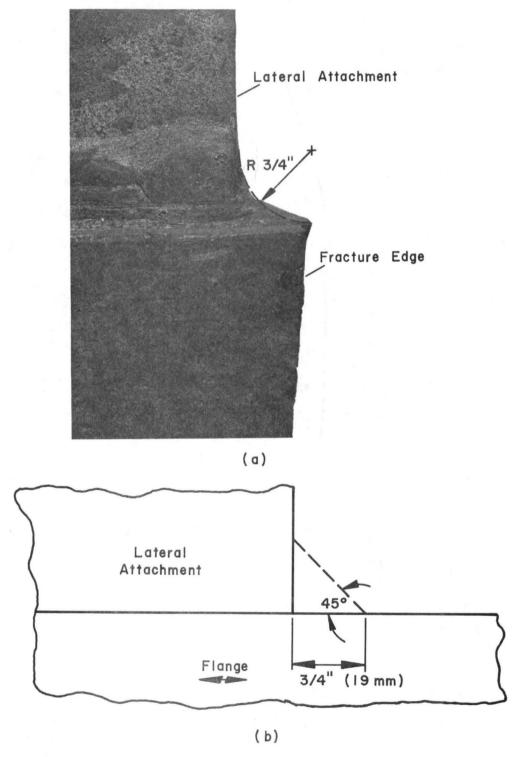


Fig. 29 Groove Weld Reentrant Corners

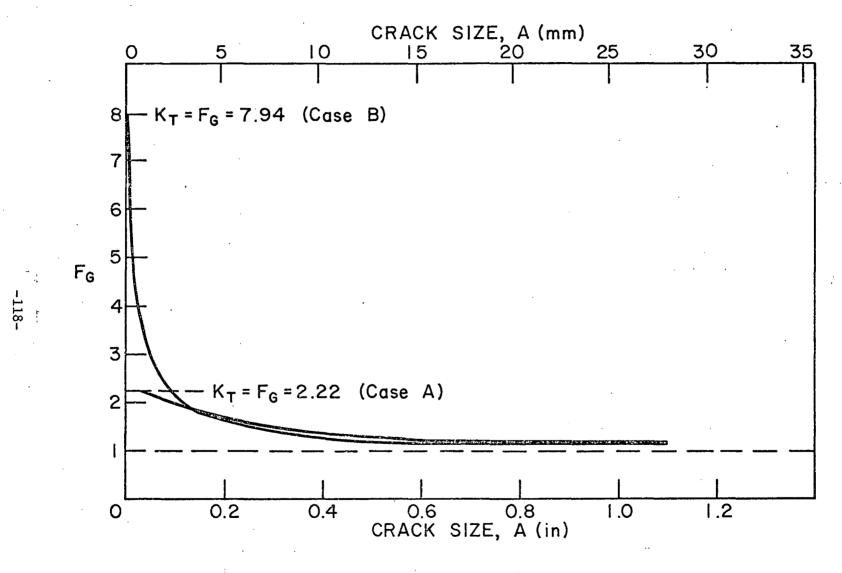


Fig. 29c Stress Concentration Decay with Crack Size

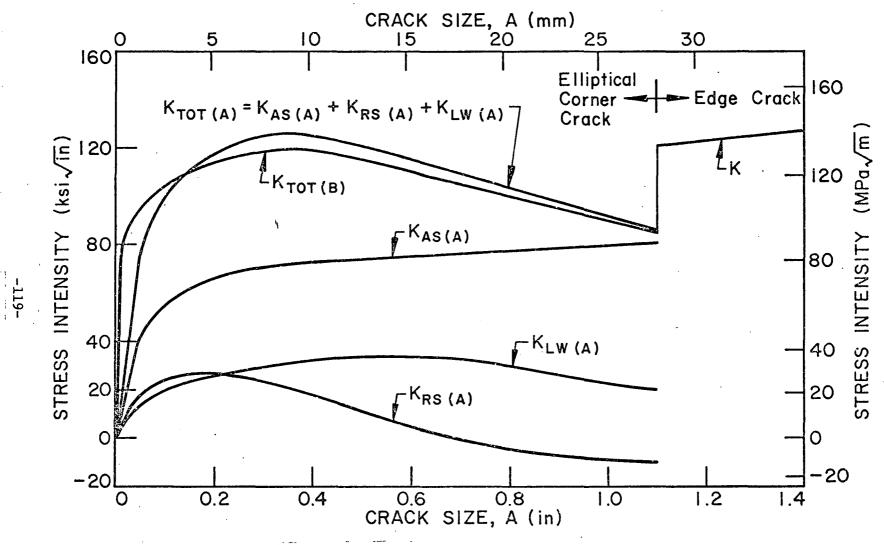


Fig. 30 Stress Intensity vs. Crack Size for Elliptical Corner Cracks (B2A, A514)

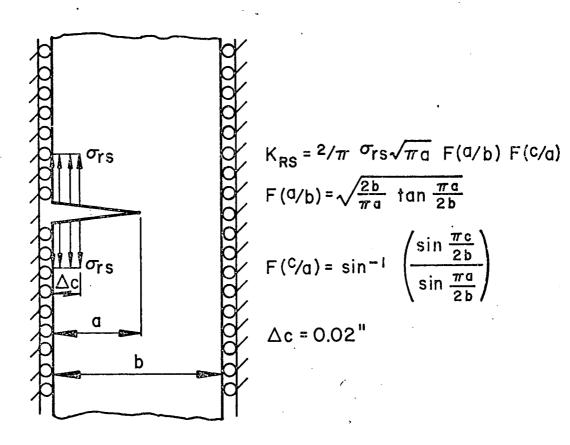


Fig. 31 Superposition Model for Residual Stresses

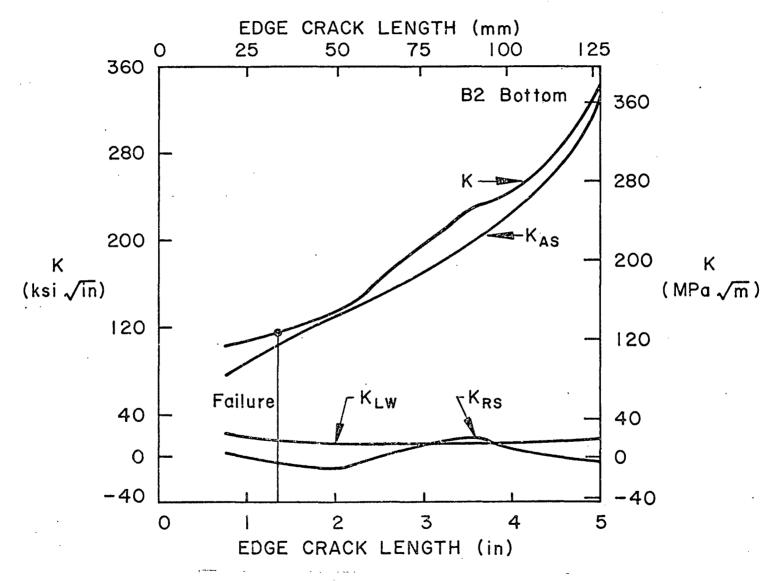


Fig. 32 K vs. Edge Crack Size (B2, A514)

Fig. 33 K vs. Edge Crack Size (B2A, A514)

Fig. 34 K vs. Edge Crack Size (B4, A36)

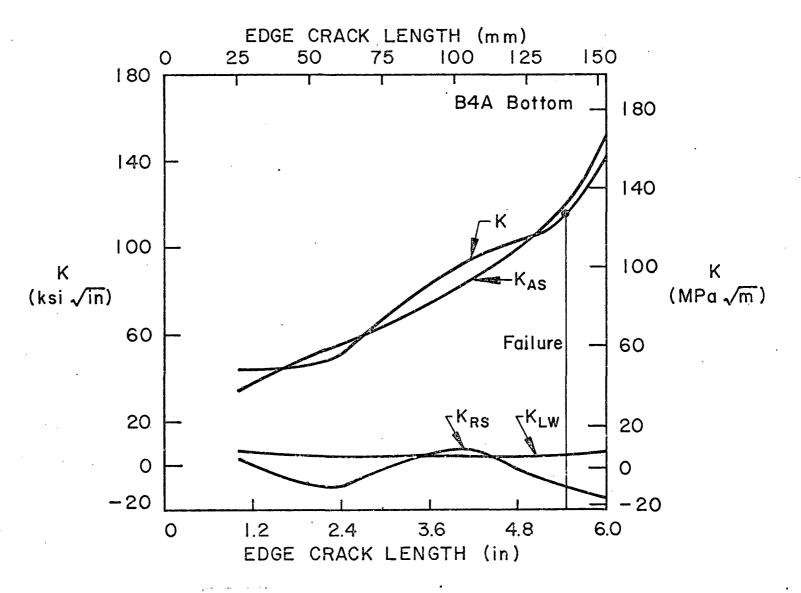


Fig. 35 K vs. Edge Crack Size (B4A, A36)

Fig. 36 K vs. Edge Crack Size (B6, A588)

Fig. 37 K vs. Edge Crack Size (B6A, A588)

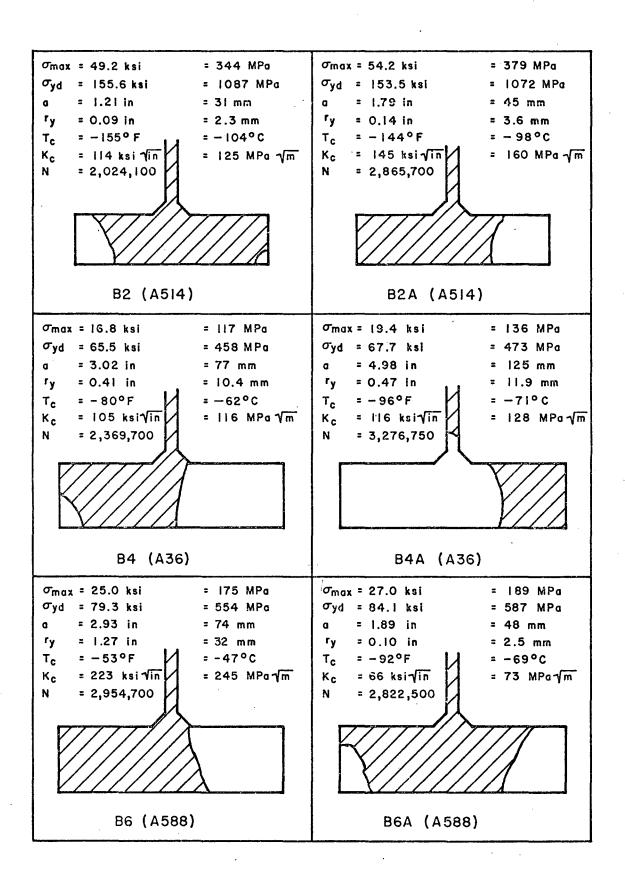


Fig. 38 Fracture Surface Sketches and Data Summary

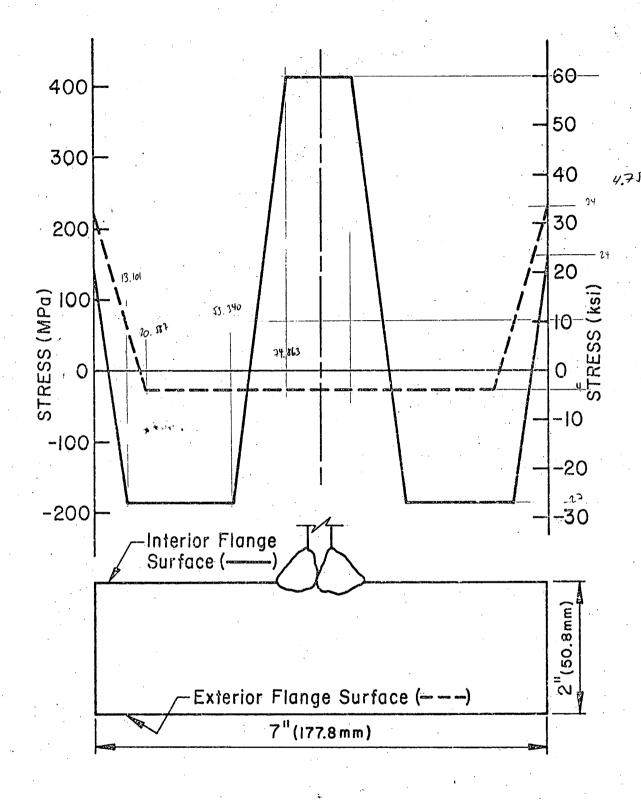


Fig. 39 Estimated Residual Stress Distribution for A36 Flange

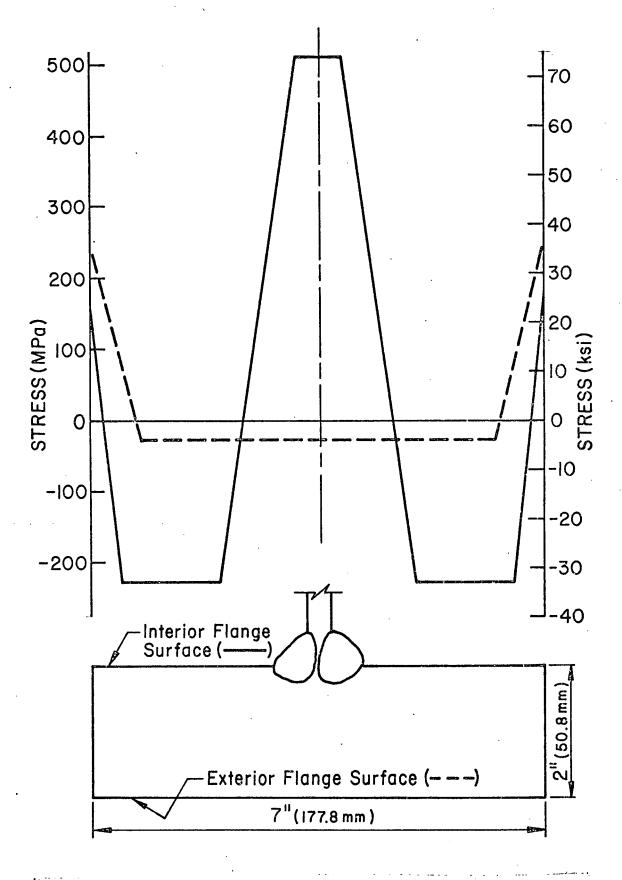


Fig. 40 Estimated Residual Stress Distribution for A588 Flange

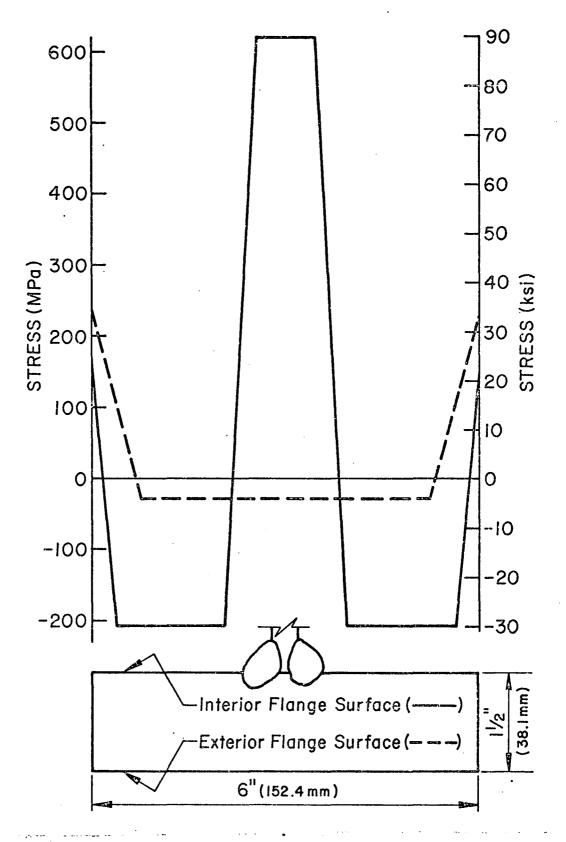
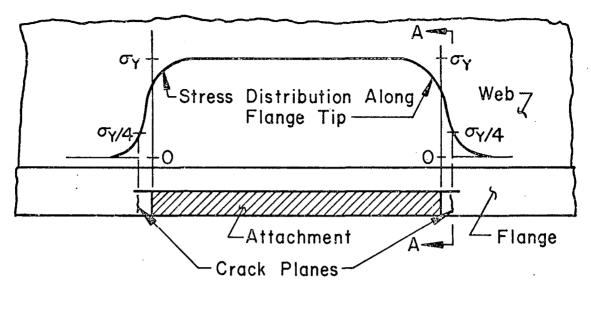
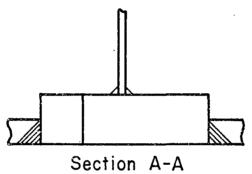


Fig. 41 Estimated Residual Stress Distribution for A514 Flange





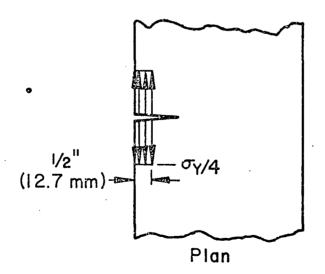
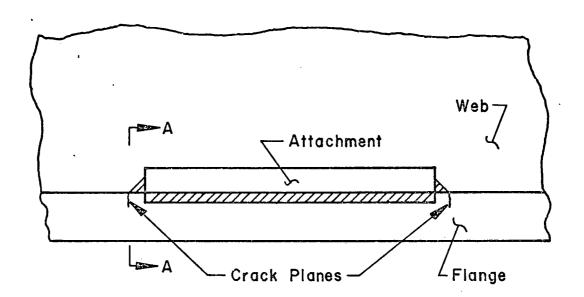


Fig. 42a Assumed Residual Stress Distribution Along Flange Tip for Groove Weld Lateral Attachment



//// Fillet Welds

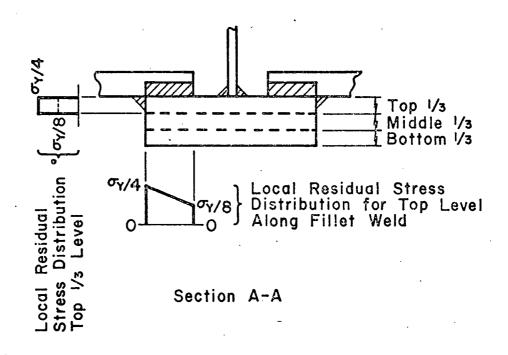


Fig. 42b Assumed Local Residual Stress Distribution for Fillet Weld Lateral Attachment

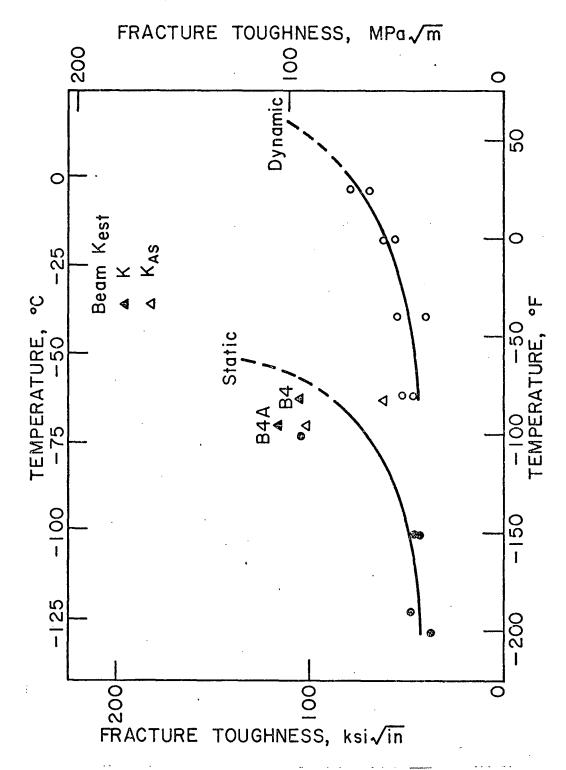


Fig. 43 Correlation of Beam K and Material
Toughness Characterization B4, B4A (A36)

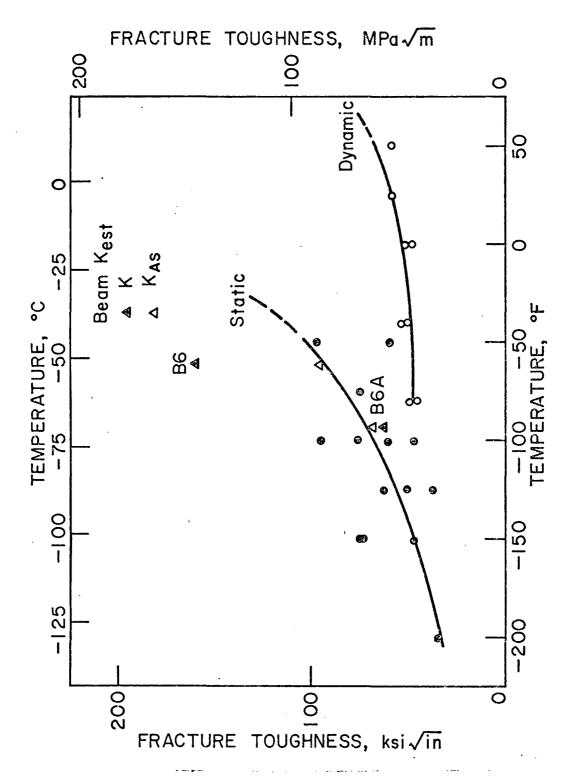


Fig. 44 Correlation of Beam K and Material Toughness Characterization B6, B6A (A588)

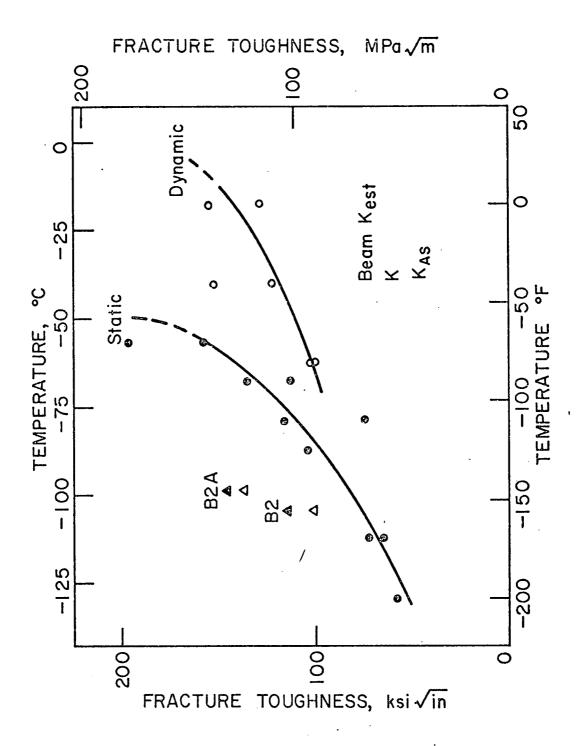
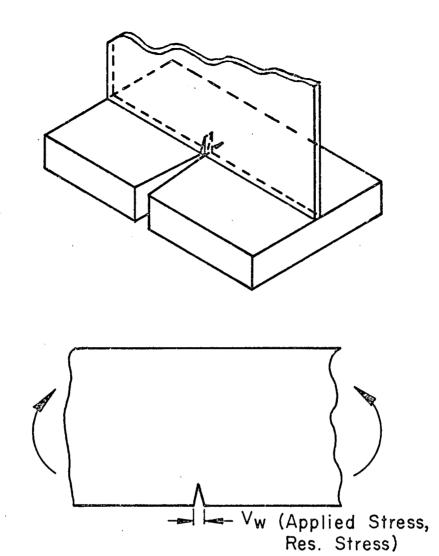


Fig. 45 Correlation of Beam $K_{\rm est}$ and Material Toughness Characterization B2, B2A (A514)



Web Crack Opening

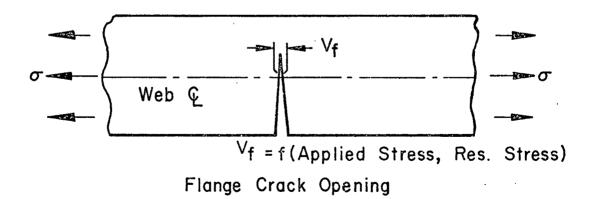


Fig. 46 Flange and Web Crack Interaction

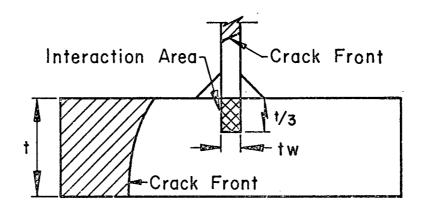


Fig. 47 Assumed Interaction Area for Web Restraint Analysis

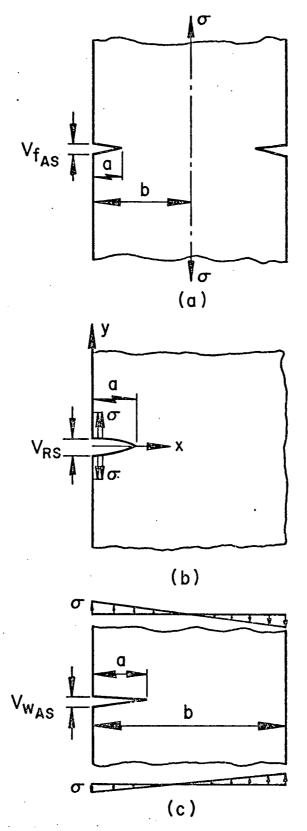


Fig. 48 Models Used for Web Restraint Analysis

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