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# **Design of active Nickel single-atom decorated MoS<sub>2</sub> as a pH-universal catalyst for hydrogen evolution reaction**

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**Abstract:** MoS<sub>2</sub> has been considered as a potential alternative to Pt-based catalysts in the hydrogen evolution reaction (HER). However, the presence of the inactive in-plane domains limits their surface area specific activity of the catalyst. Here, we demonstrate a new approach for activating these inactive sites and therefore dramatically enhancing the activity. We discover that decorating single Ni atom on MoS<sub>2</sub> can increase the HER activity in both alkaline and acidic conditions. Experimental and theoretical results indicate that single Ni atom modifiers are inclined to single dispersion in the S-edge sites and H-basal sites of MoS<sub>2</sub>, resulting in a favorable change in the adsorption behavior of H atoms on their neighboring S atoms and subsequently the HER activity. Consequently, the single-Ni-atom decorated MoS<sub>2</sub> (Ni<sub>SA</sub>-MoS<sub>2</sub>) achieved cathodic current density of 10 mA cm<sup>-2</sup> at overpotentials of 98mV and 110 mV in 1 M KOH and 0.5 M H<sub>2</sub>SO<sub>4</sub>, respectively. The dispersion of the Ni single atoms in the Ni<sub>SA</sub>-MoS<sub>2</sub> is unaffected upon 2000 cycles in both acidic and alkaline conditions. This single atom decorating approach presents a facile and promising pathway for designing active electrocatalysts for energy conversion and storage.

**Keywords:** Ni Single atom, MoS<sub>2</sub>, Hydrogen Evolution Reaction, Active sites

## 1. Introduction

Water splitting and hydrogen storage have the potential to boost the implementation of green energy technologies due to its high energy density and environmental-friendliness.[1,2] The electrocatalytic hydrogen evolution reaction (HER) is regarded as one of the most effective pathways to produce hydrogen.[3–5] Currently, the best known HER electrocatalysts are based on noble metals, in particular, platinum.[6] However, upon implementation in electrolyzers and fuel cells, the cost of the utilization of Pt in these technologies would limit their commercialization and industrialization due to its scarcity and high cost.[7] Therefore, the development of cost-effective HER electrocatalysts with high activities is of great significance to the hydrogen economy[8–10]

The identification of MoS<sub>2</sub>-based materials as efficient HER electrocatalysts presented a viable class of nonprecious materials to the field.[11–13] Nevertheless, the catalytic activities of MoS<sub>2</sub> based materials are still inferior to the traditional Pt-based catalysts, stemming from its low active sites to surface area ratio and poor conductivity.[14] The HER catalytic sites of MoS<sub>2</sub> have been demonstrated to be the low coordination metallic edges whose electronic structure significantly differs from the catalytically inactive basal planes.[14] Thus, many strategies have been employed targeting the improvement in catalytic performance of MoS<sub>2</sub>, such as controlling morphology and structure and compositing with other materials so as to expose more edge sites and accelerate electron transport speeds.[15–20] Decorating MoS<sub>2</sub> with foreign materials, such as Pt, Pd and metal oxide particles, has been shown to be an effective method to accelerate HER catalytic performance due to the activation of the basal plane via synergistic effect between MoS<sub>2</sub> and the decorating materials.[21–25] However, most of the currently reported decorating materials refer to metal particles, quite often of expensive noble metals whilst other more abundant metals suffered from instability under acidic conditions.[26,27] To date, efforts to develop low-cost MoS<sub>2</sub>-based HER catalysts with efficient and stable performance, in both acidic and alkaline conditions have faltered.

In this work, we propose an alternative strategy to engineer MoS<sub>2</sub> nanosheets, resulting in significant enhancement for the HER catalysis, via decorating with single atoms of Ni in specific sites. The preparation procedure involves the synthesis of MoS<sub>2</sub> nanosheets supported on carbon cloth by hydrothermal method and subsequent single Ni atom decorating by wet-impregnation and calcination. Our results demonstrate that single atom Ni-decorated MoS<sub>2</sub> (Ni<sub>SA</sub>-MoS<sub>2</sub>/CC) catalysts exhibit superior electrocatalytic activities toward HER in both acidic and alkaline media, which is attributed to the activation of the sites at S-edge and the originally inert basal plane of MoS<sub>2</sub> by coordination of single Ni atoms. We also demonstrate the good stability of the Ni single atom decorated on the MoS<sub>2</sub> nanosheet through potential cycling.

## **2. Material and methods**

## 2.1 Materials preparation

### 1. Synthesis of MoS<sub>2</sub> nanosheets array on carbon clothes (MoS<sub>2</sub>/CC)

0.25 g Na<sub>2</sub>MoO<sub>4</sub> and 0.3 g thiourea was added into 18 ml H<sub>2</sub>O, after stirring for 10 min, the solution was transferred into 50 ml Teflon-lined stainless autoclave. After a piece of carbon cloth was immersed in the solution for 10 min, placed in the stainless autoclave was sealed and heated at 220 °C for 24 h. the product was rinsed with deionized (DI) water and ethanol three times.

### 2. Synthesis of Ni<sub>SA</sub>- MoS<sub>2</sub>/CC

A piece of MoS<sub>2</sub>/CC was immersed into NiCl<sub>2</sub> ethanol solution (4 mg/mL) for 10s, then taken out and dried at 80 °C for 1 h. The product was transferred to a stainless autoclave and heated at 300 °C for 1 h under the 10 % H<sub>2</sub>. The product was purified with DI water and ethanol.

### 3. Synthesis of Ni<sub>C</sub>-MoS<sub>2</sub>/CC

A piece of MoS<sub>2</sub>/CC was immersed into NiCl<sub>2</sub> ethanol solution (4mg/mL) for 10s, then taken out and dried at 80 °C for 1 h. The product was transferred to a stainless autoclave and heated at 600 °C for 2 h under the 10% H<sub>2</sub>. The product was purified with DI water and ethanol.

## 2.2 Characterization

The phase of the as-prepared products was identified by XRD with Cu K $\alpha$  radiation ( $\lambda = 1.5418 \text{ \AA}$ ). Scanning electron microscope (SEM) images were taken on a FEI Helios Nanolab 600i scanning electron microscope. Room-temperature HAADF-STEM imaging and Electron Energy Loss Spectroscopy (EELS) were conducted using a double Cs-corrected Themis Z S/TEM operated at an accelerating voltage of 60-300 kV. The XPS analyses were carried out on a PHI 5000 VersaProbe II spectrometer using monochromatic Al K(alpha) X-ray source.

## 2.3 Electrochemical testing

Electrochemical measurements of the as-prepared samples towards the hydrogen evolution reaction (HER) were conducted at room temperature in a standard three-electrode cell, with a 0.5 M H<sub>2</sub>SO<sub>4</sub> solution and 1M KOH solution were used as the electrolyte. A piece of freshly-made MoS<sub>2</sub>/CC, Ni<sub>SA</sub>- MoS<sub>2</sub>/CC or Ni<sub>C</sub>- MoS<sub>2</sub>/CC as the working electrode, a graphite rod as the counter electrode and a saturated calomel electrode (SCE) as the reference electrode. In all measurements, the SCE was calibrated with respect to RHE. In 0.5 M H<sub>2</sub>SO<sub>4</sub>, E (RHE) = E (SCE) + 0.241 V. In 1.0 M KOH, E (RHE) = E (SCE) + 1.068 V. LSV curves were conducted in electrolyte with a scan rate of 5 mV·s<sup>-1</sup>. All the potentials reported in our work were expressed vs. the RHE.

## 2.4 Simulation details and methods

All the calculations were performed by using density functional theory (DFT) calculations as implemented in the Vienna *ab initio* package (VASP). Spin-polarization was considered for all the simulations. The projector augmented wave (PAW) method was used to describe electron-ion interaction, while the generalized gradient approximation using the Perdew-Burke-Ernzerhof (PBE) functional was used to describe the electron exchange-correlation.[28] A plane wave basis was set up to an energy cutoff of 520 eV. A 30 Å vacuum space was constructed to avoid the periodical image interactions between two adjacent MoS<sub>2</sub> layers. The Brillouin zone was integrated using the Monkhorst-Pack scheme with 3×3×1 *k*-grid.[29] All the atomic positions and cell parameters were relaxed using a conjugate gradient minimization until the force on each atom is less than 0.01 eV/Å.

The adsorption energy ( $E_{\text{ads}}$ ) of Ni on monolayer MoS<sub>2</sub> and slab were calculated with  $E_{\text{ads}} = E_{\text{MoS}_2+\text{Ni}} - E_{\text{MoS}_2} - E_{\text{Ni}}$ , [30] where  $E_{\text{MoS}_2+\text{Ni}}$  and  $E_{\text{MoS}_2}$  are the total energies of MoS<sub>2</sub> with and without Ni adsorption, respectively.  $E_{\text{Ni}}$  is the energy of an isolated Ni atom.

The formation energy ( $E_f$ ) for the substitution of Mo with Ni was calculated with  $E_f = E_{\text{Ni-Mo}} - E_{\text{MoS}_2} + \mu_{\text{Mo}} - \mu_{\text{Ni}}$ , where  $E_{\text{Ni-Mo}}$  and  $E_{\text{MoS}_2}$  are the total energies of MoS<sub>2</sub> with and without Ni substitution, respectively.  $\mu_{\text{Mo}}$  and  $\mu_{\text{Ni}}$  were the chemical potentials of Mo and Ni atoms, respectively.  $\mu_{\text{Mo}}$  and  $\mu_{\text{Ni}}$  values were calculated with respect to the bulk state.

Gibbs free-energy of the adsorption atomic hydrogen was calculated using equation (1):

$$\Delta G_H = \Delta E_H + \Delta E_{ZPE} - T\Delta S_H \quad (1)$$

where  $\Delta E_{ZPE}$  and  $\Delta S_H$  are the zero-point energy and entropy difference of hydrogen in the adsorbed state and the gas phase, respectively. The hydrogen adsorption energy  $\Delta E_H$  is calculated with the following expression:

$$\Delta E_H = E_{\text{MoS}_2\text{-H}} - E_{\text{MoS}_2} - \frac{1}{2}E_{\text{H}_2} \quad (2)$$

where  $E_{\text{MoS}_2\text{-H}}$  and  $E_{\text{MoS}_2}$  are the total energies of MoS<sub>2</sub> with and without hydrogen adsorption, respectively.  $E_{\text{H}_2}$  is the energy of a gas phase hydrogen molecule.

The calculated frequencies of H<sub>2</sub> gas were 4345 cm<sup>-1</sup>, 58 cm<sup>-1</sup>, and 42 cm<sup>-1</sup>. The contribution from the configurational entropy in the adsorbed state is small and is neglected. So the entropy of hydrogen adsorption as  $\Delta S_H = \frac{1}{2}S_{\text{H}_2}$  where  $S_{\text{H}_2}$  is the entropy of molecule hydrogen in the gas phase at standard conditions[31].

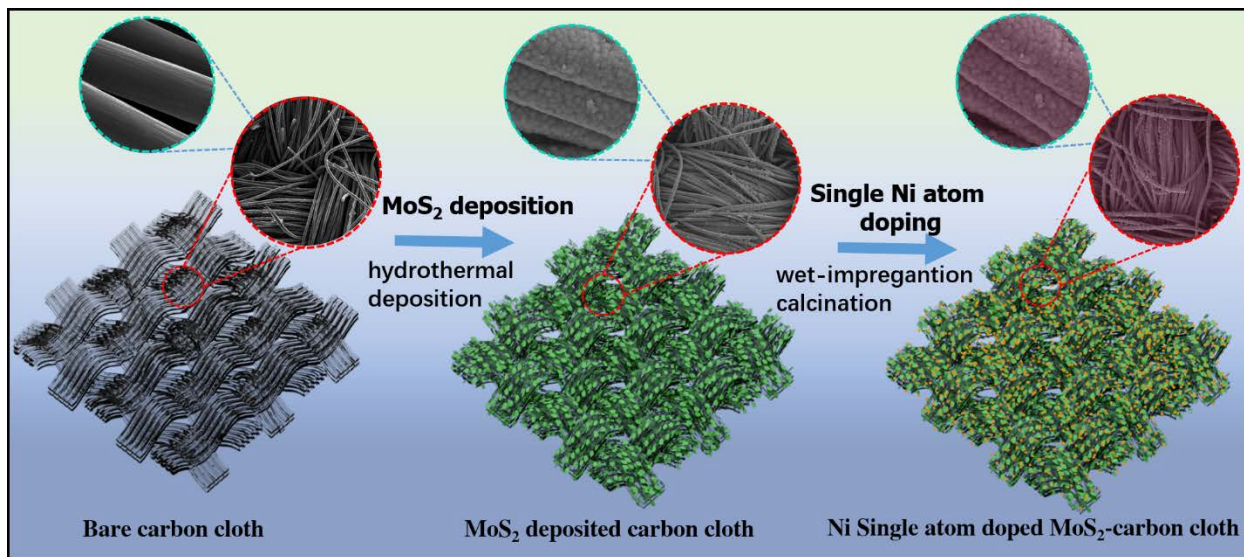
With these values the Gibbs free energy of equation (1) can be rewritten as:

$$\Delta G_H = \Delta E_H + 0.29 \quad (3)$$

## 2. Results and Discussion

The self-supported MoS<sub>2</sub>-carbon cloth (MoS<sub>2</sub>/CC) electrodes were synthesized by growing MoS<sub>2</sub> on a pre-treated carbon cloth through a hydrothermal method.[32] The single atom modification was achieved by wet-impregnation of the MoS<sub>2</sub>/CC in NiCl<sub>2</sub> solution and further calcination in a 10% H<sub>2</sub>/Ar atmosphere at 300 °C for 1h. The synthetic procedure employed in the production of Ni<sub>SA</sub>-MoS<sub>2</sub>/CC is illustrated in Figure 1. In order to demonstrate the effect of the single atom decorating on the MoS<sub>2</sub>, a Ni-cluster modified MoS<sub>2</sub> (Ni<sub>C</sub>-MoS<sub>2</sub>/CC) was prepared by minor modification of the synthetic method. In this case, the wet-impregnated

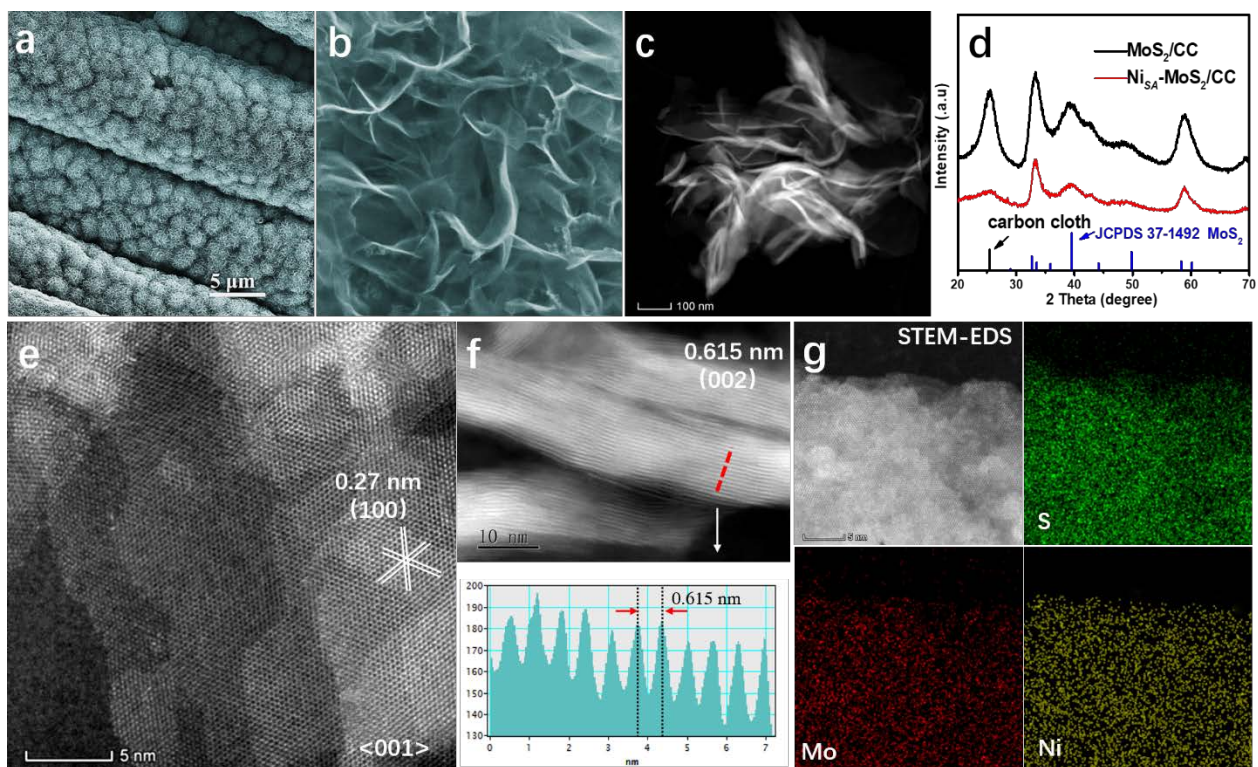
MoS<sub>2</sub>/CC sample was calcined at 600°C for 2h under 10% H<sub>2</sub>/Ar atmosphere. Full experimental details can be found in the experimental procedure section.



**Figure 1** Schematic procedure for the preparation of Ni<sub>SA</sub>-MoS<sub>2</sub> and Ni<sub>C</sub>-MoS<sub>2</sub>. The green color of the cartoon represent the MoS<sub>2</sub> deposited catalyst, the orange dots correspond to the Ni-single atom decorating of the MoS<sub>2</sub>. Top figure in circles correspond to the SEM images of the sample at each stage of the preparation.

SEM images of the original MoS<sub>2</sub> indicate that the carbon cloth substrate is fully covered with the MoS<sub>2</sub> nanosheet array (Figure S1). The low- and high-magnification SEM images of Ni<sub>SA</sub>-MoS<sub>2</sub> presented in Figure 2a-b suggest that both the morphology and integrated nature of the nanosheet array are well preserved upon Ni decorating. The high uniformity and coverage of the as-prepared sample is evidenced by SEM images captured at various magnifications (Figure S2). Figure 2d shows the X-ray diffraction (XRD) patterns of MoS<sub>2</sub>/CC and Ni<sub>SA</sub>-MoS<sub>2</sub>/CC. For the MoS<sub>2</sub>/CC precursor, the diffraction peaks at  $\sim 2\theta = 32.8^\circ$ ,  $39.7^\circ$  and  $58.3^\circ$  correspond well to the (100), (103) and (110) planes of the hexagonal MoS<sub>2</sub> phase (JCPDS no. 37-1492), respectively. The composition of Ni was also determined by ICP-AES to be  $1.8 \pm 0.1$  at.%. No obvious change in the relative signal intensity or position of the MoS<sub>2</sub> peaks is observed after Ni atom decorating. No additional peaks are observed, which can be attributed to the low concentration of highly dispersed Ni atoms.





**Figure 2** (a) Low and (b) high magnification SEM images of  $\text{Ni}_{\text{SA}}\text{-MoS}_2/\text{CC}$ , (c) HAADF-STEM image in low magnification (d) XRD spectrum of  $\text{Ni}_{\text{SA}}\text{-MoS}_2/\text{CC}$  and undecorated  $\text{MoS}_2/\text{CC}$ . Atomic STEM images of (e) basal plane and (f) edge site of  $\text{Ni}_{\text{SA}}\text{-MoS}_2/\text{CC}$ . (g) EDS-mapping of  $\text{Ni}_{\text{SA}}\text{-MoS}_2/\text{CC}$ .

Fig. 2c shows a typical STEM image of the MoS<sub>2</sub> nanosheet. The spacing of the crystal lattices, as shown in Figure 2e, is 0.27 nm, which can be attributed to basal planes of MoS<sub>2</sub> viewed along the <001> zone axis.

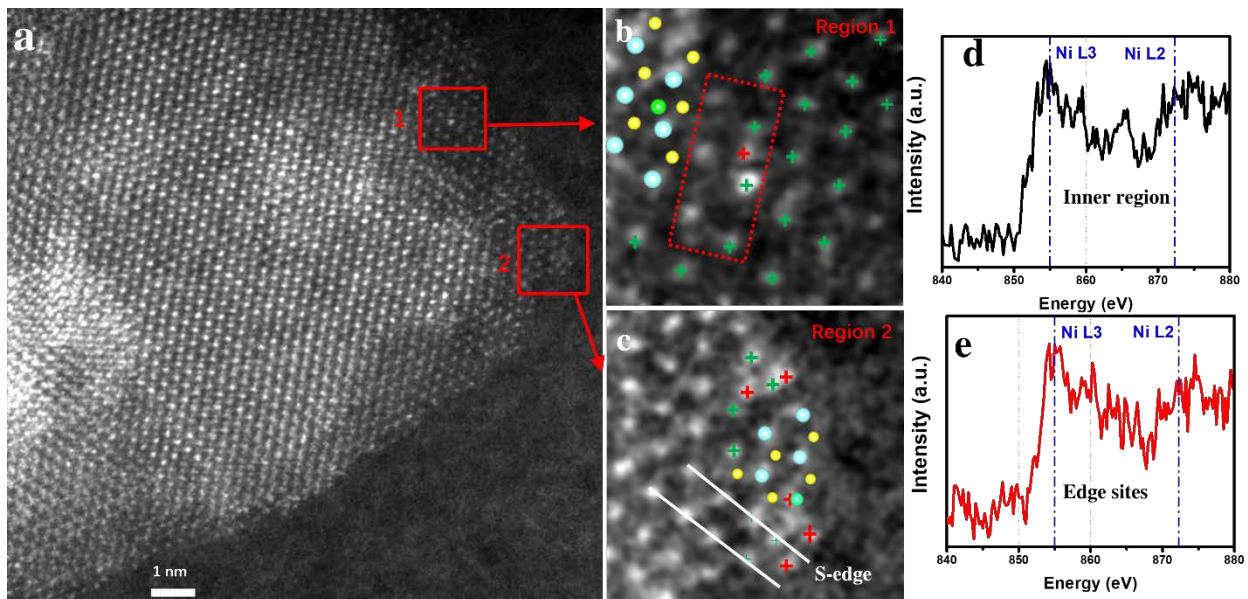
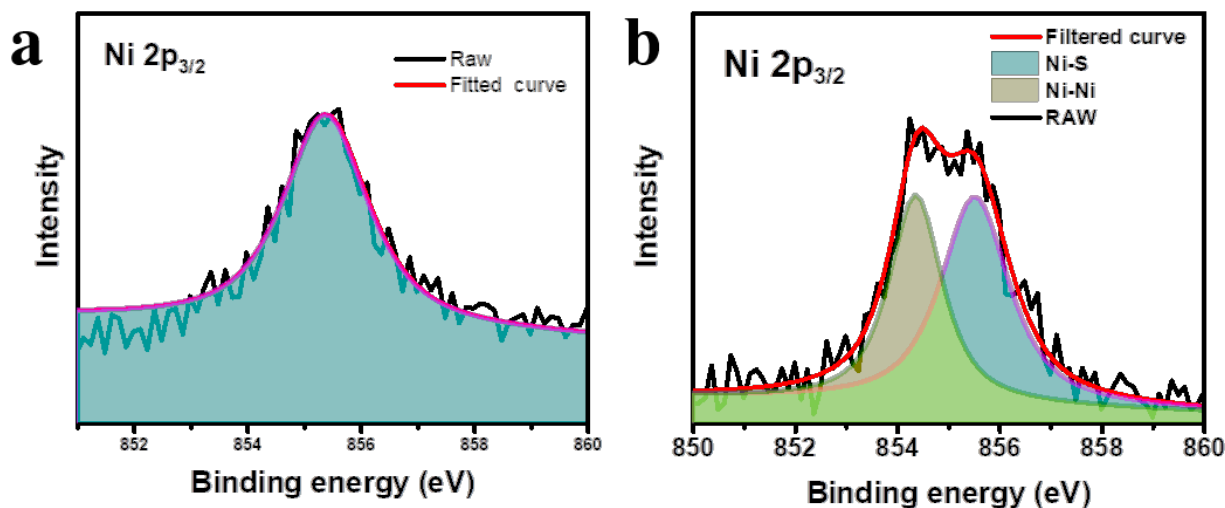


Figure 2f shows the fringes of the layered nanosheet arrays. The profile indicates the layer distance to be 0.615 nm, which is in accordance with the layer distance of MoS<sub>2</sub>.<sup>[33]</sup> The high density of edge sites observed in these regions indicates the edge-rich feature of Ni<sub>SA</sub>-MoS<sub>2</sub>/CC, which is favorable to HER activity.<sup>[14]</sup> Besides, no Ni cluster or particles were observed at any region of the Ni<sub>SA</sub>-MoS<sub>2</sub>/CC samples, indicating the highly dispersed nature of the Ni atoms. Distribution of Ni atoms in the MoS<sub>2</sub> nanosheet is further corroborated by scanning transmission electron microscopy coupled energy-dispersive X-ray spectroscopy (STEM-EDS) element mapping (Figure 2g). It is evident that Ni atoms are uniformly dispersed and that no areas of aggregation are detected. The Ni content in the product was calculated to be 1.9±0.1 at.% from the mapping, which is in good agreement with the ICP-AES result. In contrast, the Ni<sub>c</sub>-MoS<sub>2</sub>/CC shows Ni cluster size ranged between 4-13 nm and the atomic ratio of Ni in this sample was slightly larger (2.0±0.1 at.%) (see Supplementary Figure3).

**Figure 3** (a) Atomic resolution STEM images of Ni<sub>SA</sub>-MoS<sub>2</sub>/CC and corresponding close-up views of (b) region 1 and (c) region 2; Ni atoms marked with a red cross, Mo atoms marked with a green cross) Corresponding Ni EELS spectra of (d) inner region 1 and (e) edge sites, region 2

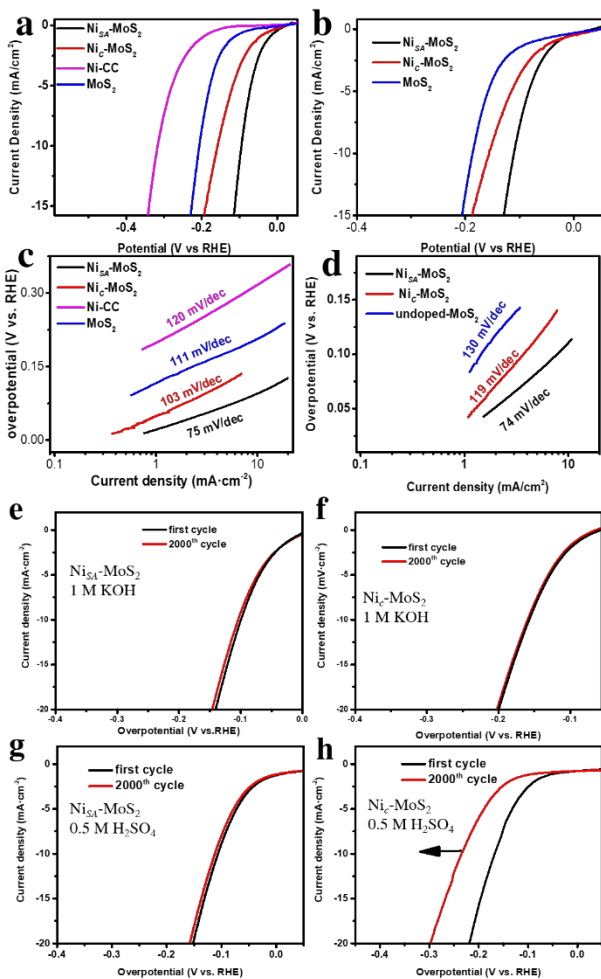
The coordination of the Ni decorating site can impose a significant effect on the HER performance of MoS<sub>2</sub>-based catalysts.[23] Some previous reported works have illuminated a Ni doping strategy to modify MoS<sub>2</sub> lattice in order to improve its HER performance.[11,23,34] In their cases, Ni atoms substitute some Mo sites in MoS<sub>2</sub> lattice. However, the 300 °C calcination of NiCl<sub>2</sub> upon a well-formed MoS<sub>2</sub> in this work may result in a totally different Ni atoms location. To verify the coordination of Ni atoms in the MoS<sub>2</sub> lattice in this work, atomic resolution STEM was applied. The HAADF-STEM image, shown in Figure 3a, presents a high magnification view of a piece of Ni<sub>SA</sub>-MoS<sub>2</sub>/CC. No obvious Ni atom could be found due to the lower atomic number of Ni in comparison to Mo. By analyzing the atomic structure of larger multiples, some embedded foreign atoms are found in the atomic column of MoS<sub>2</sub> (Figure 3b, c). The corresponding EELS spectra of this sites reveal the existence of Ni peaks at ~855 eV, suggesting the embedded atoms to be Ni atoms (Figure 3d). Figure 3b shows a magnified image of the Ni atom from Figure 3a (region 1), demonstrating the single Ni atom locates atop of a hexagonal site of the basal plane, namely H-basal site. The model inserted is an atomic simulation of the red rectangle region. Moreover, the high density of disordered atoms with lower contrast, situated at the edge of MoS<sub>2</sub>, suggests the existence of Ni atoms in the S-edge (Figure 3c). The corresponding EELS spectra of edge sites also reveal these atoms to be Ni atoms (Figure 3e).[35] The signals obtained in both regions were rough due to the atomic dispersion of Ni atoms.[36]



**Figure 4** Ni 2p<sub>3/2</sub> XPS spectrum of (a) Ni<sub>SA</sub>-MoS<sub>2</sub>/CC and (b) Ni<sub>c</sub>-MoS<sub>2</sub>/CC.

The chemical coordination of Ni atoms can largely influence their activity and stability during catalysis. The Ni surface composition and electronic binding energies were characterized by X-ray photoelectron

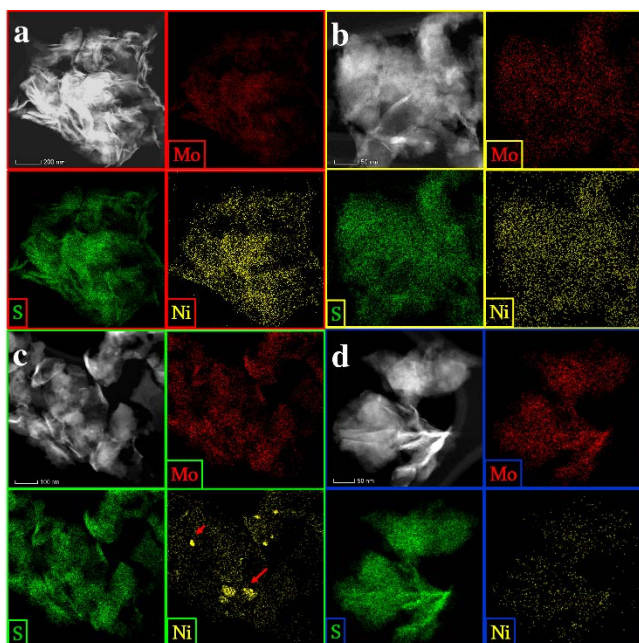
spectroscopy (XPS) with C 1s energy correction (see Figure S4). The Ni 2p<sub>3/2</sub> spectrum of Ni<sub>SA</sub>-MoS<sub>2</sub>/CC shows a single peak located at 855.5 eV, which can be assigned to Ni-S bonding (Figure 4a).[37] In comparison, the Ni 2p<sub>3/2</sub> spectrum of Ni<sub>c</sub>-MoS<sub>2</sub>/CC shows two characteristic peaks located at 854.5 eV and 855.5 eV, which can be assigned as Ni-Ni and Ni-S bonding, respectively (Figure 4b).[37] These results provide further evidence for the single atom dispersion of Ni in Ni<sub>SA</sub>-MoS<sub>2</sub>/CC. The noticeable signal noise is attributed to the high dispersion and low concentration of Ni atoms. [38]



**Figure 5** (a) Polarization curves of Ni<sub>SA</sub>-MoS<sub>2</sub>/CC, Ni<sub>c</sub>-MoS<sub>2</sub>/CC, Ni/CC and MoS<sub>2</sub>/CC in 1 M KOH solution with a scan rate of 5 mV·s<sup>-1</sup>. (b) Polarization curves of Ni<sub>SA</sub>-MoS<sub>2</sub>, Ni<sub>c</sub>-MoS<sub>2</sub> and bare MoS<sub>2</sub> in 0.5 M H<sub>2</sub>SO<sub>4</sub> solution with a scan rate of 5 mV·s<sup>-1</sup>. (c) Tafel plots obtained from the polarization curves in (a). (d) Tafel plots obtained from the polarization curves in (b). Stability tests by measuring the polarization profiles for Ni<sub>SA</sub>-MoS<sub>2</sub>/CC (e,g) catalyst and Ni<sub>c</sub>-MoS<sub>2</sub>/CC before and after 2000 cyclic potential scans in 1 M KOH (e,f) and 0.5 M H<sub>2</sub>SO<sub>4</sub> (g,h) at a scan rate of 50 mV·s<sup>-1</sup>.

Inspired by the unique structural features of Ni<sub>SA</sub>-MoS<sub>2</sub>/CC, its HER activity was firstly evaluated in 1 M KOH solution. For comparison, the HER performance of Ni/CC, MoS<sub>2</sub>/CC and Ni<sub>c</sub>-MoS<sub>2</sub>/CC were also investigated under identical conditions. Since the as-measured reaction currents cannot directly reflect the intrinsic behavior of the catalysts due to the effect of Ohmic resistance, an iR compensation was applied to all recorded data. Figure 5a shows representative linear sweep voltammograms (LSVs) of the aforementioned catalysts in 1 M KOH. For Ni<sub>SA</sub>-MoS<sub>2</sub>/CC, an overpotential of ~95 mV versus reversible hydrogen electrode (RHE) is needed to drive a cathodic current density of 10 mA·cm<sup>-2</sup>, which is 112 mV lower than that of MoS<sub>2</sub>/CC (207 mV) and 62 mV lower than that of Ni<sub>c</sub>-MoS<sub>2</sub>/CC (157 mV). Such activity surpasses most of previously reported metal sulfide HER electrocatalysts.[39–41] It is noted that the HER activity of MoS<sub>2</sub> is significantly enhanced after Ni decorating in the form of either clusters or single atoms, which can be attributed to the synergistic effect of Ni at the catalyst surface, the HER mechanism can be generalized for two pathways: (Scheme 1) the Volmer-Heyrovsky mechanism and/or (Scheme 2) the Volmer-Tafel mechanism (See Supplementary Scheme). These two pathways imply that the competitive processes of H-atom adsorption/desorption occurring at the catalyst surface should be matched in order to accelerate the HER. Hydrogen adsorption (H<sub>ads</sub>) in alkaline condition is formed from dissociation of water. This step is rather sluggish on MoS<sub>2</sub> surface, which is the main inhibitor of the reaction.[42] However, water dissociation is accelerated at Ni atom sites, resulting in the enhanced HER activity observed for Ni<sub>SA</sub>-MoS<sub>2</sub>/CC.[43] A HER catalyzed by pure MoS<sub>2</sub> follows the Volmer-Heyrovsky mechanism (scheme 1),[22] which has a lower H<sub>ads</sub> coverage (Θ<sub>H</sub>) and a Tafel slope of ~111 mV·decade<sup>-1</sup>. In contrast, the Ni<sub>SA</sub>-MoS<sub>2</sub>/CC shows a Tafel slope of 75 mV·decade<sup>-1</sup>, which is about 36 mV·decade<sup>-1</sup> lower than that of MoS<sub>2</sub>/CC and 45 mV·decade<sup>-1</sup> less than that of Ni-decorated carbon cloth (Ni/CC) (Figure 5c). The Tafel slope of 75 mV·decade<sup>-1</sup> indicates that the rate-limiting step at low overpotentials is the recombination of two adsorbed H atoms, which demonstrates that the Ni decorating promotes the HER via a Volmer-Tafel mechanism (Scheme 2). When comparing the Ni<sub>SA</sub>-MoS<sub>2</sub>/CC with the Ni<sub>c</sub>-MoS<sub>2</sub>/CC, it was observed that the presence of Ni cluster favors the HER (Tafel slope of 103 mV·decade<sup>-1</sup>) relative to MoS<sub>2</sub>/CC. However, the catalytic activity of the Ni<sub>SA</sub>-MoS<sub>2</sub>/CC is significantly superior with respect to the Ni<sub>c</sub>-MoS<sub>2</sub>/CC. This result demonstrates that single atom Ni-decorating in MoS<sub>2</sub> is preferable to cluster-decorating in the pursuit of a more efficient HER catalyst. Furthermore, the LSVs of Ni<sub>SA</sub>-

MoS<sub>2</sub>/CC and Ni<sub>c</sub>-MoS<sub>2</sub> suffers negligible depletion before and after 2000 cycles, suggesting the Ni single atoms can stably anchor on MoS<sub>2</sub> in alkaline condition (Figure 5e,f).

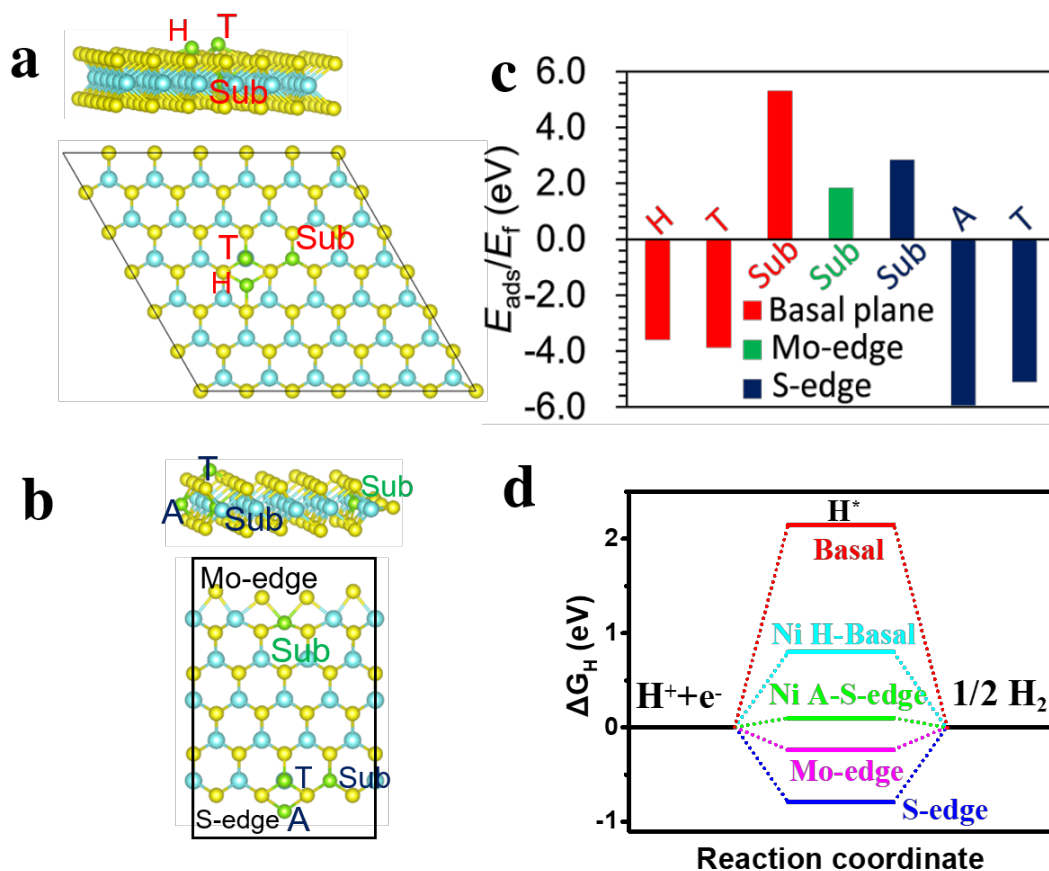


**Figure 6** EDS-mapping of Ni<sub>SA</sub>-MoS<sub>2</sub>/CC sample before (a) and after (b) 2000 CV cycle in 0.5M H<sub>2</sub>SO<sub>4</sub> solution. The Ni amount suffers no obvious change after 2000 cycles. EDS-mapping of Ni<sub>c</sub>-MoS<sub>2</sub>/CC sample before (c) and after (d) 2000 CV cycle in 0.5M H<sub>2</sub>SO<sub>4</sub> solution. A huge depletion of Ni (2.0 at.% to 0.7 at.%, evidenced by ICP-AES) was found after 2000 cycles caused by the dissolution of Ni atom in the harsh electrochemical condition.

The HER performance of as prepared sample was further studied in a 0.5 M H<sub>2</sub>SO<sub>4</sub> solution. It took an overpotential of 110 mV to achieve a current density of 10 mA·cm<sup>-2</sup>, 47 mV lower than that of Ni<sub>c</sub>-MoS<sub>2</sub>/CC (157mV) and 76 mV lower than that of the undecorated MoS<sub>2</sub>/CC (186 mV). Meanwhile, Ni<sub>SA</sub>-MoS<sub>2</sub>/CC showed a Tafel slope of 74 mV·decade<sup>-1</sup> which is lower than Ni<sub>c</sub>-MoS<sub>2</sub>/CC (119 mV·decade<sup>-1</sup>) and undecorated MoS<sub>2</sub> (130 mV·decade<sup>-1</sup>), indicating a superior HER catalytic activity of single atom decorating in acid environment. Additionally, the stability of Ni<sub>SA</sub>-MoS<sub>2</sub>/CC was examined by successive 2000 CV cycles in

a 0.5 M  $\text{H}_2\text{SO}_4$  solution (Figure 5g). Little decrease was observed after 2000 CV cycles. However, overpotential of  $\text{Ni}_c\text{-MoS}_2/\text{CC}$  suffered great depletion in the same condition. EDS results revealed that Ni atoms in  $\text{Ni}_{\text{SA}}\text{-MoS}_2/\text{CC}$  still dispersed homogeneously and the loading amount suffered little decrease (Figure 6a,b). In contrast, Ni cluster in  $\text{Ni}_c\text{-MoS}_2/\text{CC}$  vanished after CV cycles (Figure 6c,d) and the overpotential increased from  $\sim 167$  mV to  $\sim 234$  mV due to Ni cluster losing after 2000 cycles (Figure 5h). We attribute the stability of Ni single atom to the strengthened Ni-S bonding.[44]

Density functional theory (DFT) calculations were performed to gain fundamental understanding of the



**Figure 7** (a) Atomistic configuration of Ni absorbed at the top of the Mo sites (T) and the center of the hexagon sites (H) on the basal plane and substitution of Mo atom with Ni in monolayer  $\text{MoS}_2$ . (b) Atomistic configuration of Ni absorbed at T and A sites at the S-edge and substitution of Mo atom with Ni at the Mo-edge and S-edge. (c) Adsorption energy of Ni on  $\text{MoS}_2$  and formation energy of substitution of Mo atom with Ni atom in  $\text{MoS}_2$ . (d) Gibbs free energies for  $\text{H}^*$  adsorbed at basal plane, Mo edge and S edge with and without Ni adsorption/substitution.

Ni<sub>SA</sub>-MoS<sub>2</sub> active sites during the HER under alkaline conditions. First, the possible atomistic configurations of Ni-MoS<sub>2</sub> were evaluated. Two possible adsorption sites for Ni on the basal plane of MoS<sub>2</sub> are the top of the Mo site (T) and the center of the hexagonal sites (H) (Figure 7a). The Ni can also be coordinated at the S-edge in MoS<sub>2</sub>, as shown in Figure 7b. Thus, there were three substitution sites for Ni replacing Mo, i.e. the basal plane, the Mo-edge and the S-edge of MoS<sub>2</sub>. It was reported that the Mo-edge covered with S atoms is catalytically active for the HER,[45] and we adopted the same model. Details of the simulation methodology can be found in the Supporting Information.

Figure 7c shows the calculated  $E_{\text{ads}}$  and  $E_f$  values; positive values indicate unfavorable interactions. The adsorption of Ni is preferential at the A site of the S-edge, in addition to the T-site and H-site of the basal plane. The  $E_f$  values are 5.30, 1.83 and 2.83 eV for the substitution of Ni for Mo at the basal plane, the Mo-edge and the S-edge, respectively, indicating the less possibility of that the substitution of Mo with Ni. As was discussed above, the presence of Ni atoms at the A-S-edge and H-basal site was confirmed by atomic STEM, indicating good agreement between the experimental observation and the theoretical results.

The Gibbs free-energy for the adsorption of atomic hydrogen ( $\Delta G_{\text{H}}$ ), which should be close to zero for an ideal HER catalyst, is an important parameter to discuss in the evaluation of the HER catalyst. Thus, we calculated  $\Delta G_{\text{H}}$  at the basal plane, Mo-edge and S-edge in the absence and presence of Ni-decorating. The free energy diagrams for HER are shown in Figure 7d. The +2.16 eV value for the basal plane of MoS<sub>2</sub> indicates that H<sup>+</sup> cannot be efficiently adsorbed there. A  $\Delta G_{\text{H}} = -0.79$  eV at the S-edge indicates that H<sub>2</sub> cannot effectively desorb from the S-edge. On the other hand, a  $\Delta G_{\text{H}} \cong -0.24$  at the Mo-edge, is distinctive as a good HER active site. Our results are in good agreement with previous reports that the catalytically active sites of monolayer MoS<sub>2</sub> are located at the Mo-edge, while the basal plane is inert to the HER.[46] Interestingly, after single Ni atoms are anchored on S-edge, the value of  $\Delta G_{\text{H}}$  are calculated to be 0.1 eV, indicating the introduction of a great enhanced HER active sites. Additionally, the calculations on the Ni<sub>SA</sub>-MoS<sub>2</sub> show that the values of  $\Delta G_{\text{H}}$  for the H-basal sites decrease to 0.8 eV, suggesting an improvement in the HER activity of the basal plane. Thus, single Ni atom-decorating does not significantly affect the catalytic behavior of the Mo-edge, but rather promotes noticeable improvement in the catalytic activity of S-edges. The inert basal plane is activated and the activity of the edge sites is greatly enhanced. In this case, Ni<sub>c</sub>-MoS<sub>2</sub>, with the same



decorating concentration, provides less activated sites in comparison to Ni<sub>SA</sub>-MoS<sub>2</sub>. The Ni<sub>SA</sub>-MoS<sub>2</sub> can offer a synergistic effect along with activated effect so as to provide higher HER activity.[22]

### 3. Conclusions

In conclusion, through introducing Ni single atoms into the MoS<sub>2</sub> S-edge and H-sites of the basal plane, a distinct enhancement of HER activity was achieved compared with pure MoS<sub>2</sub>/CC and Ni cluster-decorated MoS<sub>2</sub>/CC. Atomic STEM indicates that the Ni single atoms were anchored on the S-edge and H sites of basal plane. Experimentally, by decorating the MoS<sub>2</sub> with Ni-single atoms, the overpotential to achieve a current density of 10 mA·cm<sup>-2</sup> during the HER is significantly reduced in comparison to undecorated MoS<sub>2</sub> and Ni-cluster decorated MoS<sub>2</sub> in acidic conditions. DFT calculations indicate that the decorated Ni atoms on the S-edge and H site of basal plane can tune the adsorption behavior of H atoms and consequently the HER activity. The Ni<sub>SA</sub>-MoS<sub>2</sub>/CC provides a low-cost but highly efficient solution to HER. This work opens the door to the design of more active catalyst based on single-atom decorating MoS<sub>2</sub> with other transition metal, which can outperform the catalytic activity of state of the art catalyst without the utilization of precious metals.

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