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ARL

**High Temperature Ceramic Microstructure and Interface Evolution during
Exposure to Particulate Laden Combustion Flows in Gas Turbine Engines**

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- **Introduction**
 - The sand – CMAS problem
 - Protective Ceramic Coatings
- **Ongoing Experimental Efforts**
 - TBCs tested in full scale engine tests
 - EBC burner rig testing and characterization
 - CMAS characterization and evolution
 - Model bulk YSZ systems under CMAS attack
- **Future Work**
 - Advanced Interfacial Studies
 - Computational Studies
- **Summary**



OBJECTIVE

To innovate sandphobic coating and surface modification for high temperature turbine blades to resist sand glaze build-up and related Calcia-Magnesia-Alumina-Silicate (CMAS) attack on Thermal/Environmental Barrier Coatings (T/EBCs)



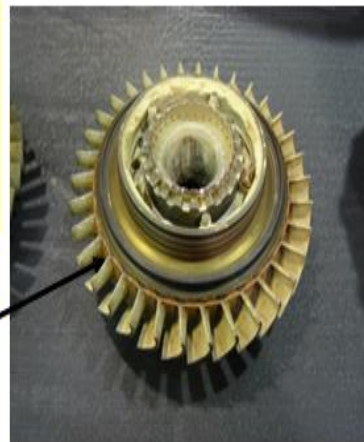
Field returned engine hardware from SWA



Turbine Nozzle

- Sand build up (glazing)
- Plugged cooling holes
- Nozzle oxidation
- CMAS attack

- Blades coated with melted sand
- Blade tip wear
- Plugged cooling holes
- CMAS attack



Turbine Rotor/Blades



Typical build-up on vane



Typical build-up on blade

Typical rotorcraft gas turbine engine nozzle and rotor blades with sand-induced damages

- **Hot section sand glazing / chemical attack is influenced by following parameters:**

- Particle size and material composition of particle
- Material properties of airfoil thermal barrier coating systems
- Fluid flow dynamics and temperature



- Various empirical methods in research/development to mitigate CMAS damage
- Lacking → quantitative physico-chemical model of the reactions
- This is a complex problem:
 - Impact + adherence
 - Infiltration
 - Glass formation via solidification
- AND, the contaminant adds complexity...
 - Natural sand → compositional variation, different grain sizes, and different morphologies based on the age and location of the desert.
- Synthetic sand developed to create a representative baseline
 - AFRL 02/03 represents the state-of-the-art
 - AFRL 02 → small grains, used for bench-level and component-level testing
 - AFRL 03 → larger grain distribution, used for engine-level testing

All occurring at elevated pressures + high temperatures





Layered and Composite TBCs Exposed to Full Scale Sand Ingestion Engine Test:

CMAS Adhesion and TBC Microstructural Evolution



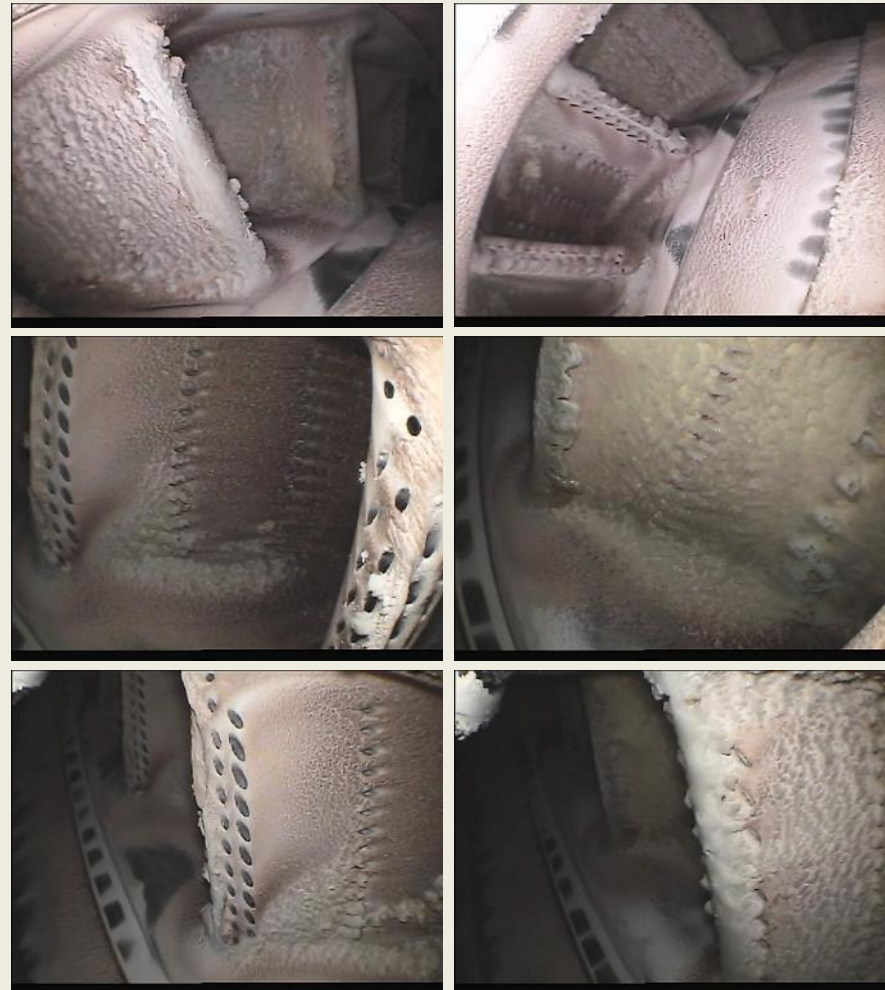
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Sand Ingestion Jet Engine Test **ARL**

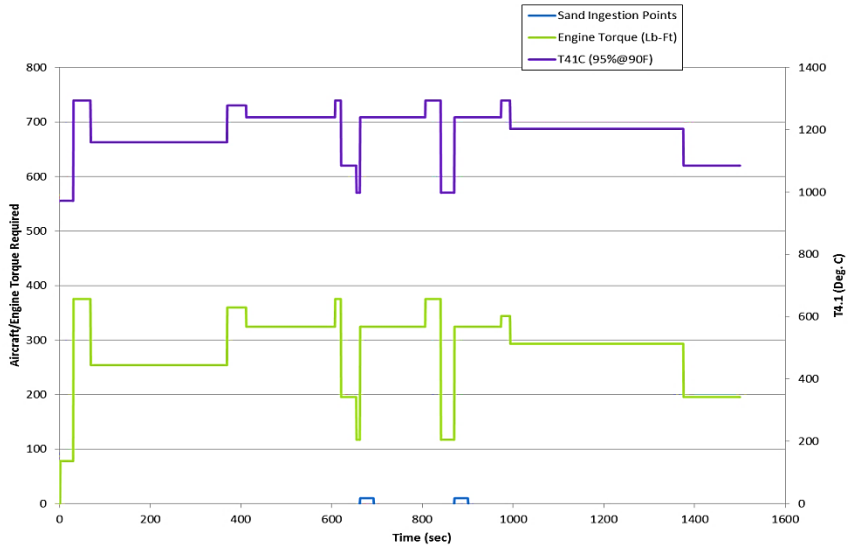
As-built nozzle ring



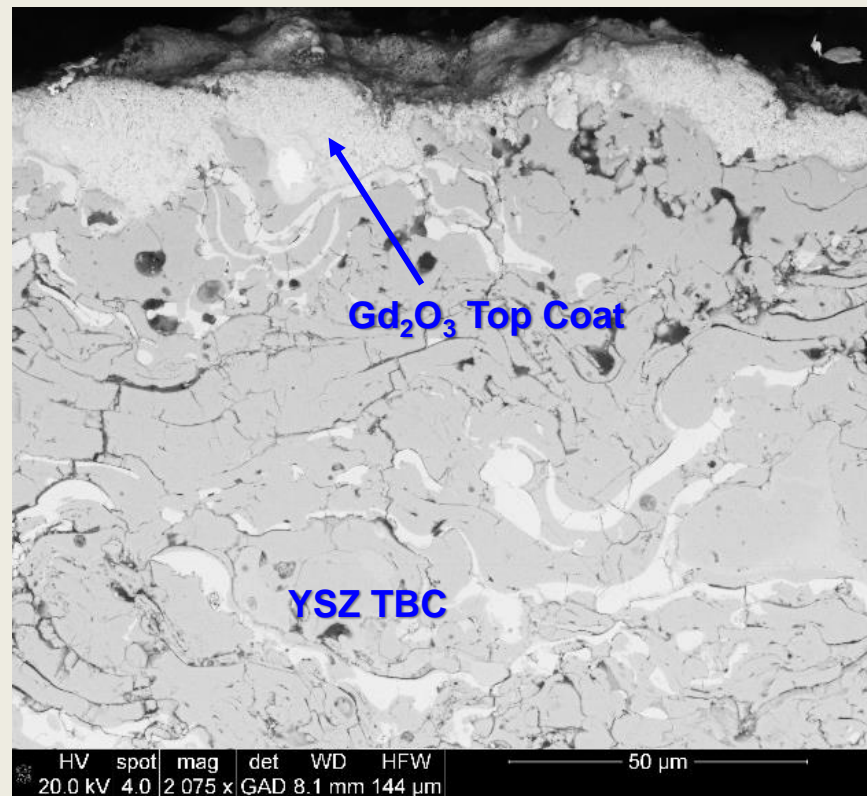
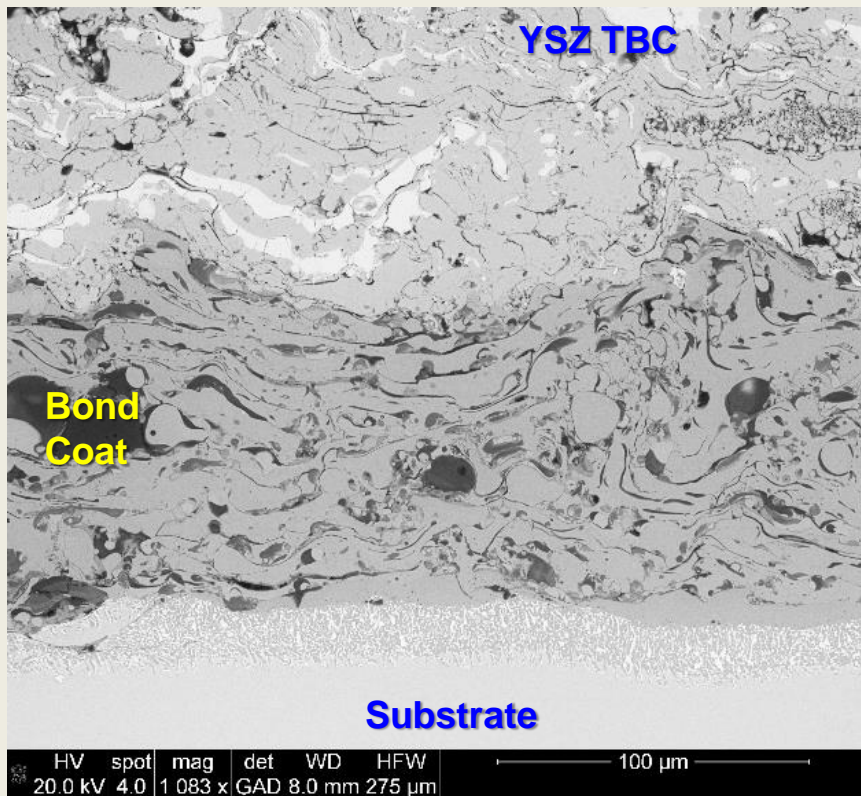
Borescope images from engine run



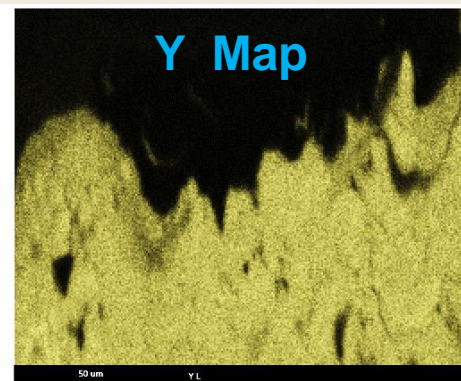
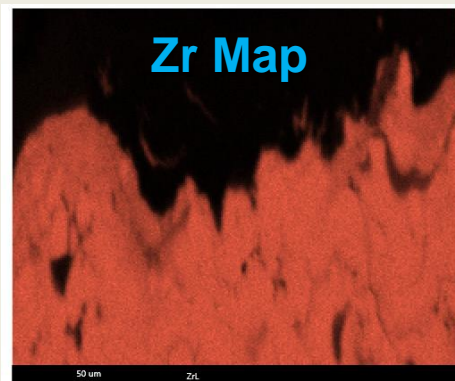
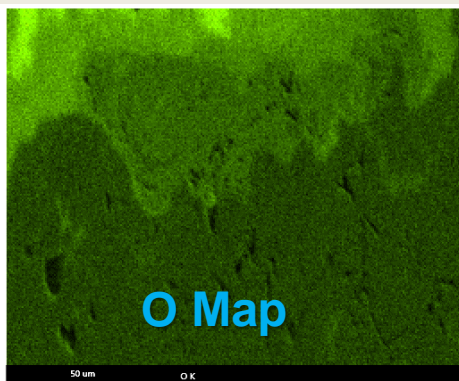
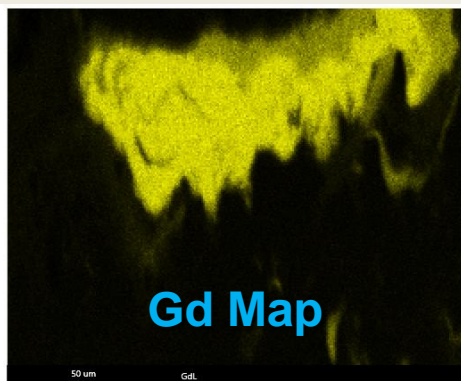
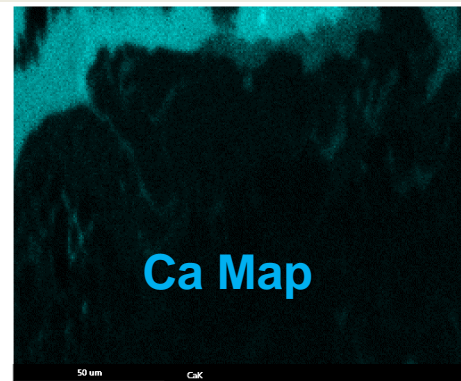
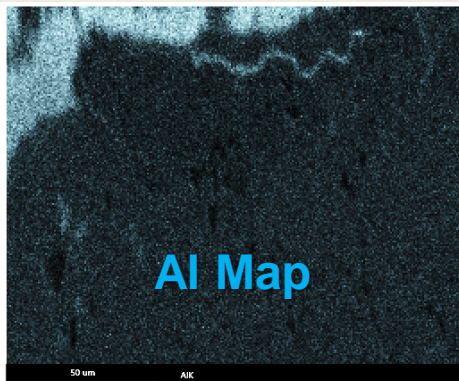
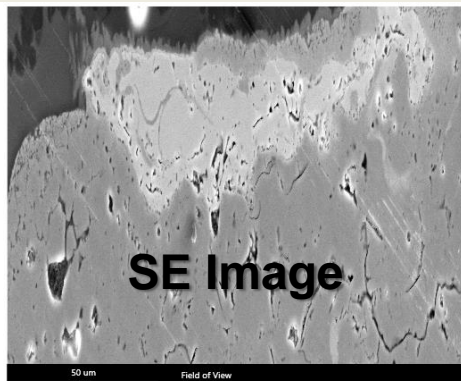
Standard Cycle

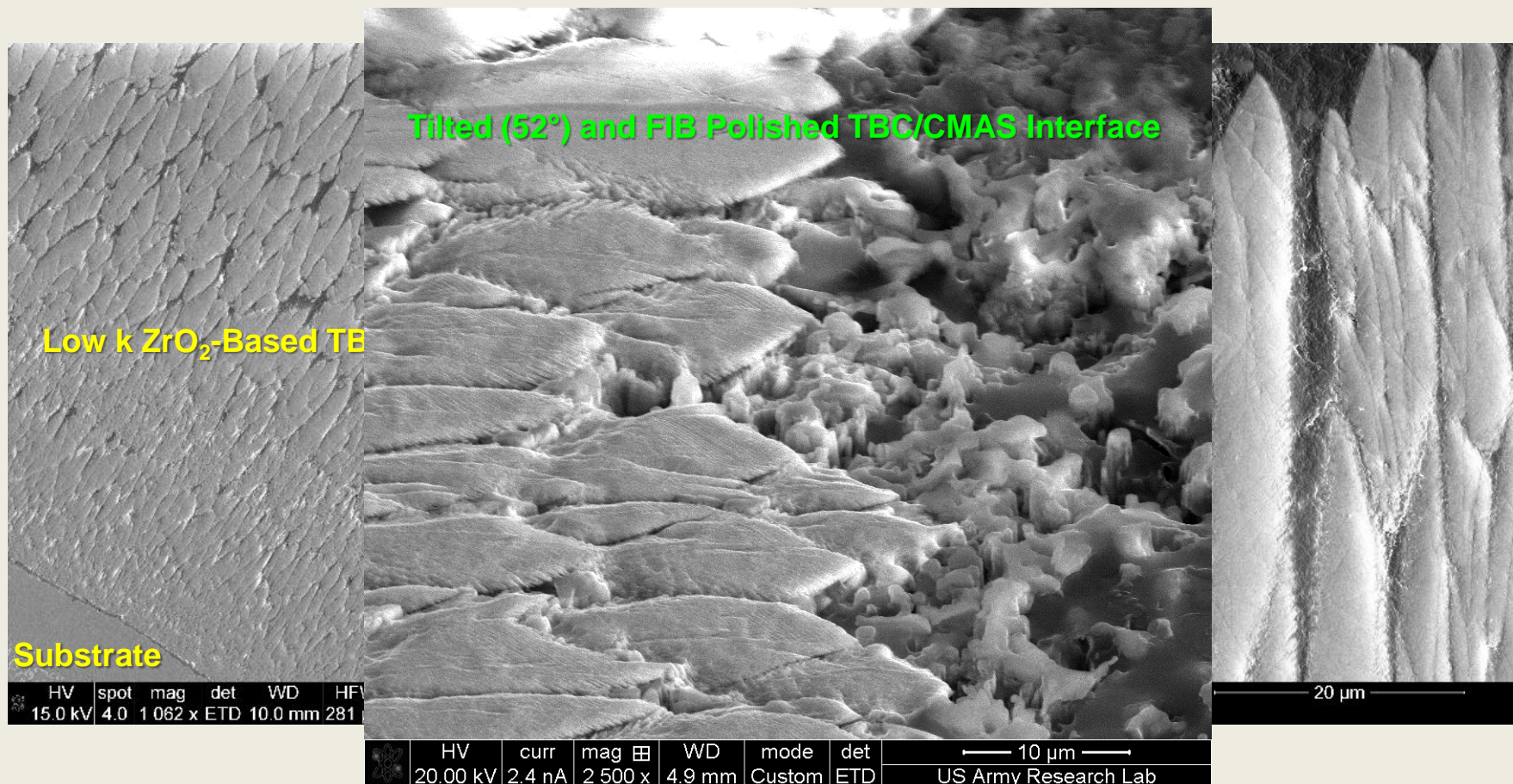


**Engine Tested Consisted of Two 25 min Cycles
MAX Temperature of 1240 °C**



- No significant damage or chemical degradation observed on ARL-NASA-02
- Thin segments (10-20 μm) of Gd₂O₃ top coat are found throughout the specimen

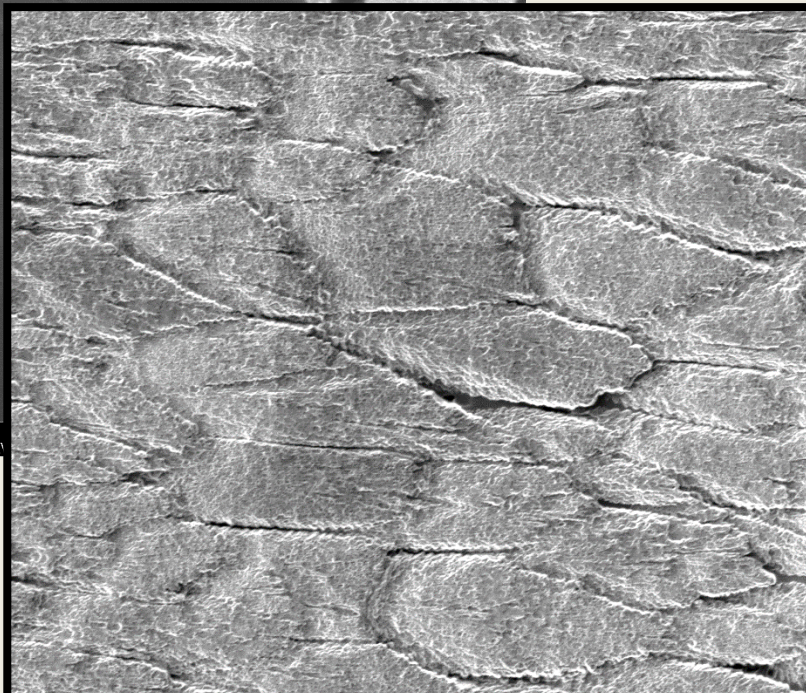




- ARL-NASA-06 exhibits minimal signs of structural damage
- Porous columnar structure does not lead to CMAS infiltration



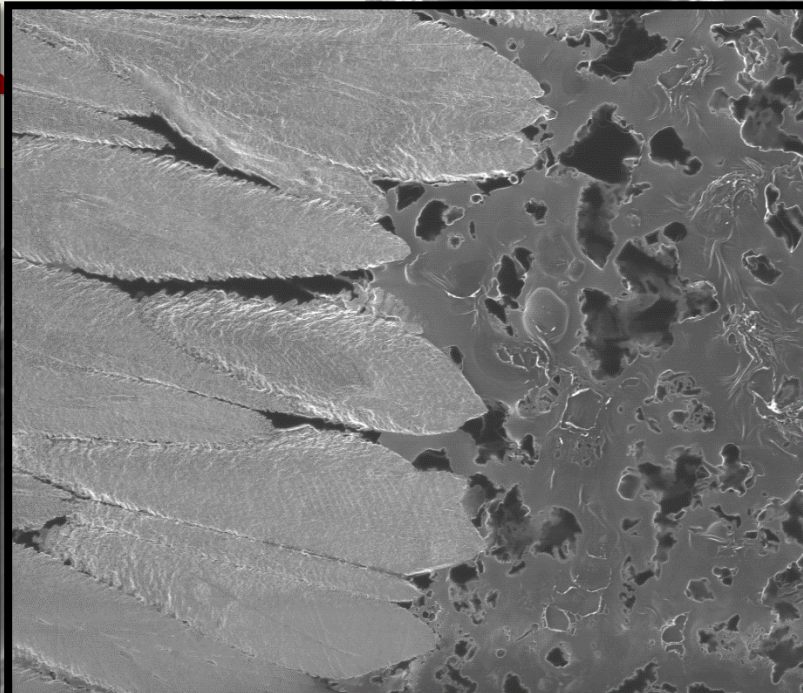
TBC



HV	curr	WD	dwell	mag	tilt	10 μm
30.00 kV	27 pA	16.4 mm	30 μs	5 000 x	53 °	US Army Research Lab

TBC

CMAS



HV	curr	WD	dwell	mag	tilt	20 μm
30.00 kV	27 pA	16.4 mm	30 μs	2 000 x	53 °	US Army Research Lab

- FIB ion milling and imaging provides distinct contrast between TBC & CMAS
 - CMAS infiltration can be assessed *w/out EDS mapping*
- FIB polishing removes surface artifacts due to mechanical preparation (e.g., debris, polishing media)

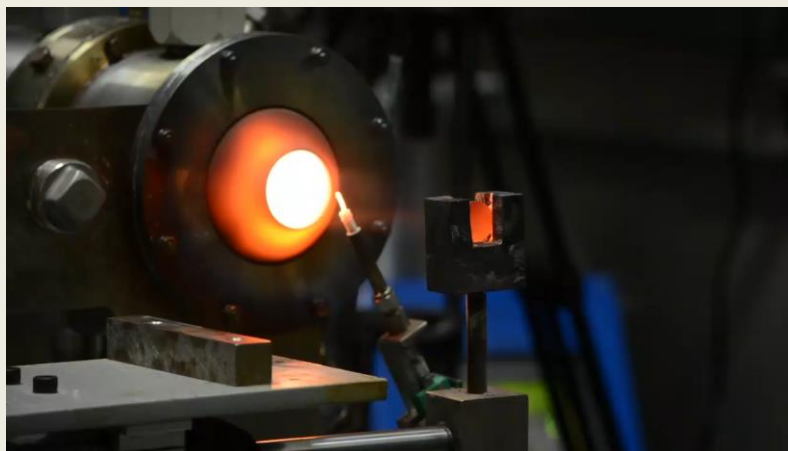


Environmental Barrier Coatings (EBC) under Sand Laden Combustion Flows

Burner Rig Testing and Microstructural Evolution



- **Burner Rig testing with sand ingestion will be conducted on several promising EBCs to evaluate resistance to CMAS infiltration**
- **EBCs will be held under continuous exposure to sand laden combustion flow at ~0.5 Ma and ~1550 °C, for set time intervals (5 – 60 min)**
 - **Times can be adjusted based on CMAS infiltration behavior and EBC durability**
- **Objective is to quantify the CMAS infiltration kinetics on different EBC systems, and if possible, within individual layers of the EBC**

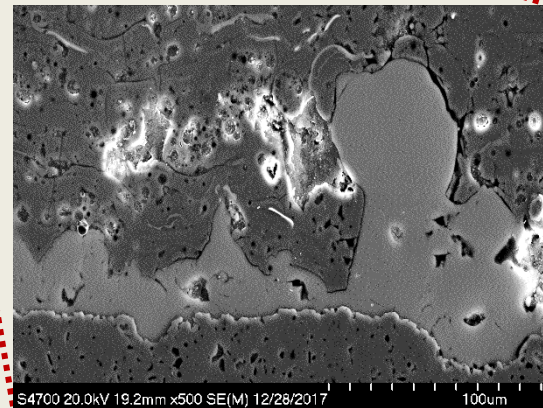
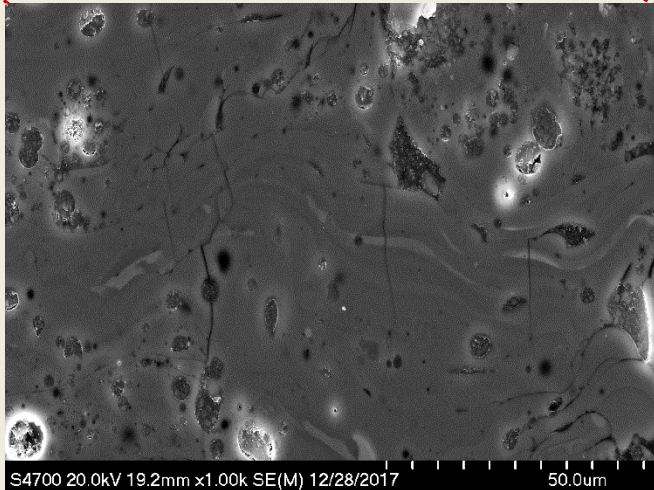
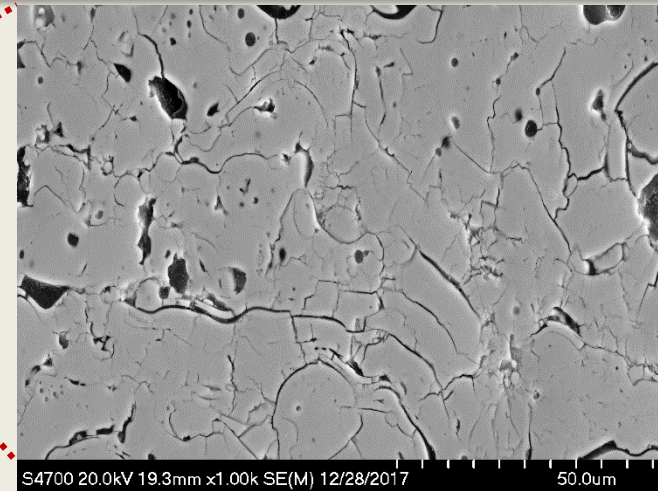
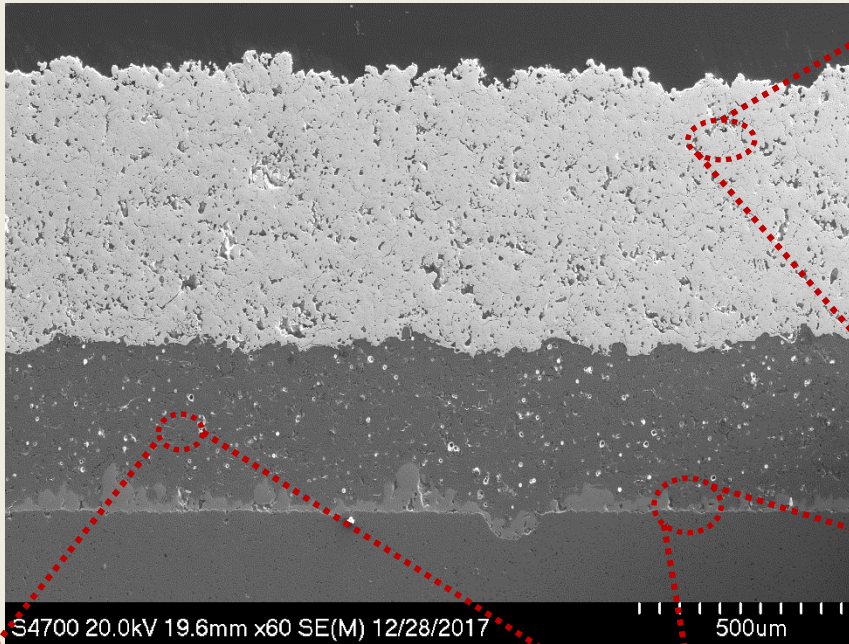




ZrO₂-Y₂O₃ based EBC Systems: As Sprayed



APS ZrO₂-Y₂O₃ based top coat provides low thermal conductivity a good damage tolerance



Alumina-rich Mullite Layer

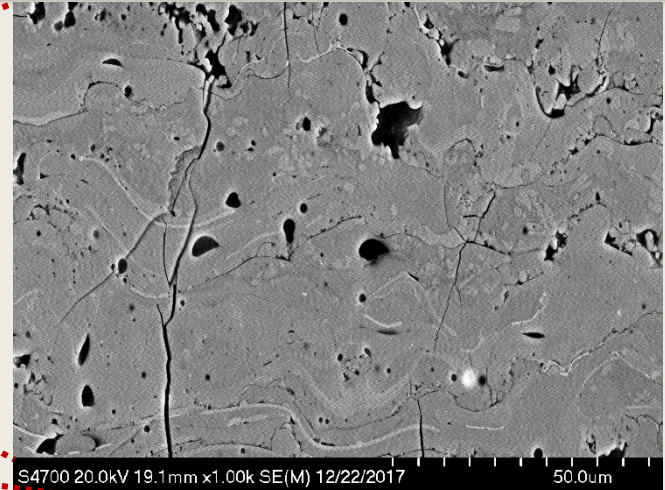
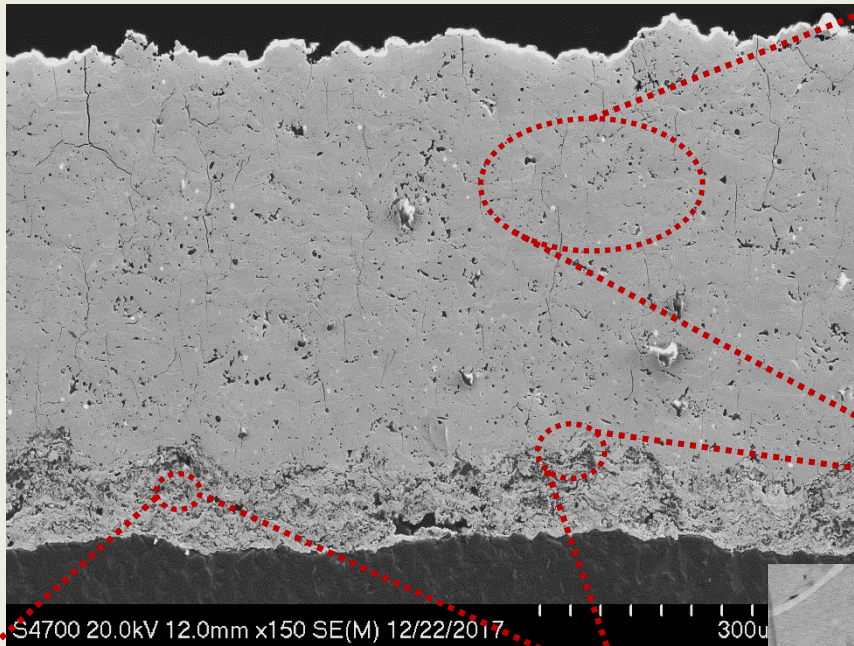
Si Layer serves as oxygen barrier for Substrate
-Sensitive to surface roughness
-Uneven thickness



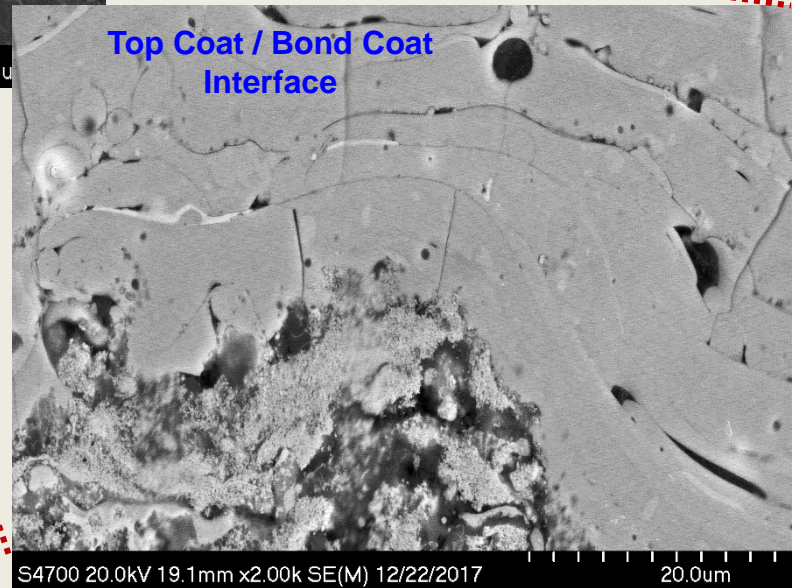
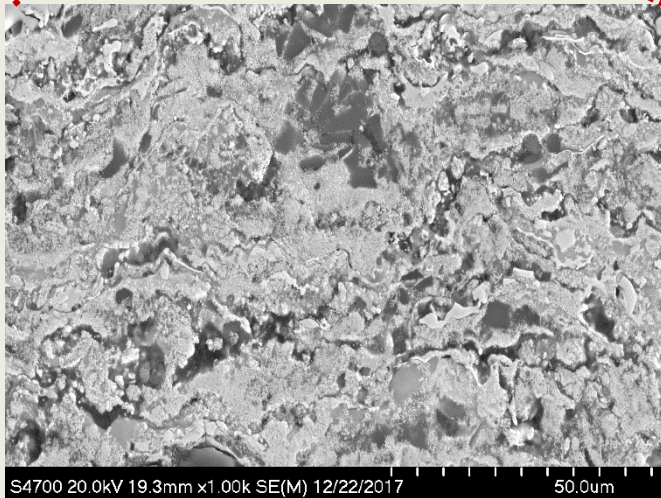
RE DS on HfO₂-Si Bond Coat EBCs: As Sprayed



RE DS Top coat w/ Vertical Cracks for Improved Damage Tolerance



HfO₂-Si bond coat for reduced CTE mismatch with SiC-SiC CMCs



Top Coat / Bond Coat Interface



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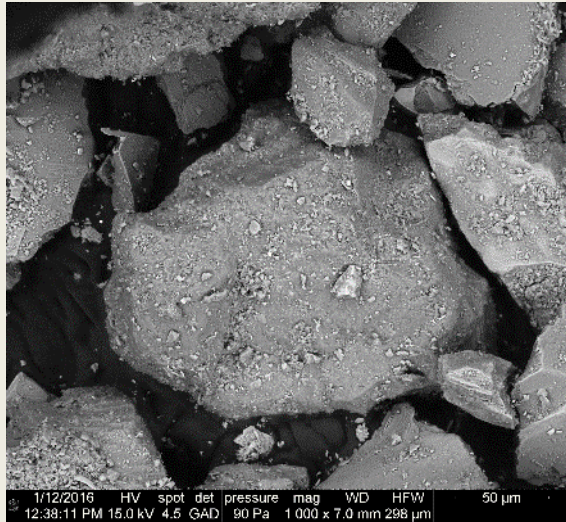
Ongoing Experimental Efforts

ARL

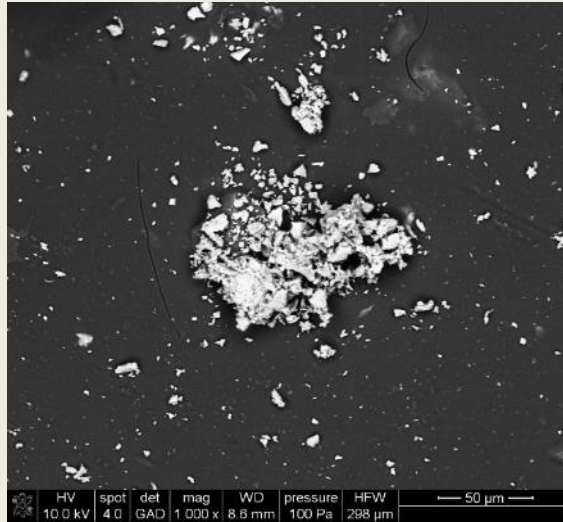
Characteristics and Chemical / Microstructural Evolution of Sand/CMAS



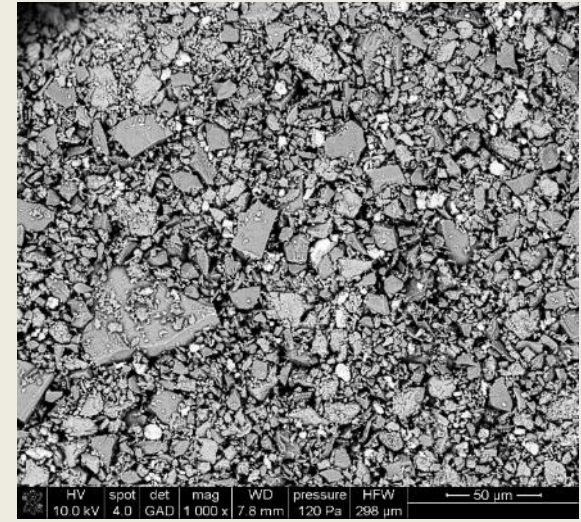
Natural Sand



AFRL 02 Sand



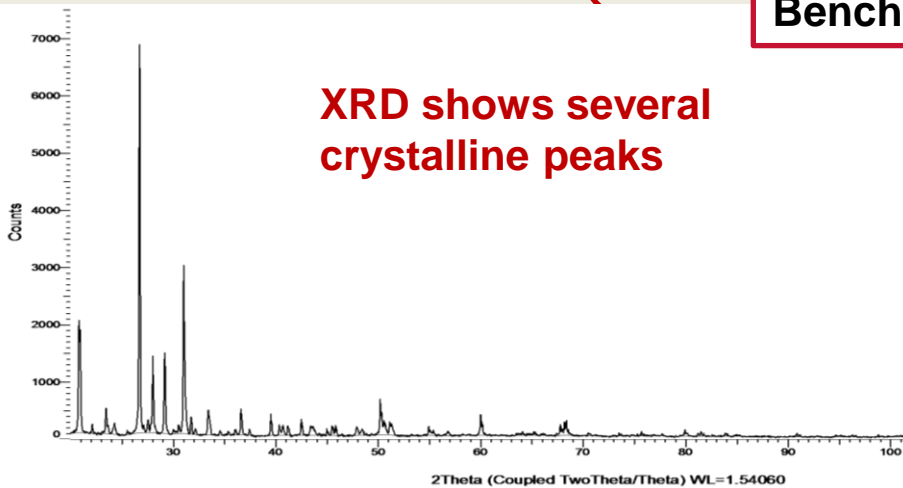
AFRL 03 Sand



Bench-level

Engine-level

XRD shows several crystalline peaks



AFRL synthetic sand

- 34 % quartz (SiO_2)
- 30 % gypsum ($\text{CaSO}_4 \cdot 2\text{H}_2\text{O}$)
- 17 % aplite ($\text{SiO}_2 + \text{KAlSi}_3\text{O}_8/\text{NaAlSi}_3\text{O}_8/\text{CaAl}_2\text{Si}_2\text{O}_8$)
- 14 % dolomite ($\text{CaMg}(\text{CO}_3)_2$)
- 5 % salt (NaCl)



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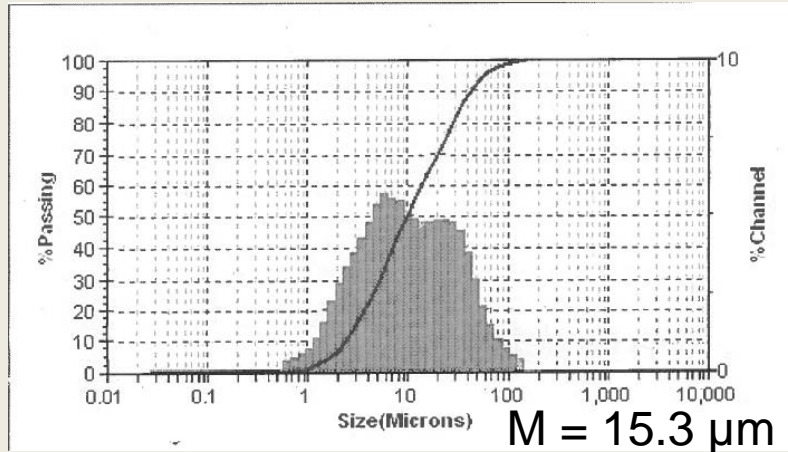
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Sand size distribution



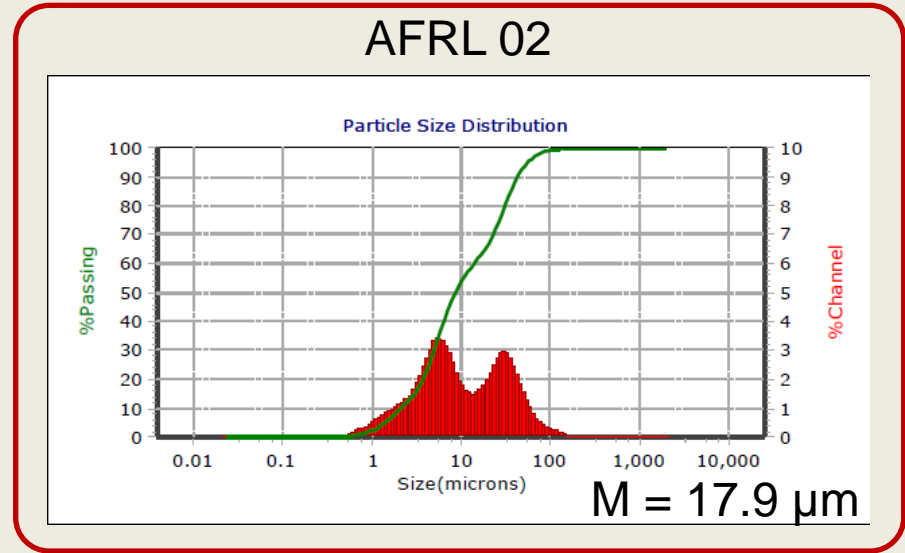
PTI Information

AFRL 02

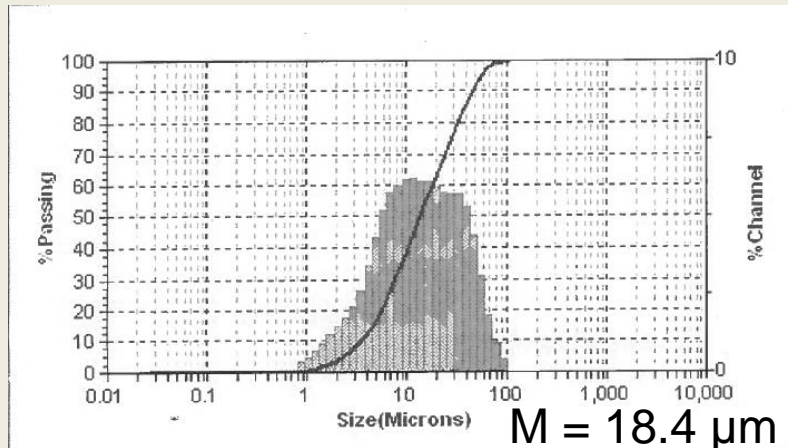


ARL Demonstration

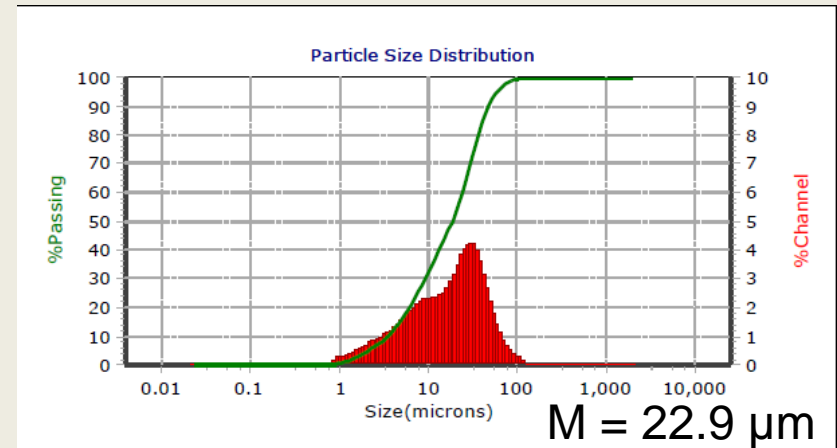
AFRL 02

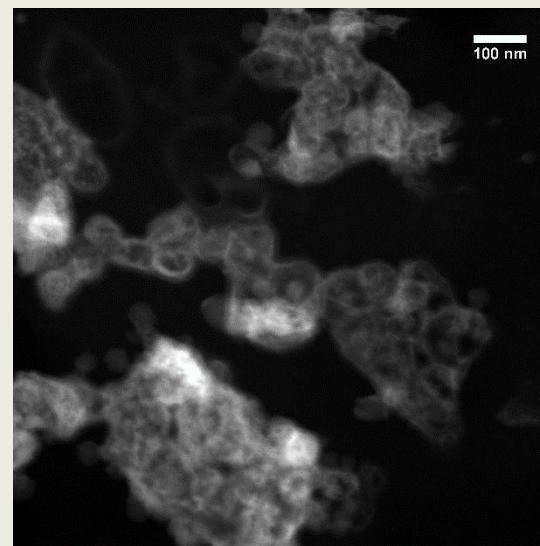
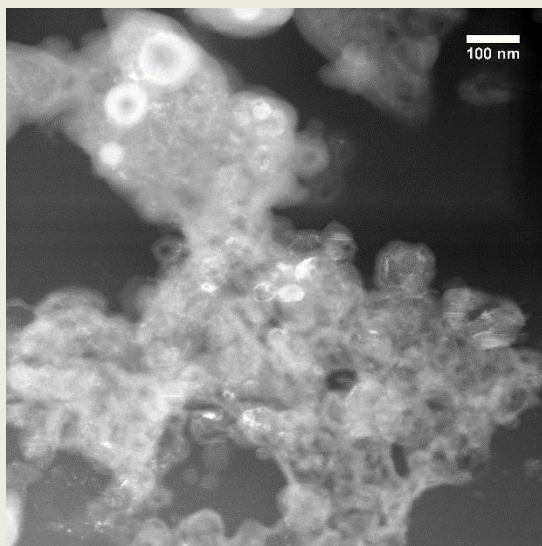
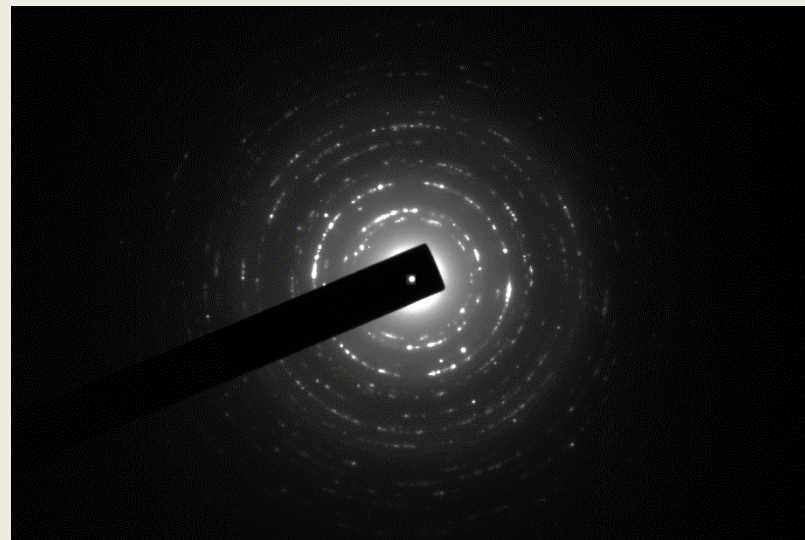
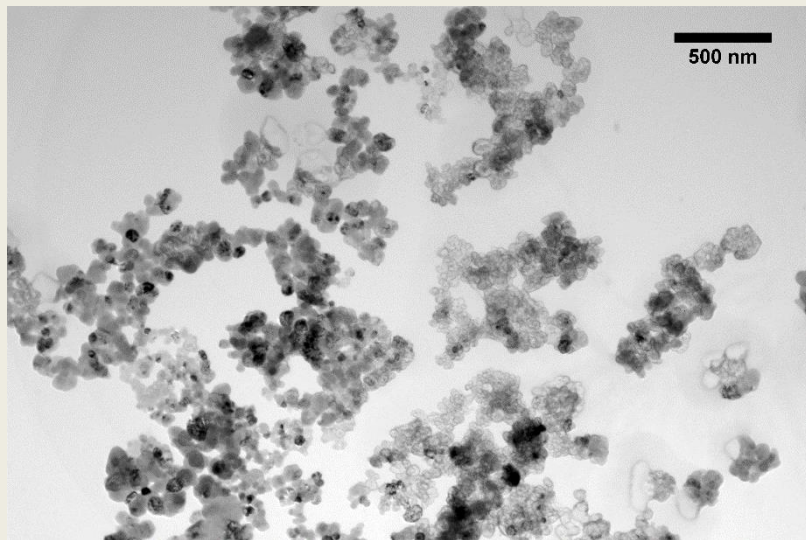


AFRL 03



AFRL 03



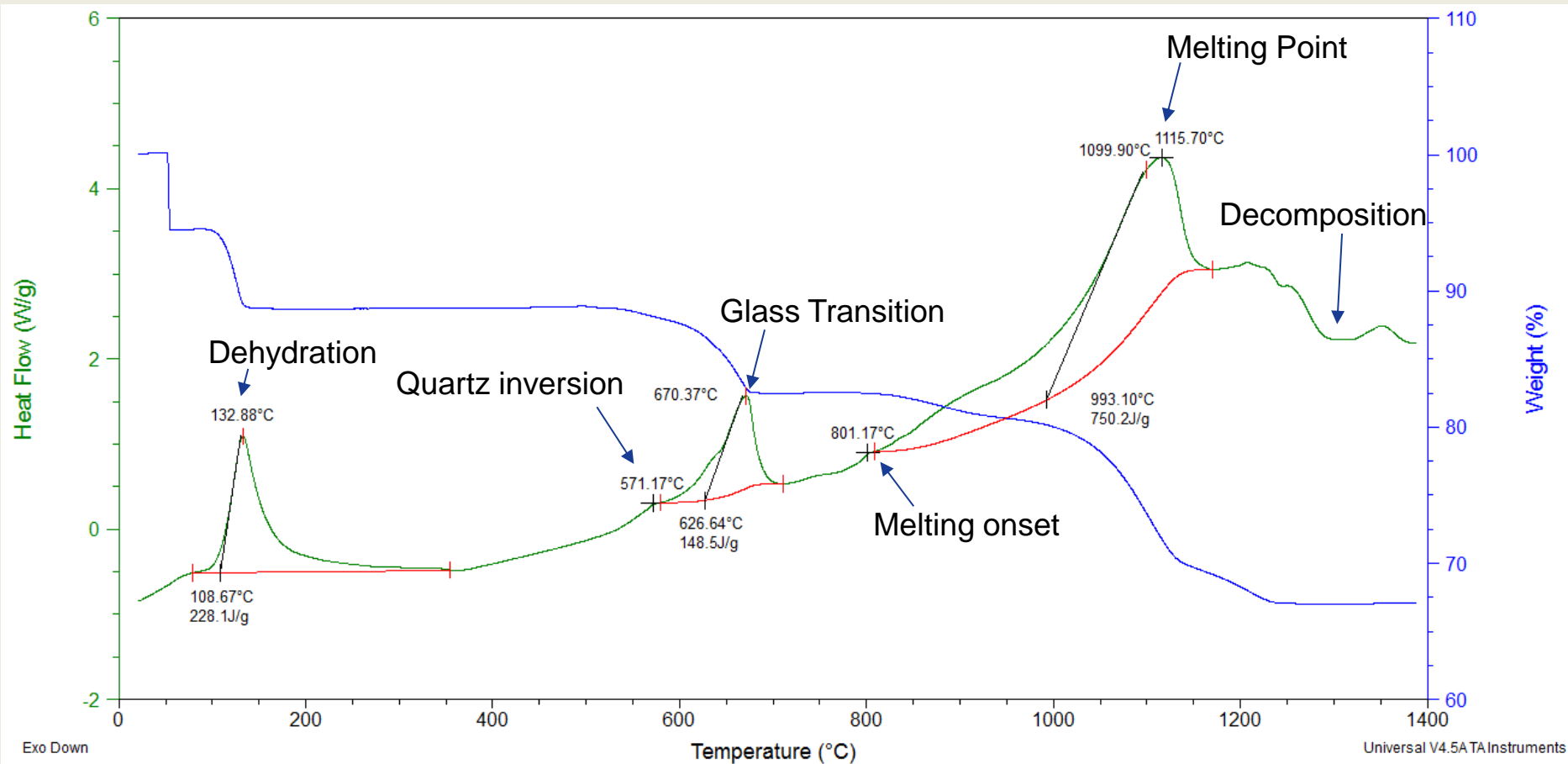




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Differential Scanning Calorimetry AFRL-02



under Ar gas flow



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Optical Microscopy (OM) - Nozzle Cross-Sections



ARL-NASA-02

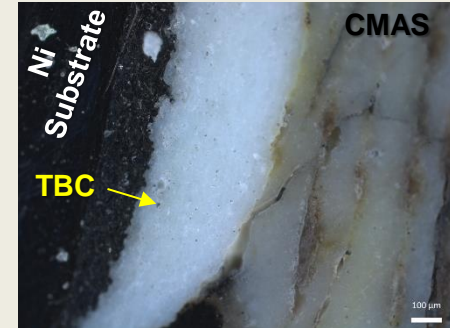
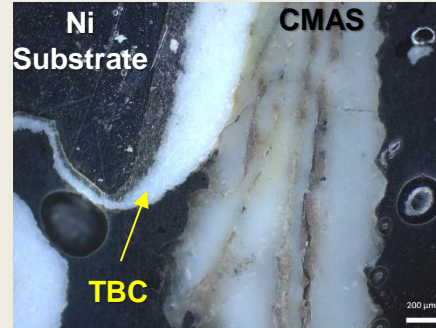
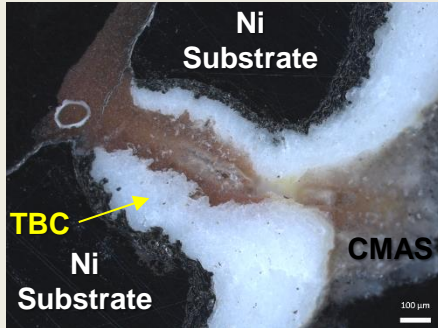
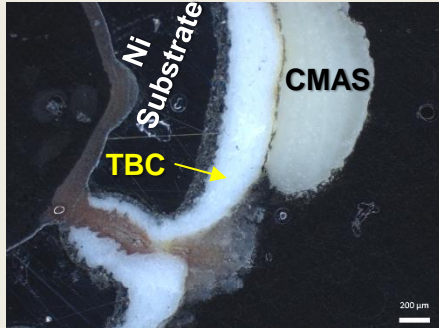
ARL-NASA-03

50x Dark Field

100x Dark Field

50x Dark Field

100x Dark Field

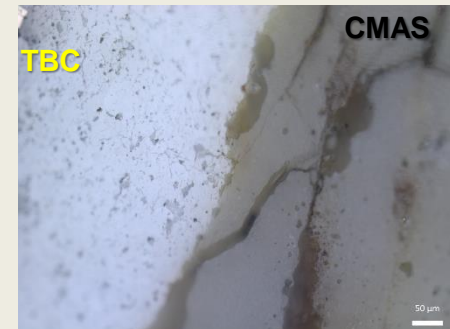
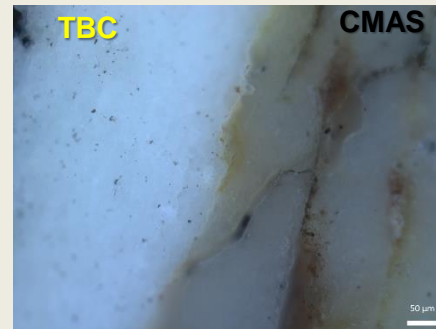
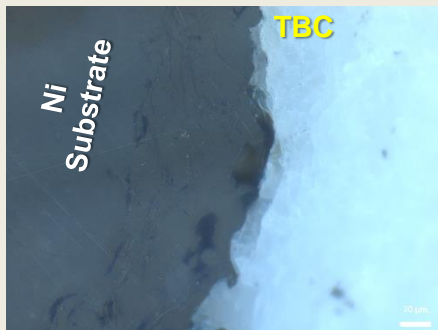
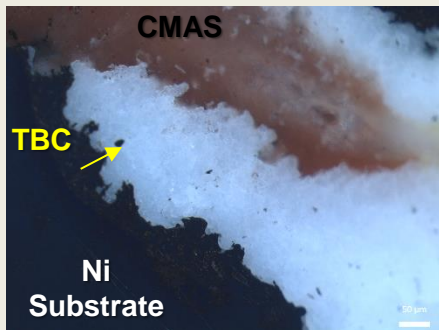


200x Polarized

500x Polarized

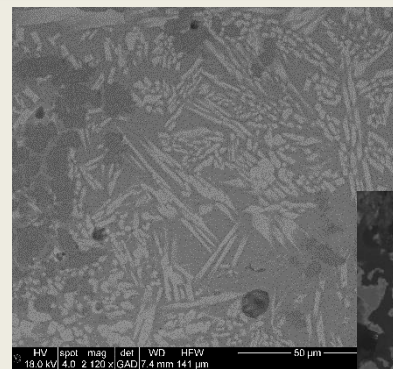
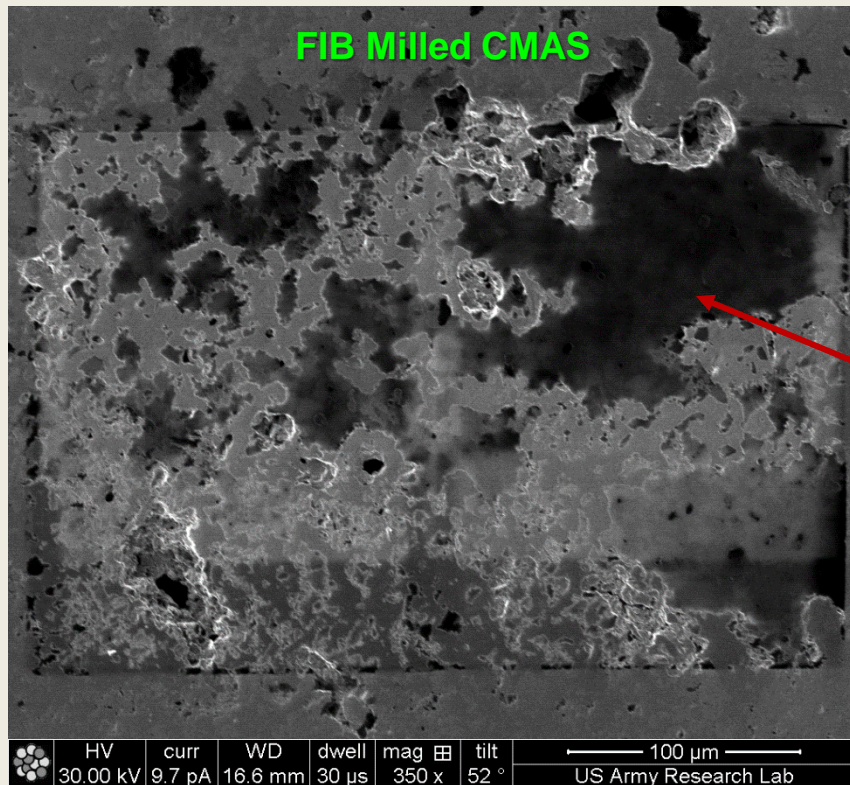
200x Polarized

200x Bright Field

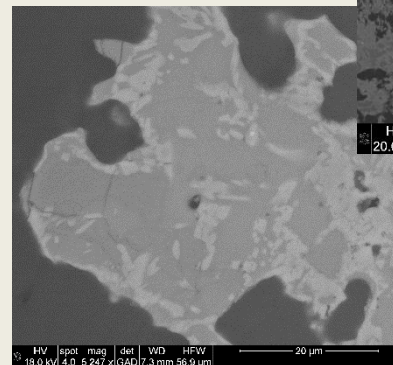
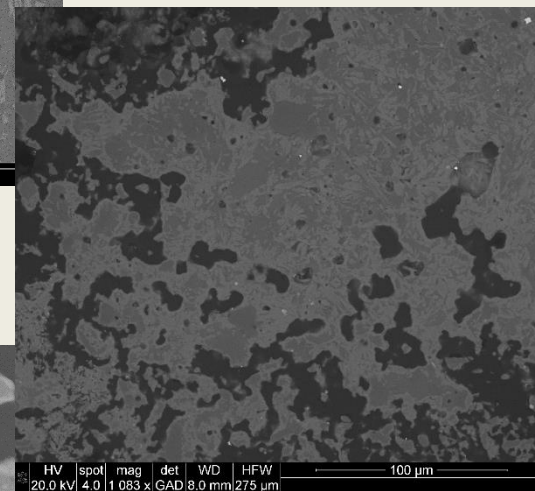


Leading Edge (LE)

Pressure Side (PS) - Near TE



**Pores
(Resin milled away)**



- **CMAS deposits exhibit chemical and microstructural variances**
- **Complex environment coupled with complex CMAS chemistry leads to a range of material responses and behaviors**



Model YSZ-based Sintered Compacts under Controlled CMAS Attack:

*Effect of Porosity on CMAS Infiltration and
YSZ/CMAS Interactions*



Approximately 25 pellets synthesized

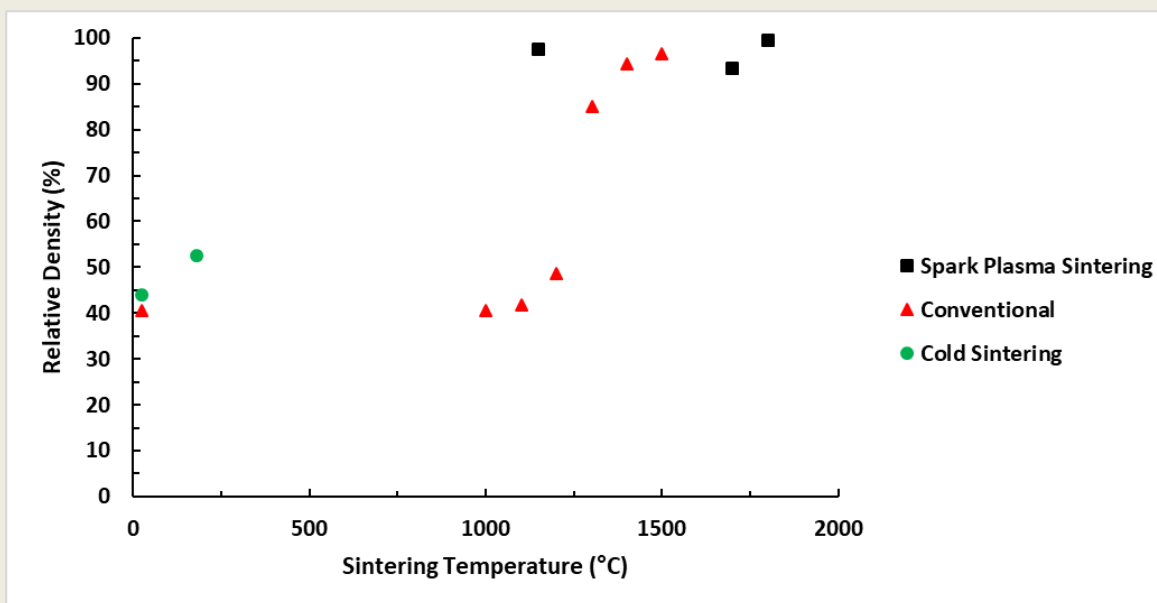
3 mol%, 8 mol%, 7 wt% (~4 mol%)

Conventional (pressureless), spark plasma, and cold sintering

Cold Sintering: ~50% relative density

Spark Plasma Sintering: ~95-99.5% relative density

Conventional Sintering: ~40-95% relative density

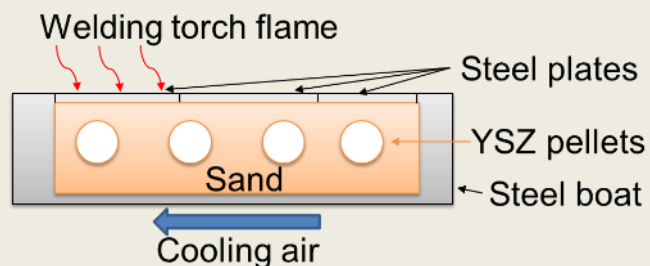


Comparison of relative densities achieved through varying sintering methods

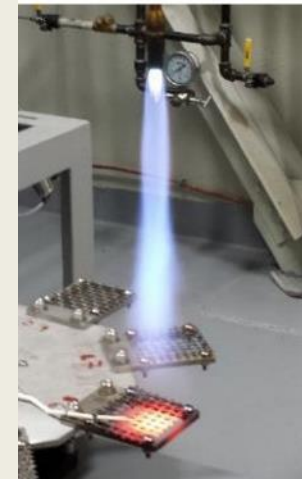


Button Cell Flame Test Rig

- Pellets buried in sand
- 1300°C
- 15 min exposure time
- Stationary contact



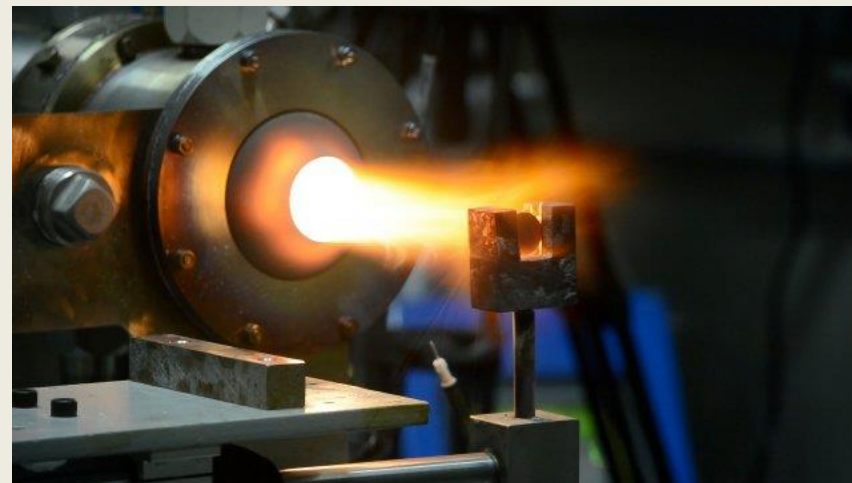
Button Cell Flame Rig



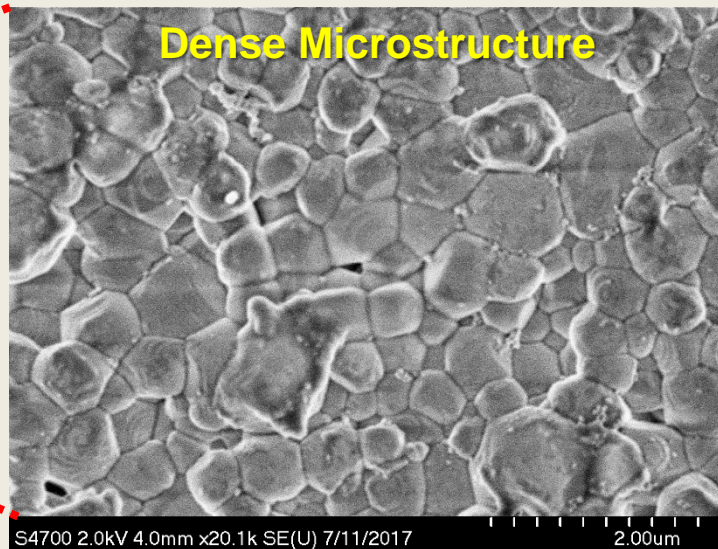
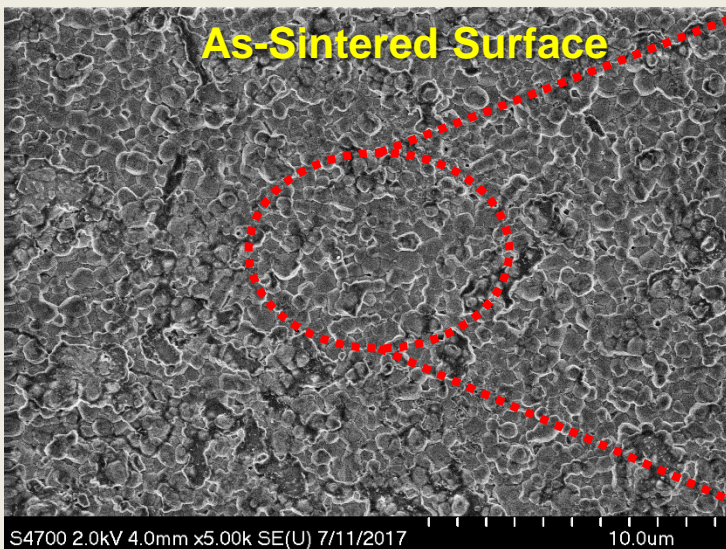
Schematic of the button cell flame rig along with the actual set-up

Hot Particulate Ingestion Rig

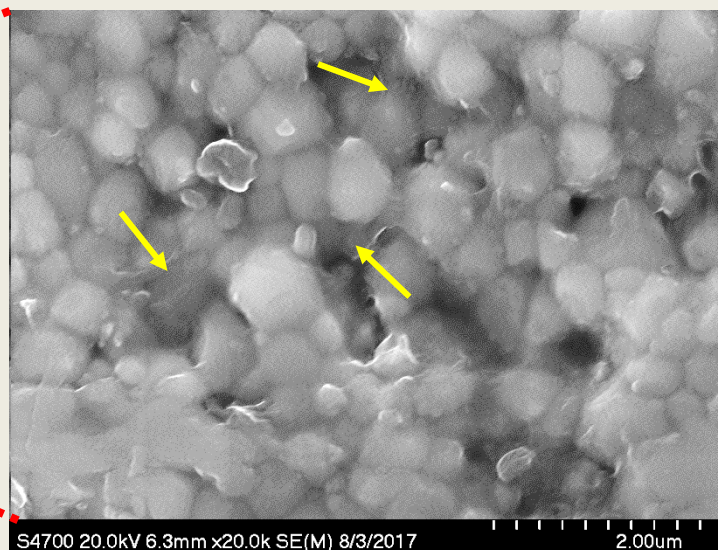
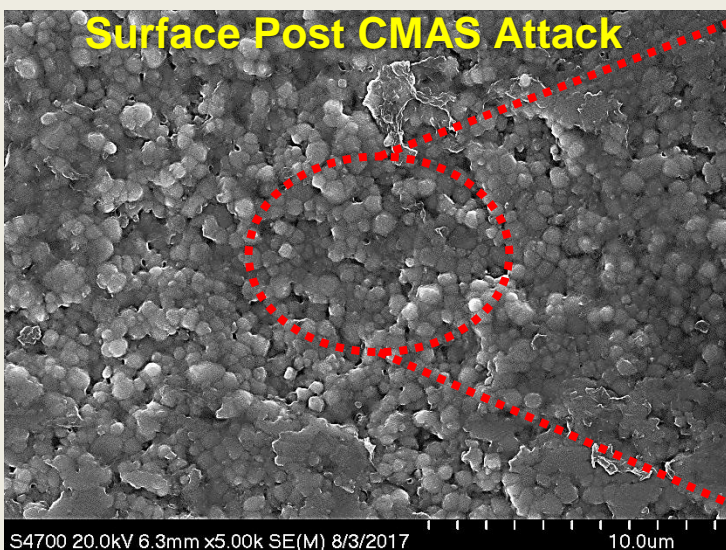
- Replicates temperature and velocity conditions of a jet engine
- Settings:
 - 1300°C
 - 0.3 M
 - 1 g of sand/min,
 - 3 cycles of 5 min exposure; 15 min total
- Dynamic CMAS Contact



Hot particulate ingestion rig in operation



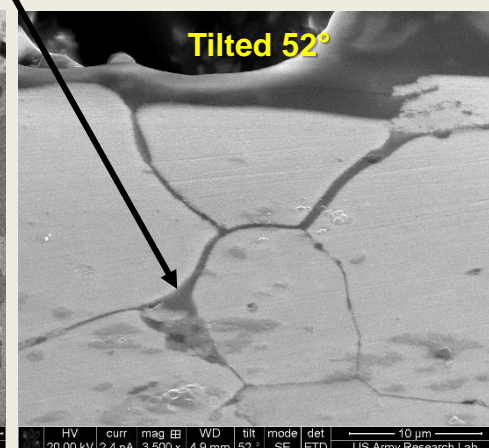
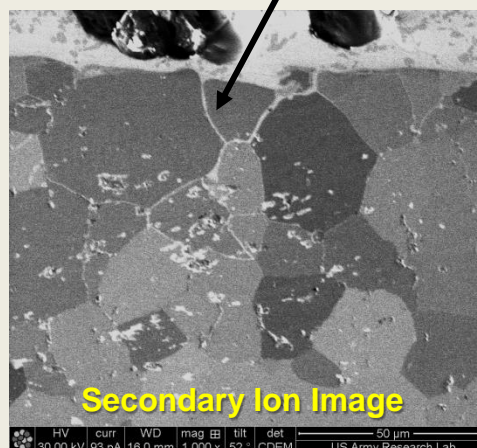
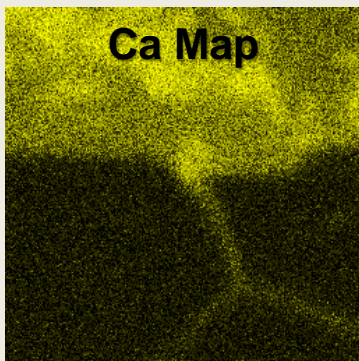
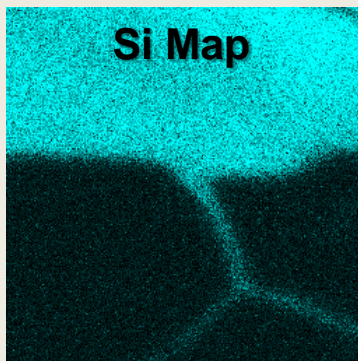
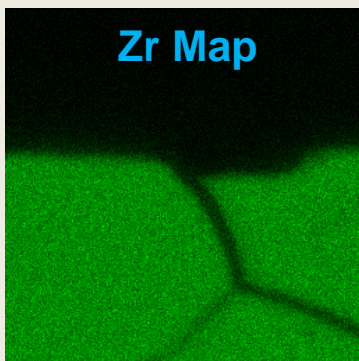
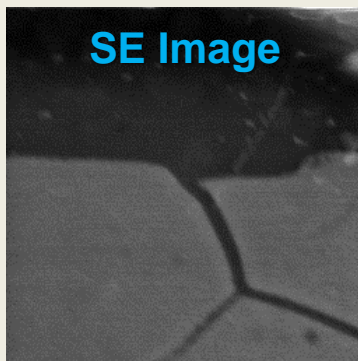
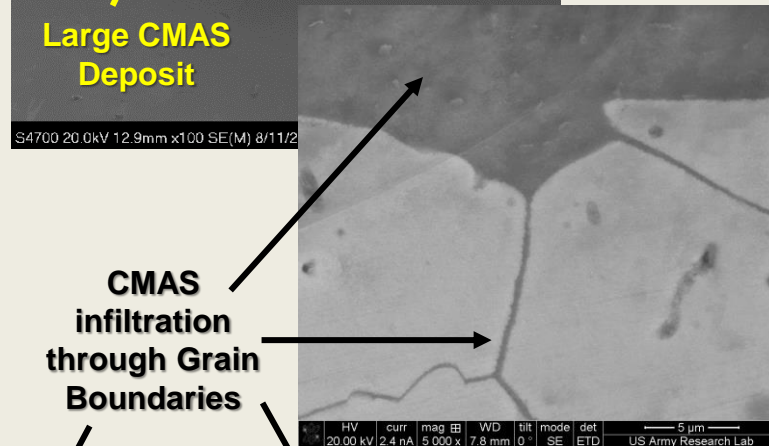
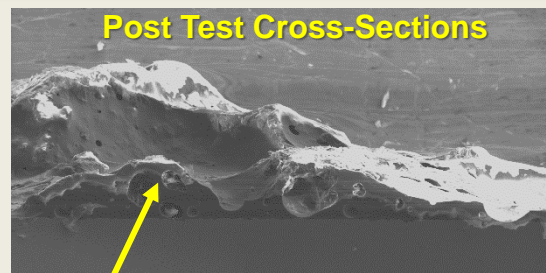
- SPS consistently produced densest pellets
 - 97% Dense pellet exhibits dense microstructure on as-sintered surface
 - Grain structure is relatively fine (~ 1 – 2 μm)



- Post CMAS attack grains appear to be covered in CMAS 'glaze'
 - Thin, transparent CMAS strands seen
 - Thin deposits not seen via visual inspection
 - Grains appear to be less faceted, suggesting possible dissolution

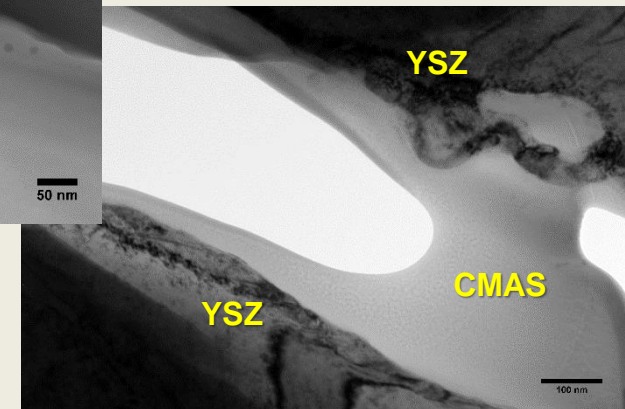
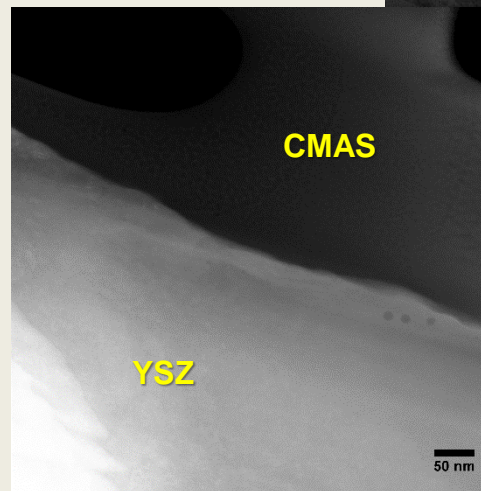
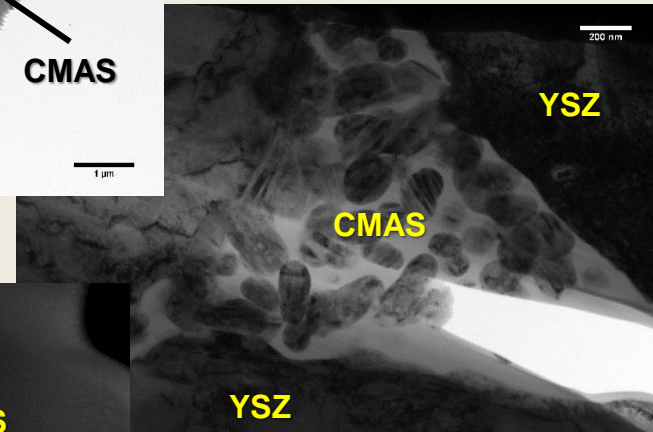
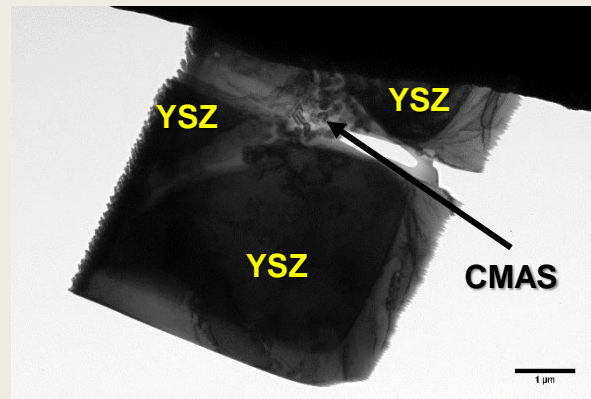


- Highly dense pellets were expected to mitigate CMAS attack by preventing infiltration
 - Lack of pores eliminates facile pathway for infiltration
- CMAS infiltration is observed on near surface, adjacent to surface deposits
 - Infiltration depth over 50 μm into bulk





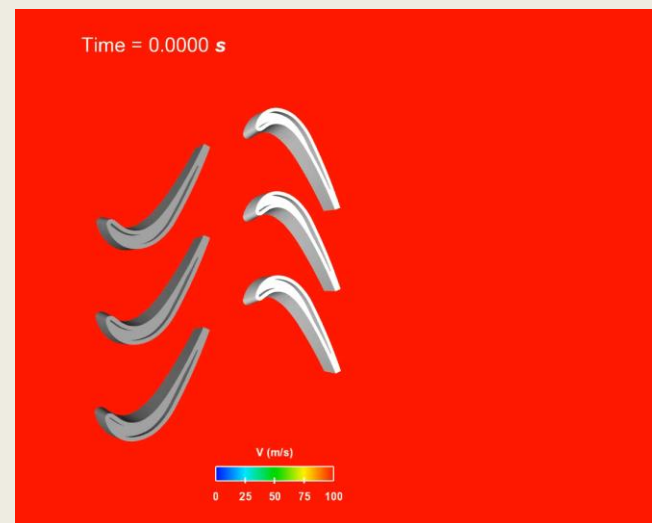
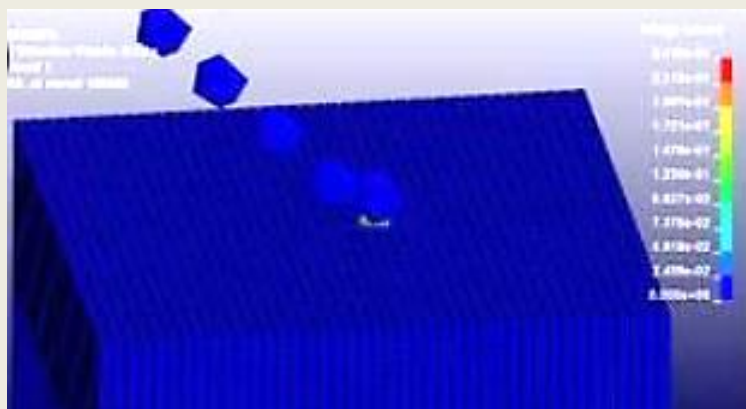
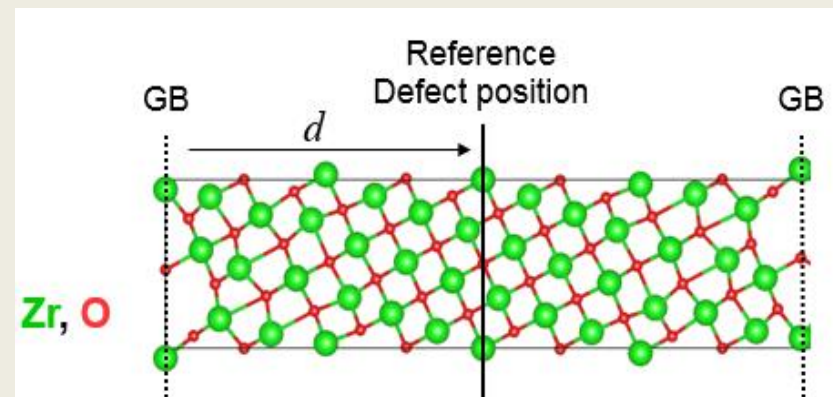
- TEM studies underway to elucidate CMAS infiltration mechanism in the absence of porosity
- Nanometric YSZ particles at triple junction exhibit severe twinning
 - Not seen in YSZ grains away from CMAS
 - Deformation twinning suggests CMAS induced strains play a role in infiltration/dissolution/reaction mechanisms
- STEM/EDS/EELS analysis will attempt to determine atomic scale diffusion mechanisms at CMAS/YSZ interface





Computational Materials Modeling

- Flow field modeling in gas turbine engine to determine regions in components exposed to most severe CMAS attack and accumulation conditions
- Molecular dynamics (MD) simulations on CMAS constituent element segregation on YSZ boundaries, with Prof. Kesong Yang at University of California San Diego
- Particle impact/adhesion simulations on EBC/CMC systems

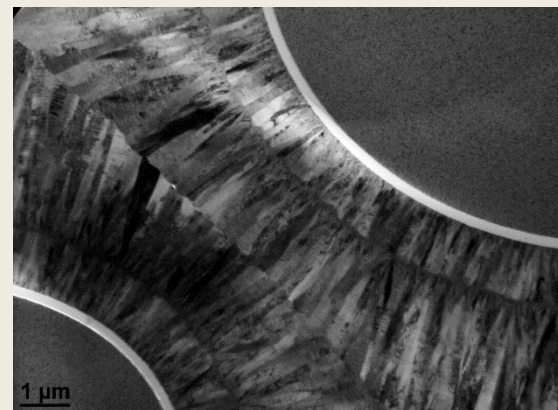
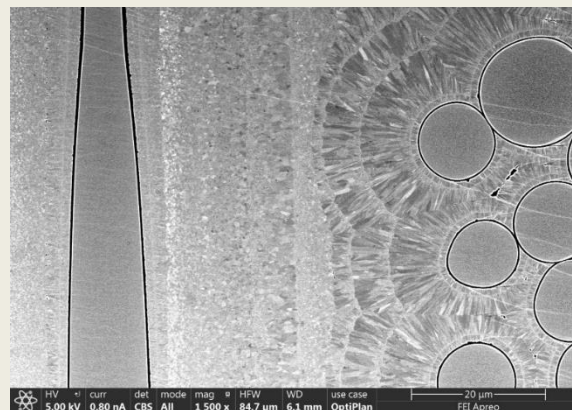




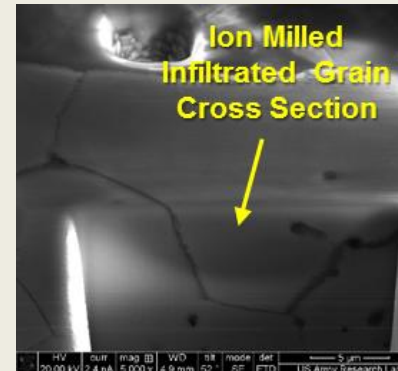
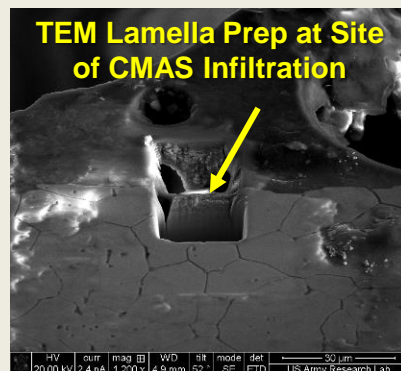
Nanoscale Interfacial Characterization

- Evaluation of CMAS kinetics in RE silicate / disilicate EBCs on SiC and CMC substrates will be conducted at ARL under engine relevant conditions
- Site specific TEM studies (EELS, SAD) will be conducted at CMAS/TBC/EBC/CMC interfaces and on CMAS phases.
- Evaluation of microstructure and properties of various commercially available CMCs and ceramic fibers.

CMC Cross Section



Site Specific TEM Characterization via FIB

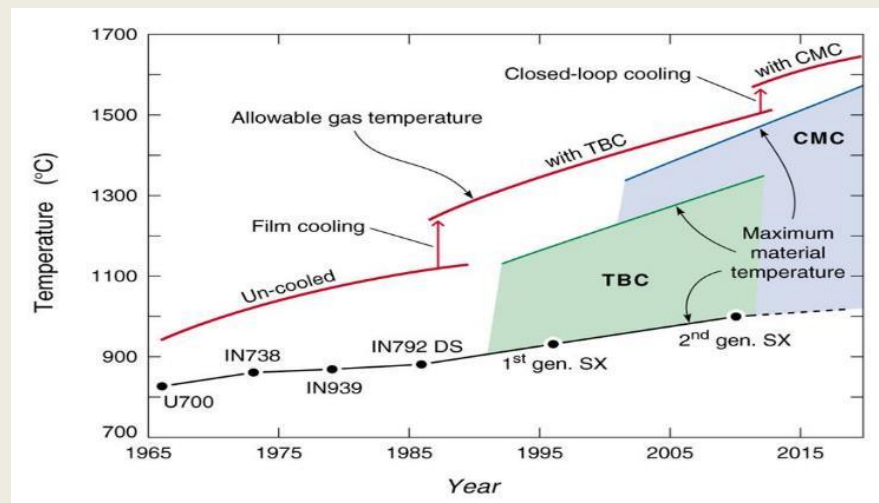
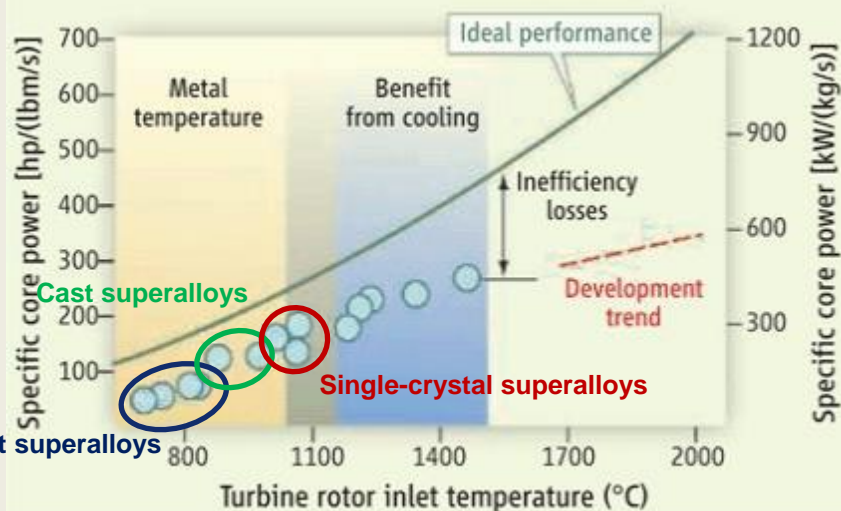




U.S. ARMY
RDECOM

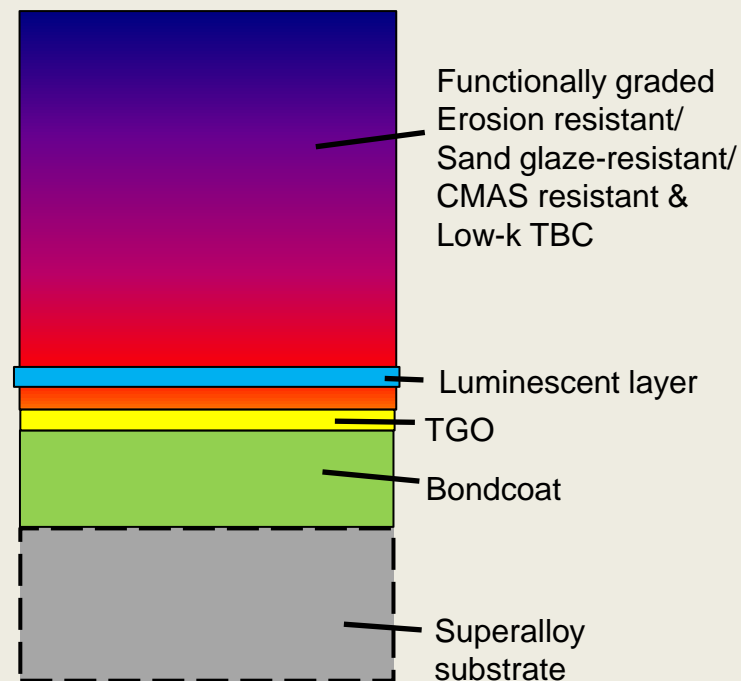
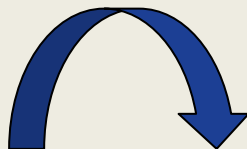
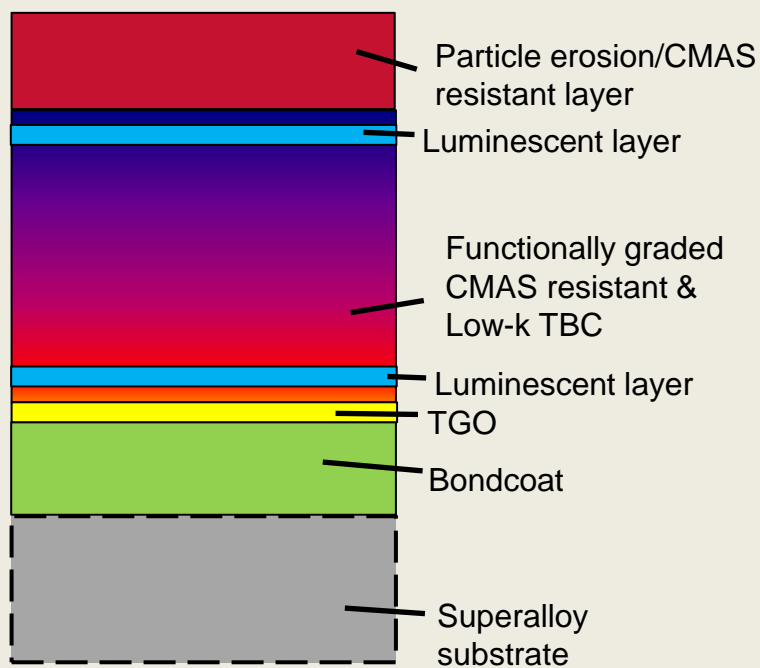
ARL

Back-up



J.H. Perepezko, Science 326, 1068 (2009); used with permission Source: <http://www.virginia.edu/ms/research/wadley/high-temp.html>

- **Current state-of-the-art turbine nozzles & blades:**
 - Single-crystal Ni superalloy blades
 - Metal bond-coat: (Pt, Pd)Al
 - Ceramic thermal barrier coating (TBC): 7 wt % Y_2O_3 : ZrO_2
- **Other hot section components (such as shrouds and combustor liners):**
 - Polycrystalline, cast Ni superalloys
 - M-CrAlY (NiCo-CrAlY) bond-coats
 - Ceramic TBC: 7 wt % Y_2O_3 : ZrO_2



Hybrid Approach (Layered & Graded)

Functionally Graded Approach

- ❖ Luminescent layers will provide self-aware coating capability
- ❖ Functionally graded layer has multifunctional capability



Identification of Primary Mechanism(s) for Sand Accumulation and its abatement include:

- **Surface finish improvement**
- **Surface debris 'wetting' reduction/repellant**
- **Ablative**
- **Limit infiltration through microstructural tuning**

Primary Mechanism(s) for Sand Melt/ CMAS Infiltration depth, glassification and mitigation:

- **Viscosity and surface tension of the melt**
- **Operational temperature and surface temperature of the substrate**
- **Shape of the inter-columnar gaps**
- **Thermal conductivity and Porosity of TBC**
- **Size and shape of original sand particulate (spherical vs nonspherical)**



- Oxy-propane torch
- Motorized rotary stage
- Temperature measurement → Optical pyrometer
- Test parameters
 - Surface temperature ~ 1300 °C
 - 3 cycles – IN/OUT: 3 min/3 min
 - AFRL-02 sand → slurry deposited on surface, and allowed to dry before test



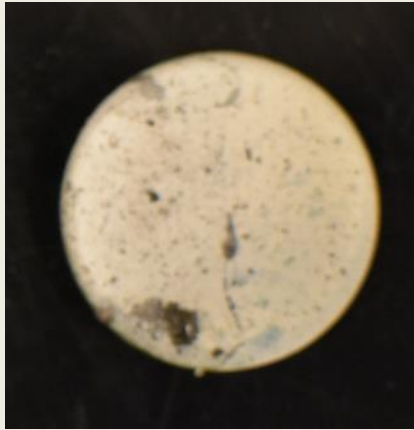


Button cell Flame Test Rig

Pre-test
YSZ, topped with YSZ-GdO



Post-test
YSZ, topped with GdO



Post-test
YSZ, topped with YSZ-GdO

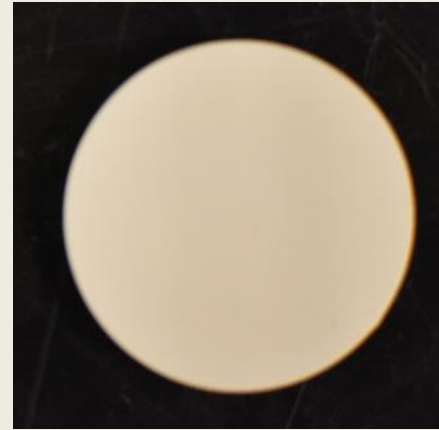


Hot Particulate Ingestion Rig

Pre-test
MAX-phase: Ti2AlC



Post-test
Low-k ZrO2-based



Post-test
MAX-phase: Ti2AlC





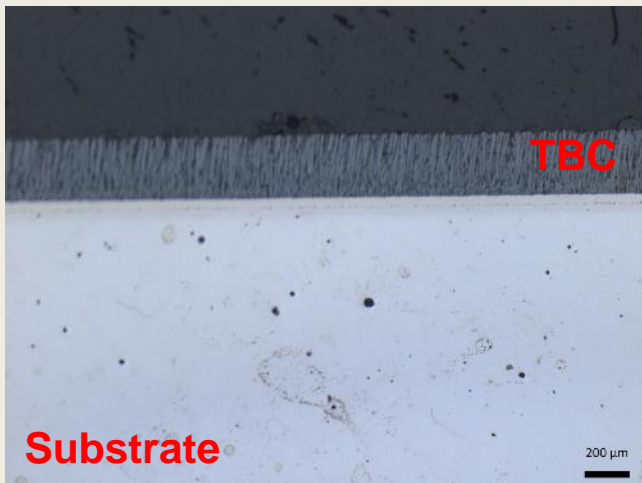
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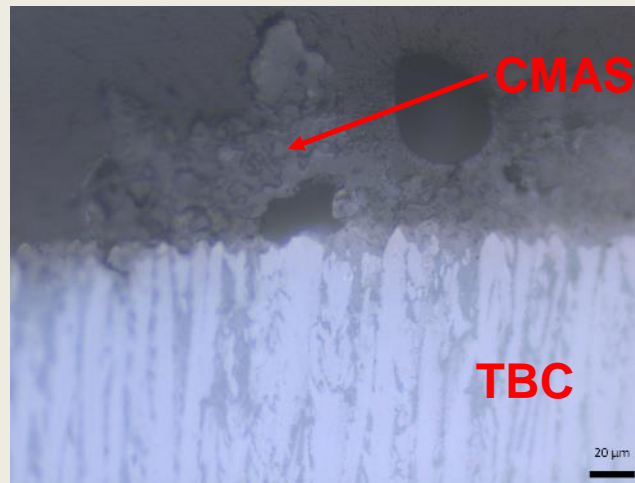
Optical Microscopy (OM) HM-3848



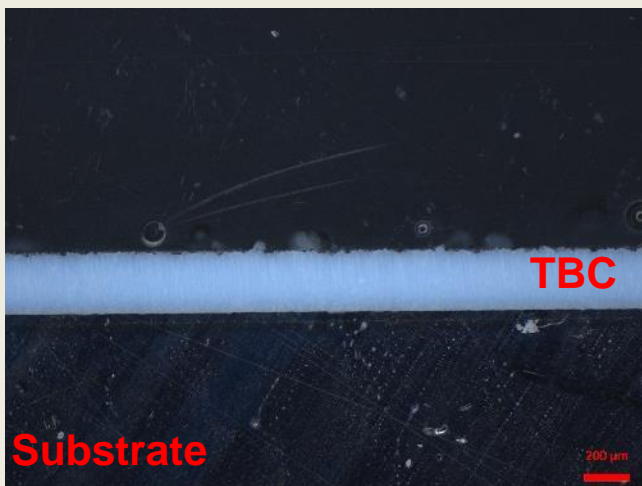
BF @ 5X



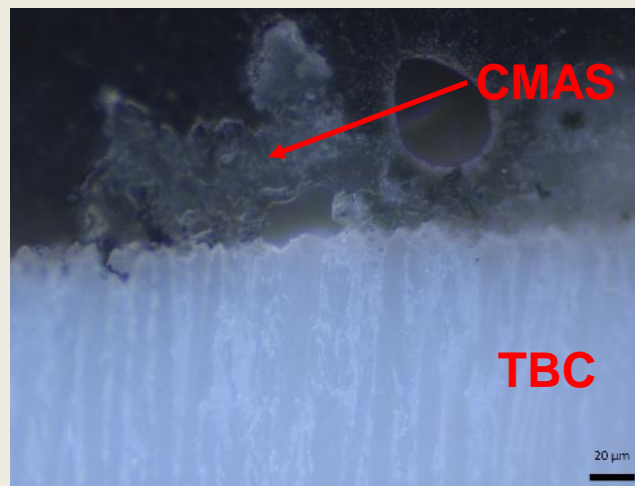
BF @ 50X

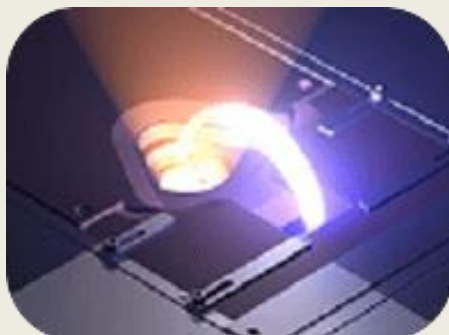
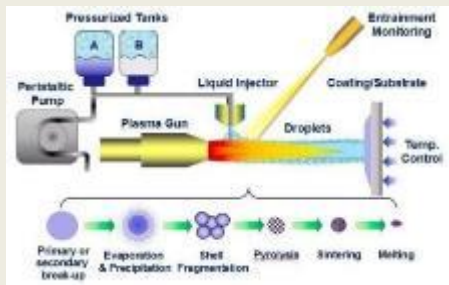
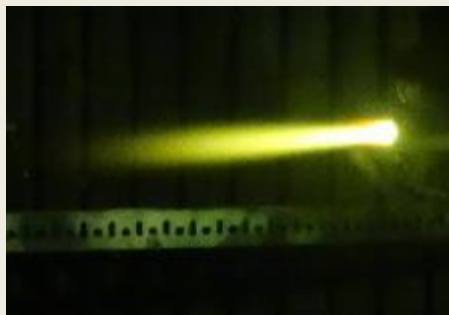


DF @ 5X



DF @ 50X





ID	Processing	Coating Architecture	Post-deposition treatment	Suspected Mitigation Mechanism
1	SPPS	8YSZ	None	Finer microstructure, with limited infiltration paths
2	APS	7YSZ / Gd ₂ O ₃	None	Reduced wetting by debris
3	APS	(7YSZ+Gd ₂ O ₃ blend) / GZO	None	Crystallization of deposits
4	APS	7YSZ / GZO	None	Crystallization of deposits
5	APS	7YSZ / (7YSZ + Gd ₂ O ₃ blend)	None	Reduced wetting by debris
6	EB-PVD	Low-k ZrO ₂ -based	None	Reduced wetting by debris
7	EB-PVD	low-k HfO ₂ -based	None	Reduced wetting by debris



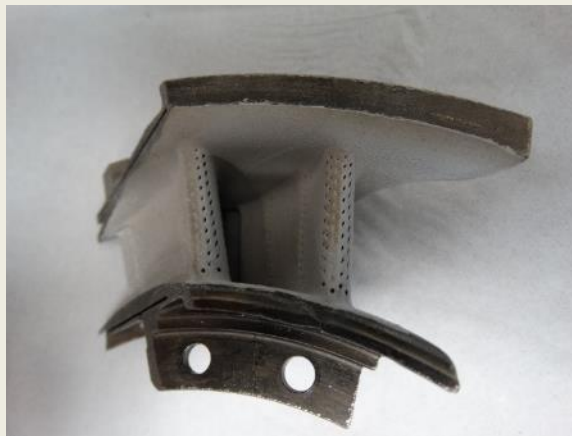
Nozzle 1 – 7YSZ via APS



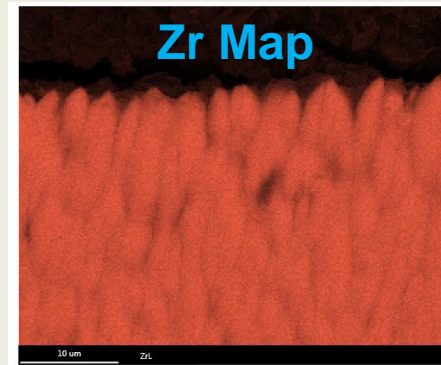
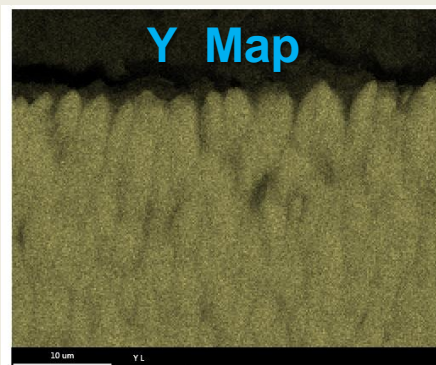
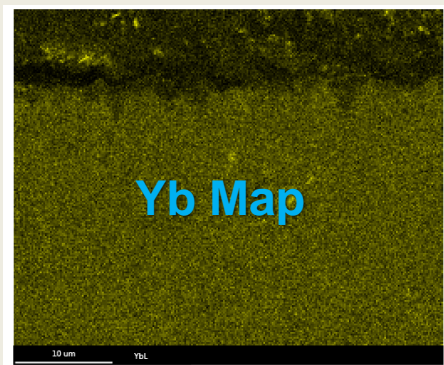
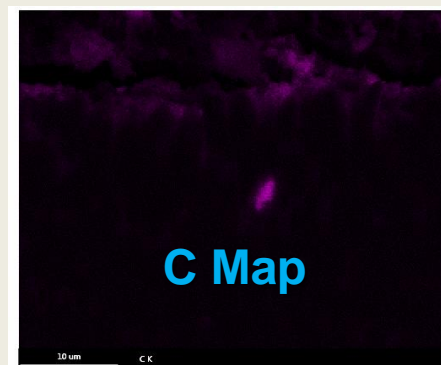
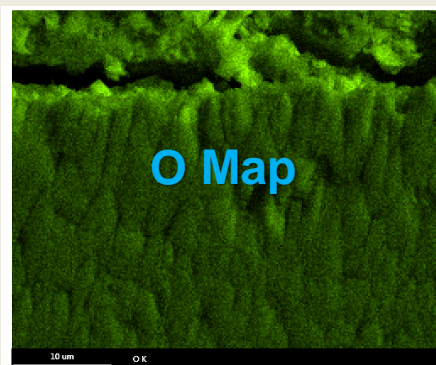
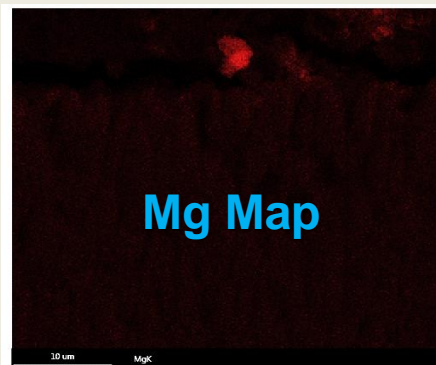
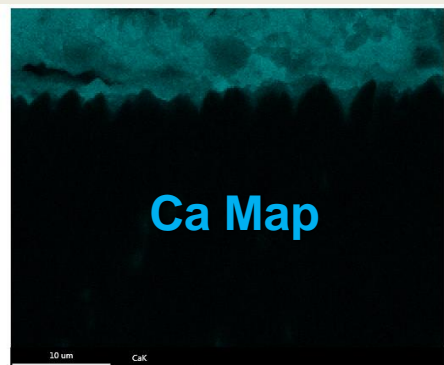
- Leading Edge Holes are largely clear
- Clogging increases as you approach the vane's trailing edge

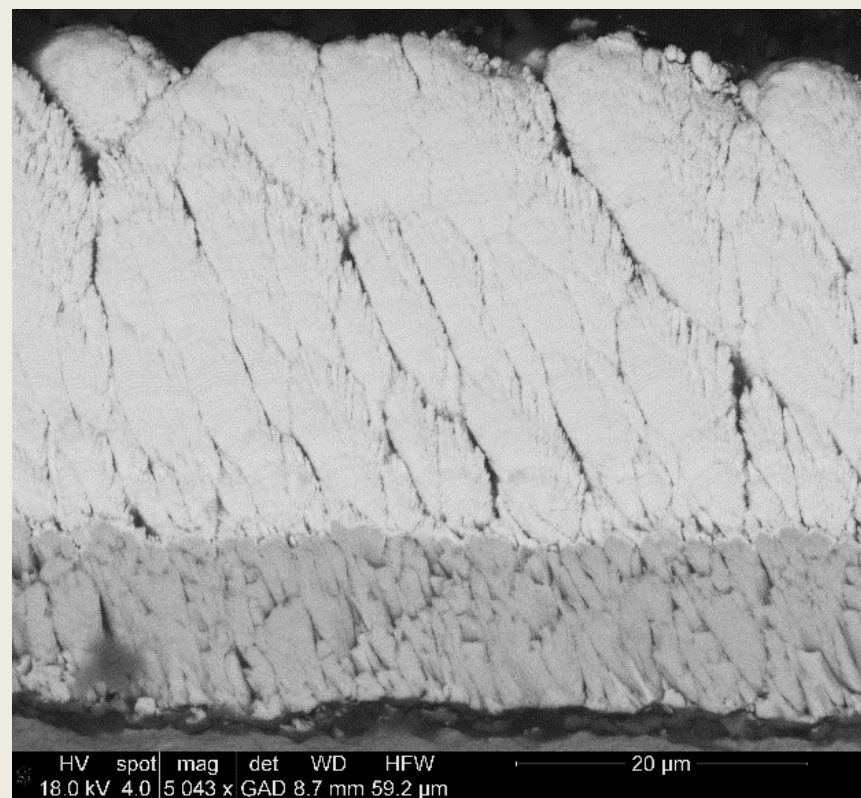
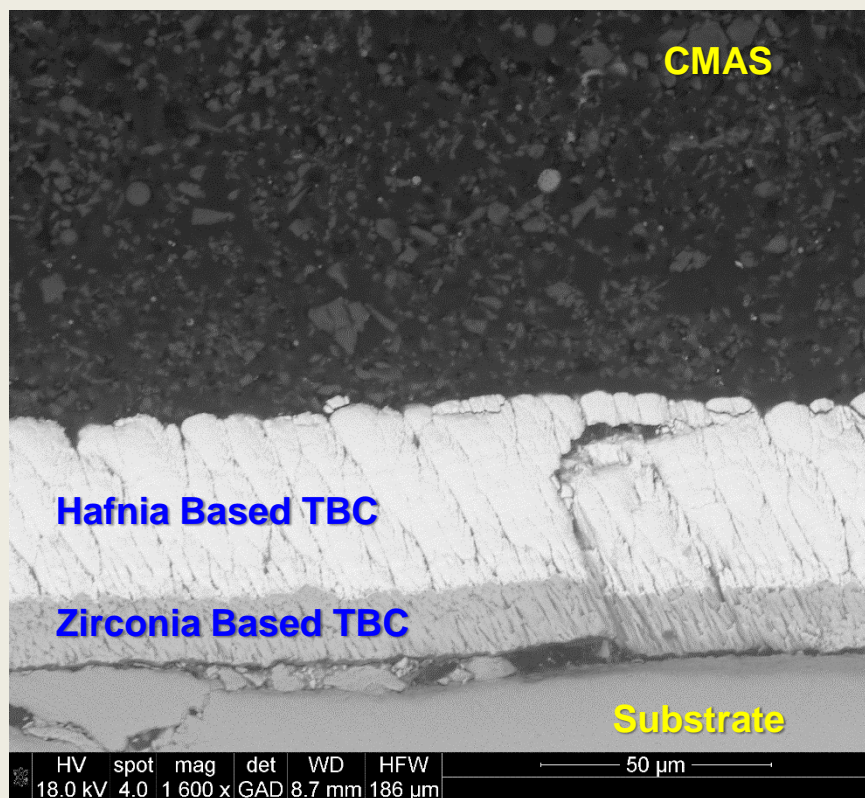
NOTE: Samples were grit-blasted for coating adhesion

Nozzle 2 – 8YSZ via SPSS



- Cooling hole row at the peak of the leading edge is clogged
- Trailing edge shows less clogging than Nozzle 1, despite identical spray path
- Finer Coating Droplets (Solution Plasma Topcoat) appear to bridge cooling holes more easily/thoroughly, despite increased bleed air inlet pressure/flow.





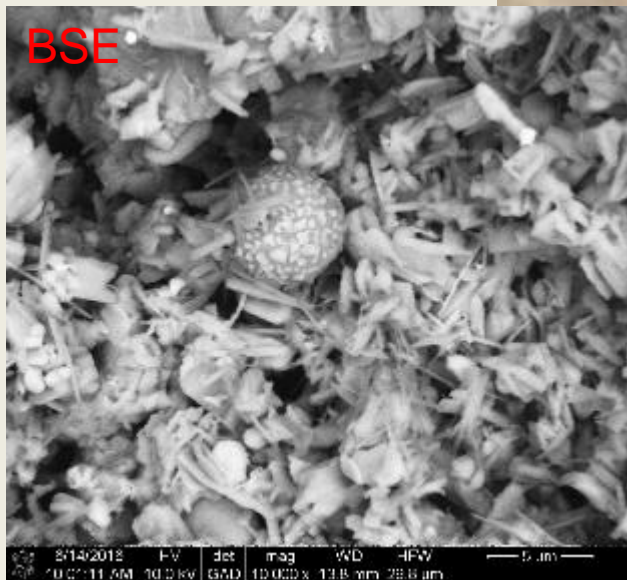
- ARL-NASA-07 shows signs of delamination from the substrate, as well as significant cracking on TBC itself (*aside from columnar structure*)
- Columnar structure of hafnia based EB-PVD TBC is significantly different from that of 5YSZ EB-PVD TBC (*ARL-NASA-07 is also much thinner*)



- CMAS constituents exhibit rich variety of morphologies and microstructures
- Different CMAS constituents will have distinct reactions to high temperatures and different interactions w/ TBCs



BF



- Most CMAS deposits are very rough and porous
- CMAS deposits can be white, red, or both.

2/14/2012 HV det mag WD HFW
10.0 kV GND 10.000 x 11.8 mm 28.6 μm



As-Sintered Pellets



**Pellets Post
CMAS Attack
Testing**



**Adhesion,
morphology, and
size of CMAS
deposits can be
seen to vary
significantly**

