

Conductometric Hydrogen Gas Sensor Based on Polypyrrole Nanofibers

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Abstract—Polypyrrole nanofibers are synthesized through a template-free chemical route and used as the active component for hydrogen gas sensing at room temperature. The synthesis of polypyrrole nanofibers was achieved by using bipyrrole as an initiator to speed up the polymerization of pyrrole with FeCl_3 as the oxidizing agent. Scanning and transmission electron microscopy studies indicate that the resulting polypyrrole forms a nanofibrous mat with average nanofiber diameter of 18 nm. Fourier transform infrared spectroscopy and elemental analysis confirms that the structure of the nanofibers is comparable to bulk polypyrrole. Gas sensing properties of polypyrrole nanofibers were investigated by depositing nanofiber dispersions on an interdigitated conductometric transducer. The sensor performance was tested through programmable exposure towards different concentrations of hydrogen gas diluted in synthetic air in an environmental cell at different temperatures. A short response time of 43 s was observed upon exposure to a concentration of 1% hydrogen with a decrease in film resistance of 312 Ω at room temperature. The sensor sensitivity was analyzed with gradual elevation of the operating temperature.

Index Terms—Conducting polymers, hydrogen sensors, polypyrrole nanofibers.

I. INTRODUCTION

CONDUCTING polymers such as polypyrrole, polyaniline, polythiophene, and their derivatives have been investigated for gas sensing applications for more than 20 years [1]. Gas sensors based on conducting polymers have many features that are superior to their metal oxides counterparts including high sensitivities, short response time, and room temperature operation. In contrast, metal oxides generally operate at elevated temperatures around 300 °C [2]. Conducting polymers are easily synthesized through chemical or electrochemical

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processes, and new polymers can be designed by the substitution of different functional groups onto the polymer backbone.

Polypyrrole and its derivatives (PPy) are formed through the oxidation of pyrrole or substituted pyrrole monomers. The oxidation is usually carried out by either electropolymerization on a conductive substrate (electrode) with the application of an external potential, or by chemical polymerization in solution through the use of a chemical oxidant. The two methods produce PPy with different chemical and electrical properties [3]. Changes to the electrical conductivity of PPy have been noted upon exposure to various organic and inorganic gases. Blanc *et al.* [4] measured the conductivity of PPy films obtained by electropolymerization method under exposure to O_2 , NH_3 , NO_2 diluted in synthetic air or N_2 at 100 °C. It was established that a wide range of organic vapors can alter the conductivity of PPy, which facilitated its application in field effect transistor (FET)-based gas sensors [5]–[8]. A CO sensor was fabricated by Liu *et al.* [9] by growing a PPy film on the surface of an interdigitated-capacitive transducer using the electropolymerization technique. Recently, Waghuley and coworkers [10] reported a CO_2 sensor based on PPy film synthesized using chemical polymerization of the pyrrole monomer.

Although PPy thin films can be very easily made by electrochemical deposition, their performance as gas sensing films is affected by the fact that the active sensing components remain embedded in the bulk, which limits both the efficiency and the sensitivity. This can be improved by making PPy in a nanofiber form to generate high surface area for a given mass or volume, which can enhance the sensitivity by providing a porous structure that has better interactions between the sensing film and the target gas molecules [11]. In order to enhance the chemical sensing properties of PPy, an extensive amount of research work has concentrated on the synthesis of nanostructured forms of PPy. The production of nanostructured PPy has been studied using templated synthetic methods employing mesoporous silica, anodized aluminum oxide membranes, and particle track-etched membranes [12], [13]. A bulk growth approach using V_2O_5 seeds as a template has also been reported recently [14]. The method of surfactant-mediated synthesis, which has been so successful in the synthesis of polyaniline nanofibers yields only nonfibrous, granular powders in the case of polypyrrole [15]. Fibrillar and tubular morphologies of polypyrrole has been observed when β -naphthalene sulfonic acid (NSA) or p-toluenesulfonate acid (TsOH) is used as a dopant during the synthesis process [16]. These fibers and tubes have diameters in the range of 50–2000 nm and are formed presumably as a result of the solution aggregation of the dopant anions. In general,

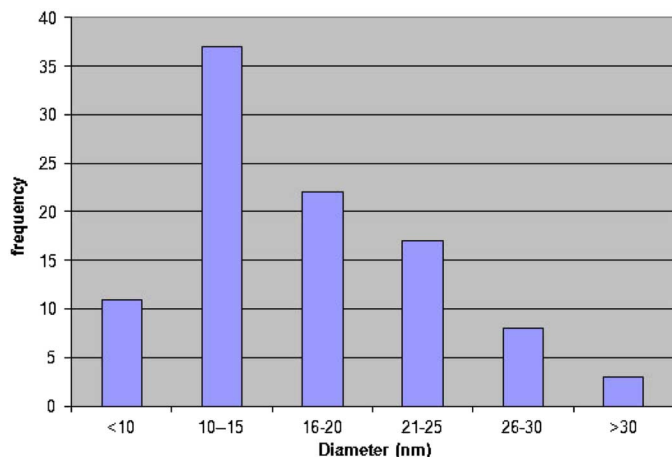


Fig. 1. Distribution of PPy nanofiber diameters resulting from a template-free synthesis using bipyrrrole as an initiator.

a straightforward bulk synthesis of nanofibers of polypyrrole directly from pyrrole monomer with average fiber diameter less than 100 nm, has been difficult to achieve.

The application of nanostructured conducting polymers in gas sensing has so far focused on polyaniline nanofibers [17], [18]. Polyaniline nanofiber films outperform their convention counterparts in terms of sensitivity and response time due to their higher surface to volume ratios, which provide better interaction between the film and the target gas molecules. According to the best of the authors' knowledge, this paper is the first study on PPy nanofibers thin films investigated for hydrogen gas sensing applications.

Here, we report the fabrication of a hydrogen gas sensor from polypyrrole nanofibers deposited on to conductometric transducers. Conductivity changes of the sensitive films upon exposure to different concentrations of hydrogen gas under variable operating temperature is measured.

II. SYNTHESIS AND CHARACTERIZATION OF POLYPYRROLE NANOFIBERS

A. Synthesis

All chemicals were purchased from Sigma-Aldrich. Monomers were distilled prior to use. Bipyrrrole was synthesized according to established procedures [19]. Reactions were performed in 20 ml glass vials in which pyrrole (50 mg, 0.74 mmole) and bipyrrrole (3 mg, 0.023 mmol) were dissolved in 10 mL of methanol and rapidly mixed with a separate solution of FeCl_3 (120 mg, 0.74 mmol) in 10 ml of deionized water. The reaction mixture was vigorously shaken for several seconds and then left unagitated for one day. The crude product was purified by centrifugation and washed multiple times with deionized water. The purified product is typically resuspended in deionized water to a concentration of 2 g/l. The resulting nanofibrillar morphology of PPy from this synthetic process was found to have a statistical distribution of nanofibrillar diameters, as shown in Fig. 1. The mechanism of polypyrrole nanofiber formation is described in great detail elsewhere [20].

B. Scanning Electron Microscope (SEM) and Transmission Electron Microscopy (TEM) Analysis

The morphology of PPy nanofibers was analyzed with a JEOL JSM-6700F Field Emission Scanning Electron microscope by drop casting a ~ 2 g/l dispersion of PPy nanofibers onto a silicon wafer and allowing it to dry in a clean room environment.

Transmission electron microscopy (TEM) (JEOL 100CX) was also used to characterize the PPy nanofibers. A scanning electron microscope (SEM) and a TEM [Fig. 2(a) and (b)] show that a nanofibrous mat of PPy has been created. The PPy nanofibers have an average diameter of 18 nm based on statistical analysis of the TEMs (Fig. 1) with lengths on the order of several micrometers.

C. Elemental Analysis

Elemental analysis (conducted with a Thermo Electron/FlashEA 1112 Elemental Analyzer) reveals that the ratio of C/N of the polypyrrole nanofibers (3.5) is consistent with the theoretical C/N ratio of pure polypyrrole (3.43) and lends further support that the observed nanofibers are indeed polypyrrole.

D. Fourier Transform Infrared Spectroscopy (FTIR) Analysis

Fourier transform infrared spectroscopy (FTIR) was used to analyze dispersions of PPy nanofibers. FTIR spectra were obtained with a JASCO FT/IR-420 using pressed potassium bromide (KBr) pellets. The spectrum shown in Fig. 3 reveals the characteristic bands of PPy nanofibers. Absorption bands are observed at 1488 and 1558 cm^{-1} (stretching vibrations of the pyrrole ring), 917 and 1200 cm^{-1} (stretching vibrations of doped PPy), and 1045 and 1315 cm^{-1} (C-N stretching vibrations and C-H deformations, respectively) indicating that the PPy nanofibers share the same molecular structure as that of established methods for making PPy [21].

III. EXPERIMENTAL

A. Sensor Fabrication

The sensor consists of two physical layers, namely, a sensitive thin film of PPy nanofibers that interact with the target gas through redox reactions and a conductometric transducer that transforms this reaction into an electrical signal to be captured by a computerized monitoring system.

For the transducer fabrication, platinum was sputtered to form the interdigitated electrode system on the front side and the heater resistor on the backside. Polished sapphire (Al_2O_3) was used as a substrate because of its excellent electrical insulating and heat conducting properties. An intermediate 30 nm thick layer of titanium was used to improve the adhesion of the platinum layer to the sapphire substrate. The structure was fabricated using a standard liftoff technique. A schematic diagram of the conductometric transducer utilized is shown in Fig. 4.

The sensing layer was formed by drop-casting a PPy nanofiber dispersion onto the interdigital fingers of the transducer surface. Afterwards, the solution was allowed to evaporate in a clean environment for one day. The thicknesses of the resulting PPy nanofiber thin films were measured and found to be $\sim 0.3\text{ }\mu\text{m}$.

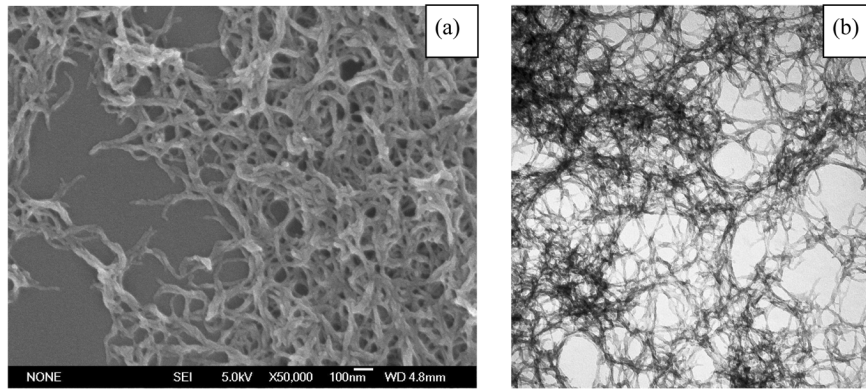


Fig. 2. (a) Scanning and (b) transmission electron micrographs of PPy nanofibers.

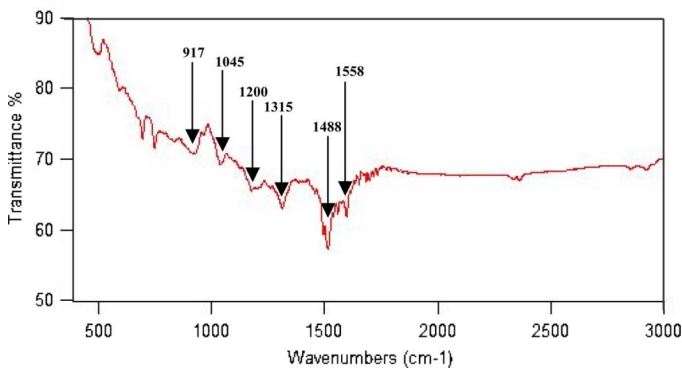


Fig. 3. FTIR spectrum of PPy nanofibers obtained by the template-free synthesis using bipyrrrole as an initiator.

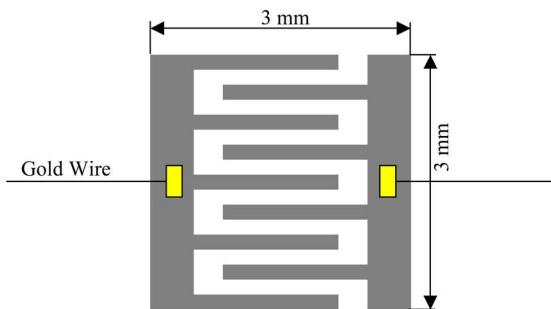


Fig. 4. Schematic diagram of the conductometric transducer pattern.

B. Computerized Gas Calibration System

The sensor was mounted inside an enclosed environmental cell. Four mass flow controllers (MFCs) were connected to form a single output that supplies gas to the cell. Teflon tubing was used to prevent atmospheric contamination. A constant flow rate of 0.2 liters per minute was delivered via the MFCs. The sensor was exposed to a hydrogen gas pulse sequence with different concentrations of hydrogen gas diluted in synthetic air at different temperatures.

A computerized gas calibration system was used to vary the concentration of H_2 gas in synthetic air. The sensor was connected in series with a Keithley 2001 multimeter and a computer was used to log data from the multimeter. The data was related to resistance variation over time. A programmable power supply was connected to the gas calibration system. Elevation

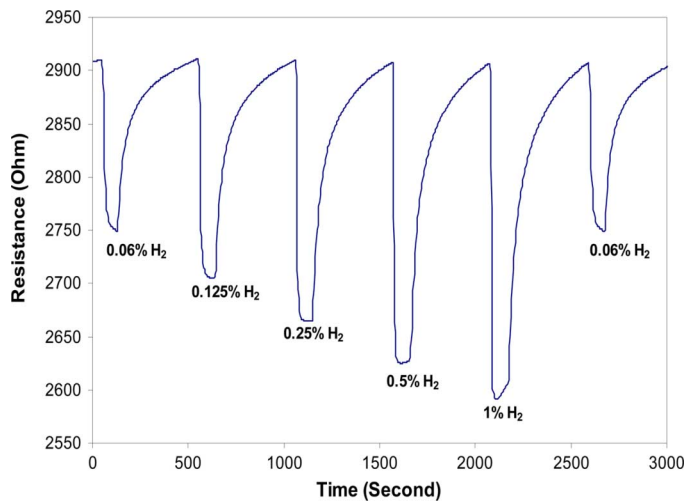


Fig. 5. The dynamic sensor response towards different concentrations of H_2 .

of the operating temperature was controlled by the same program that supplies gas pulses to the chamber. A 5 V increment at a time was used to raise the temperature from 22 °C–100 °C gradually, with synthetic air flowing in the chamber. The sensor was allowed to stabilize under synthetic air at each selected operating temperature setting before exposure to the sequence of hydrogen pulses diluted in synthetic air.

IV. RESULTS AND DISCUSSION

The sensor was found to have a short response time of 43 s upon exposure to a 1% H_2 pulse with a resistance change of 312 Ω . Fig. 5 shows the dynamic response of the sensor towards different concentrations of hydrogen gas at room temperature (22 °C). Resistance decreases of 157, 205, 245, and 282 Ω were obtained for 0.06%, 0.125%, 0.25%, and 0.5% H_2 with response times of 75, 72, 61, and 61 s, respectively. Hence, it can be concluded that the highest sensitivity is obtained for 1% H_2 with the shortest response time. Recovery time was fixed at 7 min under synthetic air after the exposure to each hydrogen pulse.

The polypyrrole nanofiber sensor is reproducible, as shown in Fig. 5, with the start of a new pulse sequence beginning with 0.06%. Testing with different operating temperature settings did not produce any significant change in the sensor response time or its selectivity towards different concentrations of H_2 gas. Although the concentration of H_2 is doubled in consecutive pulses

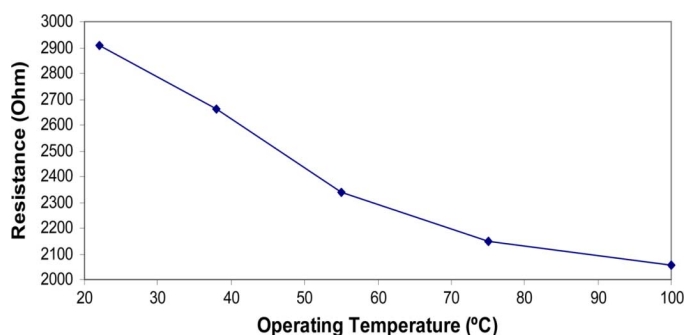
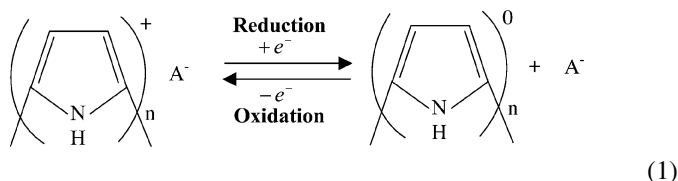


Fig. 6. The effect of elevating the operating temperature on film resistance in synthetic air only (no H_2).

the magnitude of the sensor response in Fig. 5 did not double, likely due to saturation of the sensitive layer with hydrogen. This is consistent with similar observation found for hydrogen sensors based on other conducting polymer (such as polyaniline) nanofibers [22].

The dynamic response is generated because of the interaction between the polymer layer and the adsorbed gas molecules. The interaction leads to a heterogeneous charge transfer reaction [23] and hence to a chemical modulation of the polymer doping level, which is directly related to the Fermi level of the organic semiconductor [24]. This effect results in a change in the electronic conductivity or the work function, Φ , of the organic layer. In case of Φ , the polarity of the response depends on the ability of the diffusing gas molecules to exchange charge density with the polymer matrix either by oxidation ($\Delta\Phi > 0$) or reduction ($\Delta\Phi < 0$). The gas molecules behave as electron acceptors and donors, respectively. In Fig. 5, the resistance of PPy nanofibers decreases during the exposure to H_2 because hydrogen may act as a reducing gas. Alternatively, a mechanism involving the bridging of conducting polymer chains by hydrogen has been proposed [22].

The redox reaction of PPy is responsible for large changes in its electrical resistance [25]. Most conducting polymers are inherent anion-exchange materials in their conducting form due to the positive charges delocalized over their backbones. It has been shown that PPy is a strong anion exchanger with a capacity of $7.1 \times 10^{-4} \text{ mol g}^{-1}$ [26]. Equation (1) describes the redox reaction of PPy, which causes the conductivity changes in PPy



In this work, we also studied the effect of elevating the operating temperature on the dynamic response of PPy nanofibers towards different concentrations of the hydrogen gas/synthetic air mixture. Under synthetic air only, the changes in film resistance as a function of increasing temperature from 22 °C–100 °C are shown in Fig. 6. It can be seen that the film resistance decreases, i.e., the conductivity increases, consistent with an increase in the number of thermally activated carriers. Fig. 7 presents the steady-state sensor response under hydrogen exposure.

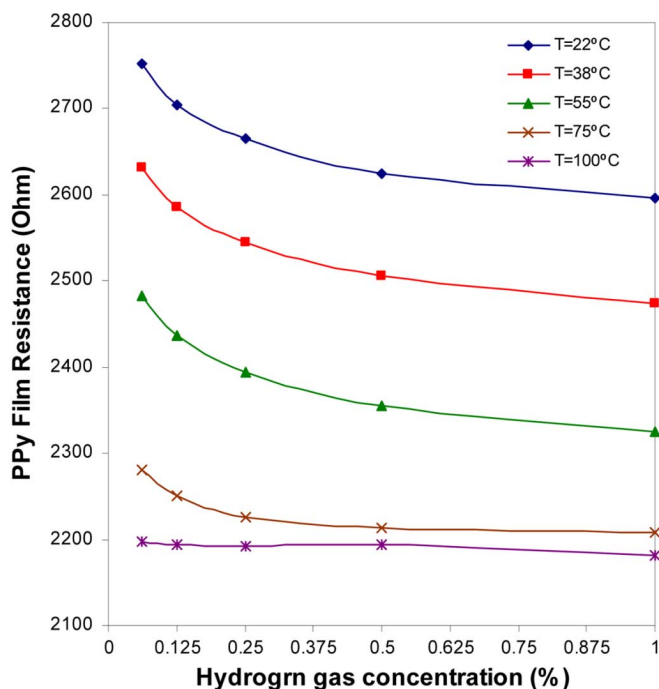


Fig. 7. Variation of the steady-state response of PPy nanofiber sensor as a function of hydrogen gas concentrations.

The nonlinearity of the curves in the temperature range between 22 °C–55 °C can be described by a Langmuir type isotherm as established in the literature [27]. At higher operating temperatures, the response is almost flat, suggesting that the material's response saturates presumably due to the creation of a large number of carriers with high energy moving freely in the polymer chain, thus preventing the polymer from exhibiting its inherent redox reaction behavior. From Fig. 7, it is evident that the conductivity of PPy nanofibers is increasing with temperature elevation and the proposed idea for this observation is that the conductivity increase is due to improvement in the nanostructured film quality because of the temperature increase. At elevated temperatures, the nanostructured film is expected to be smoother comprising dense nanofibrous morphology that has less porosity compared with the film quality at lower temperatures, resulting in an increase of the film conductivity due to the increase of charge carriers hopping between localized states along the polymer chains in accordance with Mott's Variable Range Hopping (VRH) conduction model [28]. The high conductivity of the nanostructured film degrades the sensor response significantly at high temperatures, as shown in Fig. 7, due to the lack of deep penetration of the gas molecules in the bulk of the sensitive layer.

It is worthwhile to mention that humidity has been suppressed in this study through purging the gas chamber with dry synthetic air for 1.5 h before starting the tests. The synthetic air used constitutes of nitrogen 78%, oxygen 21%, and argon 1% according to manufacturer data sheet. Diluting hydrogen in synthetic air may produce some H_2O molecules but high operating temperature maintains H_2O in the vapor state and purging the cell with dry synthetic air for 7 min after gas exposure eliminates the effect of humidity. Our decision to eliminate the effect of humidity

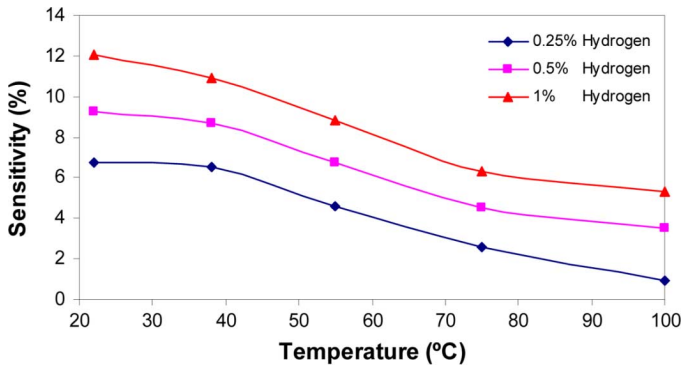


Fig. 8. The conductometric sensor sensitivity as a function of temperature increase for three concentrations of hydrogen gas in synthetic air mixture.

in our study is based on a previous report on the effect of humidity on polyaniline nanofibers that concluded that humidity can impair the sensor response [22].

The conductometric sensor sensitivity (S) is defined by (2)

$$S = \frac{R_a - R_g}{R_g} \times 100\% \quad (2)$$

where R_a is the initial resistance of the sensor under synthetic air and R_g is the sensor resistance during the exposure to hydrogen gas. In an earlier study, it was shown that variation of the nanofiber film thickness in the range of 0.2–2.0 μm did not produce any significant change in sensor sensitivities [11]. This observation was interpreted in terms of the high porosity of the nanofiber films and, hence, high surface to volume ratio which also explains the enhanced sensitivity of these films compared with conventional bulk thin films.

We found that sensitivity of the sensor degrades with increasing temperature. Fig. 8 displays the relation between variable temperature and sensor sensitivity. It can be seen that the best operating temperature for the polypyrrole nanofibers in gas sensing is at room temperature.

V. CONCLUSION

Polypyrrole nanofibers have been successfully synthesized using a template-free route. The fibrillar morphology was confirmed using both SEM and TEM. The average diameter of the polypyrrole nanofibers is ~ 18 nm with an average length of several micrometers. Elemental analysis revealed a C/N ratio of 3.5, which is consistent with the theoretical value for PPy. FT-IR confirms that the structure of the nanofibers is comparable to bulk polypyrrole. A conductometric transducer was fabricated on a sapphire substrate and PPy nanofibers were deposited on its electrode. The sensor performance was analyzed after exposure to different concentrations of hydrogen gas in a synthetic air mixture. A resistance shift of 312 Ω in the presence of a 1% hydrogen pulse was obtained with a 43 s response time. The sensor sensitivity is largely dependent on the operating temperature, as well as the concentration of the target gas. Optimum sensor performance was found to be at room temperature. However, the sensor maintained its systematic response towards different concentrations of hydrogen gas at elevated temperatures up to 100 $^{\circ}\text{C}$ despite a slow degradation in sensitivity.

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