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8	
9	Abstract

10 Titanium(IV) oxide (TiO₂) nanocrystal with brookite phase was prepared by hydrothermal treatment of titanate nanotubes under strong acidic conditions (perchloric acid, hydrochloric acid 11 12and nitric acid). The relative fractions of crystal phases (anatase, rutile and brookite) were strongly dependent on hydrothermal treatment duration and the kind of strong acid used. A high 13brookite type TiO₂ content was obtained by hydrothermal treatment under perchloric acidic solution. 14Single-phase brookite TiO₂ particles were extracted from as-prepared TiO₂ by a centrifugal 15separation procedure, and they exhibited higher photocatalytic activity for acetaldehyde 1617decomposition than TiO₂ prepared without the separation procedure.

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1 1. Introduction

 $\mathbf{2}$ Titanate nanotubes (TNTs) have attracted much attention because of their unique structure and simple preparation method, which was reported by Kasuga et al. [1]. Applications of TNT 3 powders include dye-sensitized solar cells [2], catalyst supports [3,4] and photocatalysts [5]; these $\mathbf{4}$ applications exploit the chemical and physical properties of TNT powders, such as their large $\mathbf{5}$ specific surface area and tubular structure. Another very important application of TNTs is as a 6 titanium source for material preparation. Several groups have investigated the structural 78 transformation of TNTs to titanium(IV) oxide (TiO₂) by thermal, acid and hydrothermal treatment, and they found that treatment of TNTs under specific conditions results in the formation of 9 characteristic TiO₂ nanostructures, e.g., nanorods, nanobelts, nanofibers [6-10]. 10 The main component of their crystal phase is anatase and/or rutile, while a trace amount of brookite is 11 12sometimes present [9]. Anatase, rutile and brookite are well known to be the crystal phases of natural TiO₂. However, the physical and chemical properties of brookite still remain unclear, in 13contrast with those of anatase and rutile, which have been extensively investigated. This is due to 14the low availability of single-phase brookite TiO₂ as a result of it being thermodynamically 15metastable. 16

Kominami et al. prepared single-phase brookite TiO_2 powders for the first time by solvothermal synthesis in which oxobis(2,4-pentanedionato-*O*,*O*')titanium was used as the TiO_2 precursor and sodium laurate was used as a reagent for controlling the crystal structure [11]. They subsequently investigated the photocatalytic activity of single-phase brookite TiO_2 [12,13]. Recently, other groups have also reported single-phase brookite TiO_2 having several kinds of characteristic structures, e.g., nanorods and nanotubes [14-18]. However, the photocatalytic properties of these structures were not investigated in these studies.

In the present study, TiO_2 powders were prepared by hydrothermal treatment of TNTs under several kinds of strong acidic conditions without using an organic reagent for controlling the crystal structure, such as sodium laurate. Single-phase brookite TiO_2 nanocrystals were extracted from as-prepared TiO_2 , and the photocatalytic properties of these TiO_2 powders were investigated.

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2 2. Experimental

3 2.1 Sample preparation

One gram of TNT powder (TNT-352, Catalysts & Chemicals Ind. Co., Ltd.) was dispersed in 4 50 cm³ of 2 mol dm⁻³ of an acid solution (perchloric acid (HClO₄), hydrochloric acid (HCl), and $\mathbf{5}$ nitric acid (HNO₃)) in a Teflon bottle. After stirring the solution, the Teflon bottle was sealed with 6 7a stainless steel jacket and heated at 200 °C for 1~50 h in an oven. After this hydrothermal 8 treatment, the supernatant in the bottle was removed. To extract single-phase brookite TiO₂ powders, centrifugal separation was performed by adding Milli-Q water, centrifuging the 9 suspension for 30 min at 5000 rpm, and then filtrating the supernatant (not transparent). The 10 residues obtained both by using and without using the above-mentioned procedure were washed 11 12with an aqueous NH₃ solution (1%) and Milli-Q water several times until the ionic conductivity of the supernatant was $<10 \ \mu S \ cm^{-2}$. 13

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15 2.2 Characterization

16 The crystal structures of the TNT or TiO₂ powders were characterized with an X-ray 17 diffractometer (Rigaku, MiniFlex II) with Cu K α radiation ($\lambda = 1.5405$ Å). The relative surface 18 areas of the powders (S_{BET}) were determined using a surface area analyzer (Quantachrome, 19 Autosorb-1) in conjunction with the Brunauer-Emmett-Teller (BET) equation. The morphologies 20 of the TNT or TiO₂ powders were observed using a transmission electron microscope (TEM; 21 Hitachi, H-9000NAR).

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23 2.3 Photocatalytic decomposition of acetaldehyde

Photocatalytic activities of TNT or TiO_2 samples were evaluated by photocatalytic decomposition of acetaldehyde. One hundred milligrams of TNT or TiO_2 powder, which have a complete extinction of the incident radiation, was spread on a glass dish, and the glass dish was placed in a 125 cm³ Tedlar bag (As one). Five hundred parts per million of gaseous acetaldehyde

was injected into the Tedlar bag, and photoirradiation was performed at room temperature after the 1 The gaseous composition in Tedlar bag were $\mathbf{2}$ acetaldehyde had reached adsorption equilibrium. 79 % of N₂, 21 % of O₂, < 0.1 ppm of CO₂ and 500 ppm of acetaldehyde, and relative humidity was 3 ca. 30 %. A light emitting diode (Nichia, NCCU033), which emitted light at a wavelength of ca. 4 365 nm and an intensity of 1.0 mW cm^{-2} , was used as the light source. The concentrations of $\mathbf{5}$ acetaldehyde and carbon dioxide (CO₂) were estimated by gas chromatography (Shimadzu, GC-8A, 6 7FID detector) with a PEG-20 M 20% Celite 545 packed glass column and by gas chromatography 8 (Shimadzu, GC-9A, FID detector) with a TCP 20% Uniport R packed column and a methanizer (GL Sciences, MT-221), respectively. In the photocatalytic evaluation, P-25 (Japan Aerosil Co., $S_{BET} =$ 9 50 m² g⁻¹), which is a well-known commercial TiO₂ photocatalyst with a high photocatalytic 10 activity, was employed as a standard photocatalyst. 11

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13 **3. Results and discussion**

The XRD analysis revealed that the crystal phase of the samples was transformed from TNT 14into TiO₂ (anatase, brookite and rutile) after over 1 h of hydrothermal treatment of TNT. Figures 151a and b show XRD patterns of samples before hydrothermal treatment and after 50 h of 1617hydrothermal treatment under HClO₄ solution, respectively. They indicate that a large fraction of brookite and small fractions of anatase and rutile were formed. The relative fractions of these 18crystal phases exhibited a strong dependence on the duration of the hydrothermal treatment. 19Figure 2 shows the relative fractions of the crystal phases (f_A : anatase; f_B : brookite; f_R : rutile) as a 20function of the duration of the hydrothermal treatment with HClO₄ as the acid. The relative 21fractions of the crystal phases were calculated using the following equation [19] 22

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$$f_i = k_i S_i / (k_A S_A + k_B S_B + k_R S_R) \qquad i = A,$$

where $k_A = 0.886$, $k_R = 1$, and $k_B = 2.721$ [19]. S_A , S_B and S_R are the integrated intensities of peaks attributed to anatase (101), brookite (121) and rutile (110), respectively. They were estimated by fitting the XRD patterns in the range of 22~34° with a superposition of five Lorentz functions for peaks attributed to brookite (120), anatase (101), brookite (111), rutile (110) and brookite (121).

B. R.

1 For the calculation, the ratio of integrated intensities of the three peaks attributed to brookite (120),

2 (111), (121) was fixed at 10 : 8 : 9, as determined by JCPDS data.

In Fig. 2, $f_{\rm B}$ and $f_{\rm R}$ increase monotonically with an increase in the hydrothermal treatment 3 duration, and f_A has a maximum at about 5 h. This indicates that brookite is relatively stable under 4 HClO₄ conditions, while anatase is metastable and is transformed into the other phases. However, $\mathbf{5}$ phase transformation of fine anatase particles (ST-01, Ishihara Sangyo Co., $S_{\text{BET}} = 316 \text{ m}^2 \text{ g}^{-1}$) to 6 7brookite was not observed after 5 h of hydrothermal treatment under the same conditions. This 8 result conflicts with the above-mentioned possibility of transforming anatase into brookite. 9 Another plausible reason might be that f_A is underestimated because it is difficult to accurately estimate S_A in the presence of the brookite phase due to the overlapping of the three peaks that 10occur in the range of $25 \sim 26^{\circ}$. The reason for this behavior of the anatase phase is currently under 11 investigation. 12

 TiO_2 prepared by 5 h of hydrothermal treatment of TNTs in Milli-Q water or 2 mol dm⁻³ 13aqueous solution of sodium perchlorate had a smaller $f_{\rm B}$ than that prepared in HClO₄ solution. 14Some reports have also reported the formation of a large fraction of brookite phase by hydrothermal 15synthesis from titanium(IV) tetrachloride (TiCl₄) under HCl and HNO₃ conditions [14,20]. 1617However, 50 h of hydrothermal treatment under HCl and HNO₃ conditions induced small f_B ($f_R = 1$ for HCl, $f_{\rm R} = 0.53$, $f_{\rm B} = 0.47$ for HNO₃) in the present study, though relatively large $f_{\rm B}$ was observed 18for samples prepared by 5 h of hydrothermal treatment ($f_{\rm R} = 0.72, f_{\rm B} = 0.28$ for HCl, $f_{\rm A} = 0.05, f_{\rm R} =$ 190.12, $f_{\rm B} = 0.83$ for HNO₃). These results indicate that a low pH induce the transformation of TNTs 20to brookite and the presence of ClO_4^- anions suppress the transformation into rutile, while the 2122presence of Cl⁻ anions promote the transformation into rutile due to the stabilization of rutile surfaces [21]. 23

Figures 1b and c show XRD patterns of TiO_2 powders that were prepared by 50 h of hydrothermal treatment without and with centrifugal separation, respectively. The TiO_2 produced without centrifugal separation has an appreciable amount of rutile, whereas the XRD pattern of the TiO₂ produced using centrifugal separation contains no peaks that are attributed to rutile. This

indicates that single-phase brookite TiO₂ was extracted by the separation procedure of the 1 as-prepared TiO₂. Moreover, heat treatment (300 °C, 3 h) of single-phase brookite TiO₂ induced $\mathbf{2}$ no formation of the anatase phase. This result presumably indicates that the single-phase brookite 3 TiO₂ contains very little amorphous or TNT phases. The primary particle sizes of rutile ($d_{R(110)}$) 4 and brookite $(d_{B(121)})$ were estimated from peaks attributed to rutile (110) and brookite (121) in $\mathbf{5}$ 6 XRD patterns using the Scherrer equation $(d = 0.9\lambda/\beta \cos\theta_{\rm B})$, where λ is the wavelength of X-rays, β 7is the full width at half maximum and $\theta_{\rm B}$ is the Bragg angle) and found to be 35 nm and 23 nm, 8 respectively. However, this difference in particle size is not sufficiently large to explain the reason 9 for the extraction of single-phase brookite TiO₂.

10 Figure 3 shows TEM images of TNT and TiO₂ prepared with and without centrifugal The TNTs had tubular structures with inner and outer diameters of ca. 5 nm and ca. 10 11 separation. 12nm, respectively (Fig. 3a). TiO₂ prepared without centrifugal separation consisted of two kinds of particle shapes (Fig. 3b), namely large rods (ca. 50 nm by ca. 300 nm) and small particles with an 13aspect ratio of ca. 1 (ca. 30 nm), while only small monodispersed particles with a narrow size 14distribution were observed in the TiO₂ prepared with centrifugal separation (Fig. 3c). 15TEM observation and XRD analysis indicate that larger rods and smaller particles are attributable to rutile 1617and brookite nanocrystals, respectively. Thus, the effective particle size of the rutile rods was larger than $d_{R(110)}$ due to the rod structure. This large difference in effective particle sizes explains 18why single-phase brookite TiO₂ was extracted by the centrifugal separation. The S_{BET} (61 m² g⁻¹) 19of TiO₂ prepared with the centrifugal separation procedure was slightly larger than that $(57 \text{ m}^2 \text{ g}^{-1})$ 20of the TiO₂ prepared without centrifugal separation because of the low fraction of the rutile phase 2122with large particle sizes.

The TEM image in Fig. 3c suggests that some brookite particles have specific exposed crystal faces, and lattice fringes were also observed in the TEM images of most of the single-phase brookite TiO_2 particles shown in Fig. 3d. The space of lattice fringes (ca. 0.375 nm) indicate that exposed crystal face was attributed to brookite (101) face. This fringe means that the brookite TiO₂ particles consisted of a single crystal (i.e., they were not polycrystalline) suggesting that the 1 brookite TiO₂ particles have high crystallinity, which is important for photocatalytic activity.

 $\mathbf{2}$ Figure 4 shows the time course of CO₂ evolution of acetaldehyde decomposition over TiO₂ samples as a function of the photoirradiation time. The prepared brookite TiO₂ had a higher 3 photocatalytic activity than the other TiO_2 powders. This result is presumably due to the high 4crystallinity and relatively large surface area of the prepared brookite TiO_2 [22]. Another plausible $\mathbf{5}$ reason for this may be the specific exposed crystal faces, which induce separation of redox sites that 6 enhances photocatalytic activity [23]. Some reports have revealed that a synergistic effect occurs 78 for some photocatalytic reactions performed over a mixture of small anatase and large rutile particles [24]. However, TiO₂ prepared with centrifugal separation (small brookite particles) 9 exhibited a higher activity than TiO₂ prepared without centrifugal separation (mixture of small 10 brookite particles and large rutile particles), thus a synergistic effect was not observed in the present 11 study for powders consisting of a mixture of brookite and rutile TiO_2 . On the contrary, the low 12activity of rutile rods with specific exposed crystal faces indicates that recombination rate was 13presumably increased by brookite-rutile contacts. The high photocatalytic activity of the 14single-phase brookite TiO₂ is attributable to intrinsic photocatalytic properties of brookite TiO₂ that 15has a high crystallinity and a large surface area. 16

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18 **4. Conclusion**

19 Hydrothermal treatment of TNT under $HClO_4$ conditions induced formation of brookite phase 20 with trace amounts of the other two phases. The results of hydrothermal treatment of TNT under 21 various conditions indicate that nucleation of brookite phase requires both a suitable pH of the 22 solution in the dissolution process and the presence of the ClO_4^- ion to control the formation of 23 crystal structures in the crystallization process. Single-phase brookite TiO₂ was extracted from 24 as-prepared TiO₂ by centrifugal separation. It consisted of a single crystal, and exhibited higher 25 photocatalytic activity, presumably due to its high crystallinity and relatively large surface area.

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- 9

10 Figure Captions



11

12 **Figure 1** XRD patterns of (a) TNT and TiO₂ prepared by 50 h of hydrothermal treatment under

13 HClO₄ solution (b) without and (c) with centrifugal separation.



- **Figure 2** Relative fractions of crystal phases (f_A : anatase; f_B : brookite; f_R : rutile) estimated from
- 2 XRD patterns as a function of duration of hydrothermal treatment under HClO₄ solution.



Figure 3 TEM images of (a) TNT and TiO_2 prepared (b) without and (c) with centrifugal 6 separation. (d) High-resolution image of image in (c).



Figure 4 Time course of CO_2 evolution of acetaldehyde decomposition over TiO_2 prepared (a) 10 without and (b) with centrifugal separation, commercial TiO_2 (c) ST-01 and (d) P-25, as a function 11 of photoirradiation time.