Adsorbent filled membranes for gas separation. Part 1. Improvement of the gas separation properties of polymeric membranes by incorporation of microporous adsorbents*

J.-M. Duvala, B. Folkersb, M.H.V. Muldera, G. Desgrandchampsb and C.A. Smoldersa

aUniversity of Twente, Department of Chemical Technology, P.O. Box 217, 7500 AE Enschede (The Netherlands)
bElf Aquitaine, Groupement de Recherches de Lacq, P.O. Box 34 Lacq, 64170 Artix (France)

(Received May 18, 1992, accepted in revised form August 2, 1992)

Abstract

The effect of the introduction of specific adsorbents on the gas separation properties of polymeric membranes has been studied. For this purpose both carbon molecular sieves and zeolites are considered. The results show that zeolites such as silicate-1, 13X and KY improve to a large extent the separation properties of poorly selective rubbery polymers towards a mixture of carbon dioxide/methane. Some of the filled rubbery polymers achieve intrinsic separation properties comparable to cellulose acetate, polysulfone or polyethersulfone. However, zeolite 5A leads to a decrease in permeability and an unchanged selectivity. This is due to the impermeable character of these particles, i.e. carbon dioxide molecules cannot diffuse through the porous structure under the conditions applied. Using silicate-1 also results in an improvement of the oxygen/nitrogen separation properties which is mainly due to a kinetic effect. Carbon molecular sieves do not improve the separation performances or only to a very small extent. This is caused by a mainly dead-end (not interconnected) porous structure which is inherent to their manufacturing process.

Keywords: gas separation; zeolite filled membrane; carbon molecular sieves, composite membranes

Introduction

Adsorbents such as zeolites and molecular sieving carbons are well known for the separation of gaseous mixtures by pressure-swing adsorption. This process involves the use of the specific microporous adsorbents in a semi-continuous way with sorption–desorption cycles. In the present study, the adsorbents have been incorporated into a polymeric matrix to form a heterogeneous membrane. In filled membranes the selective properties of the specific adsorbents are used in a continuous way.

Such a concept has already been shown to
improve the separation properties of polymers for pervaporation [1,2] and gas separation [3]. In the case of pervaporation of ethanol/water mixtures using silicone rubber membranes, both the ethanol selectivity and the permeability were enhanced by the incorporation of silicalite-1 particles into the polymer matrix [1]. The same type of membranes showed improved gas separation properties [2]. However these studies were limited to one type of zeolite or one polymeric phase. The present paper aims at giving a larger overview of the influence of various zeolites and molecular sieve carbons on the gas separation properties of polymer materials. Therefore four zeolites with different hydrophilic/hydrophobic properties and various pore sizes were incorporated into five different polymers.

Basic properties of zeolites and carbon molecular sieves

Zeolites

It is beyond the scope of this paper to give an extended and detailed description of the microporous adsorbents considered (see review articles [4–7]). However some basic characteristics which are of importance for the present study will be summarized briefly. Zeolites are crystalline alumino-silicates composed of AlO₄ and SiO₄ tetrahedra, which build up a network of channels and cavities. The microcrystalline voids and channels which are interconnected are responsible for the very specific properties of these adsorbents. The aperture size is typically in the range of molecular dimensions, i.e. 3 to 10 Å. Because the aluminium atom is trivalent, an excess of negative charge is introduced in the network when Si is replaced by Al in the tetrahedra. This charge is compensated by non-framework cations, located near the negative charges; the most common ones are Na⁺, K⁺, Ca²⁺. The number of cations is thus determined by the number of aluminium atoms in the framework. Because of the presence of cations, these zeolites are polar adsorbents. This means that molecules such as water, ammonia (strong dipoles), carbon dioxide, nitrogen (quadrupolar) and aromatic hydrocarbons (π layer interaction) are adsorbed more strongly than non polar species of comparable molecular weight. Zeolites with a high Si/Al ratio are hydrophobic and the adsorption is mainly governed by Van der Waals forces.

Therefore four main factors influence the properties of a zeolite:
- pore size which acts on the ability of a molecule to enter and diffuse through the zeolite framework.
- Si/Al ratio which determines the number of cations and thus the hydrophilicity of the zeolite framework.
- type of cation (valence and size)
- direction of the pores (1, 2 or 3D porous network).

Some applications of zeolites in the separation of gaseous mixtures are the dehydration of industrial gases using hydrophilic zeolites, the separation of air by preferential equilibrium sorption of nitrogen or the removal of H₂S from sour gas [4].

Carbon molecular sieves

Carbon molecular sieves are characterized by a pore size distribution, which is much narrower than in the case of common active carbons [7,8]. It means that, contrary to zeolites, carbon molecular sieves cannot be given a definite pore size but only a mean pore size which is in the range of molecular dimensions. Furthermore, their internal surface is basically hydrophobic, with possible variations due to acidic surface groups [9]. Carbon molecular sieves are industrially used to separate air by adsorption of oxygen and also to remove carbon dioxide from landfill gas [10].
### Experimental

#### Adsorbents

The adsorbents considered are the zeolites 5A (Ceca S.A., France, crystal density \( \rho_c = 1.48 \text{ g/cm}^3 \)), Silicalite-1 (Union Carbide, USA, \( \rho_c = 1.76 \text{ g/cm}^3 \)), KY (Ceca S.A., \( \rho_c = 1.52 \text{ g/cm}^3 \)), 13X (Ceca S.A., \( \rho_c = 1.54 \text{ g/cm}^3 \)) and the carbon molecular sieves Cecalite (Ceca S.A), Carbosieve (Supelco Separation Technologies, USA) and W20 (Norit N.V., The Netherlands). Furthermore, non porous silica (Sigma) was used as inert filler (N, BET surface area: 5.8 m\(^2\)/g). The zeolites were activated in air at 350°C for two hours except for silicalite where an activation temperature of 500°C was applied (calcination of the template used during synthesis). The crystallinity of the zeolites was examined by X-ray powder diffraction. The size of the zeolite particles was typically in the range 1–5 \( \mu \)m.

The carbons were activated in a vacuum oven at 150°C for 24 hr. The adsorbents were then kept in a vacuum oven at room temperature. Table 1 (and Table A-1 in the Appendix) gives some basic properties of these microporous adsorbents.

#### Polymers

Polydimethylsiloxane (PDMS, RTV 615A and B, General Electric), ethylene–propylene rubber (EPDM, Keltan 578, DSM), polychloroprene (PCP, Aldrich), and nitrile butadiene rubbers (Polysar) with two different amounts of acrylonitrile segments: 45%, Krynac 45.50 (NBR 45) and 50%, Krynac 50.75 (NBR 50) were used as polymers.

#### Membrane preparation

Membranes were prepared by dispersion of the adsorbent phase in the polymeric solution, followed by a casting–evaporation process. Hexane and chloroform were used as solvents. The evaporation step was carried out first overnight under a nitrogen stream, then in a vacuum oven at 30°C for at least 24 hr. The membranes thus obtained were 50 to 200 μm thick. The homogenous dispersion of the particles was observed by scanning electron microscopy (Fig. 1).

#### Equipment and methods

The gas separation properties were determined by means of an automated setup with a mixture of carbon dioxide and methane (25/75 vol.%), pure oxygen and pure nitrogen at room temperature (24°C). The low pressure side was kept under vacuum (10⁻²–10⁻¹ mbar) while the feed pressure was set at 5.3 bars. The permeate flux was determined by measuring a pressure increase in a calibrated volume, the permeate pressure remaining negligible com-

<table>
<thead>
<tr>
<th>Pore size (Å)</th>
<th>Si/Al</th>
<th>Cation</th>
<th>Directions</th>
<th>Pore vol (cm³/g)</th>
</tr>
</thead>
<tbody>
<tr>
<td>5A</td>
<td>4 5</td>
<td>Na⁺/Ca²⁺</td>
<td>3D</td>
<td>0.28</td>
</tr>
<tr>
<td>silicalite-1</td>
<td>5–5.6 &lt;--&gt; 5 1–5.5</td>
<td>-</td>
<td>2D</td>
<td>0.19</td>
</tr>
<tr>
<td>KY</td>
<td>7.4</td>
<td>K⁺</td>
<td>3D</td>
<td>-</td>
</tr>
<tr>
<td>13X</td>
<td>7.4</td>
<td>Na⁺</td>
<td>3D</td>
<td>0.36</td>
</tr>
</tbody>
</table>

**Table 1**

Main characteristics of the zeolitic adsorbents. Kinetic diameters of the gas molecules used are: CO₂ 3.3 Å, CH₄ 3.8 Å, O₂ 3.46 Å, N₂ 3.64 Å.
pared to the feed pressure. The permeabilities are expressed in barrer with 1 barrer = $10^{-10}$ cm$^3$ (STP)-cm-cm$^{-2}$-sec$^{-1}$-cmHg$^{-1}$. The selectivity was calculated from the gas chromatographic analyses of the feed mixture and of the permeate in the case of CO$_2$/CH$_4$, and from the ratio of the pure gas permeabilities in the case of O$_2$/N$_2$. For more details about the experimental setup, see Ref. [11]. The permeation experiments were carried out until steady state was reached, which means from 1 to 5 days depending on the zeolite considered. The gas sorption experiments were carried out with a pressure-decay setup, as described elsewhere [12,13]. The membrane samples were evacuated at 30$^\circ$C and $10^{-3}$ mbar for at least 15 hr before measurement, in addition to the normal treatment. The experiments were carried out at 30$^\circ$C.

Results and discussion

The results obtained with carbon molecular sieves are given in the Appendix.

**Gas permeation using zeolite filled membranes**

**Carbon dioxide/methane**

The open porous structure of the zeolites results in an increase both in selectivity and permeability for carbon dioxide over methane in the case of silicalite-1, KY and 13X, as can be seen from Figs. 2 (a) and (b). Here EPDM was used as polymeric phase. The improvement of the separation properties increases when the volume fraction of zeolite in the polymer matrix increases, until the amount of adsorbent in the membrane surpasses an optimum. This results in the formation of non-selective voids as can be seen from the large increase in permeability and decrease in selectivity. It occurs when the volume fraction of zeolite is in the range of 45 to 55 percent (Fig. 2).

The largest improvement in selectivity and permeability for carbon dioxide was obtained using zeolite KY (Fig. 2), for which the pore size is about twice the kinetic diameter of carbon dioxide and methane (which are 3.3 and 3.8 Å respectively). Zeolite 13X, which has basically the same pore size but a lower Si/Al ratio,
From the results obtained it can be concluded that molecular sieving by size exclusion is not the only mechanism involved and that the affinity of the gas molecules for the zeolite internal surface plays also a major role.

Screening the results of a series of polymers shows that the zeolites silicalite-1 and KY are able to increase significantly their separation properties. Especially in the case of a polymer the selectivity of which is about 15 for carbon dioxide/methane the results of the filled membranes are comparable to industrially used polymer membrane materials (cellulose acetate, polysulfone) as can be seen in Figs. 3 (a) and (b) and Fig. 4. In these figures the diagonal lines show the results if the zeolites had no effect on the separation properties of the different polymers. All polymers showed improved selectivity and permeability for carbon dioxide when filled with silicalite-1 or KY, despite the fact that for the polymers studied the permeabilities ranged from about 3 (NBR 50) to 3000 barrers (PDMS) and their selectivities from 2.7 (PDMS) to 17.5 (NBR 50). For example, the addition of 50 vol.% of silicalite-1 to NBR 50 gives a material with a selectivity of 29 and a permeability of 7 barrers. NBR 45 filled with 46 vol.% of zeolite KY results in a material with

| Table 2 |
|-----------------------|-----------------------|
|                       | $P_{CO_2}$ (barrer)   | $\alpha_{CO_2/CH_4}$ |
| PDMS                  | 2920                  | 2.7                   |
| PDMS + 50 wt % 5A     | 1800                  | 2.8                   |
| PDMS + 28 wt % silica | 2407                  | 2.8                   |
| PDMS + 59 wt % silica | 1910                  | 3.0                   |
| EPDM                  | 81                    | 4.3                   |
| EPDM + 52 wt % 5A     | 65                    | 4.4                   |
| EPDM + 62 wt % 5A     | 50                    | 4.5                   |
| EPDM + 40 wt % silica | 72                    | 4.5                   |
| EPDM + 53 wt % silica | 55                    | 4.6                   |

Contrary to the first three zeolites considered, zeolite 5A does not enhance the membrane performance (Table 2), despite the fact that the pore size (4.5 Å) is large enough to fit the kinetic diameters of the gas molecules. The properties are similar to those of non-porous silica with respect to the effect on the gas permeation properties (Table 2). It means that the porous structure of zeolite 5A is not open for the permeation of gas molecules in the particular conditions encountered in the polymer matrix. Possible reasons for this phenomenon will be given later.
Effect of the zeolites silicalite-1 (a) and KY (b) on the CO₂/CH₄ selectivity of various rubbery polymers. Figures 3 and 4 also show that filling with zeolite KY always results in a larger improvement of the CO₂ selectivity and permeability compared to silicalite-1 for a given volume fraction in a given polymer. This is in agreement with the observations for the single polymer EPDM when using different volume fractions. Silicalite-1 and type Y zeolite both show a sorption selectivity towards CO₂ [7,14], but the selectivity obtained with Y (NaY in Ref. [14]) is much larger, due to its hydrophilic surface. It may be assumed that the same is observed when the sodium cations are exchanged for potassium, which would explain the beneficial influence of KY on the permeation properties.

This concept may also be applied to improve the separation performance of very selective (glassy) polymers. This was also investigated but the results obtained were not that positive due to a very poor adhesion of the polymer matrix to the external surface of the zeolite, resulting in large voids between the polymer and the zeolite crystals. The incorporation of zeolites in glassy polymers is described elsewhere [15] and will not be considered here.

**Oxygen/nitrogen**

A very surprising effect was observed in the oxygen/nitrogen separation. Incorporation of silicalite-1 resulted in an increase in the ideal selectivity for oxygen in the case of poorly selective rubbery polymers (Table 3). This was already observed for PDMS and our results are similar to those found in literature [2]. The selectivity values obtained with EPDM as polymeric phase are comparable to the intrinsic values of polymethylpentene or polydimethyl-phenylene oxide which are industrially used.
TABLE 3

Effect of silicalite-1 on O\textsubscript{2}/N\textsubscript{2} separation properties of PDMS and EPDM

<table>
<thead>
<tr>
<th>Membrane Material</th>
<th>(P_{O_2}) (barrer)</th>
<th>(P_{N_2}) (barrer)</th>
<th>(\alpha_{O_2/N_2})</th>
</tr>
</thead>
<tbody>
<tr>
<td>PDMS</td>
<td>606</td>
<td>289</td>
<td>2.1</td>
</tr>
<tr>
<td>PDMS + 28 vol % silicalite-1</td>
<td>892</td>
<td>355</td>
<td>2.5</td>
</tr>
<tr>
<td>PDMS + 37 vol % silicalite-1</td>
<td>998</td>
<td>373</td>
<td>2.7</td>
</tr>
<tr>
<td>PDMS + 47 vol % silicalite-1</td>
<td>1370</td>
<td>521</td>
<td>2.6</td>
</tr>
<tr>
<td>EPDM</td>
<td>16</td>
<td>5.3</td>
<td>3</td>
</tr>
<tr>
<td>EPDM + 32 vol % silicalite-1</td>
<td>29</td>
<td>7.5</td>
<td>3.9</td>
</tr>
<tr>
<td>EPDM + 53 vol % silicalite-1</td>
<td>40</td>
<td>8.5</td>
<td>4.7</td>
</tr>
</tbody>
</table>

Silicalite shows a sorption selectivity for carbon dioxide over methane (separation factor: 2.7 at 25°C \([8,14]\)) but no significant equilibrium sorption selectivity for oxygen over nitrogen is reported for this zeolite \([16]\). The improvement in selectivity obtained for the filled membranes is mainly determined by a kinetic effect, i.e. the oxygen molecules can diffuse through the porous structure of silicalite-1 faster than the nitrogen molecules. This is comparable to what is observed with carbon molecular sieves which have almost no equilibrium sorption selectivity for oxygen but are used in pressure swing adsorption. Here the separation of air is based on a faster sorption rate of oxygen over nitrogen \([17]\). Such a kinetic selectivity towards oxygen is also observed with zeolite 4A (pore size: 3.8 Å), which is even sorption selective towards nitrogen at equilibrium \([18]\). It should be noticed that silicalite-1 also enhances the nitrogen permeability with the polymers investigated, but this was not observed for methane.

**Gas sorption isotherms**

Carbon dioxide sorption isotherms show that the presence of zeolite particles in the membrane matrix enhances the carbon dioxide sorption capacity drastically (see Fig. 5). However, when considering transport properties the sorbent should not only sorb the gas molecules but also allow these to diffuse under the driving force applied. This is indeed what is observed for silicalite-1, KY and 13X. On the other hand zeolite 5A also enhances the sorption capacity of EPDM, but not the gas permeation properties (Table 2). This means that the sorbed gas molecules cannot diffuse through the zeolite porous structure but are trapped into the matrix and exhibit a very low mobility. There may be two reasons for this behaviour. The first one is that under permeation conditions, the gas
molecules are very strongly bound to the sorption sites and cannot desorb (the heat of sorption of carbon dioxide in type A zeolite is greater than in type Y or in silicalite-1 [14]); therefore they block the way for other molecules. The second reason is that water of hydration remains in the zeolite structure and cannot be removed by vacuum activation at room temperature (a high temperature would degrade the polymeric phase). The latter explanation is more likely to be correct because 5A is a very hydrophilic zeolite with a high affinity for water resulting in a lower CO₂ sorption capacity than expected from pure zeolite sorption data [19]. Moreover, Paul and Kemp [19] showed that activation of a 5A filled silicone rubber membrane at 200°C prior to use does not make zeolite 5A beneficial for gas permeation, i.e. the permeability of the membrane is lowered and the diffusion time-lag drastically enhanced. Therefore even in-situ activation does not make much difference with respect to the permeation results.

In addition, it can be seen from the sorption isotherms that the zeolite resulting in the largest improvement of carbon dioxide sorption capacity (KY) also leads to the largest improvement of the carbon dioxide permeability and selectivity. Zeolite 13X gives a surprisingly low increase of the sorption capacity when considering its microporous volume and low Si/Al ratio. These two factors should favor the sorption of carbon dioxide to a much larger extent than in the case of the hydrophobic silicalite-1 (Si/Al = ∞). Here again, the presence of hydration water might play a decisive role. After the thermal activation and storage in the vacuum oven, it may be assumed that essentially all the water has been removed from the crystal structure. However, the preparation of the membrane and the gas separation experiments themselves involve water traces which will adsorb into the zeolite structure more preferentially than carbon dioxide. This effect can decrease the sorption capacity to a substantial extent (like in the case of 5A) which then reduces the ability to improve the gas permeation properties. Nevertheless some improvement was observed with respect to CO₂/CH₄ permeation which can be attributed to the large pores present in 13X which are able to accommodate both carbon dioxide and water traces without blocking the diffusion path completely.

**Conclusions**

The carbon dioxide/methane separation performances of rubbery polymers are significantly enhanced when zeolites are incorporated which are carbon dioxide sorption selective, provided that the gas molecules can diffuse through the particles. This is the case when zeolites like silicalite-1, KY and 13X are considered, leading to separation properties comparable to commonly used glassy polymer membrane materials. The effect is observed even when non-selective or poorly permeable polymers are considered. However, zeolite 5A which seems suitable because of its pore size and hydrophilic character does not show the same positive enhancement.

Furthermore, silicalite-1 was found to be beneficial to the oxygen/nitrogen separation properties of silicone rubber (PDMS) and ethylene-propylene rubber (EPDM) despite its non-selective equilibrium sorption properties.

Finally, carbon molecular sieves do not lead to any significant improvement of the gas separation properties of rubbery polymers (Appendix) due to a dead-end porous structure.

**Acknowledgment**

This work was financially supported by Elf Aquitaine (France).
References


Appendix

Investigation of carbon molecular sieves as filling material

As indicated above, carbon molecular sieves were also investigated with respect to their effect on the gas separation properties of rubbery polymers. Table A-1 gives some properties of the carbon molecular sieves used. It can be seen that a significant part of the porous volume consists of micropores (diameter < 20 Å).

Table A-2 gives some gas permeation results obtained with filled membranes using carbon
TABLE A-1

Main characteristics of the carbon molecular sieves (data from manufacturers)

<table>
<thead>
<tr>
<th></th>
<th>Mean micropore size (Å)</th>
<th>Micropore vol (cm³/g)</th>
<th>Total pore vol (cm³/g)</th>
</tr>
</thead>
<tbody>
<tr>
<td>Cecalite</td>
<td>6-7</td>
<td>0.18</td>
<td>0.52</td>
</tr>
<tr>
<td>W20</td>
<td>-</td>
<td>0.22</td>
<td>0.55</td>
</tr>
<tr>
<td>Carbosieve</td>
<td>5-7</td>
<td>-</td>
<td>-</td>
</tr>
</tbody>
</table>

TABLE A-2

Gas separation properties of membranes filled with carbon molecular sieves

<table>
<thead>
<tr>
<th></th>
<th>PCO₂ (barrer)</th>
<th>αCO₂/CH₄</th>
<th>PO₂ (barrer)</th>
<th>αO₂/N₂</th>
</tr>
</thead>
<tbody>
<tr>
<td>PDMS</td>
<td>2920</td>
<td>2.7</td>
<td>606</td>
<td>2.1</td>
</tr>
<tr>
<td>PDMS+40 wt % W20</td>
<td>3170</td>
<td>3.35</td>
<td>-</td>
<td>-</td>
</tr>
<tr>
<td>EPDM</td>
<td>81</td>
<td>4.3</td>
<td>16</td>
<td>3</td>
</tr>
<tr>
<td>EPDM+30 wt % W20</td>
<td>93</td>
<td>4.8</td>
<td>-</td>
<td>-</td>
</tr>
<tr>
<td>EPDM+45 wt % Cecalite</td>
<td>55</td>
<td>5.2</td>
<td>13</td>
<td>4.3</td>
</tr>
<tr>
<td>EPDM+50 wt % Carbosieve</td>
<td>120</td>
<td>8</td>
<td>23</td>
<td>4.3</td>
</tr>
</tbody>
</table>

molecular sieves. It can be seen that in the case of CO₂/CH₄ separation no improvement is observed with respect to the properties of PDMS and EPDM, except for Carbosieve. This lack of transport improvement is mainly due to a dead-end porous structure, i.e. pores going from the outside to the inside of the particles without being interconnected. Therefore the gas molecules cannot diffuse through the carbon particles, or to a very small extent. To understand better the origin of such a porous structure, one must think of the way carbon molecular sieves are manufactured [7,8]. Though the manufacturing processes are carefully kept secret, it is conjectured that the preparation is based on hard-coal gasification in air followed by a treatment to control the pore structure. This is achieved either by cracking hydrocarbons within the micropore system, resulting in a partial blocking of the pore mouths [21,22], or by mild steam activation which results in an enlargement of the pores.

Both ways result in a macroporous structure from which micropores are developed without interconnection. The selective pores are thus mostly dead-end pores which are effective for use in a pressure swing adsorption process, but not when included into a membrane matrix for a continuous application. Therefore, the applicability of carbon molecular sieves as filling material in a polymer matrix for separation purposes requires an open structure which seems strongly dependent on the method of preparation.

It can be concluded that carbon molecular sieves cannot compete with zeolites as filling material in polymeric membranes for gas separation. The use of a genuinely molecular sieve carbon membrane seems more promising, as described in Refs. [22,23].