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1 **Infrared and infrared emission spectroscopy of gallium oxide  $\alpha$ -GaO(OH)**  
2 **nanostructures**

3  
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9  
10 **Abstract:**

11  
12 Infrared spectroscopy has been used to study nano to micro sized gallium oxyhydroxide  $\alpha$ -  
13 GaO(OH), prepared using a low temperature hydrothermal route. Rod-like  $\alpha$ -GaO(OH)  
14 crystals with average length of  $\sim 2.5$   $\mu\text{m}$  and width of  $1.5$   $\mu\text{m}$  were prepared when the initial  
15 molar ratio of Ga to OH was 1:3.  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> nano and micro-rods were prepared through the  
16 calcination of  $\alpha$ -GaO(OH) The initial morphology of  $\alpha$ -GaO(OH) is retained in the  $\beta$ -Ga<sub>2</sub>O<sub>3</sub>  
17 nanorods.

18 The combination of infrared and infrared emission spectroscopy complimented with dynamic  
19 thermal analysis were used to characterise the  $\alpha$ -GaO(OH) nanotubes and the formation of  $\beta$ -  
20 Ga<sub>2</sub>O<sub>3</sub> nanorods. Bands at around 2903 and 2836  $\text{cm}^{-1}$  are assigned to the -OH stretching  
21 vibration of  $\alpha$ -GaO(OH) nanorods. Infrared bands at around 952 and 1026  $\text{cm}^{-1}$  are assigned  
22 to the Ga-OH deformation modes of  $\alpha$ -GaO(OH). A significant number of bands are  
23 observed in the 620 to 725  $\text{cm}^{-1}$  region and are assigned to GaO stretching vibrations.

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26 **Keywords:** infrared, infrared emission, nanofibres, nanorods, nanomaterials, gallium  
27 oxyhydroxide, gallium oxide

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## 30 Introduction

31

32 Synthesis of inorganic nanostructures has been of great interest in material science and  
33 nanotechnology in recent times because of wide applications in catalysis, adsorption,  
34 ceramics, optical and fluorescence materials, electrochemistry, and so on [1-3]. Gallium  
35 oxide ( $\text{Ga}_2\text{O}_3$ ), known as gallia, is a ceramic material with high melting point of around 1900  
36 °C [4]. Similar to alumina, gallium oxide can crystallize forming the polymorphs  $\alpha$ ,  $\beta$ ,  $\gamma$ ,  $\delta$   
37 and  $\epsilon$ -gallia.  $\text{Ga}_2\text{O}_3$  is normally an insulator at room temperature with a forbidden energy gap  
38 of around 4.9 eV and as a semi-conductor above 800 °C [5, 6]. Calcination of  $\text{Ga}_2\text{O}_3$  in a  
39 reducing atmosphere turns it into an n-type semiconductor as a result of oxygen vacancies  
40 [7].  $\text{Ga}_2\text{O}_3$  has been widely used for the preparation of phosphors [8], gas sensor [9, 10] and  
41 catalysts [8, 11-13]. Recently, it was also used in the synthesis of solid electrolytes of  
42 superior ionic conductivity [14]. Because of the large surface area/volume ratio which is very  
43 important for both sensor and catalyst applications, one-dimensional (1D) gallium oxide  
44 nanostructures such as nanowires, nanoribbons, nanotubes has been attracted much attention  
45 [15-18]. Furthermore, the interesting properties of 1D gallium oxide nanostructure will have  
46 great potential for fabricating the next generation of optoelectronic and sensing devices.

47

48 Various synthesis methods have been explored to prepare one dimensional gallium oxide  
49 nanostructures including the thermal evaporation [19, 20], thermal annealing [21, 22],  
50 chemical vapour deposition [23, 24], arc-discharge [25], carbothermal reduction laser  
51 ablation[26], oxidation of milling GaN [27], microwave plasma [28], catalyst-assisted  
52 methods [29-31]. However, most of these approaches need to be performed at high operation  
53 temperature which is around 1300 °C or even above 1600 °C. These methods also involve  
54 complex procedures, introducing expensive element as catalyst which may lead to troubles in  
55 purification and influence properties of the target materials. Most recently, synthesis 1D  
56 nanostructures via soft-chemistry routes at low temperature is attracting great interest due to  
57 the obvious advantages, such as, economic, energy efficiency and environmental friendly.  
58 Similar to aluminium, gallium oxide hydroxide ( $\text{GaOOH}$ ) is an important precursor for  
59 synthesis of gallium oxide  $\text{Ga}_2\text{O}_3$ .

60

61 So far limited effort has been devoted to synthesis 1D  $\text{GaOOH}$  or  $\text{Ga}_2\text{O}_3$  nanostructures. Sato  
62 and Nakamura [32] studied precipitation of  $\text{GaOOH}$  in solution prepared by mixing gallium

63 chloride and various alkalis including NaOH, KOH, NH<sub>4</sub>OH, NaHCO<sub>3</sub> and Na<sub>2</sub>CO<sub>3</sub>. Hamada  
64 et al. [33] reported the formation of monodispersed GaOOH particles with diameter of ~100  
65 nm at the presence of sulphate by hydrolysis at elevated temperatures. Avivi et al. [34]  
66 prepared scroll-like cylindrical layered GaOOH crystals with small amount of metallic Ga  
67 enclosed inside via sonochemical reaction. Tas et al. [8] synthesized quadrilateral prisms and  
68 spindle-like GaOOH single crystals by forced hydrolysis of Ga<sup>3+</sup> ions in pure water or in the  
69 presence of decomposing urea. The spindles lose their morphology whereas the quadrilateral  
70 prisms maintain their morphology after calcination. Cheng and Samulski [35] prepared  $\gamma$  and  
71  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> nanotubes by immersing alumina membrane in amorphous Ga<sub>2</sub>O<sub>3</sub>·H<sub>2</sub>O sol, followed  
72 by drying and heating at 500 °C. Patra et al. [36] fabricated submicrometer-sized GaOOH  
73 rods by refluxing an aqueous solution of Ga(NO<sub>3</sub>)<sub>3</sub> and NH<sub>4</sub>OH in a microwave oven. Zhang  
74 et al. [37] synthesized GaOOH nanorods through a facile large-scale hydrothermal process  
75 from GaCl<sub>3</sub>-H<sub>2</sub>O-NaOH solutions under selected pH conditions. Ristić et al. [38] studied  
76 application of sol-gel method in the synthesis of gallium oxyhydroxide and gallium oxide by  
77 hydrolysis of gallium isopropoxide and GaCl<sub>3</sub> in the presence of tetramethylammonium  
78 hydroxide in aqueous solution. Most recently, Liu et al. [39] prepared Ga<sub>2</sub>O<sub>3</sub> nanorods by the  
79 conversion of GaOOH nanorods synthesized via a hydrothermal route using Ga<sub>2</sub>O<sub>3</sub> and NaN<sub>3</sub>  
80 as precursors. Zhang et al. [40] reported a green hydrothermal method at 200 °C for the  
81 synthesis of GaOOH nanorods using Ga<sub>2</sub>O<sub>3</sub> and water as starting materials without the  
82 presence of surfactant. It is noted that gallium oxide or gallium oxide hydroxide  
83 nanostructures synthesised via non-hydrothermal routes are usually thread-like or belt-like  
84 which are much longer in length and much smaller in width than that synthesized via  
85 hydrothermal routes. To the best of our knowledge, the morphology of Ga<sub>2</sub>O<sub>3</sub> or GaOOH  
86 reported via hydrothermal approach are generally rod-like or spindle-like crystal with width  
87 of ~100 nm which resulted in a small surface area/volume ratio.

88

89 The aim of this research is to investigate the influence of experimental procedures on the  
90 synthesis and properties of gallium oxide nanomaterials. In this work, we prepared nano to  
91 micro sized gallium oxide under different experimental procedures via a soft chemistry route.  
92 To make a suitable comparison, nonionic and cationic surfactants were used to obtain more  
93 understanding of the growth of gallium oxide. Interestingly, it was observed that Ga<sub>2</sub>O<sub>3</sub>  
94 nanotubes with width of ~3 nm can be achieved via a low temperature hydrothermal route.  
95 To our best knowledge, this is the first report that nanoscaled gallium oxide nanotubes can be

96 achieved under soft-chemistry synthesis route at a low temperature (100 °C) without any  
97 calcination.

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## 100 **Experimental**

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102 A Ga(NO<sub>3</sub>)<sub>3</sub> solution was prepared by dissolving 0.01 mol Ga(NO<sub>3</sub>)<sub>3</sub>·9H<sub>2</sub>O into 20 ml ultra-  
103 pure water. The pH of the resulting Ga(NO<sub>3</sub>)<sub>3</sub> solution is in the range of 1.8~1.9. Aqueous 5  
104 M NaOH was added drop-wise to adjust and maintain the pH at fixed value of 7 or 10.

105 Sample was taken as soon as the pH reached the desired value. The resulting fluffy  
106 precipitate mixture was then hydrothermally treated at 100 °C for three days or at 180 °C for  
107 one day. After hydrothermal treatment, all samples were washed by water and dried in air at  
108 80 °C. Sample that prepared at pH 7 and treated at 100 °C for 3 days is named as S1; Sample  
109 that prepared at pH 10 and treated at 100 °C for 3 days is named as S2; sample that prepared  
110 at pH 10 and treated at 180 °C for one day is named as S3. In addition, S1, S2 and S3 were  
111 calcined at 900 °C for 2 hours. The heating temperature was raised from room temperature at  
112 a rate of 2.5 °C/min.

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## 116 **Characterisation**

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- 118 a) A Philips CM 200 transmission electron microscopy (TEM) at 200 KV is used to  
119 investigate the morphology of the boehmite nanofibres.
- 120 b) Energy-dispersive X-ray (EDX) microanalysis using the TEM was carried out using  
121 an Oxford Instruments Link ISIS microanalysis system.
- 122 c) XRD analyses were performed on a PANalytical X'Pert PRO X-ray diffractometer,  
123 with a Cu X-ray tube ( $\lambda = 1.54\text{\AA}$ ), operating at 45 kV and 35 mA.
- 124 d) Raman spectra were collected using an Olympus BHSM microscope, equipped with  
125 10 and 50×objectives and part of a Renishaw 1000 Raman microscope system, which  
126 also includes a monochromator, a filter system and a charge coupled device (CCD).  
127 Raman spectra were excited by a HeNe laser (532 nm) at a resolution of 2 cm<sup>-1</sup> in the  
128 range between 100 and 4000 cm<sup>-1</sup>. Repeated acquisition using the highest

129 magnification was accumulated to improve the signal to noise ratio. Spectra were  
130 calibrated using the  $520.5\text{ cm}^{-1}$  line of a silicon wafer.

131 e) FTIR emission spectroscopy was carried out on a Nicolet Nexus 870 FTIR  
132 spectrometer equipped with a TGS detector, which was modified by replacing the IR  
133 source with an emission cell. A description of the cell and principles of the emission  
134 experiment have been published elsewhere [41-44]. Approximately 0.2 mg of  
135 boehmite was spread as a thin layer (approximately 0.2 microns) on a 6 mm diameter  
136 platinum surface and held in an inert atmosphere within a nitrogen-purged cell during  
137 heating.

138 f) Band component analysis was undertaken using the Jandel 'Peakfit' (Erkrath,  
139 Germany) software package which enabled the type of fitting function to be selected  
140 and allows specific parameters to be fixed or varied accordingly. Band fitting was  
141 done using a Lorentz-Gauss cross-product function with the minimum number of  
142 component bands used for the fitting process. The Lorentz-Gauss ratio was  
143 maintained at values greater than 0.7 and fitting was undertaken until reproducible  
144 results were obtained with squared correlations ( $r^2$ ) greater than 0.995. Band fitting  
145 of the spectra is quite reliable providing there is some band separation or changes in  
146 the spectral profile.

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## 148 **Results and Discussion**

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### 150 **X-ray diffraction**

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152 All the XRD patterns of samples synthesized with and without surfactant are identical. Figure  
153 1 shows the XRD pattern of sample prepared with surfactant PEO. It was noted that GaOOH  
154 phase was obtained using low temperature hydrothermal treatment ( $\leq 180^\circ\text{C}$ ) and  $\text{Ga}_2\text{O}_3$   
155 generally can be prepared at high temperature  $\sim 500^\circ\text{C}$  as reported in recent studies [8, 34,  
156 36, 37, 39, 40]. All the samples in this study were treated at a relatively low temperature  
157 ( $<100^\circ\text{C}$ ). Interestingly, as shown in Figure 1, the diffraction peaks in the pattern can not be  
158 indexed to GaOOH. Although the overall diffraction pattern is broad indicating small  
159 particle size in the resulting sample, it can be clearly seen that all the peaks in Figure 1 can be  
160 assigned to cubic  $\gamma\text{-Ga}_2\text{O}_3$  gallium oxide (JCPDS card No.04-004-4118) rather than  
161 monoclinic  $\beta\text{-Ga}_2\text{O}_3$  (JCPDS card No. 00-041-1103).

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## **Infrared spectroscopy**

168           The infrared spectra of samples S1, S2 and S3 are shown in Figure 2. A broad H-O-H  
169 stretching band at around  $3400\text{ cm}^{-1}$  can be observed in spectra of samples synthesized at pH  
170 7 and pH 10 treated at both high temperature and low temperature. It was noted that  
171 compared with samples synthesized at pH 10, the H-O-H stretching band position of sample  
172 synthesized at pH 7 slightly shifted to higher wavenumbers. No difference in H-O-H  
173 stretching band position between samples synthesized at pH 10 but treated at  $100\text{ }^{\circ}\text{C}$  and  $180$   
174  $^{\circ}\text{C}$ . A broad peak at around  $3480\text{ cm}^{-1}$  was present in the spectra of samples synthesized at pH  
175 7 and pH 10 treated at low temperature but no peak can be observed for sample synthesized  
176 at high temperature. The bands at around  $2903$  and  $2836\text{ cm}^{-1}$  can be assigned to the -OH  
177 stretching vibration of GaOOH. The spectra also have a broad band at the range of  $2298$  to  
178  $2254\text{ cm}^{-1}$  and a sharp band at  $\sim 2030\text{ cm}^{-1}$ . A band at  $2038\text{ cm}^{-1}$  and  $2020$  were also observed  
179 in some recent IR studies on  $\alpha$ -GaO(OH) rod-like and spindle-like micro sized crystals [26,  
180 45], whereas a band at  $2000\text{ cm}^{-1}$  appeared for a mixture of aggregated nanosized  $\alpha$ -  
181 GaO(OH) particles. The bands at around  $1025$ ,  $945$  and  $2030\text{ cm}^{-1}$  are assigned to the Ga-  
182 OH bending mode of GaOOH and its overtones, respectively [8, 32] An additional band at  
183 around  $1630\text{ cm}^{-1}$  can be observed in some of the GaOOH samples and is assigned to the  
184 bending mode of  $\text{H}_2\text{O}$ . The presence of band  $1359\text{ cm}^{-1}$  was interpreted as symmetric C=O  
185 vibration due to specific adsorption of atmospheric  $\text{CO}_2$  onto the hydroxide samples during  
186 the preparation and processing of FTIR samples in the ambient atmosphere [8]. The group of  
187 bands including  $690$  and  $621\text{ cm}^{-1}$  appearing within the range of  $1200$ - $600\text{ cm}^{-1}$  is  
188 characteristic Ga-O bands seen in hydroxo complexes [46]. It was reported that the band at  
189  $690\text{ cm}^{-1}$  is assigned to  $\text{Ga}_2\text{O}$  bending bands [47].

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A number of studies of the synthesis and characterisation of GaO(OH) nanomaterials  
have been reported [48-53]. Previous studies by Huang et al. [50], Ristic et al. [52], and Tas  
et al. [53] only showed infrared spectra as transmittance spectra and these authors did not  
undertake any band component analysis to determine precisely how many bands were present  
in the infrared spectral profile nor did these authors determine the exact position of these  
bands. Some of the published spectra show the presence of vibrations which can only be

197 attributed to CH stretching vibrations of organic molecules. In this work, the  $\alpha$ -GaO(OH)  
198 nanomaterials were synthesised without the use of surfactants or crystal growth directing  
199 agents.

200

201 Huang noted that as the amount of GaO(OH) increased the spectral profile shifted  
202 from  $\sim 3400$  to  $3200\text{ cm}^{-1}$  [50]. This is understandable if the two bands at  $2940$  and  $2850\text{ cm}^{-1}$   
203 increased in intensity. Ristic et al. attributed two bands at  $2990$  and  $3243\text{ cm}^{-1}$  to structural  
204 OH units and it is assumed although Ristic did not state it, the bands are due to GaO(OH)  
205 stretching vibrations [52]. The spectra as shown by Tas et al. [53] are poorly resolved and it  
206 is not possible to make any valid comparisons. The spectra are further complicated by the  
207 presence of urea the presence of which will complicate the spectra through additional NH  
208 stretching bands. Liu et al. assigned to broad profile centred at  $2850\text{ cm}^{-1}$  to water stretching  
209 and GaO(OH) stretching vibrations but did not identify which part of the spectrum was due to  
210 these units [51]. These bands were not present in the thermally treated GaO(OH)  
211 nanomaterials as the material was converted to either  $\alpha$ -Ga<sub>2</sub>O<sub>3</sub> or  $\beta$ -Ga<sub>2</sub>O<sub>3</sub>.

212

213 The infrared spectra of the GaO(OH) nanomaterials in the  $500$  to  $1500\text{ cm}^{-1}$  region are shown  
214 in Figure 2. In all of the spectra shown in this figure two bands are observed at around  $948$   
215 and  $1017\text{ cm}^{-1}$ . These bands are assigned to the GaOH unit deformation modes. In the  
216 spectra shown by Tas et al. [53], two bands are observed at  $952$  and  $1026\text{ cm}^{-1}$  and were  
217 assigned to Ga-OH bending modes. These bands were present whether the GaO(OH)  
218 nanomaterials was prepared with or without urea [53]. Only when the materials were  
219 thermally treated to elevated temperatures were the bands not observed. Obviously the  
220 materials had been thermally treated to temperatures above the decomposition temperature of  
221 GaO(OH). The reason why two GaOH deformation modes are observed is because there are  
222 two different GaOH units in the unit cell of GaO(OH). Huang et al. [50] observed two bands  
223 at around  $940$  and  $1020\text{ cm}^{-1}$  for GaO(OH) and assigned the bands to OH bending modes.  
224 Ristic et al. showed that the position of these GaOH deformation modes were sensitive to the  
225 preparation and subsequent treatment of the GaO(OH) materials [52]. In this work there is  
226 some evidence that the GaOH deformation modes are also sensitive to the consequential  
227 thermal treatment. The bands are observed at  $944$  and  $1017\text{ cm}^{-1}$  for the GaO(OH) treated at  
228 pH=5 at  $100^\circ\text{C}$  for 3 days and are found at  $948$  and  $1025\text{ cm}^{-1}$  for GaO(OH) treated at pH=10  
229 at  $100^\circ\text{C}$  for 3 days. In comparison the bands are observed at  $951$  and  $1027\text{ cm}^{-1}$  when treated  
230 at  $180^\circ\text{C}$  at pH=10 for 1 day.



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232 A significant number of bands are observed in the 620 to 725  $\text{cm}^{-1}$  region. These  
233 bands are assigned to GaO stretching vibrations. For the GaO(OH) treated at pH=5 at 100°C  
234 for 3 days infrared bands are observed at 608, 624, 650, 691 and 726  $\text{cm}^{-1}$ ; for GaO(OH)  
235 treated at pH=10 at 100°C for 3 days bands are observed at 621, 639, 658 and 692  $\text{cm}^{-1}$ . The  
236 number and position of the bands depends upon the thermal treatment and pH of the  
237 GaO(OH). There appears to be a band at a position below 525  $\text{cm}^{-1}$  which is below the cut-  
238 off point of the diamond ATR cell. Ristic et al. found bands at around 500  $\text{cm}^{-1}$  [52]. These  
239 authors also identified bands at 640 and 688  $\text{cm}^{-1}$  with some variation in position depending  
240 upon the treatment of the GaO(OH). However no assignment of the bands in these positions  
241 was forthcoming. Huang et al. did not record FTIR spectra below  $\sim 800 \text{ cm}^{-1}$  [50]. Liu  
242 assigned bands at 457 and 652  $\text{cm}^{-1}$  to Ga<sub>2</sub>O bending modes [51]. This assignment appears  
243 strange and without foundation. If there is a GaOGa bending vibration then there must first  
244 be a GaO stretching vibration. A more likely assignment is the band at 692  $\text{cm}^{-1}$  is the GaO  
245 stretching vibration and the band at 457  $\text{cm}^{-1}$  is the OGaO bending mode. In this research  
246 two infrared bands were observed at around 1940 and 2033  $\text{cm}^{-1}$ . The assignment of these  
247 bands is open to question but one probable attribution is to the first fundamental overtone of  
248 the GaO(OH) deformation modes at around 950 and 1017  $\text{cm}^{-1}$ . Bands in these positions  
249 were observed in the spectra of GaO(OH) by Liu et al. but were not assigned [51]. These  
250 authors also found infrared bands at 2343 and 2380  $\text{cm}^{-1}$ ; these bands are due to atmospheric  
251 CO<sub>2</sub> absorption.

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254 The infrared spectra may be compared with the Raman spectra. Raman spectroscopy  
255 shows bands characteristic of  $\alpha$ -GaO(OH) at 950,  $\sim 1000 \text{ cm}^{-1}$  attributed to Ga-OH  
256 deformation modes. Bands at 261, 275, 433 and 522  $\text{cm}^{-1}$  are assigned to vibrational modes  
257 involving Ga-OH units. Bands observed at 320, 346, 418 and 472  $\text{cm}^{-1}$  are assigned to the  
258 deformation modes of Ga<sub>2</sub>O<sub>6</sub> octahedra. Two sharp infrared bands at 2948 and 2916  $\text{cm}^{-1}$   
259 are attributed to the GaO(OH) symmetric stretching vibrations. Raman spectroscopy of  
260 Ga<sub>2</sub>O<sub>3</sub> provides bands at 630, 656 and 767  $\text{cm}^{-1}$  which are assigned to the bending and  
261 stretching of GaO<sub>4</sub> units. Raman bands at 417 and 475  $\text{cm}^{-1}$  are attributed to the symmetric  
262 stretching modes of GaO<sub>2</sub> units. The Raman bands at 319 and 347  $\text{cm}^{-1}$  are assigned to the  
263 bending modes of GaO<sub>2</sub> units.

264

265 **Infrared emission spectroscopy**

266

267 The infrared emission spectra of  $\alpha$ -GaO(OH) are shown in Figures 3, 4 and 5. The  
268 significance of these three figures is that different chemical routes are used to synthesise the  
269  $\alpha$ -GaO(OH), yet the spectra are identical, as expected.

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271 The infrared emission spectra clearly show a loss of intensity of the bands associated  
272 with hydroxyl groups with temperature increase. The intense band at  $2950\text{ cm}^{-1}$  is assigned  
273 to a combination of water and OH units associated with the  $\alpha$ -GaO(OH) structure. No  
274 intensity remains in this band at  $450\text{ }^\circ\text{C}$ . The three bands at  $1940$ ,  $2050$  and  $2240\text{ cm}^{-1}$  are  
275 assigned to the -OH stretching vibration of  $\alpha$ -GaO(OH). The intensity of these bands is also  
276 lost by  $400\text{ }^\circ\text{C}$ .

277

278 **Thermal analysis of  $\alpha$ -GaO(OH)**

279

280 The thermal analysis of  $\alpha$ -GaO(OH) is shown in Figure 6. Thermal decomposition  
281 of  $\alpha$ -GaO(OH)  $\rightarrow$   $\beta$ -Ga<sub>2</sub>O<sub>3</sub> occurs at  $359\text{ }^\circ\text{C}$ . This mass loss is attributed to the  
282 dehydroxylation of  $\alpha$ -GaO(OH) . A small mass loss at  $225\text{ }^\circ\text{C}$  also occurs which is  
283 attributed to intercalated water.

284 The reaction is as follows:  $2\ \alpha\text{-GaO(OH)} \rightarrow \beta\text{-Ga}_2\text{O}_3 + \text{H}_2\text{O}$

285 The results of the temperature of the mass loss step is in harmony with the infrared emission  
286 results above which showed the loss of OH units by  $400\text{ }^\circ\text{C}$ .

287 **Conclusions**

288

289 Infrared emission spectroscopy was used to study the transition of micro-sized  $\alpha$ -GaO(OH) to  
290  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> rods. By using soft chemical techniques and low temperatures micro-sized  
291 GaO(OH) nanorods were synthesised. Large crystals were readily obtained. GaO(OH) rod-  
292 like micro-sized crystal  $\sim$ 2.5  $\mu$ m in length and 1.6  $\mu$ m in width were prepared at 100 °C  
293 without surfactant via hydrothermal route when the initial Ga to OH ratio was 1:3. The  
294 growth of GaO(OH) crystal mainly occurred during ageing and the growth rate dramatically  
295 decreased once the nanosized particle aggregate disappeared and the relatively large rod-like  
296 crystals were formed.  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> crystals were obtained through calcination and the rod-like  
297 morphology of the GaO(OH) was retained.

298

299 Infrared spectroscopy of  $\alpha$ -GaO(OH) enabled bands at 2848 and 2916 cm<sup>-1</sup> are  
300 attributed to the  $\alpha$ -GaO(OH) symmetric stretching vibrations.

301

302 No intensity was observed in these bands after 200°C. Raman bands at 950 and 1010  
303 cm<sup>-1</sup> are assigned to the hydroxyl deformation modes of  $\alpha$ -GaO(OH). After 350 °C the  
304 Raman spectrum shows a significantly different pattern to that of  $\alpha$ -GaO(OH). Raman bands  
305 are observed at 155, 212, 280, 430, 570 and 685 cm<sup>-1</sup>. These bands are assigned to  $\beta$ -  
306 Ga<sub>2</sub>O<sub>3</sub>. Bands at 262, 275, 430, 520 cm<sup>-1</sup> are assigned to vibrational modes involving Ga-OH  
307 units.

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325 **References**

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- 327 [1] J.H. Fendler, F.C. Meldrum, *Adv. Mat.* 7 (1995) 607-632.
- 328 [2] S. Iijima, *Nature* 354 (1991) 56-58.
- 329 [3] B.B. Lakshmi, C.J. Patrissi, C.R. Martin, *Chem. Mat.* 9 (1997) 2544-2550.
- 330 [4] S. Gowtham, A. Costales, R. Pandey, *Chem. Phys. Lett.* 431 (2006) 358-363.
- 331 [5] L. Binet, D. Gourier, C. Minot, *J. Solid State Chem.* 113 (1994) 420-433.
- 332 [6] S. Sharma, M.K. Sunkara, *J. Am. Chem. Soc.* 124 (2002) 12288-12293.
- 333 [7] L. Binet, D. Gourier, *J. Phys. Chem. Solids* 59 (1998) 1241-1249.
- 334 [8] A.C. Tas, Peter J. Majewski, F. Aldinger, *J. Am. Chem. Soc.* 85 (2002) 1421-1429.
- 335 [9] M. Ogita, N. Saika, Y. Nakanishi, Y. Hatanaka, *Appl. Surf. Sc.* 142 (1999) 188-191.
- 336 [10] T. Weh, J. Frank, M. Fleischer, H. Meixner, *Sens. Actuators B78* (2001) 202-207.
- 337 [11] K. Nakagawa, C. Kajita, K. Okumura, N.-o. Ikenaga, M. Nishitani-Gamo, T. Ando, T.  
338 Kobayashi, T. Suzuki, *J. Catal.* 203 (2001) 87-93.
- 339 [12] A.L. Petre, A. Auroux, P. Gelin, M. Caldararu, N.I. Ionescu, *Thermochim. Act.* 379  
340 (2001) 177-185.
- 341 [13] B. Xu, B. Zheng, W. Hua, Y. Yue, Z. Gao, *J. Catal.* 239 (2006) 470-477.
- 342 [14] T. Ishihara, H. Matsuda, Y. Takita, *J. Am. Chem. Soc.* 116 (1994) 3801-3803.
- 343 [15] Y. Zhao, R.L. Frost, W.N. Martens, H.Y. Zhu, *J. Therm. Anal. Cal.* 90 (2007) 755-  
344 760.
- 345 [16] Y. Zhao, R.L. Frost, W.N. Martens, H.Y. Zhu, *Lang.* 23 (2007) 9850-9859.
- 346 [17] Y. Zhao, R.L. Frost, W.N. Martens, *J. Phys. Chem. C111* (2007) 5313-5324.
- 347 [18] Y. Zhao, R.L. Frost, J. Yang, W.N. Martens, *J. Phys. Chem. C112* (2008) 3568-3579.
- 348 [19] L. Dai, L.P. You, X.F. Duan, W.C. Lian, G.G. Qin, *J. Cryst. Growth* 267 (2004) 538-  
349 542.
- 350 [20] X. Xiang, C.-B. Cao, Y.-J. Guo, H.-S. Zhu, *Chem. Phys. Lett.* 378 (2003) 660-664.
- 351 [21] W.-S. Jung, H.U. Joo, B.-K. Min, *Physica E36* (2007) 226-230.
- 352
- 353 [22] H.-D. Xiao, H.-L. Ma, C.-S. Xue, H.-Z. Zhuang, J. Ma, F.-J. Zong, X.-J. Zhang, *Mat.*  
354 *Chem. Phys.* 101 (2007) 99-102.
- 355 [23] N.H. Kim, H.W. Kim, C. seoul, C. Lee, *Mat. Sc. Eng, B* 111 (2004) 131-134.
- 356 [24] F. Zhu, Z. Yang, W. Zhou, Y. Zhang, *Appl. Surf. Sc.* 252 (2006) 7930-7933.
- 357 [25] Y.C. Choi, W.S. Kim, Y.S. Park, S.M. Lee, D.J. Bae, Y.H. Lee, G.-S. Park, W.B.  
358 Choi, N.S. Lee, J.M. Kim, *Adv. Mater.* 12 (2000) 746-750.

- 359 [26] C.-C. Huang, C.-S. Yeh, C.-J. Ho, *J. Phys. Chem. B* 108 (2004) 4940-4945.
- 360 [27] B.C. Kim, K.T. Sun, K.S. Park, K.J. Im, T. Noh, M.Y. Sung, S. Kim, *Appl. Phys.*  
361 *Lett.* 80 (2002) 481.
- 362 [28] F. Zhu, Z.X. Yang, W.M. Zhou, Y.F. Zhang, *Solid State Comm.* 137 (2006) 177-181.
- 363 [29] Y.H. Gao, Y. Bando, T. Sato, Y. F. Zhang, *Appl. Phys. Lett.* 81 (2002) 2267-2269.
- 364 [30] C.H. Liang, G.W. Meng, G.Z. Wang, Y.W. Wang, L.D. Zhang, *Appl. Phys. Lett.* 78  
365 (2001) 3202-3204.
- 366 [31] H.J. Chun, Y.S. Choi, S.Y. Bae, H.W. Seo, S.J. Hong, J. Park, H. Yang, *J. Phys.*  
367 *Chem. B* 107 (2003) 9042-9046.
- 368 [32] T. Sato, T. Nakamura, *J. Chem. Tech. Biotechnol.* 32 (1982) 469-475.
- 369 [33] S. Hamada, K. Bando, Y. Kudo, *Bull. Chem. Soc. Jpn.* 59 (1986) 2063-2069.
- 370 [34] S. Avivi, Y. Mastai, G. Hodes, A. Gedanken, *J. Am. Chem. Soc.* 121 (1999) 4196-  
371 4199.
- 372 [35] B. Cheng, E.T. Samulski, *J. Mater. Chem.* 11 (2001) 2901.
- 373 [36] C.R. Patra, Y. Mastai, A. Gedanken, *J. Nanopart. Res.* 6 (2004) 509.
- 374 [37] J. Zhang, Z. Liu, C. Lin, J. Lin, *J. Crystal Growth* 280 (2005) 99-106.
- 375 [38] M. Ristic, S. Popovic, S. Music, *Mat. Lett.* 59 (2005) 1227-1233.
- 376 [39] X. Liu, G. Qiu, Y. Zhao, N. Zhang, R. Yi, *J. Alloy. Comp.* 439 (2007) 275-278.  
377
- 378 [40] Y.C. Zhang, X. Wu, X.Y. Hu, Q.F. Shi, *Materials Letters* 61 (2007) 1497-1499.
- 379 [41] R.L. Frost, A.M. Vassallo, *Clays Clay Miner.* 44 (1996) 635-651.
- 380 [42] R.L. Frost, A.M. Vassallo, *Mikrochim. Acta, Suppl.* 14 (1997) 789-791.
- 381 [43] R.L. Frost, J.T. Kloprogge, *Spectrochim. Acta*, 55A (1999) 2195-2205.
- 382 [44] R.L. Frost, M.L. Weier, *Thermochim. Act.* 406 (2003) 221-232.
- 383 [45] X. Liu, G. Qiu, Y. Zhao, N. Zhang, R. Yi, *J. Alloys Comp.* 439 (2007) 275-278.
- 384 [46] G. Socrates, *Infrared Characteristic Group Frequencies*, Wiley, New York, 1994.
- 385 [47] A.J. Hinchclife, J.S. Ogden, *J. Phys. Chem.* (1973) 2537.
- 386 [48] S. Avivi, Y. Mastai, G. Hodes, A. Gedanken, *J. Am. Chem. Soc.* 121 (1999) 4196-  
387 4199.
- 388 [49] B.J. Clapsaddle, D.W. Sprehn, A.E. Gash, J.H. Satcher, Jr., R.L. Simpson, *J. Non-  
389 Cryst. Solids* 350 (2004) 173-181.
- 390 [50] C.-C. Huang, C.-S. Yeh, C.-J. Ho, *J. Phys. Chem. B* 108 (2004) 4940-4945.
- 391 [51] X. Liu, G. Qiu, Y. Zhao, N. Zhang, R. Yi, *J. Alloys Comp.* 439 (2007) 275-278.
- 392 [52] M. Ristic, S. Popovic, S. Music, *Mat. Lett.* 59 (2005) 1227-1233.

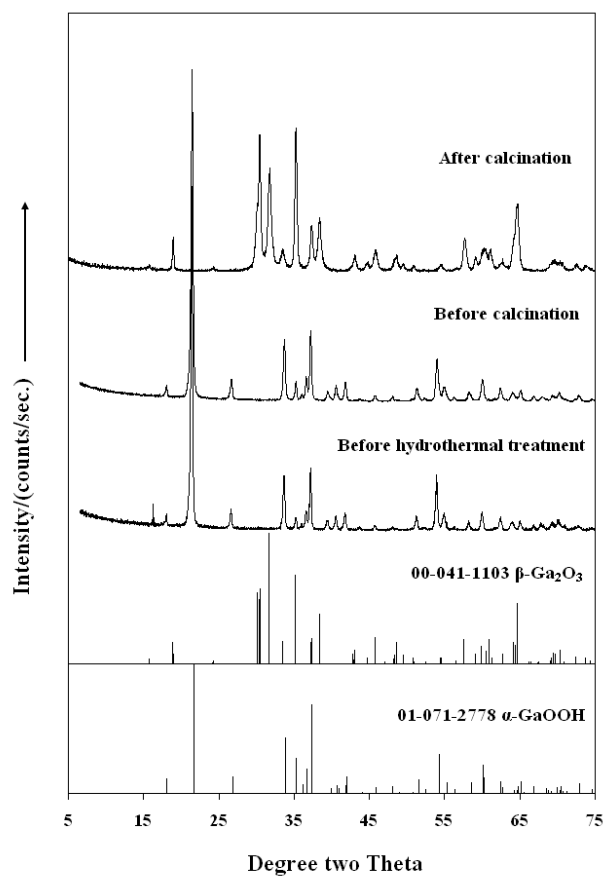
393 [53] A.C. Tas, P.J. Majewski, F. Aldinger, J. Am. Ceram. Soc. 85 (2002) 1421-1429.

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397



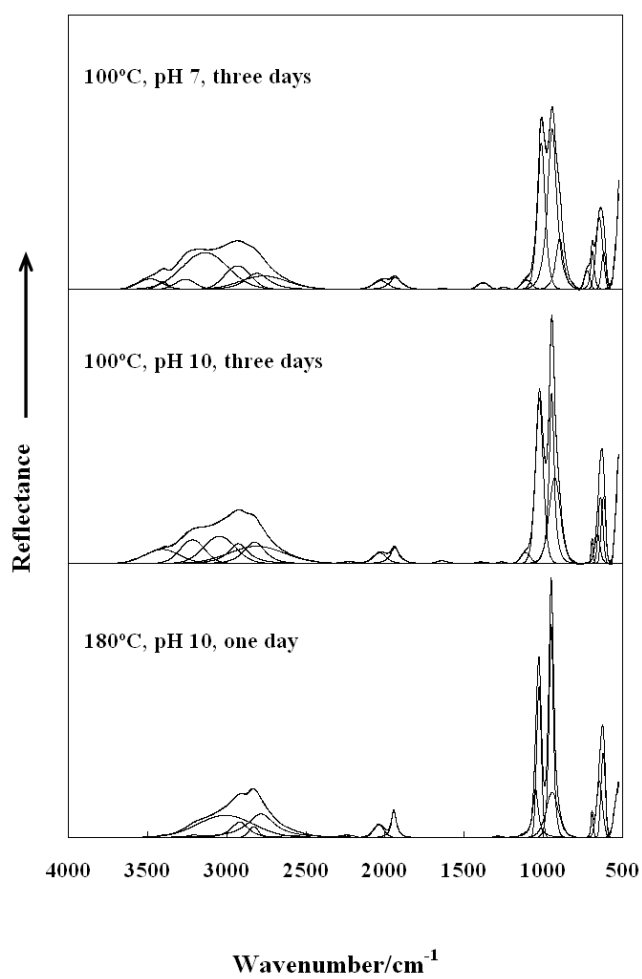
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400 **Figure 1 XRD patterns for samples synthesized**

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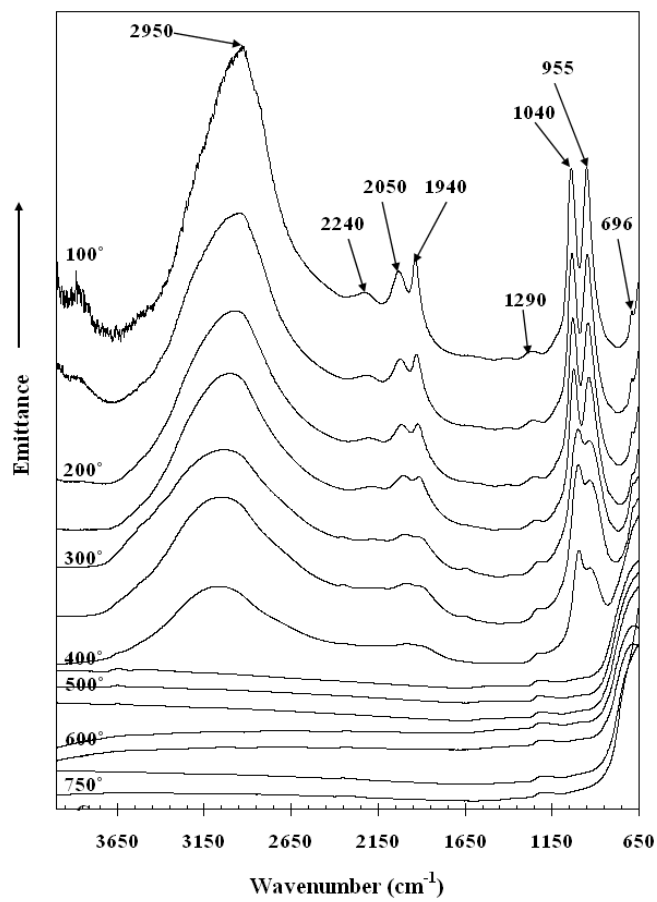




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404 **Figure 2**



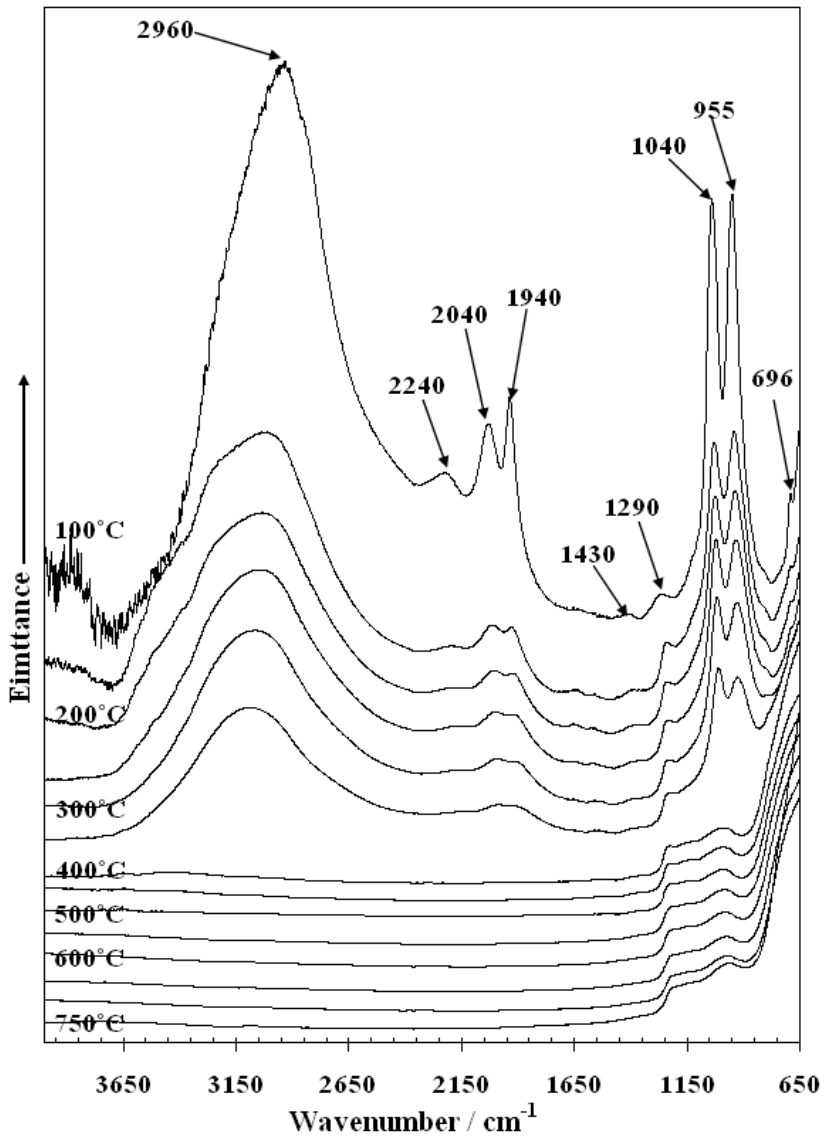
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407 **Figure 3**

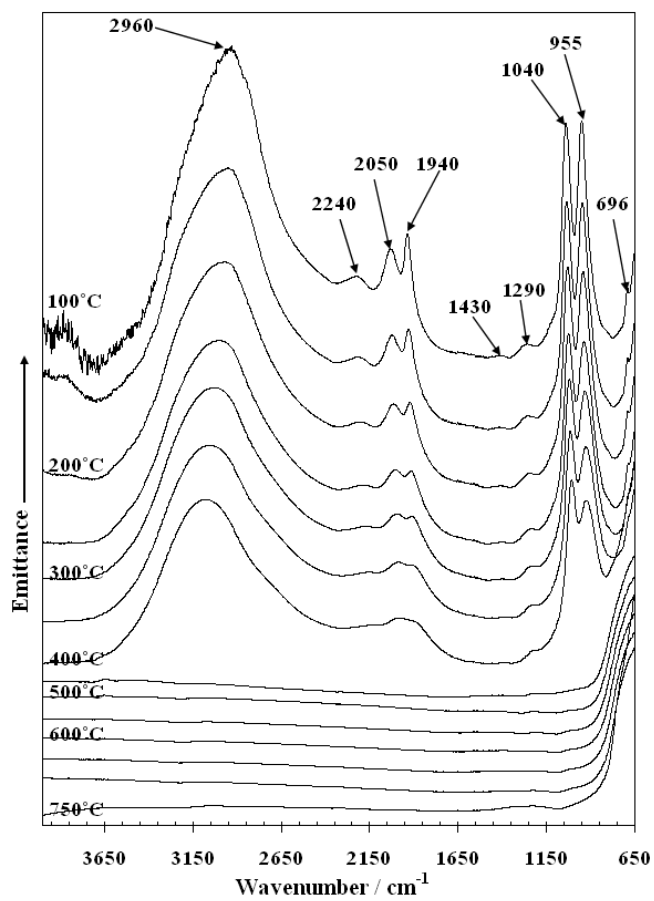
**2D3**

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**Figure 4 F34**



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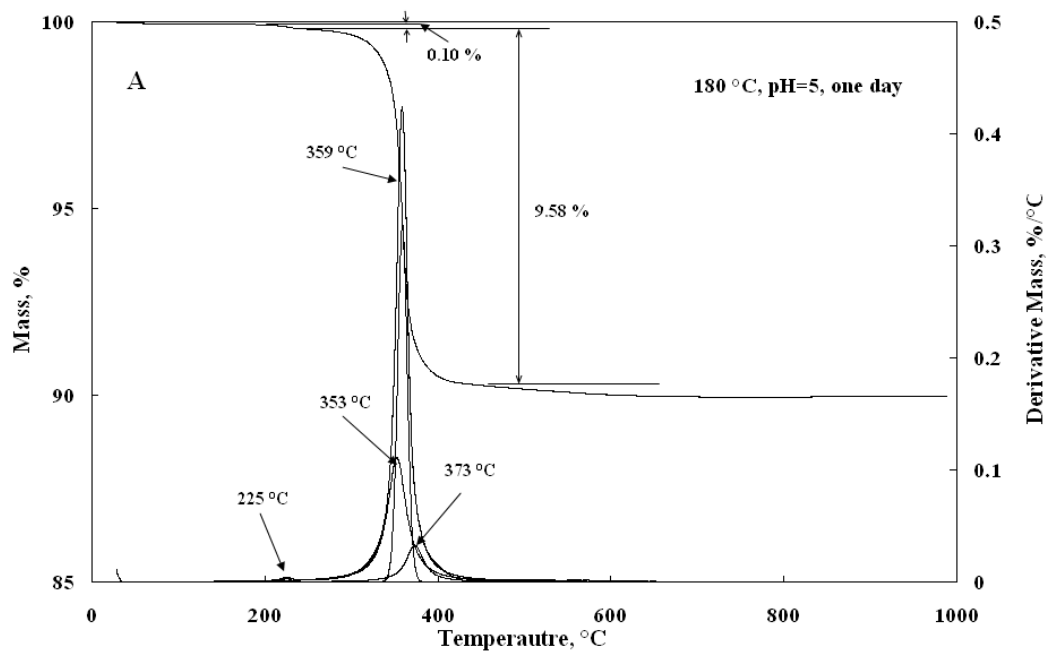
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417 **Figure 5 F32**

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423 **Figure 6 Thermal analysis of GaO(OH)**

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