

1 **A dual-application poly (DL-lactic-co-glycolic) acid (PLGA)-chitosan composite scaffold**
2 **for potential use in bone tissue engineering**

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22 **Abstract**

23 The development of patient-friendly alternatives to bone-graft procedures is the driving force
24 for new frontiers in bone tissue engineering. Poly (DL-lactic-co-glycolic acid), (PLGA) and
25 chitosan are well-studied and easy-to-process polymers from which scaffolds can be
26 fabricated. In this study, a novel dual-application scaffold system was formulated from
27 porous PLGA and protein-loaded PLGA/chitosan microspheres. Physicochemical and *in vitro*
28 protein release attributes were established. The therapeutic relevance, cytocompatibility with
29 primary human mesenchymal stem cells (hMSCs) and osteogenic properties were tested.
30 There was a significant reduction in burst release from the composite PLGA/chitosan
31 microspheres compared with PLGA alone. Scaffolds sintered from porous microspheres at
32 37°C were significantly stronger than the PLGA control, with compressive strengths of 0.846
33 ± 0.272 MPa and 0.406 ± 0.265 MPa, respectively (p < 0.05). The formulation also sintered at
34 37°C following injection through a needle, demonstrating its injectable potential. The
35 scaffolds demonstrated cytocompatibility, with increased cell numbers observed over an 8-
36 day study period. Von Kossa and immunostaining of the hMSC-scaffolds confirmed their
37 osteogenic potential with the ability to sinter at 37°C *in situ*.

38 Keywords: polymeric biomaterials, controlled delivery, poly (lactic-co-glycolic acid)
39 (PLGA), microspheres, protein delivery, tissue engineering, mechanical properties,
40 formulation.¹

¹Abbreviations

BMPs, bone morphogenetic proteins; BSA, bovine serum albumin; DCM, dichloromethane; DMSO, dimethyl sulphoxide; ECM, extracellular matrix; FTIR, Fourier transform infrared; hMSC, primary human mesenchymal stem cells; PBS, phosphate-buffered saline; PLGA, poly (lactic-co-glycolic acid); PVA, poly (vinyl alcohol); SDS, sodium dodecyl sulphate; SEM, scanning electron microscopy, TPP, sodium tripolyphosphate; ToF-SIMS, time of flight secondary ion mass spectroscopy.

41 **1. Introduction**

42 There is an urgent need for alternative approaches for the regeneration of bone
43 following fracture or orthopaedic damage *in lieu* of traditional methods, and these alternative
44 approaches constitute an important tissue engineering application (Vo et al., 2012). The
45 current ‘gold standard’ therapy is the bone graft procedure, which involves taking autologous
46 bone, usually harvested from the iliac crest of the patient, and implanting it into their defect
47 site (Martino et al., 2012; Amini et al., 2013). Alternatively, allograft bone from donors or
48 cadavers can be extracted from the femoral heads or extremities of other long bones (Delloye
49 et al., 2007). This implanted tissue acts as a scaffold for the existing bone tissue to infiltrate
50 and deposit extracellular matrix (ECM), leading to the remodelling of the fractured bone
51 (Bostrom and Mikos, 1997). Numerous drawbacks are associated with the above procedures,
52 including the limited supply of autologous bone, complications at the donor site and high
53 surgical costs (Martino et al., 2012). Furthermore, in large defects, resorption may occur
54 before osteogenesis has been completed (Burg et al., 2000). Allograft bone usage is associated
55 with incompatibility with the host, and the possible transmission of diseases and infections
56 such as hepatitis and HIV (Vo et al., 2012; Bostrom and Mikos, 1997; Chen et al., 2010;
57 Puppi 2010). The risk of disease transmission from allograft bone can be minimised by
58 processing or devitalization via freeze-drying or irradiation; however, this may reduce the
59 osteoinductivity and mechanical strength (White et al., 2013; Hau et al., 2008; Nauth et al.,
60 2011). Other options include the usage of bone morphogenetic proteins (BMPs), distraction
61 osteogenesis and bone cement; however, these are also not ideal (Amini et al., 2013). The
62 shortcomings in the current clinical options have led to concerted efforts in search of
63 alternative strategies for the repair of bone.

64 Poly (DL-lactic-co-glycolic acid) (PLGA) is a well-studied synthetic polymer used in
65 bone tissue engineering. It has favourable properties such as biodegradability (Pan and Ding,

66 2012), cytocompatibility, controllable mechanical properties (Bostrom and Mikos, 1997;
67 Burg et al., 2000; Chen et al., 2010; Puppi et al., 2010) and it can be easily processed (Burg et
68 al., 2000; Pan and Ding, 2012). Furthermore, PLGA has been approved by the FDA for use in
69 certain clinical applications (Lu et al., 2009).

70 The combination of porous and non-porous microspheres, which are able to sinter at
71 body temperature, enables the introduction of porosity within injected scaffolds, hence,
72 allowing proliferating cells access to nutrients [Qutachi et al., 2014; Boukari et al., 2015].
73 Simultaneously, the delivery of growth factors such as BMPs to the growing cells is also
74 facilitated. BMPs have been studied for their use in non-union bone defects, spinal fusion and
75 open tibial fractures (Boukari et al., 2015; Whilte et al., 2013; Hau and Wang, 2008).
76 Furthermore, it has been reported that one such BMP, BMP-2, is present during the initial
77 phase of fracture repair, and during chondrogenesis and osteogenesis (Patel et al., 2008).

78 Various strategies have been utilized for the sintering of microspheres into scaffolds.
79 These include the incorporation of plasticizers in order to reduce polymer glass transition
80 temperatures (Dhillon et al., 2011), the addition of organic solvents such as dichloromethane
81 (Pan and Ding, 2012; Wang et al., 2010) and the application of heat (Delloye et al., 2007;
82 Chen et al., 2010; Puppi et al, 2010). Although the use of high temperatures and organic
83 solvents result in mechanically strong scaffolds, these conditions are not ideal for the body
84 and so are not suitable for sintering *in-situ*. Therefore, a system capable of sintering at 37°C
85 *in situ* would be extremely beneficial.

86 Protein-loaded PLGA microspheres often exhibit an initial burst release (Boukari et al.,
87 2015; Tao et al., 2014) which is not ideal for an intended controlled release of BMP-2 at a
88 defect site. A number of strategies have been employed to control the release of proteins from
89 PLGA microspheres. These include varying the polymer molecular weight (Boukari et al.,

90 2015), the inclusion of additives such as poloxamer 188 (Paillard-Giteau et al., 2010) and the
91 use of a PLGA-PEG-PLGA triblock polymer (White et al., 2013; Kirby et al., 2011).

92 Chitosan is a natural polysaccharide derived from chitin and is popular in tissue
93 engineering applications for a variety of reasons, which include its cytocompatibility and
94 ability to promote cell adhesion (Amini et al., 2012). Chitosan microspheres show promise
95 for use in the encapsulation of proteins and have previously been shown to retain the activity
96 of a neural growth factor (Zeng et al., 2011). Moreover, due to its cationic nature and
97 propensity to slow degradation, chitosan-based materials are able to sustain the release of
98 growth factors (Qian and Zhang, 2013). Chitosan has been used in combination with PLGA
99 in various forms, including by embedding PLGA microspheres into chitosan scaffolds (Kirby
100 et al., 2011; Zeng et al., 2011; Di Martino et al., 2005; Qian, 2013). PLGA/chitosan
101 microspheres can be formulated in a variety of ways. These include the use of supercritical
102 fluid technology (Cassetari et al., 2011), the double emulsion method (Fu et al., 2012; Hu et
103 al., 2008) the solvent evaporation technique (Jian et al., 2010), an electro-dropping layer-by-
104 layer approach (Choi et al., 2013) and conjugation and adsorption methods (Chakravarthi and
105 Robinson, 2011). Porous microspheres have also been treated with chitosan (Yue et al., 2015)
106 (Chakravarthi and Robinson, 2011), whilst others have encapsulated protein-loaded chitosan
107 microspheres into large porous PLGA microspheres (Tao et al., 2014).

108 In a previous study, we reported the formulation of a novel PLGA scaffold delivery
109 system based on porous and protein-loaded microspheres that sintered at 37°C (Boukari et al.,
110 2015). There have been a number of reports utilising composites of PLGA/chitosan
111 microspheres for use in bone tissue engineering (Casettari et al., 2011; Han et al., 2015;
112 Pandey et al., 2013; Jiang et al., 2010; Choi et al., 2013; Chakravarthi and Robinson, 2011).
113 In the present work, we report the development of a ‘dual-application’ PLGA/chitosan
114 composite scaffold formulation which sinters at 37°C when injected through a hypodermic

115 needle as well as when implanted as a paste. Furthermore, we aimed to control the release
116 kinetics of a model protein for BMP-2 (BMP-2 itself was not used due to the cost
117 implications) from this system, via the inclusion of chitosan, and to investigate its
118 cytocompatibility and osteoinductive capabilities on primary human mesenchymal stem cells
119 (hMSCs).

120 2. Materials and methods

121 2.1 Materials

122 PLGA (85:15, 53 kDa) was purchased from Evonik (Morris, NJ, USA). Chitosan, low
123 molecular weight, $\geq 75\%$ deacetylation; sodium tripolyphosphate (TPP); poly vinyl alcohol
124 (PVA), 87–89% hydrolysed; phosphate buffered saline (PBS; 0.01 M phosphate buffer,
125 0.0027 M potassium chloride and 0.137 M sodium chloride; pH 7.4) tablets; sodium
126 hydroxide (NaOH) pellets; Triton X-100; goat serum; Hoechst 33258; sodium thiosulphate
127 solution; silver nitrate solution; formalin 10% v/v and paraformaldehyde 10% v/v solutions
128 were purchased from Sigma-Aldrich (St. Louis, MO, USA). Glacial acetic acid was
129 purchased from R&M Chemicals (Essex, UK). Dichloromethane (DCM), dimethyl sulfoxide
130 (DMSO) and sodium dodecyl sulphate (SDS) were purchased from Fisher Scientific UK
131 (Loughborough, UK). Bovine serum albumin (BSA) was purchased from Nacalai Tesque
132 (Kyoto, Japan). A micro BCA protein assay kit was purchased from Thermo Fisher Scientific
133 (Waltham, MA, USA). For stem cell culture, hMSCs, an MSCGM hMSC SingleQuot kit,
134 trypsin/EDTA for MSC and HEPES buffered saline were purchased from Lonza (Basel,
135 Switzerland). Presto Blue cell viability reagent was purchased from Gibco, Life Technologies
136 (Carlsbad, CA, USA). For immunostaining, anti-osteocalcin polyclonal antibody was

137 purchased from Merck Millipore (Billerica, MA, USA) and alexa flour 488 goat anti-rabbit
138 IgG was purchased from Invitrogen (Carlsbad, CA, USA).

139 **2.2 Formulation of PLGA microspheres**

140 Porous PLGA microspheres were prepared using the double emulsion solvent
141 evaporation method as described in detail elsewhere (Qutachi et al., 2014; Boukari et al.,
142 2015). Briefly, a 250- μ l aliquot of PBS was added to a 20% w/v PLGA/DCM solution and
143 homogenized at 9000 rpm using a Silverson L5M homogeniser (East Longmeadow, MA,
144 USA). This was added to 200 ml of 0.3% w/v PVA solution and homogenized at 4000 rpm
145 and then stirred at 300 rpm for 4 hours. The microspheres were washed with distilled water
146 and then exposed to ethanolic-NaOH in order to enhance the surface porosity. They were
147 then sieved (40 μ m) and washed using distilled water. Non-porous microspheres were
148 prepared in a similar way using 100 μ l of 100 mg/ml BSA solution or 100 μ l of distilled
149 water, instead of 250 μ l of PBS. BSA was chosen as a model protein as it is compatible with
150 chitosan and has previously been used as a substitute for growth factors (Song et al., 2013;
151 Yilgor et al., 2010; Yilgor et al., 2009).

152 Non-porous PLGA/chitosan composite microspheres were prepared similarly;
153 however, instead of using 200 ml of 0.3% w/v PVA solution, the aqueous phase comprised
154 150 ml of 0.4% w/v PVA solution containing 0.05 g of TPP. The primary emulsion, in
155 addition to 50 ml of 0.25% w/v chitosan solution in 2% v/v acetic acid, was added to the
156 external aqueous phase simultaneously and homogenized. All microspheres were freeze-dried
157 using a Thermo Fisher Scientific FR-Drying Digital Unit (Waltham, MA, USA) for 48 hours
158 and stored at -20°C until use.

159 **2.3 Scanning electron microscopy (SEM) and size analysis**

160 The freeze-dried samples were mounted onto aluminium stubs (Agar Scientific, UK)
161 and gold-coated using a Balzers SCD030 gold sputter coater (Balzers Union Ltd.,
162 Lichtenstein). The morphology and surface topography of the microspheres were observed
163 using a Jeol 6060L SEM imaging system (Jeol Ltd., Hertfordshire, UK) at 10 kV. The
164 particle size distribution and mean microsphere diameter were determined using a Coulter
165 LS230 particle size analyser (Beckman, UK).

166 **2.4 Fourier transform infrared (FTIR) spectroscopy**

167 FTIR spectra of the microspheres and their constituents were obtained using a
168 Spectrum RX 1 FTIR spectrophotometer (Perkin Elmer, **Waltham, MA**, USA). Samples were
169 mixed with potassium bromide (KBr) and compressed using a 5-tonne force into disks; 256
170 scans were acquired from 400 to 4000 cm^{-1} .

171 **2.5 Preparation of 3D scaffolds**

172 PLGA and PLGA/chitosan composite scaffolds were previously prepared in our
173 laboratories (Boukari et al., 2015). A 1:1 mass ratio of porous to non-porous microspheres
174 was mixed in a weighing boat followed by mixing with PBS (pH 7.4) at a ratio of 0.25:1
175 (PBS to microspheres) to form a paste. The paste was packed into a 6-mm diameter and 12-
176 mm height polytetrafluorethylene (PTFE) mould using a spatula, and then stored in a sealed
177 de-humidifying chamber at 37°C for 17 hours.

178

179 **2.6 Time of flight secondary ion mass spectrometry (ToF-SIMS)**

180 The presence and distribution of the chitosan coating on the scaffold surfaces was
181 assessed using a time of flight secondary ion mass spectrometer (ToF-SIMS IV, ION-TOF
182 GmbH, Munster, Germany). Scaffolds were placed on the ToF-SIMS stage and secured with
183 metal clips. A 25-keV Bi₃⁺ primary ion source was used to scan a 256 × 256 pixel raster,
184 while simultaneously not exceeding the limit of static, as described by Rafati et al., (2012).
185 Surface charge due to the primary ion beam on the insulating sample surface was
186 compensated using a flood gun generating low energy electrons (20 eV). Negative and
187 positive polarity data for 500 × 500 μm areas were analysed using the SurfaceLab 6 software
188 (IONTOF, Germany). PLGA was identified by the presence of C₃H₃O₂⁻ (*m/z* = 71) and
189 C₃H₅O₂⁻ (*m/z* = 73) (Rafato et al., 2012). Diagnostic secondary ion peaks for chitosan were
190 identified as CN⁻ (*m/z* = 26) from the negative polarity data, in addition to CH₄N⁺ (*m/z* = 30)
191 and C₄H₅N₂⁺ (*m/z* = 81) from the positive polarity data. For a semi-quantitative analysis, each
192 area was split into four regions of interest, and the ion intensity data for these peaks of
193 interest were exported and normalized to the total ion intensity.

194 **2.7 Encapsulation efficiency (%EE) of BSA within microspheres and scaffolds**

195 The %EE of BSA within the non-porous PLGA and PLGA/chitosan composite
196 microspheres and scaffolds were determined by gently stirring 10 mg of the microspheres or
197 one scaffold in 750 μl or 13 ml of DMSO, respectively, for 1 hour. This was followed by the
198 addition of 2.15 ml or 37.27 ml of 0.02% w/v SDS in 0.2 M NaOH to the microspheres or
199 scaffolds, respectively. The solution was left to stand at room temperature for 1 hour.
200 Standard concentrations of BSA were calibrated with a BCA reagent so that the sample
201 absorbance could be matched with standard concentrations on an Infinite 200 plate reader
202 (Tecan, Switzerland) at 562 nm. The %EE of BSA within the microspheres and scaffolds was
203 then calculated using Equation 1.

204
$$\%EE = \frac{\text{Actual mass of BSA in 10mg of microspheres OR 1 scaffold}}{\text{Theoretical mass of BSA used for 10 mg of microspheres OR 1 scaffold}} \times 100 \quad (1)$$

205

206 **2.8 Release of BSA from microspheres and scaffolds**

207 Release studies of BSA from the PLGA and PLGA/chitosan composite microspheres
208 were carried out by submerging 50 mg of microspheres in 1.5 ml of PBS in a micro-
209 centrifuge tube. The tubes were incubated at 37°C. At predetermined time intervals, the PBS
210 supernatant was removed and replaced with fresh buffer. Aliquots (150 µl) were withdrawn
211 from the supernatant and assayed for the presence of BSA at 562 nm on the microplate reader
212 using the BCA assay kit. BSA release from scaffolds was studied in 4 ml of PBS and assayed
213 as described above.

214

215 **2.9 Preparation of 3D scaffolds post-injection**

216 Microsphere mixtures were prepared at a 1:1 ratio of porous to non-porous
217 microspheres and an approximate BSA loading of 3 mg of BSA/g of mixture, as described in
218 section 2.5. The mixture was suspended in PBS at a concentration of 50 mg of
219 microspheres/ml, vortex-mixed briefly and then drawn into a 1-ml syringe (BD Fine) fitted
220 with a 19-G needle (1.1 × 50 mm, BD fine, Franklin Lakes, NJ, USA). Finally, the contents
221 of the syringe were injected into the PTFE scaffold mould.

222

223 **2.10 Compressive strength of scaffolds**

224 The compressive strength of the scaffolds was assessed using a TA.HD+ texture
225 analyser (Stable Microsystems, UK) equipped with a 50-kg load cell at a speed of 0.04

226 mm/second over a contact area of approximately 28.75 mm². Dry PLGA and PLGA/chitosan
227 BSA-loaded scaffolds prepared as described in sections 2.5 and 2.9 were tested, and the
228 compressive strength was determined as the stress at the maximum strain.

229

230 **2.11 Cell culture and seeding onto scaffolds**

231 Primary hMSCs were cultured in hMSC basal media supplemented with the contents
232 of an MSCGM hMSC SingleQuote kit. The cells were maintained in a humidified tissue-
233 culture incubator at 37°C in 5% CO₂. The cytocompatibility test was carried out on BSA-free
234 scaffolds. Scaffolds were prepared directly into a 24-well plate in a manner similar to that
235 described in section 2.5. A 1:1 porous to non-porous microsphere mixture was UV sterilised
236 for 80 minutes (Gould et al., 2013) and then transferred to the well. Basal growth medium
237 was then added at a ratio of 0.25:1 (medium to microspheres). After 17 hours of sintering,
238 each scaffold was seeded with 1×10^5 hMSCs and incubated for 2 hours, followed by the
239 addition of 1 ml of media to each scaffold/well. The cell-seeded scaffolds were maintained at
240 37°C with 5% CO₂. For all cell experiments, either 5 replicates or 2 independent repeats each
241 comprising at least 3 replicates was carried out.

242 **2.12 Cell viability assay**

243 Each scaffold was submerged in 1 ml of media and 111 µl of Presto Blue reagent and
244 the cell viability was determined at day 1, 3, 6 and 8 post-seeding using the Presto Blue cell
245 viability reagent. The well plate was protected from light and incubated at 37°C for 25
246 minutes. Aliquots of 100 µl were withdrawn from each well in triplicate and the absorbance
247 was read on an infinite 200 plate reader (Tecan, Switzerland) at excitation and emission
248 wavelengths of 560 nm and 590 nm, respectively. The Presto Blue reagent was replaced with

249 fresh media and the scaffolds were placed back in the incubator. On day 8, after measuring
250 the cell viability, the scaffolds were washed with PBS and the cells were fixed with 10% v/v
251 buffered formalin solution for 20 minutes. Fixed hMSC-scaffold constructs were viewed
252 under the SEM.

253 **2.13 Assessment of mineralization**

254 In order to determine the degree of mineralization on the scaffolds, the von Kossa
255 assay was utilized. Cells were seeded onto scaffolds as described in section 2.11 and
256 incubated in basal growth media for 21 days. On day 21, cells were fixed with 10% v/v
257 buffered formalin for 20 minutes and thoroughly washed with PBS. A 450- μ l aliquot of 1%
258 w/v silver nitrate solution was added to each scaffold and incubated under a UV light source
259 for 1 hour. The solution was then removed and the scaffolds were washed three times with
260 deionized water. This was followed by treatment with sodium thiosulphate solution for 5
261 minutes in order to remove any excess silver nitrate solution. The scaffolds were then washed
262 with PBS prior to imaging under a dissection microscope (Leica, Germany).

263

264 **2.14 Osteocalcin immunostaining**

265 Cells were seeded onto scaffolds as described in section 2.11. The scaffolds were
266 incubated in basal growth media for 21 days after which they were fixed using 10% v/v
267 paraformaldehyde for 20 minutes and then thoroughly washed with PBS. The cells were
268 permeabilised with 500 μ l of 0.1% v/v Triton X-100 solution for 40 minutes. The solution
269 was aspirated and the cells were washed with PBS. Blocking of unspecific binding sites as a
270 result of epitomes on the cell layers was carried out via the addition of 500 μ l of 3% v/v goat
271 serum in 1% w/v BSA in PBS for 40 minutes. The blocking solution was removed and 500 μ l

272 of anti-OCN primary antibody solution (1:200 dilution in 1% w/v BSA in PBS) was added.
273 The scaffolds were incubated at 4°C overnight. After incubation, the antibody solution was
274 removed, the scaffolds were washed with PBS and then incubated at room temperature for
275 two hours in 500 µl of a 1:200 solution of Alexa Fluor 488 goat anti-rabbit secondary IgG, in
276 1% w/v BSA in PBS. After incubation, the secondary antibody solution was removed and the
277 scaffolds were washed with PBS. In order to stain the DNA of cells, the scaffolds were
278 incubated for a further 15 minutes in 1 µg/ml Hoechst dye dissolved in 1% w/v BSA in PBS
279 at room temperature. After incubation, the Hoechst dye was removed and the scaffolds were
280 thoroughly washed with PBS and then viewed under a dissection microscope. The images of
281 PLGA and composite scaffolds were processed and compared using the ImageJ software
282 (Version 1.48, National Institute of Health, Bethesda, MD, USA). Four images were taken of
283 four different areas on each scaffold and then converted into binary formats so that the
284 stained areas could be calculated.

285

286 **2.15 Statistical Analyses**

287 A statistical analysis of the data was carried out using Microsoft Excel. An unpaired t test and
288 the ANOVA procedure were used and the results were deemed significant when $p < 0.05$.

289 **3. Results**

290 **3.1 Physical characterization of PLGA/chitosan composite microspheres and scaffolds**

291 BSA-encapsulated PLGA/chitosan composite microspheres were formulated using
292 TPP as a cross-linker as detailed in section 2.2. Both the PLGA and composite microspheres
293 appeared smooth, as shown in the SEM images in Figure 1A and B, respectively. Thus, the

294 addition of chitosan cross-linked with TPP did not alter the superficial appearance of the
295 microspheres and no unprocessed, free chitosan is visible from the SEM images. Size
296 analysis revealed the average diameters of the PLGA and composite microspheres to be 69.75
297 $\pm 21.47 \mu\text{m}$ and $66.85 \pm 22.68 \mu\text{m}$, respectively (Figure 1C).

298
299 The FTIR spectra of the raw materials and microspheres are presented in Figure 2.

300 The chitosan spectrum shows a high-intensity peak at 3400 cm^{-1} , which corresponds to
301 stretching vibrations of the O-H and N-H bonds, in addition to hydrogen bonding in the
302 backbone (Azevedo et al., 2011). The characteristic peak at 1647 cm^{-1} is a result of the amide
303 functionality and may be present as a consequence of the axial deformation of the C=O bond
304 (Azevedo et al., 2011) and strong N-H bending (Misch et al., 1999). Peaks present at 1019
305 and 1086 cm^{-1} (corresponding to C-O stretch vibrations), and 1152 cm^{-1} (asymmetric stretch
306 of the C-O-C bond) are also indicative of chitosan (Azevedo et al., 2011).

307 The TPP spectrum, similarly, shows a peak of significant intensity at 3390 cm^{-1} ,
308 corresponding to the stretching vibrations of the O-H bond. Peaks around the 1095 cm^{-1}
309 region are an indication of the P=O phosphate group. The PLGA spectrum presents a peak at
310 3473 cm^{-1} , which is indicative of vibration of the terminal O-H groups. Other peaks that
311 indicate PLGA are present at 743 cm^{-1} (C-H bend), 1086 and 1180 cm^{-1} (C-O stretch), 1381
312 cm^{-1} (C-H bend), 1771 cm^{-1} (the carbonyl C=O) and 2876 cm^{-1} (CH_2 bend) (Ganji and
313 Abdekhodaie, 2010). Both PLGA and PLGA/chitosan composite BSA-loaded microspheres
314 show peaks at identical wavelengths, which suggests that the microspheres are predominantly
315 PLGA. Moreover, the spectra of PLGA and PLGA/chitosan composite microspheres show
316 peaks at 1621 cm^{-1} and 1639 cm^{-1} , respectively, which are attributed to the C=O bond of the
317 amide groups that are found both in BSA and chitosan. However, there does appear to be a
318 slightly more pronounced peak at 1639 cm^{-1} on the spectrum of the PLGA/chitosan

319 composite microspheres, which corresponds to the amide C=O bond suggesting the presence
320 of chitosan in the formulation.

321 The ToF-SIMS analysis was carried out in order to ascertain the presence of chitosan
322 on the scaffold surfaces. BSA-free scaffolds were analysed based on the overlap of chitosan
323 and BSA secondary ion peaks (discussed in section 2.6). Intensities of nitrogen-containing
324 positive secondary ion peaks CH_4N^+ ($m/z = 30$) and $\text{C}_4\text{H}_5\text{N}_2^+$ ($m/z = 81$), as well as the
325 negative ion peak CN^- ($m/z = 26$) were all significantly higher in the composite
326 PLGA/chitosan scaffolds when compared to the chitosan-free scaffolds, as shown in Figure
327 3A. However, there was no significant difference between the profiles of diagnostic PLGA
328 ion peaks for the PLGA and composite scaffolds (Figure 3B).

329 The incorporation of chitosan did not elicit a significant change in the encapsulation
330 efficiency of BSA in the microspheres, with $80.58 \pm 17.06\%$ and $81.57 \pm 3.06\%$ of the
331 protein being encapsulated into the PLGA and PLGA/chitosan composite microspheres,
332 respectively. Moreover, there was no statistical difference in the encapsulation efficiencies of
333 the PLGA and composite scaffolds (2.81 mg/g [$93.68\% \pm 3.50\%$] and 2.52 mg/g [$84.02\% \pm$
334 12.08%] for the PLGA and PLGA/chitosan composite scaffolds, respectively).

335 **3.2 Release of BSA from microspheres and scaffolds**

336 The release profile of BSA was mapped over 28 days from both microspheres and
337 scaffolds sintered at 37°C (Figure 4). The initial burst release after 24 hours from the PLGA
338 microspheres was significantly higher than from the PLGA/chitosan composite microspheres,
339 $0.93 \pm 0.06 \mu\text{g/mg}$ and $0.57 \pm 0.03 \mu\text{g/mg}$, respectively ($p < 0.05$). After 28 days, 1.72 ± 0.23
340 $\mu\text{g/mg}$ of BSA was released from the PLGA microspheres, which was ~~significantly~~ higher in
341 comparison to $1.20 \pm 0.05 \mu\text{g/mg}$ from the PLGA/chitosan composite microspheres ($p = 0.05$)

342 Similarly, there was a significant retardation of the initial burst release from the
343 scaffolds containing PLGA/chitosan composite microspheres, $0.10 \pm 0.02 \mu\text{g}/\text{mg}$, in
344 comparison to the PLGA scaffolds, $0.16 \pm 0.01 \mu\text{g}/\text{mg}$ ($p < 0.05$, Figure 4B).

345

346 **3.3 Sintering of microspheres into scaffolds**

347 In order to study the effect of the scaffold preparation method on their subsequent
348 morphology and mechanical strength, the PLGA and PLGA/chitosan composite scaffolds
349 were prepared using two different methods. Firstly, a paste was formed from the
350 microspheres as previously reported (Boukari et al., 2015). In the second method, we aimed
351 to study the ability of the microspheres to sinter post-injection through a 19-G needle into a
352 scaffold mould. This was then followed by a 17-hour incubation period at 37°C . Photographs
353 of the resulting scaffolds and their compressive strengths are presented in Figure 5A and B,
354 respectively. The sintering process results in the expulsion of water so that the components
355 within close proximity. We believe that this favours ‘fusion’ and bond formation within the
356 scaffolds. This approach to scaffold sintering at 37°C is superior to the more harsh methods
357 employing elevated temperatures and reagents.

358 The overall appearances of PLGA and composite scaffolds were very similar (Figure
359 5A). However, when comparing scaffolds prepared using the paste method, the compressive
360 strength of PLGA/chitosan composite scaffolds was significantly higher ($0.846 \pm 0.272 \text{ MPa}$)
361 than the PLGA scaffolds ($0.406 \pm 0.265 \text{ MPa}$, $p < 0.05$).

362 Figure 5A shows that it was possible to successfully sinter a microsphere suspension
363 post-injection, thus, forming intact scaffolds **that retained their shape when removed from the**
364 **mould**. This confirms the injectable potential of the microspheres. When scaffolds were

365 sintered as a suspension post-injection, there was no significant difference between the
366 compressive strengths of the PLGA and PLGA/chitosan composite scaffolds, 0.086 ± 0.068
367 MPa and 0.048 ± 0.00096 MPa, respectively ($p > 0.05$, Figure 5B); however, it is likely that
368 the compressive values may be below the lower limit of threshold of the machine.

369

370 **3.4 Cell proliferation on scaffolds**

371 The culturing of primary hMSCs on the scaffolds was used as a means to test their
372 cytocompatibility. Cell proliferation was assessed using the Presto Blue viability reagent on
373 day 1, 3, 6 and 8 (Figure 6A).

374 Cell proliferation increased over time on both scaffold types. On day 1, the cell
375 numbers on PLGA and PLGA/chitosan composite scaffolds were 1.06×10^4 and 1.03×10^4 ,
376 respectively. Both types of scaffolds exhibited a very similar cell growth profile with no
377 statistically significant difference found between them ($p > 0.05$) on day 1, 3 and 6. However,
378 the cell number on day 8 was significantly higher on the PLGA scaffolds ($p < 0.05$) at $6.25 \times$
379 10^4 and 4.45×10^4 for PLGA and PLGA/chitosan composite scaffolds, respectively. SEM
380 images of the cell-scaffold constructs on day 8 are shown in Figure 6B and C, with cells
381 visibly distributed between microspheres in both scaffold types.

382

383 **3.5 Assessment of mineralization**

384 The extent of mineralization on the scaffolds after 21 days in culture media was
385 assessed using the von Kossa assay as described in section 2.14. Dark brown/black nodules
386 (indicated by the white arrow in Figure 7B) are visible on the scaffolds and represent positive
387 staining. A qualitative analysis shows that there are more nodules on the PLGA/chitosan

388 composite scaffolds (Figure 7B), which appear darker in the figure, in comparison to the
389 PLGA scaffolds (Figure 7A).

390 **3.6 Osteocalcin immunostaining**

391 The presence of the bone marker protein, osteocalcin, was detected using the
392 immunostaining technique described in section 2.14. The data obtained was processed using
393 ImageJ, which allowed us to quantify the amount of stain present on each scaffold. The
394 results of this analysis show that there was an increase in osteocalcin staining on the
395 composite scaffolds when compared to the PLGA scaffolds ($p < 0.05$, Figure 8A). When
396 osteogenic media was used (data not shown), the osteocalcin staining on the PLGA/chitosan
397 and PLGA scaffolds was not significantly different ($p > 0.05$). Processed, merged images are
398 shown in Figure 8B and C, with osteocalcin represented in green, and cell DNA in blue.

399 **4. Discussion**

400 Scaffolds made from biodegradable microspheres are a promising approach for bone
401 regeneration. However, there are several features to consider when developing such systems.
402 These include the incorporation of porosity and growth factors into the scaffolds, whilst at the
403 same time providing mechanical strength to enable the microspheres to be injectable and
404 sinter *in situ*. Some research groups have developed scaffolds with some of these properties;
405 however, most groups do not take into account all desirable features in one system. In the
406 present study, we propose a novel dual-application PLGA/chitosan composite scaffold
407 system with the potential to meet all of the above desirable criteria. The system comprises
408 porous and non-porous protein-loaded microspheres with the ability to sinter at 37°C and
409 release protein. The mechanical strength of the system is dependent upon its mode of
410 application, with a higher compressive strength achievable when it is applied as a paste, and
411 sufficient strength to maintain the shape (as evident from the fact that the microspheres

412 sintered at 37°C and were subsequently removed from the mould intact) when injected as a
413 suspension. The cytocompatibility and osteogenic potential of the formulation were
414 evaluated and compared with our previously reported system (Boukari et al., 2015).

415 Protein-loaded microspheres were formulated using PLGA and chitosan, where the
416 chitosan was cross-linked using TPP. There were no observable differences in the
417 morphology and size of the composite microspheres when compared with PLGA
418 microspheres. The presence of chitosan within the composite scaffolds formed via the paste
419 method was confirmed by ToF-SIMS, suggests that chitosan is formed as part of the
420 microstructure of the particles. Furthermore, the composite scaffolds demonstrated higher
421 compressive strength than the PLGA scaffolds. In this regard, chitosan contributes to the
422 mechanical strength of the scaffolds, due to interactions between the negatively charged
423 PLGA (Balmert et al., 2015) and the protonated amine groups in the chitosan structure.
424 Moreover, the compressive strength demonstrated by the composite scaffolds fell within an
425 acceptable range as reported by Misch et al. (1999).

426 The chitosan coating attenuated the initial burst release from the microspheres and
427 scaffolds, and this reduction may partly be attributed to chitosan complexing with BSA
428 (isoelectric point, approximately 5), thereby, impeding its release. The ability of chitosan, a
429 natural polyelectrolyte, to non-covalently bind to negatively charged proteins has been
430 reported (Boeris et al., 2010). A similar observation of a reduced burst release was made for
431 the same system when encapsulated with lysozyme, which is positively charged at a neutral
432 pH (data not shown). This suggests that other factors contribute to the reduction in burst
433 release. It has been reported that the burst release of proteins from PLGA microspheres is
434 usually due to protein residing near, or on the surface of, the delivery system (Zeng and
435 Liang, 2010). We believe that the formation of a chitosan -TPP matrix layer slows the release
436 of the protein and significantly contributes to the attenuation in the initial burst release. This

437 effect has been demonstrated in PLGA/chitosan microspheres encapsulated with a non-
438 protein drug, rifampicin, in which the addition of chitosan caused a reduction in the burst
439 release (Manca et al., 2008). The slower, steadier release of BSA from the microspheres and
440 scaffolds containing PLGA/chitosan is desirable in BMP-2 applications. The controlled
441 release reduces the need for supra-physiological loadings, which are necessary when there is
442 a huge initial loss via a burst release (Kirby et al., 2011).

443 The system described herein possesses dual-applicability arising from the
444 formulation's potential of having two application modes (i) a paste that is implanted within a
445 degenerated bone tissue, takes the shape of the defect area and then sinters at 37°C, and (ii)
446 the injection of the microsphere suspension directly into the defect area. The former would be
447 useful in applications requiring a relatively stronger scaffold, such as the regeneration of
448 cancellous bone for which the ultimate compressive strength has been reported to range from
449 0.22 to 10.44 MPa (Misch et al., 1999). However, the latter is more suited to applications in
450 which the delivery system may be injected and remain in one location, hence, allowing the
451 controlled delivery of a specific, known dose of protein to the site. To our knowledge, this is
452 the first time that the ability of microspheres to sinter at 37°C, post-injection, has been
453 demonstrated.

454 The ability of cells to attach and grow on the scaffolds is paramount in the
455 development of protein delivery systems in regenerative medicine. For this reason, the
456 cytocompatibility of the scaffolds with hMSCs was investigated. The cell number increased
457 on the composite scaffolds over the 8-day period from 1.03×10^4 on day 1, to 4.45×10^4 on
458 day 8. There was no significant difference between cell numbers on the composite and PLGA
459 scaffolds, except on day 8, by which time the cell numbers were higher on PLGA scaffolds (p
460 < 0.05). Although previous studies have investigated the cytocompatibility of sintered

461 composite PLGA/chitosan microspheres scaffolds with other cell types, these formulations
462 were not capable of sintering *in situ* (Tao et al., 2014).

463 The potential of the scaffold material to promote the differentiation of hMSCs is
464 another key factor that is crucial for the production of a successful biomaterial. Although the
465 presence of BMP-2 has been shown to promote osteogenic differentiation, the intrinsic ability
466 of the material itself to promote the process is also of interest. Chitosan has been reported to
467 have numerous biomedical properties, including its ability to improve osteogenesis in animal
468 bone defect models (Lee et al., 2008). In this study, we investigated the cell response to
469 protein-free scaffolds in basal media in order to study the effect of the scaffold material on
470 osteogenesis. The presence of a calcified ECM is a reliable way of confirming osteogenesis
471 (Declercq et al., 2005). Nodules were observed on both composite and PLGA scaffolds based
472 on von Kossa staining, which indicates the presence of calcium. To provide further
473 confirmation of the deposition of a calcified matrix, the presence of osteocalcin, a late protein
474 marker of osteogenic differentiation, was determined. Its expression is known to rise with an
475 increase in mineralization (Stein et al., (1990). The composite scaffolds showed a
476 significantly higher degree of osteocalcin staining when compared to the PLGA scaffolds.
477 Previous studies have demonstrated the ability of chitosan-containing scaffolds to induce
478 differentiation in the presence of osteogenic media (Jiang et al., 2006), which we also
479 confirmed (data not shown). However, relatively little evidence has demonstrated this in
480 basal growth media. Therefore, these results suggest that the inclusion of chitosan in PLGA
481 microspheres enhanced the osteogenic capacity of the resultant scaffolds.

482 **5. Conclusion**

483 In this study, a novel, dual-application composite microsphere system was developed
484 with the ability to fuse together as a paste, thereby forming an intact scaffold in the body at

485 37°C. Furthermore, the ability of a suspension of the microspheres to sinter post-injection
486 was also demonstrated. Composite PLGA/chitosan microspheres were shown to attenuate the
487 initial burst release and elicited a steady, slow release of protein over 28 days. The scaffold's
488 cytocompatibility and ability to promote osteogenesis were also demonstrated. This
489 technology, therefore, exhibits potential as a scaffold for bone regeneration and is an
490 excellent candidate for further *in vitro* and *in vivo* testing.

491

492 **Disclosures**

493 There are no potential conflicts of interest to disclose for this work.

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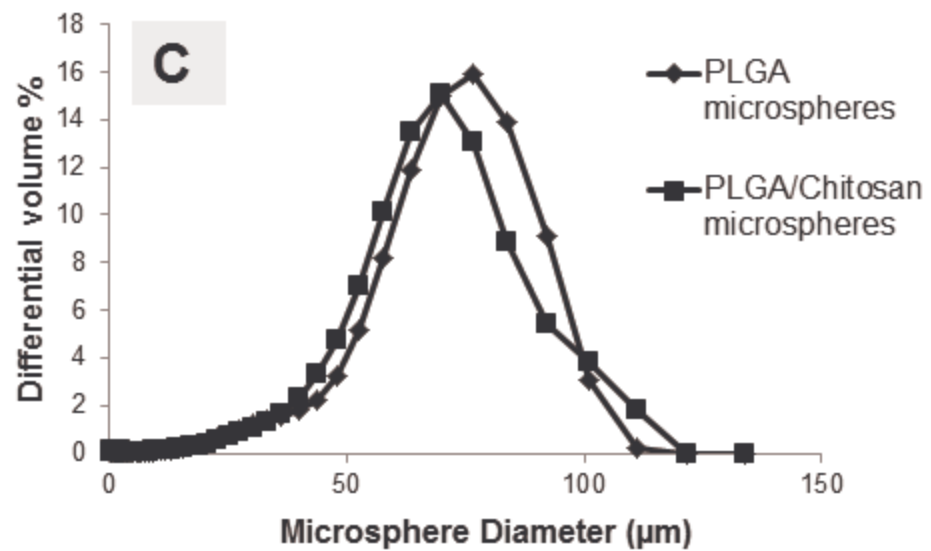
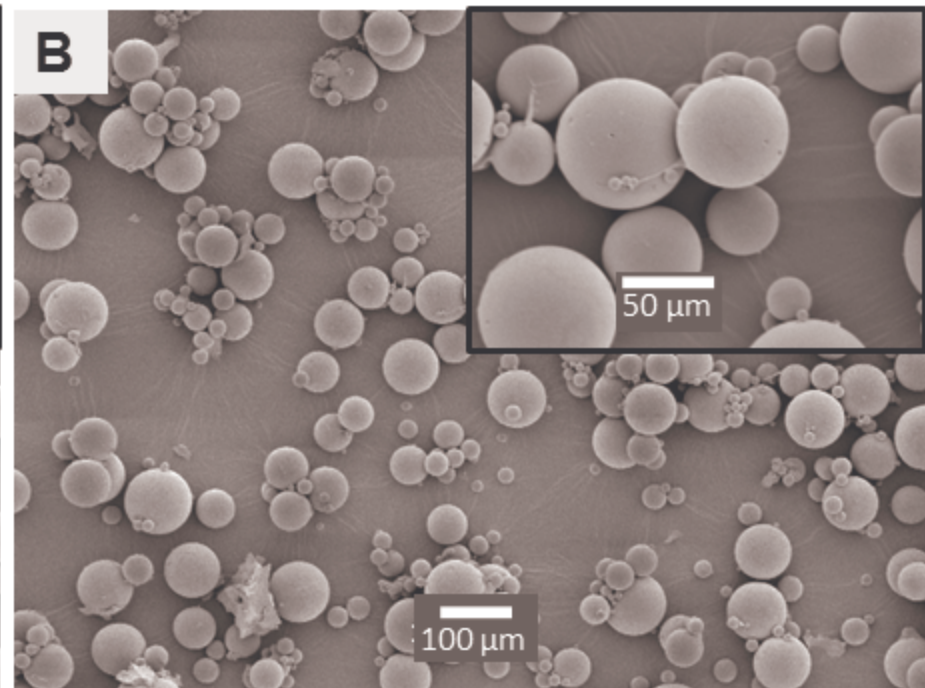
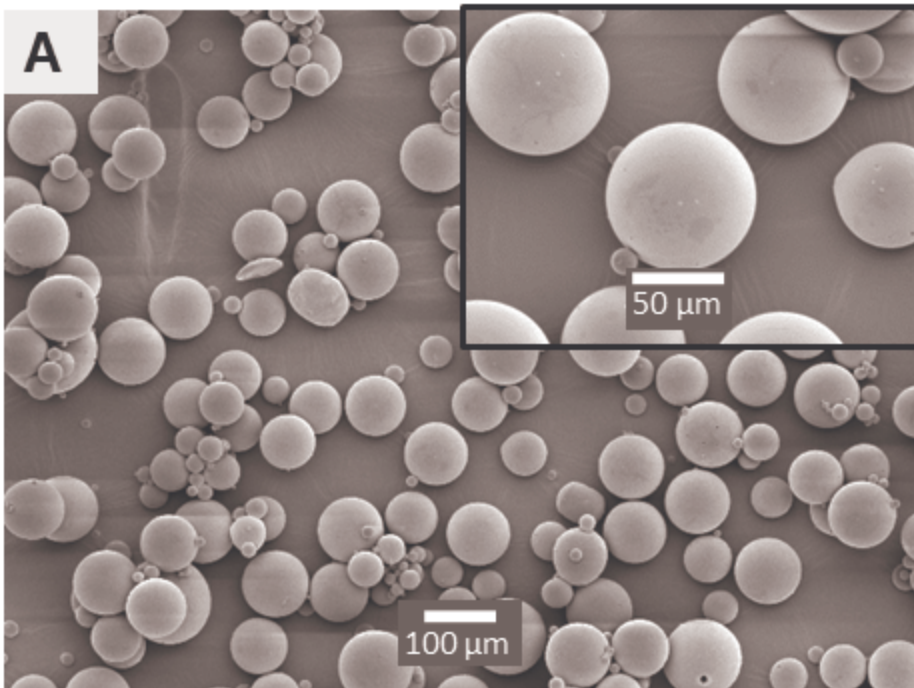
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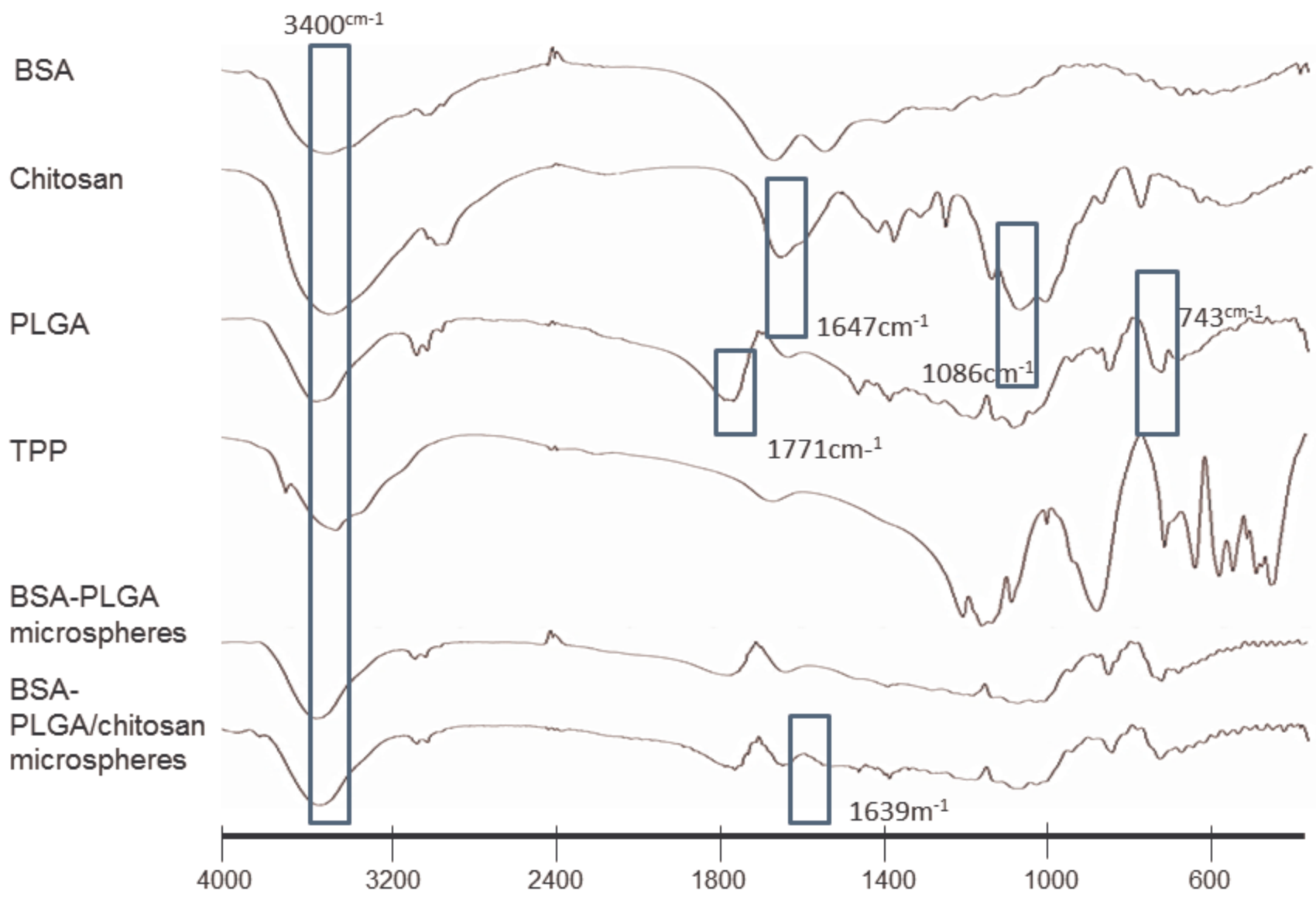
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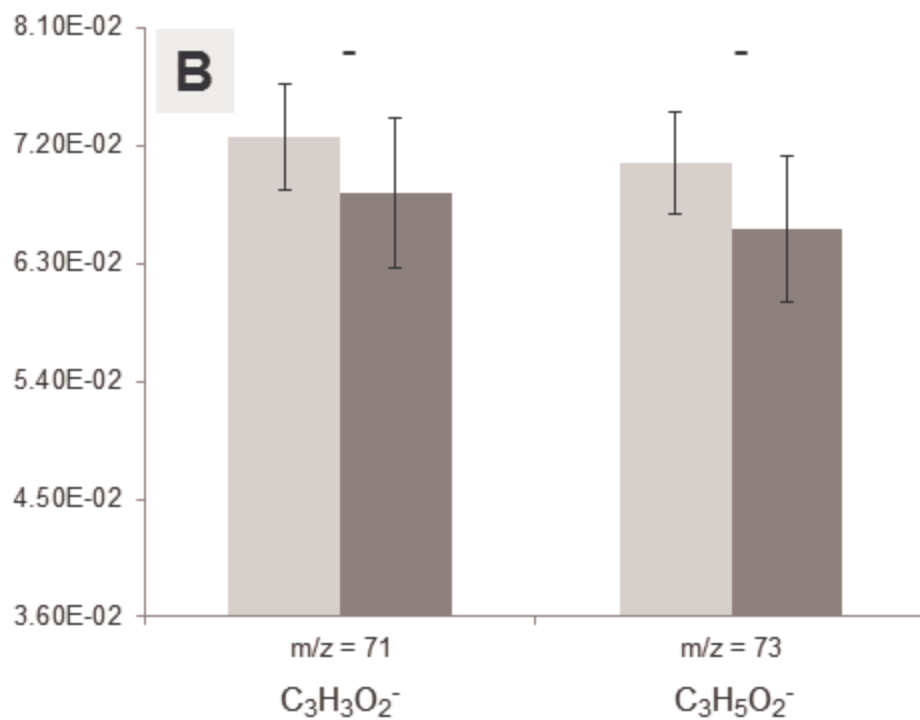
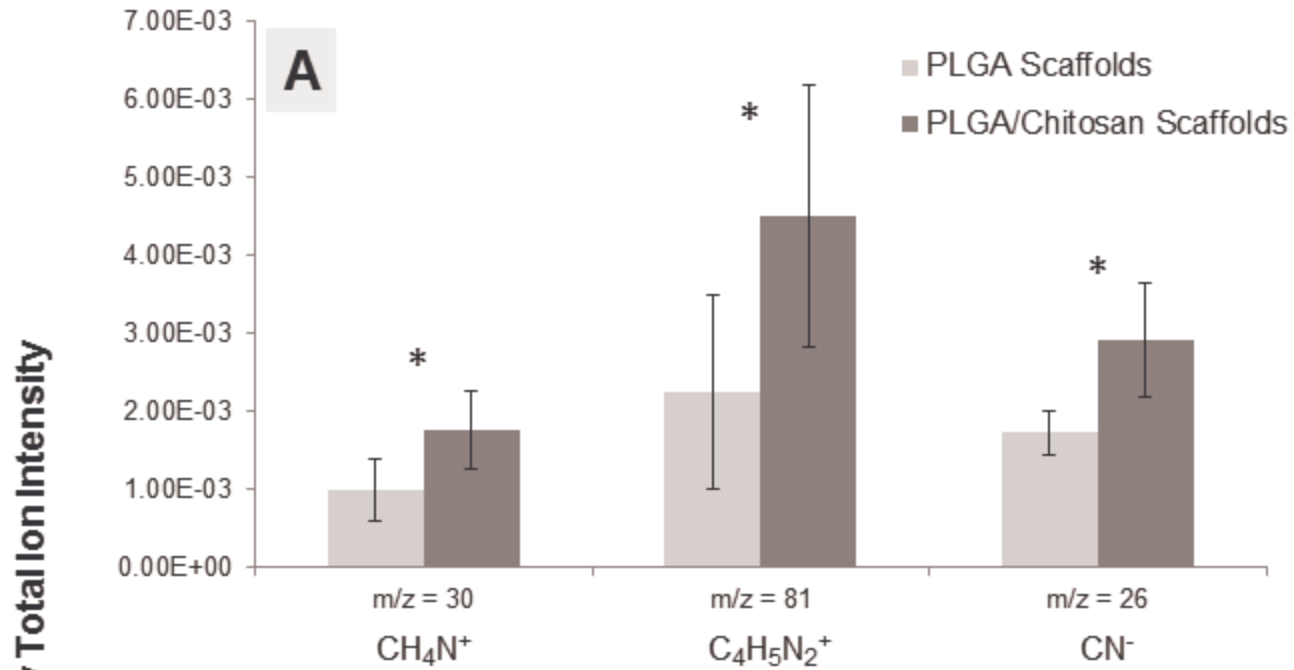
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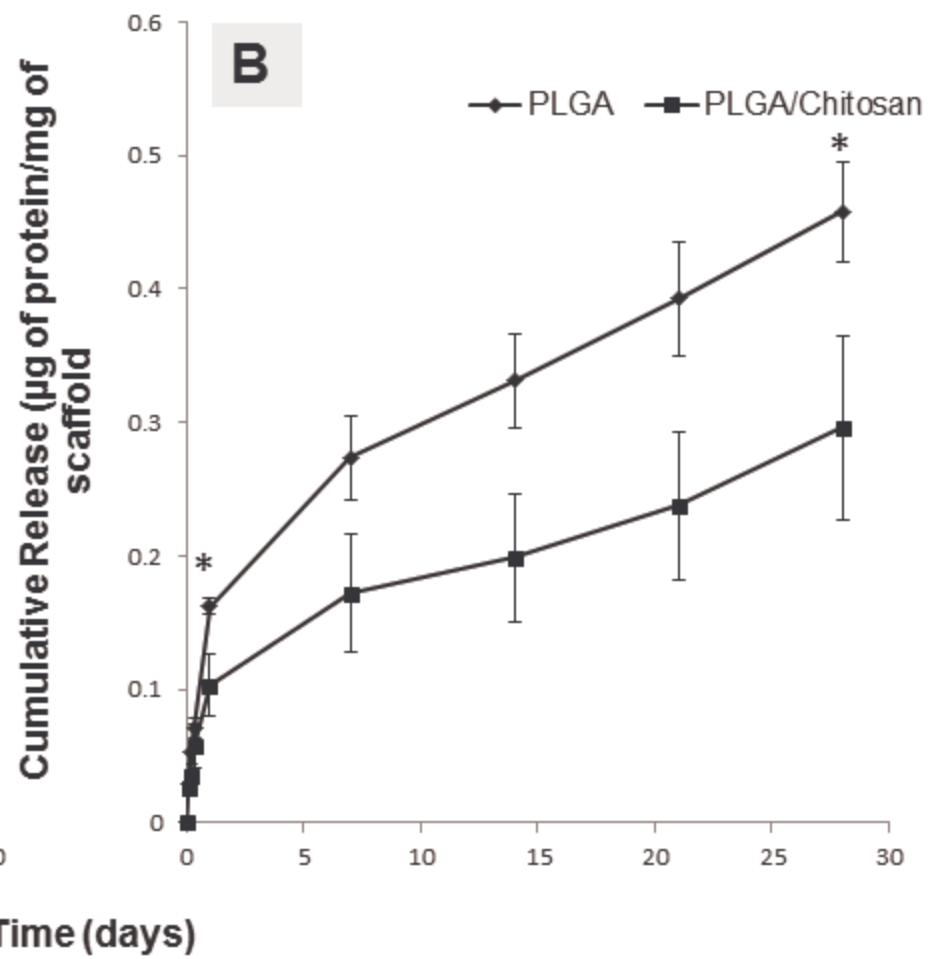
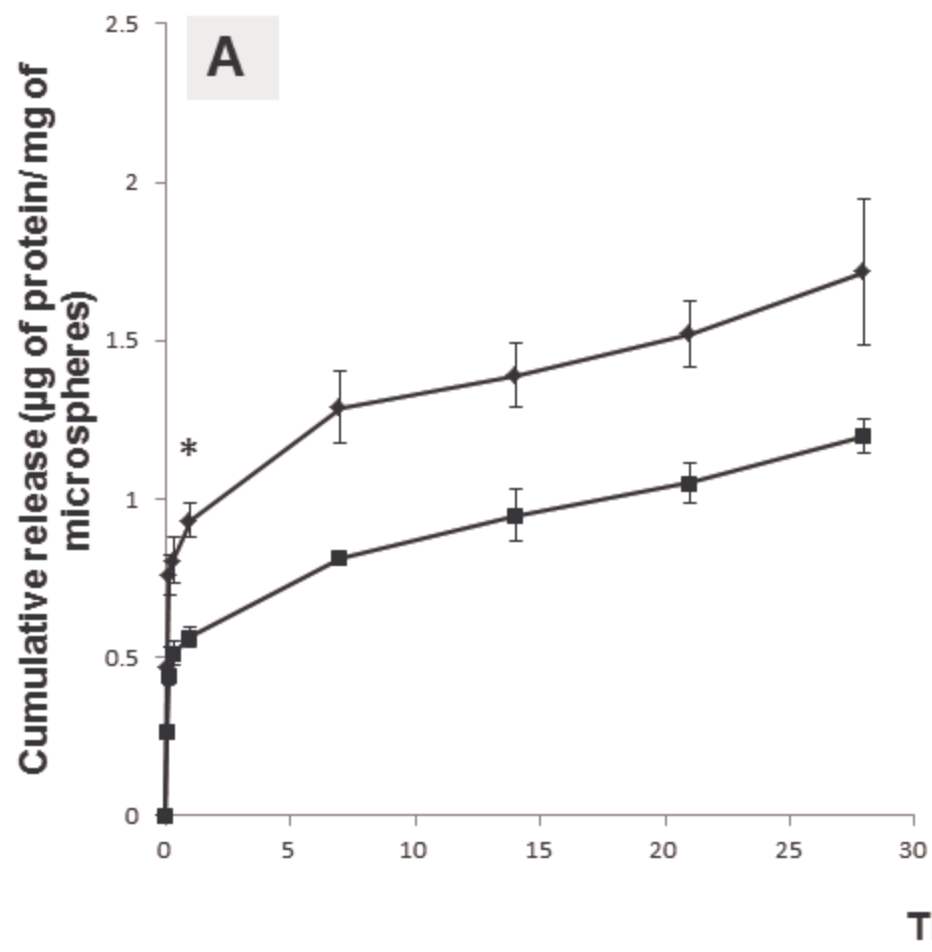
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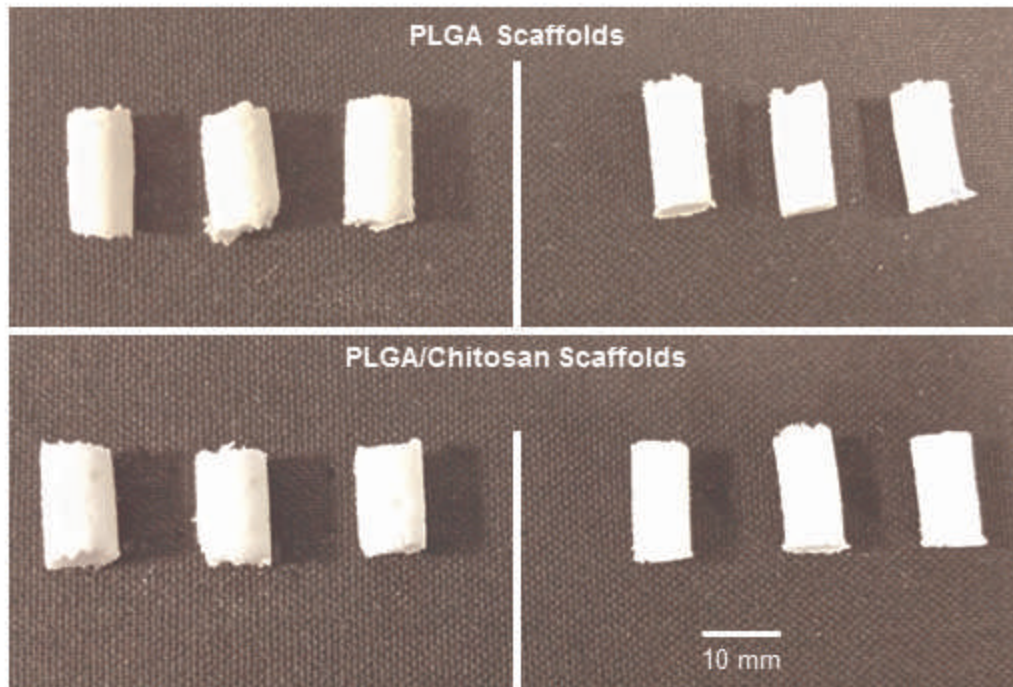
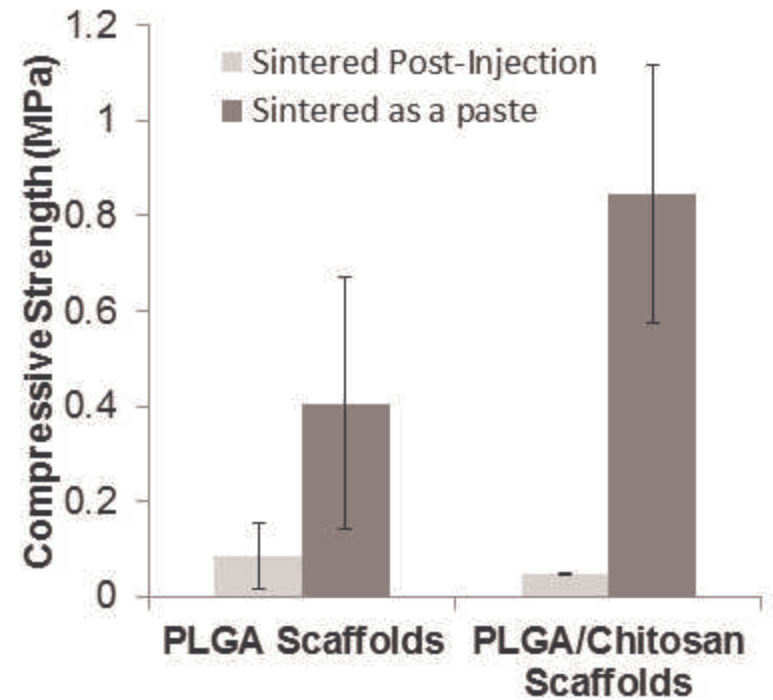
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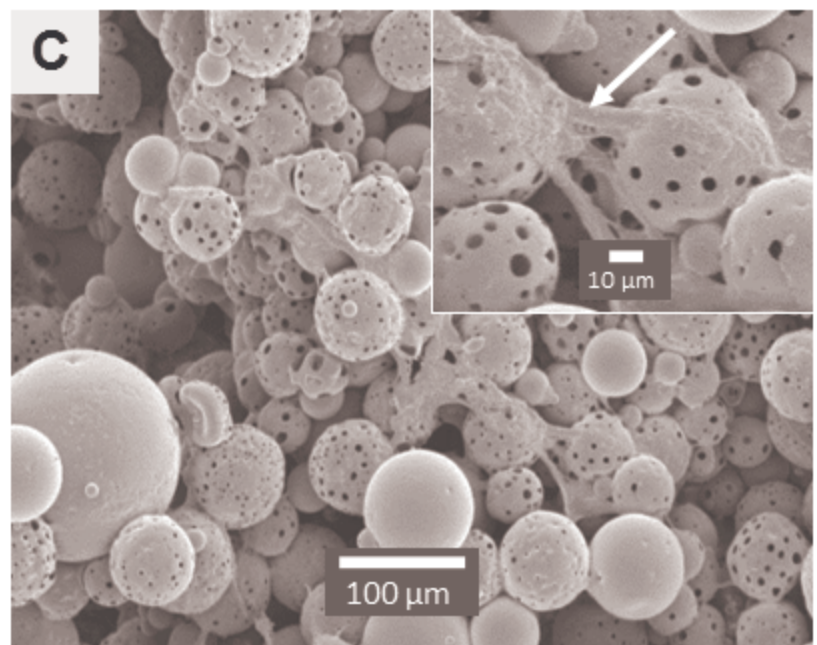
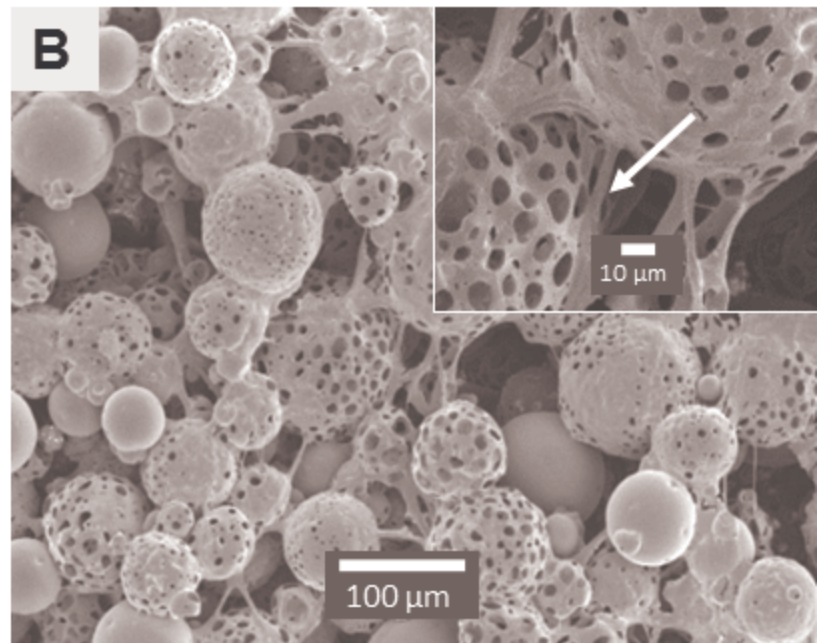
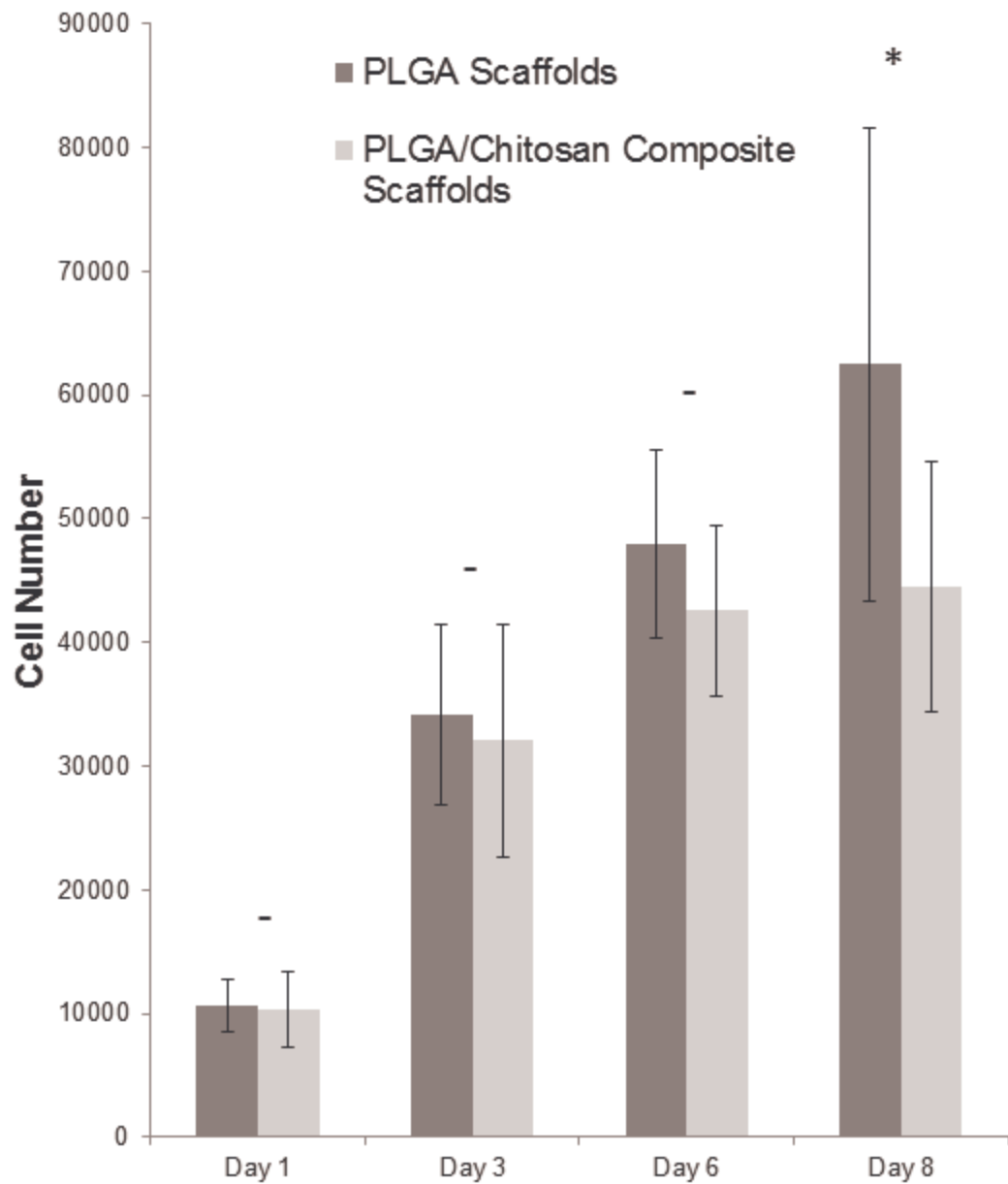


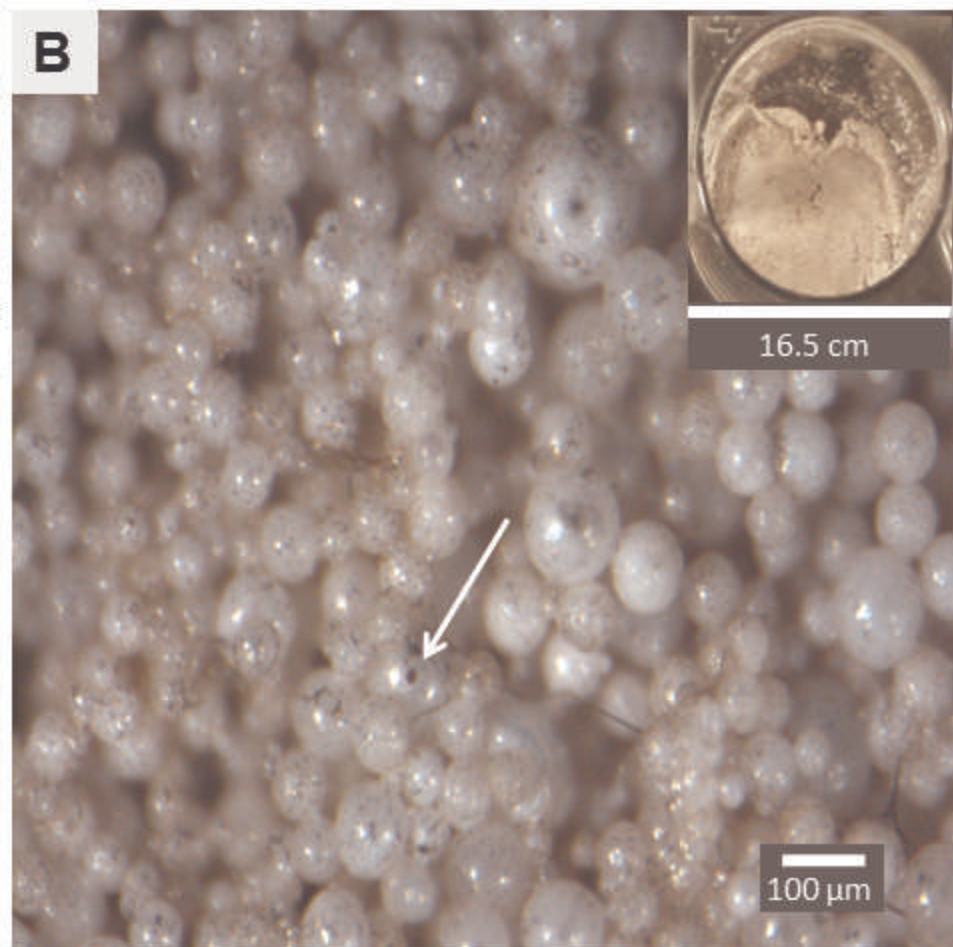
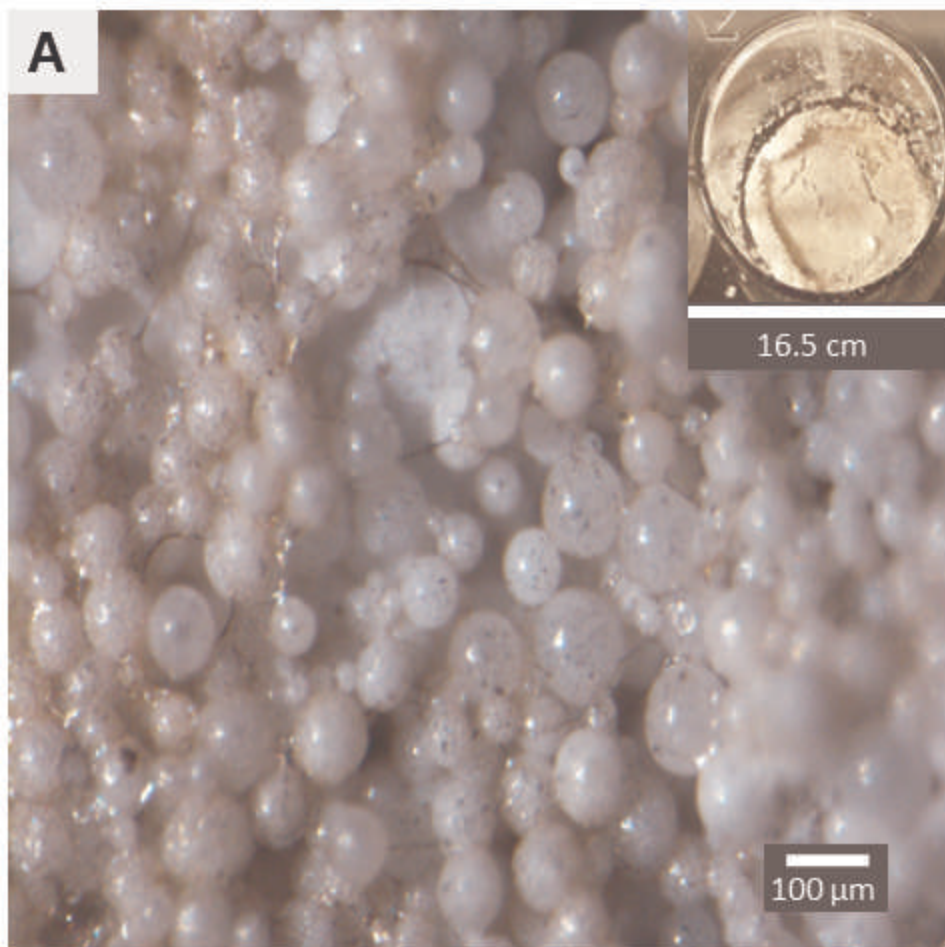


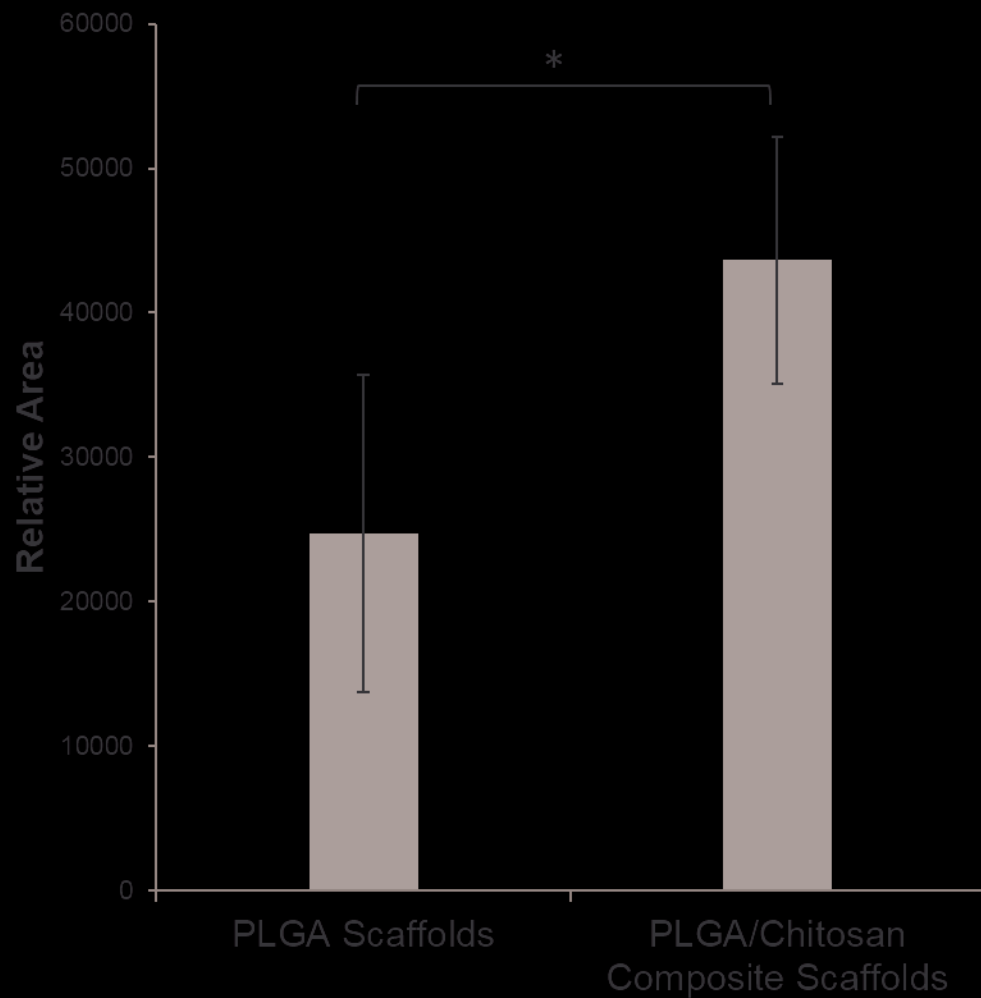
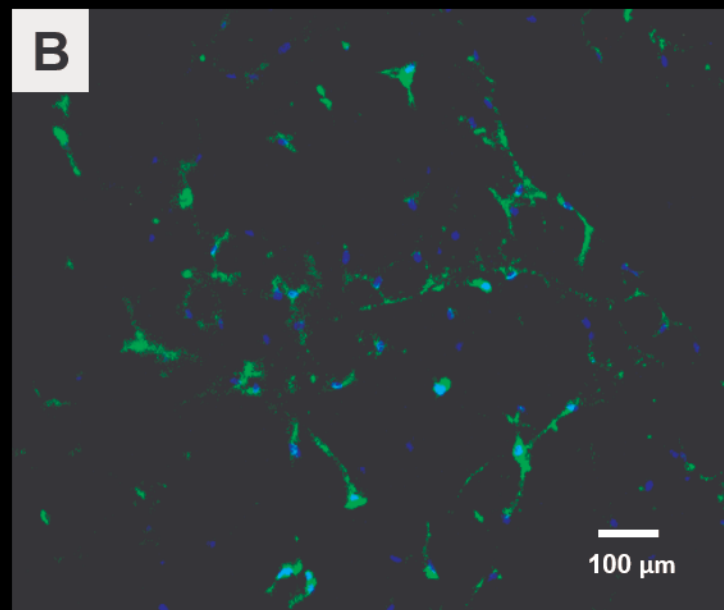




A**Sintered as a suspension
post-injection****Sintered as a paste****B**

A



A**B****C**