

Hall Effect and transient surface photovoltage (SPV) study of Cu_3BiS_3 thin films

F. Mesa¹✉, A. Dussan², B. A. Paez-Sierra³, H. Rodriguez-Hernandez³

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Abstract

Here, we present the electrical properties of the compound Cu_3BiS_3 deposited by co-evaporation. This new compound may have the properties necessary to be used as an absorbent layer in solar cells. The samples were characterized by Hall effect and transient surface photovoltage (SPV) measurements. Using Hall effect measurements, we found that the concentration of n charge carriers is in the order of 10^{16} cm^{-3} irrespective of the Cu/Bi mass ratio. We also found that the mobility of this compound (μ in the order of $4 \text{ cm}^2\text{V}^{-1}\text{s}^{-1}$) varies according to the transport mechanisms that govern it and are dependent on temperature. Based on the SPV, we found a high density of surface defects, which can be passivated by superimposing a buffer layer over the Cu_3BiS_3 compound.

Keywords: Electric transport; Cu_3BiS_3 ; transient surface photovoltage; defects.

Edited by Beynor Paez✉ & Alberto Acosta

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Received: 22-11-2013 Accepted: 25-03-2014

Published on line: 10-04-2014

Citation: Mesa F, Dussan A, Paez-Sierra BA, Rodriguez-Hernandez H (2014) Efecto hall y estudio de fotovoltaje superficial transiente (SPV) en películas delgadas de Cu_3BiS_3 . *Universitas Scientiarum* 19(2): 107-113 doi: 10.11144/Javeriana.SC19-2.ehef

Funding: Pontificia Universidad Javeriana; Banco de la República; Fundación para la Promoción de la Investigación y la Tecnología (FPIT).

Electronic supplementary material: N/A

Introduction

I-III-VI₂ chalcopyrite compounds are direct gap semiconductors and have adequate properties for use in the fabrication of single-junction cells or tandem cells (Repins et al. 2008, Kayes et al. 2011, Choi et al. 2012, Seo et al. 2012). Nevertheless, the maximum conversion efficiency reported up to date for single-junction thin film technology solar cells was obtained by fabricating the device using Cu-III-VI₂ group compounds as the absorbing layer (Contreras et al. 2009, Green et al. 2012). These solar modules which are based on a thin film of Cu (InGa)Se₂ (CIGS) offer greater stability and price competitiveness for production processes. By contrast, Cu-V-VI₂ based compounds are characterized by having a forbidden bandgap that varies between



values from 1.15 to 2.73 eV, which makes this kind of compounds potentially useful and attractive candidates for photovoltaic applications (Bar et al. 2008, Kawamura et al. 2009, Meiss et al. 2011, Mönig et al. 2011, Tadjarodi et al. 2012).

For these reasons, Cu_3BiS_3 compound has taken great importance in optoelectronic applications. This compound grows in a “wittichenite” type orthorhombic crystalline lattice, where this phase is obtained at 300°C temperatures (Naiding 1994, Razmara et al. 1997). Several techniques have been used for the preparation of Cu_3BiS_3 samples, such as evaporation (Mesa et al. 2009, Colombara et al. 2012), via PEG (polyethylene glycol) (Yan et al. 2012), reactive sputtering (Gerein et al. 2006), and CBD (Chemical Bath Deposition) (Estrella et al. 2003). Nevertheless, it is reported that with the use of CBD there is growth of a metallic Bi thin film which is deposited by thermal evaporation on CuS obtained by CBD. As a result of this study the layers presented p-type conductivity ($\sim 0.03 \Omega^{-1}cm^{-1}$) and an energy gap E_{g-} of 1.28 eV. These are photoconductive, and the product of the mobility by the life time of the photo-generated carriers is $\sim 10^{-6} cm^2V^{-1}$. In work carried out by N. Gerein and J. Haber (Gerein et al. 2006 a,b) it is reported that layers deposited by reactive sputtering in one or two phases, and then annealed in an H_2S atmosphere present electrical resistivities in the range between 3-200 Ωcm , a forbidden band gap (E_g) of 1.4 eV and an absorption coefficient of $1 \times 10^5 cm^{-1}$.

In the present work, Cu_3BiS_3 samples were deposited by evaporation processes, varying the mass ratios $m_{Cu}/(m_{Cu}+m_{Bi})$. Emphasis was made on studying the effect of the Cu/Bi ratio on the concentration of mobile charge carriers. From transient surface photovoltage (SPV) measurements, the effect of the addition of a buffer layer on the compound was also observed.

Materials and Methods

Samples of Cu_3BiS_3 were deposited by the method of evaporation of precursors varying the mass ratios $m_{Cu}/(m_{Cu}+m_{Bi})$ between 0.43 to 0.49. Additionally, Hall voltage measurements were carried out on Cu_3BiS_3 samples, using Al as electrical contact. The Hall voltage measurement system is composed of:

Chamber with prevacuum system, sample heating and cooling system, and Hall voltage measurement system, which in turn is composed of model 7065 Hall effect card and a Keithley model 7001 Switch System. The Hall voltage measurement system is described in detail in the reference (Gutiérrez et al. 2007). The concentration of mobile charge carriers is calculated from the Hall voltage measurements by following an experimental procedure developed by the Keithley firm, which is based on the Van der Pauw method and the ASTM (American Society for Testing and Materials 1991) F 76 standard.

Surface photovoltage (SPV) measurements were taken using a capacitor composed of a Cr semitransparent electrode (transparency around 70% and diameter $d=5mm$) evaporated on a quartz crystal; a mica (width $d=50\mu m$) as dielectric and the sample deposited over a conducting substrate (reference electrode). The photovoltage signal is recorded through a lock-in amplifier taking the drop in voltage over the charge resistance R. A pulsed laser NT343/1/UV (EKSPLA) was used, where the pulse duration was around 5 ns and a maximum intensity of $0.5 mJ/cm^2$ (Fu et al. 2014).

Results

HRTEM images of medium and high magnification measured in a specific area of Cu_3BiS_3 thin films, show that this compound grows in a nanocrystalline lattice, with grains of different size, whose average value is in the vicinity of 15 nm (Mesa et al. 2012). On the other hand, due to the nanocrystalline feature of the samples, it is hard to perform the corresponding indexation. Nevertheless, the diffraction pattern of the selected area is shown in the image (**Figure 1**).

With the aim of deepening in the study of electrical transport properties of Cu_3BiS_3 thin films, the p concentration and the μ mobility of charge carriers was determined from measurements of electrical conductivity and Hall coefficient as a function of temperature. The p vs T and μ vs T curves for Cu_3BiS_3 thin films deposited with varying $m_{Cu}/(m_{Cu}+m_{Bi})$ ratio are analyzed (**Figure 2**).

Additionally, transient SPV curves were obtained with Cu_3BiS_3 thin films and with Cu_3BiS_3/In_2S_3

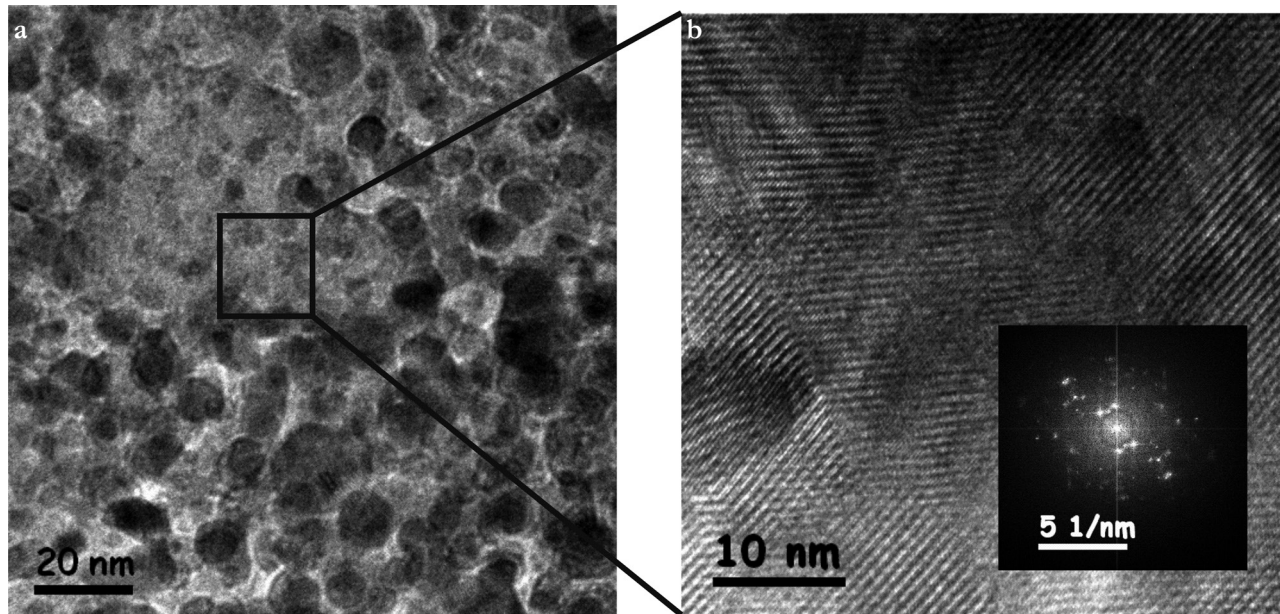


Fig. 1. a. Medium and high magnification HRTEM images, obtained with Cu_3BiS_3 thin film and b. Diffraction pattern obtained at the selected area.

system, using photoexcitation at different wavelengths (**Figure 3**). From these results it is observed that the transient SPV signal, obtained with Cu_3BiS_3 films excited with wavelengths greater than the gap

(<890 nm) is positive, with a decay time constant of $\sim 0.1 \mu\text{s}$. Nevertheless, when the sample is excited with wavelengths smaller than the energy gap (<890 nm), the SPV signal decay has negative values.

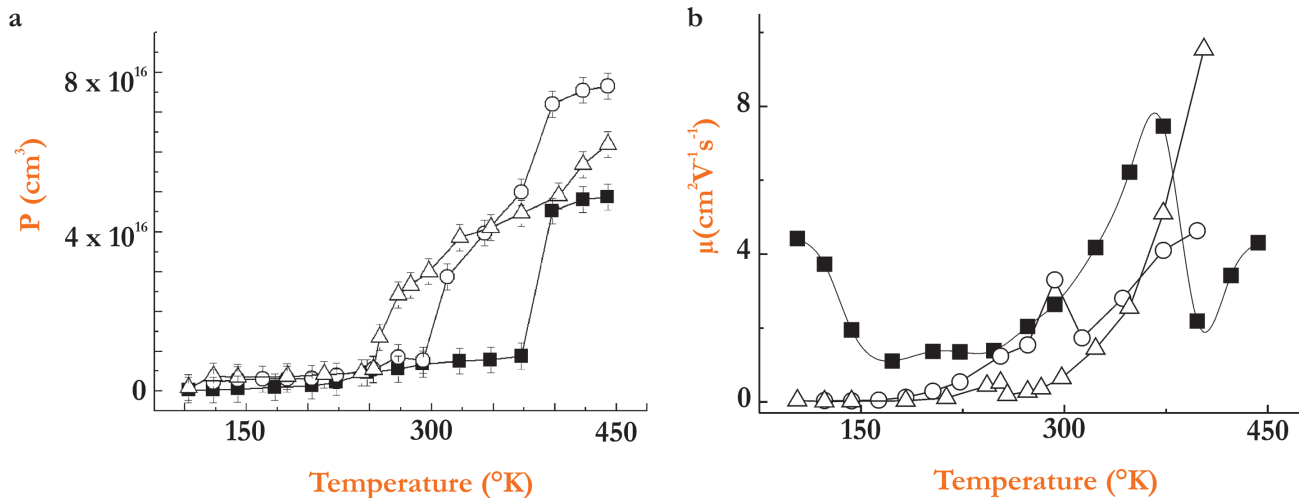


Fig. 2. a. Variation in the concentration p of holes and b. The Hall mobility in Cu_3BiS_3 samples with mass ratio $m_{\text{Cu}}/(m_{\text{Cu}}+m_{\text{Bi}})$ represented by (■- 0.43), (○- 0.46) y (△- 0.49), respectively, with a 3% dispersion.

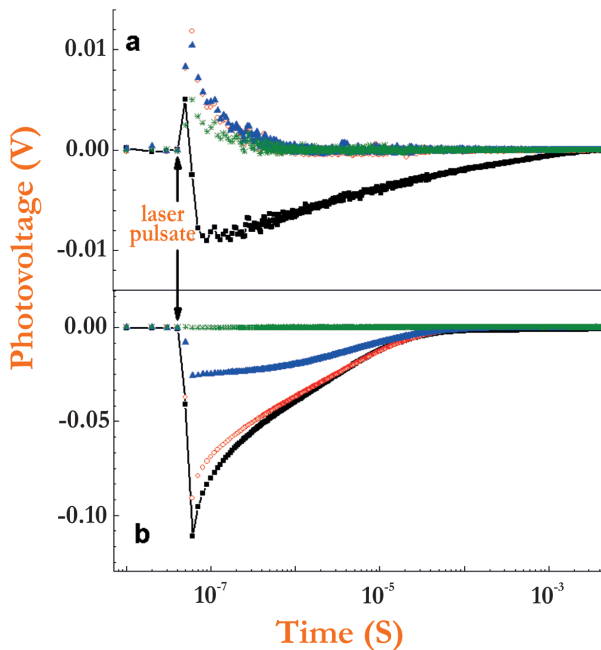


Fig. 3. Transient surface photovoltage curves taken with samples of: **a.** Al/ Cu_3BiS_3 and **b.** Al/ Cu_3BiS_3/In_2S_3 . Excitation at different wavelengths (■ 800 nm, ○ 1000 nm, △ 1300, * 1500 nm), with 40 ns pulsed laser.

Discussion

The results show that a significant increase in the p concentration of carriers is observed at temperatures greater than 250 °K, apparently caused by thermal excitation of electrons from extended states at the valence band toward accepting states within the gap, thus generating an increase in the density of negatively ionized accepting impurities (Figure 2). The results also show that the sharp increase in the conductivity of Cu_3BiS_3 thin films observed at temperatures greater than 250 °K (Mesa et al. 2009), is basically caused by an increase in the concentration of carriers at this range of temperatures (Kehoe et al. 2013).

It is also observed that the Cu_3BiS_3 samples present low mobility, probably due to the fact that the samples deposited by co-evaporation present a high density of native defects and ionized impurities affecting mobility (Figure 2). Also, it is observed that by increasing temperature, mobility increases, and the

increase is sharp at temperatures greater than 250 °K (especially in samples with a mass ratio of 0.46 and 0.49). This behavior might be explained by assuming that when temperature increases a greater quantity of carriers are excited from states at band tails towards states in a valence band region with lower effective mass (they become lighter). Nevertheless, it may also be observed that when temperature increases, carriers in grain boundary states are excited into the valence band, inducing thus a decrease in the height of the grain boundary barrier Φ_b . On the other hand, the mobility of the sample with mass ratio of 0.43 presents a different behavior at the low temperature region ($T < 150$ °K), which may be caused by a decrease in the probability that the carriers are displaced from a localized state into another via tunnel, in turn caused by some unknown mechanism.

Nevertheless, this disordered nanocrystalline feature of Cu_3BiS_3 films (Figure 1), causes the formation of a high density of associated states and point defects and incomplete bonds with high density grain boundaries, which are responsible for the low mobility values measured in this kind of samples, and for the formation of localized states allowing electrical transport via hopping.

From these results it is established that the generation of positive surface photovoltage when the Cu_3BiS_3 film is excited with photons of wavelength $\lambda > 890$ nm (Figure 3), occurs because in this case the excitation of electrons is induced from surface states towards the conduction band, which causes a decrease in the deflection of bands in the surface region, which in turn causes an increase in the SPV value. When the excitation is done with photons of wavelength $\lambda < 890$ nm, there is simultaneously an excitation of electrons from surface states to conduction bands, which causes positive SPV values, and fundamental excitation (transition of electrons from valence band to conduction band), which are diffused into surface states, giving rise to negative SPV values. As the negative effect of the fundamental electron transition is dominant, the net SPV value turns out to be negative (Mesa et al. 2010).

Moreover, it is observed that the transient SPV curves obtained with Cu_3BiS_3/In_2S_3 samples do not present change of sign and their values are generally

negative, except in the case in which the excitation is done with very low energy photons ($\lambda=1500$ nm), where the SPV change is not present (Figure 3). In this case, wavelengths longer than the In_2S_3 gap were chosen ($\lambda_G=688$ nm), in order to avoid excitation of electrons through fundamental absorption.

From obtained results it is observed that negative transient SPV of $\text{Cu}_3\text{BiS}_3/\text{In}_2\text{S}_3$ system decays further (two orders of magnitude), in comparison with the Cu_3BiS_3 negative transient SPV, apparently due to the passivation of Cu_3BiS_3 surface states, which causes elimination of the positive contribution in the SPV net value (Figure 3). Also, it is observed that the SPV value decreases as the excitation wavelength increases; this behavior is due to the fact that the decrease in energy from photons with the increase in λ , causes a decrease in the density of electrons photogenerated in the Cu_3BiS_3 , which move (by diffusion and drag) toward surface states of the In_2S_3 , thus reducing the net measured value of transient SPV. Measurements carried out with 1500 nm excitation show that there is no change in the SPV signal, indicating that the energy of the photons is lower than the activation energy of Cu_3BiS_3 and In_2S_3 surface states.

The experimental results presented here lead to the following interpretations. The Cu_3BiS_3 thin film surface presents a high density of surface states, where the emitted electrons are conducted to the

surface generating a positive SPV value. These states are distributed within the E_g , indicated by the start of the SPV at approximately 0.5 eV. The best distribution range for this work function agrees with that.

The energy band diagrams for Cu_3BiS_3 and for $\text{Cu}_3\text{BiS}_3/\text{In}_2\text{S}_3$ system show both the effect of the surface states on the surface deflection of the bands (Figure 4), causing the surface photovoltage changes observed in Cu_3BiS_3 films, and the effect of Cu_3BiS_3 surface state passivation on $\text{Cu}_3\text{BiS}_3/\text{In}_2\text{S}_3$ system surface photovoltage, which takes place when the In_2S_3 layer is deposited on it.

Conclusion

The Cu_3BiS_3 samples were grown by evaporation. From Hall effect and transient surface photovoltage measurements, it was found that the disordered nanocrystalline feature of the samples causes the formation of a high density of states associated with point defects and incomplete bonds with a high density of grain boundaries, which are responsible for the low mobility values, causing the formation of localized states which allow electrical transport via hopping. Nevertheless, when a buffer layer of In_2S_3 is included on the Cu_3BiS_3 surface, there is a passivation of defect states, improving the quality of the material.

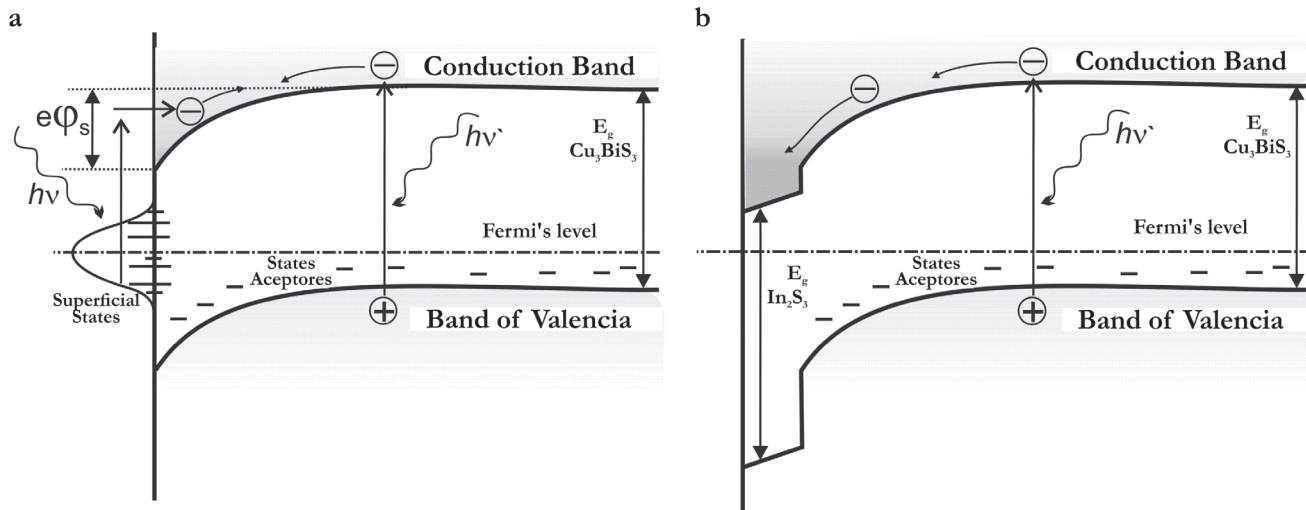


Fig. 4. Energy band diagram for: **a.** Cu_3BiS_3 and **b.** $\text{Cu}_3\text{BiS}_3/\text{In}_2\text{S}_3$ system, showing the effect of surface states and of passivation (of Cu_3BiS_3 por el In_2S_3) on the surface deflection of bands.

Acknowledgements

This work is supported by several projects from Universidad del Rosario. Special thanks also go to Universidad Nacional de Colombia, QUBITeXp International Trade SAS, Colombia. OFI at Pontificia Universidad Javeriana, projects 4513 and 4515, Banco de la República with Fundación para la Promoción de la Investigación y la Tecnología (FPIT), Project 2954 “Construcción de prototipo de pantalla plana tipo OLED con resolución 200X200”.

Conflict of Interest

This work does not present any conflicts of interest.

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Efecto Hall y estudio de fotovoltaje superficial transiente (SPV) en películas delgadas de Cu_3BiS_3

Resumen. Se presentan las propiedades eléctricas del compuesto Cu_3BiS_3 depositado por co-evaporación. Este es un nuevo compuesto que puede tener propiedades adecuadas para ser utilizado como capa absorbente en celdas solares. Las muestras fueron caracterizadas a través de medidas de efecto Hall y fotovoltaje superficial transiente (SPV). A través de medidas de efecto Hall se encontró que la concentración de portadores de carga n es del orden de 10^{16} cm^{-3} independiente de la relación de masas de Cu/Bi. También se encontró que la movilidad de este compuesto (μ del orden de $4 \text{ cm}^2\text{V}^{-1}\text{s}^{-1}$) varía de acuerdo con los mecanismos de transporte que la gobiernan en dependencia con la temperatura. A partir de las medidas de SPV se encontró alta densidad de defectos superficiales, defectos que son pasivados al superponer una capa buffer sobre el compuesto Cu_3BiS_3 .

Palabras clave: Transporte eléctrico; Cu_3BiS_3 ; fotovoltaje superficial transiente; defectos.

Efeito Hall e estudo de foto voltagem superficial transiente (SPV) em películas delgadas de Cu_3BiS_3

Resumo. Apresentam-se as propriedades elétricas do composto Cu_3BiS_3 depositado por co-evaporação. É um composto novo que pode ter as propriedades adequadas para ser utilizado como capa absorvente em células solares. As amostras foram caracterizadas através de medidas do efeito Hall e foto voltagem superficial transiente (SPV). Através de medidas do efeito Hall se encontro que a concentração de portadores de carga n é da ordem de 10^{16} cm^{-3} independentemente da relação de massas de Cu/Bi. Também se encontrou que a mobilidade des composto (μ da ordem de $4 \text{ cm}^2\text{V}^{-1}\text{s}^{-1}$) varia de acordo com os mecanismos de transporte que a governam em dependência com a temperatura. Partindo das medidas de SPV se encontrou uma alta densidade de defeitos superficiais, defeitos que são passivados a sobrepor uma capa buffer sobre o composto Cu_3BiS_3 .

Palavras-chave: Transporte elétrico; Cu_3BiS_3 ; foto voltagem superficial transiente; defeitos.