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Development and characterization of novel ZnO-loaded electrospun membranes for periodontal regeneration

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Abstract

Objectives—This study reports on the synthesis, materials characterization, antimicrobial capacity, and cytocompatibility of novel ZnO-loaded membranes for guided tissue/bone regeneration (GTR/GBR).

Methods—Poly(ϵ -caprolactone) (PCL) and PCL/gelatin (PCL/GEL) were dissolved in hexafluoropropanol and loaded with ZnO at distinct concentrations: 0 (control), 5, 15, and 30 wt. %. Electrospinning was performed using optimized parameters and the fibres were characterized via scanning and transmission electron microscopies (SEM/TEM), energy dispersive X-ray spectroscopy (EDS), Fourier transform infrared spectroscopy (FTIR), contact angle (CA), mechanical testing, antimicrobial activity against periodontopathogens, and cytotoxicity test using human dental pulp stem cells (hDPSCs). Data were analyzed using ANOVA and Tukey ($\alpha=5\%$).

Results—ZnO nanoparticles were successfully incorporated into the overall submicron fibres, which showed fairly good morphology and microstructure. Upon ZnO nanoparticles' incorporation, the PCL and PCL/GEL fibres became thicker and thinner, respectively. All GEL-containing membranes showed lower CA than the PCL-based membranes, which were highly hydrophobic. Overall, the mechanical properties of the membranes were reduced upon ZnO incorporation, except for PCL-based membranes containing ZnO at the 30 wt.% concentration.

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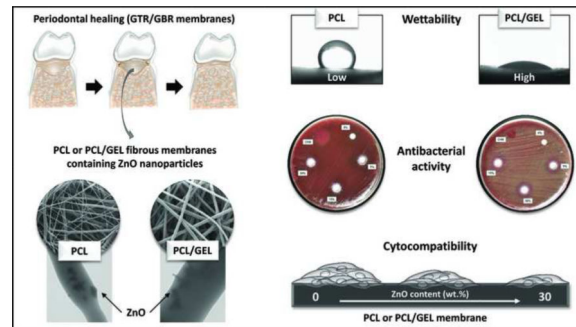
Conflict of interest

No conflicts of interest were declared.

The presence of GEL enhanced the stretching ability of membranes under wet conditions. All ZnO-containing membranes displayed antibacterial activity against the bacteria tested, which was generally more pronounced with increased ZnO content. All membranes synthesized in this study demonstrated satisfactory cytocompatibility, although the presence of 30 wt.% ZnO led to decreased viability.

Significance—Collectively, this study suggests that PCL- and PCL/GEL-based membranes containing a low content of ZnO nanoparticles can potentially function as a biologically safe antimicrobial GTR/GBR membrane.

Graphical Abstract



Keywords

electrospinning; zinc oxide; oral bacteria; periodontal regeneration; periodontitis; membranes

1. Introduction

Periodontitis is a very prevalent (ca. ~ 50% of the adult population) and aggressive dental pathology affecting both the gingiva and anchoring tissues of the tooth [1]. It is a bacterial-mediated inflammatory process, and if not treated, it may progressively destroy periodontal tissues, eventually resulting in tooth loss [2]. A recent systematic review confirmed that severe periodontitis has increased in the past two decades, and a predictable increasing burden of the disease can be expected, since life expectancy among the world population is continuously growing [3].

The traditional clinical treatment of periodontitis consists of surgical procedures and mechanical debridement (i.e. scaling and root planing), which allows healing of periodontal pockets/defects [4, 5]. Moreover, systemic and/or local antibiotic therapy may also be indicated in order to combat the infection or re-colonization of periodontopathogens [4–6]. Nonetheless, depending on the size of the pocket/inflamed site, the complete regeneration of periodontal tissues remains a difficult clinical task [4]. In this way, regenerative approaches, such as the use of bioresorbable or non-resorbable barrier membranes employed in so-called guided tissue/bone regeneration (GTR/GBR) strategies, as well as emerging technologies (e.g. cell-based and gene therapies, the use of bone anabolics, and laser treatments, among others) have been investigated in terms of their clinical potential in periodontal tissue regeneration [4, 7, 8]. Among the aforementioned approaches, GTR/GBR membranes

present the unique characteristic of resembling the extracellular matrix (ECM), thus facilitating cell adhesion, proliferation, and the differentiation of new tissues [4]. Furthermore, GTR/GBR membranes can be synthesized using a variety of natural and/or synthetic polymers, and depending on processing conditions, bioactive (e.g. antimicrobials) fibres on the nano- and micro-scale range can be fabricated [4, 6, 9–12].

Despite the fairly significant clinical success achieved with the use of membranes in GTR/GBR strategies, tissue regeneration is strongly dependent on the absence of infection, and therefore, the proper maintenance of a bacteria-free environment [4]. To that end, several controlled-release systems, such as Atridox[®] (doxycycline-based) and PerioChip[®] (chlorhexidine-based), have been thoroughly used to locally deliver antimicrobial agents to the periodontal pocket, resulting in positive clinical effects [13, 14]. More recently, antimicrobial agents have been incorporated into GTR/GBR membranes, endowing them with antibacterial activity [6, 15]. Notwithstanding, it is well recognized that antibiotics may produce some important side-effects, mainly those related to bacterial strain resistance, which is a current global concern, since bacteria are becoming resistant to several antibiotic therapies [4, 6]. Therefore, the development of biomaterials for GTR/GBR applications based on alternative antibacterial agents rather than antibiotics is paramount to translate safer regenerative technologies into clinical practice.

Inorganic ions and metallic oxides have extensively demonstrated antibacterial properties [16–19]. Silver is the most effective antibacterial metal at low concentrations and against both gram-positive (G+) and gram-negative (G-) bacteria. Meanwhile, several other metal oxides (e.g. copper, magnesium, calcium, titanium, and zinc) have also displayed important antibacterial activity against a wide variety of microorganisms. Zinc oxide (ZnO), for example, is one of the few zinc-based compounds listed as being generally recognized as a safe (GRAS) material by the Food and Drug Administration (FDA) [20]. Even though the mechanism of action is still vaguely understood, ZnO has been successfully used as an antibacterial agent in food packaging [21], restorative dental materials [22], wound dressings [23], and in tissue engineering applications [24]. Indeed, electrospun polymer-based scaffolds/membranes loaded with ZnO have recently displayed not only antibacterial properties [25], but also enhanced cell proliferation/wound healing [26]. It is also important to note that the antibacterial activity of these ZnO-loaded membranes was found to be dose-dependent (5 wt.%) [25]. Nonetheless, the earlier study tested the ZnO-scaffolds only on *Staphylococcus aureus* (G+) and *Escherichia coli* (G-), thus further highlighting the need to investigate their potential against periodontopathogens based on their application as GTR/GBR membranes.

It is well-established that degradability is one of the most important aspects involved in the clinical success of bioresorbable GTR/GBR membranes [4]. Indeed, it is expected that the membrane would be completely resorbed after treatment is completed, thus eliminating an additional surgical procedure for its removal from the regenerated site [4]. The degradability rate of GTR/GBR membranes strongly depends on the type of polymer system used. Poly(ϵ -caprolactone) (PCL) is a synthetic polymer with known degradability and hydrophobic characteristics [27], and considering the latter property influences the former, with highly hydrophobic systems displaying lower degradability, PCL usually presents a moderate

degradation rate. Recent research has utilized the strategy of blending natural polymers, such as collagen or gelatin, with PCL to grant enhanced degradability and hydrophilicity [28–30], as well as better cell-membrane response [31].

In this study, we tested the influence of ZnO nanoparticles and gelatin on the physico-chemical and mechanical properties of PCL-based electrospun membranes. Furthermore, we report for the first time on the antibacterial activity and cytocompatibility of these membranes' targeting a specific application in periodontal tissue regeneration by careful antimicrobial and cell compatibility evaluations against periodontopathogens and dental pulp stem cells, respectively.

2. Materials and methods

2.1. Materials

Poly(ϵ -caprolactone) pellets (PCL, $M_w = 80,000$), gelatine type B from bovine skin (~225 bloom, $M_w = 50,000$), 1,1,1,3,3,3-hexafluoro-2-propanol (HFP), and zinc oxide nanoparticles (ZnO, < 100 nm particle size) were all purchased from Sigma-Aldrich (St. Louis, MO, USA) and used as-received. Phosphate-buffered saline solution (PBS, 1%) was prepared by mixing PBS tablets (MP Biomedicals, Santa Ana, CA, USA) and distilled water. The plastic syringes (5 mL) and 27G dispensing needles were obtained from Becton, Dickson and Company (Franklin Lakes, NJ, USA) and CML Supply (Lexington, KY, USA), respectively.

2.2. Synthesis of PCL and PCL/Gelatin membranes loaded with ZnO nanoparticles

The electrospinning system used in this work consisted of a high-voltage source (ES50P-10W/DAM, Gamma High-Voltage Research Inc., Ormond Beach, FL, USA), a syringe pump (Legato 200, KD Scientific Apparatus, Holliston, MA, USA), and a grounded stainless steel collecting drum connected to a high-speed mechanical stirrer (BDC6015, Caframo, Warton, ON, Canada), and was described in detail elsewhere [6, 32, 33]. Processing parameters, including solution concentration, field strength, and flow rate were optimized for the distinct polymer systems to obtain defect-free fibrous membranes. Two stock polymer solutions were prepared in HFP; namely, pure PCL and a PCL/gelatin blend (PCL/GEL, ratio of 1:1, w/w). Briefly, the polymers were dissolved in HFP to produce 10 wt.% (100 mg mL^{-1}) and were stirred overnight. Next, ZnO nanoparticles were added at different concentrations (0, 5, 15, and 30 wt.%, relative to the total polymer weight); the mixtures were stirred for 24 h and then sonicated for 90 minutes before electrospinning to improve nanoparticle dispersion [7]. Each solution was then loaded into the plastic syringe fitted with the 27G metallic blunt-tip and electrospun using the following parameters: rotating mandrel (120 rpm of speed), fixed spinning distance of 18 cm, flow rate of 1.5 mL/h, and different electric voltages, depending on the solution, as determined by an optimization process that involved experiments conducted on each variable. The obtained mats (hereafter referred to as membranes), i.e. pure PCL, PCL/GEL, and the ones containing ZnO nanoparticles, were collected at room temperature (RT) and then dried under vacuum for at least 48 h to complete removal of any remaining solvent [7, 32].

2.3. Morphological (SEM/TEM) and Chemical (EDS/FTIR) Characterizations of the fibres

The electrospun membranes were observed under a field-emission scanning electron microscope (FE-SEM, Model JSM-6701F, JEOL, Tokyo, Japan) to evaluate the morphology and overall fibre architecture. In brief, samples taken from the different electrospun membranes were mounted on an Al stub and sputter-coated with Au-Pd prior to imaging. Image J software (National Institutes of Health, Bethesda, MD, USA) was used to calculate the diameter of 50 single-fibres per image obtained from the FE-SEM analyses (3 images/group) at the same magnification (5000 \times); the fibre diameter was then averaged and reported as mean \pm SD [7, 32]. Energy dispersive X-ray spectroscopy (EDX) was performed under FE-SEM to semi-quantitatively analyse the chemical composition of the fibres. Transmission electron microscopy (TEM, Model JEM-2010, JEOL) was also carried out to investigate the ZnO nanoparticles' incorporation into the polymer fibres. Fourier transform infrared spectroscopy (FTIR) was done in the attenuated total reflection mode for the PCL, PCL/GEL, and membranes containing ZnO nanoparticles (ATR/FTIR-4100, JASCO, Easton, MD, USA), over the range of 700–4,000 cm^{-1} at a resolution of 4 cm^{-1} to investigate the chemical characteristics of the membranes and ZnO incorporation [7, 32].

2.4. Contact angle (CA)

The PCL and PCL/GEL solutions loaded or not with the distinct concentration of ZnO were electrospun into fibres using the aforementioned electrospinning parameters over microscope glass coverslips (Fisherbrand, Fisher Scientific, Loughborough, UK) mounted on the rotating mandrel (n=10). Next, the surface CA of the membranes was measured using a goniometer (Model PG-2, Gardco, Pompano Beach, FL, USA) by dropping three consecutive drops of distilled water ($\sim 5 \mu\text{L}$) per sample. The measured angles were then averaged.

2.5. Mechanical properties

The mechanical properties (i.e. tensile strength, Young's modulus, and elongation at break) of all the synthesized electrospun membranes were evaluated by uni-axial tensile testing (expert 5601, ADMET, Norwood, MA, USA) [32]. Rectangular samples (15 mm \times 3 mm) were tested (n=4) under both dry (immediate testing without storage) or wet (tested after storage in PBS solution for 24 h) conditions at a crosshead speed of 1 mm min^{-1} . The specimen thickness was determined by measuring with calipers at three locations. Mechanical data were obtained from the stress-strain curves of each sample and expressed in MPa. The results are reported as mean \pm standard deviation (SD).

2.6. Antimicrobial activity

The antibacterial activity of the processed fibrous membranes was evaluated against known periodontopathogens, i.e. *Porphyromonas gingivalis* (Pg) (ATCC 33277) and *Fusobacterium nucleatum* (Fn) (ATCC 25586) using agar diffusion assays (n=3/group/bacteria) [33]. Bacteria were cultivated in Brain Heart Infusion (BHI) broth supplemented with 5 g yeast extract/L and 5% v/v vitamin K+ hemin (BHI-YE; Becton, Dickinson and Company) at 37 $^{\circ}\text{C}$ in an anaerobic GasPak jar for 24 h [33]. Before testing, the minimum inhibitory concentration (MIC) of ZnO was determined using suspensions of ZnO

nanoparticles in PBS at the following concentrations: 100, 250, 500, 1,000, 2,500, 5,000, and 10,000 µg/mL. In brief, 10 µL of each suspension was dropped on cultured blood agar plates containing the bacterial lawns, and the inhibition zones (in mm) were measured after 5 days of incubation [33]. For the antibacterial activity of the membranes, disk-shaped samples (5-mm in diameter) were prepared and sterilized through ultraviolet (UV) irradiation for 1 h (30 min each side), before placing on the cultured blood agar plates as previously indicated [33, 34]. The inhibition zone (in mm) of each sample was measured after 5 days of incubation.

2.7. Cytotoxicity test

The guidelines provided by the International Standards Organization/ISO [35] were followed in order to detect toxicity levels of the electrospun membranes [34]. Human dental pulp stem cells (hDPSCs, AllCells LLC., Alameda, CA, USA) obtained from permanent third molars were cultured in low glucose Dulbecco's Modified Eagle's Medium (DMEM, GIBCO, Invitrogen, Grand Island, NY, USA) supplemented with 10% FBS (Hyclone Laboratories Inc., Logan, UT, USA) and 1% penicillin-streptomycin (Sigma) in a humidifier incubator at 37°C, with 5% CO₂ [34]. All the processed membranes were carefully cut into squares (15 × 15 mm²) and sterilized under UV light (30 min each side) and by soaking in 70% ethanol for 30 min, followed by washing (5 min) twice in sterile PBS before testing [34]. Next, sterile samples (surface ratio = 1 cm²/mL of medium) were incubated in DMEM for 48 h at 37°C under a 5% CO₂ humidified atmosphere in order to produce extracts from each membrane [34]. Extracts of sterile ultra-high molecular weight polyethylene (UHMWPE) were similarly obtained (negative control). Supernatants were then filtered through a membrane (Millipore®), and serial dilutions (100, 50, 25, 12.5, and 6.25 vol.%) were prepared from the extracts [34, 36]. Dilutions were also performed for the positive control (i.e. 0.3 vol.% phenol solution) [34, 36]. hDPSCs at passage 9 were seeded at a density of 3 × 10³/well and allowed to adhere in the wells of 96-well tissue culture microtiter plates. The media was replaced after 4 h by the corresponding extract concentrations, and negative and positive controls, which were dispensed into each well (100 µL) [34]. Control columns of four wells were prepared with a medium without cells (blank), and a medium with cells but without the extract (100% survival) [34–36]. The microplate was then incubated again and after 48 h, 20 µL of CellTiter 96 AQueous One Solution Reagent (Promega Corporation, Madison, WI, USA) was added to the test wells and allowed to react for 2 h at 37°C in humidified 5% CO₂ atmosphere. Incorporated dye was measured by reading the absorbance at 490 nm in a microplate reader against a blank column [34–36].

2.8. Statistical analysis

The obtained data were statistically analysed (SigmaPlot version 12, Systat Software Inc., San Jose, CA, USA) using Analysis of Variance (One-Way for cytocompatibility test and Two-Way for all the other tests) and Tukey's test for multiple comparison ($\alpha = 5\%$).

3. Results

3.1. Morphological and Chemical Characterizations of the fibres

The PCL- and PCL/GEL-based membranes presented, respectively, a heterogeneous (from 93 to 2,223 nm) and homogeneous (from 56 to 1,184 nm) fibre diameter distribution (Figure 1a). The presence of gelatine produced thicker fibres compared to those gelatine-free, although only in groups containing none or low content (5 wt.%) of ZnO. The incorporation of higher amounts of ZnO (higher than 15 wt.%) resulted in rough fibres with distinct shape (e.g., swollen areas) and morphology, although the PCL membranes were more negatively affected than those constituted of gelatine (Figure 2). As shown in the EDS analysis (Figure 1b), ZnO nanoparticles were successfully incorporated into the membranes. The nanoparticles differed in shape, although most of them were rod-like and nano-sized (i.e. below 100 nm, with an average size of 65 nm) (Figure 3). While the neat fibres were visually smooth (Figure 4a), the nanocomposite fibres presented some swollen areas and several areas embedded with ZnO nanoparticles (Figure 4b). Lastly, the FTIR analysis also revealed incorporation of ZnO into the membranes (Figure 5) and gelatine into the PCL/GEL-based membranes (Figure 5b). The presence of the peak located at $2,357\text{ cm}^{-1}$ could be observed in the ZnO powder and also in the ZnO-incorporated membranes, but not in the neat membranes. FTIR spectra showed some common peaks for all membranes located at $2,944\text{--}2,861$ and $1,721\text{ cm}^{-1}$, which corresponded, respectively, to the presence of the C–H bond of saturated carbons and the ester-carbonyl group (–CO stretching) found in PCL [25]. In addition, three characteristic peaks confirmed the incorporation of gelatine into the PCL/GEL membranes: one located at $3,297\text{ cm}^{-1}$ (amide-A), another at $1,627\text{ cm}^{-1}$ (amide-I), and the last at $1,524\text{ cm}^{-1}$ (amide-II).

3.2. Contact angle

According to Figure 6, all PCL-based membranes, regardless of the ZnO content ($p = 0.056$), presented a hydrophobic surface (i.e. contact angle higher than 90°), showing poor wettability (Zhang *et al.*, 2004). By contrast, PCL/GEL membranes revealed excellent wettability/hydrophilicity, with CA ranging from 0° to 42° . The PCL/GEL membranes containing 5 and 15 wt.% of ZnO nanoparticles were more hydrophilic than the other gelatine-based membranes ($p < 0.001$).

3.3. Mechanical properties

The mechanical performance (e.g. tensile strength, Young's modulus, and elongation at break) of the synthesized membranes is presented in Figure 7. The PCL membranes showed a soft nature with moderate flexibility regardless of the storage condition. The incorporation of ZnO reduced the mechanical properties, except the elongation at break, which was increased with the addition of 30 wt.% of nanoparticles. The presence of gelatine improved rigidity of the membranes, but only under dry circumstances ($p = 0.002$); conversely, storage in PBS resulted in a significant decrease in rigidity, although a substantial gain in flexibility could be observed: wet specimens of membranes containing 0, 5, 15, and 30 wt.% of ZnO displayed, respectively, 3.4, 2.9, 2.7, and 4.3 times higher ($p < 0.001$) elongation at break than the specimens stored dry.

3.4. Antibacterial activity

Results of the MIC of ZnO nanoparticles are shown in Table 1. From all the concentrations tested, the MIC differed according to the bacteria specie: for *Pg*, the MIC was 2,500 µg/mL, whereas it was 5,000 µg/mL for *Fn*. The increase in ZnO concentration did not result in higher inhibition zones against *Pg* ($p = 0.599$). By contrast, the inhibition zones were greater against *Fn* with increased content of ZnO ($p < 0.001$).

Data and representative images from the agar diffusion test are presented in Table 1. All the membranes containing different concentrations of ZnO presented antibacterial activity against the targeted bacteria, with clear inhibition zones ranging from 6 to 15 mm in diameter. The PCL membrane containing 30 wt.% of ZnO exhibited higher ($p < 0.001$) inhibition zone than the low-concentrated group and against both bacterial species. The PCL/GEL-based membranes applied against *Pg* showed similar ($p = 0.702$) antibacterial activity regardless of the concentration of ZnO; however, when applied against *Fn*, the most concentrated group displayed higher antibacterial activity than the others ($p < 0.001$).

3.5. Cytotoxicity test

As shown in Figure 8, all membranes presented cytocompatibility (i.e. cell viability higher than 50%), which was different from the positive control (phenol solution), that was toxic to cells, even at low concentrations. Cell viability remained at an optimal level with ZnO content up to 15 wt.%; nonetheless, incorporation of a higher amount of nanoparticles resulted in a slight toxic effect on the cells, with the PCL/GEL-based membrane producing the highest cytotoxic potential.

4. Discussion

In this study, we investigated extensively both the morphology and microstructure of the electrospun membranes. According to the SEM micrographs shown in Figure 1a, the membranes demonstrated noticeable differences in regard to the morphology of the fibres, which was influenced by the polymer system and the content of the ZnO that was incorporated. The PCL-based membranes showed a more heterogeneous fibre distribution, whereas the PCL/GEL-based membranes presented fibres with a much lower distribution range, probably due to the increase in viscosity achieved through gelatine addition, leading to a more satisfactory chain entanglement, and thus to the formation of homogeneous fibres [37]. Moreover, even though we did not determine the viscosity of the polymer solution or the polymer blend, one would consider that the increased viscosity of the gelatine-based solutions might also have contributed to the occurrence of thicker fibres compared to those gelatine-free, corroborating previous findings [29]. Nonetheless, this effect was observed only within the neat fibres and the membrane containing 5 wt.% of ZnO, since the addition of higher amounts of ZnO nanoparticles (e.g. 15 and 30 wt.%) progressively reduced fibre diameter. One possible explanation is that both gelatine and ZnO nanoparticles can increase the electrical conductivity of the polymer solution [38, 39], acting synergistically toward the increase in charge density on the surface of the ejected spinning jet, leading to a reduction of self-repulsion tension, and consequently to the increase in elongation forces and the formation of thinner fibres [25, 40]. Interestingly, the incorporation of ZnO nanoparticles

into the PCL/GEL solutions produced smooth fibres, with only some areas displaying surface irregularities similar to a bead-like structure. Conversely, the addition of 30 wt.% ZnO particles into the PCL-based solution substantially modified the fibres' morphology, which became thicker and rougher compared to the other membranes, probably due to the agglomeration of ZnO (Figure 2a) on the surface and inner structure of the fibres [25]. ZnO nanoparticles present high surface energy, thus they tend to agglomerate when mixed at higher concentrations, forming clusters along the fibre length, and consequently resulting in important morphological change. These characteristics may negatively influence cell proliferation [41], thus limiting the use of PCL membranes containing high ZnO nanoparticles' content (in this study - 30 wt.%) as GTR/GBR membranes. Nonetheless, the presence of gelatine in the PCL solution allowed for the incorporation of a higher ZnO amount (i.e. 30 wt.%) without deleterious changes in the morphology and microstructure of the obtained membranes.

Though the presence of ZnO nanoparticles can be fairly well noticed in some of the SEM micrographs shown in Figures 1a and 2a, EDS (Figure 1b), TEM (Figure 4), and FTIR (Figure 5a) analyses confirmed the presence of ZnO trapped within the PCL matrix. Taken together, these results confirmed the chemical characteristics of the membranes synthesized in the study.

ZnO nanoparticles are not soluble in water. More importantly, considering that PCL is also hydrophobic [29, 31], one can expect that the ZnO-incorporated PCL membranes would demonstrate poor wettability. Nonetheless, optimal wettability is an important characteristic aimed at GTR/GBR membranes [4], since the membranes are placed in direct contact with moisture (i.e. fluids present in the periodontal surgical site). Furthermore, several studies have demonstrated that hydrophilic scaffolds can display enhanced cell affinity, thus allowing improved cell proliferation and wound healing [30, 42]. To that end, gelatine, a natural polymer, has been commonly used as an additive to PCL solutions in an attempt to increase scaffold hydrophilicity. Although cells can attach and grow in both pure PCL and PCL/GEL membranes, cell spreading and migration have been improved within gelatine-containing membranes [31]. According to Figure 6, all PCL-based membranes presented a hydrophobic surface (i.e. contact angle higher than 90°), showing poor wettability [43]. By contrast, PCL/GEL membranes revealed excellent wettability/hydrophilicity. An interesting finding of the present study was that the PCL/GEL membranes containing 5 and 15 wt.% of ZnO nanoparticles were more hydrophilic than the other gelatine-based membranes ($p < 0.001$). In fact, 10 s after the water drop (i.e. time set to allow the drop to accommodate and interact with the surface [44]), the membranes containing low (5 wt.%) and medium (15 wt.%) ZnO contents could not maintain a convex shape of the water drops, thus precluding the contact angle measurement. This, perhaps, demonstrates that the latter membranes were very hydrophilic, and therefore, they can be considered promising for GTR/GBR applications. One can also note that the same effect was not seen for the PCL/GEL membrane containing ZnO nanoparticles at 30 wt.%, which may possibly be explained by the occurrence of van der Waals physical interactions between the nanoparticles and PCL polymer chains, as previously suggested [25]. These physical interactions, which may have become more intense at higher ZnO contents (e.g. 30 wt.%), could have slightly reduced the surface energy of the membrane, thus resulting in lower wettability [43], although still

hydrophilic (i.e. contact angle lower than 90°) and comparable to the ZnO-free PCL/GEL membrane.

As previously highlighted, despite adequate morphology, microstructure, and wettability, GTR/GBR membranes should also present satisfactory mechanical properties to avoid membrane collapse into the periodontal defect as a result of compression forces of the soft tissues and mastication [4, 7]. Furthermore, GTR/GBR membranes need to display appropriate stiffness to withstand cell attachment and proliferation; otherwise, the membranes may not be clinically useful [4, 7, 9, 12]. A recent study demonstrated that cell viability and proliferation of 3T3 fibroblasts cultured on stiffer membranes were higher than on softer ones [45]. Findings from another investigation showed that the increase in elastic modulus obtained with the incorporation of alginate in a PCL-based membrane resulted in improved cell viability, as well as other important cell activity parameters (e.g. cell proliferation, alkaline phosphatase activity, and mineralization ability) [46]. Considering these foregoing statements, the mechanical performance (e.g. tensile strength, Young's modulus, and elongation at break) of the synthesized membranes was assessed, under dry and wet conditions by uni-axial tensile testing. The results obtained in this study (Figure 7) corroborate the findings of some previous studies, which demonstrated that blending gelatine with synthetic polymers, such as PCL, can increase the crystallinity [29] and mechanical strength of fibres [28–30, 42], although their storage in wet media may produce an inverse effect, since gelatine usually does not cross-link to the main polymer chain, leading to fast hydrolytic degradation of the system and a reduction of properties [29]. Worth mentioning, the use of crosslinking agents (i.e. glutaraldehyde, genipin, etc.) can potentially enhance the mechanical properties of these gelatine-containing membranes, since crosslinking reactions can be produced, leading to improved mechanical properties and degradation resistance [47, 48]. This would be an interesting focus for future studies. With regard to ZnO incorporation, an overall analysis of the tensile strength, Young's modulus, and elongation at break of the PCL and PCL/GEL-based membranes, revealed a decrease, corroborating with a previous study [25], perhaps with two exceptions: (1) the rigidity of the PCL membrane was maintained with the incorporation of 30 wt.% nanoparticles and (2) flexibility was significantly increased. This can probably result from microstructure of the aforementioned membrane, which showed thicker fibres (Figures 1a and 2a) that withstood higher loading during testing and improved flexibility. Nonetheless, and as stated before, this membrane would not be ideal for GTR/GBR application, since the microstructure/morphology of fibres lacked in satisfactory characteristics that resemble the ECM of native bone [4].

Another important goal of this study was to demonstrate that ZnO-incorporated membranes show antimicrobial activity when compared to their ZnO-free counterparts against common periodontopathogens. First, the minimal inhibition concentration (MIC) of ZnO nanoparticles was determined using suspensions of ZnO in PBS. According to Table 1, against *Pg*, the MIC was 2,500 µg/mL, whereas against *Fn* it was 5,000 µg/mL. These results are quite different from a previous study, which demonstrated that the MIC of ZnO was 250 µg/mL against both *Pg* and *Fn* [49]. That concentration, perhaps, was not effective in our study, and this may be due to the characteristics of the ZnO nanoparticles used.

According to recent studies [50, 51], differences in nanoparticle size may play a significant role in the antibacterial potential, with smaller particles demonstrating higher surface area, and therefore, enhanced inhibitory effects. The study by Vargas-Reus et al. [49] tested nanoparticles ranging from 10–50 nm, whereas the nanoparticles used in the present study had an average size of 65 nm (Figure 4). Consequently, the surface area of the ZnO particles used here was reduced, requiring an increased MIC (Table 1). Interestingly, increased concentrations of ZnO did not result in higher inhibition zones against *Pg*, whereas the antibacterial activity of ZnO against *Fn* was greater at 10,000 µg/mL when compared to the 5,000 µg/mL concentration. According to these results, the minimum ZnO dose to be tested in the present investigation for preparing the PCL- and PCL/GEL-based solutions was 5 wt. %, which corresponded to the 5,000 µg/mL concentration (i.e. the ZnO concentration that was effective against both bacteria species).

Even though the mechanism of action of ZnO is still vaguely understood, it has been reported that the nanoparticles can produce reactive oxygen species (ROS), such as hydroxyl radicals, super oxides, and hydrogen peroxide, which may penetrate the cell wall and affect bacteria integrity [52–54]. In the present study, the antibacterial activity displayed by the ZnO-incorporated membranes seemed to be dose-dependent, since the 30 wt.% concentration resulted in higher inhibition zones when compared to the less concentrated groups (Table 1). Indeed, it is expected that the higher the amount of nanoparticles incorporated, the greater the availability of active compound to diffuse into the agar medium and effectively inhibit bacterial growth [26]. Notwithstanding, the only exception was seen within the PCL/GEL-based membranes applied against *Pg*, which showed similar antibacterial activity regardless of the concentration of ZnO. These findings may suggest that gelatine-based materials can interact differently with the bacteria surface, depending on the specie evaluated. According to Bhadra et al. [55], amino-based molecules, such as gelatine, can interact with the outermost cell membrane of some bacteria by means of amino groups (–NH₂), enhancing permeability of the latter, and consequently, resulting in its destruction/death. This last mechanism might explain the greater antibacterial activity of the PCL/GEL-based membranes against *Fn*, which exhibited inhibition zones ranging from 11 to 14 mm, and in comparison with *Pg* (Table 1), whose inhibition zones ranged from 8 to 9 mm in diameter, only. This is probably due to structural differences [49] between the bacteria investigated and possible differences in their intracellular antioxidant content [56]. In light of this, the antibacterial potential of ZnO-incorporated membranes should be also investigated against different bacteria from those tested here, such as facultative anaerobic species, including but not limited to *A. actinomycetemcomitans*. Notwithstanding, and after all, the ZnO-incorporated membranes prepared in this study are promising candidates for GTR/GBR applications, showing antibacterial properties that would potentially prevent infection and/or recolonization of bacteria at the periodontal site.

Considering that cells need a proper substrate to attach and proliferate in, apart from important membrane/scaffold characteristics, such as fibre shape, fibre diameter/distribution, pore structure, wettability to facilitate protein adsorption, and cell attachment, GTR/GBR ZnO-loaded membranes should invariably be biocompatible. We investigated cell compatibility using human dental pulp stem cells, knowing their significant osteogenic

differentiation potential [57, 58], key in periodontal bone regeneration. As shown in Figure 8, all membranes presented cytocompatibility (i.e. cell viability higher than 50%), which was different from the positive control (phenol solution), that was toxic to cells, even at low concentrations. It is important to note that cell viability remained at an optimal level, even with ZnO content up to 15 wt.%, thus confirming the biocompatible nature of the ZnO-incorporated membranes. Nonetheless, incorporation of a higher amount of nanoparticles resulted in a slight toxic effect on the cells, with the PCL/GEL-based membrane producing the highest cytotoxic potential. These findings corroborate previous studies, which reported that, at low concentration (< 5 wt.%) ZnO is not harmful to cells [23, 25, 59].

Notwithstanding, the amount incorporated in the present study was significantly higher when compared to these previous studies, but it still did not show strong cytotoxicity, even at the 30 wt.% concentration. This may suggest that, perhaps the nanoparticles were homogeneously distributed within the fibres, which may, in turn, contribute to a proper and safe release of ZnO to the periodontal site.

One should note that the scope of the cell-related work presented in this study was limited. Future studies using these novel ZnO-loaded membranes should investigate cell-membrane compatibility using other cell types (e.g. periodontal ligament fibroblasts). Furthermore, the effects of ZnO should be explored from a molecular standpoint to better understand its role on DPSC cell functions and its differentiation before *in vivo* testing using GTR/GBR periodontal defect models.

5. Conclusion

The fabrication of a novel, biocompatible periodontal fibrous membrane with antibacterial properties provided by zinc oxide nanoparticles was reported. This electrospun membrane holds promise in terms of microstructure/morphology that resembles the extracellular matrix of native tissues, wettability characteristics, mechanical integrity, antibacterial activity against common periodontopathogens, and cytocompatibility to human dental pulp stem cells. It was demonstrated that the incorporation of a low amount of zinc oxide nanoparticles can improve the bioactivity of the membranes that may be potentially applied for GTR/GBR applications, therefore leading to an enhanced and predictable periodontal regeneration.

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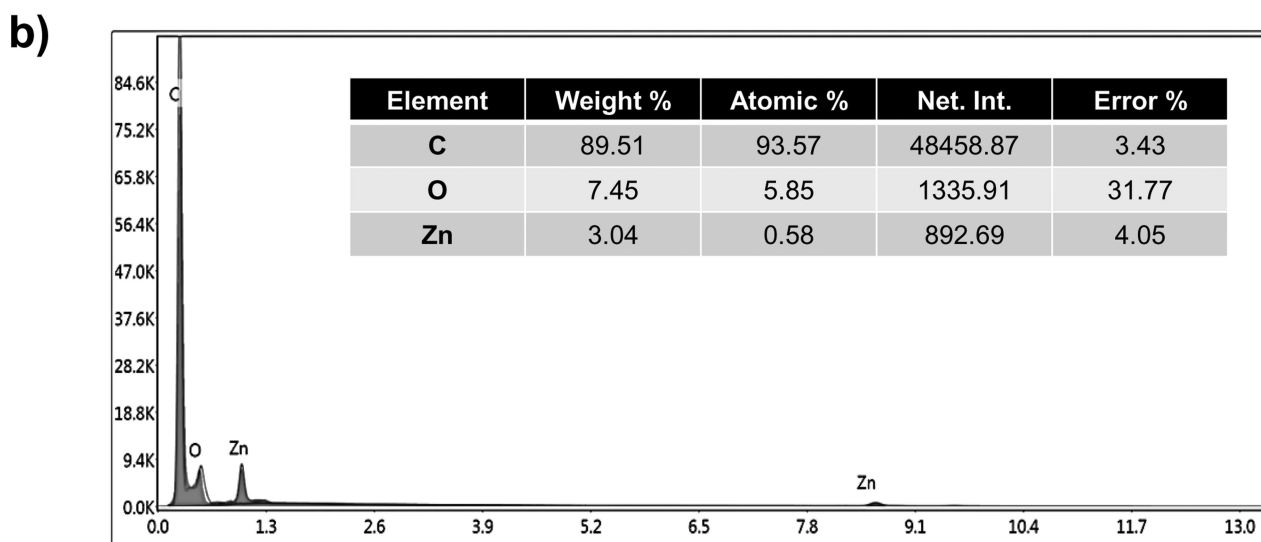
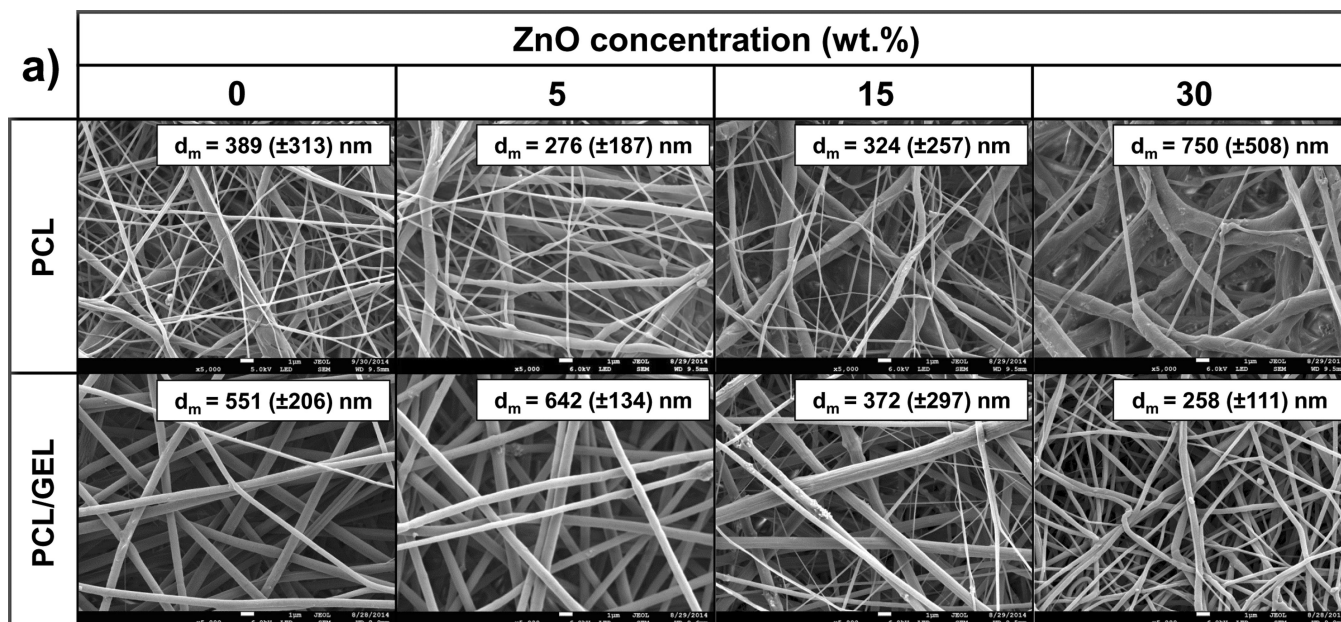
References

1. Albandar JM, Brunelle JA, Kingman A. Destructive periodontal disease in adults 30 years of age and older in the United States, 1988–1994. *J Periodontol.* 1999; 70:13–29. [PubMed: 10052767]
2. Haffajee AD, Socransky SS. Microbial etiological agents of destructive periodontal diseases. *Periodontol* 2000. 1994; 5:78–111. [PubMed: 9673164]
3. Kassebaum NJ, Bernabe E, Dahiya M, Bhandari B, Murray CJ, Marcenes W. Global Burden of Severe Periodontitis in 1990–2010: A Systematic Review and Meta-regression. *J Dent Res.* 2014; 93:1045–1053. [PubMed: 25261053]

4. Bottino MC, Thomas V, Schmidt G, Vohra YK, Chu TM, Kowolik MJ, et al. Recent advances in the development of GTR/GBR membranes for periodontal regeneration--a materials perspective. *Dent Mater.* 2012; 28:703–721. [PubMed: 22592164]
5. Keestra JA, Grosjean I, Coucke W, Quirynen M, Teughels W. Non-surgical periodontal therapy with systemic antibiotics in patients with untreated chronic periodontitis: a systematic review and meta-analysis. *J Periodontol Res.* 2014 Aug 21. [Epub ahead of print].
6. Bottino MC, Arthur RA, Waeiss RA, Kamocki K, Gregson KS, Gregory RL. Biodegradable nanofibrous drug delivery systems: effects of metronidazole and ciprofloxacin on periodontopathogens and commensal oral bacteria. *Clin Oral Investig.* 2014; 18:2151–2158.
7. Bottino MC, Thomas V, Janowski GM. A novel spatially designed and functionally graded electrospun membrane for periodontal regeneration. *Acta Biomater.* 2011; 7:216–224. [PubMed: 20801241]
8. Cochran DL, Cobb CM, Bashutski JD, Chun YH, Lin Z, Mandelaris G, et al. Emerging Regenerative Approaches for Periodontal Reconstruction: A Consensus Report. *J Periodontol.* 2014 Oct 15.:1–5. [Epub ahead of print]. [PubMed: 24364833]
9. Fujihara K, Kotaki M, Ramakrishna S. Guided bone regeneration membrane made of polycaprolactone/calcium carbonate composite nano-fibers. *Biomaterials.* 2005; 26:4139–4147. [PubMed: 15664641]
10. Li WJ, Cooper JA Jr, Mauck RL, Tuan RS. Fabrication and characterization of six electrospun poly(alpha-hydroxy ester)-based fibrous scaffolds for tissue engineering applications. *Acta Biomater.* 2006; 2:377–385. [PubMed: 16765878]
11. Schofer MD, Roessler PP, Schaefer J, Theisen C, Schlimme S, Heverhagen JT, et al. Electrospun PLLA nanofiber scaffolds and their use in combination with BMP-2 for reconstruction of bone defects. *PLoS One.* 2011; 6:e25462. [PubMed: 21980467]
12. Yang F, Both SK, Yang X, Walboomers XF, Jansen JA. Development of an electrospun nano-apatite/PCL composite membrane for GTR/GBR application. *Acta Biomater.* 2009; 5:3295–3304. [PubMed: 19470413]
13. Johnson LR, Stoller NH. Rationale for the use of Atridox therapy for managing periodontal patients. *Compend Contin Educ Dent.* 1999; 20:19–25. [PubMed: 11908360]
14. Salvi GE, Mombelli A, Mayfield L, Rutar A, Suvan J, Garrett S, et al. Local antimicrobial therapy after initial periodontal treatment. *J Clin Periodontol.* 2002; 29:540–550. [PubMed: 12296782]
15. Chen DW, Lee FY, Liao JY, Liu SJ, Hsiao CY, Chen JK. Preclinical experiments on the release behavior of biodegradable nanofibrous multipharmaceutical membranes in a model of four-wall intrabony defect. *Antimicrob Agents Chemother.* 2013; 57:9–14. [PubMed: 22948881]
16. Goudouri OM, Kontonasaki E, Lohbauer U, Boccaccini AR. Antibacterial properties of metal and metalloid ions in chronic periodontitis and peri-implantitis therapy. *Acta Biomater.* 2014; 10:3795–3810. [PubMed: 24704700]
17. Hans M, Erbe A, Mathews S, Chen Y, Solioz M, Mucklich F. Role of copper oxides in contact killing of bacteria. *Langmuir.* 2013; 29:16160–16166. [PubMed: 24344971]
18. Hwang SH, Song J, Jung Y, Kweon OY, Song H, Jang J. Electrospun ZnO/TiO₂ composite nanofibers as a bactericidal agent. *Chem Commun.* 2011; 47:9164–9166.
19. Sawai J, Yoshikawa T. Quantitative evaluation of antifungal activity of metallic oxide powders (MgO, CaO and ZnO) by an indirect conductimetric assay. *J Appl Microbiol.* 2004; 96:803–809. [PubMed: 15012819]
20. FDA. Part 182 - Substances generally recognized as safe. Food and drug administration. Washington, DC, USA: 2014. Available at: http://www.ecfr.gov/cgi-bin/text-dx?c=ecfr&sid=786bafc6f6343634fbf79fcdca7061e1&rgn=div5&view=text&node=21:3.0.1.1.13&idno=21#se21.3.182_18991 [Accessed on 26 October 2014]
21. Anitha S, Brabu B, John Thiruvadigal D, Gopalakrishnan C, Natarajan TS. Optical, bactericidal and water repellent properties of electrospun nano-composite membranes of cellulose acetate and ZnO. *Carbohydr Polym.* 2013; 97:856–863. [PubMed: 24066357]
22. Kasraei S, Sami L, Hendi S, Alikhani MY, Rezaei-Soufi L, Khamverdi Z. Antibacterial properties of composite resins incorporating silver and zinc oxide nanoparticles on *Streptococcus mutans* and *Lactobacillus*. *Restor Dent Endod.* 2014; 39:109–114. [PubMed: 24790923]

23. Shalumon KT, Anulekha KH, Nair SV, Nair SV, Chennazhi KP, Jayakumar R. Sodium alginate/poly(vinyl alcohol)/nano ZnO composite nanofibers for antibacterial wound dressings. *Int J Biol Macromol.* 2011; 49:247–254. [PubMed: 21635916]
24. Amna T, Hassan MS, Sheikh FA, Lee HK, Seo KS, Yoon D, et al. Zinc oxide-doped poly(urethane) spider web nanofibrous scaffold via one-step electrospinning: a novel matrix for tissue engineering. *Appl Microbiol Biotechnol.* 2013; 97:1725–1734. [PubMed: 22918299]
25. Augustine R, Dominic EA, Reju I, Kaimal B, Kalarikkal N, Thomas S. Electrospun polycaprolactone membranes incorporated with ZnO nanoparticles as skin substitutes with enhanced fibroblast proliferation and wound healing. *RSC Adv.* 2014; 4:24777–24785.
26. Augustine R, Malik HN, Singhal DK, Mukherjee A, Malakar D, Kalarikkal N, et al. Electrospun polycaprolactone/ZnO nanocomposite membranes as biomaterials with antibacterial and cell adhesion properties. *J Polym Res.* 2014; 21:347.
27. Dash TK, Konkimalla VB. Polymeric modification and its implication in drug delivery: poly-epsilon-caprolactone (PCL) as a model polymer. *Mol Pharm.* 2012; 9:2365–2379. [PubMed: 22823097]
28. Jin G, Prabhakaran MP, Kai D, Annamalai SK, Arunachalam KD, Ramakrishna S. Tissue engineered plant extracts as nanofibrous wound dressing. *Biomaterials.* 2013; 34:724–734. [PubMed: 23111334]
29. Nelson MT, Johnson J, Lannutti J. Media-based effects on the hydrolytic degradation and crystallization of electrospun synthetic-biologic blends. *J Mater Sci Mater Med.* 2014; 25:297–309. [PubMed: 24178985]
30. Zheng R, Duan H, Xue J, Liu Y, Feng B, Zhao S, et al. The influence of Gelatin/PCL ratio and 3-D construct shape of electrospun membranes on cartilage regeneration. *Biomaterials.* 2014; 35:152–164. [PubMed: 24135269]
31. Zhang Y, Ouyang H, Lim CT, Ramakrishna S, Huang ZM. Electrospinning of gelatin fibers and gelatin/PCL composite fibrous scaffolds. *J Biomed Mater Res B Appl Biomater.* 2005; 72:156–165. [PubMed: 15389493]
32. Bottino MC, Yassen GH, Platt JA, Labban N, Windsor LJ, Spolnik KJ, et al. A novel three-dimensional scaffold for regenerative endodontics: materials and biological characterizations. *J Tissue Eng Regen Med.* 2013 Mar 8. [Epub ahead of print].
33. Palasuk J, Kamocki K, Hippenmeyer L, Platt JA, Spolnik KJ, Gregory RL, et al. Bimix Antimicrobial Scaffolds for Regenerative Endodontics. *J Endod.* 2014; 40:1879–1884. [PubMed: 25201643]
34. Bottino MC, Kamocki K, Yassen GH, Platt JA, Vail MM, Ehrlich Y, et al. Bioactive nanofibrous scaffolds for regenerative endodontics. *J Dent Res.* 2013; 92:963–969. [PubMed: 24056225]
35. ISO 10993-5. Biological evaluation of medical devices. Part 5: Tests for Cytotoxicity: *In vitro* methods. 1993
36. Bottino MC, Coelho PG, Henriques VA, Higa OZ, Bressiani AH, Bressiani JC. Processing, characterization, and in vitro/in vivo evaluations of powder metallurgy processed Ti-13Nb-13Zr alloys. *J Biomed Mater Res A.* 2009; 88:689–696. [PubMed: 18335528]
37. Dargaville BL, Vaquette C, Rasoul F, Cooper-White JJ, Campbell JH, Whittaker AK. Electrospinning and crosslinking of low-molecular-weight poly(trimethylene carbonate-co-(L)-lactide) as an elastomeric scaffold for vascular engineering. *Acta Biomater.* 2013; 9:6885–6897. [PubMed: 23416575]
38. Janotti A, Van de Walle CG. Fundamentals of zinc oxide as a semiconductor. *Rep Prog Phys.* 2009; 72:126501.
39. Oraby MA, Waley AI, El-Dewany AI, Saad EA, Abd El-Hady BM. Electrospun gelatin nanofibers: effect of gelatin concentration on morphology and fiber diameters. *J Appl Sci Res.* 2013; 9:534–540.
40. Zong X, Kim K, Fang D, Ran S, Hsiao BS, Chu B. Structure and process relationship of electrospun bioabsorbable nanofiber membranes. *Polymer.* 2002; 43:4403–4412.
41. Christopherson GT, Song H, Mao HQ. The influence of fiber diameter of electrospun substrates on neural stem cell differentiation and proliferation. *Biomaterials.* 2009; 30:556–564. [PubMed: 18977025]

42. Lee J, Tae G, Kim YH, Park IS, Kim SH, Kim SH. The effect of gelatin incorporation into electrospun poly(L-lactide-co-epsilon-caprolactone) fibers on mechanical properties and cytocompatibility. *Biomaterials*. 2008; 29:1872–1879. [PubMed: 18234330]
43. Zhang J, Kwok DY. Lattice boltzmann study on the contact angle and contact line dynamics of liquid-vapor interfaces. *Langmuir*. 2004; 20:8137–8141. [PubMed: 15350084]
44. Fu W, Liu Z, Feng B, Hu R, He X, Wang H, et al. Electrospun gelatin/PCL and collagen/PLCL scaffolds for vascular tissue engineering. *Int J Nanomed*. 2014; 9:2335–2344.
45. Mi HY, Jing X, Salick MR, Cordie TM, Peng XF, Turng LS. Properties and fibroblast cellular response of soft and hard thermoplastic polyurethane electrospun nanofibrous scaffolds. *J Biomed Mater Res B Appl Biomater*. 2014 Aug.;30. [Epub ahead of print].
46. Kim MS, Kim G. Three-dimensional electrospun polycaprolactone (PCL)/alginate hybrid composite scaffolds. *Carbohydr Polym*. 2014; 114:213–221.
47. Bottino MC, Thomas V, Jose MV, Dean DR, Janowski GM. Acellular dermal matrix graft: synergistic effect of rehydration and natural crosslinking on mechanical properties. *J Biomed Mater Res B Appl Biomater*. 2010; 95:276–282. [PubMed: 20842698]
48. Huang GP, Shanmugasundaram S, Masih P, Pandya D, Amara S, Collins G, et al. An investigation of common crosslinking agents on the stability of electrospun collagen scaffolds. *J Biomed Mater Res A*. 2014 May 15. [Epub ahead of print].
49. Vargas-Reus MA, Memarzadeh K, Huang J, Ren GG, Allaker RP. Antimicrobial activity of nanoparticulate metal oxides against peri-implantitis pathogens. *Int J Antimicrob Agents*. 2012; 40:135–139. [PubMed: 22727529]
50. Azam A, Ahmed AS, Oves M, Khan MS, Memic A. Size-dependent antimicrobial properties of CuO nanoparticles against Gram-positive and -negative bacterial strains. *Int J Nanomed*. 2012; 7:3527–3535.
51. Khurana C, Vala AK, Andhariya N, Pandey OP, Chudasama B. Antibacterial activity of silver: the role of hydrodynamic particle size at nanoscale. *J Biomed Mater Res A*. 2014; 102:3361–3368. [PubMed: 24166739]
52. Akbar A, Anal AK. Zinc oxide nanoparticles loaded active packaging, a challenge study against *Salmonella typhimurium* and *Staphylococcus aureus* in ready-to-eat poultry meat. *Food Control*. 2014; 38:88–95.
53. Espitia PJP, Soares NFF, Coimbra JSR, Andrade NJ, Cruz RS, Medeiros EAA. Zinc oxide nanoparticles: synthesis, antimicrobial activity and food packaging applications. *Food Bioprocess Technol*. 2012; 5:1447–1464.
54. Xie Y, He Y, Irwin PL, Jin T, Shi X. Antibacterial activity and mechanism of action of zinc oxide nanoparticles against *Campylobacter jejuni*. *Appl Environ Microbiol*. 2011; 77:2325–2331. [PubMed: 21296935]
55. Bhadra P, Mitra MK, Das GC, Dey R, Mukherjee S. Interaction of chitosan capped ZnO nanorods with *Escherichia coli*. *Mat Sci Eng*. 2011; 31:929–937.
56. Applerot G, Lipovsky A, Dror R, Perkas N, Nitzan Y, Lubart R, et al. Enhanced antibacterial activity of nano-crystalline ZnO due to increased ROS-mediated cell injury. *Adv Functional Mater*. 2009; 19:842–852.
57. Miura M, Gronthos S, Zhao M, Lu B, Fisher LW, Robey PG, et al. SHED: stem cells from human exfoliated deciduous teeth. *Proc Natl Acad Sci U S A*. 2003; 100:5807–5812. [PubMed: 12716973]
58. Zhang J, Kwok DY. Lattice boltzmann study on the contact angle and contact line dynamics of liquid-vapor interfaces. *Langmuir*. 2004; 14:8137–8141. [PubMed: 15350084]
59. Liu W, Su P, Chen S, Wang N, Ma Y, Liu Y, et al. Synthesis of TiO₂ nanotubes with ZnO nanoparticles to achieve antibacterial properties and stem cell compatibility. *Nanoscale*. 2014; 6:9050–9062. [PubMed: 24971593]



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Figure 1.

(a) Representative SEM micrographs for the PCL- and PCL/GEL-based membranes containing different concentrations of ZnO. Insets within the SEM micrographs show the mean fibre diameter (d_m) and standard deviation (\pm SD). (b) Representative EDS analysis of the PCL nanocomposite membrane containing 15 wt.% of ZnO confirming the presence of ZnO nanoparticles within the fibres.

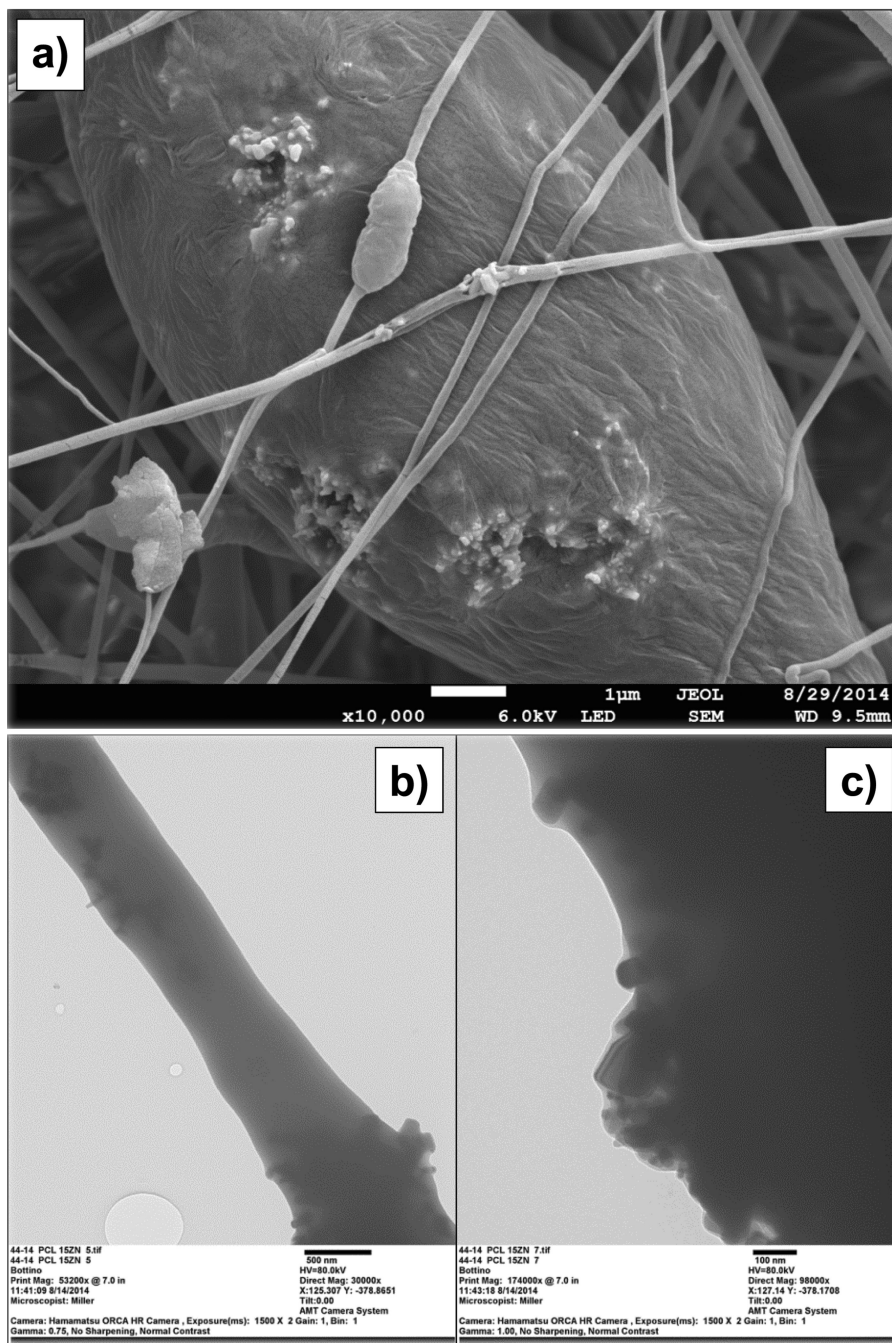


Figure 2.

(a) Representative SEM micrograph of the PCL nanocomposite membrane containing 30 wt. % of ZnO showing the accumulation/agglomeration of ZnO nanoparticles on the surface of the fibres. (b-c) TEM images showing the rough surface obtained within the ZnO-incorporated fibres at different magnifications: (b) 30,000 \times , and (c) 98,000 \times .

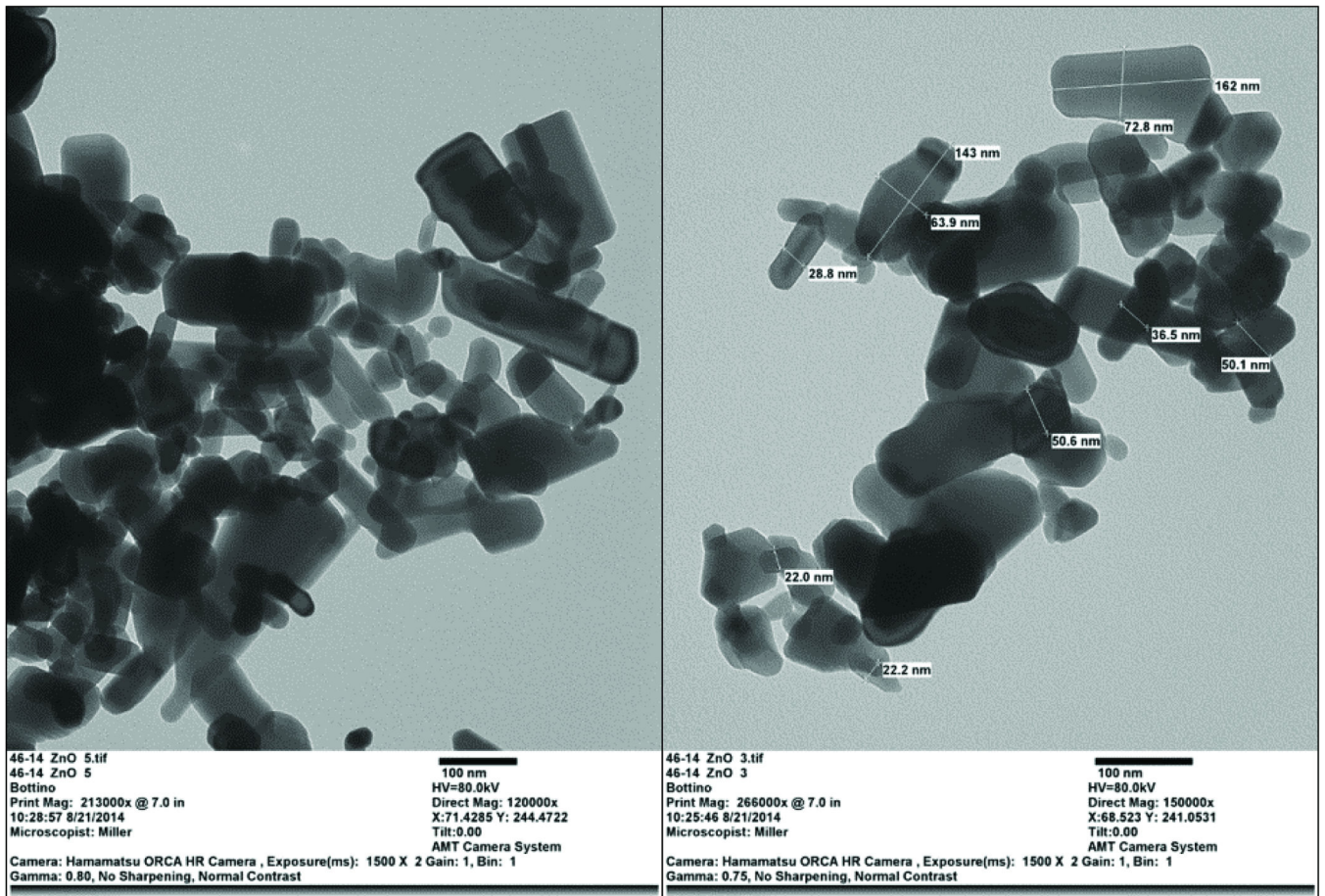
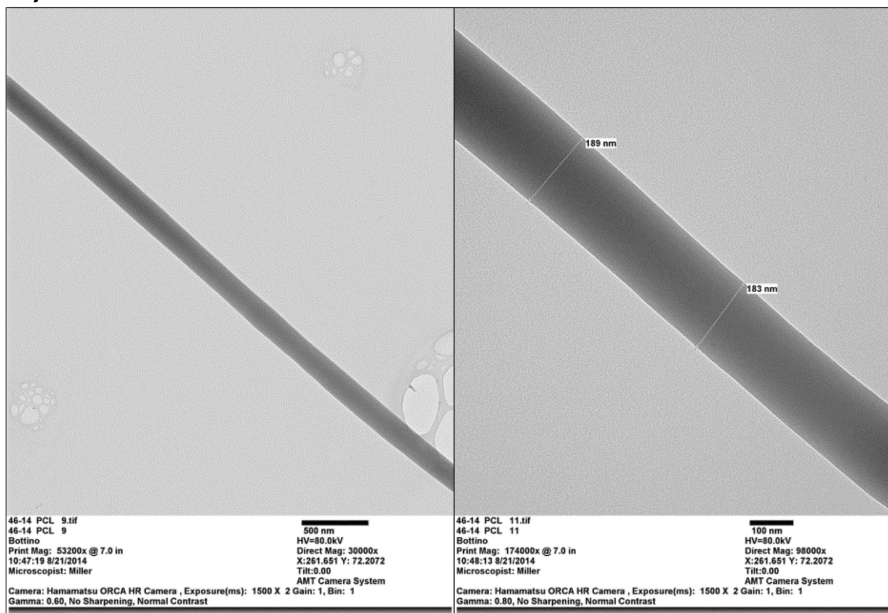


Figure 3.

TEM images showing the overall morphology and size distribution of the ZnO nanoparticles used. Image on the left and right illustrate, respectively, magnifications of 120,000 \times and 150,000 \times .

a) PCL fibers



b) ZnO-incorporated PCL fibers

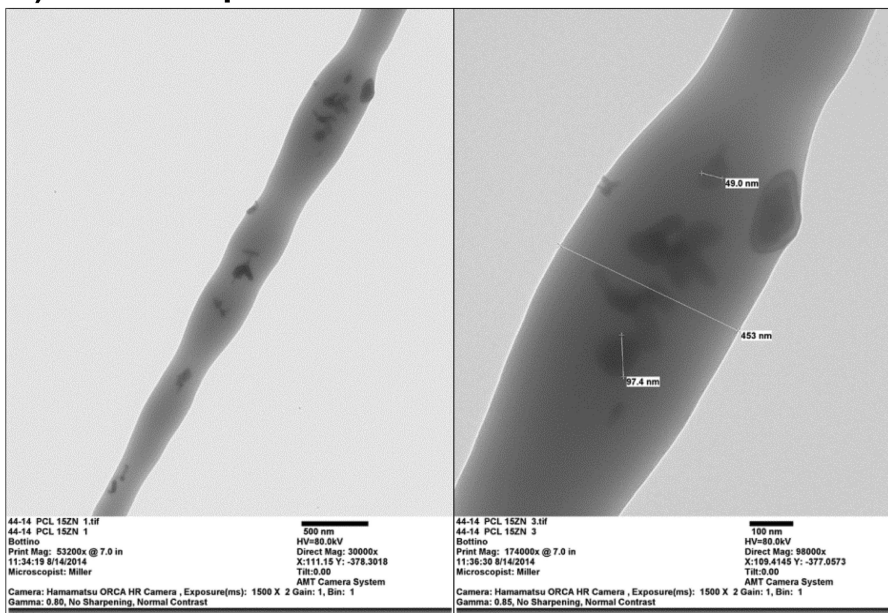


Figure 4. TEM images demonstrating, respectively, the absence and the presence of ZnO nanoparticles within the neat PCL fibres (a) and ZnO-incorporated fibres (b). Images on the left and right show, respectively, magnifications of 30,000× and 98,000×.

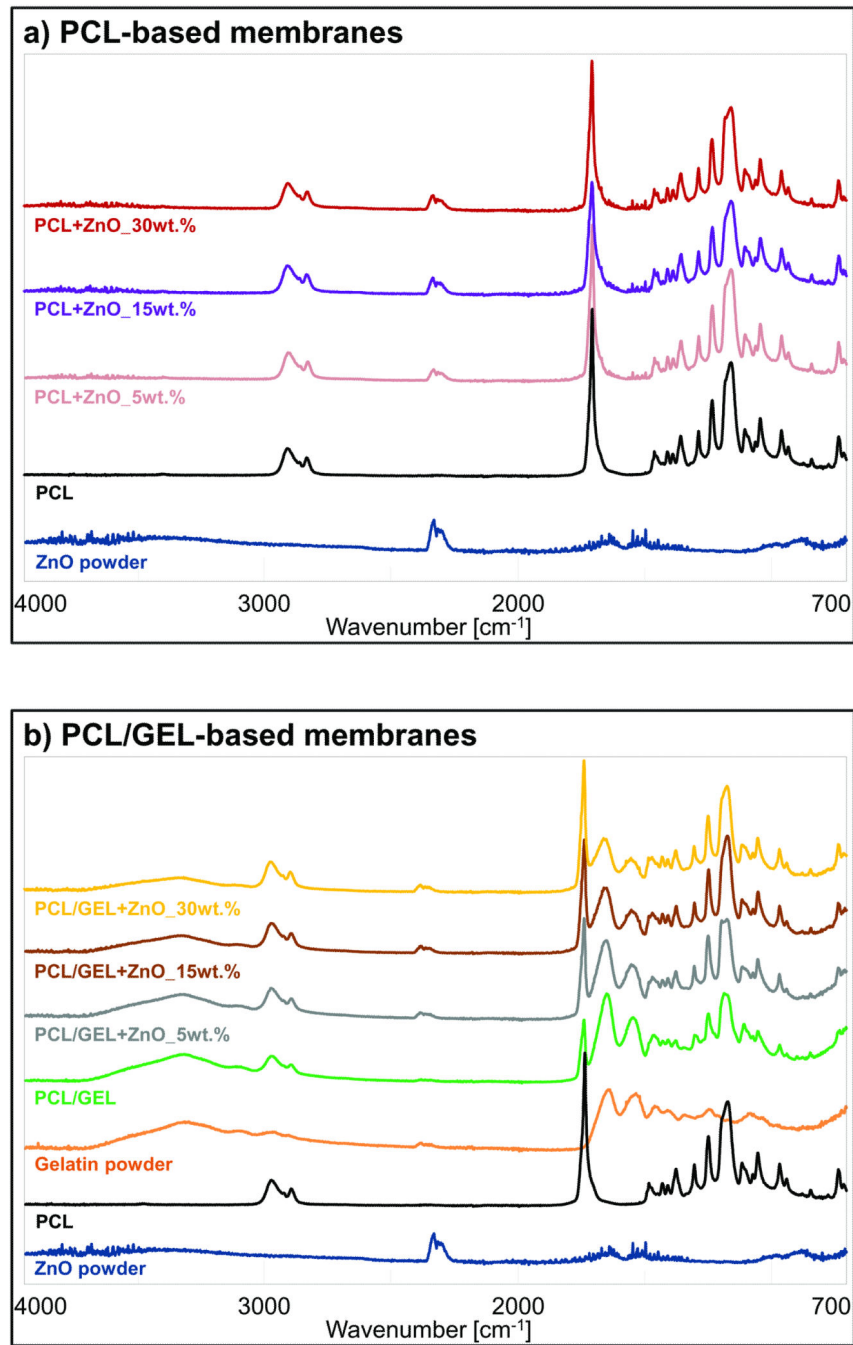
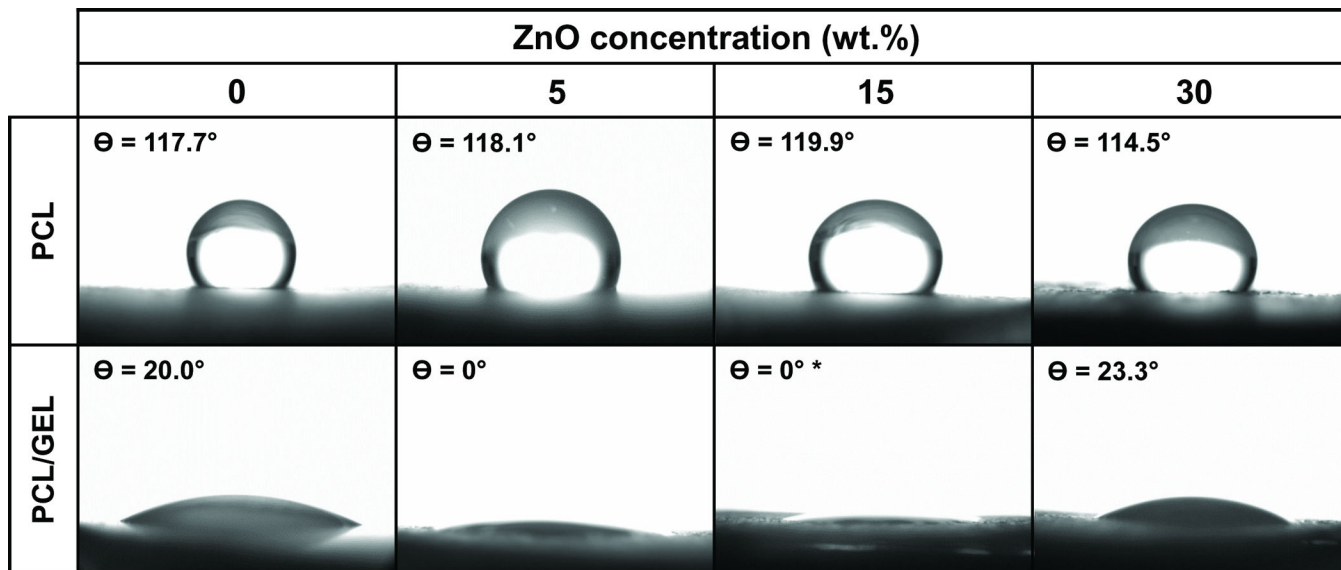


Figure 5. (a–b) FTIR spectra confirming the incorporation of ZnO and gelatin into the PCL- and PCL/GEL-based membranes.



Membrane	ZnO concentration (wt.%)			
	0	5	15	30
PCL	A 117.7° (7.6) a	A 118.1° (5.6) a	A 119.9° (6.5) a	A 114.5° (4.9) a
PCL/GEL	B 20.0° (3.7) a	B 0° (0) b	B 0° (0) b	B 23.3° (9.7) a

* Measured at time 0

Distinct uppercase letters in a same column and lowercase letters in a same line indicate statistically significant differences between membranes and ZnO content, respectively ($p < 0.05$).

Figure 6. Images of the contact angle formed between water and the PCL- or PCL/GEL-based membranes containing different concentrations of ZnO nanoparticles.

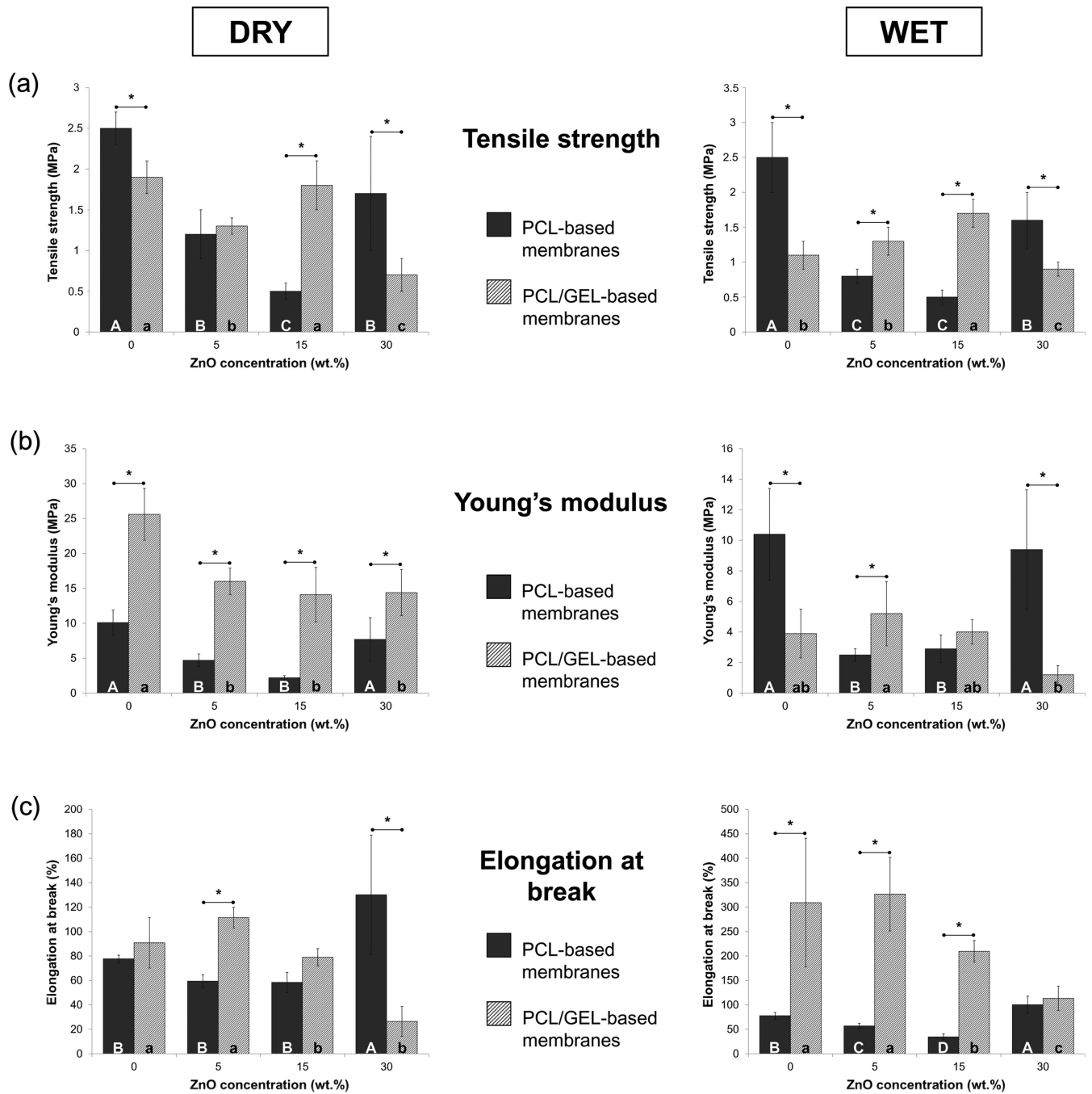
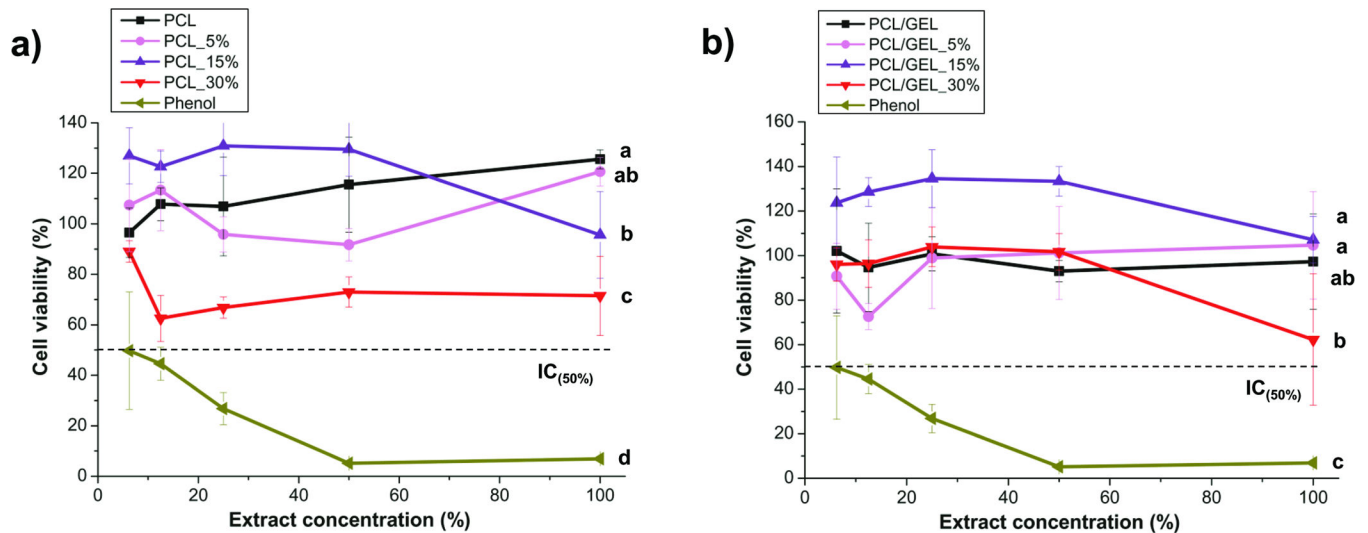


Figure 7. Mechanical characteristics of the PCL- and PCL/GEL-based membranes synthesized in the study, under dry (graphs on the left) and wet (graphs on the right) conditions: (a) Tensile strength, in MPa; (b) Young's modulus, in MPa; and (c) Elongation at break, in %.



Membranes	ZnO content (wt.%)			
	0	5	15	30
PCL	A 125.5 (±3.8) a	A 120.6 (±5.7) a	A 95.6 (±17.2) ab	A 71.5 (±15.7) b
PCL/GEL	B 97.3 (±21.4) ab	A 104.7 (±24.1) a	A 107.1 (±10.4) a	A 62.3 (±29.5) b

Distinct uppercase letters before means and lowercase letters after standard deviations (±SD) indicate, respectively, statistically significant differences between the membranes (in columns) and among zinc oxide (ZnO) content (in rows) ($p < 0.05$).

Figure 8.

Cytotoxicity assays results of the PCL- (a) and PCL/GEL-based (b) membranes and cell viability means (in %) and standard deviations (±SD) after exposure to the full-concentrated (i.e. 100% extract concentration, non-diluted) extracts (c).

Table 1

Mean (\pm standard deviation) of the minimal inhibition concentration (MIC) of the zinc oxide (ZnO) nanoparticles and antibacterial activity (inhibition halos, in mm) of the prepared electrospun membranes against *Porphyromonas gingivalis* (*Pg*) and *Fusobacterium nucleatum* (*Fn*).

Bacteria	CHX (control)	MIC, in $\mu\text{g/mL}$		
		2,500	5,000	10,000
<i>Pg</i>	^A 11.7 (± 0.6) ^a	^A 8.3 (± 1.2) ^b	^A 9.0 (± 1.0) ^b	^A 8.0 (± 0.6) ^b
<i>Fn</i>	^A 10.3 (± 0.6) ^a	^B 0.0 ^c	^B 5.3 (± 0.6) ^b	^A 9.3 (± 1.2) ^a

ZnO content (wt.%)	<i>Pg</i>		<i>Fn</i>	
	PCL	PCL/GEL	PCL	PCL/GEL
0	^C 0.0 ^a	^C 0.0 ^a	^D 0.0 ^a	^C 0.0 ^a
5	^B 6.3 (± 0.6) ^b	^B 7.7 (± 0.6) ^a	^C 6.7 (± 1.2) ^b	^B 11.3 (± 0.6) ^a
15	^{AB} 9.0 (± 1.0) ^a	^B 8.3 (± 0.6) ^a	^{BC} 9.0 (± 1.7) ^b	^B 11.7 (± 0.6) ^a
30	^A 11.7 (± 2.9) ^a	^B 8.3 (± 0.6) ^b	^A 14.3 (± 0.6) ^a	^A 13.7 (± 0.6) ^a
CHX (control)	^{AB} 10.0 (± 1.0) ^a	^A 11.0 (± 1.0) ^a	^B 9.7 (± 0.6) ^b	^{AB} 12.3 (± 0.6) ^a

Distinct uppercase letters in a same column and lowercase letters in a same row indicate statistically significant differences among groups tested ($p < 0.05$).

CHX: chlorhexidine; PCL: Poly(ϵ -caprolactone); PCL/GEL: mixture of PCL and gelatin (50:50 ratio, w/w).