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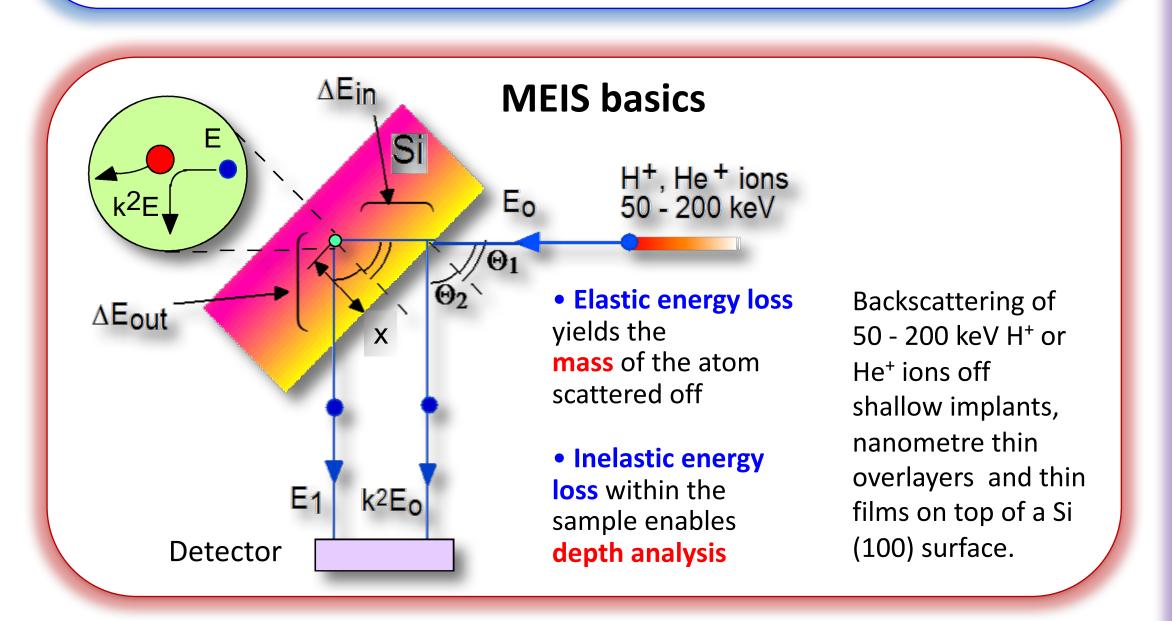
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# Quantitative considerations in medium energy ion scattering P.C. Zalm<sup>a</sup>, P. Bailey<sup>a</sup>, M. A. Reading<sup>b</sup>, A. K. Rossall<sup>a</sup>, and J. A. van den Berg<sup>a\*</sup>.

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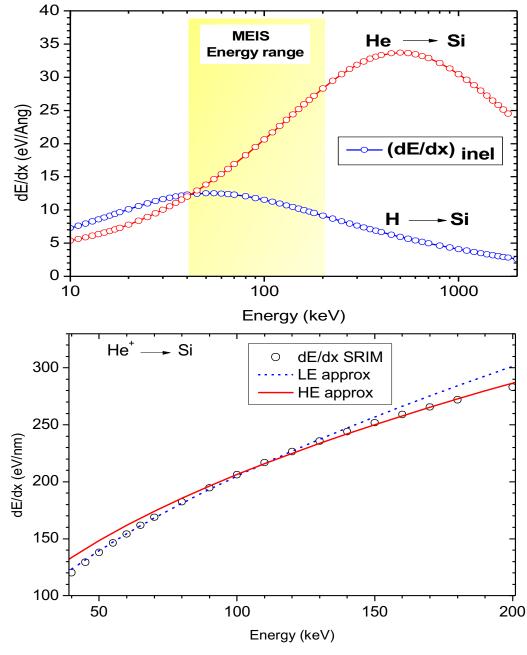
# Introduction

Due to its unique capability of providing **near-quantitative compositional and layer** structure information during depth profiling analysis, in favourable cases, with subnanometre resolution, medium energy ion scattering (MEIS) is becoming increasingly important to the characterisation of microelectronic device structures in which scaling laws have demanded the growth and doping of layers of nanometre thickness. Here we assess the quantitative accuracy in terms of both depth and concentration, that can be achieved in MEIS depth profiling.



### Inelastic energy loss to depth scale conversion

Depth scales in MEIS are derived from the inelastic energy loss of the scattered projectiles on the way in and out.



The energy loss rate dE/dx is dependent on the energy of the projectile and can be obtained from SRIM calculations, shown for for H<sup>+</sup> & He<sup>+</sup>.

For the energies used in MEIS the electronic loss rates can be well approximated by a power law:

$$-\frac{dE}{dx} = AE^B \tag{1}$$

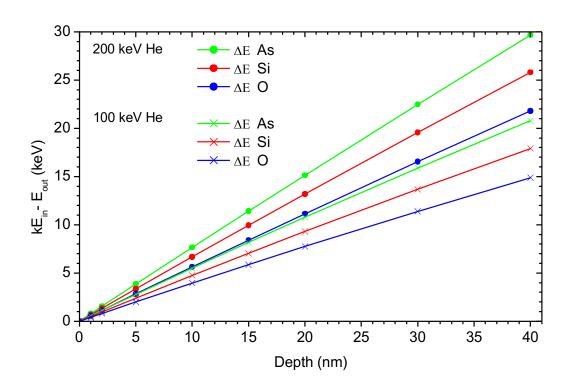
e.g. low E approx. (40-100 keV) A= 0.043, B = 0.58 (1%)

or: high E approx. (80-200 keV) A= 0.0248, B= 0.46 (1%)

NB. dE/dx in keV/nm, E in keV.

Eq.(1) enables a fully analytical conversion of inelastic energy loss to pathlength:

pathlength =	$\int_{E_{start}}^{E_{finish}} \frac{1}{-dE}$	$\frac{d}{dx} dE =$	$=\frac{1}{(1-B)A}\Big[E_{start}^{1-B}-$	$E_{finish}^{1-B}$	(2)
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Example: 100 & 200 keV He<sup>+</sup> ions incident on a <sup>28</sup>Si (100) surface, containing O and As;  $\Theta = 90^{\circ}$ ;  $\Theta_1 = 35.26^\circ \& \Theta_2 = 54.74^\circ.$ 

Apply i) eq.(2) on the way in, then ii) the kinematic factor K ( $\Theta = 90^{\circ}$ )

 $K = (M_{\text{target}} - M_{i\text{on}}) / (M_{\text{target}} + M_{i\text{on}})$ 

and finally iii) eq. (2) on the way out, This gives *kE*<sub>in</sub>–*E*<sub>out</sub> vs **depth** to 40nm (using a Si density of 5 x  $10^{22}$  at/cm<sup>3</sup>)

NB.

• The near-linear relationship between  $KE_{in}$ - $E_{out}$  & depth (for depths  $\leq$ 40 nm) • Substantial differences between slopes for Si , As and O by up to 15%

Approach is also valid for compounds. For multilayered samples, A will be different

for each layer but B changes little; but now computer simulations are required.

### **Peakheights ratios**

The absolute backscattering yield of target atoms at a depth x in a homogeneous target is according to Chu et al. (Backscattering Spectrometry, 1978)

 $H(E_{out}) = \sigma(E) \ \Omega \ \Phi \frac{D}{[\varepsilon(E)] \cos \theta_{in}} \ \frac{\varepsilon(KE)}{\varepsilon(E_{out})}$ 

the energy immediately before scattering at that depth,  $\sigma(E)$  the energy-dependent

of incident particles, D the detector energy bin or channel width,  $\varepsilon$  the stopping cross

 $[\varepsilon(E)] = \frac{K}{\cos\theta_{in}} \left. \frac{1}{N} \frac{dE}{dx} \right|_{F} + \frac{1}{\cos\theta_{out}} \left. \frac{1}{N} \frac{dE}{dx} \right|_{F}$ 

 $\boldsymbol{D} = [\boldsymbol{\varepsilon}(\boldsymbol{E})]\boldsymbol{N}\boldsymbol{\tau}$ 

N is the atom density and  $\tau$  the thickness of the surface layer across which an ion will

lose the energy equivalent to the width of a single bin within the detection system.

For scattering at the top surface  $E = E_0$  and  $E_{out} = KE_0$  so that the final term in eq. (3)

section, and K the kinematic factor for the scattering geometry adopted.

 $[\varepsilon(E)]$  is called the stopping cross-section factor (Chu et al.):

disappears; the yield off the surface is then:

where  $E_{out}$  is the energy at the detector after scattering off a target atom at depth x, E is

Rutherford scattering cross section,  $\Omega$  the detector acceptance angle,  $\Phi$  the total number

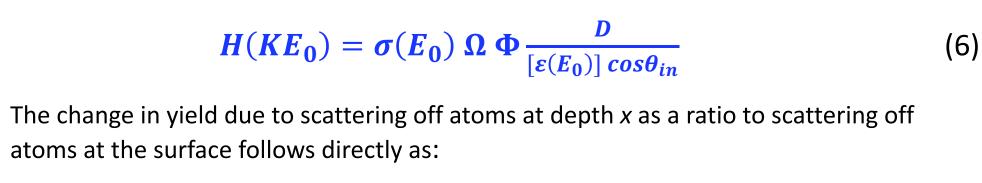
(3)

 $Z_i$  and  $Z_t$  are the atomic numbers,  $\varphi(r/a)$  is the so-called screening function and *a is* the screening length, that depends on  $Z_i$  and  $Z_t$ . At r = 0,  $\varphi = 1$  (Coulomb pot.). Expanding  $\varphi$  near r = 0 to first order in a Taylor series yields:

with  $\dot{\phi}(0)$  the gradient of the screening function r = 0. Inserting eq. (10) into eq. (9) gives:

 $\dot{\phi}(0)$  is negative; screening *decreases* the Coulomb potential by a constant amount. Andersen et al. incorporated this by *increasing* the kinetic energy of the projectile in the CoM system by exactly the same amount.

0.85 0.80



$$\frac{H(E_{out})}{(H(KE_0))} = \frac{\sigma(E)}{\sigma(E_0)} \cdot \frac{[\varepsilon(E_0)]}{[\varepsilon(E)]} \cdot \frac{\varepsilon(KE)}{\varepsilon(E_{out})}$$
(7)

as long as the detector channel width D is independent of energy (as in RBS - surface barrier detectors). Factors like  $\Omega$ ,  $\Phi$  and  $\cos\theta_{in}$  have cancelled in eq. (7). The Rutherford cross section is  $\sigma \propto E^{-2}$ . The scattering geometry dependent stopping cross section factors [ɛ], are defined in eq. (4); again constants such as the atomic density are ignored in this analysis which is permissible since only ratios of either  $\varepsilon$ 's or  $[\varepsilon]$ 's are considered here. The benefit of the power law eq.(1) is now clear since:

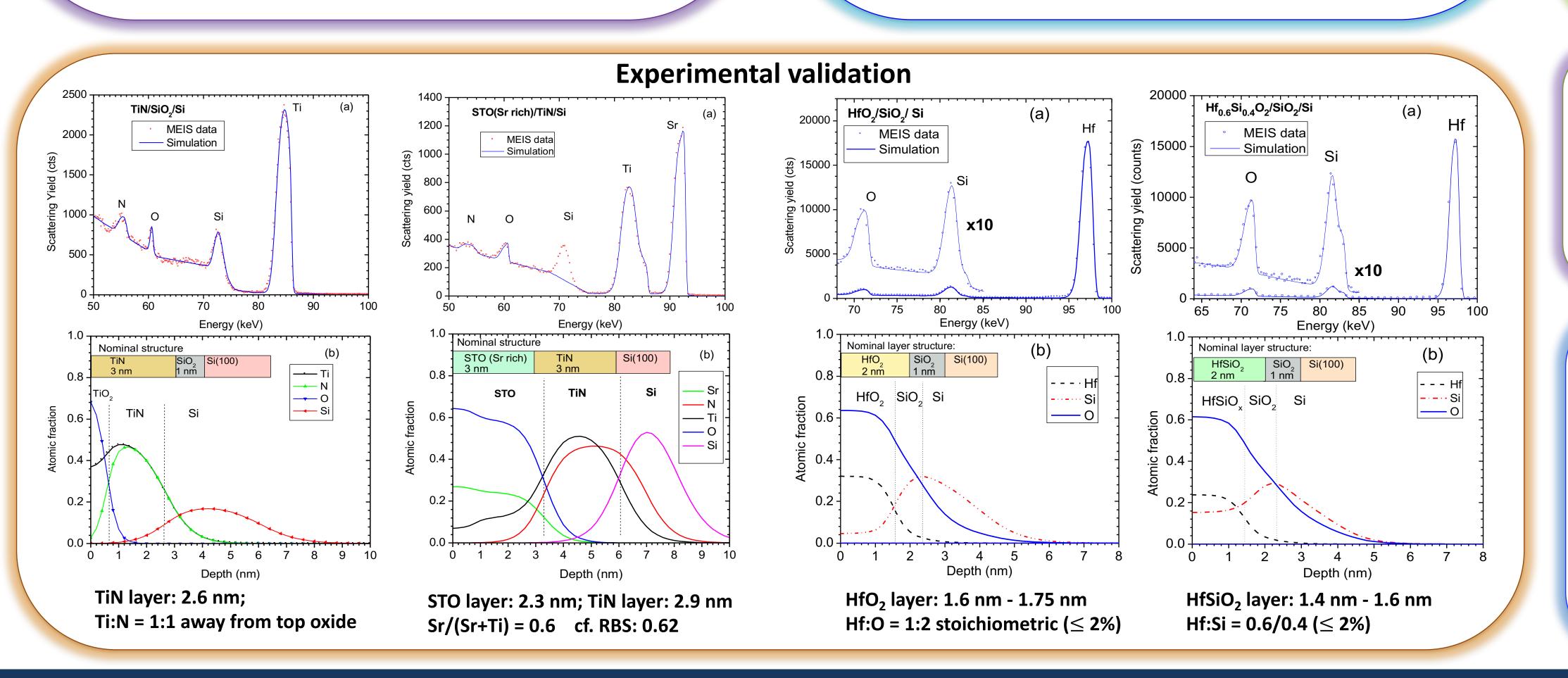
$$\frac{[\varepsilon(E_0)]}{[\varepsilon(E)]} = \left(\frac{E_0}{E}\right)^B \text{ and } \frac{\varepsilon(KE)}{\varepsilon(E_{out})} = \left(\frac{KE}{E_{out}}\right)^B$$
  
uces to:  
$$\frac{H(E_{out})}{H(KE_0)} = \left(\frac{E_0}{E}\right)^2 \cdot \left(\frac{KE_0}{E_{out}}\right)^B$$

so that eq. (7) redu

The term *D* is given as:

I.e. deviations from the energy dependence of the yield according to the Rutherford prediction (1<sup>st</sup> term eq. (8)) scale with the inverse ratio of the energies at the detector (*E<sub>out</sub>*) to a power  $B \approx \frac{1}{2}$  (2<sup>nd</sup> term eq. (8)). B depends mainly on energy, little on material type. Acquisition software makes D independent of E in our MEIS instrument.

Using eq.2 for the previous example we get: H(x)/H(0) = 1 + 0.0055x i.e < 1% /nm Slope varies with geometry but is almost independent of atom type.



(4)

(5)

(8)

### **Screening correction**

In the MEIS energy regime the nuclei are partially screened during the interaction. The Rutherford scattering cross section is not exactly valid and need a correction. Following Andersen [Phys Rev A21 (1980)1891] for a screened Coulomb potential:

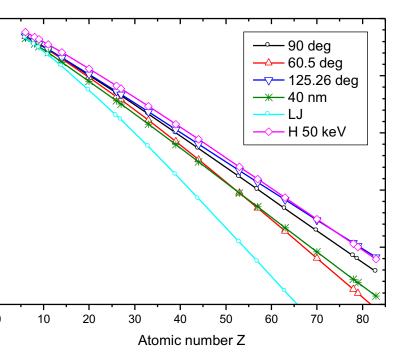
$$V(r) = \frac{Z_i Z_t e^2}{r} \phi\left(\frac{r}{a}\right)$$

$$\varphi\left(\frac{r}{a}\right) \approx 1 + \frac{r}{a} \dot{\varphi}(0)$$
 (10)

$$V(r) = \frac{Z_i Z_t e^2}{r} + \frac{Z_i Z_t e^2}{a} = V_{coulomb} + V_{const}$$
(11)

*E* is now replaced by  $E+V_{const}$  in the Rutherford cross section. The net effect is to decrease the scattering cross section progressively with increasing  $Z_r$ . In the CoM system the screening correction factor P<sub>scr</sub> can now written as:

$$P_{scr} \equiv \frac{\sigma(E_0 + V_{const})}{\sigma(E_0)} = (1 + \frac{V_{const}}{E_0})^{-2} \approx 1 - f \frac{Z_i Z_t}{E_0}$$
(12)



Example: • 100 keV He<sup>+</sup> ions scattered off surface C to Bi (Z<sub>t</sub>= 6-83) @ 60.5°, 90° and 125.26°.

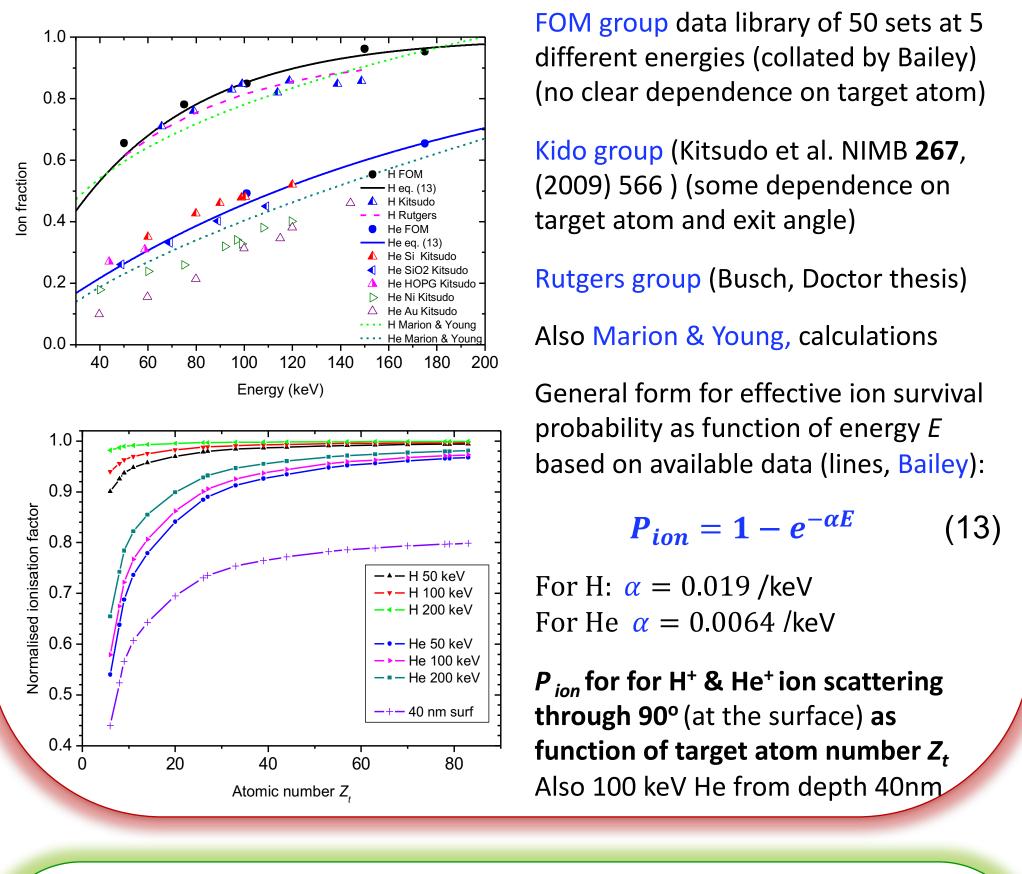
• 100 keV He<sup>+</sup> ions @ 90° off depth 40 nm

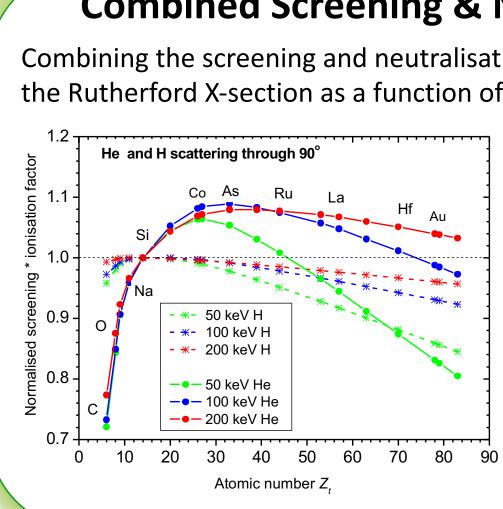
• 50 keV H<sup>+</sup> ions @ 90<sup>o</sup>

using the Biersack-Ziegler universal potential (5x) and Lens-Jensen (L J) 1x.

•  $P_{scr}$  depend almost linearly on  $Z_t$ •  $P_{scr}$  is similar for  $Z_t/E_0$  = constant (cf. 50 keV H<sup>+</sup> similar to 100 keV He<sup>+</sup>). • f is almost constant and depends mainly on choice of potential (e.g. L J) • f depends weakly on scattering angle and depth. •  $f = 0.16 \pm 0.04$  (max intrinsic error)

In MEIS the velocities of the H<sup>+</sup> or He<sup>+</sup> ions are moderate and a fraction of the scattered projectiles leave the surface neutralised. When using electrostatic analysers (unlike for TOF ones) these particles are not counted, but cannot be ignored in quantitative analysis. Fractions of particles leaving ionised are not generally accurately known. Shown below are data measured by various groups for H and He projectiles as as a function of energy.





### Spectrum simulation & Experimental confirmation

More complex layer systems can only be effectively interpreted using computer simulation of spectra. The simulation model used at IIAA, based on the IGOR graphing software incorporates the corrections considered above. Its outputs are the fitted spectra and best fit depth profiles of species in a layer. In all cases depth scales have been used which require assumptions re the correct densities of atom species. Layer thickness and composition compare well with reference data, as shown



### **Neutralisation correction**

### **Combined Screening & Neutralisation correction**

Combining the screening and neutralisation corrections gives correction curves for the Rutherford X-section as a function of target atom number, normalised to 1 for Si

> Correction curves for 50, 100 and 200keV H<sup>+</sup> & He<sup>+</sup> ions scattered through 90° (at surface) off target atom number Z,

For H: correction  $\leq 10\%$ For He: correction  $\leq 20\%$ 

**Overall correction is mass and energy** dependent and needs to be included to vield quantitative data.

## Conclusions

the **MEIS yield ratio** of particles scattered off surface atoms to those at depth x is given by the Rutherford E<sup>-2</sup> prediction \* the inverse ratio of the particle exit energies to a power B The impact of screening of the potential on the backscattering yield in MEIS has been evaluated for H<sup>+</sup> and He<sup>+</sup> ions at different energies using the Andersen approach. For a fixed beam energy the screening correction depends mainly and almost linearly on Z, Using **experimental neutralisation data** convolved with the **screening correction** leads to a correction factor to the Rutherford backscattering cross section ratio. Its dependence on both projectile energy and the atomic number of the scattering atom is presented. This **approach has been validated** for a number of representative examples that yield quantitative depth profiles derived from MEIS spectra of nanolayers. Although absolute quantification especially when using He ions, may not always be

achievable, relative quantification in which all species in a layer add up to 100%, is.