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Temperature dependent photoreflectance study of Cu_2SnS_3 thin films produced by pulsed laser deposition

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Temperature dependent photoreflectance study of Cu_2SnS_3 thin films produced by pulsed laser deposition

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The energy band structure of Cu_2SnS_3 (CTS) thin films fabricated by pulsed laser deposition was studied by photoreflectance spectroscopy (PR). The temperature-dependent PR spectra were measured in the range of $T = 10\text{--}150\text{ K}$. According to the Raman scattering analysis, the monoclinic crystal structure (C1c1) prevails in the studied CTS thin film; however, a weak contribution from cubic CTS (F-43m) was also detected. The PR spectra revealed the valence band splitting of CTS. Optical transitions at $E_A = 0.92\text{ eV}$, $E_B = 1.04\text{ eV}$, and $E_C = 1.08\text{ eV}$ were found for monoclinic CTS at low-temperature ($T = 10\text{ K}$). Additional optical transition was detected at $E_A^C = 0.94\text{ eV}$, and it was attributed to the low-temperature band gap of cubic CTS. All the identified optical transition energies showed a blueshift with increasing temperature, and the temperature coefficient dE/dT was about 0.1 meV/K . *Published by AIP Publishing.* [<http://dx.doi.org/10.1063/1.4990657>]

There is a continuous search for alternative absorber materials with an excellent environmental profile for highly efficient photovoltaics. One possible candidate is Cu_2SnS_3 (CTS) that is a direct-bandgap p-type semiconductor with a high absorption coefficient (10^4 cm^{-1}) consisting of abundantly available and non-toxic elements. The best solar cell efficiency obtained from monoclinic CTS is 4.63% .¹

Cu_2SnS_3 has been reported to crystallize in three different polymorphs: cubic (F-43m),^{2,3} tetragonal (I-42m),^{2,4} and monoclinic (C1c1)^{5–8} depending on the synthesis temperature where higher temperature leads to a higher degree of symmetry of the unit cell. These polymorphs are barely distinguishable by XRD, whereas Raman spectroscopy has been proved to be more sensitive for determining the phase composition of CTS. It has been shown that CTS synthesized at temperatures below $500\text{--}550^\circ\text{C}$ adopts a cubic or a tetragonal crystal structure, while at higher temperatures, a monoclinic structure is obtained.^{2,5,9} Additionally, other groups have mainly observed the monoclinic phase at $520\text{--}570\text{ C}$,^{7,10,11} but it has also been reported at lower temperatures,⁹ and the exact temperature might depend on the state of the precursors and the details of the annealing system. Monoclinic CTS absorbers for solar cells have to be grown Cu-poor to avoid alloying with metallic Cu_3SnS_4 ,¹² and at the same time, the Sn-rich composition may lead to the formation of $\text{Cu}_2\text{Sn}_3\text{S}_7$.¹³

The theoretical density functional theory (DFT) calculations predict a triple optical absorption onset for monoclinic CTS,^{10,14} indicating a valence band splitting. From the DFT calculations, de Wild *et al.*¹⁰ found the theoretical bandgap value of 0.76 eV for monoclinic CTS and the valence band splitting at the Γ -point between the topmost and the second valence band of $\Delta_1 = 0.10\text{ eV}$ and between the topmost and

the third valence band of $\Delta_2 = 0.15\text{ eV}$. Experimentally obtained results, calculated from the absorption spectrum, were slightly different. The room temperature bandgap energy value for monoclinic CTS was found to be $E_A = 0.96\text{ eV}$, whereas the onset between the main and the second and third uppermost valence bands of $\Delta_1 = 0.11\text{ eV}$ and $\Delta_2 = 0.16\text{ eV}$, respectively, is in good accordance with theoretical calculations. Theoretical calculations by using the DFT method of the CTS band structure have also been done by Crovetto *et al.*¹⁴ They reported a double absorption onset in monoclinic CTS and showed that the double onset is related to the intrinsic nature of this compound and not to other CTS phases or compounds. They found a theoretical bandgap value of $E_A = 0.83\text{ eV}$ for monoclinic CTS and the splitting between the main and the second and third uppermost valence bands of $\Delta_1 = 0.12\text{ eV}$ and $\Delta_2 = 0.16\text{ eV}$, respectively. These values are in good accordance with the results obtained by de Wild.¹⁰ A similar bandgap value of $E_A = 0.84\text{ eV}$, found from theoretical hybrid nonlocal exchange-correlation functional (HSE) calculations, has been reported also by Zhai *et al.*¹⁵ Crovetto *et al.*¹⁴ have also reported the experimental values of optical transition energies determined from absorption spectra at room temperature. Their results showed the bandgap energy $E_A = 0.91\text{ eV}$ and the second optical transition at 0.99 eV , which is related to E_B transition, while the theoretically predicted third transition E_C was not seen in this study. Similar to de Wild's results, theoretical transition energies calculated by Crovetto *et al.*¹⁴ are much smaller compared with experimental results. As it was mentioned before, CTS has three different polymorphs: besides monoclinic, there are also a cubic and a tetragonal crystal structure that have bandgap energy values higher than monoclinic CTS. Fernandes *et al.*¹⁶ have reported an experimental bandgap value of $E_A^C = 0.96\text{ eV}$ for cubic CTS and $E_A^T = 1.35\text{ eV}$ for tetragonal CTS. On the other hand, Guan *et al.*³ have found bandgap energies of

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$E_A^C = 1.19$ eV and $E_A^T = 1.05$ eV, for cubic and tetragonal CTS, respectively. The contradictory data about the band gap energies of the different polymorphs of CTS available in the literature were one of the main motivations for this study.

In this report, the modulation spectroscopy, namely, a photoreflectance method, is used to study the energy band structure of CTS. Photoreflectance (PR) spectroscopy has proven to be a useful tool to characterize the optical properties of semiconductors, especially the bandgap energy and valence band splittings.^{17–20}

The thin film of CTS with a thickness of 80 nm was grown by pulsed laser deposition (PLD) on a Mo covered glass substrate with 7 ns laser pulses at 355 nm from a Q-switched, frequency-tripled 1064 nm Nd:YAG laser (Coherent). The laser incidence angle on the target was 45° , the fluence was 1.6 J/cm², and the target-substrate distance was 4 cm. The fluence had been adjusted to yield the correct Cu:Sn ratio in the films.²¹ The target was a multiphase sintered pellet made from binary sulfides mixed for an average target composition of 2Cu:Sn:3S (PVD products; manufactured by Testbourne, Ltd.). The target sintering temperature was about 750°C , and the resulting pellet had microdomains of copper sulfides, tin sulfides, and copper tin sulfides. The deposition took place at room temperature in vacuum ($p < 3 \times 10^{-6}$ mbar). The target was rotated while the laser beam was moved in a raster pattern to improve film uniformity. In order to obtain the monoclinic CTS, the film was annealed with excess S in a graphite box inside a tube furnace with an N₂ atmosphere at 570°C for 10 min using a slow ramp-up rate of $2^\circ\text{C}/\text{min}$ to 570°C and natural cooling. Before annealing, the material is an amorphous mixture of Cu, Sn, and S according to XRD and energy dispersive x ray (EDX) on samples made in the same way as the ones studied here.

Before the PR spectroscopy study, Raman measurements were made at room temperature with a high-resolution micro-Raman spectrometer Horiba Jobin Yvon HR800 equipped with a multichannel CCD detection system in the backscattering configuration. The incident Nd-YAG laser light with $\lambda = 532$ nm and a spot size of $10 \mu\text{m}$ in diameter was used for excitation. The PR measurements were made with a traditional setup,¹⁸ where the $f = 64$ cm grating monochromator together with a 250 W halogen bulb was used as a primary beam and a 70 mW He-Cd laser ($\lambda = 441$ nm) as a secondary beam. The reflectance signal at 120 Hz was detected using a Ge detector with a lock-in amplifier in the spectral range from 0.85 to 1.15 eV. In order to make low-temperature PR measurements, the CTS sample under study was glued with a cryogenic grease onto the cold finger of the closed-cycle He cryostat.

The Raman spectrum of CTS together with the fitting result is presented in Fig. 1. The Raman spectrum was fitted using Lorentzian line shape functions to resolve the peaks since it can be seen that the dominating Raman peaks at about 290 cm^{-1} and 350 cm^{-1} have an asymmetric shape. Based on the available information in the literature about the Raman modes of CTS polymorphs,^{2,11,16} our Raman spectra indicate that the monoclinic CTS prevails in the studied film showing the most intense peak at 292 cm^{-1} and other peaks at 222 , 254 , 317 , 350 , and 371 cm^{-1} . However, the cubic

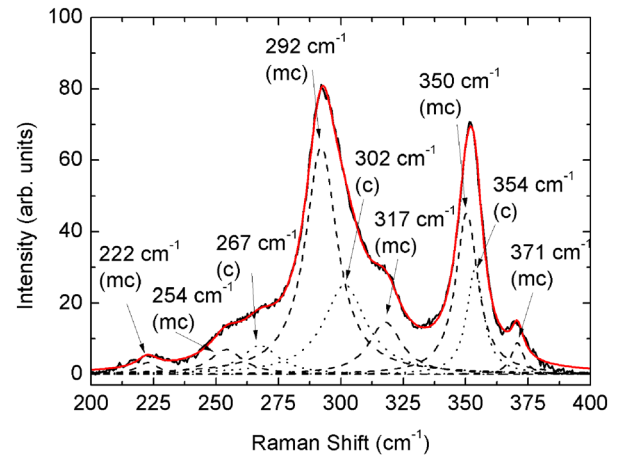


FIG. 1. Room temperature Raman spectrum of the CTS thin film together with fittings. (mc) indicates monoclinic - and (c) a cubic phase. The red line is the sum of the fitting.

CTS phase was also found with the corresponding, weaker Raman peaks detected at 267 , 302 , and 354 cm^{-1} .^{11,16}

For determining the energy band structure of CTS, photoreflectance spectroscopy was used. In PR spectroscopy, a periodic perturbation, in our case the modulated laser beam, produces sharp derivative-like features in the optical response of the sample, making this method very useful for detecting transitions having very weak oscillator strengths. The derivative nature of the spectra enhances the features in the energy region of interband transitions at the critical point in the Brillouin zone. For CTS, the valence band states arise due to the hybridization of the Cu 3d orbitals and the S 3p orbitals. The conduction band minimum is composed of Sn 5s and S 3p components.²² At the Γ point, the degeneracy of the three uppermost valence bands is lifted due to the combination of a crystal field and a spin orbit splitting. The first is induced by the lowered symmetry compared to the zinc blende structure which mainly stems from the tetragonal distortion of the unit cell due to the anion displacement. Photoreflectance measurements enable us to detect optical transitions related to the three valence bands.

The photoreflectance spectrum measured at 10 K together with fittings is shown in Fig. 2. The three derivative-like features corresponding to the multiple optical transitions of CTS that result from the band gap and valence band splittings are clearly visible in the reflectance spectrum. A weak feature at around 0.94 eV was also detected. Each PR spectrum has been analyzed by the low-field line-shape function with a third derivative functional form, developed by Aspnes²³

$$\frac{\Delta R}{R} = \sum_{j=1}^n \text{Re} [A_j e^{i\varphi_j} (E - E_{g,j} + i\Gamma_j)^{-m}], \quad (1)$$

where E is the photon energy and A_j , φ_j , $E_{g,j}$, and Γ_j are the amplitude, phase, optical transition energy, and broadening parameter of the spectrum of the j th structure, while n is the number of critical points, in our case 4. The exponent m depends on the type of the critical point, and determining its proper value is of particular importance in analyzing PR spectra. $m = 2.5$, corresponding to a three-dimensional critical point, was used for the fitting of all PR spectra. In order

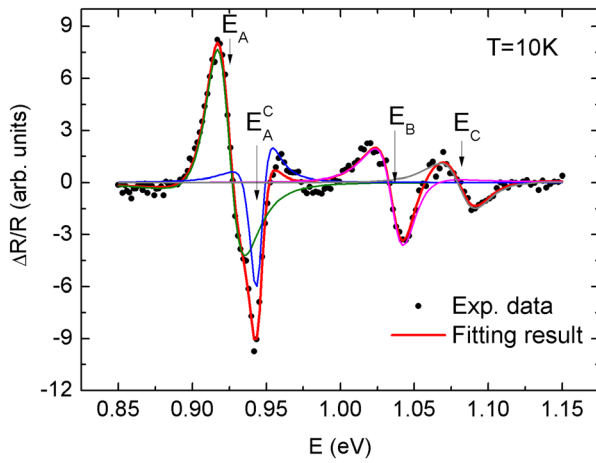


FIG. 2. Photorefectance spectrum of CTS at 10 K together with fittings with Eq. (1). From the fittings, three optical transitions of monoclinic CTS, E_A , E_B and E_C , were determined. An additional optical transition E_A^C was attributed to the band gap of cubic CTS.

to study the temperature dependencies of the observed optical transitions, PR measurements were performed in the temperature range of $T = 10\text{--}150\text{ K}$.

The detailed fitting with Eq. (1) revealed four optical transitions at $E_A = 0.92\text{ eV}$, $E_B = 1.04\text{ eV}$, $E_C = 1.08\text{ eV}$, and $E_A^C = 0.94\text{ eV}$. The latter optical transition E_A^C is attributed to the bandgap energy of cubic CTS since the Raman spectra revealed the presence of the cubic CTS phase in the studied film. Indeed, Fernandes *et al.*¹⁶ have found that the cubic phase of CTS has a bandgap of $E_A^C = 0.96\text{ eV}$. A slightly higher value of $E_A^C = 0.98\text{ eV}$ has been reported by Zhai *et al.*¹⁵ According to the literature, we can attribute E_A , E_B , and E_C to the bandgap and valence band splitting of the monoclinic CTS, and the band structure can be seen in Fig. 3. Experimentally determined offsets between optical transition energies E_A , E_B , and E_C were $\Delta_1 = 0.12\text{ eV}$ and $\Delta_2 = 0.16\text{ eV}$, and they are in good accordance with the results of the theoretical calculations.^{10,14}

Interestingly, we detected a shift of the PR spectrum towards higher energies with increasing temperature, indicating the widening of the band gap of CTS in the range of $T = 10\text{--}130\text{ K}$, as can be seen in Fig. 4. Measurements at

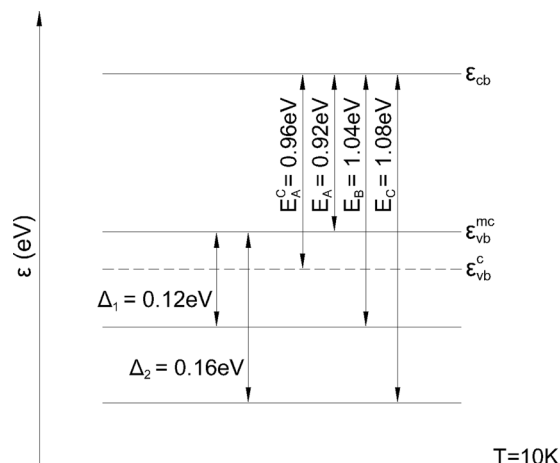


FIG. 3. Band diagram of studied CTS at 10 K, together with optical transition energies and onsets.

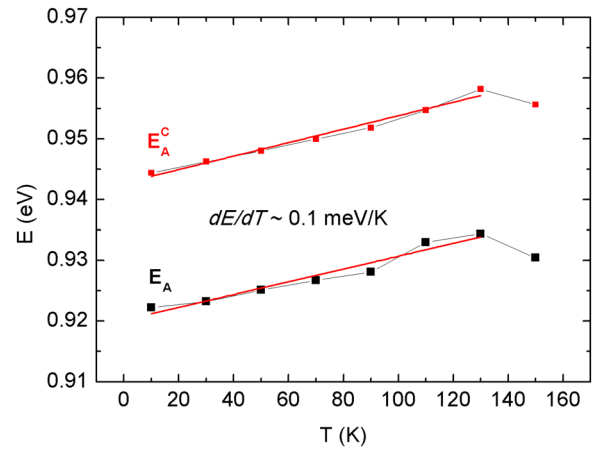


FIG. 4. Temperature dependence of the bandgap energies E_A and E_A^C , together with a linear fit (red line).

higher temperatures ($T > 150\text{ K}$) were not possible due to the low PR signal. The bandgap widening with increasing temperature is not very common among the semiconductor materials; however, a similar behavior at low temperatures has also been detected for example in Ag containing ternaries,¹⁸ black phosphorus,²⁴ some chalcogenides [CuInS₂,²⁵ CuInSe₂,²⁶ and CuGaSe₂,²⁶], and hybrid perovskites.²⁷ The blueshift of the free exciton PL band of CTS with increasing temperature was also witnessed by Aihara *et al.*^{7,8} The observed widening of the energy gap appears to be the result of the competition between the thermal expansion and the electron-phonon interaction.²⁵ For example, the thermal expansion coefficient at low temperatures can become negative due to the reduction in the unit cell volume, causing a blueshift of bandgap with increasing temperature.²⁸ This was seen for example in CuBr by Lewonczuk *et al.*²⁹ However, the origin of the blueshift of the bandgap energy of CTS is not yet clarified and needs to be studied further.

In conclusion, a photorefectance study of a CTS thin film grown by PLD was performed in the temperature range from 10 to 150 K. From the PR spectra, three low-temperature ($T = 10\text{ K}$) optical transitions of monoclinic CTS at $E_A = 0.92\text{ eV}$, $E_B = 1.04\text{ eV}$, and $E_C = 1.08\text{ eV}$ were found. An additional optical transition at $E_A^C = 0.94\text{ eV}$ was attributed to the band gap of cubic CTS. All the identified optical transition energies showed the blueshift with increasing temperature, and the temperature coefficient was $\sim 0.1\text{ meV/K}$.

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