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The pH-dependent adhesion of nanoparticles to self-assembled monolayers on gold

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Abstract

The effect of pH on the adhesion of silica and polystyrene latex nanoparticles, presenting hydroxyl and carboxyl acid surface chemistries respectively, to self-assembled monolayers (SAMs) has been investigated. The SAMs studied were 1-dodecanethiol, 11-mercaptoundecanoic acid and an original pyridine-terminated SAM. Adhesion of nanoparticles to the SAMs was found to decrease with increasing pH due to increased repulsive forces between surfaces, as a result of the deprotonation of surface moieties on the nanoparticles. A range of surface morphologies for the adsorbed nanoparticles was observed for the systems studied.

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Keywords: Self-assembled monolayer; Nanoparticle; Adhesion; Surface chemistry; pK_a

1. Introduction

The formation of nanostructures is a prime example of the ‘bottom-up’ fabrication process and is currently a subject of significant research activity, with number of techniques typically being employed in ‘bottom-up’ processes. The deposition of nanoparticles onto surfaces is an obvious example of such research, with the ability to assemble nanoparticles into patterns and arrays being one step on the road towards the construction of nanodevices and nanofunctional materials [1]. Mendes et al. [2] discussed the challenge of understanding how ordered or complex structures can form spontaneously by self-assembly, and the problems inherent with controlling such processes. Similarly, Jonas et al. [1] discussed the possible applications of these nanodevices and nanofunctional materials,

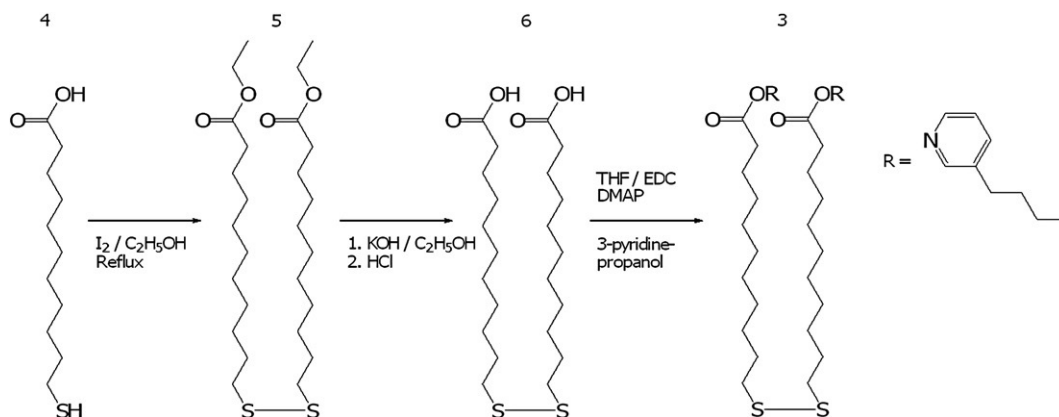
which include anti-reflective materials [3,4], biosensors [5], and superhydrophobic surfaces [6]. Other examples of ‘bottom-up’ research include the production of nanopatterned surfaces [7,8], perhaps employing nanolithographic techniques [9]. The formation of deoxyribonucleic acid-mediated artificial nano-biostructures has also been reported [10].

The selective arrangement of nanoparticles on patterned surfaces displaying two or more surface chemistries has been reported by a number of authors. For example, Krüger et al. [11] reported the pH-selective adsorption of latex nanoparticles onto photolithographically patterned silane self-assembled monolayers (SAMs), while Mendes et al. [12] reported the preferential adsorption of citrate-passivated Au nanoparticles onto NH_2 -terminated regions of a chemically modified NO_2 -terminated silane SAM, which had been patterned using e-beam lithography. Au and Ag nanoparticles have been used in this research area [13] as they can be passivated with thiols and can offer a variety of surface chemistries once passivated. For example, thiol-passivated Au nanoparticles, which were first reported by Brust et al. [14], have been employed in the

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Scheme A1. Synthetic route employed for the synthesis of SAM compound 3. Scheme uploaded electronically.

55 formation of thin films on silane SAMs, yielding structures with
56 different films providing a range of colours and reflectivities
57 [15].

58 Nanoparticles are one example of colloidal materials, or
59 colloids. Colloids are found in many aspects of new and old
60 technologies. For example, colloidal clay, polymer latex and
61 calcite particles are used in paper manufacture, each conferring
62 a different function to the finished product. Colloids are found
63 in many other day-to-day items, such as the ink in ball point
64 pens, photocopiers, paints, cosmetics and bricks, and are also
65 an important aspect of biological, medicinal and agricultural
66 systems [16]. The selective deposition of colloids onto surfaces,
67 particularly patterned surfaces, is often controlled by electro-
68 static, hydrophobic or biospecific interactions [17]. Understand-
69 ing the mechanisms behind such interactions and the variables
70 which will affect their adhesion is a key part of working towards
71 nanodevices and nanofunctional materials. The work presented
72 here investigates the deposition of colloidal nanoparticles with
73 different surface chemistries onto SAMs which also present a
74 range of surface chemistries. These studies are performed over the
75 pH range 1–11, as pH is often an important parameter in con-
76 trolling the adhesion between surfaces [7].

77 2. Experimental details



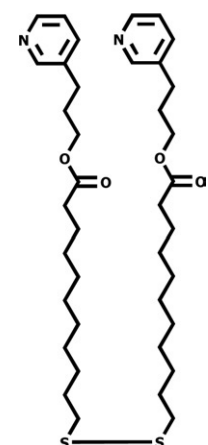
78 2.1. Chemical reagents

79 Three SAM compounds were employed for the deposition of
80 SAMs on Au. 11-mercaptoundecanoic acid (1, Sigma, UK) and
81 1-dodecanethiol (2, Sigma, UK) were used as received. An
82 original dialkyl disulfide (3) containing a terminal pyridine
83 moiety was synthesised as described in the appendix, the
84 synthetic route being shown in Scheme A1. The structures,
85 contact angle behaviour and the pK_a s (in aqueous solution) of the
86 terminal moieties of the SAMs are listed in Table 1. For the
87 pyridine-terminated SAM a prediction of the terminal moiety pK_a
88 was made using the Hammett and Taft equations for heteroaromatic
89 acids and bases [18]. The assumed terminal pK_a s for the
90 other SAMs were based on their shorter chain analogues, whose
91 pK_a s are well established, because inductive effects fall off rapidly

with distance in saturated hydrocarbons [18]. Therefore, the
92 terminal methyl moiety of a 1-dodecanethiol SAM was assumed
93 to have a pK_a of 50 in aqueous solution, analogous to ethane
94 [19]. Likewise, the terminal carboxylic acid moiety of a
95 11-mercaptoundecanoic acid SAM was assumed to have a pK_a of
96 4.75 in aqueous solution, analogous to ethanoic acid [20,21].
97

The organic solvent used for SAM formation was HPLC grade
98 ethanol (Fisher Scientific, UK). Piranha solution was used for
99 glassware cleaning and for cleaning Au slides prior to SAM
100 formation. Piranha solution was made as a 3:7 mixture of 30%
101 laboratory reagent grade hydrogen peroxide (Fisher Scientific,
102 UK) and analytical reagent grade concentrated sulfuric acid
103 (Fisher Scientific, UK). Piranha solution is a very strong oxidising
104 agent and has been known to detonate spontaneously upon contact
105 with organic material. Therefore, eye protection (Fisher Scientific,
106 UK) and nitrile gloves (Bodyguards, UK) were worn at all times,
107 and as a precaution H₂O ice was used as a quenching agent.
108

Table 1
Chemical structures of SAM compounds 1–3, their H₂O contact angle
behaviour and terminal moiety pK_a s in aqueous solution

SAM compound	1	2	3	t1.3
Compound structure				t1.4
θ_a (°)	8±3	113±2	54±4	t1.5
θ_r (°)	~0	95±2	20±4	t1.6
pK_a of terminal moiety	4.75	50	5.60	t1.7

When required for pH adjustments, NaOH solutions were made by dissolving NaOH pellets (Fisher Scientific, UK) in Ultra-High Quality (UHQ) H₂O at room temperature, followed by dilution as required. HCl solutions were made by diluting 11.65 M HCl solution (Fisher Scientific, UK) with UHQ H₂O at room temperature. All pH measurements were performed using an IQ150 pH meter (IQ Scientific Instruments) operating at room temperature.

2.2. Deposition of Au thin films and formation of SAMs

Au was deposited onto clean glass microscope slides (BDH, UK) by thermal evaporation using an Auto 306 vacuum evaporation chamber (Edwards, UK). Cr was used as an adhesion promoter, as priming the glass surface with Cr or Ti improves adhesion of Au, which has been reported by various authors when describing their preparation of Au films [8,22–30]. The chamber pressure was reduced to $\sim 10^{-5}$ Pa using a two-stage pumping system. Cr pieces of 99.99% purity (Agar Scientific, UK) were heated by electrical resistance using a voltage of 30 V and a current of 3 A until ~ 5 nm of Cr had been deposited onto the glass surface. Au wire of 99.99% purity (Advent Research Materials, UK) of 0.5 mm diameter was placed into a Mo boat (Agar Scientific, UK) and was heated by electrical resistance using a voltage of 10 V and a current of 3 A until ~ 100 nm of Au had been deposited onto the desired surface. Deposition was monitored using an *in situ* quartz crystal microbalance thickness monitor. The deposition rate for both Cr and Au was in the range 0.05–0.10 nm s⁻¹. Nitrile gloves (Bodyguards, UK) were worn during all handling procedures and Dumostar tweezers (Agar Scientific, UK) were employed to minimise contact with the samples whenever it was practical to do so. Where Au substrates were required to be cut up into smaller pieces, a diamond-tipped scribe (Agar Scientific, UK) was used. Any dust produced was blown away with Ar gas.

All glassware used in SAM formation was cleaned prior to use by immersion in piranha solution at room temperature for ~ 1 h. Cleaning with piranha solution was followed by rinsing with copious amounts of 18 M Ω deionised H₂O (Elga UHQ-PS) and drying in an oven at 140 °C. SAMs were prepared by immersing Cr-primed, Au-coated glass microscope slides in 1 mM solutions of the SAM compounds for 24 h (11-mercaptopundecanoic acid and 1-dodecanethiol) and 48 h (pyridine SAM compound), using ethanol as a solvent. All Au substrates were cleaned prior to SAM formation by immersion in piranha solution at room temperature for 10 min. Cleaning with piranha solution was followed by rinsing with copious amounts of 18 M Ω deionised H₂O (Elga UHQ-PS) and rinsing with copious amounts of ethanol. After the desired immersion time, Au substrates were removed from the SAM solution and rinsed with copious amounts of ethanol, before being blown dry using Ar gas.

2.3. SAM characterisation procedures

Characterisation of SAMs formed on Au substrates involved assessing their wetting behaviour, elemental composition and thickness, employing dynamic water contact angle measure-

ments, X-ray photoelectron spectroscopy (XPS) and ellipsometry, respectively. Figs. A1–A3 show the XPS spectra obtained for SAMs 1–3 respectively, while Table A1 lists the SAM thicknesses measured using ellipsometry.

Dynamic H₂O contact angles were measured using a home-made stage apparatus, employing a Charge-Coupled Device (CCD) KP-M1E/K camera (Hitachi) and FTA Video Analysis software v1.96 (First Ten Angstroms) for analysis of the contact angle of a droplet of UHQ H₂O at the three-phase intersection point. All data was collected at room temperature and pressure under ambient humidity conditions. A 25 μ L gastight syringe (Hamilton) was used for changing the volume of the droplet for all measurements, allowing volume adjustments of ~ 1 μ L to be performed manually, if necessary. The droplet was released onto the sample surface from a blunt-ended needle of ~ 1 mm diameter (Hamilton). Frames for the video analysis were captured at a rate of 0.12 Hz, usually yielding a minimum of ten frames for both the advancing contact angle and the receding contact angle. Mathematical analysis of the contact angle was performed assuming a non-spherical droplet shape, with manual designation of the baseline for each surface analysed. Data for the advancing contact angle were only chosen when the droplet width was increasing. Similarly, data for the receding contact angle were only chosen when the droplet width was decreasing. The calculated contact angles for each frame during the advancing or receding droplet movement were averaged to give mean values for both the advancing and receding contact angle behaviour of the surface. A minimum of 7 measurements were performed for each sample.

Ellipsometry measurements were performed using a spectroscopic ellipsometer (Jobin-Yvon/Horiba) operating with DeltaPsi2 v2.0.8 software. Ellipsometer calibration and alignment of the Polariser and Detector were performed using an Al reference sample, which has a thermally grown Al₂O₃ layer. The angle of incidence between the analyser and the polariser was set to 70° and was maintained for all subsequent measurements. The light wavelength range used for all measurements was 280–800 nm. All measurements were made under conditions of ambient temperature, pressure and humidity. SAM thicknesses are averages of a minimum of six measurements, each made at a different location on the substrate. Precautions were made to

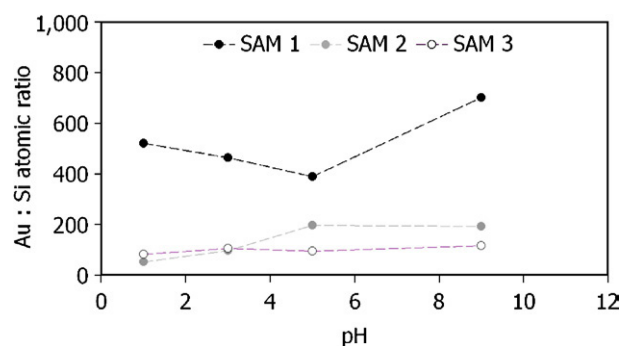


Fig. 1. XPS results for SiO₂ nanoparticles deposited onto SAMs 1–3: Au:Si peak area ratio.

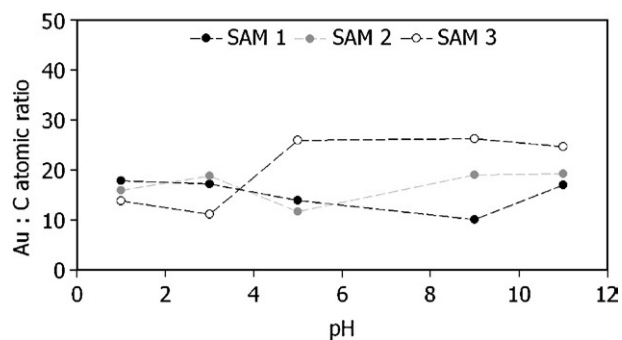


Fig. 2. XPS results for COOH-PL nanoparticles deposited onto SAMs 1–3: Au : C peak area ratio.

using a Cauchy transparent layer, whose initial thickness was 213 varied using a multiguess iterative calculation procedure. The 214 single outcome of each iteration process was the result with the 215 lowest χ^2 . A minimum of five different initial values for the 216 SAM thicknesses were chosen for each SAM measurement. 217 Those results with the lowest χ^2 for each measurement made 218 were averaged to give a mean SAM thickness. 219

XPS analysis of SAMs was performed using an Escalab 250 220 system (Thermo VG Scientific) operating with Avantage v1.85 221 software. An Al K α X-ray source was used, providing a 222 monochromatic X-ray beam with incident energy of 1486.68 eV. 223 All measurements were made at a pressure of $\sim 5 \times 10^{-9}$ mbar. A 224 circular spot size of $\sim 0.2 \text{ mm}^2$ was employed throughout all 225 measurements. Samples were immobilised onto stainless steel 226 sample holders, using both double-sided carbon sticky tape 227 (Shintron tape, Shinto Paint Company) and stainless steel or 228 copper sample clips (Thermo VG Scientific). The use of clips 229 provided conductivity between the sample surface and the 230 sample holder, because although the Au film is conductive, the 231 glass substrate is insulating. By providing a conductive link 232 between the sample surface and the sample holder, surface 233 charge retention during measurement was minimised. 234

Low resolution survey spectra were obtained using a pass 235 energy of 150 eV over a binding energy range of -10 eV to 236 1200 eV , obtained using 1 eV increments. Recorded low 237 resolution spectra would typically be an average of 5 scans. All 238

203 avoid performing measurements on visibly defective locations 204 on the sample. Mathematical modelling of the SAM thickness 205 was performed for each measurement. The SAM thickness 206 calculations were based on a three-phase ambient/SAM/Au 207 model, in which the SAM was assumed to be isotropic and 208 assigned an initial refractive index of 1.50 [31–33]. The 209 refractive index of a SAM has also been reported as 1.45 210 [34,35]. However, it was found that whether the starting value for 211 the iterative calculation process was 1.45 or 1.50, the outcome of 212 the modelling process did not vary. The SAM was modelled

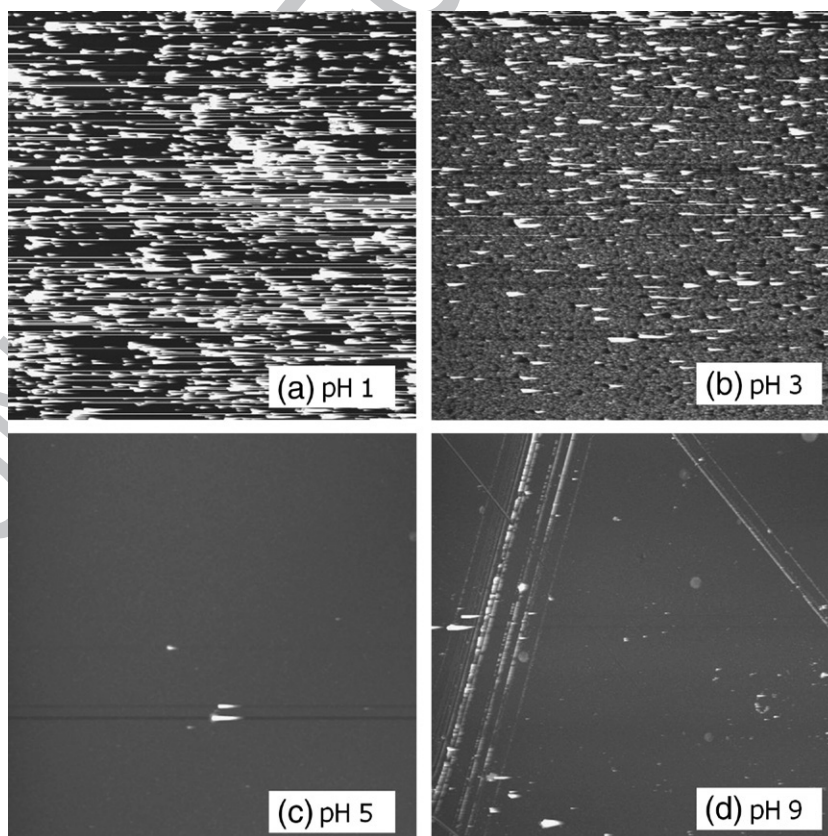


Fig. 3. $50 \mu\text{m} \times 50 \mu\text{m}$ AFM images for SiO_2 nanoparticles deposited onto SAM 1 at pH 1–9 (height scale is 100 nm).

239 high resolution spectra were obtained using a pass energy of
240 20 eV over a binding energy range of 20–30 eV, centred around
241 a chosen photoelectron binding energy, obtained using 0.1 eV
242 increments. A dwell time of 20 ms was employed when
243 collecting data from each binding energy increment for all
244 measurements. Recorded high resolution spectra would typi-
245 cally be an average of at least 10 scans.

246 2.4. Nanoparticle deposition and sample analysis procedures

247 Two types of nanoparticles were deposited onto SAMs. The
248 nanoparticles were used as received (0.5 g in 10 mL suspension)
249 and were SiO₂ (160 nm diameter, Bangs Labs, USA) and

COOH-terminated polystyrene latex (PL) (40 nm diameter, 250
Bangs Labs, USA). Deposition was performed over the pH 251
range 1–11 for the polystyrene latex nanoparticles, but over the 252
pH range 1–9 for the SiO₂ nanoparticles, because SiO₂ 253
dissolves at pH > 10 [36]. In each case, 0.1 mL of nanoparticle 254
suspension was added to 20 mL of aqueous solution at the 255
desired pH. Each SAM was immersed in the nanoparticle 256
solution for 2 h before being removed, whereupon it was rinsed 257
with aqueous solution of the same pH as the immersion 258
solution, followed by drying under a stream of Ar gas. 259

Samples were analysed using atomic force microscopy 260
(AFM) using a Dimension 3100 Nanoscope AFM (Veeco, UK) 261
operating in tapping mode under ambient conditions. The AFM 262

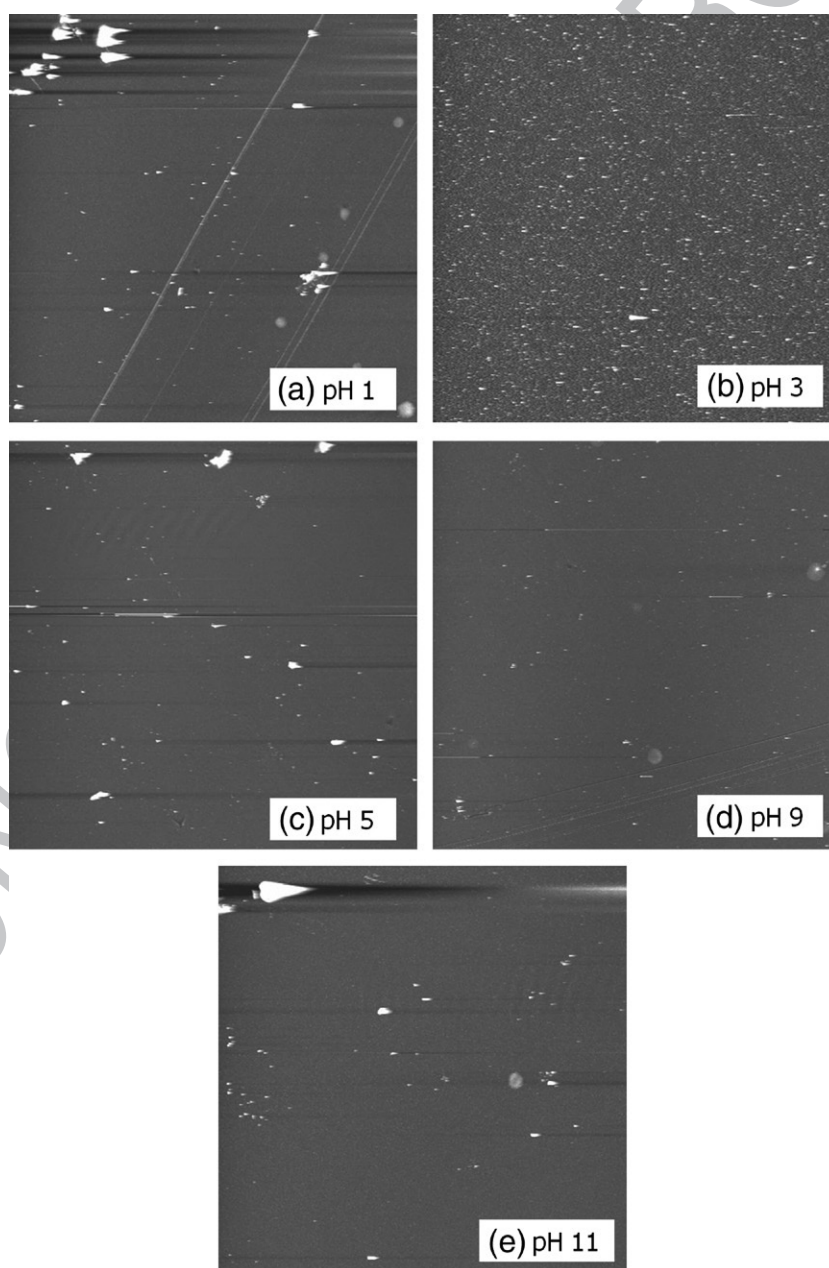


Fig. 4. 50 $\mu\text{m} \times 50 \mu\text{m}$ AFM images for COOH-PL nanoparticles deposited onto SAM 1 at pH 1–11 (height scale is 100 nm).

263 was housed on a vibration isolation table to minimise the effect
 264 of ambient noise on imaging quality. Nanoscope v5.12 software
 265 (Veeco, UK) was used throughout for both real-time analysis
 266 and post-capture image processing. Tapping Mode AFM
 267 imaging was performed using rectangular 180 μm length
 268 pyramidal-tipped Si cantilevers (Veeco, UK) with nominal
 269 spring constants of 40 N m^{-1} and resonant frequencies in the
 270 range 250–350 kHz. All images were acquired at scan rates
 271 between 0.2–2.0 Hz, each image being composed of
 272 512×512 pixels. Samples were immobilised onto steel SPM
 273 specimen disks (Agar Scientific, UK) using double-sided sticky
 274 tape (3M, UK) prior to AFM imaging. All sample handling was
 275 carried out using Dumostar tweezers (Agar Scientific, UK) to
 276 minimise the risk of sample contamination.

277 Samples were analysed by XPS as described in Section 2.3.
 278 The area of the Au 4f photoelectron peaks from the Au surface was
 279 calculated from the spectra recorded from all samples. For those
 280 samples with SiO_2 nanoparticles adhered on them, the area of the
 281 Si 2p photoelectron peaks was calculated. Similarly, for those
 282 samples with COOH–PL nanoparticles adhered on them, the area of
 283 the C 1s photoelectron peaks was calculated. The contribution
 284 to the C 1s photoelectron peaks from the underlying SAM was
 285 assumed to be negligible in comparison to the contribution from
 286 the COOH–PL nanoparticles. The Au:Si or Au:C ratios for each
 287 SAM/nanoparticle/pH combination were then calculated using
 288 relative sensitivity factors according to Wagner et al. [37].

3. Results and discussion

289

The adhesion of SiO_2 nanoparticles and COOH–PL
 290 nanoparticles to SAMs 1–3 is presented in the following
 291 sections. The Au:Si ratios for the SAM/ SiO_2 nanoparticle
 292 systems, and the Au:C ratios for the SAM/COOH–PL
 293 nanoparticle systems, as determined by XPS, are presented in
 294 Figs. 1 and 2 respectively. 295

3.1. SAM 1 (carboxylic acid-terminated)

296

SAM 1 presents a terminal carboxylic acid moiety with an
 297 assumed pK_a of 4.75 in aqueous solution. Therefore the pH of
 298 the aqueous electrolyte from which the nanoparticles are
 299 deposited will determine the protonation state of the SAM
 300 and may affect the observed patterns of adhesion. Figs. 3 and 4
 301 show the AFM images SiO_2 , COOH–PL and $\text{R}_3\text{N–PL}$
 302 nanoparticle adhesion to SAM 1 as a function of pH. 303

Fig. 3 reveals that there is significant adhesion of SiO_2
 304 nanoparticles to SAM 1 at pH 1 and pH 3, while at pH 5 and pH
 305 9 there is little adhesion of SiO_2 nanoparticles. Such behaviour
 306 suggests that as the pH increases from 3 to 5 the dissociation of
 307 the SiOH groups on the SiO_2 nanoparticles introduces a
 308 sufficiently repulsive interaction to prevent adhesion of SiO_2
 309 nanoparticles to the SAM through hydrogen bonding. Similarly,
 310 at pH 5 the terminal COOH groups of the SAM may have
 311

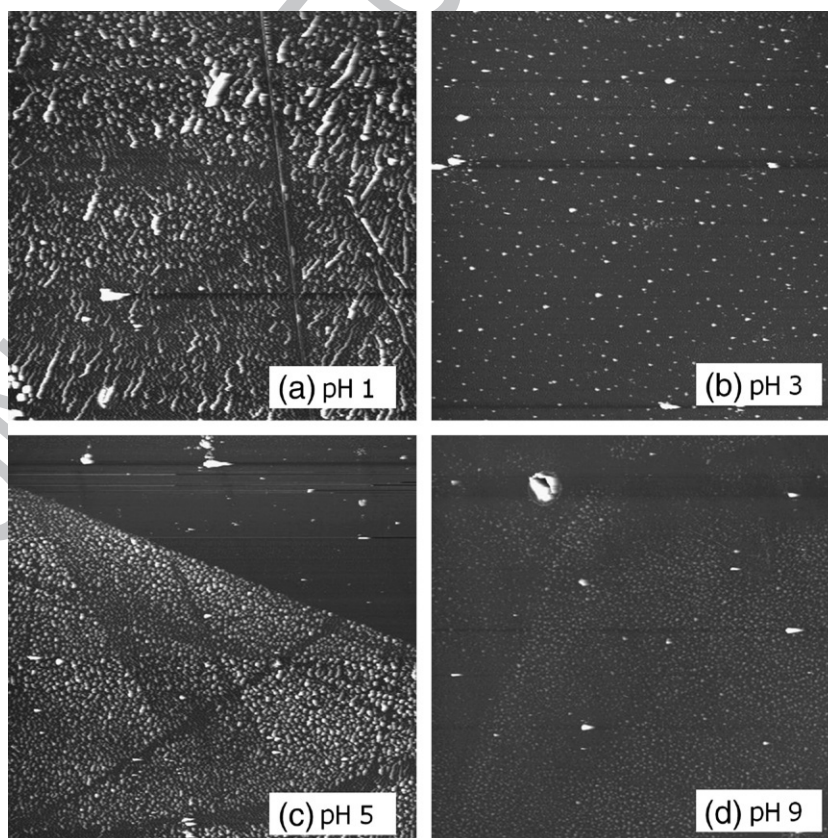


Fig. 5. $50 \mu\text{m} \times 50 \mu\text{m}$ AFM images for SiO_2 nanoparticles deposited onto SAM 2 at pH 1–9 (height scale is 100 nm).

312 dissociated to the carboxylate anion, which would further
313 increase the repulsive interaction between the SiO₂ nanoparti-
314 cles and the SAM. However, the dissociation of the COOH
315 moiety in SAMs has been investigated by Smith et al. [38,39]
316 and has been reported to increase from 4.75 towards 8.0. The
317 SiO₂ nanoparticle surface consists of hydrophilic SiOH (silanol)
318 groups [40], at a density of 4–5 SiOH groups nm⁻² [36]. SiO₂
319 has a negative zeta potential in aqueous solution at pH > 3.5, due
320 to the dissociation of SiOH groups to SiO⁻, and the zeta
321 potential will become increasingly negative as pH increases,
322 causing the electrostatic repulsion between nanoparticles to
323 increase. Nevertheless, for the system investigated here it is
324 anticipated that a repulsive interaction will exist between the

SAM and the SiO₂ nanoparticles whether the SAM surface 325
consisted of either dissociated or undissociated COOH 326
moieties. The Au:Si ratios obtained from XPS analysis, as 327
shown in Fig. 1, do not agree entirely with the results of the 328
AFM analyses, but it is believed that the morphology of the 329
nanoparticles is a contributory factor in this situation. The 330
aggregation of SiO₂ nanoparticles on the SAM, at pH 1 in 331
particular, may give rise to a higher Au:Si ratio than might be 332
expected, as Si 2p photoelectrons leaving the surface will have 333
to pass through a substantially greater amount of surface 334
material than the Au 4f photoelectrons leaving the surface. 335
Hence, the Au:Si ratio appears to be higher than it actually is. A 336
similar situation may also have occurred at pH 3, at which pH 337

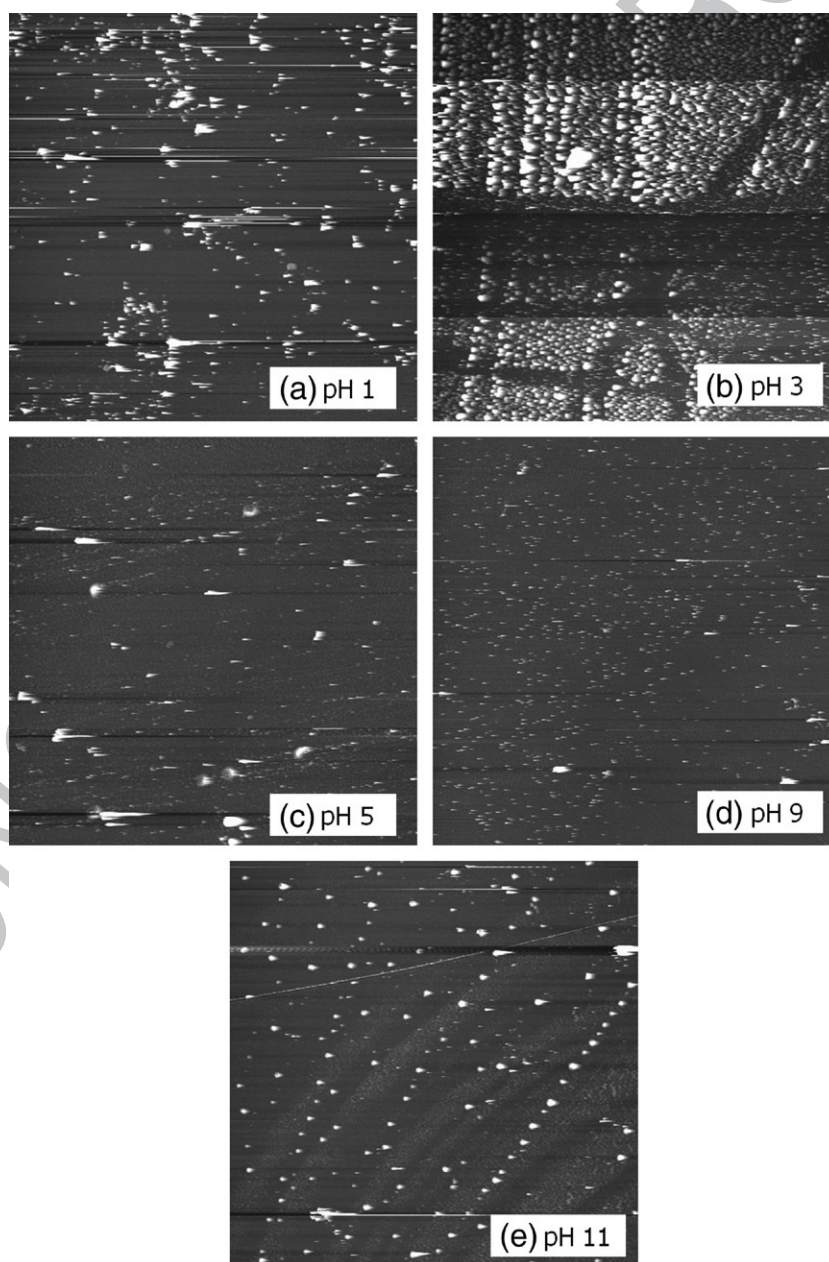


Fig. 6. 50 $\mu\text{m} \times 50 \mu\text{m}$ AFM images for COOH-PL nanoparticles deposited onto SAM 2 at pH 1–11 (height scale is 100 nm).

aggregation was also observed, although the nanoparticle aggregates appeared more loosely bound together, which lead to the streaking visible in the AFM image for these pHs.

Figs. 2 and 4 reveal that COOH–PL nanoparticles exhibit little adhesion to SAM 1 at all pHs. However, it would appear from Fig. 2 that there are slightly more COOH–PL nanoparticles adhered to the SAM at pHs 5 and 9 than at the other pHs, which suggests that adhesion was greatest at those pHs around the pK_a of the COOH moiety.

3.2. SAM 2 (methyl-terminated)

SAM 2 presents a terminal methyl moiety with an assumed pK_a of 50 in aqueous solution. Therefore the pH of the aqueous electrolyte from which the nanoparticles are deposited will not affect the protonation state of the SAM, and will not affect the adhesion. Figs. 5 and 6 show the AFM images for SiO₂ and COOH–PL nanoparticle adhesion to SAM 2 as a function of pH.

Figs. 1 and 5 reveal that adhesion of SiO₂ nanoparticles to the methyl-terminated SAM surface occurs at pH 1–9 and decreases somewhat with increasing pH. There is no electrostatic repulsion between the SAM and the SiO₂ nanoparticles, due to the nature of the methyl moiety, hence the adhesion of SiO₂ nanoparticles to the SAM will be dominated by van der Waals forces.

Figs. 2 and 6 reveals that the adhesion of COOH–PL nanoparticles to SAM 2 varies little with pH, although the AFM images

indicate differences in the morphology of the deposited COOH–PL nanoparticles, particularly at pH 3. It may be that at pH 3 there exists insufficient electrostatic repulsion between the COOH–PL nanoparticles to prevent aggregation and deposition onto the SAM. At all other pHs, adhesion of COOH–PL nanoparticles to the SAM will be dominated by van der Waals forces.

3.3. SAM 3 (pyridine-terminated)

SAM 3 presents a terminal pyridine moiety with a predicted pK_a of 5.60 in aqueous solution. Therefore the pH of the aqueous electrolyte from which the nanoparticles are deposited will determine the protonation state of the SAM. Figs. 7 and 8 show the AFM images for SiO₂ and COOH–PL nanoparticle adhesion to SAM 3 as a function of pH.

Fig. 1 reveals that as pH increases the Au:Si ratio remains approximately constant, which suggests that the adhesion of SiO₂ nanoparticles to SAM 3 remained approximately constant with pH. This is in contrast to the AFM results, presented in Fig. 7, which reveal that as pH increases the number of SiO₂ nanoparticles deposited decreases. It is believed that the morphology of the nanoparticles is a contributory factor in this situation, as discussed for SAM 1. Briefly, the aggregation of SiO₂ nanoparticles on the SAM increases the amount of surface material which the photoelectrons leaving the surface must pass through before being detected, which can serve to

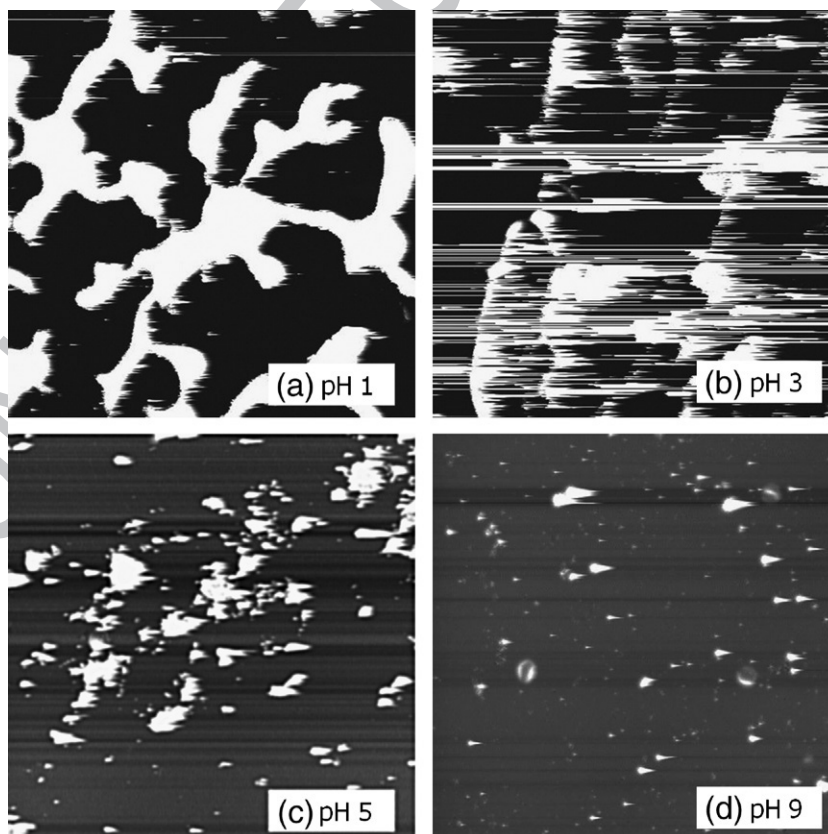


Fig. 7. 50 $\mu\text{m} \times 50 \mu\text{m}$ AFM images for SiO₂ nanoparticles deposited onto SAM 3 at pH 1–9 (height scale is 100 nm).

386 distort the measured photoelectron counts. At pH 1 and pH 3 it
387 would be expected that the SiO₂ nanoparticles have a little or no
388 surface charge, and therefore the observed aggregation of the
389 SiO₂ nanoparticles, leading to the deposition of aggregates on
390 the SAM, might also be expected. At pH 5 and pH 9 the SiO₂
391 nanoparticles will exhibit negatively charged surfaces and
392 therefore will repel each other, preventing aggregation. The
393 terminal pyridinium moieties of SAM 3 will become increas-
394 ingly dissociated with increasing pH, which will serve to
395 decrease the cationic surface charge of the SAM. This
396 decreasing surface charge will also promote the adhesion of
397 fewer SiO₂ nanoparticles to the SAM.

398 Fewer SiO₂ nanoparticles are adsorbed onto SAM 1 than
399 to SAMs 2 and 3, particularly at pH > 5, due to the de-

400 protonation of both the SiO₂ nanoparticle surface and the
401 SAM terminal moieties, leading to electrostatic repulsion
402 between the SAM and the SiO₂ nanoparticles, preventing
403 adhesion. For SAM 2 there will be no deprotonation of its
404 terminal moieties with increasing pH. For SAM 3, the
405 deprotonation of the terminal pyridine moiety will still leave
406 a surface capable of some electrostatic attraction with the
407 SiO₂ nanoparticles. Therefore, only SAM 1 does not present
408 a surface capable of adhering SiO₂ nanoparticles at increased
409 pH.

410 Fig. 2 reveals that the Au:C ratio increases with increasing
411 pH, suggesting a decrease in adhesion of COOH-PL nano-
412 particles to SAM 3 between these pHs. This trend is in
413 approximate agreement with the AFM results, presented in

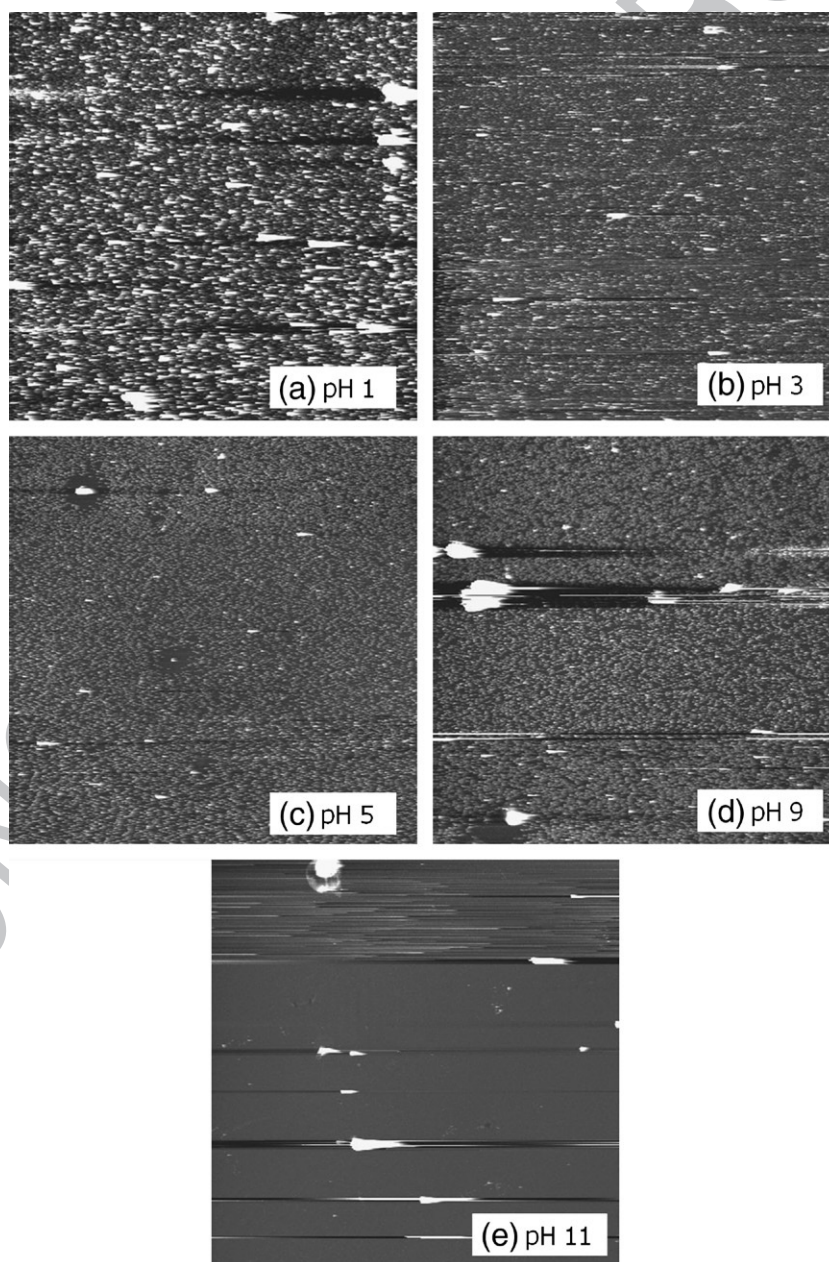


Fig. 8. 50 μm × 50 μm AFM images for COOH-PL nanoparticles deposited onto SAM 3 at pH 1–11 (height scale is 100 nm).

Fig. 8. The decreasing adhesion of COOH–PL nanoparticles to SAM 3 with increasing pH may be linked to the increasing dissociation of the surface COOH groups on the COOH–PL nanoparticles with increasing pH. Adhesion of COOH–PL nanoparticles to the SAM is reduced at pHs above the pK_a of the COOH moiety, although the apparent pK_a of the moiety, which is 4.75 in free solution, has been shown to increase when present at a surface [38,39], for example when it is the surface moiety of a SAM. For both the XPS and the AFM data, the decrease in adhesion of COOH–PL nanoparticles to the SAM occurs at pH greater than the pK_a of the COOH moiety.

4. Conclusions

The effect of electrolyte pH on the deposition and adhesion of nanoparticles to SAMs presenting carboxylic acid, methyl and pyridine surface moieties has been investigated using AFM and XPS. Adhesion was found to vary with pH for many combinations of SAM and nanoparticle, with a number of different surface morphologies of nanoparticles being observed. The adhesion behaviour is believed to be linked to the protonation state of the surface moieties on the nanoparticles, rather than simply the contact angle behaviour of the SAM, whereby increasing pH often led to decreased nanoparticle adhesion, probably due to increased repulsive interactions between the SAMs and the nanoparticle, and also between nanoparticles. Differences in nanoparticle adhesion were observed between SAMs for the same nanoparticle and pH, due to the different surface properties of each SAM, such as pK_a . Further investigation on systems such as these could include the study of nanoparticle adhesion to hydroxyl-terminated SAMs and assessment of the mass of adhered nanoparticles using a quartz crystal microbalance.

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Appendix A

The following supporting information is presented.
(i) Synthesis and characterisation data for SAM compound 1. (ii) Characterisation results for SAMs 1–3, consisting of elemental composition as determined by X-ray photoelectron spectroscopy and thickness data as determined by ellipsometry.

A.1. Synthesis and characterisation data for SAM compound 3

A.1.1. Compound 5

To a solution of 11-mercaptopundecanoic acid 4 (7.0 g, 32.1 mM) in C_2H_5OH (100 mL) heated under reflux was added

a solution of iodine (4.07 g, 16.03 mM) in C_2H_5OH (50 mL). Heating was continued for 12 h after which the reaction was allowed to cool to room temperature and washed with a saturated aqueous solution of $Na_2S_2O_3$ (50 mL). The products were extracted into CH_2Cl_2 (3×50 mL) and dried ($MgSO_4$), filtered and the solvent was removed *in vacuo*. The residues were purified by recrystallisation from CH_2Cl_2 /hexane. The feathery white crystals were filtered from the mother liquor, washed with ice-cold hexane and dried *in vacuo* affording 5 (6.47 g, 82%). m/z (ES) 513 $[M+Na]^+$. δ_H (500 MHz, $(CD_3)_2SO$) 4.09 (4H, q , $J=7.3$, 14.3 Hz), 2.65 (4H, t , $J=7.3$ Hz), 2.26 (4H, t , $J=7.3$ Hz), 1.61 (8H, m), 1.34 (30H, m). δ_C (400 MHz, $CDCl_3$) 173.9, 60.1, 39.1, 34.4, 29.3, 29.2, 28.5, 24.9, 14.2. Elemental analysis of $C_{26}H_{50}O_4S_2$ requires C 63.67%, H 10.20%. Elemental analysis found C 63.52%, H 10.48%.

A.1.2. Compound 6

To a vigorously stirred solution of 5 (4.75 g, 9.68 mM) in tetrahydrofuran (THF) (100 mL) was added a solution of potassium hydroxide (1.63 g, 29.0 mM) in H_2O/C_2H_5OH (1:1, 20 mL). The reaction was stirred for 12 h, and acidified with HCl (aq, 2 M, 20 mL) upon which a white solid precipitated. The solid was filtered off, washed with H_2O (100 mL), cold C_2H_5OH (100 mL) and dried *in vacuo* affording 6 as white plate-like crystals (4.20 g, 99%). m/z (ES) 457 $[M+Na]^+$. δ_H (400 MHz, $(CD_3)_2SO$) 3.5 (2H, s), 2.68 (4H, t , $J=8.0$ Hz), 2.14 (4H, t , $J=8.0$ Hz), 1.62 (4H, m), 1.46 (4H, m), 1.2 (24H, m). δ_C (400 MHz, $(CD_3)_2SO$) 174.5, 38.0, 33.7, 28.9, 28.8, 28.6, 28.6, 27.8, 24.5. Elemental analysis of $C_{22}H_{42}O_4S_2$ requires C 60.82%, H 9.67%. Elemental analysis found C 60.78%, H 9.70%.

A.1.3. Compound 3

To a solution of 6 (0.100 g, 0.23 mM) in dry THF (10 mL) cooled to 0 °C under an N_2 atmosphere was added 1–(3–dimethylaminopropyl)–3–ethyl–carbodiimide hydrochloride (0.272 g, 1.38 mM) and a catalytic amount of 4–dimethylaminopyridine. The mixture was stirred for 30 min and 3–pyridinepropanol (0.094 g, 0.69 mM) was added over 10 min, followed by further stirring for 24 h under an N_2 atmosphere at room temperature. The white precipitate was filtered and the filtrate was diluted with CH_2Cl_2 (30 mL) and washed with H_2O (3×30 mL), followed by 10% aqueous $NaHCO_3$ (10 mL) and saturated (aqueous) $NaCl$ (5 mL). The organic phase was dried ($MgSO_4$), filtered and the filtrate evaporated to dryness under reduced pressure. The residue was purified by silica gel column chromatography (eluent: $CH_2Cl_2/EtOAc$, 3:1) to yield 3 (0.045 g, 29%) as a white solid. m/z (ES) 695 $[M+Na]^+$. δ_H (400 MHz, $CDCl_3$) 8.4 (4H, m), 7.47 (2H, m), 7.19 (2H, m), 4.07 (4H, t , $J=6.4$ Hz), 2.66 (8H, m), 2.27 (4H, t , $J=8.0$ Hz), 1.94 (4H, m), 1.66 (8H, m), 1.32 (24H, m). δ_C (400 MHz, $CDCl_3$) 173.7, 149.9, 147.5, 136.3, 135.6, 123.2, 63.1, 39.0, 34.2, 29.8, 29.3, 29.1, 28.4, 24.9. Elemental analysis of $C_{38}H_{60}O_4N_2S_2$ requires C 67.75%, H 8.91%, N 4.16%. Elemental analysis found C 67.80%, H 8.82%, N 4.07%.

A.2. Characterisation results for SAMs 1–3

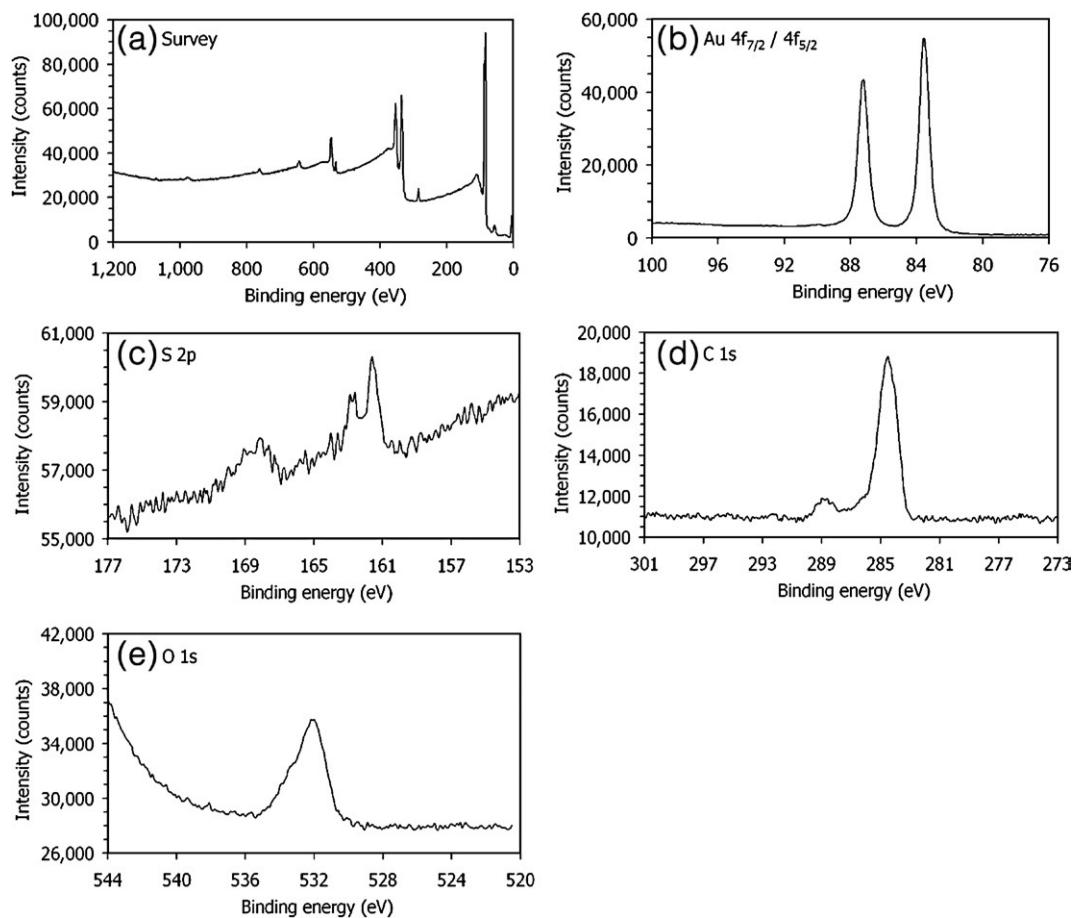


Fig. A1. XPS spectra for SAM 1. Figure uploaded electronically.

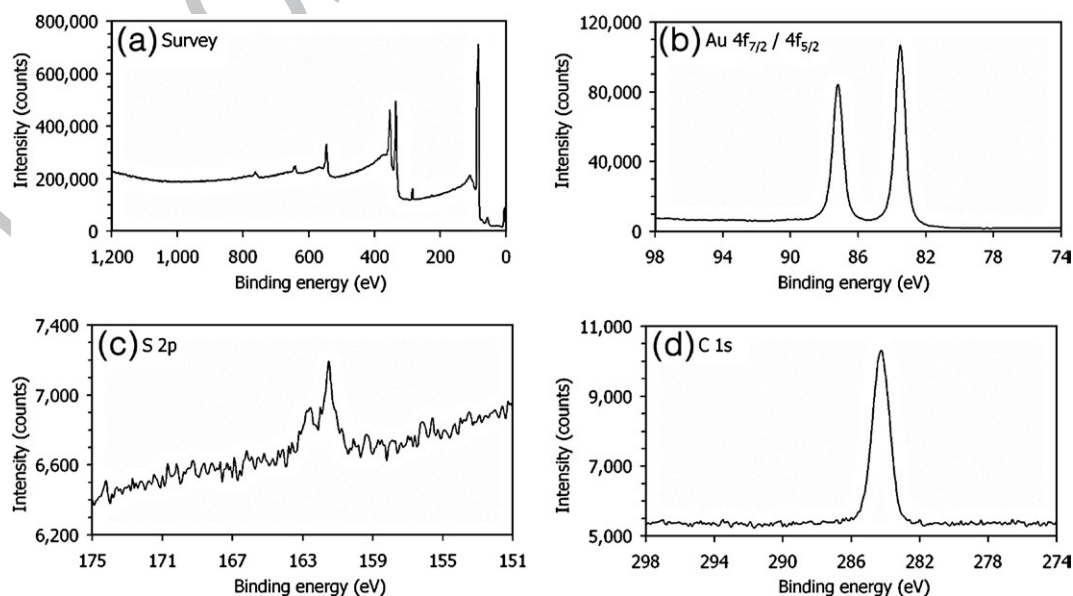


Fig. A2. XPS spectra for SAM 2. Figure uploaded electronically.

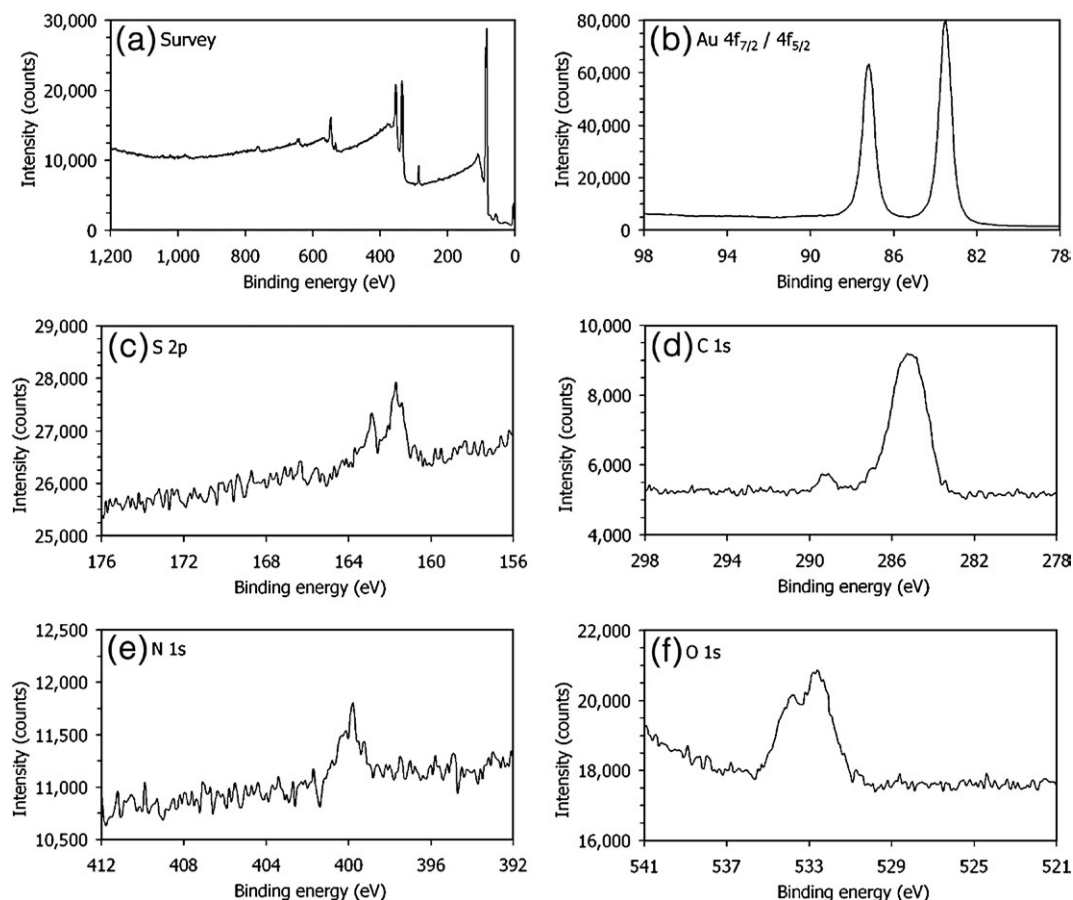


Fig. A3. XPS spectra for SAM 3. Figure uploaded electronically.

518 Table A1

519 Ellipsometrically measured thicknesses for SAMs 1–3

524 SAM	Calculated thickness range (nm)	Measured thickness (nm)
526 1	1.32–1.53	1.09±0.16
528 2	1.34–1.55	1.45±0.15
531 3	1.94–2.24	1.42±0.31

533 A thickness range for each SAM was calculated by
 534 estimating the length of the molecular structures of compounds
 535 1–3 using ChemDraw Ultra (v7.0.1, CambridgeSoft, UK) and
 536 Chem3D Ultra (v7.0.0, CambridgeSoft, UK) software. The
 537 upper limit of the range is the full length of the SAM
 538 molecule. The chosen lower limit of the range is the height of the SAM
 539 molecule at a tilt angle of 30° to the surface normal.

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