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Heterogeneous precipitation of Cu in Fe-Cu alloys

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LAWRENCE LIVERMORE NATIONAL LABORATORY

Heterogeneous precipitation of Cu in Fe-Cu alloys

M. Caro, A. Caro

November 1, 2006

IGRDM-13 Tsukuba, Japan October 15, 2006 through October 20, 2006

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Heterogeneous precipitation of Cu in Fe-Cu alloys

IGRDM-13

15-20 October 2006

M. Caro, A. Caro

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This work was performed under the auspices of the U.S. Department of Energy by the University of California Lawrence Livermore National Laboratory under Contract No. W-7405-Eng-48. We are developing models and computational tools for alloys to understand radiation damage in nuclear materials

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Motivation

 Copper is found as an impurity in reactor pressure vessel steels

 Radiation induced Cu precipitation leads to changes in the microstructure and controls damage effects such as embrittlement and radiation hardening

Nuclear materials

Microstructural evolution of structural steels under irradiation

Reactor Pressure Vessel



We investigate the formation of Cu precipitates in Fe-Cu alloys of low Cu concentration using :

• a classic potential that has been fully characterized thermodynamically

• a parallel Monte Carlo code with displacements that allows to perform simulations of heterogeneous precipitation IGRDM-13 Classic interatomic potentials for MD / MC simulations in alloys



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Our objective is to relate *ab initio* energetics, alloy potentials, thermodynamics, and microstructures



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2Computational Thermodynamics

We evaluate free energies

$$F = -kT \ln\left(\int_{\Omega} \exp(-H(x))dx\right)$$

Switching Hamiltonians Gibbs – Duhem integration Gas expansion

Phys. Rev. **B 66**, 054201, (2002) Phys. Rev. **B. 68**, 214205 (2003) J. Nuc. Mat. **336**, 233 *(*2005) J. Nuc. Mat. **349**, 317 (2006)

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Phase diagram corresponding to Ludwig-Farkas Fe-Cu EAM Potentials

CALPHAD Fe-Cu Phase Diagram Experiment





We developed a massively parallel displacement Monte Carlo code to predict microstructures

3



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Experimental determination of cluster composition in industrial ferritic steel

a) b) Mn Chr MmCu, Mn, Ni, Si Ni 5nm Ni Cu, Mn, Ni, Si

Figure 2. Enlarged view of a) cluster 2 (extents of outline box 8x8x8nm³) from 3DAP analysis of SH and b) cluster 8 from 3DAP analysis of WV012 (extents of outline box 6x6x6nm³) showing distribution of Cu, Mn, Ni, Si and P.

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Strategy to show you several HETEROGENEOUS precipitation examples: a nanophase Fe sample



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Heterogeneous precipitation in Fe-Cu



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Heterogeneous Cu precipitation in the presence of a free surface

Free surface (dark blue) in a nanophase sample

Cu goes preferentially to grain boundaries and triple-junctions that emerge at free surfaces





ame 001 | 27 Aug 2006



3% Cu precipitation at the core of a screw dislocation in Fe



Heterogeneous Cu precipitation on a screw dislocation





Heterogeneous Cu precipitation on a free (100) surface





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Heterogeneous Cu precipitation on a free (100) surface

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Conclusions: New models and tools for computational modeling of alloys

Developments:

(1)

(2)

- ✓ Formalism to design classic potentials for complex alloys
- ✓ Thermodynamic package to evaluate free energies of alloys in any phase
- 3 Parallel Monte Carlo code with displacements, to predict equilibrium microstructures in alloys

Applications:

✓ An accurate description of the thermodynamics of the system allows us to study heterogeneous precipitation and obtain physically based microstructures

✓ We are studying the effect of precipitation on mechanical properties





