

Enhanced low voltage nonlinearity in resonant tunneling metal-insulator-insulator-metal nanostructures

A.D. Weerakkody¹, N. Sedghi¹, I.Z. Mitrovic¹, H. van Zalinge¹, I. Nemr Nouredine¹, S. Hall¹, J.S. Wrench², P.R. Chalker², L.J. Phillips³, R. Treharne³, K. Durose³

¹University of Liverpool, Department of Electrical Engineering & Electronics, Liverpool, UK

²Centre for Materials and Structures, School of Engineering, University of Liverpool, Liverpool, UK

³Stephenson Institute for Renewable Energy, University of Liverpool, Liverpool, UK

E-mail address of corresponding author: d.a.d.weerakkody@liverpool.ac.uk

The electrical properties of bi-layer Ta₂O₅/Al₂O₃ and Nb₂O₅/Al₂O₃ metal-insulator-insulator-metal nanostructures as rectifiers have been investigated. The ultra-thin (1-6 nm) insulator layers were deposited by atomic-layer deposition or rf magnetron sputtering with Al as metal contacts. Variable angle spectroscopic ellipsometry was performed to extract the optical properties and band gap of narrow band gap insulator layers while the surface roughness of the metal contacts was measured by atomic force microscopy. Superior low voltage large signal and small signal nonlinearities such as asymmetry of 18 at 0.35 V, rate of change of non-linearity of 7.5 V⁻¹, and responsivity of 9 A/W at 0.2 V were observed from the current-voltage characteristics. A sharp increase in current at ~2 V on Ta₂O₅/Al₂O₃ device can be ascribed to resonant tunneling.

Keywords: metal-insulator-insulator-metal nanorectifiers, nonlinearity, resonant tunneling

1. Introduction

Metal-insulator-metal (MIM) rectifying devices using double [1–5], triple [6] or quadruple [5] insulator layers are currently the focus of attention for the development of next-generation antenna-coupled infrared detectors [7] and optical rectennas [8]. The interest is driven by their distinctive attributes, such as nanoscale footprint, room temperature operation, zero bias voltage requirement, and ease of integration with Complementary Metal Oxide Semiconductor (CMOS) technology. Superior diode characteristics such as large responsivity, asymmetry and nonlinearity can be achieved via thermionic emission and quantum mechanical tunneling, however for less temperature dependent high-speed operation [9], the dominant conduction should be quantum tunneling. Devices with two insulator layers, viz. metal-insulator-insulator-metal (MIIM) enhance the rectification by increasing the nonlinearity of the current-voltage (*I**V*) characteristics. There are two mechanisms that allow MIIM rectifiers to have a high nonlinearity while keeping the resistance low [1–5]: (i) use of resonant tunneling (RT) of electrons through a quantum well formed between the two insulators (Fig. 1a). This occurs when the metal Fermi level on the higher barrier side is biased positive creating a right-triangular well at the interface of the two insulators. When an allowed energy level in the quantum well aligns with the metal Fermi level on the negative side, it causes a sharp turn-on of the rectifier; (ii) use of step tunneling (ST) occurs for the opposite bias polarity (Fig. 1b), when an abrupt increase in current occurs, when the metal Fermi level on the higher

barrier side rises above the conduction band of the lower barrier, thereby decreasing the tunnel distance. In a particular device, the choice of insulator materials, metals and thicknesses determines the mechanism that dominates. The highest reported rectification parameters for MIIM devices based on Cr/Al₂O₃/HfO₂/Cr are: asymmetry 10 @ 3 V, and responsivity <2.5 A/W for the range of voltages 0.5–2.5 V [3]. A responsivity of 11 A/W at 0.02 V has been predicted from simulations on 4 nm resonant tunneling W/Nb₂O₅/Ta₂O₅/W, however not experimentally demonstrated [4]. Even for bilayer Ta₂O₅/Al₂O₃ MIIMs with dissimilar metal electrodes of work function difference of 0.6 eV, the highest reported low voltage (<0.8 V) asymmetry is 10 [1,2,5].

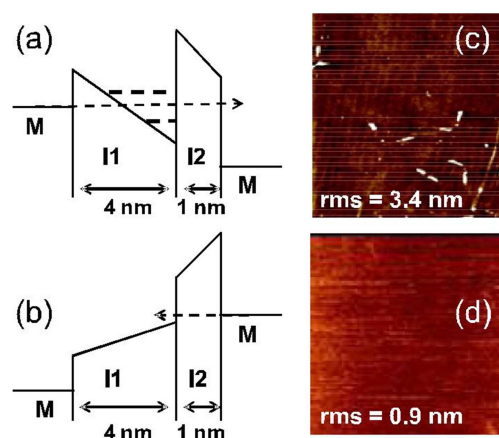


Fig. 1. Energy band diagram of the MIIM structure (a) resonant tunneling in forward bias (b) step tunneling in reverse bias. AFM images of the Al blanket films (c) before, and (d) after optimizing the thermal evaporation parameters. The scanned area is 40 μ m \times 40 μ m.

This paper demonstrates fabrication of bi-layer Ta₂O₅/Al₂O₃ and Nb₂O₅/Al₂O₃ nanostructures using atomic-layer deposition (ALD) and rf sputtering with a focus on achieving low voltage nonlinearity due to the effect of resonant tunneling. We observe superior low voltage asymmetry (18 @ 0.35 V) and responsivity (9 A/W @ 0.2 V) for Ta₂O₅/Al₂O₃ and Nb₂O₅/Al₂O₃ MIIM devices respectively, fabricated using Al metal electrodes.

2. Device fabrication

MIIM devices were fabricated on 4 cm × 4 cm cleaned Corning glass substrates with an rms surface roughness of 0.32 nm. Surface roughness was extracted by both variable angle spectroscopic ellipsometry (VASE) and atomic force microscopy (AFM). Room temperature VASE measurements were performed in the energy range 0.7–5.1 eV and three angles 65°, 70° and 75° to maximize the accuracy in extracting thickness, optical properties and band gap of Nb₂O₅ and Ta₂O₅ layers. The 60 nm thick Al metal contacts were deposited by thermal evaporation and defined either by photolithography or shadow mask. The rms surface roughness after Al evaporation was found to improve substantially from 3.40 nm (Fig. 1c) to 0.87 nm (Fig. 1d) using a lower evaporation rate of 0.4 nm/s. The surface roughness of the substrate and the bottom metal contact was found to have a large impact on device performance in agreement with recent work of Alimardani et al. [10].

We have previously reported theoretical calculations using the multi-barrier transfer matrix method [11], that 1 nm of Al₂O₃ and 3–6 nm of Ta₂O₅ or Nb₂O₅ using Al, can bring optimum low voltage rectification performance. Hence, these thicknesses were used as nominal values for depositing insulators either by ALD or rf magnetron sputtering. The dry etching of Al native oxide was done using Ar plasma at a power of 50 W for 35 min. Room (293 K) and high temperature (293–370 K) current–voltage measurements were performed on fabricated devices in a dark probe station using the Agilent B1500 semiconductor parameter analyzer.

3. Results and discussion

3.1 Optical properties of Ta₂O₅ and Nb₂O₅ films

Optical properties and band gap of Ta₂O₅ and Nb₂O₅ were determined from VASE measurements on bulk oxide films (≥10 nm) deposited by ALD on Si reference samples. The dielectric function of the Si film with the native oxide as Cauchy layer (1.34 nm) was modeled first. Then, another Cauchy layer was

added and fit-ted for the thickness of Ta₂O₅ (9.9 nm), or Nb₂O₅ (32.7 nm). Subsequently, the optical constants (real and imaginary parts of the dielectric function) were extracted. The dielectric function converts to refractive index (n) and extinction coefficient (k) using the Kramers–Kronig [12] relations. The absorption coefficient (α) is calculated from the extinction coefficient as $\alpha = 4\pi k/\lambda$, where k is wavelength. The n, k and α vs photon energy (E) plots are shown in Fig. 2a–c respectively. The band gap can be estimated by linear extrapolation of the segments on the curves in the non-absorbing regions, and is found to be 3.71 eV for Nb₂O₅ and 4.45 eV for Ta₂O₅.

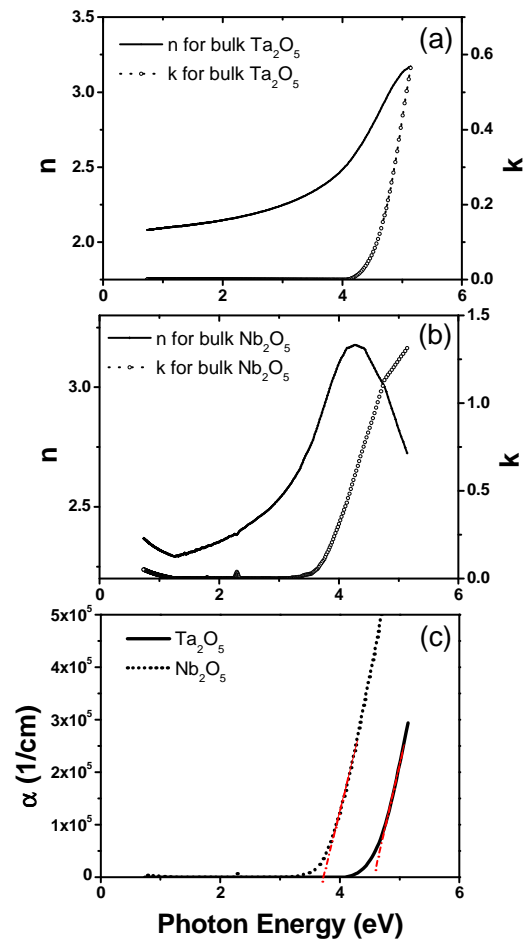


Fig.2: Refractive index (n) and extinction coefficient (k) vs photon energy for bulk (> 10 nm) (a) Ta₂O₅, and (b) Nb₂O₅ films. (c) The extraction of band gap from the absorption coefficient vs photon energy plots for respective Ta₂O₅ and Nb₂O₅ films deposited by ALD.

3.2 Rectification parameters of fabricated Ta₂O₅/Al₂O₃ and Nb₂O₅/Al₂O₃ MIIM devices

The typical current density vs. voltage (JV) characteristics of the MIIM devices with 4 nm Ta₂O₅

or Nb₂O₅, 1 nm deposited Al₂O₃ and ~1.6 nm native Al₂O₃ are shown in Fig. 3(a). It can be seen that Nb₂O₅/Al₂O₃ MIIM device shows larger current due to smaller band gap (see Fig. 2(c)) and smaller barrier height at the Al/Nb₂O₅ (up to 0.2 eV) than that at the Al/Ta₂O₅ (~ 0.5 eV) interface [4,13]. Moreover, the Nb₂O₅/Al₂O₃ devices shows negligible IV temperature dependence (Fig. 3(b)) indicating the dominance of tunneling conduction, in agreement with findings in [1-3].

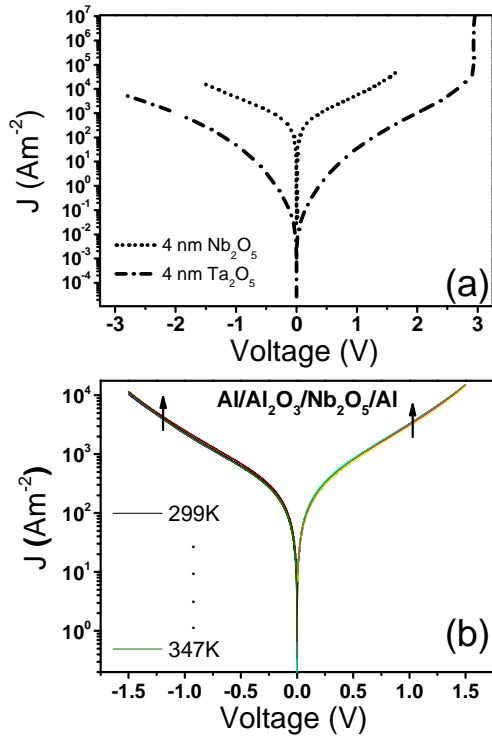


Fig.3: (a) JV plots of the Al/native oxide/1 nm Al₂O₃/Al MIIM devices, where I refers to Ta₂O₅ or Nb₂O₅. (b) Temperature dependent JV plots of the Nb₂O₅/Al₂O₃ MIIM device. The device area is: A = 10⁻⁸ m².

The large signal rectification is ascribed to the device asymmetry defined as the ratio of current at positive (I₊) or negative (I₋) bias, whichever larger, to that at opposite bias

$$\eta_{asym} = |I_+/I_-| \text{ or } |I_-/I_+| \quad (1)$$

Small signal rectification, however, is governed by nonlinearity around the operating point and is usually realized by square law rectification. A measure of small signal nonlinearity is responsivity, defined as the ratio of dc rectified current, I_{dc} to input ac power, P_{in} [14].

$$Resp = \frac{I_{dc}}{P_{in}} = \frac{1}{2} \frac{I''}{I'} \bigg|_{V_p} = \frac{1}{2} \frac{dg_d/dV}{g_d} \quad (2)$$

where I' and I'' are the first and second derivatives of current and g_d is dynamic conductance at operating point V_p. Maraghechi *et al.* [3] have defined a nonlinearity factor as the ratio of dynamic to static conductance i.e.

$$\chi = \frac{dI/dV}{I/V} \quad (3)$$

and also used the rate of change in nonlinearity, to reflect the small signal nonlinearity. The large signal asymmetry i.e. |I/I₊| for Nb₂O₅/Al₂O₃ and Ta₂O₅/Al₂O₃ MIIM devices is shown in Fig. 4. The Ta₂O₅ device has peak value of 18 at around 0.3 V and at voltages close to 0 V. The Nb₂O₅/Al₂O₃ MIIM device shows the peak value of 7.6 around 0 V and the asymmetry reaches to ~ 6 at larger voltages.

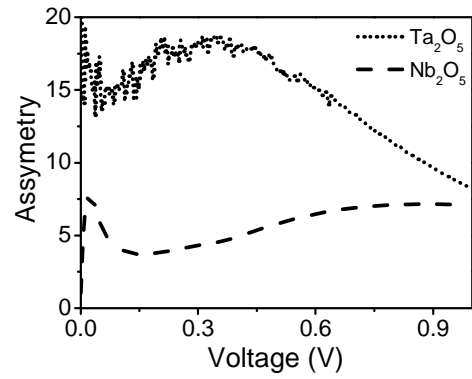


Fig.4: Device asymmetry (= |I/I₊|) for: 1 nm Al₂O₃/4 nm Ta₂O₅, and 1 nm Al₂O₃/4 nm Nb₂O₅ MIIM devices.

The responsivity vs voltage graphs for the MIIM devices are shown in Fig. 5, with peak values of 9 A/W and 6 A/W for Nb₂O₅ and Ta₂O₅ devices, respectively. The thickness of dielectric layers in these devices is 5.9 nm for Nb₂O₅, 5.3 nm for Ta₂O₅, and 1.6 nm for Al₂O₃, measured by VASE.

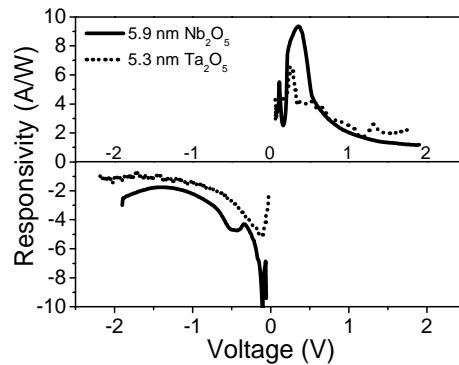


Fig.5: Device responsivity (= I''/2I') for two MIIMs based on Al₂O₃/Ta₂O₅ & Al₂O₃/Nb₂O₅.

The nonlinearity factor for the above devices is shown in Fig. 6. The rate of change in nonlinearity factor at low voltage range <0.8 V was found to be 7.5 V^{-1} and 6.8 V^{-1} for Ta_2O_5 and Nb_2O_5 MIIM devices, respectively, higher than reported value of 5.5 V^{-1} [3].

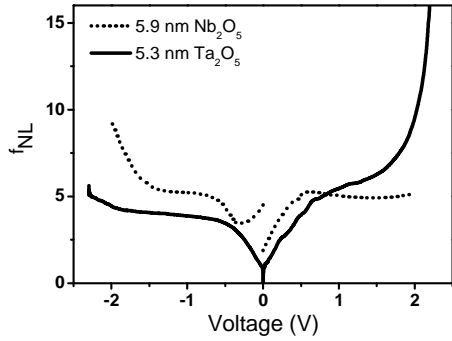


Fig.6: Device nonlinearity factor for two MIIMs based on $\text{Al}_2\text{O}_3/\text{Ta}_2\text{O}_5$ & $\text{Al}_2\text{O}_3/\text{Nb}_2\text{O}_5$.

3.3 Observation of resonant tunneling

A fingerprint of resonant tunneling can be found from a change of curvature and significant increase of current in IV characteristics. A larger current for positive bias is evident from Fig. 7a for $\text{Ta}_2\text{O}_5/\text{Al}_2\text{O}_3$ MIIM device. The native oxide grown unintentionally on bottom Al contact (~ 1.6 nm measured by VASE) has a large impact on the MIIM device behavior, particularly on resonant tunneling. It increases the thickness of Al_2O_3 layer, resulting in a shift of RT to larger voltages and in some cases beyond the operating voltages of the device. As can be seen from Fig. 7(b), after etching native Al_2O_3 , the device shows a significant increase of current for about two orders of magnitude. The change in slope of the JV characteristics due to RT has moved from 2 V for the unetched sample to ~ 1.5 V for the etched sample.

Referring to the experimental data in Fig. 7 and theoretical calculations [11], the band diagrams shown in Fig. 8 can be derived. At small positive bias voltages (Fig. 8a), the conduction in both oxide layers is by direct tunneling (DT). By increasing the voltage, the conduction in the Ta_2O_5 layer is by Fowler-Nordheim (FN) and in Al_2O_3 by DT (Figs. 8b,d). However, the change in current at negative bias (Fig. 7(b)) is higher due to lowering of the barrier for Ta_2O_5 , and at ~ -1.2 V the conduction changes to step tunneling (Fig. 8e). At larger positive voltages, > 1.5 V, when the bound states are established in the quantum well (Fig. 8c), the RT becomes dominant. A sharp increase in current and change of curvature can be observed at ~ 2 V in Fig. 7. Since the breakdown voltage for $\text{Ta}_2\text{O}_5/\text{Al}_2\text{O}_3$ MIIM devices is ~ 3 V (see Fig. 3(a)), the latter can be assigned to the onset of RT. It is

worth mentioning that by calculations [11], the onset of resonant tunneling is expected to be at 1.5 V. The larger onset of around 2 V is presumably due to thicker alumina native oxide layer or a slight difference in work functions for the Al top and bottom contacts.

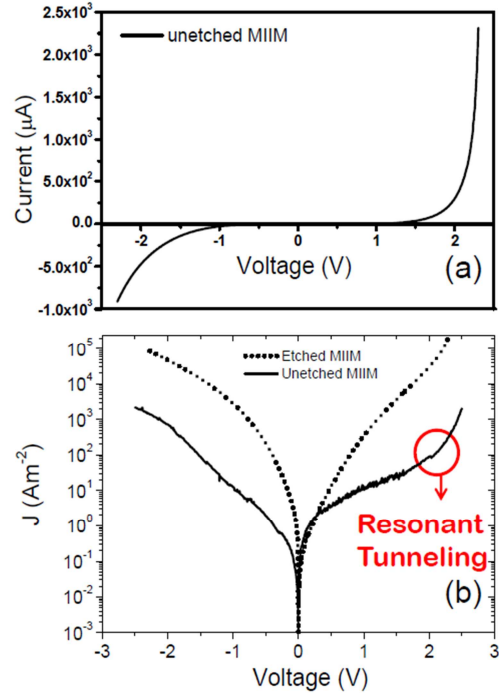


Fig.7: (a) IV (linear scale) and (b) $J-V$ (log scale) plots with $\text{Al}_2\text{O}_3/\text{Ta}_2\text{O}_5$ MIIM prior and post etching of native Al_2O_3 on bottom Al electrode ($A = 10^{-8} \text{ m}^2$). The etching step and subsequent insulator deposition were done without breaking the vacuum. The onset of resonant tunnelling is marked.

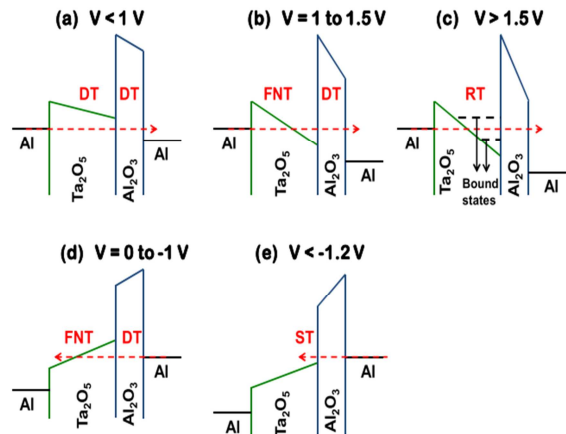


Fig.8: Energy band diagrams of $\text{Al}/\text{native oxide} + 1 \text{ nm } \text{Al}_2\text{O}_3/4 \text{ nm } \text{Ta}_2\text{O}_5/\text{Al}$ MIIM tunnel rectifier depicting various conduction scenarios being dependent on external bias: (a)-(c) positive voltages; (d)-(e) negative voltages. DT, FN, ST, and RT refer to direct, Fowler-Nordheim, step and resonant tunneling, respectively.

4. Conclusions

We demonstrate experimentally, enhanced low voltage nonlinearity in ALD and rf sputtered nanometer scale bilayer $\text{Ta}_2\text{O}_5/\text{Al}_2\text{O}_3$ and $\text{Nb}_2\text{O}_5/\text{Al}_2\text{O}_3$ tunnel barriers. The results indicate how the layers can be designed to tune IV asymmetry and nonlinearity via resonant tunneling in these next-generation high speed rectifiers.

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