

# Facing the challenge of liquid phase crystallizing silicon on textured glass substrates

Grit Köppel<sup>a,\*</sup>, Daniel Amkreutz<sup>a</sup>, Paul Sonntag<sup>a</sup>, Guangtao Yang<sup>b</sup>, René van Swaij<sup>b</sup>, Olindo Isabella<sup>b</sup>, Miro Zeman<sup>b</sup>, Bernd Rech<sup>a</sup>, and Christiane Becker<sup>a</sup>

<sup>a</sup>Helmholtz-Zentrum Berlin für Materialien und Energie GmbH, Kekuléstr. 5, D-12489 Berlin, Germany

<sup>b</sup>Laboratory of Photovoltaic Material and Device, Delft University of Technology, P.O. Box 5031, Delft 2600 GA, The Netherlands

**Abstract** — A major limitation in current liquid phase crystallized (LPC) silicon thin-film record solar cells are optical losses caused by their planar glass-silicon interface. In this study, silicon is grown on nanoimprinted periodically as well as on randomly textured glass substrates and successfully implemented into state-of-the-art LPC silicon thin-film solar cell stacks. By systematically varying every layer the whole sample stack is optimized regarding its anti-reflection ability. Compared to an optimized planar reference device, a reduction of reflection losses by -3.5% (absolute) on the random and by -9.4% (absolute) on the periodic texture has been achieved in the wavelength range of interest.

**Index Terms** — silicon, thin-film solar cells, light management, absorption enhancement.

## I. INTRODUCTION

Recently, a new in-house record of 12.1 % efficiency for a 10- $\mu\text{m}$  thick liquid phase crystallized (LPC) silicon thin film solar cell on glass has been achieved by texturing the silicon absorber backside as well as reducing reflection losses at the air-facing surface [1]. In this cell design the optical absorption potential is not fully tapped due to losses at the planar glass-silicon interface. Hence, effective measures for light management are needed at this interface to further increase photo-generated current densities. It has been demonstrated for other thin-film silicon solar cell types [2]–[4] that an enhancement of cell efficiencies ensues. However, the challenge is to identify structures that increase the incoupling of light while at least maintaining the electronic material quality of the silicon absorber layer [5]–[9].

Texturing glass substrates by nanoimprint lithography using high-temperature stable sol-gels has proven to be a suitable method for texturing the glass-silicon interface in LPC silicon solar cells [10], [11]. A smooth random ZnO:Al (AZO) texture as well as a periodic U-shaped texture with an aspect ratio of 0.5 have been transferred to glass substrates. On one hand, the smooth AZO texture maintained the electronic silicon material quality but did not significantly enhance light incoupling into the cell. On the other hand, the high aspect ratio U-shaped texture remarkably enhanced the optical properties of the cells but disturbed the silicon material quality resulting in reduced quantum efficiencies. Recently, a hexagonal sinusoidal structure with an aspect ratio of 0.2 has shown to be a balanced compromise between these two extreme cases.

Another approach in this work is to implement modulated surface textured (MST) glass substrates [12]. These glasses have already been integrated successfully into 3- $\mu\text{m}$  thick  $\mu\text{c-Si:H}$  solar cells, enabling a cell efficiency above 10 % [13].

In this study both textures, the periodic hexagonal sinusoidal texture (“Sine”) as well as the MST substrates are integrated into 10- $\mu\text{m}$  thick LPC silicon thin-film solar cells and compared to state-of-the-art planar reference devices. All cells are analyzed regarding their optical and electronic properties from which the quality of the silicon layer can be inferred.

## II. EXPERIMENTAL

As glass substrates, 0.7-mm and 1.1-mm thick Corning Eagle XG<sup>TM</sup> glasses have been used. After periodical or random texturing, the substrates were coated with a 250-nm  $\text{SiO}_x$  layer acting as diffusion barrier against substrate impurities. As periodic texture, a 750-nm pitched hexagonal sinusoidal structure has been transferred to glass substrates using nanoimprint lithography in combination with a high-temperature stable sol-gel. Details on the process can be found in [11]. On the other hand, differently sized random textures were transferred into the substrates by wet chemical etching using sacrificial layers. Depending on the sacrificial layer type differently sized crater-like features can be achieved: an ITO induced process results in a morphology with typical average lateral features size of 15  $\mu\text{m}$  (“IIT”) and a ZnO:Al induced process results in a morphology with typical average lateral features size of 2  $\mu\text{m}$  (“ZIT”) [14]. In addition, both textures can be superimposed to produce a modulated surface texture (“MST”) [13].

These textured substrates as well as planar reference substrates have subsequently been coated with  $\text{SiN}_x$  and  $\text{SiO}_x$  layers, where the  $\text{SiN}_x$  layer provides anti-reflective properties while the  $\text{SiO}_x$  acts as a passivation layer as well as wetting agent for silicon during liquid phase crystallization [15], [16]. For optical optimization both interlayer thicknesses as well as the thickness of the subsequently deposited silicon absorber layer have been varied between 50 nm and 80 nm for the  $\text{SiN}_x$  layer, 5 nm to 20 nm for the  $\text{SiO}_x$  layer and 5  $\mu\text{m}$  to 30  $\mu\text{m}$  for the silicon absorber layer. The silicon absorber layers are deposited by electron beam evaporation at a heater temperature of 600°C and capped with a 200-nm thick  $\text{SiO}_x$

layer. Afterwards the samples were crystallized by scanning a line-shaped CW laser emitting at 808 nm over the preheated substrate (700 °C) with a constant velocity of 3 mm/s. By means of this capping layer the substrate texture is preserved at the top of the silicon absorber layer after liquid phase crystallization resulting in a double-sided textured silicon absorber layer [17]. The capping layer is removed by wet chemical etching in buffered-HF for nine minutes. A schematic sample stack as well as atomic force microscope (AFM) images of sinusoidal and MST textured glass substrates are depicted in Fig. 1.

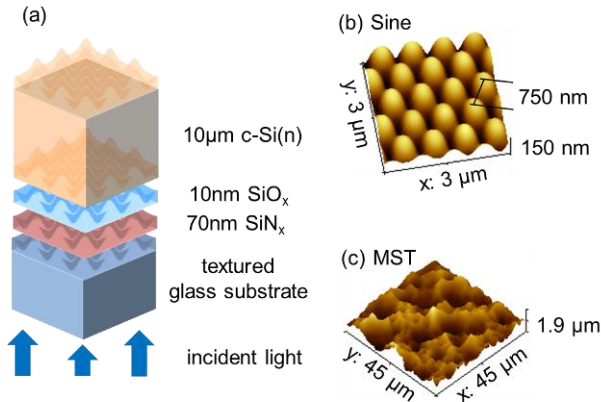


Fig. 1. (a) Schematic of the sample stack. AFM images of textured glass surfaces patterned with (b) a hexagonal sinusoidal texture ('Sine') and (c) a modulated surface texture (MST).

Optical analysis has been conducted using a Perkin Elmer LAMBDA 1050 spectrometer featuring an integrating sphere. Average reflectance values have been calculated for a wavelength range of 300 nm to 600 nm, the substantial wavelength range for anti-reflective properties at the glass-silicon interface. Surface morphology determination of the structures was realized using a Park Systems XE-70 AFM (Fig. 1 b+c).

### III. RESULTS AND DISCUSSION

Here, we implement periodic and random textures into state-of-the-art LPC silicon thin-film absorber layers in order to address reflection losses at the glass-silicon interface and enhance solar cell performance. For this purpose the structure type, the thicknesses of the  $\text{SiN}_x$  /  $\text{SiO}_x$  interlayer stack as well as the silicon absorber thicknesses are varied (section A). Based on these results optimal thicknesses are chosen and analyzed regarding their optical properties (section B).

#### A. Texture, Interlayer and Absorber Thickness Variation

The periodic sinusoidal texture has already been successfully implemented into state-of-the-art thin-film LPC silicon solar cells [11]. For this device, the anti-reflective effect of the glass-silicon texture as well as the light-trapping effect of the double-sided textured absorber layer led to an enhancement of external quantum efficiency in the short and long wavelength range, respectively. However, in the wavelength range between 400 nm and 700 nm the conversion efficiency declined due to recombination losses caused by insufficient interface passivation. In this study, we additionally implement random textures into state-of-the-art LPC silicon thin film solar cell devices, which have already successfully been integrated into 3- $\mu\text{m}$  thick  $\mu\text{-Si:H}$  devices [13]. These random textures have been chosen due to their promising smoother texture flanks compared to the sinusoidal texture.

In Fig. 2 the optical properties of both texture types are compared regarding their anti-reflection potential in state-of-the-art 10- $\mu\text{m}$  thick LPC silicon absorber layers with a  $\text{SiN}_x$  70 nm /  $\text{SiO}_x$  10 nm interlayer stack [18]. The sample stack under investigation is schematically depicted in Fig. 1a as well as in every graph as an inset featuring a horizontal arrow indicating which layer has been varied.

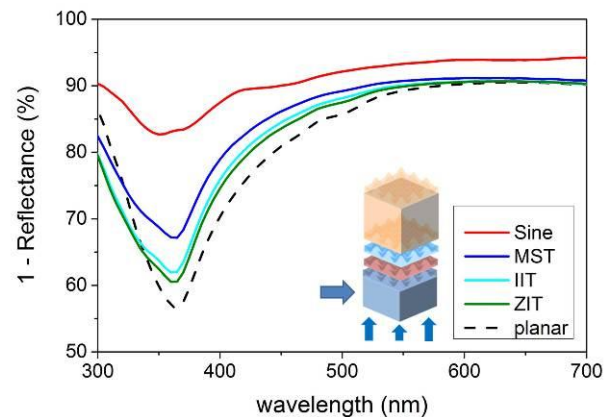


Fig. 2. Reflectance data ( $R$ , plotted as  $1-R$ ) of a planar reference (black, dashed), different random textures with large features (IIT, cyan), small features (ZIT, green) and a combination of both (MST, blue) as well as a hexagonal sinusoidal texture (Sine, red).

Compared to a planar reference (black, dashed line) with optimized interlayer stack, all textures provide additional anti-reflective properties (solid lines) over the entire wavelength range. The highest reflection reduction is found for the periodic sinusoidal texture (Sine, red). Comparing the different random textures with large features (IIT, cyan), small features (ZIT, green) and combined texture (MST, blue) the largest reflection reduction is found for the combined MST texture. Therefore, the MST texture is chosen for further analysis as random texture.

Incident light with wavelengths longer than 600 nm reaches the backside of the absorber and is either transmitted without being absorbed or scattered back. Hence, the light scattering effect of the double-sided textured absorber layer becomes visible and both light trapping effects superimpose. Therefore, the analysis of the anti-reflective effect is restricted to shorter wavelengths than 600 nm. There, average reflection losses of the planar reference of 24.2% are reduced by -7.1% (absolute) for the MST texture (blue) and -14.0% (absolute) for the sinusoidal texture (red), which was chosen based on simulations of Lockau et al. predicting optimum anti-reflective properties in thin-film LPC silicon devices for a hexagonal sinusoidal texture [19].

To find optimum optical properties the subsequently deposited 70 nm SiN<sub>x</sub>, 10 nm SiO<sub>x</sub> and 10 μm Si absorber layer have been varied systematically by keeping two layer thicknesses fixed and varying one layer thickness (Fig. 3). A schematic sample stack is depicted as inset, where the varied layer is highlighted by a horizontal arrow.

For wavelength shorter than 450 nm reflection losses increase with increasing SiN<sub>x</sub> interlayer thickness (Fig. 3a) resulting in the highest reflection losses for the thickest barrier (80 nm, orange). However, for longer wavelength this trend inverts and the highest reflection losses are found for the thinnest barrier (50 nm, dark blue) while all other barrier thicknesses are about equal without clear trend. While the 60 nm barrier (purple) features slightly lower reflection losses than the 70 nm barrier (red) for wavelength smaller than 500 nm, for longer wavelength the reflection losses of the thicker barrier are slightly lower, such that overall no clear difference between these two SiN<sub>x</sub> barrier thicknesses can be found. Hence, both medium interlayer thicknesses (60 nm and 70 nm) are suited for implementation into LPC-Si solar cell devices.

For the SiO<sub>x</sub> layer thickness variation (Fig. 3b) reflection losses increase with increasing SiO<sub>x</sub> thickness for wavelength shorter than 550 nm. For longer wavelength no difference between the different SiO<sub>x</sub> layer thicknesses is found. Therefore, the thinnest 5 nm SiO<sub>x</sub> barrier (red) is favorable for implementation into LPC-Si solar cell devices. Lower SiO<sub>x</sub> thicknesses were not evaluated since they are known to degrade the electronic properties of a solar cell [18].

In order to compare different silicon absorber layer thicknesses (Fig. 3c) on sinusoidal (diamond, red) and on MST (circle, blue) textured substrates to a planar reference device (square, black), the maximum achievable short-circuit current density ( $j_{sc,max}$ ) has been calculated based on the measured absorption data assuming that every incident photon generates an electron-hole pair using

$$j_{sc,max} = e \int_{280nm}^{1100nm} \frac{S(\lambda)}{hv} A(\lambda) d\lambda \quad (1)$$

where  $e$  represents the elementary charge,  $h\nu$  the photon energy,  $S(\lambda)$  the spectral intensity under AM1.5g and  $A(\lambda)$  the measured absorption.

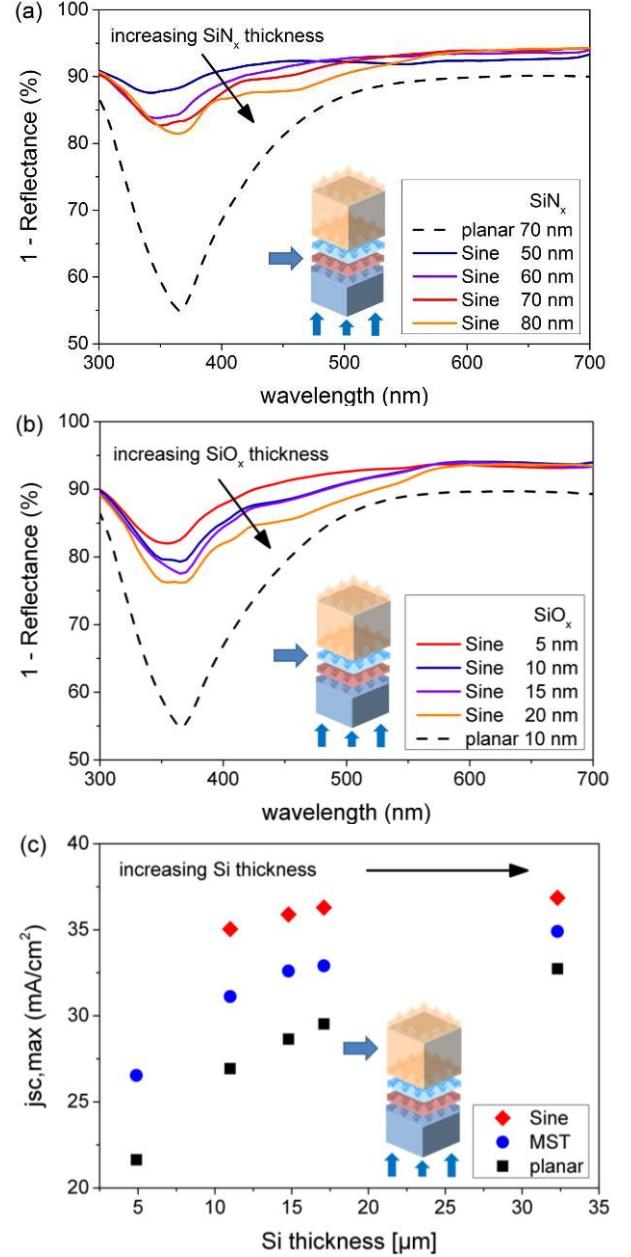


Fig. 3. Reflectance data (plotted as 1-R) of (a) SiN<sub>x</sub> and (b) SiO<sub>x</sub> interlayer thickness variations representatively shown on the sinusoidal texture (solid lines) in comparison to a planar reference (dashed line). The actual thicknesses of the varied SiN<sub>x</sub> / SiO<sub>x</sub> layer are given in the figure. (c) Maximum achievable short-circuit current density ( $j_{sc,max}$ ) of planar references (black, square), randomly MST textured (blue, circle) as well as sinusoidal textured (red, diamond) as a function of silicon absorber layer thickness.

For all texture types  $j_{sc,max}$  rises with increasing silicon absorber layer thickness since the light path inside the absorber layer increases allowing for more electron-hole pairs to be generated. However, for processing time and cost reasons the silicon absorber layer should be chosen as thin as possible. The data point at 5  $\mu\text{m}$  of the sinusoidal texture is missing because this sample delaminated during the LPC process. Nevertheless, independent of the silicon thickness, using a double-sided textured absorber layer increases absorption. For thicker layers this enhancement is less pronounced because with rising amount of light being already absorbed in the silicon absorber layer the amount of light being left to be scattered at the rear side declines. In addition, the rear side texture of the double-side textures smoothens with increasing amount of silicon being deposited on top of the textured substrate reducing the scattering ability of the structures. For all thicknesses the absorption enhancement is more pronounced for the sinusoidal texture (diamond, red) than for the MST texture (circle, blue).

For further experiments a silicon thickness of 15- $\mu\text{m}$  has been selected, where the MST texture enhances the maximum achievable short-circuit current density ( $j_{sc,max}$ ) of the planar reference of 28.6  $\text{mA}/\text{cm}^2$  by +4.0  $\text{mA}/\text{cm}^2$  and the sinusoidal texture by +7.3  $\text{mA}/\text{cm}^2$ .

In summary, based on the results presented in this section, interlayer thicknesses of  $\text{SiN}_x$  70 nm /  $\text{SiO}_x$  5 nm and an absorber layer thickness of 15  $\mu\text{m}$  have been chosen for further processing and analysis regarding their optical (section B) and opto-electric (section C) properties. The corresponding schematic sample stack is depicted in Fig. 1a.

### B. Optical Properties of Optimized Textured LPC Si Absorber Layers

The optical properties of the optimized LPC-Si absorber layer are depicted in Fig. 4.

For wavelengths shorter than 600 nm both texture types reduce average reflection losses (Fig. 4a, reflectance data, dashed lines) from 20.7% for the planar case (black) down to 16.8% for the MST sample (blue) and 11.3% for the sinusoidal sample (red), respectively. This reduction in reflection losses as well as light scattering at the rear side of the absorber layer are translated into an absorption enhancement over the entire wavelength range (solid lines). The corresponding maximum achievable short-circuit current density enhancement is depicted in Fig. 3c.

For wavelength larger than the band gap of silicon at around 1100 nm absorptance is still present for the sinusoidal texture being a hint for defect absorption and, hence, a lower material quality compared to the planar and MST textured device. On the contrary, that the absorptance curves of the planar and MST textured samples do not reduce to zero can be attributed to imperfections of the integrating sphere like opening slits for detectors and incident light. At these long wavelengths

reflection losses are more pronounced for textured devices than for the planar reference because light reaching the rear side of the double-sided textured absorber layer is scattered into higher angles and can partially escape through the side walls of the glass substrate and thus, contributes to the measured reflectance.

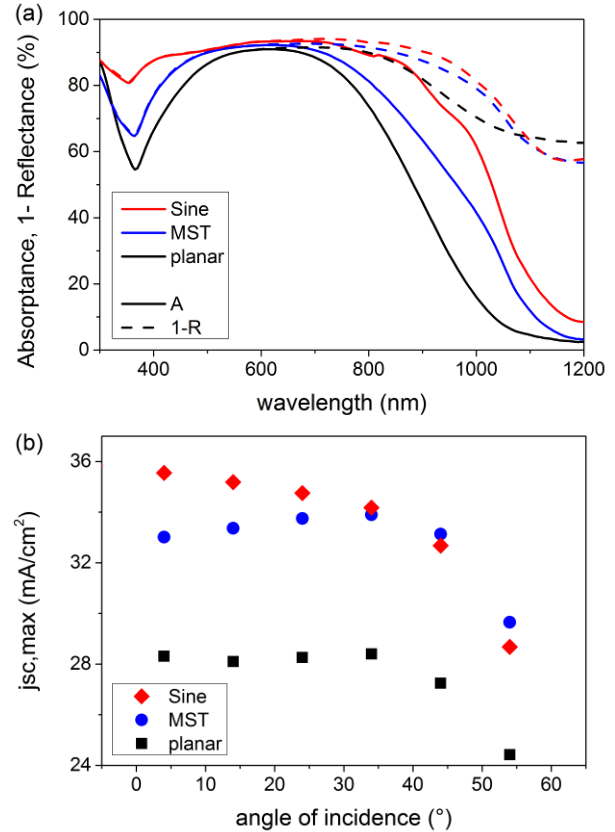


Fig. 4. Comparison of an absorber layer deposited on a planar (black) substrate and on a MST (blue) as well as on a sinusoidal (red) texture. (a) Absorptance (solid lines) and reflectance (dashed lines, plotted as 1-R) as well as (b) absorptance behavior under different incident angles of light are shown.

For usage in a solar cell device the optical behavior under different angles of light (Fig. 4b) plays a crucial role. For comparison  $j_{sc,max}$  has been calculated according to (1) based on the measured absorption spectra for every angle of incidence. While absorptance in the sinusoidal textured absorber layer (red, diamond) declines with increasing angle of incidence, absorption in the MST textured layer (blue, circle) increases. The decline for the sinusoidal texture can be explained by small differences in refractive index between glass substrate, sol-gel and interlayers [20]. Absorptance of the planar reference (black) stays constant until an incident angle around 40 $^\circ$ , for which all structure types start to decline strongly.

Further enhancement of the optical properties should be possible by using a different texture with larger pitch on the rear side of the absorber layer in order to optimize the rear side for light-trapping while keeping the glass-silicon interface texture optimized for anti-reflection [21]–[23].

## V. CONCLUSION

We successfully integrated nanoimprinted 750 nm -periodic hexagonal sinusoidal and wet-etched random modulated surface (MST) textures into liquid phase crystallized (LPC) silicon thin-film solar cell stacks. The optical properties of both textures exceed the optimized planar reference cell over the entire wavelength range, which can be attributed to anti-reflective and light scattering effects of the double-sided textured silicon absorber layers grown on textured substrate. In addition, the influence of the  $\text{SiN}_x / \text{SiO}_x$  interlayers as well as the silicon absorber layer thickness on reflectance has been studied and optimized thicknesses for  $\text{SiN}_x$  (70 nm),  $\text{SiO}_x$  (5 nm) and silicon (15  $\mu\text{m}$ ) have been found. In the wavelength range between 300 nm and 600 nm average reflectance could be reduced by 3.5% (absolute) using a MST texture and 9.4 % (absolute) using a sinusoidal texture. Those results indicate that both texture types are suited for reducing reflection losses in state-of-the-art LPC-Si thin-film solar cells on glass and are promising candidates for future device designs.

## ACKNOWLEDGEMENT

The authors gratefully acknowledge the support of M. Krüger and H. Rhein, Helmholtz-Zentrum Berlin für Materialien und Energie GmbH (HZB), in solar cell preparation. The German Federal Ministry of Education and Research (BMBF) is acknowledged for funding the research activities of the Young Investigator Group ‘Nano-SIPPE’ at HZB in the program NanoMatFutur (no. 03X5520).

## REFERENCES

- [1] T. Frijnts, S. Kühnapfel, S. Ring, O. Gabriel, S. Calnan, J. Haschke, B. Stannowski, B. Rech, and R. Schlattmann, “Analysis of photo-current potentials and losses in thin film crystalline silicon solar cells,” *Sol. Energy Mater. Sol. Cells*, vol. 143, pp. 457–466, Dec. 2015.
- [2] M. Despeisse, C. Battaglia, M. Boccard, G. Bugnon, M. Charriere, P. Cuony, S. Haenni, L. Loeffgren, F. Meillaud, G. Parascandolo, T. Soederstroem, and C. Ballif, “Optimization of thin film silicon solar cells on highly textured substrates,” *Phys. Status Solidi A*, vol. 208, no. 8, pp. 1863–1868, Aug. 2011.
- [3] B. Liu, X. Liang, J. Liang, L. Bai, H. Gao, Z. Chen, Y. Zhao, and X. Zhang, “Broadband light trapping based on periodically textured ZnO thin films,” *Nanoscale*, vol. 7, no. 21, pp. 9816–9824, 2015.
- [4] V. E. Ferry, M. A. Verschuuren, M. C. van Lare, R. E. I. Schropp, H. A. Atwater, and A. Polman, “Optimized spatial correlation for broadband light trapping nanopatterns in high efficiency ultrathin film a-Si:H solar cells,” *Nano Lett.*, vol. 11,

- pp. 4239–4245, 2011.
- [5] C.-M. Hsu, C. Battaglia, C. Pahud, Z. Ruan, F.-J. Haug, S. Fan, C. Ballif, and Y. Cui, “High-Efficiency Amorphous Silicon Solar Cell on a Periodic Nanocone Back Reflector,” *Adv. Energy Mater.*, vol. 2, no. 6, pp. 628–633, 2012.
- [6] M. Sever, J. Krč, and M. Topič, “Prediction of defective regions in optimisation of surface textures in thin-film silicon solar cells using combined model of layer growth,” *Thin Solid Films*, vol. 573, pp. 176–184, 2014.
- [7] V. Jovanov, U. Planchoke, P. Magnus, H. Stiebig, and D. Knipp, “Influence of back contact morphology on light trapping and plasmonic effects in microcrystalline silicon single junction and micromorph tandem solar cells,” *Sol. Energy Mater. Sol. Cells*, vol. 110, pp. 49–57, 2013.
- [8] H. Tan, E. Psomadaki, O. Isabella, M. Fischer, P. Babal, R. Vasudevan, M. Zeman, and A. H. M. Smets, “Micro-textures for efficient light trapping and improved electrical performance in thin-film nanocrystalline silicon solar cells,” *Appl. Phys. Lett.*, vol. 103, no. 17, p. 173905, 2013.
- [9] H. Sai, K. Saito, N. Hozuki, and M. Kondo, “Relationship between the cell thickness and the optimum period of textured back reflectors in thin-film microcrystalline silicon solar cells,” *Appl. Phys. Lett.*, vol. 102, no. 5, pp. 53505–53509, 2013.
- [10] V. Preidel, D. Amkreutz, J. Haschke, M. Wollgarten, B. Rech, and C. Becker, “Balance of optical, structural, and electrical properties of textured liquid phase crystallized Si solar cells,” *J. Appl. Phys.*, vol. 117, no. 22, p. 225306, Jun. 2015.
- [11] G. Köppel, B. Rech, and C. Becker, “Sinusoidal nanotextures for light management in silicon thin-film solar cells,” *Nanoscale*, vol. 8, no. 16, pp. 8722–8728, 2016.
- [12] O. Isabella, J. Krč, and M. Zeman, “Modulated surface textures for enhanced light trapping in thin-film silicon solar cells,” *Appl. Phys. Lett.*, vol. 97, no. 10, p. 101106, 2010.
- [13] G. Yang, R. a. C. M. M. van Swaaij, H. Tan, O. Isabella, and M. Zeman, “Modulated surface textured glass as substrate for high efficiency microcrystalline silicon solar cells,” *Sol. Energy Mater. Sol. Cells*, vol. 133, pp. 156–162, Feb. 2015.
- [14] G. Yang, R.A.C.M.M van Swaaij, O. Isabella, “WO2015093966A1.pdf,” WO 2015093966 (A1).
- [15] J. Dore, R. Evans, B. Eggleston, S. Varlamov, and M. A. Green, “Intermediate Layers for Thin-Film and Polycrystalline Silicon and Solar Cells and on Glass and Formed by Diode Laser Crystallization,” in *MRS Proceedings*, 2012, vol. 1426, pp. 63–68.
- [16] I. Höger, M. Himmerlich, A. Gawlik, U. Brückner, S. Krischok, and G. Andrä, “Influence of intermediate layers on the surface condition of laser crystallized silicon thin films and solar cell performance,” *J. Appl. Phys.*, vol. 119, no. 4, p. 045306, Jan. 2016.
- [17] C. Becker, V. Preidel, D. Amkreutz, J. Haschke, and B. Rech, “Double-side textured liquid phase crystallized silicon thin-film solar cells on imprinted glass,” *Sol. Energy Mater. Sol. Cells*, vol. 135, pp. 2–7, Apr. 2015.
- [18] D. Amkreutz, W. D. Barker, S. Kühnapfel, P. Sonntag, O. Gabriel, S. Gall, U. Bloeck, J. Schmidt, J. Haschke, and B. Rech, “Liquid-Phase Crystallized Silicon Solar Cells on Glass: Increasing the Open-Circuit Voltage by Optimized Interlayers for n- and p-Type Absorbers,” *IEEE J. Photovoltaics*, vol. 5, no. 6, pp. 1757–1761, Nov. 2015.
- [19] D. Lockau, M. Hammerschmidt, J. Haschke, M. Blome, F. Ruske, F. Schmidt, and B. Rech, “A comparison of scattering and non-scattering anti-reflection designs for back contacted polycrystalline thin film silicon solar cells in superstrate configuration,” in *SPIE Proceedings*, 2014, vol. 9140, no. 914006, p. 914006.

- [20] K. Jäger, C. Barth, M. Hammerschmidt, S. Herrmann, S. Burger, F. Schmidt, and C. Becker, "Simulations of sinusoidal nanotextures for coupling light into c-Si thin-film solar cells," *Opt. Express*, vol. 24, no. 6, p. A569, Mar. 2016.
- [21] K. X. Wang, Z. Yu, V. Liu, Y. Cui, and S. Fan, "Absorption Enhancement in Ultrathin Crystalline Silicon Solar Cells with Antireflection and Light-Trapping Nanocone Gratings," *Nano Lett.*, vol. 12, no. 3, pp. 1616–1619, 2012.
- [22] M. Z. Pakhuruddin, J. Dore, J. Huang, and S. Varlamov, "Effects of front and rear texturing on absorption enhancement in laser-crystallized silicon thin-films on glass," *Jpn. J. Appl. Phys.*, vol. 54, no. 8S1, p. 08KB04, Aug. 2015.
- [23] X. Meng, E. Drouard, G. Gomard, R. Peretti, A. Fave, and C. Seassal, "Combined front and back diffraction gratings for broad band light trapping in thin film solar cell," *Opt. Express*, vol. 20, no. S5, p. A560, 2012.