

PANI-derived polymer/ Al_2O_3 nanocomposites: Synthesis, characterization and electrochemical studies

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Abstract

This paper presents the physicochemical, conductive, and electrochemical properties of different polyaniline(PANI)-derived polymer/Al₂O₃ nanocomposites synthesized by chemical oxidation polymerization method carried out in two-stages. First, activation of the surface of the Al₂O₃ nanoparticles by hydrochloric acid. Second, polymerization of 2-chloroaniline (2ClANI), aniline (ANI) and the co-polymer (2ClANI-ANI) in the presence of Al₂O₃ using ammonium persulfate as oxidant in aqueous hydrochloric acid. XRD and TEM results reveal the growth of the polymers on Al₂O₃ nanoparticles and the formation of PANI-derived polymer/Al₂O₃ nanocomposites. FTIR and UV-Vis show a systematic shifting of the characteristic bands of the polymers with the presence of Al₂O₃ nanoparticles. Moreover, these nanoparticles enhance the thermal stability of the polymers, as found by thermogravimetric analysis (TGA). Although the incorporation of Al₂O₃ nanoparticles reduces the electric conductivity of the polymers, the resulting nanocomposites still keep high conductivities, ranging between 0.3×10^{-2} to 9.2×10^{-2} S cm⁻¹. As a result, the polymer/Al₂O₃ nanocomposites exhibit a good voltammetric response. All these synergetic features of the nanocomposites are assigned to the effective interaction of the polymers and Al₂O₃ particles at nano-scale.

Keywords: Polyaniline; 2-Chloroaniline; Aluminum oxide; Nanocomposite; Electrochemical properties.

Introduction

Conducting polymers have been extensively studied for various applications [1, 2]. Among them, polyaniline is one of the most promising due to its unique properties: easy synthesis in aqueous media, excellent ambient stability and simplicity in doping [3, 4]. The doping level can be controlled through a non-redox acid doping/base dedoping process [5]. By changing the doping level, the conductivity of polyaniline can be modified to suit specific applications. On the other hand, unlike acids and bases, oxidizing and reductant chemicals can change the conductivity of polyaniline by changing its inherent oxidation state. The conductivity of polyaniline depends on both the oxidation state of the main polymer chain and the degree of protonation on imine sites [6].

Furthermore, functional additives incorporated into the polymer matrix, such as metals, metal oxides and enzymes can also affect the characteristics of polyaniline [7]. Such a high versatility for tailoring its conductivity has made polyaniline an attractive material for a broad scope of design and development of smart sensors [7].

Magnetic nanostructures have been intensively studied due to the fact that they can be used in a wide range of applications such as clinical diagnosis, mineral separation, magnetic storage devices, absorption of microwave radiation, magneto-optic materials, microwave filters and hybrid materials [8-10].

The high purity metal oxides nanoparticles were synthesized as yet by variety methods such as precipitation, hot-air spray pyrolysis, sol-gel, hydrothermal and sonochemical [9-15].

Regarding these additives, the development of inorganic/polymer nanocomposite materials has received significant interest due to the wide range of potential applications in optoelectronic devices [16, 17] and electromagnetic interference shielding [18]. These

composites have been prepared by different techniques and they exhibit a synergetic behavior between the polymer and the inorganic material [8-12, 45,46, 19]. Since at nano-scale the surface to volume ratio is large, the nano-sized inorganic material is expected to largely modify the thermal, electrical, optical and dielectric properties of the polymer.

In addition, the different properties of components provide the nanocomposites with preferential capabilities for specific applications. For example, alumina has a large surface area, adsorption capacity, and shows good wear resistance [20, 21], so it is commonly used as catalyst, catalyst support, filter, and filler for polymers to improve mechanical properties [22, 23]. At least, seven different transitional aluminas have been reported and Al_2O_3 nanoparticle is perhaps, the most important one for its industrial applications [13,15, 24, 45].

In the present investigation, different polymers and copolymers based on aniline and chloroaniline and their composites with Al_2O_3 were prepared by in situ chemical oxidative polymerization using hydrochloric acid as dopant and ammonium persulphate as an oxidant. The resulting samples were characterized by FTIR, UV and XRD. The thermal properties of the materials were studied by TGA analysis and the electrical conductivities were measured by using the four-point probe method. The electrochemical behavior of the nanocomposites has been analyzed by cyclic voltammetry.

Experimental

Materials

The monomer aniline (ANI) (Aldrich) was distilled under vacuum prior to use, whereas 2-Chloroaniline (2ClAni) (Aldrich) was used as received. Perchloric acid and hydrochloric acid (Merck) were suprapur quality and all the solutions were freshly prepared with distilled-deionised water obtained from an Elga Labwater Purelab Ultra system.

Aluminum oxide nanoparticles (Al_2O_3) (99%) was purchased from Sigma-Aldrich. Ammonium persulfate (APS) and ammonia solution (NH_4OH) were of analytical purity and used without further purification.

Chemical synthesis of nanocomposites

The homopolymers of chloroaniline (poly(2ClAni)) and aniline (PANI), the copolymer of aniline with chloroaniline (poly(2ClAni-co-Ani)), and their corresponding nanocomposites with Al_2O_3 : poly(2ClAni)/ Al_2O_3 , poly(2ClAni-co-Ani)/ Al_2O_3 and PANI/ Al_2O_3 , respectively, were synthesized by the in-situ chemical oxidative polymerization method [25]. This polymerization procedure is schematized in Fig. 1 [26]. Firstly, a certain amount of Al_2O_3 nanoparticles was dispersed in HCl (0.1 M) under simultaneous mechanical stirring and sonication for 1 h, to activate the surface of Al_2O_3 . Then, an equimolar solution (0.25 mol) of the monomer (2ClAni or/and Ani) and hydrochloric acid was prepared and kept at 5 °C. 0.5 g of activated Al_2O_3 nanoparticles were added to the above solution and subsequently submitted to vigorous stirring (30 min) to keep the activated Al_2O_3 suspended in the solution. Afterwards, a pre-cooled solution of APS (1 M) was added drop-wise under constant stirring. The reaction was allowed to process at about 5 °C for 24 h. Since the surface charge of Al_2O_3 is positive in acidic conditions, an amount of Cl^- is adsorbed on the surface of nanoparticles to compensate the positive charges. In the same acidic conditions, the monomers (2-ClAni and/or Ani) are converted to cationic anilinium ions. This leads to electrostatic interactions between the adsorbed anions and cationic anilinium ions. The precipitates were filtered and washed several times with distilled water and acetone until the filtrate was colorless. The final products were maintained in 50 mL of 1 M NH_4OH at room temperature while magnetically stirring for 2 h. The precipitate product was collected by

filtering and after washing with deionized water and drying under vacuum at 60°C for 24 h [25, 27].

Physicochemical Characterization

The X-ray diffraction of the powder nanocomposites were obtained using a Bruker CCD-Apex equipment with a X-ray generator (Cu $K\alpha$ and Ni filter) operated at 40 kV and 40 mA. A Hitachi U-3000 spectrophotometer was used for recording the UV-Vis spectra. The homopolymers and co-polymers were dissolved in N-methyl-2-pyrrolidone (NMP). Fourier transform infrared (FT-IR) spectroscopy was recorded using a Bruker Alpha.

For Transmission Electron Microscopy (TEM) observations, the samples were dispersed in ethanol and supported on TEM grids. The images were collected using a JEOL (JEM-2010) microscope, working at an operation voltage of 200 kV. The TEM is coupled with Energy Dispersive X-ray Spectroscopy (EDS) for the elucidation of chemical composition of the samples.

The thermogravimetric analysis (TGA) was performed with a Du Pont thermogravimetric analyzer at a heating rate of 20 °C/min under nitrogen. About 10 mg of sample was heated up to 900°C.

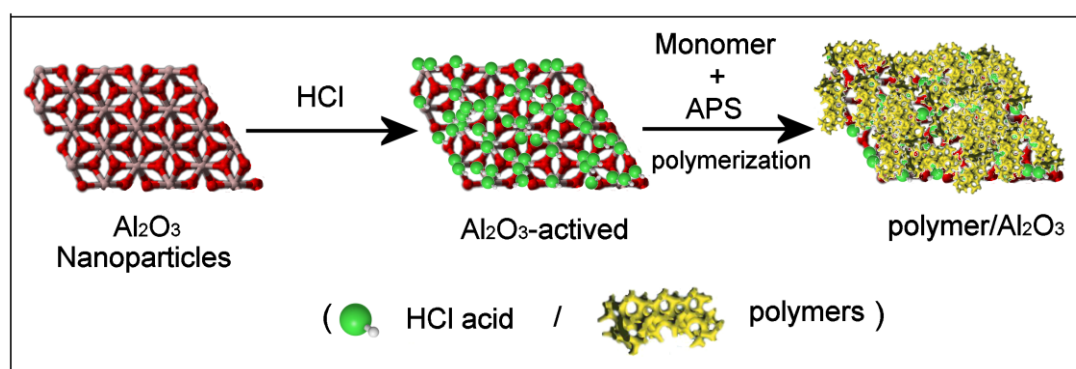


Fig. 1. Scheme of polymer/Al₂O₃ nanocomposites preparation.

Electrochemical characterization

The electrochemical behavior of the nanocomposites and the polymers was studied by cyclic voltammetry. The nanocomposite was first dissolved in N-methyl-2-pyrrolidone (NMP) in which the polymers are soluble; then the dissolved polymers are extracted from the nanocomposites [25]. Then, a drop of resulting solution was placed on the glassy carbon electrode (0.07 cm^2 of geometrical area) and dried in air under an infrared lamp to remove the solvent. The electrochemical measurements were carried out using a conventional three-electrodes cell. The counter and reference electrodes were a platinum wire and a reversible hydrogen electrode (RHE) immersed in the supporting electrolyte, respectively. The electrolyte used was 1 M HClO_4 and all experiments performed at 50 mV s^{-1} .

Electrical conductivity measurements

Electrical conductivity measurements were carried out by using a Lucas Lab resistivity equipment of four probes in-line. The samples were dried under vacuum for 24 h and shaped into pellets (0.013 m diameter) by using a FTIR mold and applying a pressure of $7.4 \times 10^8\text{ Pa}$.

Results and discussion

Fourier-transform infrared spectra (FTIR)

The IR spectra of poly(2ClAni), poly(2ClAni-co-Ani), Al_2O_3 nanoparticles and poly(2ClAni)/ Al_2O_3 , poly(2ClAni-co-Ani)/ Al_2O_3 and PANI/ Al_2O_3 nanocomposites are shown in Fig. 2. The IR characteristic bands of all samples are tabulated in Table 1.

The $-\text{N}-\text{H}$ stretching bands of pure polymers appear at around 3231 cm^{-1} [27]. The characteristic bands due to quinoid and benzenoid rings appear around 1563 cm^{-1} and 1495 cm^{-1} , respectively [27, 28]. In the case of poly(2ClAni), the band at 1297 cm^{-1} is due to

aromatic -C-N- stretching vibration, whereas that at around 1174 cm^{-1} corresponds to charged species ($\text{B-N}^+\text{H-B/Q=N}^+\text{H-B}$) present in the emeraldine salt structure [28]. The bands at 815 cm^{-1} have been assigned to -C-H out of plane bending vibration of 1, 2, 4 tri-substituted aromatic rings [29]. The band at 615 cm^{-1} is attributed to the chlorine atom attached to the phenyl ring [29]. All the above bands confirm the formation of poly(2-ClAni). The ratio of the area of benzenoid and quinonoid band is 1:3 which indicates the formation of emeraldine salt structure [30].

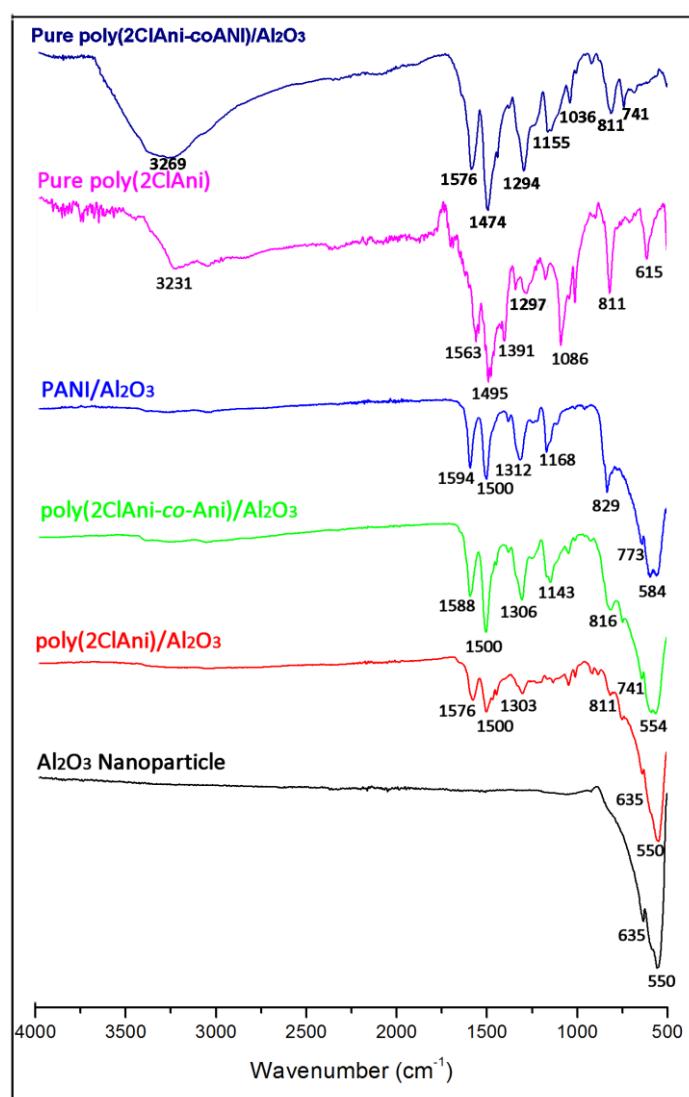


Fig. 2. FT-IR adsorption spectra of Al_2O_3 nanoparticle, poly(2ClAni), poly(2ClAni-co-Ani), poly(2ClAni)/ Al_2O_3 , poly(2ClAni-co-Ani)/ Al_2O_3 and PANI/ Al_2O_3 nanocomposites.

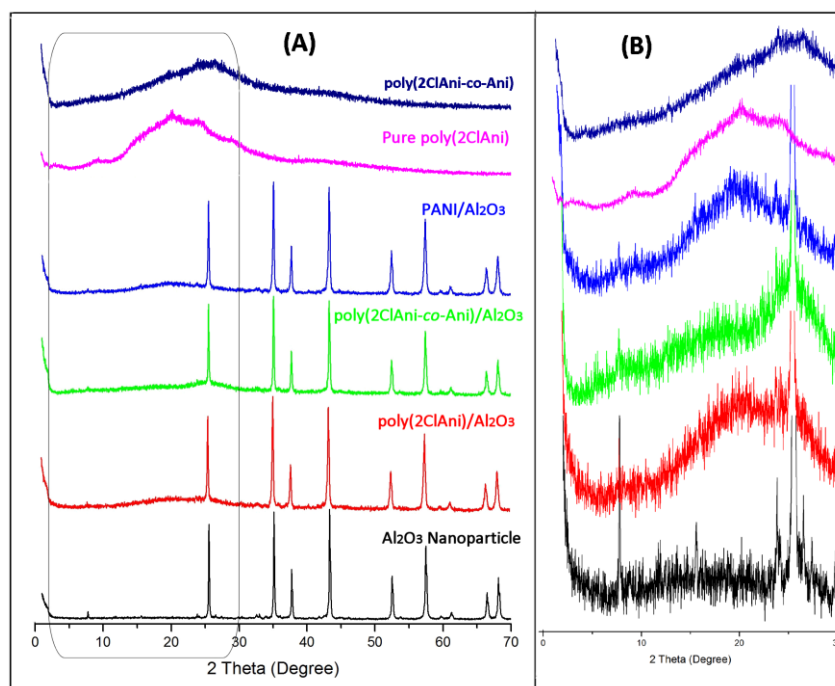


Fig. 3. XRD diffraction patterns of poly(2ClAni), poly(2ClAni-co-Ani), Al₂O₃ nanoparticle, poly(2ClAni)/Al₂O₃, poly(2ClAni-co-Ani)/Al₂O₃ and PANI/Al₂O₃ nanocomposites : (A) Experimental result between 2-70 (2 θ) ; (B) zoom-in inset for the angles 2–30 (2 θ).

In the case of poly(2ClAni-co-ANI), these bands are also observed and the specific positions are included in Table 1.

The IR spectrum of Al₂O₃ nanoparticles is represented by two clear bands located at 550 cm⁻¹ and 635 cm⁻¹. These bands have been assigned to condensed AlO₆ octahedral, i.e. the building units in alumina structure, and are considered a fingerprint of this material [31].

The presence of Al₂O₃ bands in the IR spectrum of poly(2ClAni)/Al₂O₃ and poly(2ClAni-co-ANI)/Al₂O₃ nanocomposites confirms the incorporation of the nanoparticles into the polymer matrix. Although with lower intensity, the band related to charged species at 1174 cm⁻¹ and 1155 cm⁻¹ points out that the polymers in the poly(2ClAni)/Al₂O₃ and poly(2ClAni-co-ANI)/Al₂O₃ nanocomposites are in the emeraldine base form. Nevertheless, the bands characteristic of -N-H and benzenoid and quinonoid rings are all shifted in the

polymer/ Al_2O_3 blend. These results suggest that the Al_2O_3 nanoparticles effectively interact with the polymer chains, affecting their vibrational modes and, therefore, their structural and physicochemical properties.

X-ray diffraction (XRD) studies

The crystalline structure of the different blends was characterized by XRD measurements. As shown in Fig. 3a, the diffraction peaks at $2\theta = 25.55^\circ$, 35.14° , 37.74° , 43.38° , 52.51° and 57.53° are in good agreement with the standard profile of Al_2O_3 , corresponding to the crystal planes of (012), (104), (110), (113), (024) and (116) (JCPDS card No. 87-0245), respectively. However, some obvious changes are observed on the spectra of these blends at low bragg angles, with the appearance of a broad peak between 10 to 30° (Fig. 3b) after polymerization. These broad peaks are characteristic of amorphous structures, so they are attributed to the presence of polymers, confirming the formation of polymer/ Al_2O_3 nanocomposites by the in-situ polymerization approach [32].

Electrical conductivity characterization

The electrical conductivity of poly(2ClAni), poly(2ClAni-co-Ani), PANI, poly(2ClAni)/ Al_2O_3 , poly(2ClAni-co-ANI)/ Al_2O_3 and PANI/ Al_2O_3 nanocomposites was found to vary between 14.8×10^{-2} and $2.52 \times 10^{-3} \text{ S}\cdot\text{cm}^{-1}$ (Table 2). Seanor et al. [33] observed that the conductivity range for semi-conductive materials is known to be $10^{-7} - 10^2 \text{ S}\cdot\text{cm}^{-1}$. Conductivities of most doped conducting polymers lie approximately in this region [34]. The addition of Al_2O_3 nanoparticles in the different polymer matrices causes a conductivity decrease with respect to the pure polymers. This result is expected, because alumina is a non-conducting material. Polymers like (poly(2ClAni), poly(2ClAni-co-ANI) or PANI have a conjugated system which enables easy transportation of electrons. In this sense, the decreased

conductivity of nanocomposites suggests that Al_2O_3 nanoparticles interrupt and/or hindrance the electron transportation path of polymers, probably by decreasing the length of polymer chains. Once again, these results are indicative of a strong interaction at nano-interphase between the polymer chains and the Al_2O_3 nanoparticles.

UV-Vis spectroscopy

The UV-visible spectra of PANI/ Al_2O_3 , poly(2ClAni)/ Al_2O_3 , poly(2ClAni-co-ANI)/ Al_2O_3 nanocomposite and Al_2O_3 nanoparticles are shown in Fig. 4. Several absorption bands are observed, the first one between 314 and 372 nm is assigned to the π - π^* transition and it is related to the extended conjugation between adjacent rings in polymeric chain. Considering that the polymers studied are less conjugated than PANI, it is reasonable that this band occurs at lower wavelength (in PANI appears at 326 nm) [26, 27]. The second band between 539 and 616 nm, in the visible zone, is assigned to the transition of the exciton of the quinone and is related to the hopping electronic intra and interchain [35, 36]. The maximum of this band depends on the polymer oxidation, and also shows a hypsochromic shift from 630 nm (in the case of PANI [36]) to 615 nm for the poly(2ClAni) and poly(2ClAni-co-ANI). This effect can be due to steric hindrance of chlorine in the PANI structure which disturbs the coplanarity of the π system avoiding electron hopping, decreasing the conductivity. Finally, the intensity of this second band decreases drastically for the nanocomposites in comparison to the polymers and also are shifted to lower wavelength, this result is in agreement with the decrease in conductivity of the nanocomposites.

Thermogravimetric analysis (TGA)

The thermal stability of the different materials was evaluated by thermogravimetry. Fig. 5. shows the TG of poly(2ClAni), poly(2ClAni-co-ANI), PANI/ Al_2O_3 ,

poly(2ClAni)/Al₂O₃ and poly(2ClAni-co-ANI)/Al₂O₃ nanocomposites. The curve of pure Al₂O₃ nanoparticles only shows a small weight loss, occurring below 400°C, that can be ascribed to the elimination of water and ethanol and the partial dehydroxylation of the Al₂O₃ nanoparticles [37]. In fact, the influence of the inorganic nanoparticles on the thermal stability of polymer/Al₂O₃ nanocomposites is very complex; which is related to many factors; such as the synthetic method; type of inorganic nanoparticles; structure of composites; the interaction between two components; and so on [38].

At temperatures lower than 130 °C, poly(2ClAni) presents a weight loss of 1.4 wt% , attributed to the volatilization of water absorbed by the polymer in the form of moisture [39]. In the temperature range of 130-210 °C, poly(2ClAni) loses about 3.1 wt% of its weight and losing in total about 54.5 wt% between 210-550 °C.

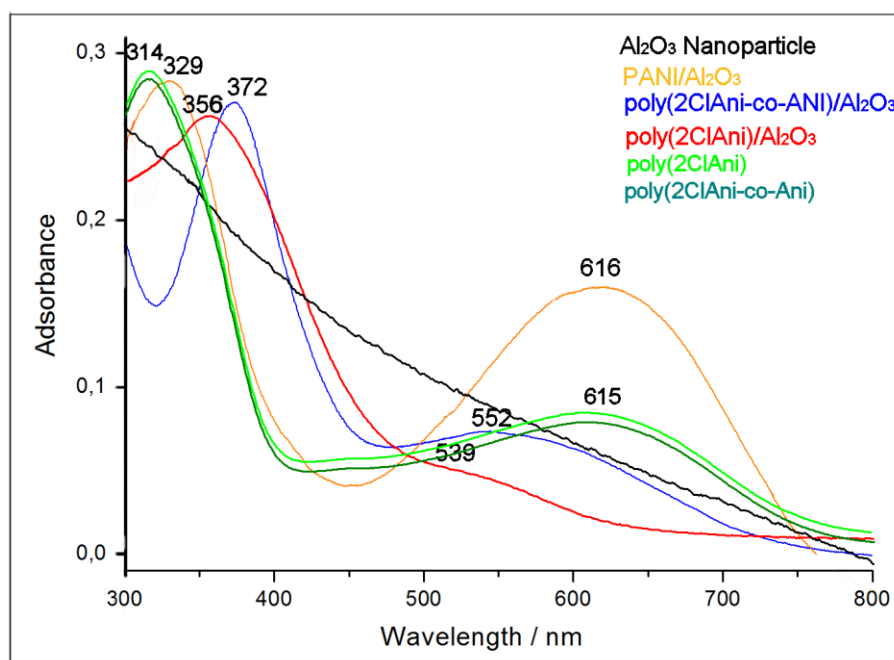


Fig. 4. UV-vis spectra of Al₂O₃ nanoparticle, poly(2ClAni)/Al₂O₃, poly(2ClAni-co-ANI)/Al₂O₃ and PANI/Al₂O₃ nanocomposites.

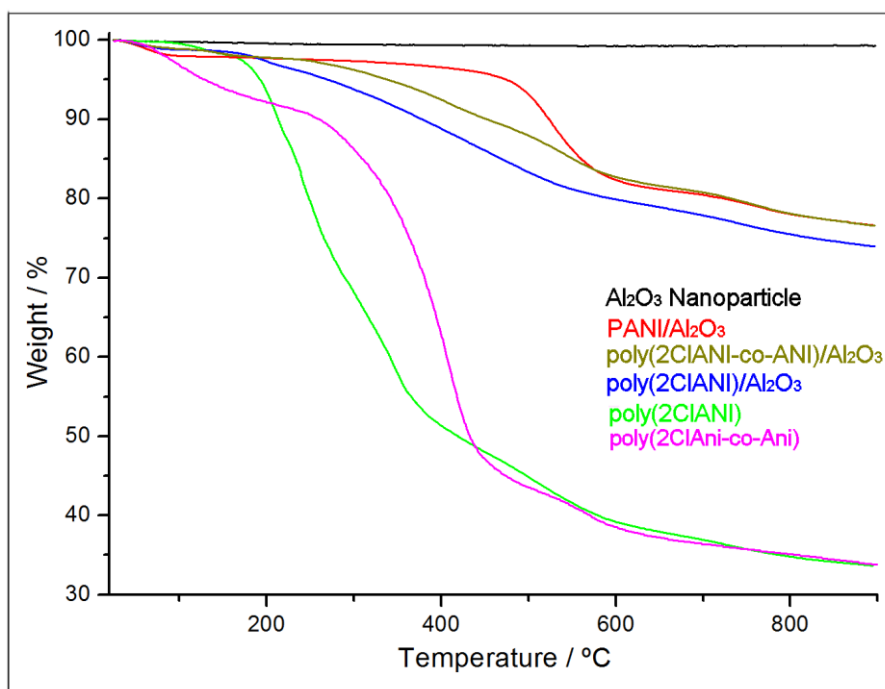


Fig. 5. Thermogravimetric analysis of poly(2ClAni), poly(2ClAni-co-Ani), Al₂O₃ nanoparticle, poly(2ClAni)/Al₂O₃, poly(2ClAni-co-Ani)/Al₂O₃ and PANI/Al₂O₃ nanocomposites obtained in nitrogen atmosphere at heating rate of 10°C/min.

The thermogram of poly(2ClAni-co-Ani) shows three distinct regions of weight loss. The initial weight loss below 120 °C (ca. 4.5 %) is attributed to the loss of water molecules. The weight loss of about 9.2 % by 240°C corresponds to the loss of acid. Beyond 240°C, the rate of decomposition is found to be maximum at 560°C and the weight loss observed is about 59.6 % wt.

The TGA curve of the poly(2ClAni)/Al₂O₃ nanocomposite also shows a weight loss between 25°C and 110°C attributed to the loss of water from both Al₂O₃ and polymer matrix. Besides a differentiated behaviour marked by a strong weight loss in the 100-450°C range is observed. Comparing poly(2ClAni)/Al₂O₃ nanocomposite with poly(2ClAni) pure; it can be seen that the incorporation of Al₂O₃ nanoparticles results in a lower weight-loss.

TGA of PANI pure showed that the first stage of weight loss occurred below 100°C, and this could be related to evaporation of water molecules from PANI. The second weight loss occurred at higher temperature, from 230 to 430°C. It was assigned to loss of dopants from PANI [40]. The third weight loss from 430°C to 640°C was due to the decomposition of PANI chains themselves.

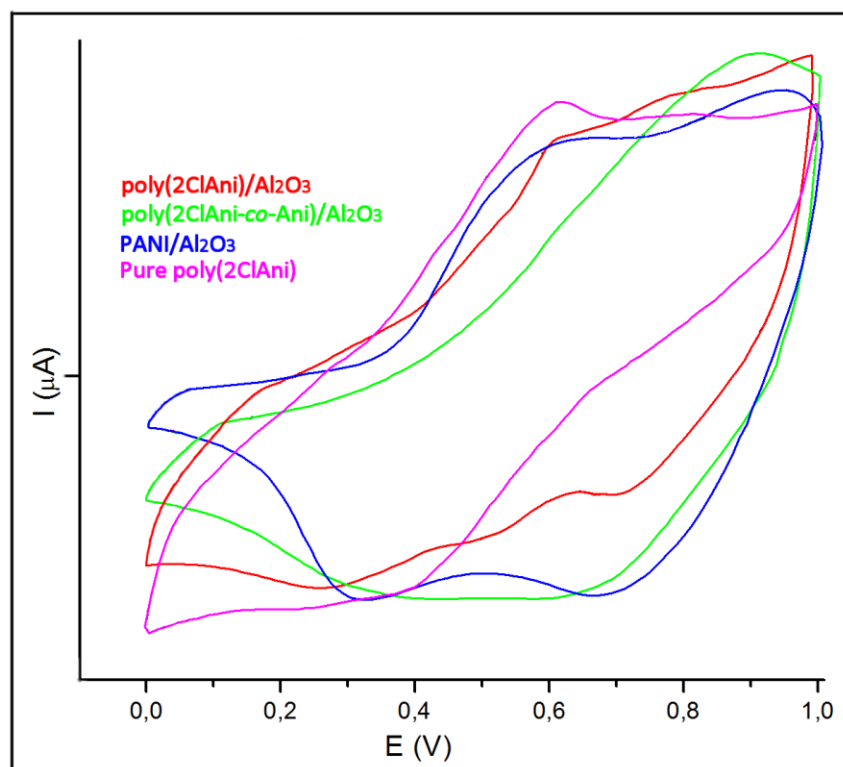


Fig. 6. Cyclic voltammograms recorded for a graphite carbon electrode covered by: poly(2ClAni) pure, poly(2ClAni)/Al₂O₃, poly(2ClAni-co-Ani)/Al₂O₃ and PANI/Al₂O₃ nanocomposites in 1M HClO₄ solution. Scan rate 50 mV/s.

Table 2. The electrical conductivity values of poly(2ClAni), poly(2ClAni-co-Ani), poly(2ClAni)/Al₂O₃, poly(2ClAni-co-Ani)/Al₂O₃ and PANI/Al₂O₃ nanocomposites

Samples	PANI	poly(2ClAni)	poly(2ClAni-co-ANI)	poly(2ClAni)/Al ₂ O ₃	poly(2ClAni-co-ANI)/Al ₂ O ₃	PANI/Al ₂ O ₃
Conductivity (S.cm ⁻¹)	14.8x10 ⁻²	3.89x10 ⁻²	8.22x10 ⁻²	2.52x10 ⁻³	6.37x10 ⁻³	9.21x10 ⁻²

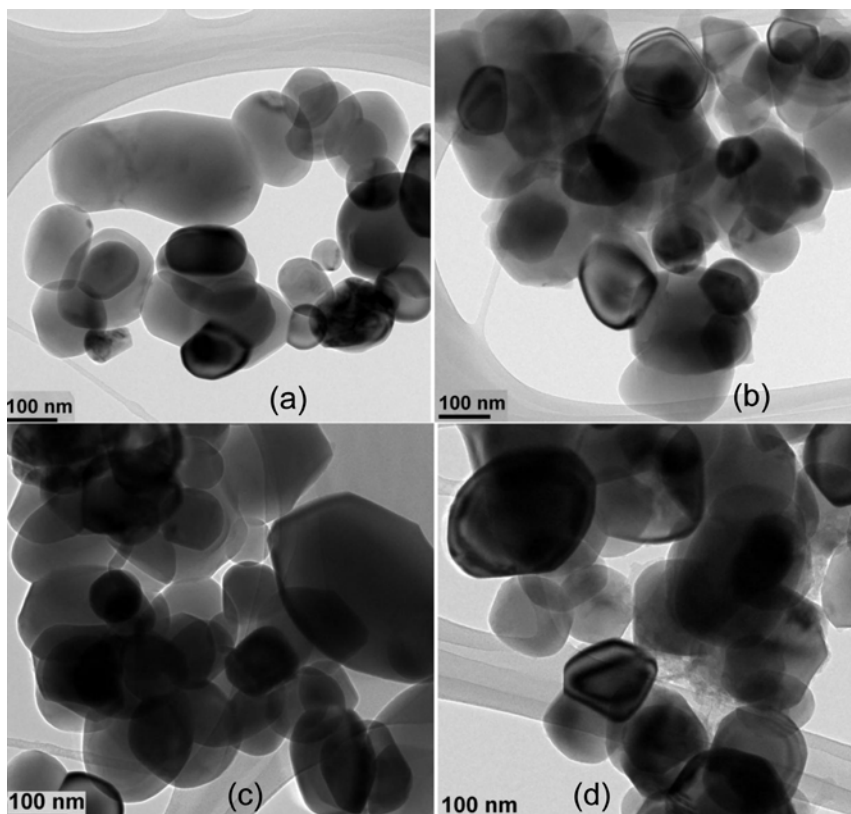


Fig. 7. TEM images of Al_2O_3 nanoparticles, poly(2ClAni)/ Al_2O_3 , poly(2ClAni-co-Ani)/ Al_2O_3 and PANI/ Al_2O_3 nanocomposites.

The TGA of PANI/ Al_2O_3 nanocomposite shows three weight losses. The first decomposition step occurs from 80°C to 150°C , incurring about 2.2% weight loss, corresponding to the loss of water coming from alumina and the polymer matrix. The second weight loss occurs in the temperature range from 150°C to 500°C , which may be due, at lower temperatures to the dedoping from the polymer chain and finally to the degradation and decomposition of organic moiety and the total mass loss at up to 760°C can be estimated to be about 23.07%. These results are similar to those obtained by Y. He [41] who found that ZnO nanoparticles improve the thermal stability of PANI to a certain extent. However; Senthilkumar et al. [42] observed that the thermal stability of PANI was relatively enhanced by forming the nanocomposites with NiFe_2O_4 . It seems that the influence of the inorganic

nanoparticles on the thermal stability of PANI nanocomposites is very complicated; its mechanism remaining to be addressed.

On the other hand, TG of poly(2ClAni)/Al₂O₃ nanocomposite shows a 3.94% mass loss from 208 to 305°C, which could correspond to the dedoping from the polymer chain. After 640 °C, another 18.17% mass-loss is observed due to the complete degradation and decomposition of the poly(2ClAni) backbone, and the total mass loss at up to 800°C can be estimated to be about 22.42%.

Electrochemical characterization

Cyclic voltammetry experiments were performed to test the electroactivity of the polymer and nanocomposites. Fig. 6 shows the steady-state voltammograms of poly(2ClAni) pure, poly(2ClAni)/Al₂O₃, poly(2ClAni-co-Ani)/Al₂O₃ and PANI/Al₂O₃ nanocomposites. In general, all the materials show a good voltammetric response. In particular, the observed redox processes are related to PANI-derived polymers, indicating that the polymerization on Al₂O₃ nanoparticles produces electroactive polymers. Such a good electrochemical response is in agreement with the high conductivities exhibited by the nanocomposites (Table 2). This is a very interesting result, since it demonstrates that a non-conducting material like Al₂O₃ can be successfully integrated into a conductive one by effective interaction with conductive polymers.

The addition of 2ClAni content in polymer, shows a shift of the potential peak of the redox processes associated to PANI being these redox processes less defined. It is thought that an increase in 2ClAni content in the polymer would result in a thicker coating of polymer onto Al₂O₃. Therefore, it was expected that such a thicker polymer coating could not endow effective surface area or a suitable pore structure for easy charge transfer and ion transport. In

the case of PANI/Al₂O₃, the CV presents also two overlapped redox processes. The first one appears at 0.59/0.31V, which results in a potential peak separation (ΔE_p) close to 280mV; the second process is observed at 0.94/0.67V and gives an ΔE_p value of 270mV. These redox processes are attributed to leucoemeraldine/emeraldine and emeraldine/permanganine transitions, respectively [43, 44]. With presence of the 2ClAni, we observe one redox peak.

Transmission Electron Microscopy

Fig. 7 depicts the TEM image of Al₂O₃ nanoparticles, poly(2ClAni)/Al₂O₃, poly(2ClAni-co-Ani)/Al₂O₃ and PANI/Al₂O₃ nanocomposites prepared by the in-situ chemical oxidative polymerization method. The Al₂O₃ nanoparticles (Fig.7a) display nearly spherical morphology and uniform particle size of ca. 100 nm diameter, similar to that observed in literature [45]. On the other hand, TEM images of nanocomposite samples show that the Al₂O₃ nanoparticles are finely dispersed into polymer matrix. This is attributed to intense turbulence and micro-mixing caused by cavitation effects due to polymerization leading to the formation of finely dispersed polymer/Al₂O₃ nanocomposites [46]. Also the particle size confirms that the composites obtained are nanostructured.

Conclusions

The polymer/Al₂O₃ nanocomposite was successfully prepared by using chemical oxidation polymerization using 2Cl-Aniline and/or Aniline monomers and Al₂O₃ in the presence of hydrochloric acid as a dopant with ammonium persulfate as oxidant. The results confirm the existence of polymer and Al₂O₃ in nanocomposites. The FT-IR studies revealed changes on the characteristic vibration bands of the polymers, confirming that they effectively interact with the surface of the Al₂O₃ nanoparticles. TGA studies revealed that the thermal stability of polymers is enhanced in the nanocomposites. The electric conductivity of the

nanocomposites has been studied. The electrochemical properties of PANI/Al₂O₃ and poly(2ClAni-co-ANI)/Al₂O₃ are better than those of poly(2ClANI)/Al₂O₃. This may be attributed to the difference in the length of the polymer chain, which results in the structural change of the polymer/Al₂O₃ interface.

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Statement on disclosure of potential conflicts of interest

We have no conflict of interest to declare on this manuscript

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