

Broadband optical properties of large-area monolayer CVD molybdenum disulfideWei Li,^{1,2} A. Glen Birdwell,³ Matin Amani,³ Robert A. Burke,³ Xi Ling,⁴ Yi-Hsien Lee,^{4,5} Xuelei Liang,^{2,*} Lianmao Peng,² Curt A. Richter,¹ Jing Kong,⁴ David J. Gundlach,¹ and N. V. Nguyen^{1,†}¹*Semiconductor and Dimensional Metrology Division, National Institute of Standards and Technology, Gaithersburg, Maryland 20899, USA*²*Key Laboratory for the Physics and Chemistry of Nanodevices and Department of Electronics, Peking University, Beijing 100871, China*³*Sensors and Electron Devices Directorate, US Army Research Laboratory, Adelphi, Maryland 20783, USA*⁴*Department of Electrical Engineering and Computer Science, Massachusetts Institute of Technology, Cambridge, Massachusetts 02139, USA*⁵*Material Sciences and Engineering, National Tsing-Hua University, Hsinchu, 30013, Taiwan*

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Recently emerging large-area single-layer MoS₂ grown by chemical vapor deposition has triggered great interest due to its exciting potential for applications in advanced electronic and optoelectronic devices. Unlike gapless graphene, MoS₂ has an intrinsic band gap in the visible which crosses over from an indirect to a direct gap when reduced to a single atomic layer. In this paper, we report a comprehensive study of fundamental optical properties of MoS₂ revealed by optical spectroscopy of Raman, photoluminescence, and vacuum ultraviolet spectroscopic ellipsometry. A band gap of 1.42 eV is determined by the absorption threshold of bulk MoS₂ that shifts to 1.83 eV in monolayer MoS₂. We extracted the high precision dielectric function up to 9.0 eV, which leads to the identification of many unique interband transitions at high symmetry points in the MoS₂ momentum space. The positions of the so-called A and B excitons in single layers are found to shift upwards in energy compared with those of the bulk form and have smaller separation because of the decreased interactions between the layers. A very strong optical critical point predicted to correspond to a quasiparticle gap is observed at 2.86 eV, which is attributed to optical transitions along the parallel bands between the *M* and Γ points in the reduced Brillouin zone. The absence of the bulk MoS₂ spin-orbit interaction peak at ~ 3.0 eV in monolayer MoS₂ is, as predicted, the consequence of the coalescence of nearby excitons. A higher energy optical transition at 3.98 eV, commonly occurring in bulk semiconductors, is associated with a combination of several critical points. Additionally, extending into the vacuum ultraviolet energy spectrum are a series of newly observed oscillations representing optical transitions from valence bands to higher conduction bands of the monolayer MoS₂ complex band structure. These optical transitions herein reported enhance our understanding of monolayer MoS₂ as well as of two-dimensional systems in general and thus provide informative guidelines for MoS₂ optical device designs and theoretical considerations.

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I. INTRODUCTION

MoS₂, an emerging new class of atomically thin material down to a unit cell thickness (monolayer), has shown intriguing physical properties and exciting prospects for a variety of applications [1–7]. The monolayer MoS₂ is composed of a hexagonal plane of Mo atoms sandwiched between two hexagonal planes of S atoms in a trigonal prismatic arrangement, while its bulk counterpart can be considered as a stack of many S-Mo-S sheets weakly bonded by van der Waals force [8]. Unlike graphene, there is an intrinsic band gap in MoS₂, which changes from an indirect gap of $E_g \sim 1.3$ eV in bulk to a direct gap in monolayer [8,9]. Therefore, monolayer MoS₂ is more attractive than graphene for future transistors and logic circuit applications where a high on/off current ratio is required [2,5,10–12]. More importantly, due to the fact that the band gap of monolayer MoS₂ is direct with the energy in the visible range of the spectrum, it could be used in light emitting diodes, photodetectors or solar cells, and transparent, light, and flexible devices [1,2,4,6,13]. With such a broad potential application base, a comprehension of the optical properties of MoS₂ will elucidate its electronic

band structure which is critical to electronic and optoelectronic devices researchers. It is known that the connection of the microscopic details manifested in the electronic band structure of a solid with experimental macroscopic observables can be established through a response function. In the case of probing photons and interacting electrons, the commonly used response function is the wave vector and frequency dependent complex dielectric function [14]. In most previous experimental studies of monolayer or ultrathin MoS₂, small flakes (several to tens of micrometers) were produced by mechanical exfoliation [3,5,8,9]. Such a small sample size generally hinders the measurement of the complex dielectric function of MoS₂ by optical techniques. An up-to-date literature survey shows the optical response of MoS₂ has been reported by only a few research groups, and most are theoretical studies [8,9,15–20]. The reported experimental studies focused on photoluminescence (PL), absorbance, optical contrast, and photoconductivity in the vicinity of the *K* point in its Brillouin zone, and the corresponding spectral ranges are rather limited [8,9,19,21–23]. On the other hand, there are limited reports on the broad band optical dielectric function of monolayer MoS₂ in the published literature [24], and it is expected to be distinct from that of bulk MoS₂. Not until very recently could large-area monolayer MoS₂ with sample sizes up to millimeters be grown by chemical vapor deposition (CVD) [25–28]. Such a large-sized sample

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facilitates the optical measurement of MoS₂ optical response function.

Spectroscopic ellipsometry (SE) is a nondestructive and submonolayer-sensitive characterization method that can be used to accurately measure material dielectric functions. Spectroscopic ellipsometry is an important optical method to characterize two-dimensional (2D) materials. For instance, it has been shown to successfully investigate the dielectric functions of graphene [29–31]. Recently, SE has also been used to determine the dielectric function of CVD-grown MoS₂ [28,29]. Yim *et al.* [32] have found that, in their four-phase ellipsometric model, the refraction index (n) and the extinction coefficient (k) of their CVD MoS₂ exhibit a dependence on the MoS₂ film thickness. Shen *et al.* [24], using terahertz (THz) absorption and SE, investigated the charge dynamics to show the semiconductor behavior and a blueshift of ~ 0.2 eV of charge transfer bands compared with those of the bulk material and also determine the dielectric function from 0.73 to 6.42 eV.

In this paper, a comparative study of broadband (1 to 9 eV) dielectric function of monolayer and bulk MoS₂ by SE is experimentally reported. The optical band gap of 1.42 eV was observed for bulk MoS₂, but shifts to a higher energy of 1.83 eV for single-layer MoS₂. The so-called A and B excitons due to the direct d - d transitions in monolayer are accurately derived to be 1.88 and 2.02 eV by lineshape fitting of the experimental data, which are blueshifted as compared with those of the bulk form. A series of optical critical points (CPs) at higher energy than the A and B excitons are clearly observed with a very strong absorption at 2.86 eV, which is attributed to optical transitions along the parallel bands between the M and Γ points in the reduced Brillouin zone. Above these transitions, a critical point at 3.98 eV is observed and can be associated with a combination of several critical points as seen for common semiconductors [14]. Deep in the ultraviolet (UV) region, the dielectric function shows a series of absorption peaks or fine features separated by about 0.7 eV. Our high precision dielectric function data facilitates the accurate optical properties analysis, and the results agree well with previous work including ellipsometry [24], absorption measurements [8], PL measurements [9], and theoretical calculation [16–18].

II. EXPERIMENTAL SECTION

Chemical vapor deposition in a hot-wall furnace is adopted to synthesize a monolayer of MoS₂ directly on SiO₂/Si substrates using MoO₃ and S powders as the reactants. The substrates are treated to make the surface hydrophilic prior to the application of perylene-3,4,9,10-tetracarboxylic acid tetrapotassium salt (PTAS). As discussed in Ref. [33], the PTAS promotes monolayer growth of MoS₂ by serving as a nucleation site for the precursors. Large-area MoS₂ layers can be directly obtained on amorphous SiO₂ surfaces without the need to use highly crystalline metal substrates or an ultrahigh vacuum environment. The resulting MoS₂ films are highly crystalline, and their size is up to several millimeters. Samples were transferred from the growth substrate using the poly(methyl methacrylate) (PMMA) mediated wet method [26]. PMMA was spin-coated on the samples and allowed to dry at room temperature overnight. The PMMA-supported MoS₂ was released from the Si/SiO₂ substrate in a 0.2 M

KOH solution heated to 60°C. The sample was cleaned in three deionized (DI) water baths and then placed on the fused silica substrate. After the sample was allowed to dry at 40°C, it was baked at 180°C to improve adhesion between the MoS₂ and the substrate. The PMMA was removed via an overnight soak in acetone. Bulk MoS₂ was commercially purchased from SPI Supplies.¹

Spectroscopic ellipsometry measurements were performed in a nitrogen gas filled chamber at room temperature on a vacuum UV spectroscopic ellipsometer with a light photon energy from 1.0 to 9.0 eV in steps of 0.01 eV. The SE data were taken at multiple angles of incidence of 65°, 70°, and 75°. The three-phase ellipsometric model consisting of the substrate (SiO₂), thin film (MoS₂), and ambient (air) was employed to extract the dielectric function of MoS₂. Without explicitly expressing the ellipsometric equation of the three-phase structure, it can be written as

$$\tan \psi(\lambda) \exp[i \Delta(\lambda)] = f(\varepsilon_{\text{SiO}_2}, \varepsilon_{\text{MoS}_2}, \varepsilon_{\text{Air}}, d_{\text{MoS}_2}, \phi, \lambda) \quad (1)$$

The known parameters in this equation are the dielectric function of SiO₂ ($\varepsilon_{\text{SiO}_2}$) and air (ε_{Air}), the thickness of MoS₂ (d_{MoS_2}), the angle of incidence (ϕ), and wavelength (λ). The only two unknown parameters are the real and imaginary parts of the complex dielectric function of MoS₂ ($\varepsilon_{\text{MoS}_2}$). Therefore, with two measured parameters ψ and Δ at each photon energy and with the known monolayer MoS₂ thickness 0.65 nm, ε_1 and ε_2 of MoS₂ can be directly calculated. For bulk MoS₂ (thickness > 50 μm), ε_1 and ε_2 are directly calculated from the measured ψ and Δ . Since the thickness of monolayer MoS₂ is only 0.65 nm, it is presumed that the film is optically uniaxial with the out-of-plane dielectric function component different from the in-plane component. Experimentally, for such a ultrathin film, ellipsometry is known to be insensitive to the out-of-plane component and its ellipsometric response is determined by the in-plane component [34,35]. Therefore, in our experiment, ε_1 and ε_2 are considered to be pseudodielectric functions representing the in-plane component. Nonetheless, unlike graphene being a sheet of single atom, and MoS₂ has three atoms in the out-of-plane direction, its out-of-plane dielectric characteristic is arguably similar to that of the in-plane component [36].

Micro-Raman and PL measurements were performed with a WITec Alpha 300RA system using the 532 nm line of a frequency-doubled Nd:YAG laser as the excitation source. The spectra were measured in the backscattering configuration using a 100 \times objective and either a 600 or 1800 grooves/mm grating. The spot size of the laser was ~ 342 nm, resulting in an incident laser power density of ~ 140 $\mu\text{W}/\mu\text{m}^2$. A low laser power was used in order to avoid any significant heat-related effects in the Raman or PL signatures [37]. Raman and PL maps were developed over 25×25 μm regions using a 250 nm

¹Certain commercial equipment, instruments, or materials are identified in this report in order to specify the experimental procedure adequately. Such information is not intended to imply recommendation or endorsement by the National Institute of Standards and Technology, nor is it intended to imply that the materials or equipment identified are necessarily the best available for the purpose.

grid spacing. The integration time was 5 s for Raman and 1 s for PL.

III. RESULTS AND DISCUSSION

Figure 1(a) shows the optical image of the transferred sample, which appears to be free from contaminations and uniform (also see Fig. S1 [38]). Raman spectroscopy is commonly used to characterize the thickness of MoS₂. Since there is no center of inversion for one molecular layer of MoS₂, E' and A' are used instead of the common E_{2g}^1 and A_g^1 notations. These two characteristic Raman modes, E' at ~ 385 cm⁻¹ and A' at ~ 405 cm⁻¹, are sensitive to MoS₂ layer thickness. As the number of MoS₂ layers increases, the frequency of E' decreases, while the frequency of A' increases [39,40]. Thus, the frequency difference between E' and A' is often used to identify the number of MoS₂ layers. Figure 1(b) shows the Raman spectra averaged over the 5×5 μm center area (400 spectra) of the sample in Fig. 1(a). The wave number difference between the E' and A' peak is 19 cm⁻¹, thus confirming that our MoS₂ sample is monolayer [39]. Figures 1(c) and 1(d) display the Raman intensity mapping images of the E' and A' peak, respectively, which were taken from the whole area shown in Fig. 1(a), confirming the uniformity of a monolayer across the area.

It is known that, as the thickness of MoS₂ decreases from bulk to a monolayer, the band gap crosses over from an indirect to direct gap [8,41]. Thus, the PL quantum efficiency of monolayer MoS₂ enhances significantly, more than four orders of magnitude compared with the bulk material [8]. Photoluminescence spectra were also obtained for our CVD-grown monolayer MoS₂ samples. Shown in Fig. 2(a) are the

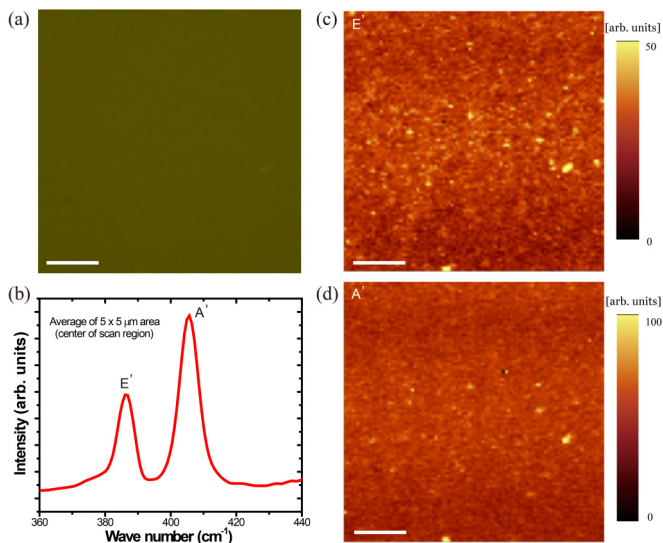


FIG. 1. (Color online) Raman characterization of the CVD-grown MoS₂ on fused silica using a 532 nm laser line with a 342 nm spot size and 12.5 μW power: (a) Optical image of the monolayer MoS₂ on fused silica substrate. (b) Averaged Raman spectroscopy of a 5×5 μm area in the center of the area shown in (a). (c) and (d) Raman mapping images of the E' and A' peak intensity, respectively, taken in the whole area shown in (a) with a 250 nm grid spacing. Scale bar: 5 μm .

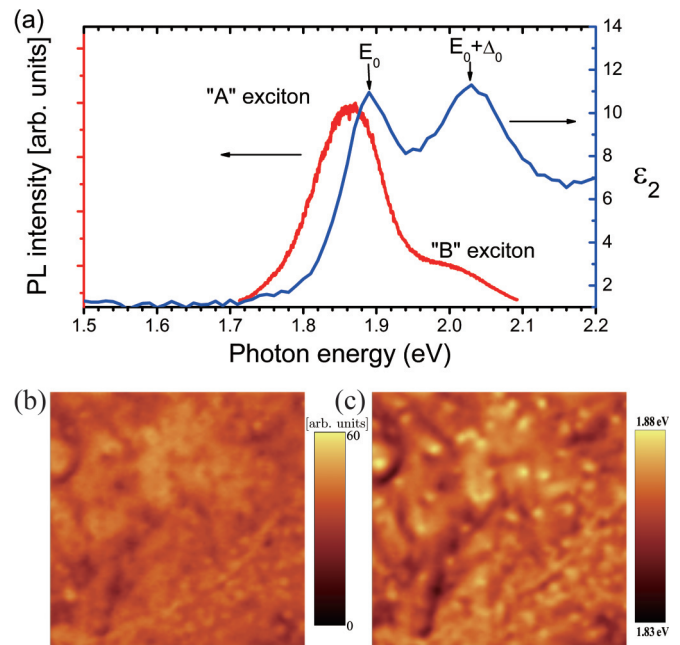


FIG. 2. (Color online) PL characterization of the CVD-grown MoS₂. (a) Averaged PL spectrum (red curve) measured over 5×5 μm area in the center of the area in Fig. 1(a) and the imaginary part (ϵ_2 ; blue curve) of the dielectric function measured by SE. (b) and (c) PL mapping images of the A exciton peak intensity and peak energy, respectively, taken in the area shown in Fig. 1(a) with a 250 nm grid spacing. Scale bar: 5 μm .

PL spectra averaged over the 5×5 μm center area (400 spectra) pictured in Fig. 1(a). Two PL peaks, known as A and B excitons [8,16] are observed at ~ 1.86 and ~ 2.00 eV, respectively, which are consistent with a previous report [9]. Shown in Figs. 2(b) and 2(c) are the PL mapping images of the A exciton peak intensity and energy, respectively. These maps were taken on the whole area of Fig. 1(a), which revealed slightly more variations, up to 50 meV in energy (minimum to maximum) across the sample, than the uniform Raman maps as shown in Figs. 1(c) and 1(d). Such variations may be due to the structural defects in the sample, which are difficult to detect by the E' and A' Raman peaks intensity mapping, but more easily by the PL energy and intensity mapping [22]. Another possible reason for the variations is the result of the fluctuating nature of the interfacial contact between the substrate and MoS₂ that might affect the dielectric screening of the long-range Coulomb interaction [1,21].

The broadband dielectric function is measured by vacuum UV SE in this paper, as schematically illustrated in Fig. 3(a). Spectroscopic ellipsometry is a nondestructive and noncontact technique widely used to characterize optical properties of thin films [34]. Compared to other optical techniques such as transmission and reflection spectroscopy, absorbance and photoconductivity, SE is more advantageous in that both the real (ϵ_1) and imaginary (ϵ_2) part of the complex dielectric function are obtained simultaneously. Once ϵ_1 and ϵ_2 are known, other optical parameters of the material, e.g. refractive index, absorption coefficient, reflectivity, and loss function can be easily derived. The large area and high quality of MoS₂ films can easily accommodate the ellipsometry light

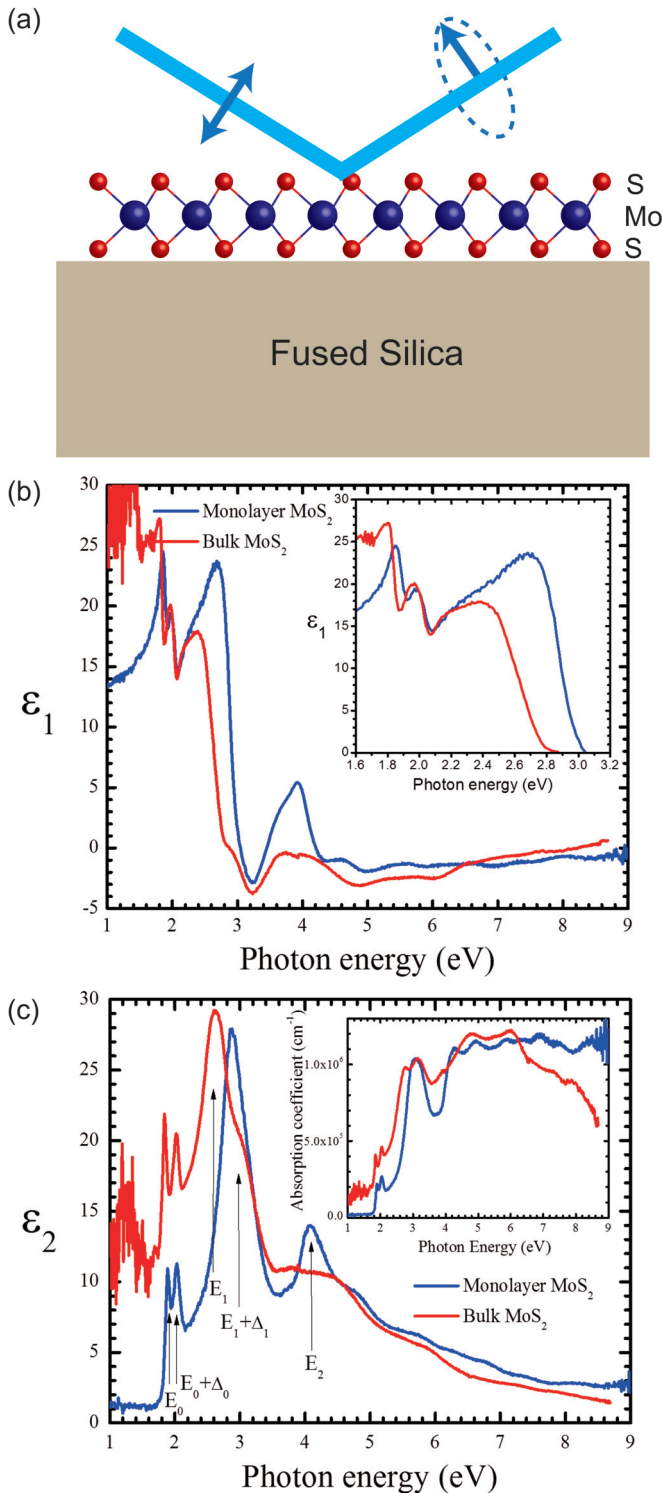


FIG. 3. (Color online) The schematic of ellipsometry measurement and the dielectric function of the CVD-grown monolayer MoS₂. (a) Schematic of SE measurement. (b) The real (ϵ_1) part of the dielectric function of CVD-grown monolayer and bulk MoS₂, inset: zoom-in of the spectra range from 1.6 to 3.2 eV. (c) The imaginary (ϵ_2) part of the dielectric function of CVD-grown monolayer and bulk MoS₂. The inset is the absorption spectrum of monolayer and bulk MoS₂ calculated from the measured dielectric function.

beam size and yield very low noise signals in a very wide energy range (1–9 eV), from which high-order numerical differentiations of the dielectric function can be carried out with high precision. Shown in Figs. 3(b) and 3(c) are the extracted real (ϵ_1) and imaginary parts (ϵ_2), respectively, of the monolayer MoS₂. For comparison, the dielectric function of the bulk material was also measured with SE. For monolayer MoS₂, the most prominent features in the spectra are a series of sharp peaks, i.e., CPs, in the lower energy range, and a series of broad and low oscillations at higher photon energies. The sharp features in ϵ_2 are classified according to a standard nomenclature, where the lowest energy structural feature is denoted by E_0 corresponding to the direct-gap transitions from the valence band maximum to the conduction band minimum at K point in the Brillouin zone [14]. E_0 is immediately followed by the $E_0 + \Delta_0$ peak, which corresponds to the spin-orbital splitting of the valence band at the same K point [42]. These two features are designated as A and B excitons by PL measurements, and they are attributed to be direct-gap transitions [8,9,43]. There appears to be a slight shift in energy of the peak E_0 and $E_0 + \Delta_0$ when compared with peaks A and B obtained by PL shown in Fig. 2(a). In fact, there is a slight variation of about 50 meV of the A peak position across the sample as shown in Fig. 2, which is larger than the shift mentioned above. As a result, within the variation, it can be said that the energies of A and B peaks are the same as those of the E_0 and $E_0 + \Delta_0$ CPs. It is interesting to note that, in contrast to the PL spectra reported by other groups [8,22], where the B peak was absent from monolayer MoS₂ but present for thicker layers, our monolayer PL clearly shows the B peak with a lower luminescence intensity than A peak, which was similarly observed in Ref. [9].

The optical absorption coefficient derived from the extinction coefficient for monolayer and bulk MoS₂ are shown in the inset of Fig. 3(c). Both are comparable with each other, whereas Shen *et al.* [24] obtained much higher values for the monolayer. Their monolayer MoS₂ has higher absorption coefficient than that of bulk MoS₂ in the energy range from 3 to ~ 6.42 eV as a result of much higher extinction coefficient extracted from the SE measurement. Our extinction coefficient at 3 eV is ~ 3.4 (see Fig. S3 [38]), which is comparable with that (~ 3.2) of Yim *et al.* [32] for ultrathin CVD MoS₂, but much lower than that (~ 5.1) of Shen *et al.* [24]. The higher coefficient could be a result of sample structural quality or the way it was extracted from their four-phase ellipsometric model, even though it contains all similar absorption peaks [24].

It is just not accurate to extract the energy critical point by simply locating the energy at which the maximum of absorption spectrum occurs [44]. To accurately determine the energy position of the CPs, it is necessary to locally isolate a CP from the polarization contributions from other nearby CPs. One direct experimental approach is to apply an external perturbation as modulations such as electric or magnetic field or stress to suppress uninteresting background effects and strongly reveal structures localized in the energy [14,45]. Another approach is to produce a similar modulation response by numerical differentiation of the dielectric function that in

effect filters out the above mentioned background [14]. In the vicinity of a CP, the dielectric response can be represented by the following analytic form [46,47]:

$$\varepsilon(E) = C + A\Gamma^{-n}e^{i\theta}(E - E_t + i\Gamma)^n, \quad (2)$$

where each CP is described by five energy independent parameters: the amplitude A , phase angle θ , threshold energy E_t , and phenomenological broadening Γ , which relates to scattering rates, and n . The exponent n characterizes the analytical shape of $\varepsilon(E)$ near its minima or maxima and depends on the signs of the reduced masses of the electron and hole. The value of n is $-1/2$ for one-dimensional (1D), 0 [logarithmic: $\ln(E - E_t + i\Gamma)$] for 2D, and $+1/2$ for three-dimensional (3D) CP. The CP dimension is classified based on the electron energy dispersion at Van Hove singularities, which can be expanded as a function of momentum vector \mathbf{k} about the CP as $E(\mathbf{k}) \sim E(0) + \alpha_1 k_1^2 + \alpha_2 k_2^2 + \alpha_3 k_3^2 + \dots$. The dimension is assigned by the number of nonzero coefficients α . Here, 1D, 2D, and 3D represent one, two, or three, respectively, nonzero α [14,46]. The direct discrete bound exciton is included in the same equation for which $n = -1$. Since the measured $\varepsilon(E)$ is a superposition of different spectra calculated from Eq. (1) and in order to eliminate the effect of the constant background, we perform lineshape fitting to the second derivative of the dielectric function with respect to photon energy

$$\frac{d^2\varepsilon}{dE^2} = \begin{cases} n(n-1)A\Gamma^{-n}e^{i\theta}(E - E_t + i\Gamma)^{n-2}, & n \neq 0 \\ -A\Gamma^{-n}e^{i\theta}(E - E_t + i\Gamma)^{-2}, & n = 0 \end{cases} \quad (3)$$

To ensure an accurate determination of E_t , the CP energy, the fitting was carried out on the real and imaginary parts of the $\frac{d^2\varepsilon}{dE^2}$ simultaneously. Shown in Figs. 4(a) and 4(b) are the fitting of Eq. (2) to the second derivative of the measured dielectric function of the monolayer MoS₂.

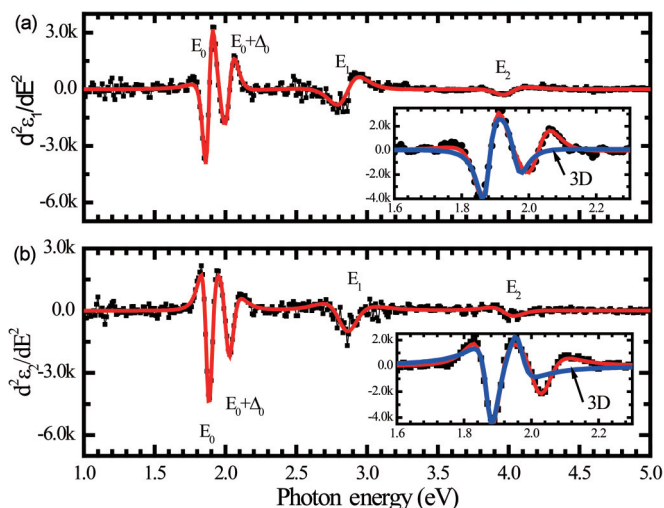


FIG. 4. (Color online) CP determination. Red curves are the fits to the second derivatives (black curves) of the real (ε_1) and imaginary (ε_2) parts of the dielectric function of monolayer MoS₂. The inset is the comparison of the fitting results when using exciton and 3D lineshapes.

For the E_0 and $E_0 + \Delta_0$ peaks, it was found that $n = -1$ yields the best fit, which explicitly confirms that they are excitons. First-principle calculations have predicted that the exciton peak at K point in the Brillouin zone is split into two peaks at ~ 1.88 and ~ 2.02 eV, respectively, by spin-orbital coupling [16]. These two peaks around those energies are observed experimentally and ascribed to the A and B excitons as we discussed above. The insets of Figs. 4(a) and 4(b) are intended to show that the best fit is obtained by using a discrete exciton when compared with the 1D (not shown), 2D (not shown), and 3D [shown as blue lines in the inset of Figs. 4(a)–4(b)] lineshape. The effect of the electron-hole interaction, i.e., excitonic effects, is to enhance the oscillator strength of the transitions that result from the interaction of the discrete exciton with a continuous background [46,47]. Without the excitonic effects, the theory reproduces the low-energy exponential absorption edge of ε_2 but fails to account for the strength of the negative and positive peak. By including Coulomb effects, the discrepancy is eliminated and a good fit is obtained. As a fitting result of Eq. (2) by using $n = -1$, the fitted peak positions are 1.88 and 2.02 eV for the E_0 and $E_0 + \Delta_0$ peaks, respectively. Table I is a compilation of recent theoretical and experimental results showing SE E_0 and $E_0 + \Delta_0$ peaks agree with the latest first-principles calculations [16]. Compared with bulk MoS₂, where the separation of the A and B peak is about 190 meV, these two peaks in monolayer blueshift 50 and 10 meV, respectively [see Fig. 3(b)], which leads to a narrower separation. Such a blueshift and smaller separation in a monolayer result from the decreased interactions between the layers [8,48].

The next higher energy feature in the ε spectra is denoted by E_1 . For the monolayer MoS₂, it is found that the lineshape fitting to this critical point by Eq. (2) yields the same figure of merit for different values of n and yields a resonance of 2.86 eV (see Fig. 4). In contrast to monolayer MoS₂, a lower or redshifted E_1 of ~ 2.60 eV is observed for bulk MoS₂. From the first-principles full-potential linearized augmented plane wave (FLAPW) band structure calculation by Yun *et al.* [49], they show that the lowest conduction band and the highest valence band are parallel over a wide range in the Brillouin zone between M and Γ points, thus leading to a maximum in the joint density of states, and therefore giving rise to the CP E_1 . The energy of E_1 for the bulk MoS₂ is consistent with the first-principles calculated value of 2.60 eV along the M - Γ direction [49]. An additional peak labeled as $E_1 + \Delta_1$ at ~ 3.0 eV is observed in bulk MoS₂, which is due to the large spin-orbit interaction in the valence band, but absent in the spectrum of monolayer [50]. For the monolayer, it is interesting to note that the fitted resonant energy at 2.86 eV (E_1) matches quite well with the theoretical quasiparticle gap (~ 2.84 eV) at K point [16]. It has been also theoretically predicted that electron-phonon interaction in monolayer MoS₂ will produce few excitons in the energy range from 2 to 3 eV. These excitons are expected to coalesce into a much broader peak [16]. In other words, it might be reasonable to speculate that the $E_1 + \Delta_1$ peak observed in bulk, but absent in monolayer, converges into E_1 peak of monolayer. Also, it might explain that all line shape dimensions (1D, 2D, 3D, and exciton) have the same goodness-of-fit to the peak at E_1 as discussed above (see Fig. 4).

TABLE I. The optical transition energies of peaks A and B and quasiparticle band gap from different studies: G_1W_0 , G_0W_0 -Bethe-Salpeter equation (BSE), and $sc-GW_0$ -BSE are theoretical models all using GW -BSE but with different physical parameters (see Ref. [16]).

	This work (SE) Experiment	G_1W_0 [16] Theory	Absorption [8] Experiment	SE [24] Experiment	PL [9] Experiment	G_0W_0 -BSE [17] Theory	$sc-GW_0$ -BSE [18] Theory
Optical transition energies (eV) A	1.88	1.88	1.88	1.88	1.83	1.78	2.22
B	2.02	2.02	2.03 ^a	2.03	1.98	1.96	2.22
Quasiparticle band gap (eV)	2.86	2.84	2.86 ^b	3.00	...	2.82 (G_0W_0)	2.80 ($sc-GW_0$)

^aExtracted from Fig. 4(a) of Ref. [8].

^bExtracted from Fig. S1(a) of Ref. [8].

Above E_1 is the E_2 CP, whose energy resonance position obtained from the fitting of Eq. (2) to monolayer MoS₂ yields a value of 3.98 eV [see Figs. 4(a) and 4(b)]. Critical points at this energy range or above were seldom reported experimentally [24], and the origin of the E_2 edge is not well identified. We speculate it is associated with a combination of several critical points next to each other similarly observed for common semiconductors [14].

Above E_2 critical points, a series of broad and low oscillations are observed in the ϵ_2 spectra. These oscillations, with an energy separation of about 0.7 eV, can be seen more clearly by the corresponding absorption coefficient, as shown in the inset of Fig. 3(c). These oscillations are actual responses from the MoS₂ film, not artifacts from the silica substrate or the possible interferences with the backside of the substrate (see Fig. S2 [38]). They are the results of optical transitions from valence bands to higher conduction bands and are difficult to identify with other higher CPs in its complex band structure [49]. To the best of our knowledge, these high energy fine structures were neither discussed in theory nor reported in experiments before. Therefore, further efforts are needed to elucidate these oscillations, as well as the origin of the E_2 critical point.

For direct gap semiconductors, ϵ_2 can be expressed as $\epsilon_2 = C(\frac{E}{E_g})^{-2}(\frac{E}{E_g} - 1)^{1/2}$ near the direct energy gap (E_g), where C is a constant related to the transition probability from the valence band to conduction band and E is the photon energy

[14]. The expression can be rewritten as $\epsilon_2^2 E^4 \propto E - E_g$, and therefore the plot of $\epsilon_2^2 E^4$ as a function of photon energy E yields the optical band gap value E_g highly precisely. This is illustrated by the red curve in Fig. 5, from which the optical band gap of the monolayer MoS₂ is extracted to be 1.83 eV. To obtain the band gap or absorption edge of the bulk MoS₂ which is an indirect gap semiconductor, the SE transmission measurement was performed instead of the ellipsometry reflectivity measurement because the surface of our bulk sample is not quite specular on a large surface scale. The ellipsometric parameter, $\tan \psi$ shown in Fig. 5, is proportional to the reflectivity [41] of the light incidence on bulk MoS₂, yielding a band gap of 1.42 eV, which is within the range of values reported in literature as an indirect band gap [8,50].

IV. CONCLUSIONS

In summary, the optical properties of CVD-grown large-area monolayer MoS₂ are presented in this paper. Raman and PL spectroscopy studies proved our MoS₂ samples are monolayer with high crystalline quality and uniformity. Strong PL peaks are observed due to the well-known A and B excitonic transition. The broadband optical dielectric function, from 1 to 9 eV, is measured by the extended spectral range SE. In addition to the E_0 and $E_0 + \Delta_0$ CPs corresponding to A and B PL peaks, a series of higher energy sharp peaks, E_1 , $E_1 + \Delta_1$, and E_2 , are observed in the dielectric function of MoS₂. The energies of these CPs were precisely obtained by a line shape fitting method, and the results agree with theoretical calculation very well. Also, a series of absorption peaks separated by ~ 0.7 eV are clearly detected at higher UV photon energies. We believe that the optical features reported here will stimulate further investigations, especially to shed light on their origins from the band structure calculation perspectives, and in addition, will provide better understanding of the optical responses of MoS₂ to their optoelectronic device designs and applications.

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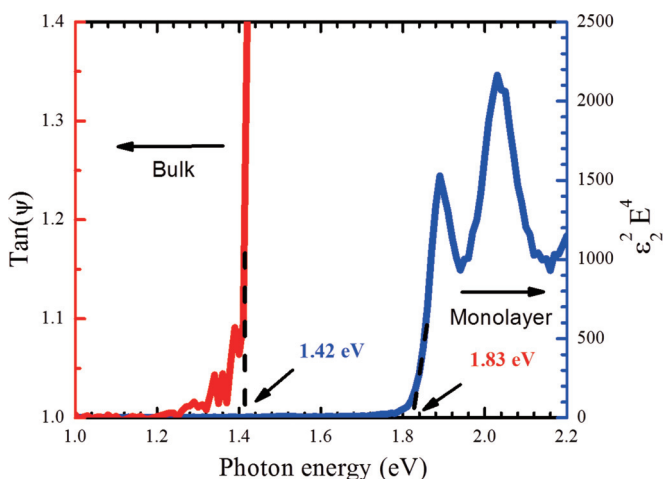


FIG. 5. (Color online) Determination and comparison of the optical band gap of monolayer (blue curve) and bulk (red curve) MoS₂.

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