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Tunable localized surface plasmon-enabled broadband light harvesting enhancement for high-efficiency panchromatic dye-sensitized solar cells

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Abstract

In photovoltaic devices, light harvesting (LH) and carrier collection have opposite relations with the thickness of the photoactive layer, which imposes a fundamental compromise for the power conversion efficiency (PCE). Unbalanced LH at different wavelengths further reduces the achievable PCE. Here, we report a novel approach to broadband balanced LH and panchromatic solar energy conversion using multiple-core-shell structured oxide-metal-oxide plasmonic nanoparticles. These nanoparticles feature tunable localized surface plasmon resonance frequencies and the required thermal stability during device fabrication. By simply blending the plasmonic nanoparticles with available photoactive materials, the broadband LH of practical photovoltaic devices can be significantly enhanced. We demonstrate a panchromatic dye-sensitized solar cell with an increased PCE from 8.3% to 10.8%, mainly through plasmon-enhanced photo-absorption in the otherwise less harvested region of solar spectrum. This general and simple strategy also highlights easy fabrication, and may benefit solar cells using other photo-absorbers or other types of solar-harvesting devices.

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Author contributions

X.D., J.Q., P.T.H., and A.M.B. conceived the idea and designed the experiments. X.D. and J.Q. performed the synthesis. X.D., J.Q., and D.S.Y. performed the TEM imaging. M.T.K. and N.X.F. performed the simulation. X.D., M.T.K., and N.X.F. analyzed the simulation results. X.D., J.Q., and P.-Y.C. performed the fabrication and characterization of the solar cells. X.D., J.Q., P.T.H., and A.M.B. co-wrote the paper and all authors discussed the results and commented on the manuscript.

Supporting Information. Supplementary discussion, supplementary materials and methods, supplementary figures, supplementary tables, and supplementary references. This material is available free of charge via the Internet at <http://pubs.acs.org>.

Keywords

Tunable localized surface plasmon; light harvesting enhancement; panchromatic dye-sensitized solar cells; multiple-core-shell nanoparticles

Developing efficient and inexpensive solar cells is of paramount importance for preserving non-renewable energy and lowering carbon dioxide emissions. The power conversion efficiency (PCE) of the solar cells is generally governed by the tradeoff between light harvesting (LH) and carrier collection. To break this compromise, strategies of improving either LH or carrier collection while maintaining the other have been successful^{1, 2}. However, most photo-absorbing materials possess unbalanced LH in high- and low-absorption wavelength regions (λ_{Hi} and λ_{Lo} , figure 1a, b), which results in another tradeoff. At λ_{Lo} , long absorption lengths are required, which reduce carrier collection; at λ_{Hi} , solar energy is fully-absorbed by thin photoactive layers, which inefficiently exploit energy at λ_{Lo} (usually the red to near-infrared (NIR) region, figure 1c). Consequently, the ability to achieve broadband balanced LH is desirable for high-efficiency solar cells.

Achieving such broadband balanced LH would lead to panchromatic solar energy conversion, an ideal strategy for maximizing overall LH. As a promising solution-processed photovoltaic technology, dye-sensitized solar cells (DSSCs) have demonstrated PCE exceeding 12%³⁻⁸; to further improve the PCE, several approaches toward panchromatic DSSCs⁹ have been attempted, including the development of panchromatic dyes¹⁰, co-adsorbing dyes¹¹, and energy transfer systems¹². However, these methods usually require extensive synthesis and optimization for various parameters (*e.g.*, spectral overlapping and spatial arrangement) and are limited to small ranges of materials. Therefore, a general and simple approach for broadband balanced LH and panchromatic DSSCs is required.

Localized surface plasmons (LSPs) are the elementary excitation states in noble metal nanoparticles (NPs), and can improve LH of photo-absorbers^{2, 13-18}. Here, we introduce a general and simple strategy for broadband LH enhancement and panchromatic DSSCs by matching LSP resonance (LSPR) wavelength (λ_{LSPR}) of plasmonic NPs with λ_{Lo} of existing photo-absorbers. Demonstrated by both simulations and experiments, matching λ_{LSPR} with λ_{Hi} or λ_{Lo} affects LH differently (figure 1d). Although matching λ_{LSPR} with λ_{Hi} readily improves LH and PCE for optically-thin photo-absorbing layers, matching λ_{LSPR} with λ_{Lo} maximally increases LH and PCE for practical photovoltaic devices with optically-thick layers, through enhancing photo-absorption in the weakly-absorbing region. The multiple-core-shell oxide-metal-oxide plasmonic NPs developed and utilized here are advantageous over other geometries, by featuring: adjustable λ_{LSPR} of 600–1,000 nm, substantially enhanced near-field electromagnetic (EM) intensity, and preservable plasmonic properties during device fabrication. By matching λ_{LSPR} with λ_{Lo} of a common dye (N719), a panchromatic DSSC with broadband balanced LH and a PCE of 10.8% is achieved (30% increase comparing to DSSCs without plasmonic NPs).

Synthesis and optical characterization are performed for the core-shell ($d_{\text{core}} \sim 15$ nm, $d_{\text{T}} \sim 2$ nm) Ag@TiO₂ (AgT) and Au@TiO₂ (AuT) and multiple-core-shell ($d_{\text{core}} \sim 15$ nm, $d_{\text{Au}} \sim 0-4$ nm, and $d_{\text{T}} \sim 2$ nm) TiO₂-Au-TiO₂ (TAuT) NPs (figure 2a, S3–S6). d_{core} , d_{Au} , and d_{T}

represent core diameter, gold-shell thickness, and TiO₂-shell thickness, respectively. When synthesizing the gold-shell of T AuT NPs, Au seeds gradually cover the core and coalesce to form a continuous layer, which is stable at 500°C (annealing condition for DSSCs). The outer TiO₂-shells impede the metals from promoting recombination in the DSSCs^{2, 14}. The λ_{LSPR} of AgT and AuT NPs are 420 and 550 nm respectively, and the λ_{LSPR} of T AuT NPs is continuously tunable from 1,000 to 600 nm by increasing d_{Au} to 4 nm (figure 2b). Tunable λ_{LSPR} across visible-NIR is thus achieved.

The interaction of the NPs with light was investigated using the finite-difference time-domain (FDTD) method. All plasmonic NPs strongly amplify the near-field EM intensity (figure 2c). Therefore, surrounding dye-molecules would experience a significantly increased light intensity near the LSPR frequency^{16, 19} and a higher photon flux (ϕ_{ph}), which increases electron-hole pair generation. The enhancement factors (derived by integrating over near-field space, figure S7) of plasmonic NPs as functions of wavelength (figure 2d) show similar results of λ_{LSPR} to those obtained from absorption spectra. The characteristic absorption peak of the T AuT NPs originates from the LSPs arising in the gold-shell structure^{20–23}. By tuning d_{Au} and d_{T} of T AuT NPs in 2–6 nm, λ_{LSPR} of 600–800 nm is achieved; as expected, thinner gold- and thicker TiO₂-shells result in longer λ_{LSPR} (figure 2e, S8). Furthermore, an optimum $d_{\text{Au}} \sim 3$ nm maximizes the spectrum-integrated enhancement factor and thicker TiO₂-shells tend to weaken the enhancement (figure 2f, the enhancement factor at λ_{LSPR} is shown in figure S9). The λ_{LSPR} and enhancement factors are also obtained for AgT and AuT NPs, which extend the LSP-enhanced spectral range to vis-NIR. These results represent the intrinsic near-field enhancement of plasmonic NPs and do not take the spectral response of dye-molecules into account.

The impact on the photo-absorption of surrounding dye-molecules is investigated by comparing the enhancement factors at different wavelengths (figure 2g, S10). At $\lambda_{\text{Lo}}=600\text{--}800$ nm (for N719), the greatest enhancement occurs by using T AuT NPs (*e.g.*, $d_{\text{Au}} \sim 3$ nm and $d_{\text{T}} \sim 2$ nm for 700 nm), where λ_{LSPR} matches λ_{Lo} ; at $\lambda_{\text{Hi}}=400\text{--}600$ nm, the maximum enhancement occurs by using AuT and AgT NPs, where λ_{LSPR} matches λ_{Hi} . The overall dye absorption is also maximally improved by AgT and AuT NPs (bottom part of figure 2g). These results and the simulated LSP-enhanced dye absorption ($\lambda_{\text{LSPR}}=400\text{--}700$ nm, figure S11) suggest that, on the level of individual plasmonic NP, matching λ_{LSPR} with λ_{Hi} enhances overall LH of dye-molecules more efficiently, whereas matching λ_{LSPR} with λ_{Lo} balances LH in a broad spectral range.

Moreover, the LSP-enhanced LH of N719-sensitized mesoporous TiO₂ films (figure 2h) agrees with the simulation results. AgT and AuT NPs improve LH mostly at λ_{Hi} , whereas T AuT NPs (T AuT-700 NPs with $\lambda_{\text{LSPR}}=700$ nm are used for all thin film LH and device performance characterizations) increase LH at λ_{Lo} , resulting in balanced spectra. The similar enhancements of simulated EM intensity and LH of thin films around λ_{LSPR} (supplementary discussion) confirm that the LH improvement stems from the interaction between dye-molecular dipoles and the LSP-enhanced near field and scattering cross-section of the NPs.

The multiple-core-shell oxide-metal-oxide plasmonic NPs are advantageous over other geometries (*e.g.* nanorods, nanodisks, and hollow nanoshells) which also possess tunable

λ_{LSPR} in vis-NIR^{20, 24}. Under thermal treatment, the T AuT NPs maintain geometric and plasmonic properties with the rationally-designed templated metal-shell structure, while other geometries melt to spheres or spheroids and the characteristic λ_{LSPR} blue-shifts (figure 2i, S12, S13), severely reducing the ability to enhance LH at λ_{Lo} . In addition, little change in synthesis is required to tune λ_{LSPR} of T AuT NPs due to their high sensitivity to geometry (supplementary discussion). Moreover, metal-shell structures (e.g. T AuT, hollow gold-shell (figure S15, S16), and T AgT (figure S17, S18)) possess larger EM field enhancement than spherical solid metal structure (e.g. AuT and AgT). Therefore, the multiple-core-shell plasmonic NPs can benefit many LSP-enhanced applications which require tunable spectral response and high temperature fabrications. In addition, unlike the method using the inter-particle coupling²⁵ to tune λ_{LSPR} , the current approach does not require sophisticated chemistry to control particle aggregation and the enhanced EM field uniformly surrounds the plasmonic NPs.

To study the LH of tunable LSP-enhanced DSSCs, different plasmonic NPs-incorporated TiO₂ photoanodes (thickness of 1–20 μm and plasmonic NPs-TiO₂ ratio of 0.01–3.2 wt%) are assembled into DSSCs^{1, 2}. In 1.5- μm -thick optically-thin photoanodes, AgT, AuT, and T AuT NPs increase the incident photon-to-current conversion efficiency (IPCE) at λ_{max} (maximum absorption wavelength~530 nm) by 45%, 60%, and 50%, respectively (figure 3a-left). Enhancement is maximized when λ_{LSPR} is closest to λ_{max} . Similarly, at 700 nm, T AuT NPs increase IPCE the most by 80%, whereas AgT and AuT NPs enhance IPCE only by 21% and 22% due to large mismatch between λ_{LSPR} and λ_{Lo} . By correlating the absorbance, LH efficiency (LHE), and IPCE (supplementary discussion), we find that the experimental results of absorbance and IPCE enhancement are similar, which indicate that the IPCE improvement mainly arises from LSP-enhanced LH.

While IPCE (specifically LHE) of thin photoanodes is readily increased by LSPs, optically-thick photoanodes (10~15 μm) for practical devices are required to ensure balance between LH and carrier collection for maximized PCE²⁶, where the LHE at λ_{Hi} and internal quantum efficiency are close to unity. At λ_{max} , less than 5% enhancement of IPCE for all plasmonic NPs is observed (figure 3a-middle, supplementary discussion). In contrast, at λ_{Lo} , the T AuT NPs increase IPCE by up to 100%, whereas AgT and AuT NPs increase IPCE slightly. Additionally, the IPCE increases monotonically with the concentration of T AuT NPs (figure 3a-right).

Therefore, both experiments and simulations have demonstrated that matching λ_{LSPR} with λ_{Hi} or λ_{Lo} impacts the spectral response of LH differently. Matching λ_{LSPR} with λ_{Hi} enhances LH in already strongly-absorbing region, which benefits optically-thin photoanodes. In contrast, matching λ_{LSPR} with λ_{Lo} improves the weakly-absorbed part of solar spectrum, which is of great importance for broadband balanced LH in practical panchromatic solar cells.

The PCEs of tunable LSP-enhanced DSSCs are measured (figure S19, table S1) ($\text{PCE} = J_{\text{SC}} V_{\text{OC}} \text{FF} / P_{\text{in}}$, J_{SC} , V_{OC} , FF, and P_{in} represent short-circuit current density, open-circuit voltage, fill factor, and incident power density, separately). $J_{\text{SC}} (=q \int \text{IPCE}(\lambda) \varphi_{\text{ph}}(\lambda) d\lambda)$ is improved with increasing concentrations of plasmonic NPs (within the

concentration range studied); the largest J_{SC} is achieved by incorporating the highest concentration of TAU T NPs (figure 3b). Besides the improved LH, thinner photoanodes for optimized PCEs (the optimized thicknesses for TiO₂, AgT, AuT, and TAU T DSSCs are 13.2, 10.9, 10.1, and 11.0 μm) are also responsible for the improved IPCE and J_{SC} , by improving the carrier collection (electron diffusion lengths are not changed significantly, figure S22).

Additionally, V_{OC} is increased by introducing plasmonic NPs (figure 3c). Generally, V_{OC} is limited by the material properties of electronic structure (*e.g.*, the quasi-Fermi level of TiO₂ and redox potential of electrolyte); enhancing V_{OC} usually requires exploiting new materials³. Two reasons could be responsible for the LSP-induced V_{OC} enhancement: thinner optimized photoanodes reduce the voltage loss from charge recombination; and the quasi-Fermi level is lifted due to the equilibrium between quasi-Fermi level of TiO₂ and LSP energy level of plasmonic NPs^{15, 27}. Our result is also in agreement with a recent study on plasmoelectronics²⁸. In addition, we observe that TAU T NPs increase V_{OC} less significantly than AgT and AuT NPs, which is likely due to lower LSP energy level (supplementary discussion). The different V_{OC} enhancements could assist in the elucidation of the origin of LSP-enhanced V_{OC} , and the ability of rationally increasing V_{OC} could further improve PCE. In addition, although we suggest that the V_{OC} increase is due to quasi-Fermi level equilibrium, we do not yet have direct evidence of charge transfer from metal to TiO₂. Regarding this subject, we plan to perform further investigation.

Since LSPs have a larger effect on enhancing LH than changing V_{OC} and FF (no significant impact on FF is observed, figure S23), the PCEs are improved (figure 3d) with increasing concentrations of NPs (0–3.2 wt%, higher concentrations could reduce performance, supplementary discussion). Thus, maximum PCEs of 10.1%, 10.3%, and 10.8% are achieved for AgT, AuT, and TAU T NPs-incorporated DSSCs respectively, comparing to 8.3% without plasmonic NPs.

The optimized device performance helps distinguish the improvement from different plasmonic NPs. AgT and AuT NPs used here and similar materials in previous studies on LSP-enhanced DSSCs^{2, 14–16} improve IPCE and PCE, mainly through improved LH at λ_{Hi} . Although the previous approach is responsible for achieving ultrathin solar cells²⁹, it could not break the compromise imposed by unbalanced LH at λ_{Hi} and λ_{Lo} . In contrast, our general and simple strategy of matching λ_{LSPR} with λ_{Lo} balances and optimizes broadband LH, achieves panchromatic DSSCs with the existing dye, and further enhances LH, J_{SC} , and PCE. In fact, the distribution of core sizes and shell thicknesses result in the broad absorption spectrum of TAU T NPs with $\lambda_{LSPR} \sim \lambda_{Lo}$ (figure 2b), which benefits panchromatic LH and carrier collection simultaneously. Our approach toward panchromatic DSSCs can apply to other photo-absorbers and other types of solar cells. For instance, a PCE of 12% was achieved by using a porphyrin dye in the DSSC³. We believe that the same strategy can be used to improve the overall LH by matching the λ_{LSPR} with the λ_{Lo} and a similar degree of improvement would be observed. Furthermore, since LH is the initial stage of a series of physical, chemical, or biochemical processes in many solar energy conversion devices, our strategy of matching λ_{LSPR} with λ_{Lo} may benefit many other devices³⁰ including artificial photosynthesis¹⁸, solar heating, solar thermal electricity, and solar fuels³¹.

Supplementary Material

Refer to Web version on PubMed Central for supplementary material.

Acknowledgments

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References

1. Dang X, Yi H, Ham MH, Qi J, Yun DS, Ladewski R, Strano MS, Hammond PT, Belcher AM. *Nature Nanotech.* 2011; 6(6):377–384.
2. Qi J, Dang X, Hammond PT, Belcher AM. *ACS Nano.* 2011; 5(9):7108–7116. [PubMed: 21815674]
3. Yella A, Lee HW, Tsao HN, Yi C, Chandiran AK, Nazeeruddin MK, Diao EWG, Yeh CY, Zakeeruddin SM, Grätzel M. *Science.* 2011; 334(6056):629–634. [PubMed: 22053043]
4. Hardin BE, Snaith HJ, McGehee MD. *Nature Photon.* 2012; 6(3):162–169.
5. Ito S, Zakeeruddin SM, Comte P, Liska P, Kuang D, Grätzel M. *Nature Photon.* 2008; 2(11):693–698.
6. Hagfeldt A, Boschloo G, Sun L, Kloo L, Pettersson H. *Chem Rev.* 2010; 110(11):6595–6663. [PubMed: 20831177]
7. Chung I, Lee B, He J, Chang RPH, Kanatzidis MG. *Nature.* 2012; 485(7399):486–489. [PubMed: 22622574]
8. Kim HS, Lee CR, Im JH, Lee KB, Moehl T, Marchioro A, Moon SJ, Humphry-Baker R, Yum JH, Moser JE, Grätzel M, Park NG. *Sci Rep.* 2012; 2:591. [PubMed: 22912919]
9. Yum JH, Baranoff E, Wenger S, Nazeeruddin MK, Grätzel M. *Energy Environ Sci.* 2011; 4(3):842–857.
10. Nazeeruddin MK, Péchy P, Renouard T, Zakeeruddin SM, Humphry-Baker R, Comte P, Liska P, Cevey L, Costa E, Shklover V, Spiccia L, Deacon GB, Bignozzi CA, Grätzel M. *J Am Chem Soc.* 2001; 123(8):1613–1624. [PubMed: 11456760]
11. Cid JJ, Yum JH, Jang SR, Nazeeruddin MK, Martínez-Ferrero E, Palomares E, Ko J, Grätzel M, Torres T. *Angew Chem Int Ed.* 2007; 46(44):8358–8362.
12. Hardin BE, Hoke ET, Armstrong PB, Yum JH, Comte P, Torres T, Frechet JMJ, Nazeeruddin MK, Grätzel M, McGehee MD. *Nature Photon.* 2009; 3(7):406–411.
13. Atwater HA, Polman A. *Nature Mater.* 2010; 9(3):205–213. [PubMed: 20168344]
14. Brown MD, Suteewong T, Kumar RSS, D’Innocenzo V, Petrozza A, Lee MM, Wiesner U, Snaith HJ. *Nano Lett.* 2010; 11(2):438–445. [PubMed: 21194204]
15. Choi H, Chen WT, Kamat PV. *ACS Nano.* 2012; 6(5):4418–4427. [PubMed: 22494109]
16. Chen H, Blaber MG, Standridge SD, DeMarco EJ, Hupp JT, Ratner MA, Schatz GC. *J Phys Chem C.* 2012; 116(18):10215–10221.
17. Aydin K, Ferry VE, Briggs RM, Atwater HA. *Nat Commun.* 2011; 2:517. [PubMed: 22044996]
18. Gao H, Liu C, Jeong HE, Yang P. *ACS Nano.* 2011; 6(1):234–240. [PubMed: 22147636]
19. Mahmoud MA, Snyder B, El-Sayed MA. *J Phys Chem C.* 2010; 114(16):7436–7443.
20. Hu M, Chen J, Li ZY, Au L, Hartland GV, Li X, Marquez M, Xia Y. *Chem Soc Rev.* 2006; 35(11):1084–1094. [PubMed: 17057837]
21. Kodali AK, Llorca X, Bhargava R. *Proc Natl Acad Sci USA.* 2010; 107(31):13620–13625. [PubMed: 20634428]
22. Milton GW, Nicorovici NAP, McPhedran RC, Podolskiy VA. *Proc R Soc A.* 2005; 461(2064):3999–4034.
23. Prodan E, Radloff C, Halas NJ, Nordlander P. *Science.* 2003; 302(5644):419–422. [PubMed: 14564001]

24. Skrabalak SE, Chen J, Sun Y, Lu X, Au L, Cogley CM, Xia Y. *Acc Chem Res.* 2008; 41(12): 1587–1595. [PubMed: 18570442]
25. Shanthil M, Thomas R, Swathi RS, George Thomas K. *J Phys Chem Lett.* 2012; 3(11):1459–1464.
26. Halme J, Vahermaa P, Miettunen K, Lund P. *Adv Mater.* 2010; 22(35):E210–E234. [PubMed: 20717984]
27. Takai A, Kamat PV. *ACS Nano.* 2011; 5(9):7369–7376. [PubMed: 21819038]
28. Warren SC, Walker DA, Grzybowski BA. *Langmuir.* 2012; 28(24):9093–9102. [PubMed: 22385329]
29. Hägglund C, Apell SP. *J Phys Chem Lett.* 2012; 3(10):1275–1285.
30. Linc S, Christopher P, Ingram DB. *Nature Mater.* 2011; 10(12):911–921. [PubMed: 22109608]
31. Thomann I, Pinaud BA, Chen Z, Clemens BM, Jaramillo TF, Brongersma ML. *Nano Lett.* 2011; 11(8):3440–3446. [PubMed: 21749077]

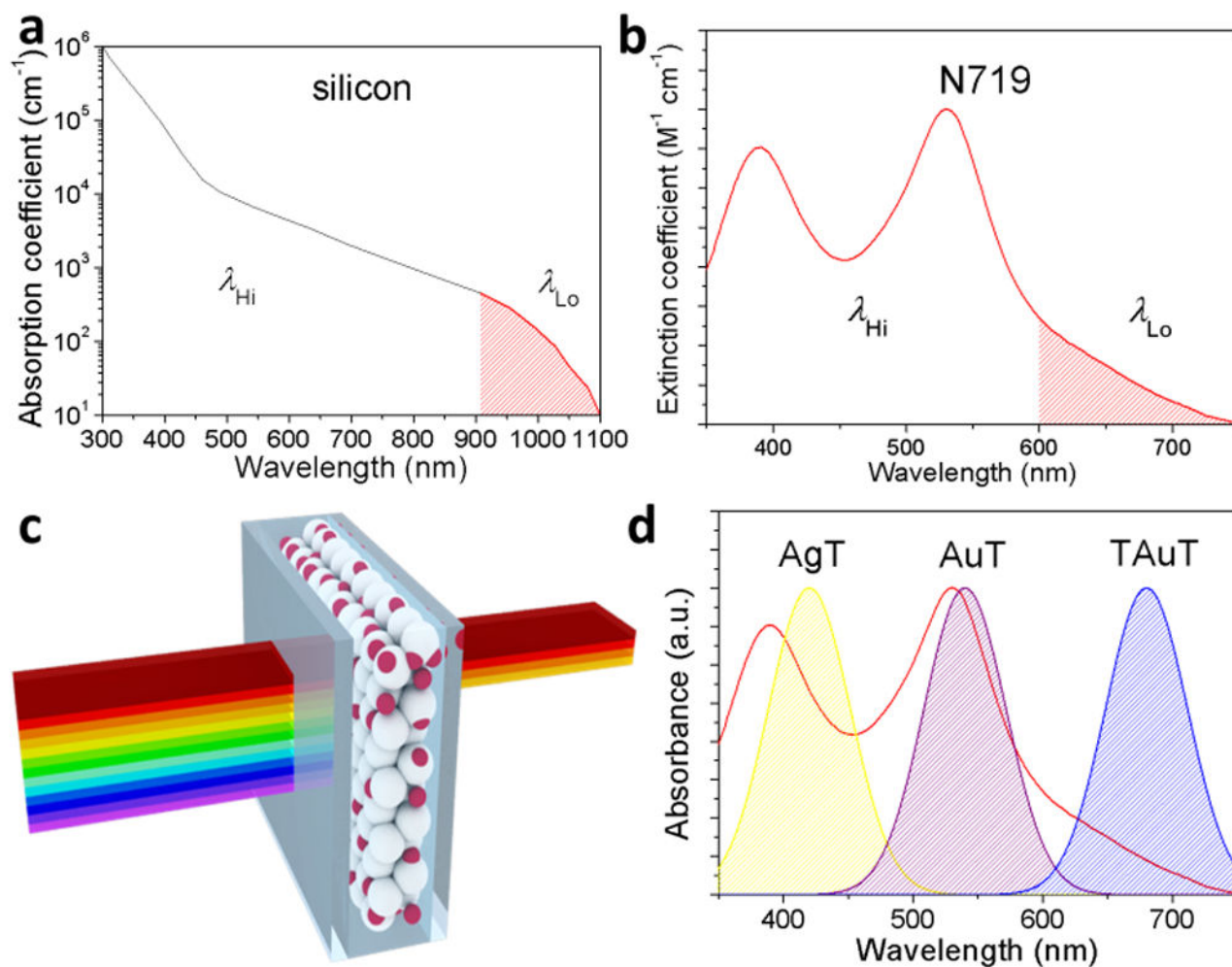


Figure 1. Spectral response of the LH of a solar cell and tunable LSP-enhanced broadband LH
a, b, the absorption coefficient of bulk silicon (**a**) and extinction coefficient of N719 (**b**); the shaded areas are low LH regions, which require thicker photoactive layers to achieve efficient LH. **c,** a schematic of the spectral response of a solar cell. The solar energy is less utilized at λ_{Lo} (usually red-NIR). **d,** illustrations of enhancing EM intensity by AgT (yellow), AuT (purple), and TAUt (blue) plasmonic NPs. λ_{LSPR} of AgT and AuT overlaps with λ_{Hi} of N719, maximizing the effect of LSP-enhanced LH. λ_{LSPR} of TAUt matches λ_{Lo} of N719, balancing LH at different wavelengths.

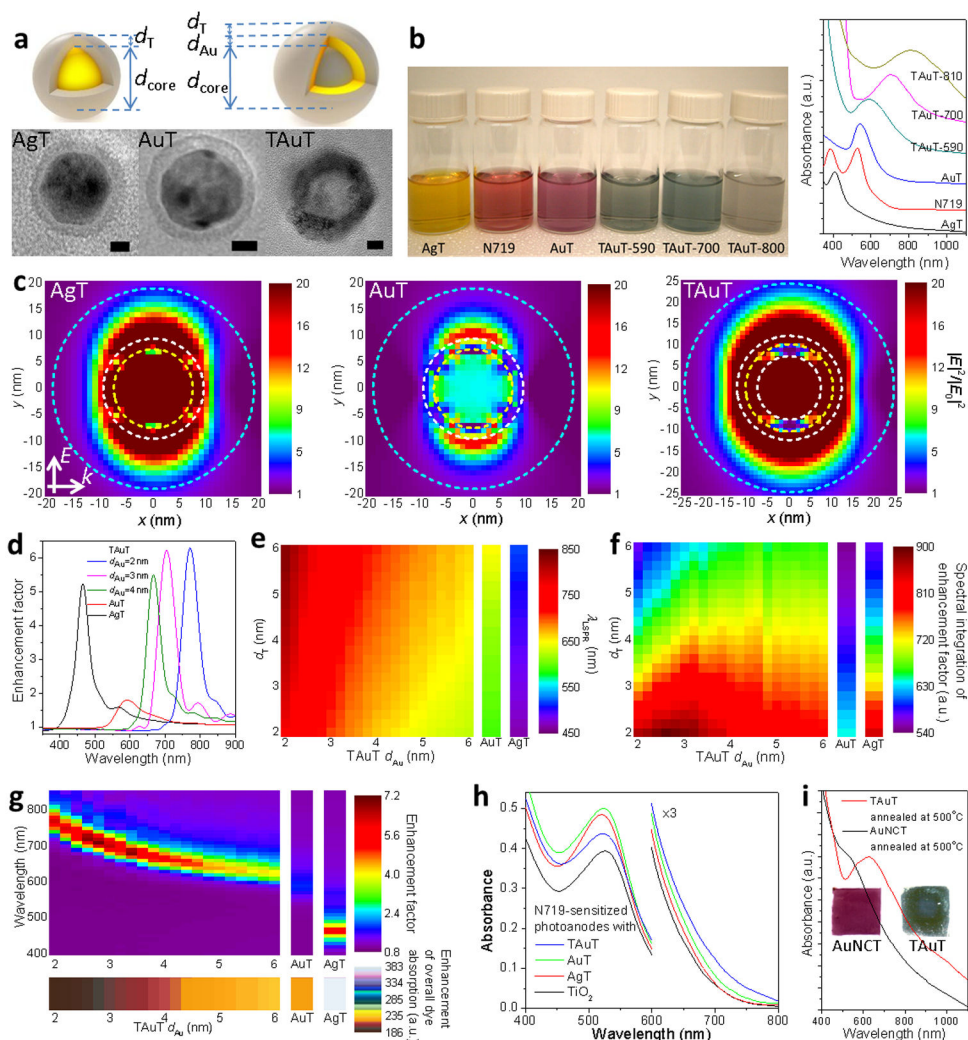


Figure 2. Synthesis, optical characterization, and FDTD simulation of AgT, AuT, and TAUt plasmonic NPs

a, illustrations and transmission electron microscope (TEM) images of AgT, AuT, and TAUt NPs. The scale bars are 5 nm. **b**, photograph picture and absorption spectra of AgT, N719, AuT, TAUt-590 ($\lambda_{\text{LSPR}}=590$ nm), TAUt-700, and TAUt-810 in ethanol solutions. **c**, simulated EM intensity enhancement ($|E|^2/|E_0|^2$) in near field at λ_{LSPR} for AgT, AuT, and TAUt NPs. The inner circles (white and yellow) represent different layers of NPs, and the outermost circles (cyan) represent the volume of integration. $d_{\text{core}}=15$ nm (used for all simulations unless specified), $d_{\text{Au}}=3$ nm, and $d_{\text{T}}=2$ nm. **d**, enhancement factor as a function of wavelength for AgT, AuT, and TAUt NPs with different d_{Au} ($d_{\text{T}}=2$ nm). **e**, **f**, simulated λ_{LSPR} (**e**) and spectrum-integrated enhancement factor over 300–900 nm (**f**) for AgT, AuT, and TAUt NPs as a function of d_{Au} (of TAUt) and d_{T} (of AgT, AuT, and TAUt). **g**, enhancement factor at different wavelengths (top) and enhancement of overall dye absorption (bottom) for AgT, AuT, and TAUt with different d_{Au} ($d_{\text{T}}=2$ nm). **h**, absorption of N719 in sensitized 3- μm -thick TiO_2 films improved by AgT (1.0 wt%), AuT (1.8 wt%), and TAUt (3.2 wt%) NPs. **i**, absorption spectra and photograph images of thin films of Au

nanocage@TiO₂ (AuNCT, left) and T AuT-700 (right) on glass substrates after 500°C annealing; T AuT NPs maintain optical properties, whereas λ_{LSPR} of AuNCT blue-shifts toward λ_{LSPR} of AuT NPs.

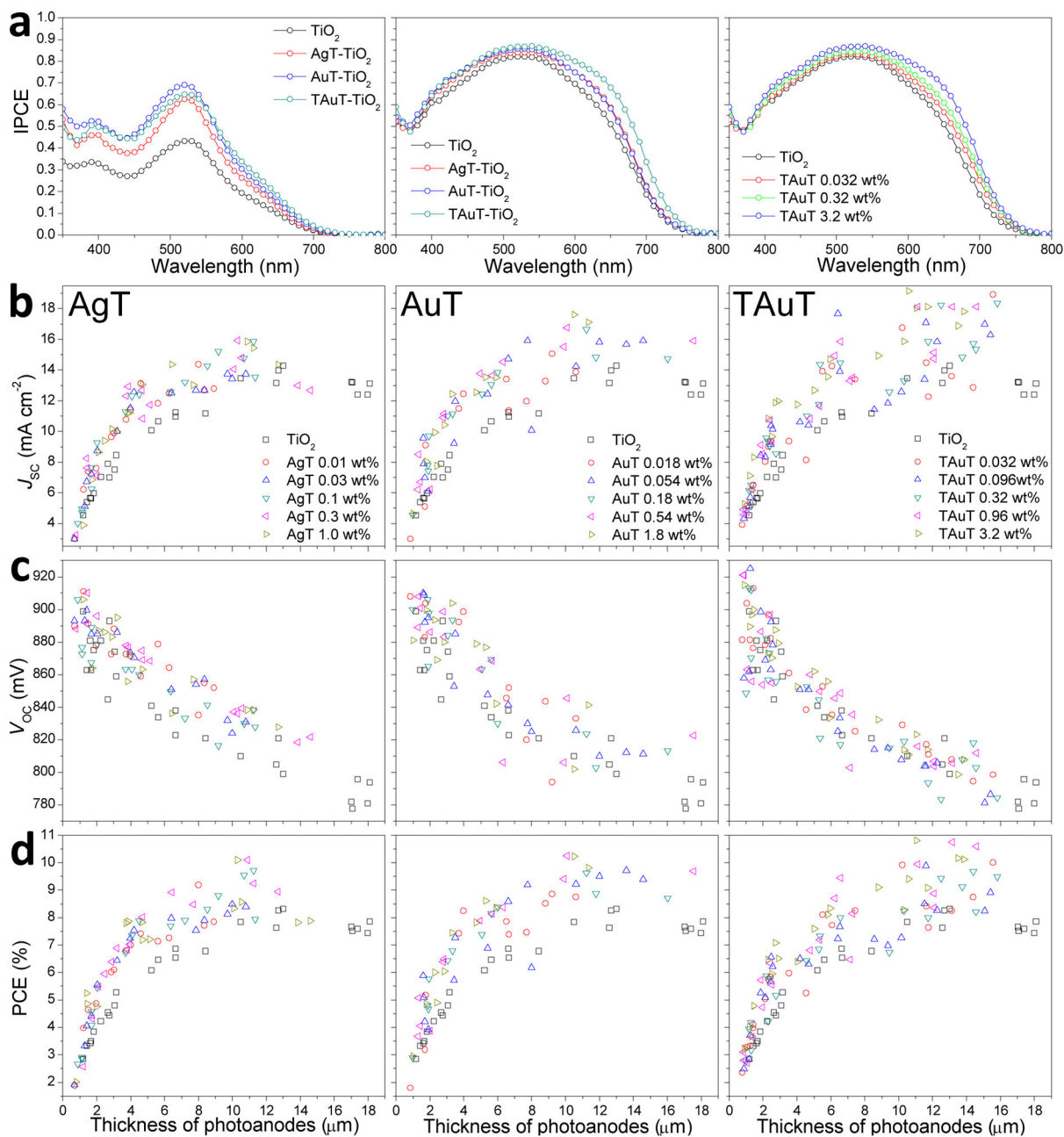


Figure 3. Tunable LSP-enhanced LH and photovoltaic performance of DSSCs

a, IPCE spectra of DSSCs with AgT, AuT, and TAUt NPs-incorporated photoanodes of 1.5 μm thickness (left), of optimized thickness for maximum PCE (middle), and as a function of concentration of TAUt NPs in optimized TAUt-DSSCs (right). **b**, **c**, **d**, J_{sc} (**b**), V_{oc} (**c**), and PCE (**d**) of LSP-enhanced DSSCs with AgT (left), AuT (middle), and TAUt (right) NPs-incorporated photoanodes, as a function of the concentration of plasmonic NPs (0–3.2 wt%) and thickness of photoanodes (1–20 μm). The particle densities for all plasmonic NPs are

the same and in the range of 8.4×10^{12} – $8.4 \times 10^{14} \text{ cm}^{-3}$, and the difference in weight percent is due to the different geometries of NPs (supplementary discussion).