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2 TITLE

3 **Optical properties of fluid hydrogen at the transition to a conducting state**

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23 ABSTRACT

24 We use fast transient transmission and emission spectroscopies in the pulse laser heated diamond
25 anvil cell to probe the energy-dependent optical properties of hydrogen at pressures of 10-150
26 GPa and temperatures up to 6000 K. Hydrogen is absorptive at visible to near-infrared
27 wavelengths above a threshold temperature that decreases from 3000 K at 18 GPa to 1700 K at
28 110 GPa. Transmission spectra at 2400 K and 141 GPa indicate that the absorptive hydrogen is
29 semiconducting or semi-metallic in character, definitively ruling out a first-order insulator-metal
30 transition in the studied pressure range.

31

32 TEXT

33 Realizing metallic hydrogen and understanding its properties is fundamental for
34 achieving predicted high temperature superconductivity [1], exploring the regime of inertial
35 confinement fusion [2], and resolving the structure and dynamics of giant planetary interiors [3-
36 7]. The metallic state has not been reached yet in the solid at pressures as high as 360 GPa [8-
37 10], but experiments [3,11-16] and theoretical calculations [5,16-27] probing the fluid state at
38 high temperature document an insulator-metal transition (IMT). This fluid metallic state has been
39 theorized to be even the ground state at sufficiently high pressures [19,20], however recent
40 experiments suggest more complex behavior [16,28].

41 While the underlying physics of metallization in hydrogen is thought to be related to a
42 Mott-like mechanism (band overlap), the essential parts of this phenomenon remain uncaptured
43 because of difficulties in finding appropriate theoretical approximation methods [25-27] and
44 experimental challenges. With increasing pressure, the fluid IMT is expected to exhibit a *critical*
45 *point* where it transitions from being continuous to discontinuous (first-order), and merge with

46 the melting line in the limit of high densities [19,20]. Different theoretical studies agree about the
47 transition character, but the location of the critical point varies substantially, with modern
48 estimates ranging as low as 90 GPa [4,5,19,21,22,25-27].

49 Experiments on fluid hydrogen using shock compression measured gradual increases in
50 electrical conductivity and optical reflectivity to constant, metallic values with increasing
51 temperature and pressure up to 90 GPa [11,13,14], evincing a continuous IMT below this
52 pressure. Between 90 and 140 GPa shock experiments were conducted without direct
53 temperature measurements, leaving the gradual increase and saturation of conductivity detected
54 in this region [3,12,29] open to interpretation: the data are consistent with a continuous IMT
55 [3,12,29] but also show characteristics of a first order IMT naturally broadened by adiabatic
56 compression (e.g. Ref. [30]). Recent isentropic compression measurements suggest the IMT
57 becomes first order by 285 GPa [16], but also assumed temperature, leaving a broad pressure
58 range [3,12,16,29] where the nature of the IMT remains poorly characterized. Static
59 compression, diamond anvil cell (DAC) experiments showed that direct temperature
60 measurements are possible in the metallization regime at high pressure, and detected a fluid
61 phase transition at ~ 120 GPa, though were not able to provide any characteristics of the
62 transformed state [15].

63 Hydrogen is a highly reactive and diffusive material, so is challenging to contain in high
64 temperature and pressure experiments for long periods [28,31]. Dynamic compression has
65 probed hydrogen beyond several thousand K at high pressures on microsecond or faster
66 timescales [3,11-14,29], whereas DAC experiments limited to longer timescales reached 1000
67 and 1800 K using resistive [28] and laser heating [15,31,32], respectively.

68 In this Letter we describe microsecond, single-pulse laser heating DAC experiments on
69 hydrogen that reach novel conditions not previously characterized by dynamic or static studies
70 (Fig. 1). Time-resolved optical emission and transmission spectroscopy determines sample
71 temperature T and corresponding optical absorptivity α during heat cycles [33,34]. A 4-10 μs
72 long laser pulse heats a metallic (Ir) foil in a hydrogen sample, and heat propagates across the
73 adjacent hydrogen creating a localized heated excited state of several μm in linear dimensions
74 and a few μs long. Transient absorption probing using a continuous laser (CW: 532 nm) and
75 pulsed broadband supercontinuum (BB: 1 MHz, 150 ps, 400-900 nm) was performed by
76 transmission through a hole in the foil at the heated region. Fits of emission spectra to a Planck
77 distribution determined temperature with a time resolution of 0.5-5 μs .

78 To ensure our measurements probed pristine hydrogen, several precautions were taken.
79 Pressure was measured before and after the heat cycles using Raman spectra of the hydrogen
80 vibron [35] and diamond edge [36], and ruby fluorescence [37]. Vibron signal from the heated
81 area was confirmed before and after heating [34]. Continued heating resulted in decreasing
82 vibron signal, pressure changes (usually but not always negative), decreasing foil hole diameters
83 [34], and occasional anvil fracturing, evincing rapid hydrogen diffusion and loss. Complete loss
84 occurred within ~ 1 ms of total heating time. Weak Raman lines attributed to Ir hydride [38]
85 appeared in one sample subjected to prolonged heating at high temperature [34], but not in
86 reported experiments.

87 Upon increasing laser power, time histories of thermal emission during heat cycles
88 exhibited a drastic shift in behavior, similar to that seen in noble gases as a consequence of high-
89 temperature absorption onset [33]. For low peak laser power, the temperature followed the laser
90 power history (Fig. 2a), having a distinct initial peak. With increasing power, there was a

91 transition to a different thermal response, where temperature did not follow laser power, but
92 instead rose and remained roughly constant, forming a plateau that persisted for an especially
93 long duration (Fig. 3). To examine this transition we performed finite element (FE) models
94 [33,34,39] to investigate how properties of hydrogen samples, such as a temperature-dependent
95 absorption, control temperature history. The lower-temperature behavior is expected for a
96 transparent sample, i.e. where the laser is absorbed entirely in the foil surface. The higher-
97 temperature behavior could not be explained if the sample remained transparent; instead an
98 abrupt increase in sample absorption with temperature (to $\alpha \approx 0.1$ to $1 \mu\text{m}^{-1}$) is needed to
99 reproduce the long temperature plateau, which occurs near the temperature of transition to the
100 absorptive state. In this regime, hydrogen is heated directly by bulk absorption of laser energy,
101 and this delocalization of heat energy compared to absorption at the foil surface limits the
102 achievable temperature, producing the plateau effect.

103 Transient absorption measurements (Fig. 2) confirm the change in thermal history is
104 correlated with increased optical absorption. Here, absorption coefficient $\alpha = -\ln(I_H/I_C)/d$,
105 where d is the thickness of the hot region (estimated from FE calculations, and of order $1 \mu\text{m}$ at
106 141 GPa), while I_C and I_H are transmitted probe intensities through cold and hot samples,
107 respectively. Peak α near $1 \mu\text{m}^{-1}$ are consistently inferred, with total uncertainty of about an
108 order of magnitude largely due to thickness uncertainty and reproducibility.

109 To compare our optical measurements in a wide, previously unexplored region of the
110 phase diagram to prior data, we interpolated direct-current (DC) conductivity (σ_0) measurements
111 on fluid hydrogen [3,11,12,29,40] using an experimentally-consistent model [34] having the
112 form $\sigma^* = \sigma_m^* - \sigma_j^* \{1 - 0.5 \operatorname{erfc}[(T^* - T_c^*)/T_w^*]\}$, where $\sigma^* = \log(\sigma_0)$ and $T^* = 1/T$. This
113 model has a sigmoidal temperature dependence that reproduces the Arrhenius- or semiconductor-

114 like proportionality of $\sigma^* \propto T^*$ during the IMT [11,12,29], with constant conductivity in purely
115 metallic (σ_m^*) [13,14,29] and insulating ($\sigma_m^* - \sigma_j^*$) [40] states; the transition temperature (T_c^*) and
116 width (T_w^*) were taken to vary linearly with density [34].

117 Absorption spectra at 141 GPa and 2,400 K show increasing absorption with photon
118 energy across the visible (Fig. 4a). Semiconductor-like absorption is one possible explanation:
119 electronic band gaps on the order of the present optical energies have been reported in dense
120 hydrogen [3,8,9,12,16,29,41,42]. The data do not permit the exact assignment to existing
121 semiconductor or semi-metal models. However, given the disordered nature of the material and
122 rather large values of the absorption coefficients (up to $\sim 10^6 \text{ m}^{-1}$), we suggest that observed
123 absorption is due to optical processes between extended states, which are well described by
124 Tauc's relation $\alpha = A(\hbar\omega - E_g)^2/\hbar\omega$. This well fits the data, implying a gap E_g of $0.9 \pm 0.3 \text{ eV}$.
125 In this semiconductor picture, hydrogen is electrically conductive due to thermal excitation of
126 electrons. Assuming an effective carrier mass of $0.5-1 m_e$ [13,33] the DC conductivity at these
127 conditions is predicted to be 5-23 S/cm for $E_g=0.9 \text{ eV}$, in agreement with that determined from
128 shock data ($\sim 15 \text{ S/cm}$) [43]. The spectral character is consistent with theory for semiconducting
129 hydrogen at similar pressure and lower temperature [16] which may be similarly described by the
130 Tauc model.

131 Conductivity at optical frequencies is $\sigma = n\alpha c \epsilon_0$, where n is the real index of refraction
132 [44,45] which is weakly dependent on material properties, and always of order 10^0 [34]. Thus, σ
133 is determined principally by α , which varies by many orders of magnitude during electronic
134 transformation. The conductivity at 2400 K and 141 GPa varies between ~ 70 and $\sim 220 \text{ S/cm}$
135 from 1.55 to 2.3 eV, and this extrapolated to zero energy is consistent with the DC conductivity

136 of ~ 15 S/cm (Fig. 4b). The decrease in conductivity with energy is inconsistent with the simple
137 Drude model of free carriers widely used for hydrogen at extreme conditions [2,11-14,16,17].

138 A modified Drude model, after Smith [46], given by $\sigma = \sigma_0 [1 + C (1 - \omega^2 \tau^2) / (1 +$
139 $\omega^2 \tau^2)] / [(1 + C)(1 + \omega^2 \tau^2)]$ and incorporating reduced electron mobility through a
140 backscattering term C , does provide an adequate representation of the data including the DC
141 limit (Fig. 4b). This model has features typically observed in poor metals at the boundary of
142 metallization transitions such as mercury [46] and argon [33], suggesting its applicability for
143 hydrogen at the IMT. The parameter C , a measure of how closely the spectrum follows the
144 Drude (free-electron) approximation, ranges from 0 to -1, with $C = 0$ (minimum backscattering)
145 corresponding to the Drude form. Fits to our data show C is closer to -1 at conditions of incipient
146 metallization (Fig. 4c). This is consistent with theories for conducting hydrogen [17,21,23,24],
147 which are well described by a Smith-Drude model with $C \neq 0$ [34]. Scattering times τ from
148 Smith-Drude fits are insensitive to pressure and temperature (Fig. 4d) despite conditions sampled
149 by experiment and theory ranging from 24-6,000 GPa, 1,000-125,000 K, and 0.3-5.4 g/cc in
150 pressure, temperature, and density, respectively [17,21,23,24], and are consistent with the
151 expected minimum scattering time (Ioffe-Regal limit) [12,13] where scattering occurs at the
152 interatomic spacing. Conductivity peaks at $\omega_m \approx 1/\tau$ when $C \approx -1$, or $\hbar\omega_m \approx 10$ eV for the
153 present data. The fact that conduction is maximized in conjunction with the shortest-distance
154 carrier motion possible indicates that transport is dominated by motion of bound carriers, such as
155 hopping [18], as opposed to unimpeded long-distance flow.

156 The temperature at which absorbing hydrogen appears (at detection limit $\alpha \approx 0.1 \mu\text{m}^{-1}$)
157 decreases weakly with pressure, remaining at 1700-2500 K at 30-110 GPa (Fig. 1). Here
158 $\sigma_0 \approx 10^{-3}$ S/cm, which is below the optical conductivity, $\sigma \approx 10^0$ S/cm. The data at 141 GPa

159 and 2400 K have $\sigma_0 \sim 10^1$ S/cm, and $\sigma \sim 10^2$ S/cm at visible frequencies (Fig. 4b). Fluid
160 hydrogen thus shows optical properties characteristic of a weak metal [17,21,23,24] and a
161 semiconductor undergoing gap closure [16] (σ increasing with frequency) throughout the
162 observed pressure range at temperatures of 1700-2500 K. Measured optical conductivities (Fig.
163 4) are less than those of the metallic state (~ 2000 S/cm) [12,29], whereas optical reflectivity R ,
164 estimated by assessing the Fresnel reflectivity between insulating (cold) and optically
165 transformed (hot) states in the experiment, is $R \sim [(4n\omega/\alpha c)^2 + 1]^{-1}$ or less than $\sim 1\%$ at
166 presently examined conditions.

167 Our data directly show hot fluid hydrogen retains a significant band gap to above 140
168 GPa pressure (Fig. 4) and temperatures of 2000-3000 K. Prior interpretations of conductivity
169 data, assuming a density-dependent, temperature-independent gap, predicted metallization at
170 these conditions (densities above 0.32 mol-H₂/cc) via compressive gap closure [3,12,29]. The
171 difference between our direct measurement and the prior model result is attributed to temperature
172 dependence of the gap. Indeed, the temperature at which absorption appears in fluid hydrogen is
173 nearly density- and pressure-independent between 30 – 110 GPa, suggesting gap closure is
174 primarily thermal rather than compressive.

175 Our definitive observation of a weakly conducting, semiconductor-like state of hot fluid
176 hydrogen in measurements to 150 GPa rules out the possibility of a rapid or first-order
177 transformation between insulator and metal at these pressures. This is inconsistent with some *ab-*
178 *initio* theoretical predictions [5,19,21,22] and supports more recent theories employing nonlocal
179 density functionals and nuclear quantum effects [25] or quantum Monte Carlo molecular
180 dynamics [27], which place a critical point at 250-375 GPa. Isentropic compression
181 measurements find the IMT becomes first order by 285 GPa [16], suggesting together with our

182 results an experimental critical point between 150 and 285 GPa. Also, the gap in temperature
183 between insulating and metallic conditions appears to be decreasing with pressure in the studied
184 range, consistent with the transition sharpening towards a critical point at higher pressures (Fig.
185 1): at 22 GPa, reflectivity [14] onsets 3710 K above absorption; at 45 GPa, the difference is 1540
186 K). Parallel behavior is seen in the DC conductivity (Fig. 1).

187 Prevailing first-principles models for hydrogen and hydrogen-bearing systems at high
188 pressure and temperature in giant planets [4,5,47] thus require a significant reassessment.
189 Compared with these theories, metallic conditions occur at higher pressure and temperature (i.e.
190 deeper within the planets), potentially influencing atmospheric coupling with the metallic layer
191 [6,7] and the conditions of hydrogen-helium phase separation. For example, as conditions of
192 phase separation are correlated with the location of the critical point [4,5,47], the increased
193 pressure of the critical point required by our direct observations to 150 GPa suggests phase
194 separation is unlikely to have occurred in Jupiter [34].

195 Our optical properties measurements on hydrogen cover a wide, previously unexplored
196 region of the phase diagram and bridge large gaps between prior dynamic and static compression
197 measurements of transformation and transport properties. Our data show the presence of an
198 intermediate absorptive but not metallic state of hydrogen at the boundary between insulating
199 and metallic regimes in a wide pressure range (10-150 GPa). This is inconsistent with first-order
200 insulator-metal transition and compression-driven gap closure that were previously inferred in
201 this region from experiments and theory.

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211 FIGURE CAPTIONS

212

213 FIG. 1. (color) Phase diagram of hydrogen. Black lines are phase boundaries. Present
214 measurements are filled circles for transparent (white), and absorbing (grey, black) hydrogen;
215 black points are characterized via direct transient absorption measurement (Fig. 2) whereas grey
216 points correspond to anomalous temperature responses observed upon increasing heating laser
217 power (Fig. 3). A thermal pressure of 2.5 GPa/1000 K [48] is included. The heavy black line is
218 onset of absorbing hydrogen in the present data. Prior measurements are the onset of reflectivity
219 in shock compression [14] (crosses and dotted line), the onset of visible absorption in isentropic
220 compression [16] (squares and dashed line), the location of anomalies in temperature with
221 increasing heating laser power in the DAC [15] (stars), and the DC conductivity (color map)
222 based on interpolated data [3,11,12,29,34,40]. The melting curve is taken from Ref. [28] and the
223 metallization line is the saturation of DC conductivity. White lines are interior conditions of
224 Jupiter [49] and Saturn [50].

225

226 FIG. 2. (color) Transient absorption and emission measurements in hydrogen at 141 GPa. (a)
227 Laser power (upper panel) and spectrogram showing transient absorption (lower panel). (b) Time
228 histories of absorption at different wavelengths using pulse referencing [33,34]. (c) Transmission
229 spectrum averaged over 2 to 5 μ s where absorption (and temperature) is roughly constant. (d)
230 Emission spectrogram (20 spectrograms stacked), with inset showing gray-body Planck fit to
231 data at 2 to 5 μ s. Temperature in this time interval was 2400(300) in a series of heat cycles at this
232 laser power.

233

234 FIG. 3. (color) Temperature histories at 30 GPa with finite element model predictions. Two
235 measurements (open symbols: vertical bars are temperature uncertainty, horizontal bars are time
236 resolution) are presented with finite element models [33,34,39] with and without an onset of
237 infrared absorption in hydrogen at a critical temperature of ~ 3300 K (solid and dashed lines,
238 respectively). Below the critical temperature (blue points), models (grey) are indistinguishable
239 and follow behavior typical for a transparent sample with laser energy absorption on the foil
240 surface [39]. For experiments achieving the critical temperature (red points), models (black)
241 show the result of sample absorption: rather than an initial peak and decay that scaled with laser
242 power, temperature is limited to values near the critical temperature [33]. Laser power increased
243 from 65 to 155 W between the models. Above 100 GPa transient absorption occurred without
244 this effect, since thinner samples at high pressure did not become infrared-optically thick when
245 heated.

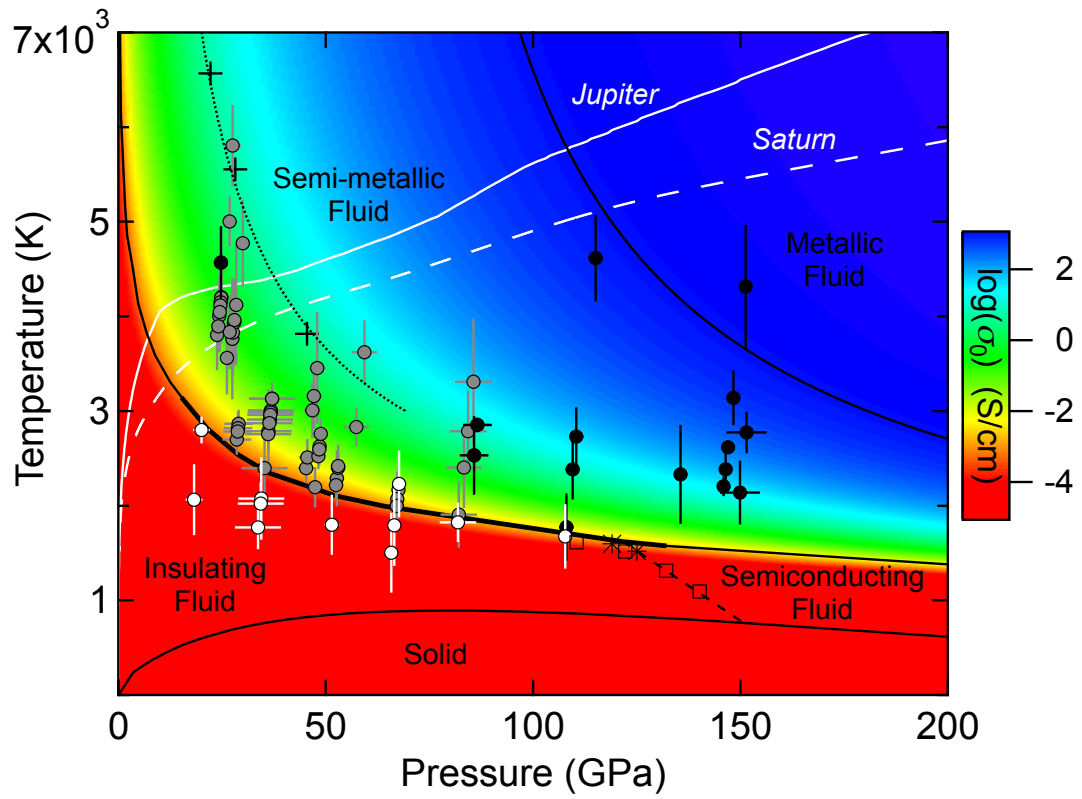
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247 FIG. 4. Optical properties of hydrogen. Data at 141 GPa and 2400(300) K are open circles (error
248 bars are systematic), theoretical predictions are crosses, and fits are lines. (a) Absorption spectra
249 with Tauc fits, with theory for semiconducting states at 1600-1700 K, 101-159 GPa [16]. (b)
250 Conductivity spectra with Smith-Drude fits. The DC conductivity corresponding to the present
251 data and used in the fitting is $\sigma_0 = 15$ S/cm (triangle). Theory for metal and nonmetal states are
252 for 1000 K, 170 GPa [21]. (c) Smith-Drude backscattering parameter C and (d) scattering time τ
253 are from theory [17,21,23,24,34] and experiment; shaded region in (c) is the conditions for
254 metallization [12,21,29] and in (d) the calculated minimum scattering time (Ioffe-Regal limit)
255 [12,13] for relevant conditions.

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258 FIGURES
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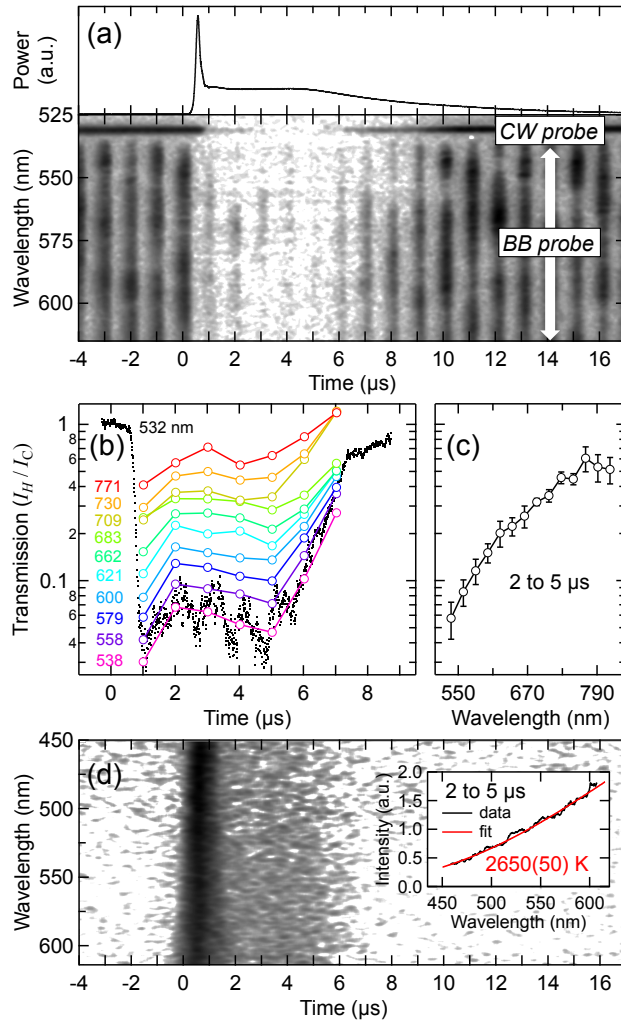
FIG. 1



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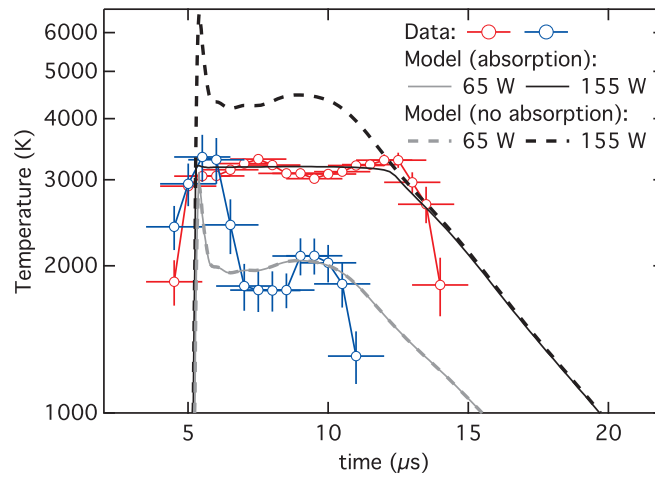
FIG. 2



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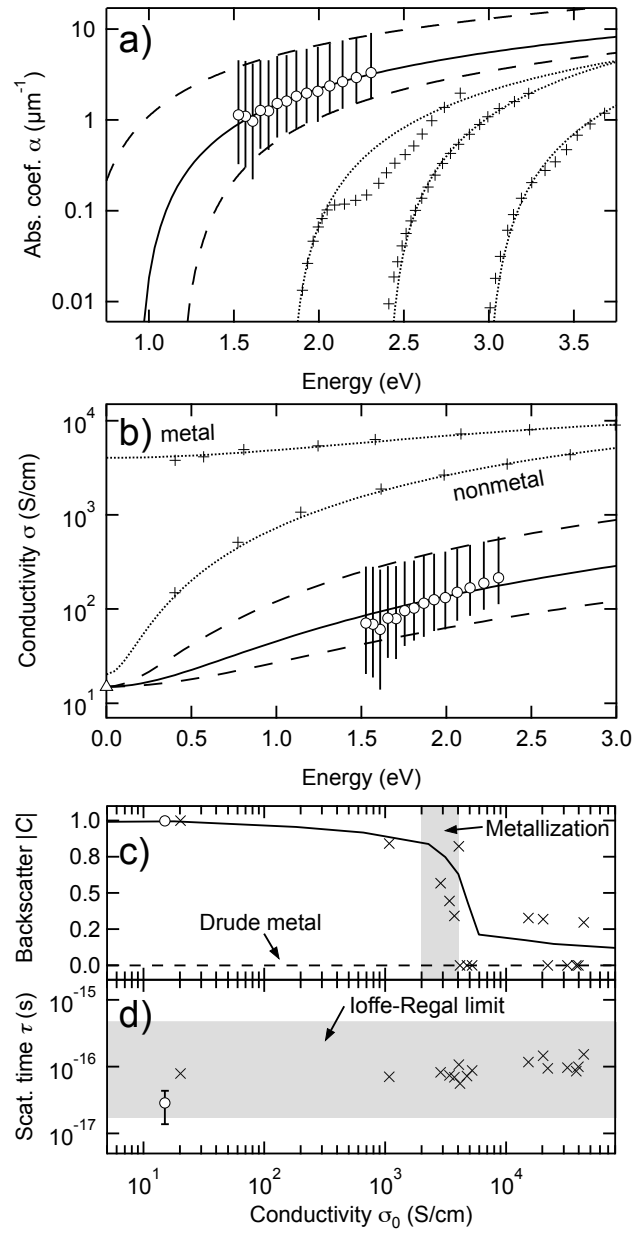
FIG. 3



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FIG. 4



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364

365 SUPPLEMENTARY TEXT

366 **Sample Preparation and Characterization**

367 Diamond anvil culets diameters of 300 μm flat, or beveled 300 μm with 100 μm culet
368 were used with Re gaskets. Foils were 30-60 μm across with holes of 5-10 μm diameter, and 3-
369 12 μm thick, whereas cavities were at least several μm thicker than the foil in typical
370 experiments. High-purity hydrogen gas was pressure-loaded. In total, 9 sample loadings were
371 made.

372 Samples were prepared, as in the prior experiment on noble gases [33], by placing
373 metallic couplers containing small holes directly on a diamond culet, prior to pressure-loading of
374 samples. This was done to add stability to coupler position while performing high-temperature
375 experiments, and to ensure the coupler surfaces were orthogonal to the optical axis of the DAC.
376 Interference fringes indicate that the gaps between the coupler and anvil on which it rested was
377 of order a few wavelengths of light at most, so ~ 1 μm . Laser heating was performed by heating
378 the coupler on the opposite surface (Fig. S1).

379 Low-pressure experiments ($P < 60$ GPa) were performed with the large culets, and so
380 corresponded to larger sample chambers, with total cavity thicknesses of 10-25 μm , and coupler
381 thicknesses of 6-12 μm . Higher pressure experiments ($P > 60$ GPa) with small culets
382 corresponded to significantly smaller samples of 5-10 μm total thickness and couplers of 3-6
383 microns in thickness.

384 Samples were originally configured to permit transient absorption measurement by
385 reflection from foils [33], in addition to transmission, however foil optical changes during
386 heating prevented successful measurements. Specifically, a permanent darkening of the reflective
387 foil, even after a few μs heating, was observed and was attributed to fast interfacial reaction.

388 Anvils often fractured when subjected to the highest achievable temperatures (4,000 to 6,000 K),
389 which was attributed to rapid hydrogen diffusion into the anvil. Above ~60 GPa hydrogen loss
390 from the sample was reduced and anvil failure eliminated, improving sample stability, though
391 total heating duration remained limited to less than ~1 ms.

392 *DC conductivity model for hydrogen*

393 The phenomenological model fitted to the DC conductivity data (see text) assumed
394 transition temperature and width varied with density ρ as $T_c^* = a_c + b_c\rho$ and $T_w^* = a_w + b_w\rho$,
395 where $\rho(P) = \rho_0 + c P^n$ was given as the density on the 300 K isotherm at pressure P [51].
396 With these assumptions, the best fit parameters, for σ in S/cm, T in K, ρ in mol-H₂/cc, and P in
397 GPa are $\sigma_m^* = 3.41$, $\sigma_j^* = 20.9$, $a_c = 2.22 \times 10^{-4}$, $b_c = 1.41 \times 10^{-3}$, $a_w = 5.25 \times 10^{-4}$,
398 $b_w = -3.86 \times 10^{-4}$, $\rho_0 = 0.0412$, $c = 0.0417$, $n = 0.437$. Note that these fit parameters are
399 based on shock temperature calculations [3,29] that assumed a continuous IMT, which are
400 considered to be consistent with the present direct observations.

401 We now review the design, physical significance, and phenomenological basis of this
402 model.

403 (1) During the IMT, it was previously found that $\log(\sigma_0)$ is proportional to $1/T$ [11,12,29]. This
404 model using the *erfc* function has such a linear relationship during the transition, such that in the
405 transformation region it is equivalent to a semiconductor excitation model similar to that used
406 previously for hydrogen during the IMT [3,11,12,29].

407 (2) The conductivity of hydrogen shows saturation when it becomes fully metallic [3,12-
408 14,16,29]. The *erfc*-function of the model captures this saturation, which cannot be described
409 using only the semiconductor treatment [3,11,12,29]. This model thus describes metallic,

410 semiconducting and insulating (see below) regimes and so includes semiconductor-like behavior
411 only during the transition.

412 (3) The insulating phase in our model fit is found at $T < 1000$ K, conditions where hydrogen is
413 known to be a good insulator in both fluid [15,16,40], and solid [28] phases.

414 (4) Our model implies a density- and temperature- dependent band gap for hydrogen (see below).
415 A gap depending linearly on density only, as used in prior analytical models of the IMT
416 [3,12,29], cannot describe the collected results. However, a linear density dependence of the
417 temperatures of transformation (T_w^* , T_c^*) does provide good consistency between the model and
418 available data.

419 5) The increases in conductivity in our model fit follow closely related changes in optical
420 properties. The onset of visible absorption occurs at $\sigma_0 = 8 \times 10^{-4}$ S/cm; in the phase diagram,
421 the contour of this conductivity accurately describes absorption onset (solid/bold line in Fig. 1);
422 similarly, a contour of conductivity at 0.65 S/cm (dashed line in Fig. 1) describes accurately the
423 onset of optical reflectivity [14]. Such topological agreement provides good phenomenological
424 support for this model.

425 6) While not occurring in the interpolated pressure range, our model allows for a discontinuous
426 transition (i.e. where transition width $T_w^* \rightarrow 0$). Instead, our model has a weak decrease in T_w^*
427 with pressure in the studied pressure range, consistent with the closing gap between absorption
428 and reflection onset observed with pressure (Fig. 1 and text). Extrapolating to higher pressures,
429 the model suggests $T_w^* \rightarrow 0$ at $P > 270$ GPa (in consideration of fit uncertainty) consistent with
430 dynamic compression observations [16] as well as recent theory [25,27]. The transition
431 temperature ($1/T_c^*$) similarly decreases with pressure, also consistent with optical data and
432 theoretical expectations [5,19-22,25-27].

433 We now consider in more detail how our DC conductivity model relates to a
 434 semiconductor model, as developed previously for the IMT [3,11,12,29]. Within the
 435 transformation our model

$$\sigma^* = \sigma_m^* - \sigma_j^* \{-0.5 \operatorname{erfc}[(T^* - T_c^*)/T_w^*] + 1\}$$

436 is well described by a linear Taylor expansion about the central transition temperature T_c^* , i.e.

$$\sigma^* = \sigma_m^* - \frac{1}{2} \sigma_j^* + \frac{1}{\sqrt{\pi}} \frac{\sigma_j^* T_c^*}{T_w^*} - \frac{1}{\sqrt{\pi}} \frac{\sigma_j^*}{T_w^*} T^*$$

437 If we take $\sigma^* = \ln(\sigma_0)$, as opposed to $\log(\sigma_0)$, then the form of this equation is identical to that
 438 for conductivity in a semiconductor [3,11,12,29]

$$\sigma^* = \sigma_0^* - \frac{E_g}{2k_B} T^*$$

439 and thus

$$\sigma_0^* = \sigma_m^* - \frac{1}{2} \sigma_j^* + \frac{1}{\sqrt{\pi}} \frac{\sigma_j^* T_c^*}{T_w^*}$$

440 and

$$E_g = \frac{2k_B}{\sqrt{\pi}} \frac{\sigma_j^*}{T_w^*}$$

441 Here fit parameters using $\sigma^* = \ln(\sigma_0)$ become $\sigma_m^* = 7.84$, $\sigma_j^* = 48.1$ (all other model
 442 parameters are unchanged). Following this treatment, our model is nearly identical to the
 443 semiconductor approach [11,12,29] in the low-pressure limit: at 20 GPa, our model implies $E_g =$
 444 10.4 eV during the transition (near T_c^*), whereas the earlier semiconductor analysis found
 445 11.7±1.1 eV at these conditions [11,12]. However, a very different behavior is found with
 446 increasing pressure, with the band gap remaining large (10.0-12.6 eV at T_c^*) through 140 GPa in
 447 our model, rather than closing by this pressure [12,29].

448 Thus in this model a semiconducting intermediate phase exists between the insulating and
449 metallic states at all studied pressures, in the vicinity of T_c^* , as required by our measurements.
450 Gap closure occurs with increasing pressure at constant temperature (similar to earlier studies),
451 but also with increasing temperature at constant pressure. These features are necessary to fit the
452 current experimental results on hydrogen (this study and Refs. [3,11-14,16,29]) as well as
453 provide general consistency with theory.

454 *Finite Element Modeling*

455 The approach for time-dependent finite-element modeling of temperature in the laser
456 heated diamond cell has been discussed previously [33,39]. The thermochemical parameters used
457 for H₂ are similar to Ref. [39], where heating of hydrogen through the melting transition was
458 modeled. Here we tested possible material models for H₂ with and without an onset of electronic
459 transformation at high temperature. A representative set of finite element model parameters is
460 provided in Table S1. Basic thermochemical parameters were taken from literature tables
461 whereas transport parameters for hydrogen (absorption and thermal conductivity) were varied to
462 obtain the best agreement with the experimental temperature histories. Measured temperature is
463 compared to the maximum temperature in the FE models, located at the coupler surface.
464 Geometrical parameters were chosen based on visual observations and visible interferometry
465 measurements. Two different sample configurations were modeled: a coupler without a hole (e.g.
466 at 30 GPa, Fig. 3) and coupler with a hole (e.g. at 141 GPa, Figs 2, 4, and S2). Our conclusions
467 were not found to be very sensitive to the selected parameterization. To reproduce observed
468 temperature plateaus (Fig. 3), a major increase in absorption (or, alternatively, thermal
469 conductivity) with temperature was required, indicating a high-temperature phase transition to a
470 conductive state.

471

472 **Table S1.** Finite element model parameters for Fig. 3. Parameters for the temperature-dependent
 473 absorption model in H₂ are $T_c=3300$ K and $T_0=100$ K, with $\alpha_0=10^6$ m⁻¹ (with absorption) or $\alpha_0=0$
 474 m⁻¹ (without absorption).

Property/Material	Sample (H ₂)	Absorber (Ir)	Anvil (C)
Density (kg/m ³)	420	25220	3500
Thermal Conductivity (W/mK)	100	226	2000
Specific Heat Capacity (J/kg K)	15000	130	509
Surface Emissivity	N/A	.35	N/A
Bulk Absorptivity (1/m)	$\alpha_0 \left(1 - 0.5 \operatorname{erfc} \left[\frac{T - T_c}{T_0} \right] \right) \times \left(\frac{T - 0.8T_c}{0.2T_c} \right)$	N/A	0

475

476 Index of refraction of hydrogen

477 Compared to the imaginary index k (or equivalently, α), which varies by many orders of
 478 magnitude in the IMT, the real index n can only vary by, at most, about an order of magnitude,
 479 so is always of order 1. Thus to estimate the conductivity of hydrogen during the IMT, we have
 480 used n of cold hydrogen at high pressure [44,45], with deviations from this value expected to be
 481 small compared with uncertainty. To further examine the validity of this assumption, we have
 482 used the Smith-Drude model to compute expected variations in real index as metallization occurs
 483 (Fig. S8). This model describes warm dense hydrogen sufficiently well (Fig. 4 and S6) and
 484 satisfies the Kramers-Kronig relations, so can accurately treat the relative variations of real and
 485 imaginary indices as electronic properties change. This analysis (Fig. S8) suggests the real index
 486 should not change by more than a factor of two as the sample becomes absorbing. Melting,
 487 which corresponds to no major electronic or density change [28], should also not correspond to

488 any substantial change in real index. Thus variation in the real index from the ambient
489 temperature values are considered negligible in our experiments, compared to total uncertainty
490 (about an order of magnitude, Fig. 4).

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494 SUPPLEMENTAL FIGURES OVERVIEW

495 Fig S1 is a schematic summary of our technique [33].

496 Fig. S2 is a detailed discussion of Fig. 2 in the main text, containing the full data and
497 analysis.

498 Fig. S3 contains example Raman spectra discussed in the text, including a typical before-
499 and-after spectrum (a), and a spectrum from the single sample where reaction occurred during
500 heating (b).

501 Fig. S4. shows transient absorption data at several pressures with varying degrees of
502 hydrogen loss.

503 Fig. S5. is the data compilation [3,11,29,40] and fit used to assess DC conductivity, as in
504 Fig. 1.

505 Fig. S6 is the theory compilation [17,21,23,24] used to assess the suitability of the Smith-
506 Drude model in describing warm dense hydrogen, as referenced in Fig. 4.

507 Fig. S7. shows implications of our results for gas giant planets, as discussed in the
508 conclusions.

509 Fig. S8. considers plausible variations in the real index of refraction of hydrogen as
510 electronic properties change.

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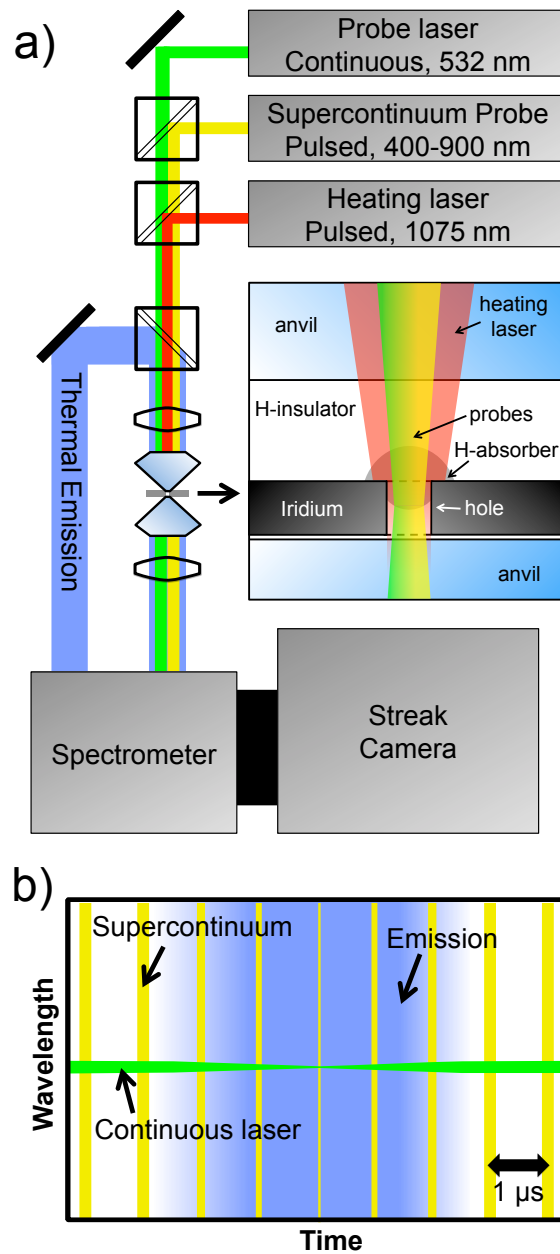


FIG. S1. Schematic of the experiment [33]. (a) Laser heating of a metallic (Ir) foil heats the surrounding, initially-transparent hydrogen. Simultaneously, the sample is probed with a combined beam for transient absorption spectroscopy comprised of a continuous (CW) laser and supercontinuum broadband (BB), which is transmitted through a hole in the foil containing heated hydrogen. Thermal emission is collected through either anvil, and together with probe signal is delivered to a spectrometer with a streak-camera detector; temperatures were measurable from the heated side of the foil for transparent samples or either side for absorbing (emissive) samples. (b) Streak-camera spectrogram of transient absorption during an emissive heat cycle. Laser and supercontinuum probes are discriminated from emission by being monochromatic and pulsed, respectively.

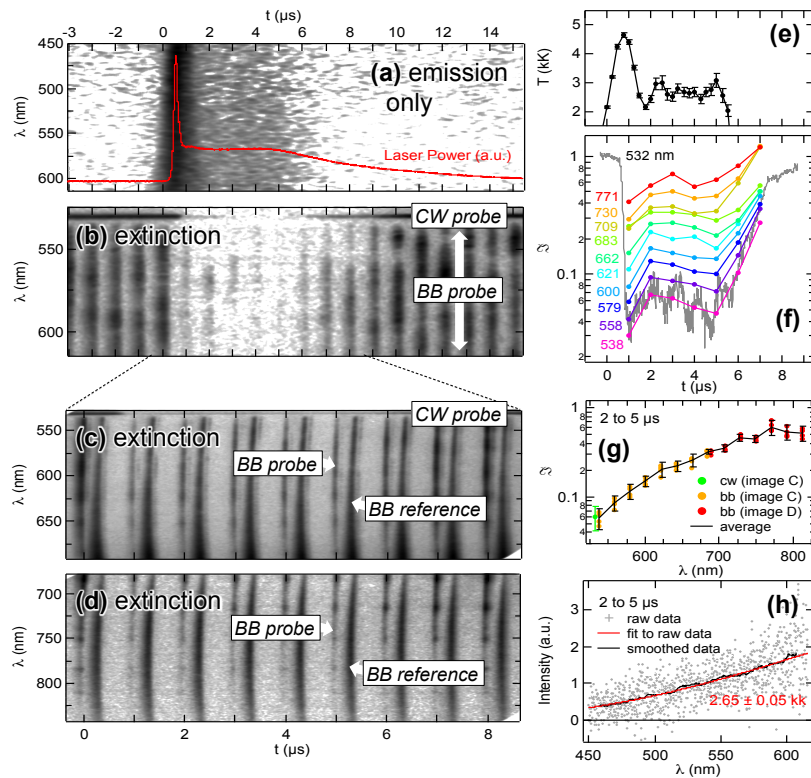


FIG. S2. Detail on Fig. 2. Transient absorption and emission measurements in warm dense hydrogen at 141 GPa and constant laser power. Spectrograms (a-d) are (a) emission alone (20 cycles integrated), (b) Transient absorption (1 cycle) without referencing BB pulses, and (c-d) transient absorption (10 cycles integrated each) with BB pulse references. The latter images are taken using a faster sweep to include a reference pulse at slightly different time delay, appearing immediately after the probe pulse, to track probe spectral instability [33]. A representative temperature history (e) is taken from (a). Transient absorptions (f,g) are taken from (c,d); only the central 7 pairs in (c,d) can be reliably analyzed. Transmission spectra (g) are based on stable temperature zone (e) between 2 and 5 μs . The spectra in two bands (c,d) were combined by assuming agreement at the region of overlap (690 nm) and including systematic offset in systematic error (Fig. 4). The integrated emission from 2-5 μs (h) closely matches a Planck distribution. The temperature between 2 and 5 μs (e) varies in the range $2,400 \pm 300$ K at this laser power. Temperature errors in (e,h) are fitting error; transmission error in (g) is random uncertainty (1-sigma).

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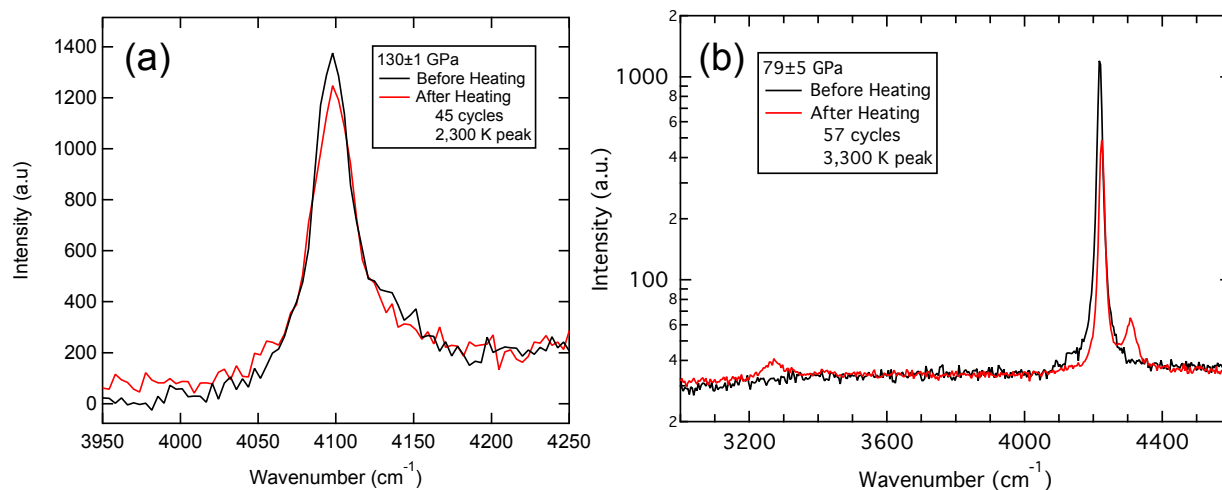


FIG. S3. Raman spectra at (a) 130 GPa (linear scale) and (b) 80 GPa (log scale) before and after heating. The post-heat spectra in (b) contain a hydride signal near 3300 cm⁻¹ and related vibron signature near 4300 cm⁻¹ as well the pure H₂ vibron (most intense peak, by 1-2 orders of magnitude). Raman evidence of reaction was only seen after the heating series performed in (b); we conclude that bulk reaction was normally prevented by the short timescales of our experiments.

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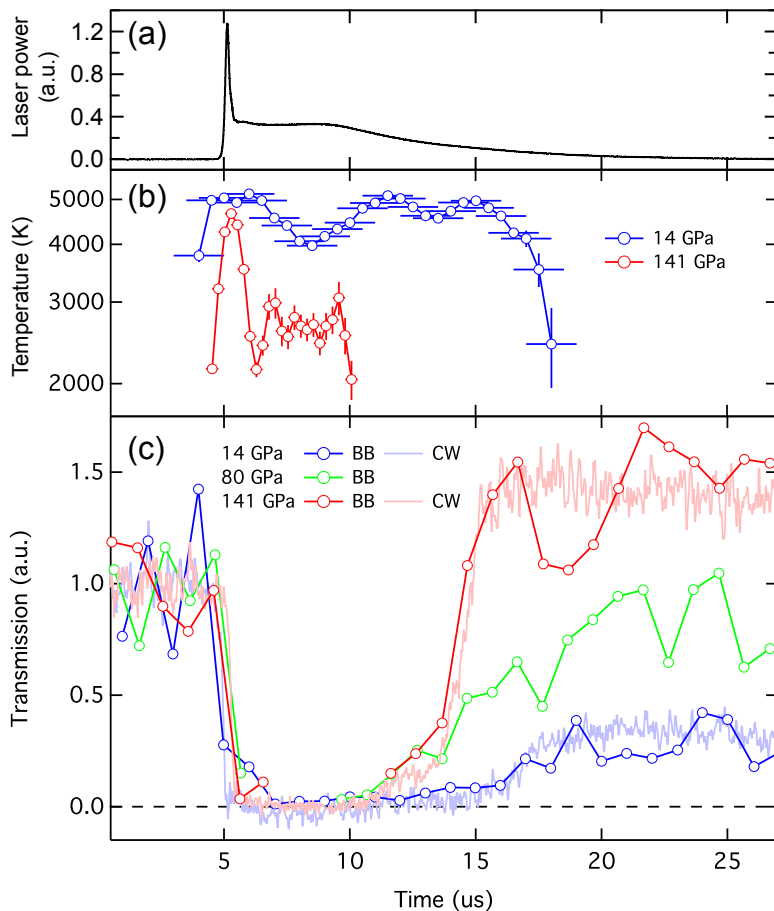


FIG. S4. Transient absorption measurements at 14, 80, and 141 GPa. (a) Laser power. (b) Time resolved temperature, with vertical bars showing temperature fit error and horizontal bars showing time resolution. (c) Transmission behavior during single shots. As a consequence of rapid hydrogen diffusion from the heated area, foil hole diameters often decreased during heat cycles, such that transmission did not always return to the original level. However a high transient extinction due to sample absorption could still be resolved. Slower diffusive loss at higher pressure allowed for relatively stable hole dimensions over many heat cycles. Note that for the 80 GPa experiment, high-time resolution temperature data was not obtained, and sample cavity interference [33] prevented use of the CW probe data. At 14 and 80 GPa, the initial hole size was larger than the probe spot, and closure of the hole was clearly evident visually and in the transmission data; at 141 GPa, the initial hole size was similar in size to the probe spot and remains so after heating. In this experiment, small changes in hole shape during heating (attributed to local melting around the rim of the hole) contributed to a small permanent increase in transmission during heating.

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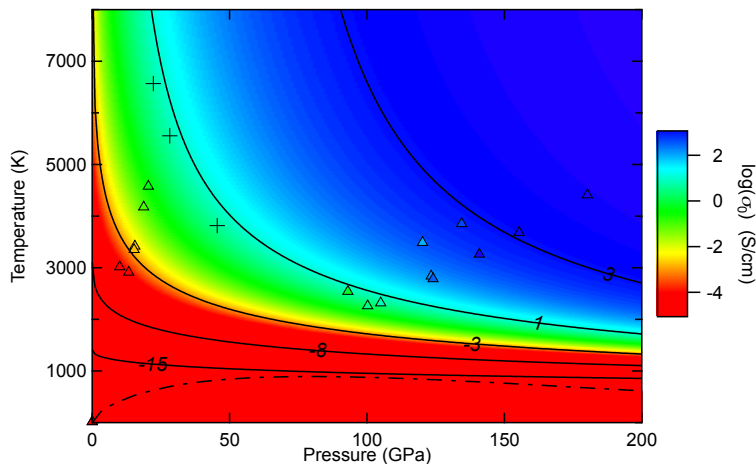


FIG. S5. Detail on DC conductivity model. Dynamic compression [3,11,29] and cryogenic [40] conductivity data on dense hydrogen and deuterium fluid used in our fit are given by colored triangles. The colors (as in Fig. 1) and black solid contours labeled with corresponding conductivity in log units ($\log(\sigma_0)$, in S/cm) are the conductivity model fit. The melt curve is the dashed dotted line [28]. The onset of optical reflectivity in shock compression (crosses) [14] follows the trend of fitted conductivities, with the critical conductivity for reflectivity ~ 1 S/cm. Pressure and temperature conditions of transformation are similar in deuterium and hydrogen [3,12,14,29], allowing for this global fit; small differences in the phase diagrams are averaged over, as there is good coverage of data for both isotopes in the examined domain.

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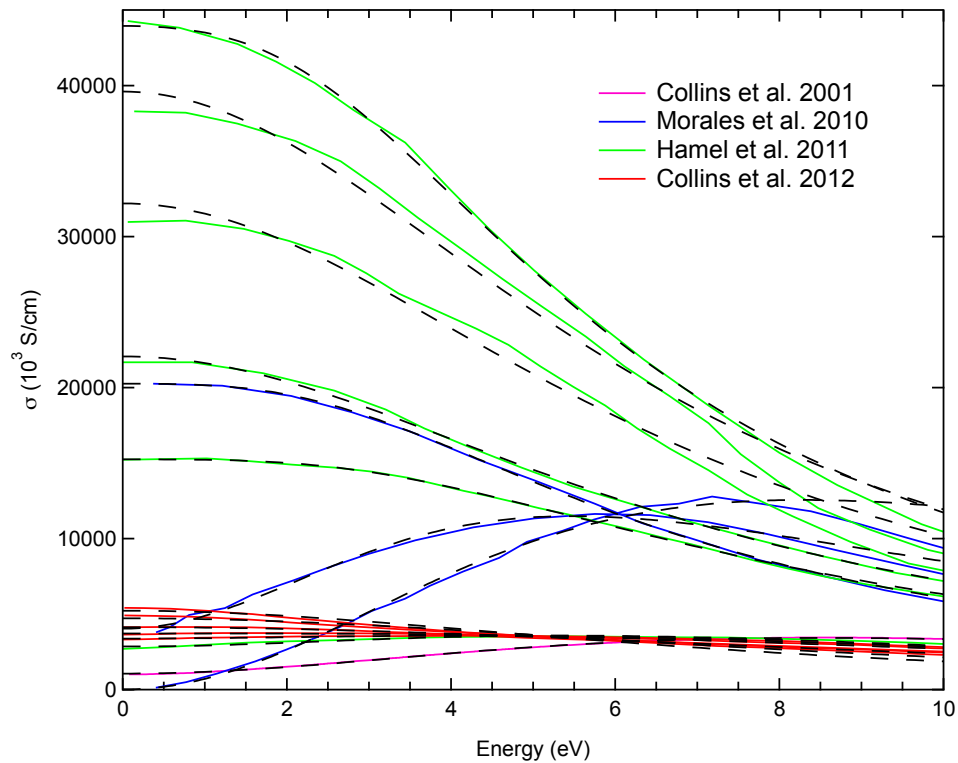


FIG. S6. Detail on Fig. 4. Smith-Drude [46] fits (dashed curves) to conductivity theory for warm dense hydrogen and deuterium at metallization; results are for 24-6000 GPa, 1000-125000K, and 0.3 to 5.4 g/cc [17,21,23,24]. Smith-Drude fits were obtained from the range 0-8 eV. In some cases where spectra had Drude-like character the fit was not fully constrained, in which instances C was set to zero for the fit (Drude model).

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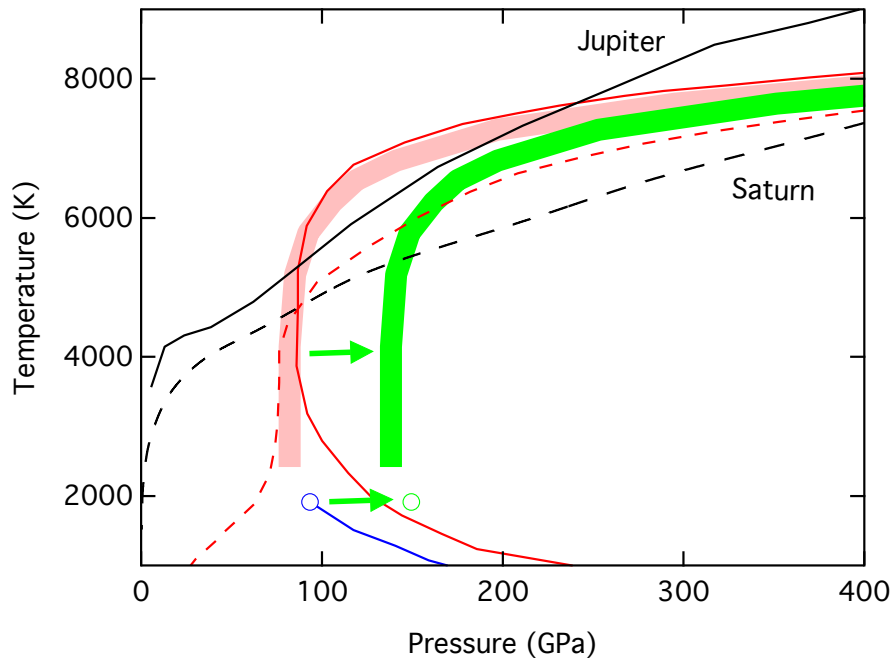


FIG. S7. Effect of high-pressure metallization in hydrogen on phase separation in giant planets. Shown are the predicted maximum temperatures of immiscibility in the hydrogen-helium system (red solid [5] and dashed [47] curves, red band showing upper limit) with corresponding location of the critical point (blue circle) and first-order metallic transition (blue line) [5,47]. Shifting the critical point to higher pressures (green arrow) as required by our measurements to 150 GPa, the upper limit on immiscibility, tied to the first-order metallization [4,5], shifts (green band) to fall outside the conditions of Jupiter's [49] interior (solid black line). Also shown is Saturn's interior [50] (dashed black), where immiscibility conditions remain plausible.

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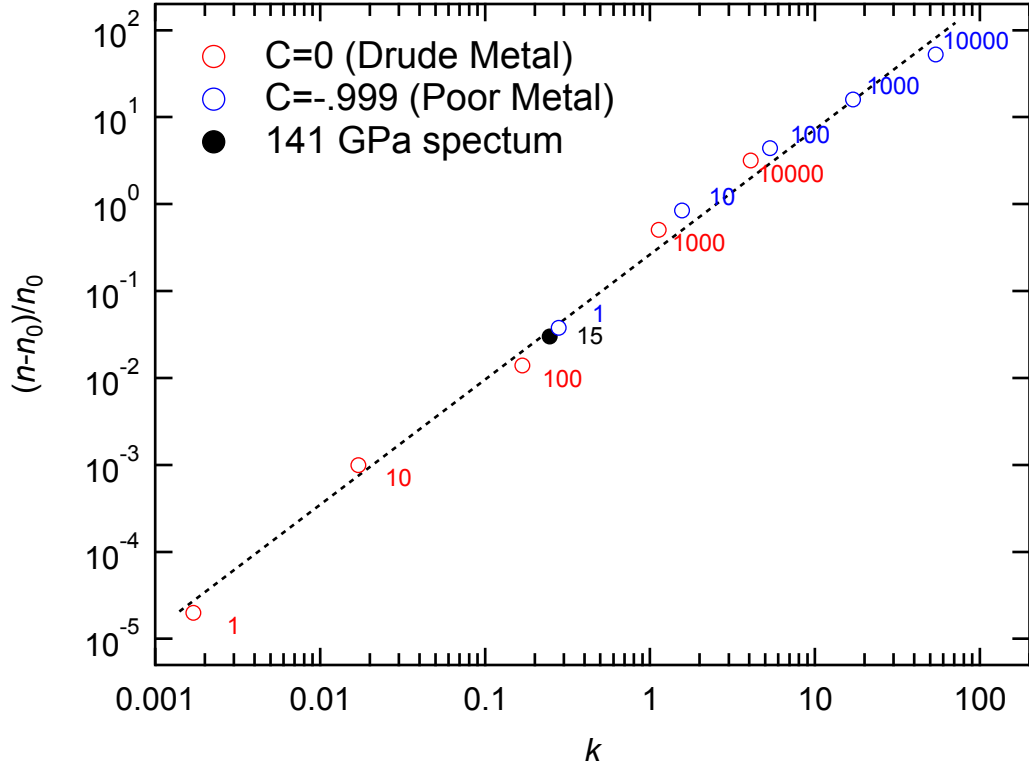


Fig. S8. Effect of metallization on the index of refraction, according to the Smith-Drude model. Results are at optical energy of 2 eV, assuming $\tau=10^{-16}$ s, a range of σ_0 (values listed in S/cm), and $C=0$ (red) or $C=-0.999$ (blue); fit to present data at 141 GPa is the black point. The fractional change in real index n relative to that of the insulating state (n_0) is given as a function of the imaginary index $k = \alpha c/2\omega$. At conditions where $k < 1$ the change in n is less than a factor of two, and it is a few percent at conditions of the 141 GPa measurements (Fig. 4).

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