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Effect of shear angles on material flow during incremental ECAP of Al-1050 billets

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Abstract. Incremental equal channel angular pressing is an evolution of the traditional equal channel angular pressing method used to produce ultra-fine grain materials with significant mechanical properties compared to its coarse grain counterpart. In this paper we look at effects of using 90° and 120° after a first pass for a pair of 10X10X60mm billets. The forces required to produce the billets was examined and compared. Hardness maps were created to examine the change in materials mechanical properties. It was found that using the 90° configuration results in higher press forces but greater uniformity of hardness distribution when compared to 120°. The results correlated to the findings of the simulations that were carried out prior to the experimental investigation.

1 INTRODUCTION

Severe plastic deformation (SPD) is gaining significant attention by researchers as a method for refining the average grain size of material and to obtain ultrafine grain (UFG) material where grain size is below 1 micron [1-3]. UFG material exhibit enhanced physical properties compared to their coarse grained counterpart. Among the various SPD processes, Equal channel angular pressing (ECAP), is one of the only few techniques capable of producing fully dense bulk UFG samples in large enough sections that can be used for structural applications [4-6]. First introduced by Segal and his co-workers in the former Soviet Union [7, 8]. During ECAP the billet material undergoes shear plastic deformation without change in cross-sectional area. The process has to be repeated to impose large cumulative strains in the material before obtaining UFG size. Different routes [9] may be employed in order to process the material; Route A: in which the orientation of specimen remains unchanged in successive passages; Route B: in which the specimen is rotated 90° between consecutive passages; Route C: in which the specimen is rotated 180°. Depending on the processing route, microstructural characteristics will differ from one route to another. The other factor that can affect the grain refinement and induced

mechanical properties is the extrusion angle. In this paper we will systematically examine the mechanical, microstructural properties, material flow as well as finite element modelling the differences of 90° and 120° Incremental equal channel angular pressing of AA-1050

2 I-ECAP PROCESS

In Incremental equal channel angular pressing (I-ECAP), the material feeding stage and the deformation stage are separated as opposed to classical ECAP [10]. The illustration of I-ECAP process shown in Figure 1 is a two billet version used in this study. A punch which follows a sine wave form (oscillates at a certain frequency and amplitude) comes cyclically in contact with the billets. The billet material is pushed into the deformation zone in increments of distance ‘a’ in Figure 1 (also known as feeding stroke) by the pusher tool, this material feeding is done when the punch is retracting. The punch then comes down and deforms the billets during the deformation stage. To facilitate material flow into the output channels the punch has a spike in the middle (not shown in the illustration). The mode of deformation is similar to that in classical ECAP i.e. simple shear, provided the feeding stroke is not large. By separating the feeding and deformation stages, reduces or eliminates friction during feeding. Thereby substantially decreasing the feeding force and enables processing of very long/infinite billets.

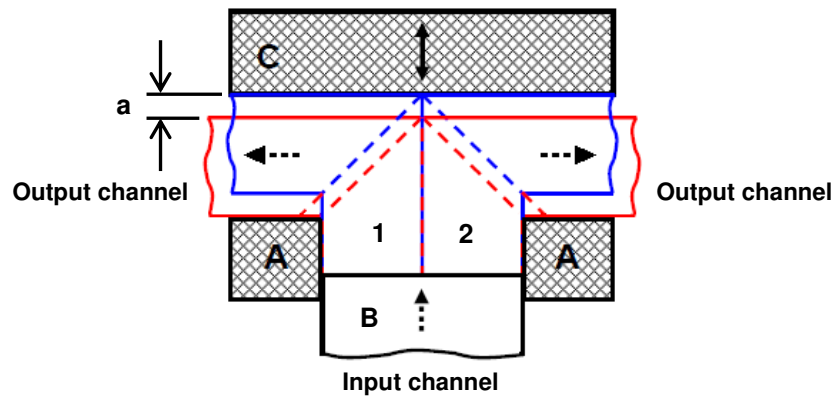


Figure 1: Illustration of I-ECAP process (A=Die, B=Pusher and C=Punch) [10].

The Figure 2 shows the relative motion of tools (punch and pusher) in a typical I-ECAP process. Here the punch is oscillating with an amplitude of 1.6mm at 1 Hz and feeding stroke is 0.5mm/cycle.

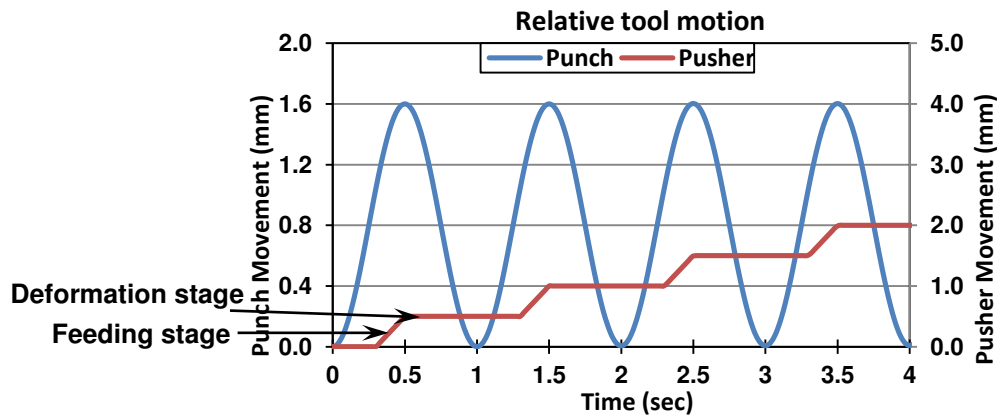


Figure 2: Relative movement of punch and pusher tools during the I-ECAP process

3 EXPERIMENTAL SETUP

The I-ECAP process is carried on a customized 1000KN servo-hydraulic press which is controlled by Zwick's Control Cube via Cubus software. The punch is attached to the press actuator and follows a sinusoidal cyclic command during processing. The material feeding is realized by a screw jack which is driven by a servo-motor. A dedicated LabVIEW application controls and synchronizes the material feeding by monitoring the punch oscillation. The application also records and captures deformation and feeding force during processing.

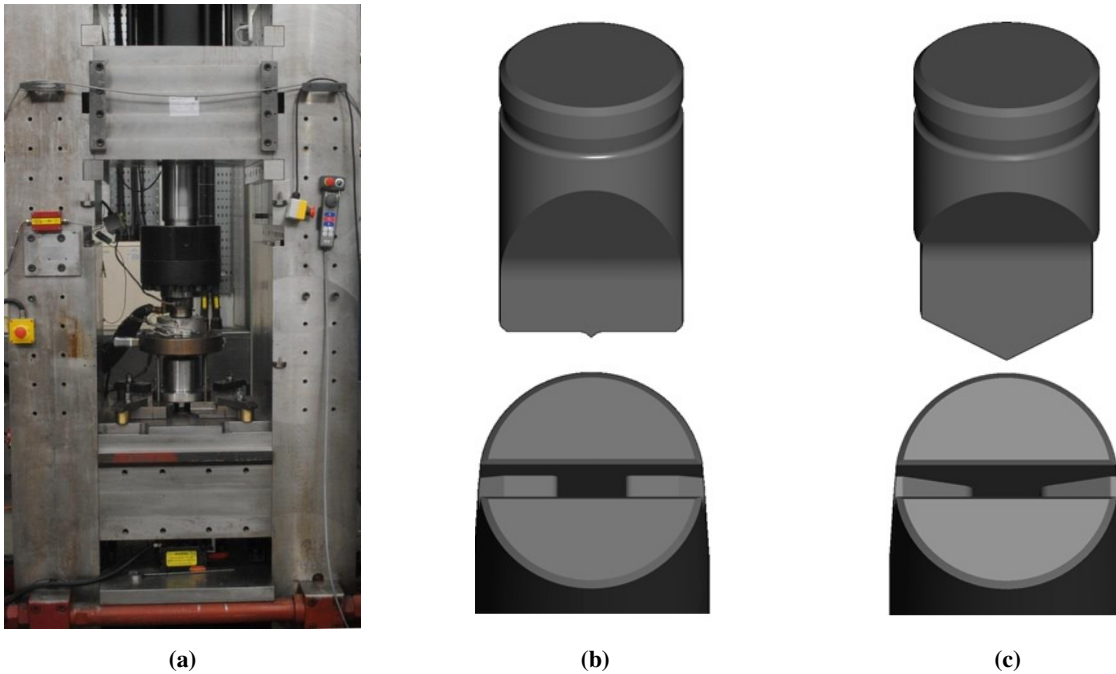


Figure 3(a): The I-ECAP experimental rig on 1000KN servo-hydraulic press. The punch and split die design for (b) $\phi=90^\circ$ I-ECAP die and (c) $\phi=120^\circ$ I-ECAP die.

3.1 Experiments:

Prior to starting the experiments, the chemical composition and microstructure of the starting AA-1050 billets were examined. AA-1050 is 99.5% Aluminum with traces of the following elements Cu, Mg, Si, Fe, Mn, Zn, Ti. A sample of the material was polished and prepared for Scanning Electron Microscope (SEM) examination. The image obtained can be seen in figure 4a. Image analysis was used map the grains (as shown in Figure 4b), to determine the mean grain size of material to be $140\ \mu\text{m}$.

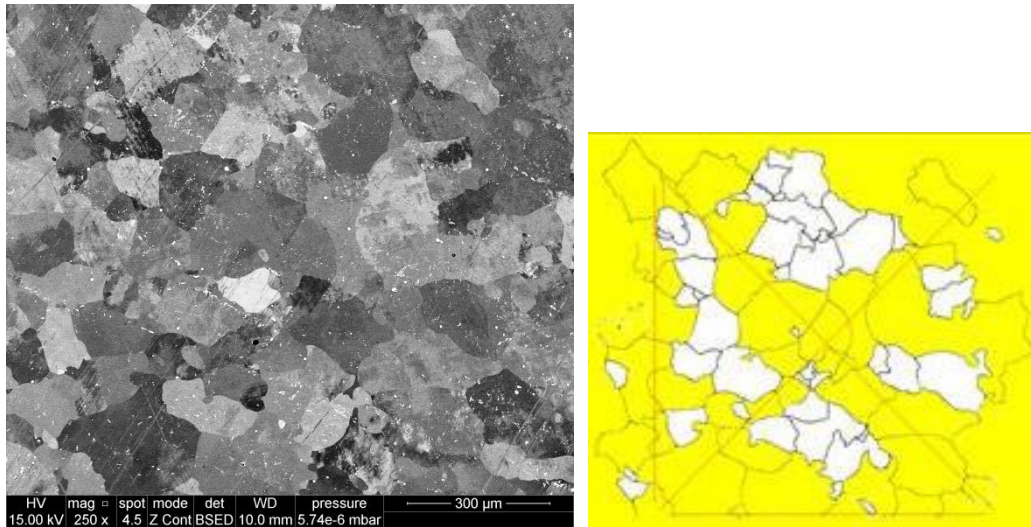


Figure 4: initial microstructure of the AA-1050 material

Experiments were performed using AA-1050 billets measuring 10x10x60mm, the billets were machined from round bars in the extrusion direction. Before the experiments could be conducted, it was necessary to perform some auxiliary steps which were related to billet preparation. These included sandblasting of billets, conversion coating the billets with Calcium Aluminate and finally applying a thin layer of Loctite 8009 (a graphite based anti-seize lubricant) from Henkel technologies. The purpose of these auxiliary steps was to reduce the friction during the processing and to avoid the sticking of billet material onto the die walls. I-ECAP process was carried out at room temperature with a cycle frequency of 30 strokes per minute (0.5 Hz), peak to peak punch amplitude of 1.6 mm and a feeding stroke of approximately 0.2 mm.

Figure 5(a) and (b) represents the shape of the billets after 50% processing with $\phi=90^\circ$ and $\phi=120^\circ$. Figure 6(a) and (b) shows the recorded forces during the two experiments.



Figure 5: I-ECAPed billets halfway processed (a) $\phi=90^\circ$ and (b) $\phi=120^\circ$

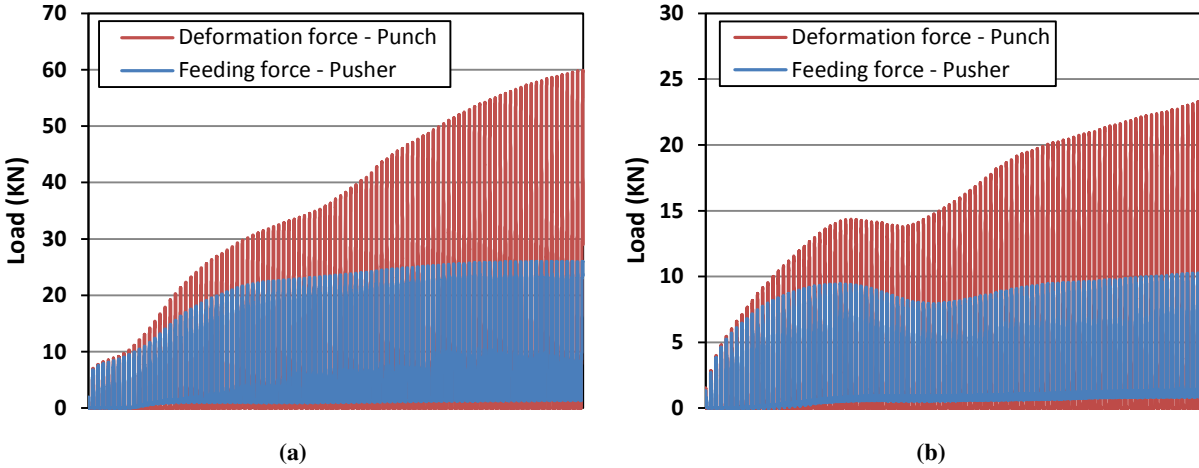


Figure 6: Reordered forces during I-ECAP processing for (a) $\phi=90^\circ$ and (b) $\phi=120^\circ$

4 FINITE ELEMENT SIMULATION

In order to analyse the behavior of the billets in the channel die, the total equivalent plastic strain, material flow and temperature rise in billets and tools during I-ECAP processing, FEM simulations have been carried out. A 3D model of I-ECAP process for two channel intersection angles (ϕ) of 90° and 120° has been developed using commercial FE code QForm. QForm was used as it had previous experience showed the capability of the software to complete simulations quickly and accurately. The material model used for these simulations can be found in the standard QForm software therefore they did not have to be found experimentally. Figure 7(a) and (b) shows the FE model for 90° and 120° channel intersection respectively, consisting of the tools and initial position of billets.

All tools (punch, die and pusher) were modelled with H13 tool steel and for the sake of simplicity are considered as rigid bodies; however with heat transfer capability. The two die elements were constrained in all degrees of freedom, however the punch and pusher were allowed to translate only along the Z-direction. The two AA-1050 billets were modelled with the built-in elastic-plastic material model in QForm which includes strain hardening behavior and strain rate effects. The billets were divided into four node tetrahedral elements. Heating of billets due to plastic deformation and friction during I-ECAP process was considered. The friction between the inner surfaces of the tools and the outside surface of the billet was modelled using Levanov law with friction factor taken as 0.3 under the graphite based lubricating conditions. However frictionless conditions were assumed between billet-billet interfaces, as the two billets are moving relative to each other. Also no interactions were considered between the tools. All simulations used automatic remeshing in billets to replace excessively distorted elements due to large strain and the occurrence of flow localizations. Volume constancy was also selected to ensure overall billets volume remains same after each remeshing step. Explicit method of integration was selected for solution.

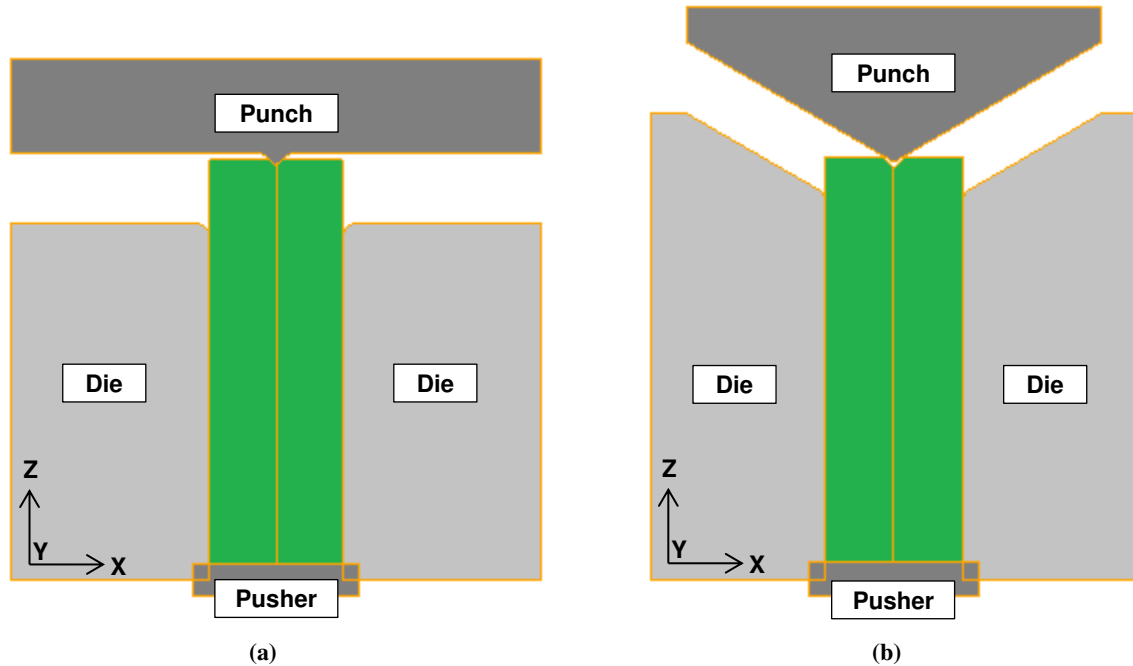


Figure 7: FE model for channel intersection angles of (a) $\phi=90^\circ$ and (b) $\phi=120^\circ$ I-ECAP process.

5 RESULTS

5.1 Strain distribution

The Figure 8(a) and (b) shows the evolution of equivalent plastic strain at various stages of processing for $\phi=90^\circ$ and $\phi=120^\circ$ channel intersections respectively. As predicted the overall plastic strain is higher for the $\phi=90^\circ$ as compared to $\phi=120^\circ$. For both cases, along the billet axis from left to right, there are three distinct deformation regions: tail, steady state and head. The steady state region of $\phi=90^\circ$ seems to show little strain in-homogeneity along both the billet axis and also along the transverse axis from top to bottom. However $\phi=120^\circ$ shows strain in-homogeneity along both axes, it is also showing some bowing which was observed during actual experiments as well. This means the billets have to be straightening before they can undergo other passes. The front surfaces of the two billets in the head region are quite different from each other because of the different ϕ .

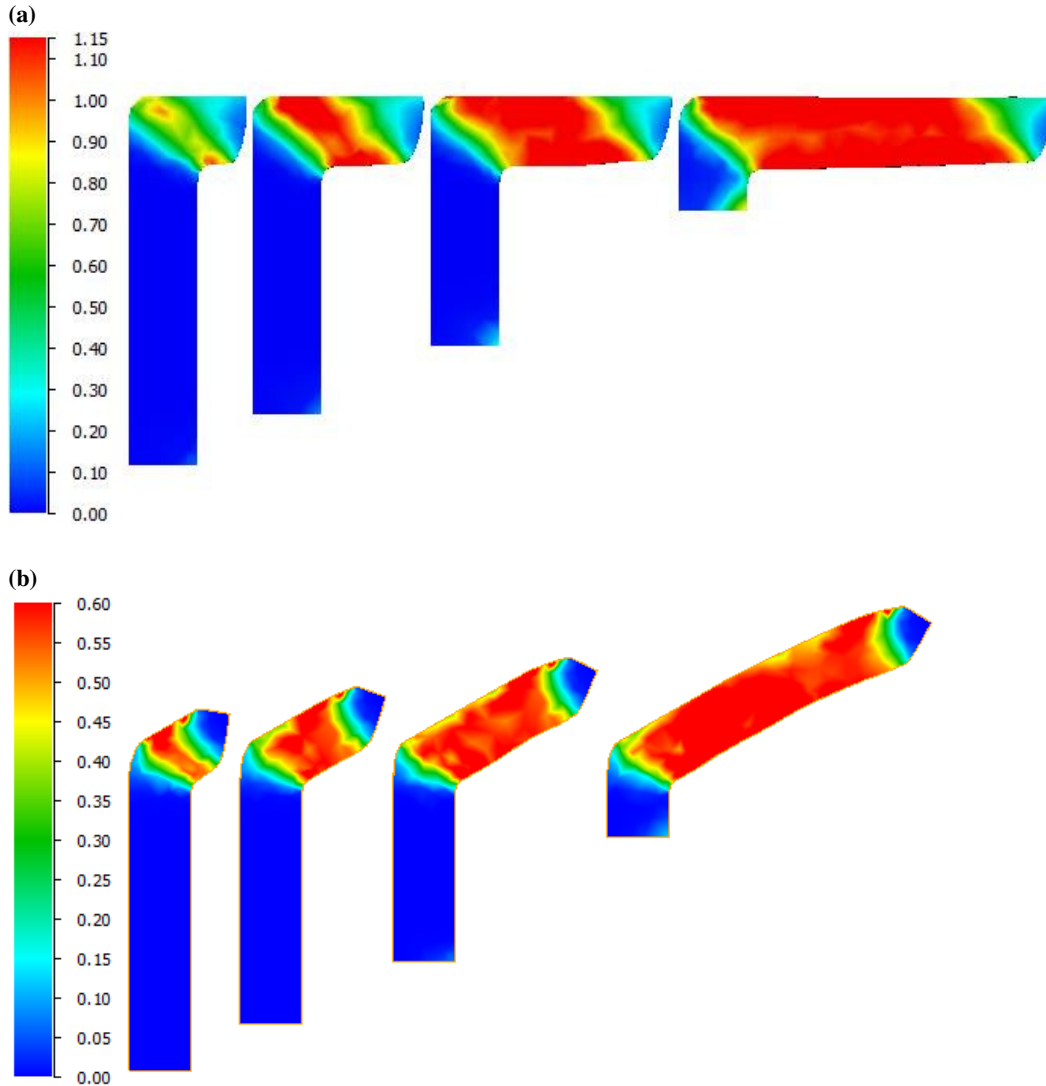


Figure 8: Evolution of equivalent plastic strain distribution at 15%, 30%, 50% and 90% processed state for (a) $\phi=90^\circ$ and (b) $\phi=120^\circ$.

5.2 Material flow

The material flow pattern and the ability of the billets to fill the die is examined by dividing the billets in a grid format using 30 lines along the billet axis and 7 lines along the transverse axis. Figure 9(a) and (b) shows the influence of ϕ on the material flow at 50% processed state for $\phi=90^\circ$ and $\phi=120^\circ$. The figure also shows an enlarged single element passing through the shear zone. Shear angle 'a' in case of $\phi=90^\circ$ is smaller as compared to shear angle 'b' in case of $\phi=120^\circ$. However compared to the $\phi=90^\circ$, $\phi=120^\circ$ exhibits better die filling and has almost no corner gap.

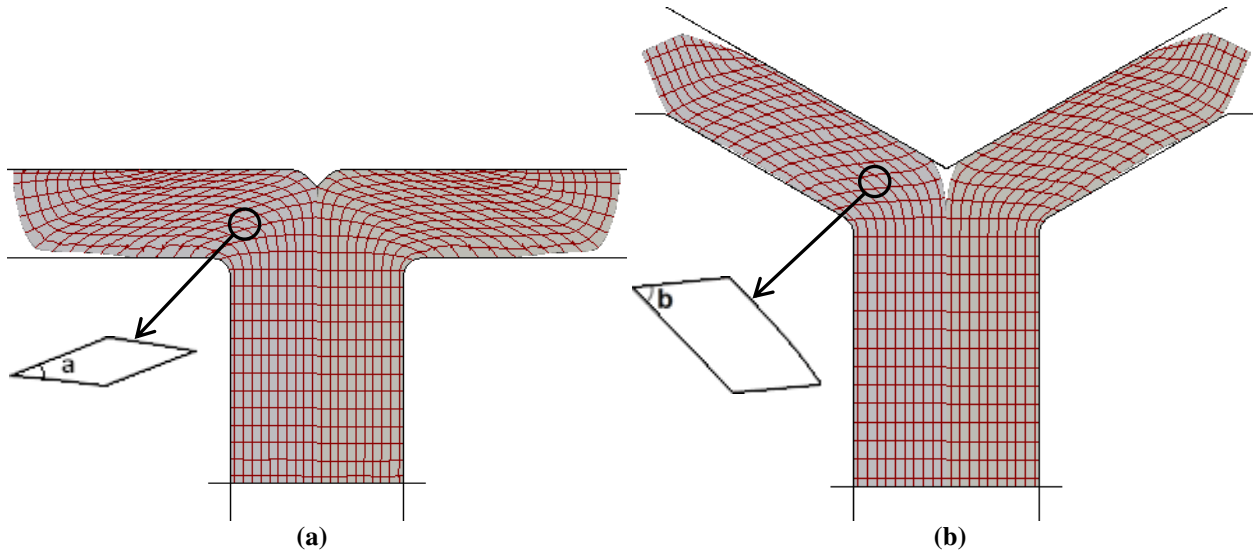


Figure 9: Influence of channel intersection angle ϕ on the material flow (a) $\phi=90^\circ$ and (b) $\phi=120^\circ$.

5.3 Hardness measurement

A detailed hardness measurement study was undertaken to further the understanding of material change behavior during the incremental ECAP process and comparisons could be drawn between the 90° and 120° configuration. A Zwick ZHV μ micro hardness tester was used to produce the hardness maps presented in figure 6. The ZHV μ Micro Vickers hardness tester covers Vickers and Knoop hardness tests to ISO 6507, ISO 4545 and ASTM E 384 with a test load range of 0.01 to 2Kg.

The specimens were cut in half along the flow plain. The cut surface was ground and polished. A large array was configured to measure the hardness of the specimen with 1 mm intervals. The hardness of the outer 1mm of the specimens was not measured. Performing hardness measurements close to the edge of a sample will result in incorrect low readings being achieved. About 500 hardness measurements were performed for each configuration.

After the first incremental ECAP pass, the specimens show significant hardening both for 90° and 120° . The 90° first pass showed both the highest hardness and the greatest uniformity.

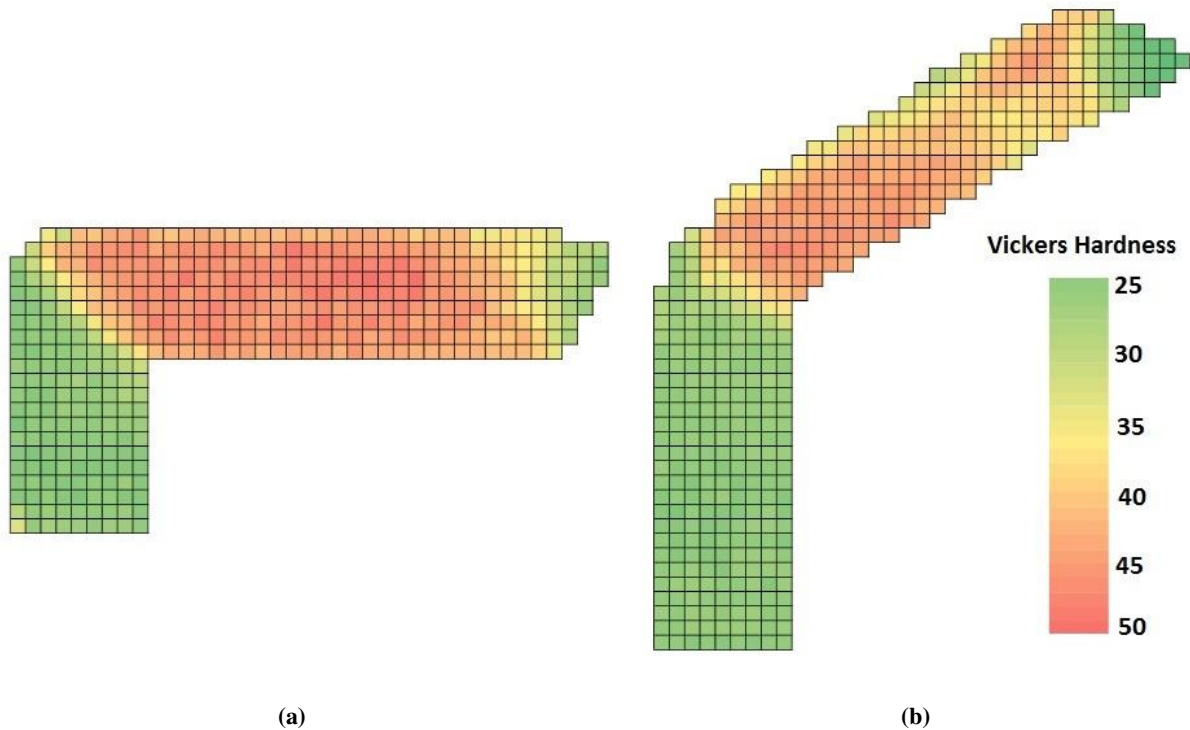


Figure 10: FE hardness map for billets during Incremental ECAP s of (a) $\phi=90^\circ$ and (b) $\phi=120^\circ$ I-ECAP process.

There is a distinct region separating the softer unprocessed material and the post processed material in the 90° configuration but for the 120° configuration there is a widening “V” shape transitional area separating the processed and unprocessed material. It is also noticeable that a larger section of the starting billet has remained softer in the 120° configuration and the following material exhibits less uniformity in its hardness as well as a lower peak hardness value. For both configurations, it is possible to see the headrest measurements were achieved in the center section of the billets with the outer 1mm of the material remaining 5HV softer.

5.4 Microstructure

First observation: The E112 was used to measure the grain size. The mean grain size in the first pass of I ECAP of 90° and 120° are 14 and 17 micron respectively. This illustrates that smaller shear angles lead to smaller grains, even after the first pass. The figure 8 indicates that the shear deformation and consequently equivalent strain in 90° is higher than 120° which results in the grain refining at different rates. At the initial stages of I-ECAP, as a general accepted mechanism, new dislocations generated via shear deformation resulting in the sub-grains structure. By preceding the I-ECAP, misorientation of the sub-grains increase and become grain boundaries. This could explain the increase in the mean hardness in 90° as compared to that of 120° . This attributed to the grain size as well as dislocation density Figure 10.

The differences between the microstructure in Figure 11a, and smaller mean grain size of 90° compared to the 120° is due to the number of dislocations produced in each process. Second observation: Through the I-ECAP, the microstructure is more uniform than the conventional ECAP (as hardness results is shown in Figure 10).

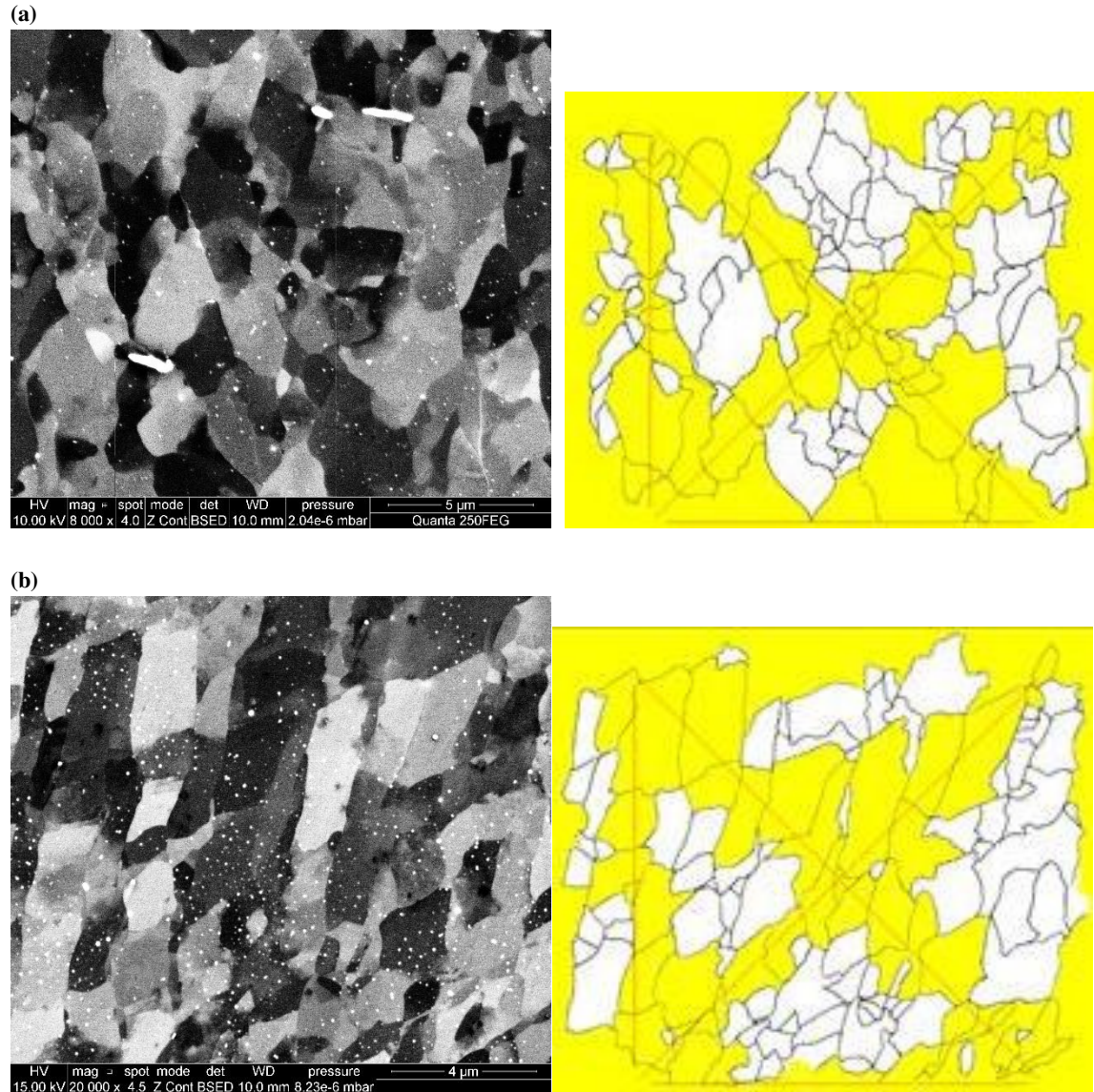


Figure 11: SEM examination of the microstructure of billets during (a) $\phi=90^\circ$ and (b) $\phi=120^\circ$ I-ECAP

5.5 Temperature rise

Figure 12 shows the contour plots of transient temperature distribution after 25%, 50% and 90% processed state for the two cases at the initial processing temperature of 20°C . It is obvious that there is an abrupt temperature rise within the shear zone and the distribution of temperature rise in the billet during the different extrusion stages is inhomogeneous. The maximum increase in temperature due to friction and plastic deformation for $\phi=90^\circ$ is 6.0°C compared to 3.8°C for $\phi=120^\circ$.

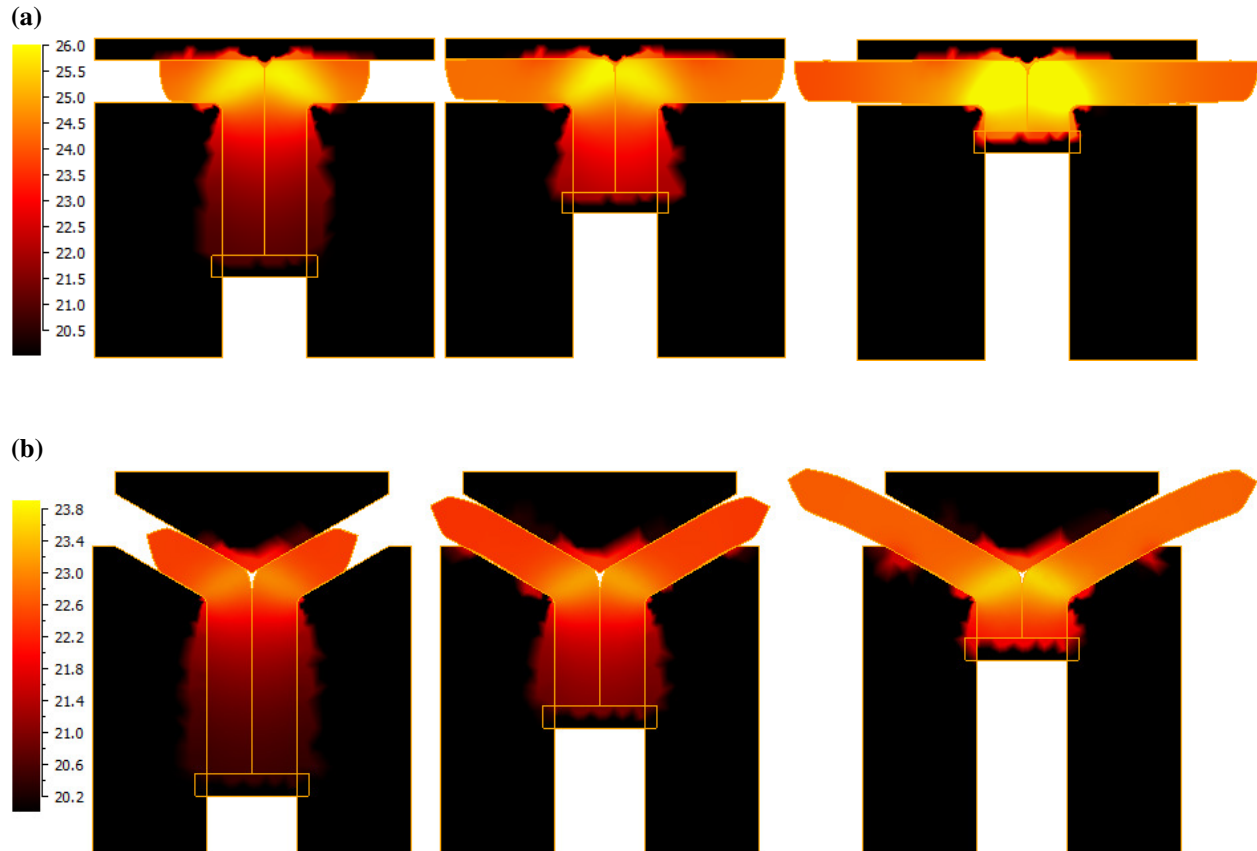


Figure 12: Temperature distribution in tools and billet at 25%, 50% and 90% processed (a) 90° and (b) 120°

6 CONCLUSION

1. First pass of I-ECAP experiments were performed on AA-1050 using the channel intersection angle (ϕ) 90° and 120°.
2. Hardness measurement suggested significant hardness increase in the deformed part compared to the un-deformed part.
3. FE simulation was performed using QForm for the first pass of I-ECAP process to predict strain distribution. The accumulated shear strain in 90 is higher than 120.
4. There is significant grain size reduction in both 90° and 120° I-ECAP.
5. After the first pass the billets that were processed using 90° I-ECAP configuration showed smaller and more uniform grains as compared to 120° I-ECAP.
6. The FEA results would also suggest the 90° configuration experience greater and more uniform shear strain throughout the billet.

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