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Data Article

Dataset concerning the analytical approximation of the Ae_3 temperature



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ABSTRACT

In this paper we present a new polynomial function for calculating the local phase transformation temperature (Ae_3) between the austenite+ferrite and the fully austenitic phase fields during heating and cooling of steel:

$$Ae_3(^{\circ}C) = c_0 + \sum_{X,k} c_{Xk} X^k + \sum_{X,Y,k,m} c_{XkYm} X^k Y^m + \sum_{X,Y,Z,k,m,n} c_{XkYmZn} X^k Y^m Z^n$$

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Keywords:

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The dataset includes the terms of the function and the values for the polynomial coefficients for major alloying elements in steel. A short description of the approximation method used to derive and validate the coefficients has also been included. For discussion and application of this model, please refer to the full length article entitled “The role of aluminium in chemical and phase segregation in a TRIP-assisted dual phase steel” <http://dx.doi.org/10.1016/j.actamat.2016.05.046> (Ennis et al., 2016) [1].

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Specifications Table

Subject area	<i>Steel metallurgy</i>
More specific subject area	<i>Phase transformations</i>
Type of data	<i>Tables and equations</i>
How data was acquired	<i>The approximation of the Ae₃ temperature was constructed in two steps: in the first step a large number of compositions with the associated Ae₃ temperatures were generated; this was followed by multiple regression to find a suitable approximation</i>
Data format	<i>Analysed – Contributions to polynomial coefficients in carbon para-equilibrium equation</i>
Experimental factors	<i>Numerical analysis was carried out on model alloys generated from MTDATA [2] and resulted in the polynomial function, which is described in more detail in this paper.</i>
Experimental features	<i>The approximation of the Ae₃ temperature was constructed in two steps: in the first step a large number of compositions with the associated Ae₃ temperatures were generated; this was followed by multiple regression to find a suitable approximation</i>
Data source location	<i>N/A</i>
Data accessibility	<i>Data is within this article.</i>

Value of the data

- Improved polynomial relationship of phase transformation temperature for major alloying elements in steel.
- Can be directly used to compute phase transformation temperature for any alloy within the computed range.
- Compares well with full thermodynamic model data, but with simple polynomial function.

Table 1

Maximum valid compositions (wt. %) and calculated value of Ae₃ from the approximation.

[C]	Mn	Cr	Si	Al	Maximum calculated Ae ₃
0.8	2.5	1.0	1.5	2.0	910 °C

Table 2
Contributions to polynomial coefficients in carbon para-equilibrium equation.

Contributes to	Product of elements	Constant	Units
c_0°	[intercept]	918.6	$^{\circ}\text{C}$
	Al	161.4	$^{\circ}\text{C}/\text{wt. \%}$
	Cr	−9.4	
	Mn	−57.1	
	Si	50.2	
	AlCr	−4.2	$^{\circ}\text{C}/(\text{wt. \%})^2$
	AlMn	−18.2	
	AlSi	16.0	
	CrMn	−3.6	
	MnSi	−1.9	
	Al ²	19.4	
	Cr ²	1.1	
	Mn ²	1.5	
	Si ²	5.0	
	Al ³	−0.9	$^{\circ}\text{C}/(\text{wt. \%})^3$
	Mn ³	0.4	
	Al ² Cr	1.1	
	Al ² Mn	3.5	
	Al ² Si	−1.2	
	Mn ² Cr	0.8	
Mn ² Si	−0.5		
c_1°	[C]	−720.0	$^{\circ}\text{C}/\text{wt. \%}$
	[C]Al	−380.2	$^{\circ}\text{C}/(\text{wt. \%})^2$
	[C]Cr	−12.4	
	[C]Mn	108.6	
	[C]Si	−122.1	
	[C]MnCr	9.7	$^{\circ}\text{C}/(\text{wt. \%})^3$
	[C]Al ²	−11.3	
	[C]Si ²	−5.9	
	c_2°	[C] ²	1608.4
[C] ² Al		399.9	$^{\circ}\text{C}/(\text{wt. \%})^3$
[C] ² Mn		−212.4	
[C] ² Si		71.4	
c_3°	[C] ³	−2981.2	$^{\circ}\text{C}/(\text{wt. \%})^3$
	[C] ³ Al	−188.1	$^{\circ}\text{C}/(\text{wt. \%})^4$
	[C] ³ Mn	259.7	
c_4°	[C] ⁴	4051.0	$^{\circ}\text{C}/(\text{wt. \%})^4$
	[C] ⁴ Cr	17.3	$^{\circ}\text{C}/(\text{wt. \%})^5$
	[C] ⁴ Mn	−94.5	
c_5°	[C] ⁵	−3388.1	$^{\circ}\text{C}/(\text{wt. \%})^5$
c_6°	[C] ⁶	1227.8	$^{\circ}\text{C}/(\text{wt. \%})^6$

- This function can be seen as an extension of the Andrews expression [3], see Eq. (1), to include the role of carbon and aluminium on critical transformation temperature:

$$Ae_3 (^{\circ}\text{C}) = 910 - 25C_{Mn} + 60C_{Si} - 11C_{Cr} \quad (1)$$

- Where Ae_3 temperature is expressed in $^{\circ}\text{C}$ and concentrations in wt. %.

1. Data

There are three tables used to describe the numerical approximation of the Ae_3 temperature:

Table 3
Relationship of c_i^* terms in the carbon polynomial in Eq.(3) to the constants, c_i .

$c_0^* = c_0 + \sum_{X,k} c_{Xk} X^k + \sum_{X,Y,k} c_{XkYm} X^k Y$
$c_1^* = c_{C,1}[C] + \sum_{X,k} c_{Xk(C1)} X^k [C] + c_{Mn1Cr1(C1)} MnCr[C]$
$c_2^* = c_{C,2}[C]^2 + \sum_{X \neq [C]} c_{X1(C2)} X[C]^2$
$c_3^* = c_{C,3}[C]^3 + \sum_{X \neq [C]} c_{X1(C3)} X[C]^3$
$c_4^* = c_{C,4}[C]^4 + \sum_{X \neq [C]} c_{X1(C4)} X[C]^4$
$c_5^* = c_{C,5}[C]^5$
$c_6^* = c_{C,6}[C]^6$

Table 1 gives the maximum valid composition range based on the model alloys used.

Table 2 lists the contribution of each element to the polynomial coefficients in the derived function given in Eq. (6) in Ref. [1]:

$$Ae_3(^{\circ}C) = c_0 + \sum_{X,k} c_{Xk} X^k + \sum_{X,Y,k,m} c_{XkYm} X^k Y^m + \sum_{X,Y,Z,k,m,n} c_{XkYmZn} X^k Y^m Z^n \tag{2}$$

where Ae_3 temperature is expressed in $^{\circ}C$ and concentrations in wt. %. Under para-equilibrium conditions carbon is the only chemical element that changes its concentration during transformation and to avoid repetitive calculations it is advantageous to write Ae_3 as a polynomial in carbon, $[C]$, as follows:

$$Ae_3 = \sum_i c_i^* [C]^i \tag{3}$$

The relationships of c_i^* to the constants, c , are listed in Table 3.

2. Experimental design, materials and methods

The approximation of the Ae_3 temperature was constructed in two steps: in the first step a large number of compositions with the associated Ae_3 temperatures were generated; this was followed by multiple regression to find a suitable approximation. For each run, a total of 100,000 Ae_3 temperatures were generated with $[C] < 0.8$ wt. % and within the range of validity for all other chemical elements given in Table 1. The value of each chemical element was chosen independently of all the other elements and was taken from a uniform distribution between 0 and the maximum allowed content. The SAS procedure ‘reg’ with the option ‘selection=stepwise’ chose terms from a large bank that contributed significantly to Ae_3 . Terms that did not improve the fit to the data were not included. The bank of terms consisted of:

- Chemical elements.
- Chemical elements squared.
- $[C]$, Mn, Si and Al to the third power.
- $[C]^4$, $[C]^5$, and $[C]^6$.
- The product of $[C]$, Mn, Al and Si with all other elements.
- The product of $[C]^2$, $[C]^3$, $[C]^4$, Mn^2 , Al^2 , and Si^2 with the other elements.
- $[C]MnCr$.

Since the starting temperature for the model is $910^{\circ}C$, all calculated Ae_3 temperatures higher than this value are assigned the starting value. Calculated Ae_3 temperatures higher than $910^{\circ}C$ should be approached with caution, because some extrapolation will have taken place. This is especially true for Al and Si compositions at the upper end of the valid range.

A measure of success of the approximation is the difference between the full MTDATA expressions and the values obtained from the approximation. The standard deviation of the approximation is $4.9^{\circ}C$,

which is much smaller than the undercooling at which nucleation is assumed to take place, with an offset of 0.0 °C. The fit for Ae_3 was also determined for a second, independent set of 100,000 Ae_3 temperatures. The differences between the two sets were small; the differences with Andrews' expression, Eq. (1) are somewhat larger, with an average difference of 11 °C and a standard deviation of 22 °C.

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Transparency document. Supplementary material

Transparency document data associated with this paper can be found in the online version at <http://dx.doi.org/10.1016/j.dib.2016.11.073>.

References

- [1] B.L. Ennis, E. Jimenez-Melero, R. Mostert, M.B. Santillana, P.D. Lee, The role of aluminium in chemical and phase segregation in a TRIP-assisted dual phase steel, *Acta Mater.* 115 (2016) 132–142. <http://dx.doi.org/10.1016/j.actamat.2016.05.046>.
- [2] R.H. Davies, A.T. Dinsdale, J.A. Gisby, MTDATA – thermodynamic and phase equilibrium software from the National Physics Laboratory, *Calphad* 26 (2002) 229–271. [http://dx.doi.org/10.1016/S0364-5916\(02\)00036-6](http://dx.doi.org/10.1016/S0364-5916(02)00036-6).
- [3] K.W. Andrews, Empirical formulae for the calculation of some transformation temperatures, *J. Iron Steel Inst.* 203 (1965) 721–729.